

**Mineralized collagen scaffolds induce hMSC osteogenesis
and matrix remodeling**

Journal:	<i>Biomaterials Science</i>
Manuscript ID:	BM-ART-11-2014-000397.R1
Article Type:	Paper
Date Submitted by the Author:	22-Dec-2014
Complete List of Authors:	Weisgerber, Daniel; University of Illinois at Urbana-Champaign, Materials Science and Engineering Caliari, Steven; University of Illinois at Urbana-Champaign, Chemical and Biomolecular Engineering Harley, Brendan; University of Illinois at Urbana-Champaign, Chemical and Biomolecular Engineering

ARTICLE

Mineralized collagen scaffolds induce hMSC osteogenesis and matrix remodeling

Cite this: DOI: 10.1039/x0xx00000x

D.W. Weisgerber,^a S.R. Caliri,^b B.A.C. Harley^{b,c}Received 00th January 2012,
Accepted 00th January 2012

DOI: 10.1039/x0xx00000x

www.rsc.org/

Biomaterials for bone tissue engineering must be able to instruct cell behavior in the presence of the complex biophysical and biomolecular environments encountered *in vivo*. While soluble supplementation strategies have been identified to enhance osteogenesis, they are subject to significant diffusive loss *in vivo* or the need for frequent re-addition *in vitro*. This investigation therefore explored whether biophysical and biochemical properties of a mineralized collagen-GAG scaffold were sufficient to enhance human mesenchymal stem cell (hMSC) osteogenic differentiation and matrix remodeling in the absence of supplementation. We examined hMSC metabolic health, osteogenic and matrix gene expression profiles, as well as matrix remodeling and mineral formation as a function of scaffold mineral content. We found that scaffold mineral content enhanced long term hMSC metabolic activity relative to non-mineralized scaffolds. While osteogenic supplementation or exogenous BMP-2 could enhance some markers of hMSC osteogenesis in the mineralized scaffold, we found the mineralized scaffold was itself sufficient to induce osteogenic gene expression, matrix remodeling, and mineral formation. Given significant potential for unintended consequences with the use of mixed media formulations and potential for diffusive loss *in vivo*, these findings will inform the design of instructive biomaterials for regenerative repair of critical-sized bone defects, as well as for applications where non-uniform responses are required, such as in biomaterials to address spatially-graded interfaces between orthopedic tissues.

Introduction

Critical-sized bone defects present unique challenges for tissue engineering. Large in size and often irregular in shape, these defects often occur as a result of acute trauma or surgical resection, and can be marked by significant potential for infection as well as severe functional deficits.¹⁻⁴ Allogenic or autogenic bone repair remain the current gold-standard.⁵⁻⁷ However, concerns remain about the resultant secondary wound site, supply and disease transmission, as well as the need for shaping of donor bone to fit often complex defects. With world demand topping two million bone replacement procedures per year,^{8, 9} tissue engineering approaches offer the tantalizing possibility of generating a mechanically robust, patient-customized implant to enhance bone regeneration. However, the development of appropriate biomaterial substrates requires balancing structural, mechanical, and biomolecular design concerns as well as considering the appropriate selection of clinically-relevant cell populations.¹⁰⁻¹³

Given the well-described composition of bone, many tissue engineering approaches have explored mineral constructs consisting entirely or in part of a calcium phosphate (CaP),^{14, 15} polymeric (synthetic or natural) scaffolds,^{16, 17} or, increasingly, polymer-mineral composites.^{9, 18-20} Such biomaterial substrates

offer the advantages of *tunability*, namely allowing the manipulation of biomaterial morphology, such as fiber alignment,²¹ topology, like surface roughness,²² and substrate stiffness.^{23, 24} However, the inclusion of growth factors such as BMP-2 or BMP-7 (known commercially as OP-1)^{25, 26} or full-biochemical supplementation strategies such as osteogenic media^{14, 27-29} remain a common, if not ubiquitous, practice of soluble supplementation. Such strategies are particularly common in approaches utilizing adipose or marrow derived mesenchymal stem cells (MSCs).^{25, 26, 28, 30} However, soluble supplementation strategies – defined here as addition of an exogenous factor to the media – introduce two major concerns. The first is the need for repeated supplementation due to short biomolecule half-lives and diffusive loss of the added factor away from the biomaterial,³¹ a requirement that introduces expense as well as significantly complexity for clinical applications. Second, soluble supplementation methods limit the potential for inducing spatially-selective responses, such as for applications looking to repair orthopedic interfaces between hard and soft tissues. Soluble supplementation also presents a number of concerns for clinical translation. Notably, while BMP-2 has received FDA approval for use, the current need of supra-physiological doses, frequent need for repeated doses due

to diffusive loss, and extensive off-label use has raised a range of safety concerns.⁹

Our laboratory has recently developed a series of collagen-glycosaminoglycan (CG) scaffolds for musculoskeletal tissue engineering applications. These efforts have included optimization of scaffold pore structure (porosity and pore anisotropy),^{32, 33} soluble factor cocktails to balance proliferation and phenotypic stability,^{34, 35} glycosaminoglycan (GAG) content to mediate transient sequestration of growth factors, and structural reinforcement via the incorporation of CG membranes structures.³⁶ Recently we described a calcium phosphate mineralized CG (CGCaP) scaffold with potential application for bone tissue engineering and as a component of an osteotendinous repair biomaterial.³³ In this formulation, CaP crystallites are precipitated with collagen and GAG into a precursor solution which is subsequently lyophilized to form the final CGCaP scaffold.³⁷ While we have previously reported the impact of addition of a CaP mineral phase on scaffold biophysical properties (mechanics, permeability),³³ we had not explored the osteogenic potential of the scaffold itself, notably the potential to avoid the need for osteogenic or BMP-2 media supplementation. Here, the CGCaP scaffold design offers two potential routes to impact MSC osteogenesis. First, given the known impact of substrate stiffness on MSC osteogenesis,²³ the increased stiffness of CGCaP (vs. non-mineralized CG) scaffolds has the potential to impact MSC osteogenesis.³³ Second, the precipitation-based method for introducing CaP into the scaffold offers the potential for long term release of Ca and P ions during cell culture, which have previously been shown to impact osteoblast bioactivity.^{14, 15, 38, 39}

This study therefore examines the potential for a mineralized collagen-GAG scaffold platform to bias osteogenic differentiation of human bone marrow derived MSCs in the absence of conventional media supplementation. We compared the metabolic health and osteogenic potential of hMSCs in two scaffold groups, a non-mineralized CG scaffold control and a 40wt% CaP mineralized CGCaP scaffold in the presence of conventional growth media. We also examined hMSC osteogenic capacity in mineralized CGCaP scaffolds in the presence of conventional osteogenic media and BMP-2 supplemented growth media to determine whether addition of such factors enhanced osteogenesis. MSCs were maintained in scaffolds for 8 weeks in culture, with osteogenic differentiation assessed via periodic assessment of metabolic activity, gene expression, and new matrix synthesis (histology, microCT, mechanics) metrics. Using this approach we asked whether scaffold mineral content was sufficient to promote hMSC osteogenesis.

B. Materials and Methods

B.1. Fabrication of non-mineralized and mineralized scaffolds. Non-mineralized (CG) and mineralized (CGCaP) scaffolds were fabricated via lyophilization from non-mineralized and mineralized CG precursor suspensions. Briefly, the non-mineralized precursor suspension was created by

homogenizing type I collagen from bovine Achilles tendon (0.5 w/v%; Sigma-Aldrich, St. Louis, MO) and chondroitin sulfate from shark cartilage (GAG; 0.04 w/v%, Sigma-Aldrich, St. Louis, MO) in acetic acid (Sigma-Aldrich) at a collagen:GAG ratio of 11.25:1.^{33, 40} The mineralized suspension was fabricated from collagen (1.9 w/v%) and GAG (0.84 w/v%) as before along with calcium salts (0.9 w/v% Ca(OH)₂, 0.4 w/v% Ca(NO₃)₂·4H₂O, Sigma-Aldrich) in phosphoric acid.³³ Precursor suspensions were stored at 4°C and degassed prior to use. Lyophilization was performed in a Genesis freeze-dryer (VirTis, Gardener, NY) using a custom 3" x 3" polysulfone trays. Briefly, the suspensions were solidified via cooling at a constant cooling rate of 1°C/min to a final freezing temperature of -10°C.⁴¹ After allowing 2 h at the final freezing temperature to complete solidification, the frozen suspension was then sublimated at 0°C and 200 mTorr. CG scaffold variants were then lightly crosslinked using a dehydrothermal treatment at <25 Torr and 105°C for 24 h in a vacuum oven (Welch, Niles, IL).⁴² Cylindrical CG and CGCaP scaffold specimens were then cut from the resulting 4 mm thick sheets using an 8 mm biopsy punch (Integra-Miltex, York, PA). Unless otherwise noted, all scaffolds were then sterilized in ethanol for 1 h, hydrated in PBS overnight, crosslinked in a solution of 1-ethyl-3-[3-dimethylaminopropyl] carbodiimide hydrochloride (EDC) and N-hydroxysulfosuccinimide (NHS) at a molar ratio of 5:2:1 EDC:NHS:COOH where COOH represents the amount of collagen in the scaffold, then washed prior to use.^{42, 43}

B.2. Calcium and phosphate ion release from mineralized collagen scaffolds. Ca and P ion release profiles from the CGCaP scaffolds were determined via Calcium Colorimetric and Phosphate Colorimetric Assay Kits (BioVision, Milpitas, CA). Acellular CG and CGCaP scaffolds were cultured in complete mesenchymal stem cell growth media (Lonza, Walkersville, MD) at 37°C and 5% CO₂ for up to 8 weeks. Media was changed regularly, with the removed media (n = 6 specimens per timepoint) stored at 4°C for analysis. All media specimens were then analyzed together. Media aliquots from each time point were analyzed separately via a Tecan M200 fluorometer (Tecan, Männedorf, Switzerland) and compared to series of prepared calcium and phosphate standards to compute total calcium and phosphate concentration. Media cultured with non-mineralized CG scaffolds was used as a negative control.

B.3. Human mesenchymal stem cell culture. Human mesenchymal stem cells (hMSCs, Lonza) were expanded in standard flasks in complete mesenchymal stem cell growth media (Low glucose Dulbecco's Modified Eagle Medium; 10% MSC Fetal bovine serum; 1% Penicillin-Streptomycin; 1% L-glutamine; Invitrogen, Carlsbad, CA) at 37°C and 5% CO₂. After expansion to passage 6, hMSCs were seeded onto 8 mm diameter (4 mm thick) scaffold specimens. A total of 7.5 x 10⁴ cells were seeded onto the scaffold disc via a previously described static seeding method.^{32, 41} MSC seeded scaffolds were subsequently cultured at 37°C and 5% CO₂ for up to 56 days. CGCaP scaffolds were cultured in one of three media

formulations throughout: (1) complete MSC growth media (*CGCaP Growth*); (2) complete MSC growth media supplemented with 100 ng/mL BMP2 (*CGCaP BMP2*); or (3), complete MSC osteogenic media (*CGCaP Osteo*). Human recombinant BMP-2 was obtained from ProSpec (Israel). Osteogenic media contained 50 μ M ascorbic acid, 0.1 μ M Dexamethasone, and 10mM β -glycerophosphate added to growth media. MSC-seeded CG scaffolds were cultured in standard MSC growth media (*CG Growth*) as a control. The chosen dose of BMP-2 (100ng/mL) was based on previous studies by our group that documented a pro-osteogenic effect in CG scaffolds.^{32, 34, 44} All cell seeded scaffold groups were fed their respective media every 3 days.

B.4. Quantification of hMSC metabolic activity. The mitochondrial metabolic activity of MSCs seeded within the CG and CGCaP scaffolds was quantified using the alamarBlue® assay.⁴⁵ Briefly, cell-seeded scaffolds were washed in PBS to remove non-attached and dead cells. Cell-seeded scaffolds were incubated in alamarBlue® solution (Invitrogen) for 105 min at 37°C under mild shaking. The reduction of resazurin to the fluorescent byproduct resorufin by metabolically active cells was measured on a F200 spectrophotometer (Tecan) at 540(52)/580(20) nm (excitation/emission). Results were compared to a prepared standard to compute equivalent cell number. Results (n = 6 per timepoint) were reported as the relative metabolic activity normalized to the number of originally seeded cells.

B.5. hMSC gene expression profiles. The gene expression of osteogenic and matrix synthesis markers was determined via PCR using previously described methods⁴⁶. Briefly, RNA was isolated from scaffolds using 1% β -mercaptoethanol lysis solution and the RNeasy Plant Mini Kit (Qiagen, Valencia, CA). RNA was then reverse transcribed into cDNA using a Bio-Rad S1000 thermal cycler (BioRad, Hercules, CA) via QuantiTect Reverse Transcription Kit (Qiagen). A QuantiTect SYBR Green PCR Kit (Qiagen) and an Applied Biosystems 7900HT Fast Real-Time PCR System (Carlsbad, CA) were used to perform real time PCR. Gene expression profiles were obtained for Collagen 1 (*COL1A1*; matrix related marker), *BSP* (bone sialoprotein), *OPN* (osteopontin), and *RUNX2* (runt-related transcription factor 2). *GAPDH* (glyceraldehyde 3-phosphate dehydrogenase) was used as a housekeeping gene. Previously validated primer sequences were chosen from the literature (**Supp. Table 1**) then synthesized by Integrated DNA Technologies (Coraville, IA). All markers were quantified after 14, 28, and 56 days in culture and analyzed using Sequence Detection Systems software v2.4 (Applied Biosystems, Carlsbad, CA) via the delta-delta Ct method. Results were expressed as fold changes normalized to MSCs cultured in the non-mineralized CG scaffolds at day 14.

B.6. Mechanical behavior of hMSC seeded scaffolds. The elastic modulus of hMSC-seeded CG and CGCaP scaffolds was determined after 14, 28, and 56 days in culture via hydrated unconfined compression using a TA.XTplus Texture Analyzer

(StableMicro Systems Ltd., Surrey, UK). Briefly, samples were compressed to 50% strain at a rate of 0.05 % strain/s to capture the linear elastic response of the scaffold. Cell-seeded scaffolds behaved as low-density open cell foams, with elastic moduli obtained from the linear elastic regime of the resulting stress-strain plots.^{42, 47} The estimated densification ($\rho_{cells}/\rho_{acellular}$) of each scaffold was subsequently calculated from differences in hMSC-seeded vs. unseeded scaffold moduli at each time point (**Supp. Table 2**). Here, an established cellular solids theory method that relates changes in scaffold elastic modulus to changes in scaffold density was used to determine the relative change in scaffold densification through cellular remodeling (ρ_{cells}) as compared to the acellular scaffold control ($\rho_{acellular}$) based on changes in elastic modulus (E_{cells} and $E_{acellular}$ respectively): $E_{cells} / E_{acellular} = (\rho_{cells} / \rho_{acellular})^2$.^{42, 47, 48}

B.7. Histology. Histological assessment of scaffold mineral content was performed via Alizarin red. Scaffold specimens were obtained post fabrication (day 0, control) as well as after 14, 28, and 56 days in culture. Scaffolds were fixed in 10% formalin (Polyscience). Scaffolds were then embedded in paraffin; 5 μ m thick transverse sections were then stained with Alizarin Red (Sigma-Aldrich). Representative sections of each scaffold were captured using an optical microscope (Leica; Buffalo Grove, IL). Histograms of average red pixel intensity calculated via Matlab for each scaffold group and time point.

B.8. Quantification of mineral content and distribution in hMSC-seeded scaffolds. The mineral content of CG and CGCaP scaffolds was assessed post-fabrication (day 0) via micro-computed tomography (microCT). Additionally, hMSC-seeded scaffold specimens were removed from culture at days 14, 28, and 56 for microCT analysis. Scaffolds were stored in 10% formalin (Polysciences Inc., Warrington, PA) at 4°C prior to analysis. Subsequently, scaffolds were washed in de-ionized water to remove the formalin then re-lyophilized to obtain a dry scaffold specimen. Mineral content and distribution was assessed from microCT data gathered using an Xradia

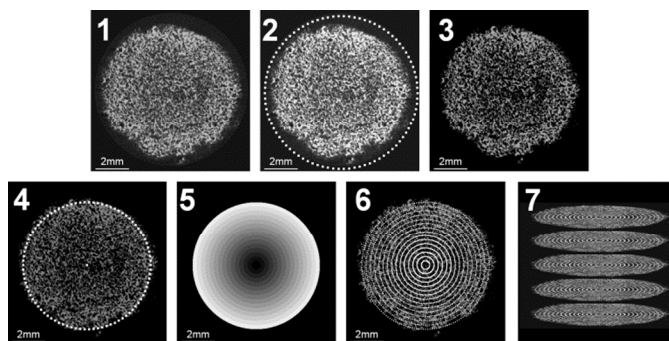


Figure 1. Step by step schematic of microCT z-stack imaging process. Process proceeds by: (1) original image is obtained from the z-stack; (2) user selects a large ROI containing both scaffold and background; (3) background not selected in step 2 is subtracted from the image; (4) user selects only scaffold ROI; (5) the scaffold ROI is divided into 15 equally spaced concentric rings; (6) voxels within each concentric ring are averaged; (7) process is repeated for each image within the z-stack. Example image is dimmed to 80% intensity to better define outlined ROIs. Scale bar: 2mm.

MicroXCT-400. Scans consisted of 793 projections performed at 25keV and 5 watts (corresponding voxel size, 20 μm). The image z-stacks were analyzed via a defined image processing sequence: (1) selection of image file from the z-stack; (2) identification of the scaffold edge and surrounding margin; (3) removal of image background (calculated from the maximum of the non-scaffold margin region); (4) selection of the scaffold edge; (5) partitioning the scaffold into 15 concentric rings of equal width; and (6) calculating the average pixel intensity within each concentric ring (Fig. 1). This analysis was performed through the image z-stack, with results reported as either mean mineral content for the entire scaffold or as a function of radial position.

B.9. Statistics. Statistical analysis was performed via two-way analysis of variance (ANOVA) tests after which a Tukey-HSD post-hoc test was used. Independent factors included time, scaffold type (CG vs. CGCaP), and treatment (growth media, osteogenic media, BMP-2 supplemented growth media). Mechanical characterization and gene expression experiments used at least $n = 3$ scaffolds per group while metabolic activity and scaffold compositional analyses used $n = 6$ scaffolds per group. Significance was set at $p < 0.05$. Error bars are reported as standard deviation unless otherwise noted.

C. Results

C.1 Release of calcium and phosphate ions from the mineralized scaffold. The release of Ca and P ions from acellular CGCaP scaffolds (vs. non-mineralized scaffold

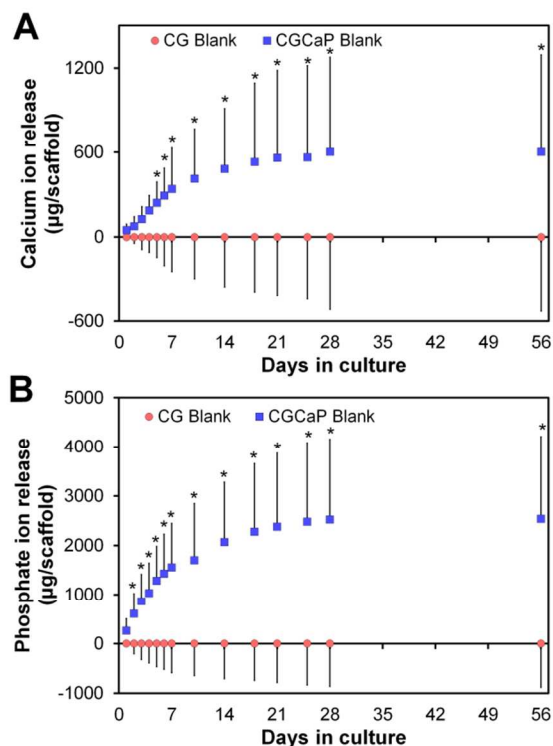


Figure 2. Cumulative release of (A) calcium and (B) phosphate ions in CGCaP relative to CG scaffolds. *: significant ($p < 0.05$) difference at given time point between scaffold groups.

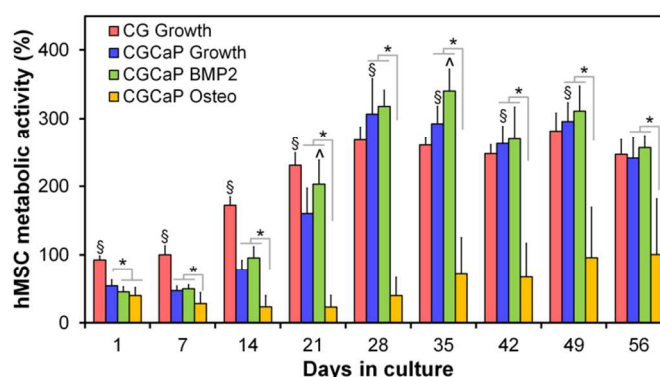


Figure 3. Metabolic activity of hMSCs within non-mineralized CG scaffolds in growth media (CG Growth), mineralized scaffolds in growth media (CGCaP Growth), mineralized scaffolds in BMP2 supplemented growth media (CGCaP BMP2), and mineralized scaffolds in osteogenic media (CGCaP Osteo). *: significant ($p < 0.05$) difference between mineralized scaffold groups. ^: significant ($p < 0.05$) up-regulation compared to all other mineralized groups at the same time point. §: significant ($p < 0.05$) difference between non-mineralized and mineralized (growth media) scaffold groups at a given time point. Metabolic activity results normalized to the number of originally seeded cells.

control) was tracked through 56 days in culture. A significant ($p < 0.05$) increase in Ca and P ion release was observed from mineralized (vs. CG) scaffolds as early as day 2 (phosphate) or day 5 (calcium) (Fig. 2). Ion release appeared to reach an asymptote, with little change after day 21 in culture. Total ion release corresponded to $\sim 80\%$ of the mineral content incorporated during fabrication. The heavy washing associated with EDC crosslinking steps leads to a burst release of mineral (Supp. Fig. 1A, days 1 – 3). However, after this initial burst release, mineral release during subsequent periods of culture was unchanged as a result of EDC crosslinking (Supp. Fig. 1B, days 4 – 7).

C.2. Metabolic activity of hMSCs within non-mineralized and mineralized scaffolds. Metabolic activity of hMSCs in all scaffold-media combinations increased throughout the 56 day culture period (Fig. 3). When cultured in growth media, the metabolic activity of hMSCs within mineralized scaffolds initially lagged behind hMSCs within non-mineralized scaffolds over the first three weeks of culture, likely due to differences in scaffold permeability.³³ However, long term results suggest that the mineralized scaffold supports long term hMSC metabolic health at the level of non-mineralized scaffolds. Addition of BMP-2 to the media did not impact the metabolic health of the MSCs in mineralized collagen scaffolds. However, use of osteogenic media led to a significant ($p < 0.001$) reduction in the metabolic health of hMSCs compared to growth media in mineralized scaffolds as early as day 1.

C.3. hMSC gene expression profiles. Trends of increasing expression levels with time for all genes tested (*COL1A1*, *BSP*, *OPN*, *RUNX2*) were found across all treatment groups (Fig. 4). Significant increases in *BSP* expression was observed after 8 weeks in culture for all scaffold conditions, but were

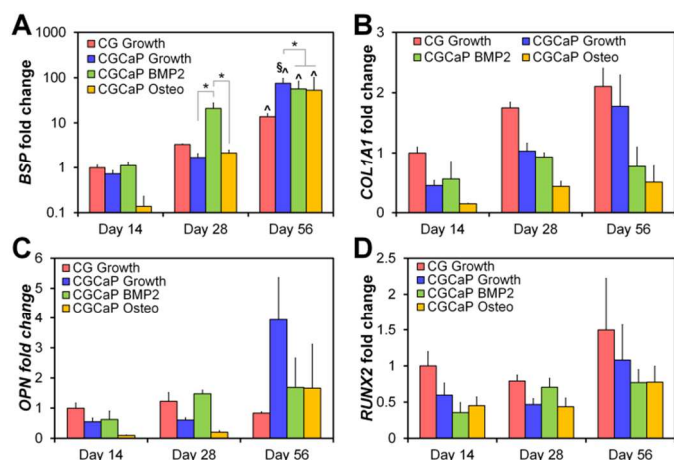


Figure 4. Gene expression levels of (A) bone sialoprotein (*BSP*), (B) collagen 1 (*COL1A1*), (C) osteopontin (*OPN*), and (D) runt-related transcription factor 2 (*RUNX2*). Expression levels were normalized to MSCs in non-mineralized scaffolds at day 14 to compare the effects of both scaffold mineralization and time. *: significant ($p < 0.05$) difference at between scaffold groups at a single time point. ^: significant ($p < 0.05$) difference versus day 14 for a given scaffold group. §: significant ($p < 0.05$) difference between non-mineralized and mineralized scaffolds (growth media) at a given time point.

significantly ($p < 0.05$) greater for hMSCs in mineralized versus non-mineralized scaffolds (maintained in growth media). Interestingly, hMSCs cultured in mineralized scaffolds with BMP-2 supplemented media showed signs of earlier osteogenesis, with a significant increase ($p < 0.05$) in *BSP* expression versus all other scaffold groups by 4 weeks.

C.4. Changes in scaffold modulus with hMSC culture. The modulus of hMSC-seeded mineralized collagen scaffolds increased significantly ($p < 0.0005$) with culture time for all media formulations relative to the unseeded scaffold control (CGCaP blank) (Fig. 5). Changes in elastic moduli of scaffolds at the end of culture versus unseeded scaffolds at the end of culture (1.83 ± 0.5 kPa, 23.00 ± 0.6 kPa for CG, CGCaP scaffolds, respectively), suggest a 50 to 80 percent increase in the density of the construct over 8 weeks of culture (Supp. Table 2). No significant increase in modulus was observed in hMSC-seeded non-mineralized scaffolds, suggesting that addition of hMSCs did not result in appreciable net matrix production. hMSC-seeded mineralized scaffolds were also significantly ($p < 0.05$) stiffer than the hMSC seeded non-mineralized scaffold at all time points (Fig. 5). The modulus of unseeded non-mineralized or mineralized scaffolds remained unchanged over the entire experimental period, consistent with previous observations that EDC-crosslinked CG scaffolds have greater than 6 month half-lives *in vivo*.⁴⁹ Addition of pro-osteogenic supplements impacted the modulus of hMSC-seeded mineralized scaffolds. Mineralized scaffolds maintained in osteogenic media consistently demonstrated a significantly ($p < 0.05$) increased modulus compared to mineralized scaffolds maintained in growth or BMP-2 supplemented media (Fig. 5).

C.5. Analysis of mineral via Alizarin red staining. Histological sections were taken after days 14, 28, and 56 to

evaluate new mineral formation via Alizarin red (Fig. 6; Supp. Table 1). Significant mineral content was observed in mineralized scaffolds regardless of media (growth, osteogenic, BMP-2). However, limited to no mineral content was observed in the hMSC-seeded non-mineralized CG scaffolds.

C.6. CaP mineral content and distribution. The relative mineral content of hMSC seeded non-mineralized and mineralized scaffold groups was calculated via microCT (Fig. 7). All analyses were performed at identical scanner settings and thresholded to reveal mineral content. Depth-averaged results of mineral distribution were calculated as a function of radial position, as no significant differences were observed in radial mineral content as a function of vertical position within the scaffold (data not shown). The average scaffold mineral content of hMSC-seeded mineralized CG scaffolds was significantly ($p < 0.0005$) higher than hMSC-seeded non-mineralized scaffolds for all time points. Choice of media impacted mineral content in hMSC-seeded mineralized scaffolds, but for all conditions significant increases in mineral content were observed with culture time (growth, osteogenic media: day 56 vs. 14; BMP-2 supplemented media: day 56 vs. 28 vs. 14) (Fig. 7B). Analysis of the relative radial distribution of mineral content suggested maximal new mineral formation took place towards the edges of the scaffold for all mineralized scaffold groups, with negligible mineral content found within the non-mineralized scaffold (Supp. Fig. 2).

D. Discussion

This study examined the impact of selective incorporation of a CaP mineral component into collagen-GAG scaffolds on the viability, osteogenic differentiation, and matrix biosynthesis capacity of hMSCs. Previously, extensive work in the literature has suggested that calcium phosphate mineral is a critical component of a range of biomaterials for osteogenic repair applications.⁵⁰⁻⁵⁴ Previous work with CG scaffolds suggested that increasing scaffold stiffness and mineral content can resist

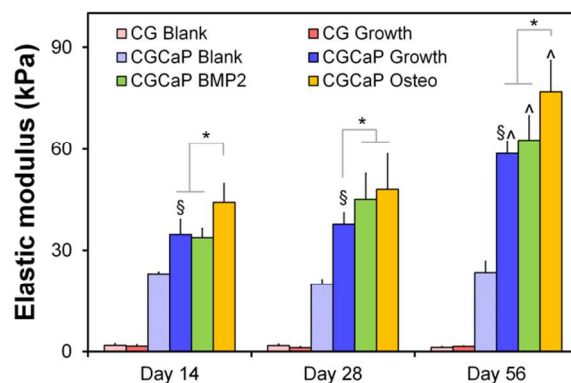


Figure 5. Elastic moduli of hMSC seeded mineralized and non-mineralized collagen scaffolds in compression as a function of time and media supplementation. *: significant ($p < 0.05$) difference at given time point compared to scaffold groups. ^: significant ($p < 0.05$) increase compared to other time points of the same scaffold group. §: significant ($p < 0.05$) difference between non-mineralized and mineralized scaffolds (growth media) at a given time point.

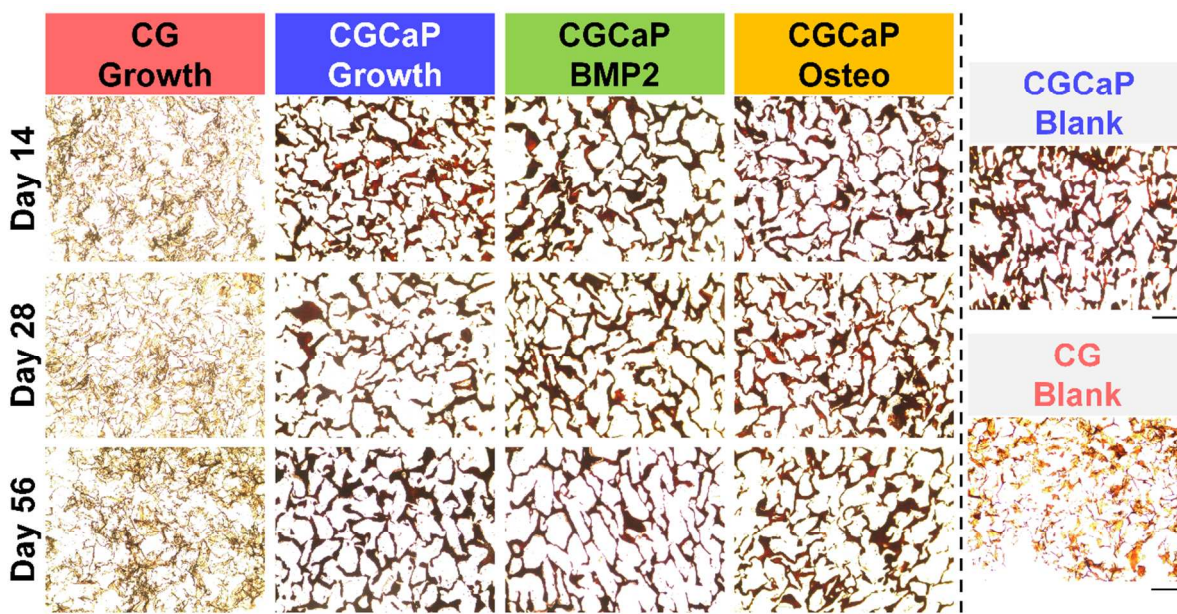


Figure 6. Alizarin red analysis of mineral synthesis. Representative histology sections from days 14, 28, and 56 showing increased mineral content in the mineralized scaffold groups (versus non-mineralized scaffold). Scale bar: 200 μ m.

cell-mediated contraction and promote a more osteogenic phenotype.^{28, 32, 55, 56} However, these efforts all used supplemented media (osteogenic) formulations. For this study we tested the hypothesis that the mineralized CG scaffold could facilitate osteogenesis independent of osteogenic or BMP-2 supplemented media. And while the main goal of this study was to compare the results of hMSC-seeded mineralized versus non-mineralized scaffolds, we included hMSC-seeded mineralized scaffolds maintained in either BMP-2 supplemented (100ng/mL) or osteogenic media as positive controls. The results of our study support that mineralized collagen-GAG scaffolds can themselves promote hMSC osteogenesis.

While we have previously reported increased mechanical properties due to the precipitation based incorporation of CaP mineral content prior to lyophilization,³³ we had not yet examined the long term stability of this mineral content over long term culture as in other material systems.⁵⁷ We therefore examined the kinetics of release of Ca and P ions into the surrounding culture media and their supporting role in cell behavior.^{58, 59} We observed a significant release of Ca and P ions into the surrounding media as early as 2 days in culture (Fig. 2). Though release was quantified for a full 8 weeks *in vitro*, Ca and P release was largely concluded within the first three weeks for mineralized CG scaffolds. Comparing overall phosphate to calcium release, these results further suggested that Ca and P ions were released both due to dissolution of brushite CaP from the scaffold³³ as well as release of residual phosphoric acid trapped in the scaffold following lyophilization.^{37, 60} EDC crosslinking of the mineralized scaffolds had no impact on Ca and P release rate (Supp. Fig. 1) beyond that observed during the extensive washing steps during crosslinking. This suggests the potential to tune scaffold mechanics and degradation rate via crosslinking independent of Ca and P ion release. These results also suggest that this

precipitation-based scheme allows fabrication of mineralized collagen scaffolds whose mineral content may contribute to its osteogenic potential in multiple manners: (1) greater initial stiffness; (2) slow release of pro-osteogenic Ca and P ions over time.

While subject to burst diffusive losses like other soluble supplementation methods, scaffold mineral content was able to instruct pro-osteogenic phenotype. Analysis of osteogenic gene expression profiles (Fig. 4) and matrix remodeling (Fig. 5, 6) indicated that the mineralized scaffold significantly contributed to an enhanced pro-osteogenic phenotype at later time points (day 56) in culture and without the need for conventional osteogenic supplements. The kinetics of Ca and P release also suggest the need for future studies to more closely examine mechanisms underlying scaffold-mediated MSC osteogenesis as a function of ion release.

Measures of metabolic activity give a good representation of the overall metabolic health of the cell-seeded biomaterials,^{32, 34, 61} they are not exact measures of total cell number. Given previous studies favorably comparing explicit measures of cell number versus metabolic activity of cell-seeded CG scaffolds^{62, 63} as well as the large number of scaffold groups and experimental end-points in this study, we chose not to explicitly determine cell number via destructive assays, but rather tracked construct metabolic activity via non-destructive AlamarBlue. Future efforts will explicitly measure overall MSC number. Regardless, all scaffold variants supported a significant increase in cell metabolic activity over the course of the 8 week *in vitro* culture (Fig. 3). Examining the impact of scaffold type (mineralized vs. non-mineralized), non-mineralized CG scaffolds supported significantly enhanced MSC metabolic health up to day 28 of culture, while mineralized CG scaffolds supported enhanced metabolic health at later time points (Fig. 3). These results suggest an important

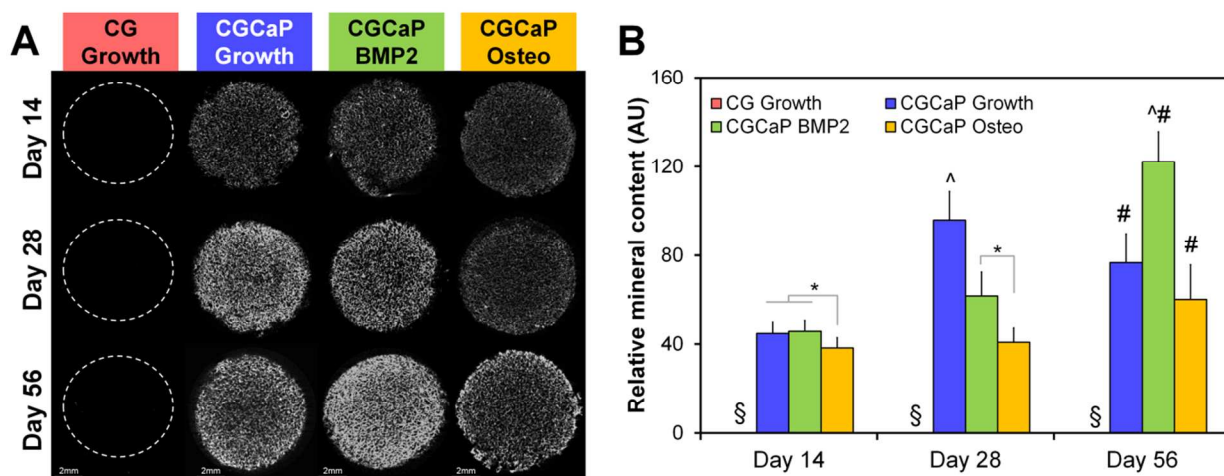


Figure 7. Micro-CT analysis of new mineral formation as a function of time and scaffold culture conditions. **(A)** Representative micro-CT slices. **(B)** Average pixel intensity was used as a quantitative metric of relative mineral content within each scaffold at each time point. *: significant difference between groups at a given time point. ^: significant ($p < 0.05$) increase compared to all other groups at the same time point. §: significantly ($p < 0.0005$) lower than all other groups. #: significant ($p < 0.05$) increase compared same group at day 14.

trade-off in initial scaffold properties versus cell-mediated remodeling. The increased relative density of the mineralized CG scaffolds reduces construct permeability³³ and likely initial cell attachment,^{64, 65} leading to initially reduced MSC metabolic health in the mineralized scaffold during initial culture (**Fig. 3**). Cell-mediated contraction can significantly impact CG scaffold microstructural properties as well as resultant cell activity at the functional and genomic level over longer periods of culture.⁶⁶ However mineralized, highly crosslinked, and high-density variants of CG scaffolds are substantially less subject to significant cell-mediated contraction,³² suggesting that the relative improvement of MSC metabolic activity in mineralized scaffolds at late culture times may be due to reduced contraction-mediated remodeling. Comparing the impact of media supplementation on MSC metabolic health in mineralized CG scaffolds, MSCs supplemented with osteogenic media showed significantly reduced metabolic activity versus growth media or BMP-2 supplemented growth media (**Fig. 3**), in line with previous findings in the literature regarding trade-offs in cell proliferation versus phenotype.⁶⁷⁻⁶⁹

Given observed differences in metabolic activity, we therefore examined whether: (1) the mineralized CG scaffold was sufficient to induce osteogenic MSC differentiation in the absence of osteogenic supplementation; and (2) osteogenic media supplementation enhanced the speed of osteogenesis. An analysis of osteogenic gene expression profiles suggested the mineralized scaffold supported higher levels of pro-osteogenic gene expression (*BSP*) versus the non-mineralized scaffold. Comparing expression profiles within the mineralized scaffold groups, we observed an earlier increase in bone sialoprotein expression for MSCs in BMP-2 supplemented relative to growth or osteogenic media mineralized scaffolds. By day 56, however, hMSCs in mineralized scaffolds without osteogenic supplementation showed the highest expression levels of *BSP* (**Fig. 4**). We also observed the increased expression of *COL1A1* within both non-mineralized and mineralized CG scaffolds in growth media versus scaffolds maintained in osteogenic of

BMP-2 supplemented media, suggesting that hMSCs may be more inclined to deposit or remodel their matrix without supplementation (**Fig. 4**). Recent efforts have shown enhanced osteogenesis and osteogenic gene expression notably increased *COL1A1* and *RUNX2* expression, in presence of calcium phosphate,^{70, 71} suggesting that the phase of mineral and the mode of incorporating it into a collagen scaffold network may alter MSC response. Ongoing efforts are looking to identify mechanisms associated with MSC bioactivity within these mineralized collagen scaffolds.

To better understand if both the mineral content and/or media supplementation enhanced osteogenic cell remodeling we examined changes in scaffold mechanical properties (**Fig. 5**) as well as changes in scaffold mineral content via histological and microCT analysis (**Figs. 6-7**). Mechanical analysis provided insight into changes within the matrix such as protein and CaP mineral content deposition. We observed no significant difference in elastic modulus between acellular CG scaffolds and those cultured in growth media with hMSCs. While some cell-mediated degradation of the CG scaffold is expected given the long culture time, these results suggest little to no excess ECM deposition occurred over the 8 week period of culture. In contrast, all mineralized CG scaffolds seeded with MSCs exhibited significant increases in elastic modulus relative to the acellular control. Consistent with the concept of a trade-off in metabolic expansion and functional activity, while hMSCs maintained in osteogenic media showed relatively little increases in metabolic activity, these scaffolds showed the greatest gains in elastic moduli at each time point (**Fig. 5**). However the mechanical properties of mineralized scaffolds maintained increased significantly over the course of the 8 week experiment regardless of media formulation used.

The observed change in scaffold mechanical properties correlated with analysis of scaffold mineral content via Alizarin Red histological staining (**Fig. 6**) and microCT (**Fig. 7**). Despite significant release of initial scaffold mineral content, extensive Alizarin Red staining was observed in all in mineralized

scaffolds with little to no staining in non-mineralized scaffolds (Fig. 6). Analysis of the micro-CT scans of all scaffold groups suggested significant new mineral formation in the mineralized CG scaffolds regardless of media supplementation, while again little to no new mineral content was observed in the non-mineralized CG scaffold (Fig. 7). Examining trends in time, it appears that the mineralized CG scaffold was able to support the largest increase in mineral content over the first 4 weeks of culture, while BMP-2 supplementation led to a larger increase in new mineral formation at the later time points (8 weeks). This result suggests that the significant increase in modulus of mineralized CG scaffolds in osteogenic media may be due to a combination of protein and mineral deposition. Given the significant release of initial scaffold mineral content over the course of 8 weeks of culture (~80% of the initial mineral, Fig. 2), the presence of significant amounts of mineral and significant modulus increases suggests that the mineralized scaffold provides the correct microenvironment to support significant new mineral formation and matrix synthesis.

A radial analysis of the mineral content (Supp. Fig. 2) also provided information regarding both mineral deposition and, by proxy, cell distribution and ingrowth. The largest increases in mineral were concentrated towards the edges of the scaffold disks, with radial plots of mineral content suggesting substantial additional mineral content in the external 1.5 mm of the 8 mm diameter disks. In agreement with the average mineral content within a scaffold (Fig. 7) no mineral content was detected within the CG growth scaffolds. Assuming uniform release of mineral content within a fully hydrated scaffold, radial differences in mineral content suggest non-uniform matrix remodeling via hMSCs. Such results motivate future optimization of mineralized scaffold microstructural properties to support enhanced cell penetration as well as subsequent metabolic support via diffusive transport. These results also motivate future efforts to explore strategies such as incorporation of biomolecular agonists (e.g., VEGF) to enhance the speed of vascular ingrowth *in vivo*. Together these results suggest that the mineralized collagen scaffold described here is sufficient to induce pro-osteogenic behavior of hMSCs in the absence of conventional media supplements. Ongoing efforts are examining both the mechanisms associated with these findings, hypothesized to be mechanotransduction and TGF- β superfamily signal transduction pathway, and their use *in vivo* for bone regeneration. However, these results have significance regarding the developing of biomaterials for orthopedic insertional repair (e.g., osteotendinous, osteochondral interface). Identifying a mineralized scaffold able to support enhanced hMSC osteogenesis in the absence of media supplementation provides the foundation for developing instructive biomaterials containing spatially-graded mineral content to locally enhance osteogenic specification.

Conclusions

Biomaterials for bone tissue engineering must be able to instruct cell behavior in the presence of the complex

biophysical and biomolecular environments encountered *in vivo*. While osteogenic media or exogenous BMP-2 supplementation are often used as an essential element of many bone regeneration studies, diffusive loss and rapid degradation are primary concerns of efforts requiring exogenous supplementation. Using a series of collagen-GAG scaffolds, we investigated whether incorporation of a transient CaP mineral content was sufficient to enhance hMSC osteogenic differentiation and matrix remodeling in the absence of lineage-specific media supplementation. We found that the presence of CaP mineral enhances hMSC osteogenesis compared to non-mineralized scaffold in the absence of traditional osteogenic supplements. While osteogenic media or BMP-2 supplementation may enhance the speed or long term intensity of the response, the mineralized CG scaffold is able to support extensive hMSC osteogenesis, matrix remodeling, and new mineral formation in the absence of lineage specific media. These efforts are informing ongoing work in our lab exploring the regenerative potential of mineralized CG scaffolds in critical sized bone defects as well as methods to selectively immobilize osteo-inductive signals such as BMP-2 in the mineralized CG scaffold to enhance this effect.

Acknowledgements

The authors acknowledge Donna Epps for assistance with histology, IGB Core Facilities for assistance with RT-PCR, Dr. Leilei Yin for assistance in micro-computed tomography, as well as Yue Wang and Dr. Michael Insana for assistance with mechanical testing of cell-seeded scaffolds. This material is based upon work supported by the National Science Foundation under Grant No. 1105300. Research reported in this publication was supported by the National Institute of Arthritis and Musculoskeletal and Skin Diseases of the National Institutes of Health under Award Numbers R21 AR063331. The content is solely the responsibility of the authors and does not necessarily represent the official views of the National Institutes of Health. DW was funded at UIUC from National Science Foundation (NSF) Grant 0965918 IGERT: Training the Next Generation of Researchers in Cellular & Molecular Mechanics and BioNanotechnology. We are also grateful for funding for this study provided by the Chemistry-Biology Interface Training Program NIH NIGMS T32GM070421 (SRC). This research was carried out in part in the Microscopy Suite, Imaging Technology Group, Beckman Institute for Advanced Science and Technology, University of Illinois at Urbana-Champaign.

Notes

^a Dept. of Materials Science and Engineering

^b Dept. of Chemical and Biomolecular Engineering

^c Institute for Genomic Biology

University of Illinois at Urbana-Champaign

Urbana, IL 61801, USA

Email: bharley@illinois.edu

Phone: 217-244-7112

Fax: 217-333-5052

Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/b000000x/

References

- M. J. Miller, D. P. Goldberg, A. W. Yasko, J. C. Lemon, W. C. Satterfield, M. C. Wake and A. G. Mikos, *Tissue Eng*, 1996, **2**, 51-59.
- L. Krishnan, N. J. Willett and R. E. Guldborg, *Ann Biomed Eng*, 2014, **42**, 432-444.
- P. M. Mountziaris and A. G. Mikos, *Tissue Eng Part B Rev*, 2008, **14**, 179-186.
- E. M. Christenson, K. S. Anseth, J. J. van den Beucken, C. K. Chan, B. Ercan, J. A. Jansen, C. T. Laurencin, W. J. Li, R. Murugan, L. S. Nair, S. Ramakrishna, R. S. Tuan, T. J. Webster and A. G. Mikos, *J Orthop Res*, 2007, **25**, 11-22.
- M. H. Smith, C. L. Flanagan, J. M. Kemppainen, J. A. Sack, H. Chung, S. Das, S. J. Hollister and S. E. Feinberg, *Int J Med Robot*, 2007, **3**, 207-216.
- T. B. Dodson, R. A. Bays, R. C. Pfeffle and D. L. Barrow, *J Oral Maxillofac Surg*, 1997, **55**, 260-267.
- G. Zimmermann and A. Moghaddam, *Injury*, 2011, **42**, S16-21.
- C. Laurencin, Y. Khan and S. F. El-Amin, *Expert Rev Med Devices*, 2006, **3**, 49-57.
- F. G. Lyons, J. P. Gleeson, S. Partap, K. Coghlan and F. J. O'Brien, *Clin Orthop Relat Res*, 2014, **472**, 1318-1328.
- D. W. Huttmacher, *J Biomater Sci Polym Ed*, 2001, **12**, 107-124.
- E. S. Place, N. D. Evans and M. M. Stevens, *Nat Mater*, 2009, **8**, 457-470.
- W. L. Grayson, M. Fröhlich, K. Yeager, S. Bhumiratana, M. E. Chan, C. Cannizzaro, L. Q. Wan, X. S. Liu, X. E. Guo and G. Vunjak-Novakovic, *Proc Natl Acad Sci U S A*, 2010, **107**, 3299-3304.
- X. L. Xu, J. Lou, T. T. Tang, K. W. Ng, J. H. Zhang, C. F. Yu and K. R. Dai, *J Biomed Mater Res B Appl Biomater*, 2005, **75B**, 289-303.
- P. Muller, U. Bulnheim, A. Diener, F. Luthen, M. Teller, E. D. Klinkenberg, H. G. Neumann, B. Nebe, A. Liebold, G. Steinhoff and J. Rychly, *J Cell Mol Med*, 2008, **12**, 281-291.
- S. Nakamura, T. Matsumoto, J. Sasaki, H. Egusa, K. Y. Lee, T. Nakano, T. Sohmura and A. Nakahira, *Tissue Eng Part A*, 2010, **16**, 2467-2473.
- S. Musah, S. A. Morin, P. J. Wrighton, D. B. Zwick, S. Jin and L. L. Kiessling, *ACS Nano*, 2012, **6**, 10168-10177.
- J. Hu, X. Sun, H. Ma, C. Xie, Y. E. Chen and P. X. Ma, *Biomaterials*, 2010, **31**, 7971-7977.
- A. A. Al-Munajjed, N. A. Plunkett, J. P. Gleeson, T. Weber, C. Jungreuthmayer, T. Levingstone, J. Hammer and F. J. O'Brien, *J Biomed Mater Res B Appl Biomater*, 2009.
- H. W. Kim, J. H. Song and H. E. Kim, *J Biomed Mater Res A*, 2006, **79**, 698-705.
- S. S. Liao, F. Z. Cui, W. Zhang and Q. L. Feng, *J Biomed Mater Res B Appl Biomater*, 2004, **69B**, 158-165.
- C. A. Bashur, R. D. Shaffer, L. A. Dahlgren, S. A. Guelcher and A. S. Goldstein, *Tissue Eng Part A*, 2009, **15**, 2435-2445.
- M. Massumi, M. Abasi, H. Babaloo, P. Terraf, M. Safi, M. Saeed, J. Barzin, M. Zandi and M. Soleimani, *Tissue Eng Part A*, 2012, **18**, 609-620.
- A. J. Engler, S. Sen, H. L. Sweeney and D. E. Discher, *Cell*, 2006, **126**, 677-689.
- Y. Sun, L. G. Villa-Diaz, R. H. Lam, W. Chen, P. H. Krebsbach and J. Fu, *PLoS ONE*, 2012, **7**, e37178.
- I. Pountos, T. Georgouli, K. Henshaw, H. Bird, E. Jones and P. V. Giannoudis, *J Orthop Trauma*, 2010, **24**, 552-556.
- W. Shen, X. Chen, Z. Yin, B. C. Heng, W. Chen and H.-W. Ouyang, *Biomaterials*, 2010, **31**, 7239-7249.
- M. H. You, M. K. Kwak, D. H. Kim, K. Kim, A. Levchenko, D. Y. Kim and K. Y. Suh, *Biomacromolecules*, 2010, **11**, 1856-1862.
- E. Farrell, F. J. O'Brien, P. Doyle, J. Fischer, I. Yannas, B. A. Harley, B. O'Connell, P. J. Prendergast and V. A. Campbell, *Tissue Eng*, 2006, **12**, 459-468.
- J. H. Lee, J. W. Rhie, D. Y. Oh and S. T. Ahn, *Biochem Biophys Res Commun*, 2008, **370**, 456-460.
- L. D. Solorio, A. S. Fu, R. Hernandez-Irizarry and E. Alsberg, *J Biomed Mater Res A*, 2010, **92**, 1139-1144.
- Y. H. Shen, M. S. Shoichet and M. Radisic, *Acta Biomater*, 2008, **4**, 477-489.
- S. R. Caliani and B. A. C. Harley, *Advanced healthcare materials*, 2014, **3**, 1086-1096.
- D. W. Weisgerber, D. O. Kelkoff, S. R. Caliani and B. A. C. Harley, *J Mech Behav Biomed Mater*, 2013, **28**, 26-36.
- S. R. Caliani and B. A. C. Harley, *Tissue Eng A*, 2014, **20**, 2463-2472.
- S. R. Caliani and B. A. Harley, *Tissue Eng Part A*, 2013, **19**, 1100-1112.
- S. R. Caliani, L. C. Mozden, O. Armitage, M. L. Oyen and B. A. C. Harley, *J Biomed Mater Res A*, 2014, **102**, 917-927.
- B. A. Harley, A. K. Lynn, Z. Wissner-Gross, W. Bonfield, I. V. Yannas and L. J. Gibson, *J Biomed Mater Res A*, 2010, **92**, 1066-1077.
- S. An, J. Ling, Y. Gao and Y. Xiao, *J Periodontal Res*, 2012, **47**, 374-382.
- Y. Zhou, T. L. Lewis, L. J. Robinson, K. M. Brundage, R. Schafer, K. H. Martin, H. C. Blair, J. Soboloff and J. B. Barnett, *J Cell Physiol*, 2011, **226**, 1082-1089.
- I. V. Yannas, Lee, E., Orgill, D.P., Skrabut, E.M., Murphy, G.F., *Proc. Natl. Acad. Sci. USA*, 1989, **86**, 933-937.
- F. J. O'Brien, B. A. Harley, I. V. Yannas and L. J. Gibson, *Biomaterials*, 2005, **26**, 433-441.
- B. A. Harley, J. H. Leung, E. C. Silva and L. J. Gibson, *Acta Biomater*, 2007, **3**, 463-474.
- L. H. Olde Damink, P. J. Dijkstra, M. J. van Luyn, P. B. van Wachem, P. Nieuwenhuis and J. Feijen, *Biomaterials*, 1996, **17**, 765-773.
- S. R. Caliani, E. A. Gonnerman, W. K. Grier, D. W. Weisgerber, J. M. Banks, A. J. Alsop, J.-S. Lee, R. C. Bailey and B. A. C. Harley, *Advanced healthcare materials*, 2014.
- C. M. Tierney, M. J. Jaasma and F. J. O'Brien, *J Biomed Mater Res A*, 2009, **91**, 92-101.
- G. P. Duffy, T. M. McFadden, E. M. Byrne, S. L. Gill, E. Farrell and F. J. O'Brien, *Eur Cell Mater*, 2011, **21**, 15-30.
- B. P. Kanungo and L. J. Gibson, *Acta Biomater*, 2009, **5**, 1006-1018.
- B. P. Kanungo and L. J. Gibson, *Acta Biomater*, 2009, **6**, 344-353.
- B. A. Harley, M. H. Spilker, J. W. Wu, K. Asano, H. P. Hsu, M. Spector and I. V. Yannas, *Cells Tissues Organs*, 2004, **176**, 153-165.
- A. M. Barradas, V. Monticone, M. Hulsmann, C. Danoux, H. Fernandes, Z. Tahmasebi Birgani, F. Barrere-de Groot, H. Yuan, M. Reinders, P. Habibovic, C. van Blitterswijk and J. de Boer, *Integr Biol (Camb)*, 2013, **5**, 920-931.
- C. V. Brovarone, E. Verne and P. Appendino, *J Mater Sci Mater Med*, 2006, **17**, 1069-1078.
- A. K. Lynn, T. Nakamura, N. Patel, A. E. Porter, A. C. Renouf, P. R. Laity, S. M. Best, R. E. Cameron, Y. Shimizu and W. Bonfield, *Journal of Biomedical Materials Research Part A*, 2005, **74**, 447-453.
- S. J. Polak, S. K. Levengood, M. B. Wheeler, A. J. Maki, S. G. Clark and A. J. Johnson, *Acta Biomater*, 2011, **7**, 1760-1771.
- G. M. Cunniffe, F. J. O'Brien, S. Partap, T. J. Levingstone, K. T. Stanton and G. R. Dickson, *J Biomed Mater Res A*, 2010, **95**, 1142-1149.
- C. M. Curtin, G. M. Cunniffe, F. G. Lyons, K. Bessho, G. R. Dickson, G. P. Duffy and F. J. O'Brien, *Adv Mater*, 2012, **24**, 749-754.
- C. M. Murphy, A. Matsiko, M. G. Haugh, J. P. Gleeson and F. J. O'Brien, *J Mech Behav Biomed Mater*, 2012, **11**, 53-62.
- Y. J. Seol, J. Y. Park, J. W. Jung, J. Jang, R. Girdhari, S. W. Kim and D. W. Cho, *Tissue engineering. Part A*, 2014, **20**, 2840-2849.

58. A. M. Barradas, H. A. Fernandes, N. Groen, Y. C. Chai, J. Schrooten, J. van de Peppel, J. P. van Leeuwen, C. A. van Blitterswijk and J. de Boer, *Biomaterials*, 2012, **33**, 3205-3215.
59. G. Y. Jung, Y. J. Park and J. S. Han, *J Mater Sci Mater Med*, 2010, **21**, 1649-1654.
60. A. K. Lynn, S. M. Best, R. E. Cameron, B. A. Harley, I. V. Yannas, L. J. Gibson and W. Bonfield, *J Biomed Mater Res A*, 2010, **92**, 1057-1065.
61. C. M. Tierney, M. G. Haugh, J. Liedl, F. Mulcahy, B. Hayes and F. J. O'Brien, *J Mech Behav Biomed Mater*, 2009, **2**, 202-209.
62. S. R. Caliri, D. W. Weisgerber, M. A. Ramirez, D. O. Kelkhoff and B. A. C. Harley, *J Mech Behav Biomed Mater*, 2012, **11**, 27-40.
63. S. R. Caliri and B. A. C. Harley, *Biomaterials*, 2011, **32**, 5330-5340.
64. C. M. Murphy, M. G. Haugh and F. J. O'Brien, *Biomaterials*, 2010, **31**, 461-466.
65. F. J. O'Brien, B. A. Harley, M. A. Waller, I. V. Yannas, L. J. Gibson and P. J. Prendergast, *Technol Health Care*, 2007, **15**, 3-17.
66. S. M. Vickers, T. Gotterbarm and M. Spector, *J Orthop Res*, 2010, **28**, 1184-1192.
67. H. Wang, B. Pang, Y. Li, D. Zhu, T. Pang and Y. Liu, *Cytotherapy*, 2012, **14**, 423-430.
68. S. Thomopoulos, F. L. Harwood, M. J. Silva, D. Amiel and R. H. Gelberman, *J Hand Surg Am*, 2005, **30**, 441-447.
69. P. Malladi, Y. Xu, M. Chiou, A. J. Giaccia and M. T. Longaker, *Am J Physiol Cell Physiol*, 2006, **290**, C1139-1146.