ChemComm

Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/chemcomm

Cite this: DOI: 10.1039/coxx00000x

www.rsc.org/chemcomm

Communication

ChemComm Accepted Manuscript

A Facile Synthetic Route to Poly(*p*-phenylene terephthalamide) with Dual Functional Groups

Shuming Du,^a[‡] Wenbin Wang,^a[‡] Yan Yan,^b Jie Zhang,^a Ming Tian,^b Liqun Zhang,^b and Xinhua Wan^a*

50

Received (in XXX, XXX) Xth XXXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX 5 DOI: 10.1039/b000000x

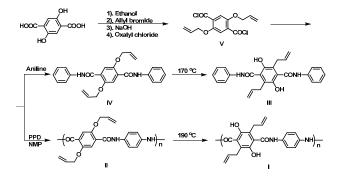
Claisen rearrangement reaction was employed for the first time to obtain a novel PPTA bearing reactive allyl and hydroxyl groups, that may act as a sizing agent of Kevlar fiber to improve the interface structure and interfacial 10 adhesion of rubber or epoxy based composites.

Poly(*p*-phenylene terephthalamide) (PPTA) is a famous aramid polymer. Its totally aromatic, rod like structure facilitates the fabrication of fibers (e.g. Kevlar) with a very high orientation from a liquid crystalline solution.¹ These fibers have ultrahigh ¹⁵ tensile strength and modulus, exceptional heat and flame resistance as well as chemical, solvent and environmental

- resistance as well as chemical, solvent and environmental stabilities, and have been used as reinforcing materials of composites in wide applications such as military, aeronautics, astronautics, automobiles, and sports.²⁻⁶
- 20 One main challenge in manufacturing high performance composites reinforced by Kevlar fibers is their low affinity for interfacial adhesion towards resin matrix due to the highly oriented, extended-chain morphology and the lack of reactable substituent. It is a common practice to introduce substituents on 25 PPTA chains through either amide groups or phenyl rings to
- ²⁵ PPTA chains through either annue groups of phenyl rings to overcome this difficulty. The *N*-substitution of amide group works at the cost of damaging interchain hydrogen bondings.⁷ The substitution on phenyl ring does not have such a drawback and therefore has been extensively investigated. The substituents
- ³⁰ employed include alkyl⁸⁻¹⁰, aryl,¹¹⁻¹³ halogen,^{8,14} nitro,^{15,16} sulfo,¹⁷⁻²⁰ hydroxyl,²¹ propargyl,²² and other groups. However, all these chemically modified PPTAs contain only mono-functional groups, which are just effective to one type matrix, e.g. rubber or epoxy. It is highly desired to develop a facile method to obtain
- ³⁵ PPTA with dual functional substituents, which may act as a reinforcing material alone or a sizing agent of Kevlar fiber to improve the interface structures and interfacial adhesions of composites based on various resins such as rubber and epoxy.

Claisen rearrangement (**CR**) is a powerful and mild carbon-⁴⁰ carbon bond-forming chemical reaction.²³ The aromatic variation with allyl phenyl ether has the advantage of introducing allyl and hydroxyl groups on phenyl ring simultaneously, but has never been applied in the preparation of high performance polymer. Herein, we report the synthesis of a novel PPTA bearing both

⁴⁵ apolar allyl and polar hydroxyl groups via low temperature polycondensation of *p*-phenylenediamine (PPD) with 2,5-diallyloxyterephthaloyl chloride (V) and subsequent thermally induced CR.



Scheme 1 Syntheses of model compounds and polymers

The monomer V was obtained from inexpensive, commercially available 2,5-dihydroxyterephthalic acid through esterification with ethanol, etherification with allyl bromide, hydrolization under alkaline condition, and chloroformylation by oxalyl 55 chloride (Scheme 1). All the reactions happened in high yields and the intermediates and products were easily purified by recrystallization. To test the feasibility of synthetic strategy and help characterize the structure of target polymer, 2,5-diallyloxy-N,N'-diphenylterephthalamide (IV) was first prepared as a model 60 compound by the amidation of V with aniline and then subjected to CR at 170 °C, which was monitored by NMR and FTIR spectroscopy. Figure 1 presents ¹H NMR spectra of IV annealed at 170 °C for various time. The assignment of each resonance peak was made by Heteronuclear Multiple Quantum Coherence 65 (HMQC) and Heteronuclear Multiple Bond Correlation (HMBC) (Figs. S1 and S2). The resonance absorptions at 7.47 ppm were attributed to the aromatic protons (e) of central phenyl ring and 4.73 ppm for the methylenic protons (f) of allyloxy group, respectively. As CR continued, these two absorptions became 70 weak and disappeared completely after 6 h. Meanwhile, two new absorptions from benzylic protons (f') and hydroxyl protons (e') at 3.36 and 8.25 ppm, separately, were observed, implying the formation of 2,5-diallyl-3,6-dihydroxy-N,N'diphenylterephthalamide (III). Although the resonance peaks of 75 other protons of IV did not change remarkably during CR, the signals of vinylic protons shifted upfield. The ¹³C NMR and FTIR spectra gave the consistent results (Figs. S3 and S4).

The polycondensation of **V** and PPD was carried out in *N*methyl-2-pyrrolidone (NMP) solution containing 8 wt% CaCl₂ at ⁸⁰ -10 °C. Due to the tight wrapping of high population of apolar allyloxy groups,¹⁰ which requires distinct solvent to polar main

This journal is © The Royal Society of Chemistry [year]

chain, the resultant polymer **II** was insoluble in either concentrated H_2SO_4 or common organic solvents. It was therefore not possible to characterize its molecular weight by means of viscometry applied for PPTA and other techniques. To increase s its solubility, *N*-butylation was carried out with Katayose's method.⁷ The derived polymer showed number-averaged molecular mass (M_n) of 9100 Dalton and molecular mass polydispersity index (PDI) of 1.79, estimated by gel permeation chromatography using tetrahydrofuran as an eluent.

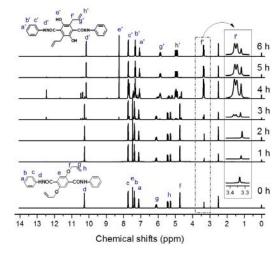
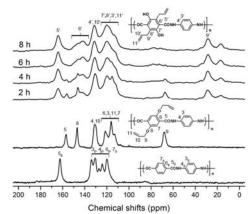


Fig.1 ¹H NMR spectra of model compound **IV** annealed at 170 °C for various time. Measured in d6-DMSO at room temperature.

10

The **CR** reaction of **II** was carried out in NMP at 190 °C for various time under inert atmosphere. The presence of hydroxyl ¹⁵ groups enabled the target polymer **I** to dissolve readily at room temperature in NMP, dimethyl formamide, dimethyl sulfoxide (DMSO), *m*-cresol, and concentrated H₂SO₄ (Table S1). Before the reaction began, **II** was suspended in NMP as fine powders. As **CR** proceeded, the solids became smaller gradually and clear

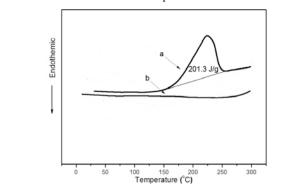
- ²⁰ solution was obtained after 3 h. The reaction was allowed to continue for another 3 h to make sure completion. Since **II** was insoluble, solid state NMR spectroscopy was employed to monitor the reaction process. Fig. 2 displays solid state ¹³C NMR spectra of **II** annealed at 190 °C for 2, 4, 6 and 8 h, respectively,
- ²⁵ together with PPTA. The assignment of each resonance peak was made according to those of the model compound (Fig. S5) and PPTA.²⁴ For **II**, the peak at 157 ppm was attributed to the carbon atom of carbonyl group, whereas that at 147 ppm to the carbon atom on the phenyl ring linked to the ether oxygen atom. The
- ³⁰ methylenic carbon of allyloxy group gave a peak at 68 ppm. After **CR**, the signals belong to carbon atom of amide group shifted downfield to 164 ppm due the decreased shielding effect of *ortho* substituents. The benzylic carbon atom showed absorption at 28 ppm. The absence resonance peaks at 68 and 147 ppm suggested
- ³⁵ that the polymer **II** was completely converted to the polymer **I**. The proton NMR of rearranged samples in deuterated DMSO gave a consistent conclusion (Fig. S6).



⁴⁰ Fig.2 Solid state ¹³C NMR spectra of polymers PPTA and II annealed at 190 °C for 2, 4, 6 and 8 h.

The structures of **II** and **I** were also confirmed by FTIR (Fig. S7). The absorptions at 1193 and 1000 cm⁻¹, attributed to the symmetrical and asymmetrical stretching vibrations of ether bond ⁴⁵ C-O-C of **II**, respectively, vanished after rearrangement reaction. Whereas **I** displayed a broad absorptions centered at 3000 cm⁻¹ typically for hydroxyl group and the bending vibration of O-H

bond at 1320 cm⁻¹. A new absorption at 1166 cm⁻¹ due to the stretching vibration of C-OH bond was also observed.
⁵⁰ The polymers I and II were quite thermal stable and had temperature of 5wt% loss under nitrogen atmosphere above 360 °C (Fig. S8). Fig. 3 exhibits differential scanning calorimetry (DSC) curves of II recorded during first and second heating scans. The first heating curve displays a large exothermic peak centered ⁵⁵ at 220 °C, which was attributed to CR reaction. However, no discernible phase transition was observed on the second heating curve, implying the complete conversion from II to I. This



provided another tool to track CR process of II.

Fig. 3 DSC curves of II recorded during first (a) and second (b) heating scans. Heating rate: $5 \, ^{\circ}C/min$; atmosphere: N_2 .

The simultaneous introduction of allyl and hydroxyl groups not only improved the solubility of PPTA but also changed its surface ⁷⁵ property and intermolecular packing manner. Fig. 4 exhibits the images of water droplets on the surfaces of PPTA and I films, respectively. Because of the dense substitution of polar hydroxyl groups, the polymer I showed a smaller contact angle of 62° than PPTA (72°), an indicative of increase in affinity to water. In the ⁸⁰ powder wide-angle X-ray diffraction pattern (Fig. 5), II presented an intensive peak at $2\theta = 9.2°$ in the low angle region and another one at $2\theta = 26.3°$ in the wide angle region. In addition, two ChemComm Accepted Manuscript

15

60

80

shoulder peaks at $2\theta = 18.4$ and 21.0° , respectively, existed. A highly ordered aggregated structure of macromolecules was implied. After **CR**, **I** showed only two diffused halos in the low and wide angle regions, separately, suggesting an amorphous ⁵ nature. The patterns of both polymer **I** and **II** were totally different from that of PPTA,²⁵ indicating the remarkable roles of side groups. It should be noted that although PPTA is a lytropic liquid crystalline aromatic polyamide, the polymer **I** generated no mesophase. It is probably due to the highly crowded substituents ¹⁰ around polymer backbone.

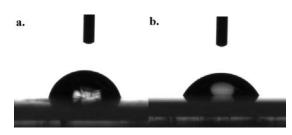


Fig. 4 Water droplets on the films made up of (a) PPTA and (b) I

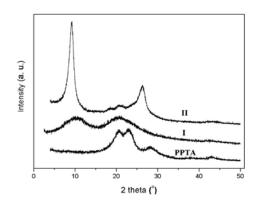


Fig. 5 Wide angle X-ray diffraction curves of polymers II, I, and PPTA

One potential utility of I is to improve the interface structures and interfacial adhesions of Kevlar fiber reinforced composites, as the sizing agent or as a reinforcing material alone, through the

- ²⁰ reactions of allyl and hydroxyl groups with rubber and epoxy, respectively. To demonstrate such an application, the interface property of Kevlar fiber-rubber composite was studied by single fibre pull-out force method. Under identical and un-optimized conditions, the pullout force of fibers treated with as low as 0.1
- ²⁵ and 0.5 wt% solution of I in NMP was 25.0 and 30.1 N, respectively, obviously larger than that of the fiber without treatment (20.8 N) (For details, see supporting information).

In summary, we reported a facile and efficient method to prepare a novel PPTA bearing both apolar allyl and polar

- ³⁰ hydroxyl groups by a combination of the step-growth polymerization and subsequent **CR**. The rearrangement processes of the model compound, **IV**, and the precursor of target polymer, **II**, were completed in 6 h at 170 °C for the former while at 190 °C for the latter without addition of any catalyst as monitored by
- ³⁵ NMR and FTIR. This polymer showed a remarkably improved solubility and hydrophilicity, and may find applications in rubber or epoxy resin based high performance composite acting as a reinforcing material alone or a sizing agent of Kevlar fiber. The

strategy of introducing simultaneously two kinds of reactive ⁴⁰ groups to polymer backbone, developed in the present work, should also be applicable to other aromatic engineering polymers by improving the interface structures and interfacial adhesions. The corresponding work is under way.

This work was supported by National Basic Research Program ⁴⁵ of China (2011CB606106).

Notes and references

^a Beijing National Laboratory for Molecular Sciences, Key Laboratory of Polymer Chemistry and Physics of Ministry of Education, College of Chemistry and Molecular Engineering, Peking University, Beijing 50 100871, China.

E-mail: <u>xhwan@pku.edu.cn</u>.

- ^b State Key Laboratory of Organic Inorganic Composites, Beijing University of Chemical Technology, Beijing 100029, China
- † Electronic Supplementary Information (ESI) available: [The 55 experimental details, ¹³C NMR and FTIR spectra of model compound and its product of Claisen rearrangement reaction, the solubilities, thermal properties and single fibre pull-out force test results of polymers]. See DOI: 10.1039/b000000x/

‡ These authors contributed equally to this work.

- J. K. Fink, High performance polymers. New York: William Andrew Inc.; 2008.
- 2 J. M. García, F. C. García, F. Serna, J. L. de la Peña, *Prog. Polym. Sci.* 2010, **35**, 623.
- 65 3 J. M. García, F. C. García, F. Serna, J. L. de la Peña, in *Handbook of Engineering and Specialty Thermoplastics*. John Wiley & Sons, Inc. 2011, p. 141-181.
- 4 U. S. Agarwal, D. V. Khakhar, *Nature* 1992, **360**, 53.
- 5 S. Du, J. Zhang, Y. Guan, X. Wan, Aust. J. Chem. 2013, 67, 39.
- 70 6 Y. Guan, Y. Zheng, J. Cui, X. Wan, *Chinese J. Polym. Sci.*, 2010, 28, 257.
- 7 M. Takayanagi, T. Katayose, J. Polym. Sci.: Polym. Chem. Ed. 1981, 19, 1133.
- 8 L. Kong, R. B. Li, Y. Zhao, H. F. Shi, D. J. Wang, D. F. Xu, J. G. Wi, 75 Spectrosc. Spect. Anal. 2012, **32**, 2647.
- 9 M. Steuer, M. Hörth, M. Ballauff, J. Polym. Sci. Part A: Polym. Chem. 1993, **31**, 1609.
- 10 H. R. Kricheldorf, A. Domschke, Macromolecules 1994, 27, 1509.
- 11 J. Y. Jadhav, W. R. Krigbaum, J. Preston, Macromolecules 1988, 21,
- 538.
- 12 H. R. Kricheldorf, B. Schmidt, *Macromolecules* 1992, 25, 5471.
- 13 H. R. Kricheldorf, B. Schmidt, R. Buerger, *Macromolecules* 1992, 25, 5465.
- 14 T. I. Bair, P. W. Morgan, F. L. Killian, *Macromolecules* 1977, **10**, 1396.
 - 15 A. K. Chaudhuri, B. Y. Min, E. M. Pearce, J. Polym. Sci.: Polym. Chem. Ed. 1980, 18, 2949.
 - 16 S. Kim, E. M. Pearce, *Makromol. Chem.* 1989, 15, 187.
 - 17 S. Viale, A. S. Best, E. Mendes, W. F. Jager, S. J. Picken, Chem. Commun. 2004, 1596.
- 18 S. Viale, A. S. Best, E. Mendes, S. J. Picken, *Chem. Commun.*2005, 1528.
- 19 S. Viale, W. F. Jager, S. J. Picken, *Polymer* 2003, 44, 7843.
- 20 S. Viale, N. Li, A. H. M. Schotman, A. S. Best, S. J. Picken, *Macromolecules* 2005, **38**, 3647.
- 21 K. S. Lee, US. Patent 20070015896-A1, 2007.
- 22 B. H. Glomm, A. G. Oertli, C. Rickert, P. Neuenschwander, U. W. Suter, *Macromol. Chem. Phys.* 2000, 201, 1374.
- 23 A. M. Martín Castro, Chem. Rev. 2004, 104, 2939.
- 100 24 A. D. English, J. Polym. Sci. Polym. Phys., 1986, 24, 805.
 - 25 M. Panar, P. Avakian, R. C. Blume, K. H. Gardner, T. D. Gierke, H. H. Yang, J. Polym. Sci.: Polym. Phys. Ed., 1983, 21, 1955.

This journal is © The Royal Society of Chemistry [year]