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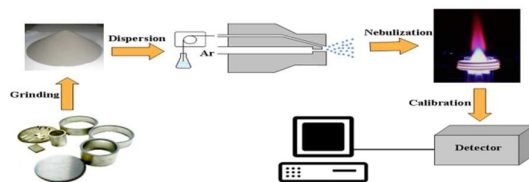
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### A table of contents entry



The current state of research on slurry nebulization in plasmas for the analysis of advanced materials is thoroughly surveyed.

# Slurry Nebulization in Plasmas for Analysis of Advanced Ceramic Materials

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**Abstract:** Slurry nebulization in plasmas has advantages of simplicity, high speed, low cost, minimized analyte loss, and low risk of sample contamination, but it has not been very widely adopted. However, the study of advanced ceramic materials has recently renewed enthusiasm for this technique. In this paper, the current state of research on slurry nebulization in plasmas for the analysis of advanced materials is thoroughly surveyed. Sample preparation, sample characterization, and modifications of the instrumentation necessary for slurry nebulization are reviewed, along with calibration procedures and studies on fundamental issues. Finally, the applications of this method to a variety of advanced materials are summarized, and the outlook for this method is discussed.

## 1. Introduction

The development of structural and functional ceramics for modern industrial applications and electronic devices requires interdisciplinary investigations in the fields of mineralogy, materials science, materials processing, and materials testing in general, as well as in analytical science in particular<sup>1</sup>. Advanced ceramics are synthesized from pure or alloyed oxides, nitrides, carbides, and borides with rigidly defined structures, shapes, and chemical compositions. In particular, oxide-based materials (aluminium oxide (Al<sub>2</sub>O<sub>3</sub>), titanium dioxide (TiO<sub>2</sub>), zirconium oxide (ZrO<sub>2</sub>)), non-oxide-based nitrides, carbide materials (aluminium nitride (AlN), titanium nitride (TiN), silicon carbide (SiC), silicon nitride (Si<sub>3</sub>N<sub>4</sub>), boron carbide (B<sub>4</sub>C)), and their chemical compounds have all been used as raw materials for advanced ceramic synthesis.

The fantastic properties of advanced ceramic materials, such as their high-temperature stability, strength, wear resistance, corrosion/oxidation resistance, and electrical optical and/or magnetic properties, rely greatly on their chemical composition and purity.<sup>2</sup> In particular, advanced ceramics are usually of high purity, and in a number of cases even a trace level of impurities severely affects ceramic performance.<sup>3,4</sup> Furthermore, small amounts of dopants, sintering aids, and other additives are normally spiked into these ceramics to improve their processing and ultimate properties.<sup>5</sup> Therefore, the chemical composition of the powders used

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3 for the production of advanced ceramics must be determined with a high precision, down to  
4 trace concentrations, for manufacturing control, property improvement, failure prevention, and  
5 quality assurance. Furthermore, because the chemical composition of advanced ceramics,  
6 especially the impurities, should be exactly controlled for their continuing high quality, analysis  
7 of not only the raw material but also the sintered material is required.  
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11 Therefore, powerful analytical methods and strategies are required to determine the main  
12 elements and the impurities of raw and sintered materials. Since many elements have important  
13 effects on the properties of advanced ceramics, only methods with a high multi-element capacity  
14 should be applied, such as inductively coupled plasma optical emission spectrometry/mass  
15 spectrometry (ICP-OES/MS). Traditionally, samples have been prepared as solutions using  
16 digestion steps such acid dissolution or fusion. This facilitates introduction, calibration, and  
17 homogenization. However, advanced ceramics are difficult to digest even by microwave-assisted  
18 digestion.<sup>6</sup> Alkali fusion procedures have the disadvantages of high reagent blanks due to the  
19 fluxes and large dilution factors, both of which lead to a sensitivity decrease. Meanwhile, some  
20 elements may not be detected, especially trace elements. Methods based on, for example, fusion  
21 and digestion using acids can result in incomplete dissolution of the sample, evaporative losses  
22 of the more volatile elements, and contamination problems. For these reasons, sample  
23 preparation for advanced ceramic materials is still the Achilles' heel of ICP-OES/MS as a  
24 microquantity analysis technique. The direct introduction of solids or slurries into plasmas  
25 would circumvent these difficulties and markedly reduce the sample preparation time by  
26 combining the matrix destruction and analyte atomization and excitation into a single step. The  
27 solid samples can be introduced into the plasma using a number of well-established techniques.  
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31 Direct sample insertion with a graphite rod as a support and a sample elevator is commonly  
32 used to introduce a solid into an ICP.<sup>7</sup> The inductive heating of the carbon vaporizes the sample  
33 directly into the plasma. Direct insertion may provide low limits of detection and a wide  
34 dynamic range, but it suffers from the high temperature stability (i.e., high vaporization  
35 temperature) of refractory materials and ceramics consisting of stable carbide-forming elements.  
36 Another important requirement is the need for closely matching standards. Because of these  
37 practical limitations, direct introduction of powders into the ICP-OES has not been exploited  
38 successfully to its full potential for routine analysis in spite of its many import advantages.<sup>8,9</sup>  
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42 Laser ablation (LA) is increasing in popularity as a method of direct solid sample introduction.  
43 Specially, LA coupled with ICP-MS has been become a well-established, powerful, rapid, and  
44 sensitive method for analysing trace elements without extensive sample preparation.<sup>10-12</sup>  
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46 However, the lack of appropriate standards or certified reference materials for a wide variety of  
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3 samples of interest is the most important restriction for quantitative analysis by LA-ICP-MS. For  
4 ceramics in particular, suitable standards are often not available, resulting in a need of for  
5 calibration methods. LA-ICP-MS also suffers from non-stoichiometric effects during sampling,  
6 aerosol transport, vaporization, atomization, and ionization in ICP, which are collectively  
7 described as elemental fractionation. Furthermore, quantitative analysis of powdered materials  
8 like advanced ceramic powders is especially difficult. For these reasons, laser ablation has  
9 usually been applied to compact ceramic samples but not to ceramic powders.

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11 Electrothermal vaporization (ETV) has often been combined with other techniques such as  
12 ETV-ICP-OES or ETV-ICP-MS to provide a high sample introduction efficiency, low sample  
13 volume, high absolute detection power, and the possibility of removing certain sample  
14 components before the vaporization step or of using chemical modifiers.<sup>13,14</sup> However, the  
15 condensation step during transport to the plasma, memory effects, soaking of the sample into the  
16 furnace material, blanks, chemical reactions, and spectral interference caused by the furnace  
17 material can limit the applicability of this technique. Generally, it is difficult to completely  
18 vaporize ceramic powder from the ETV device because of its high thermal stability.

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20 Another approach to solid sample introduction is slurry nebulization, which involves the  
21 nebulization of (usually aqueous) suspensions of fine powders into the ICP. This technique also  
22 requires only simple preparation procedures.<sup>15-17</sup> Extreme temperature conditions are usually  
23 utilized in ICP, which is a tremendous potential advantage that has been neglected in other  
24 methods. In general, the plasma is only used to ignite the light source. However, the  
25 high-temperature environment could also be used to thermally decompose samples into atoms or  
26 ions. Therefore, nebulization of suspensions of powdery samples and subsequent introduction of  
27 the nebulized samples into the ICP torch, i.e., slurry nebulization, has been received particular  
28 attention in the past two decades.<sup>16,17</sup> Because of the similarities between the rheological  
29 properties of suspensions and solutions, the technique is simple to operate, and only a few  
30 instrumentation modifications (nebulizer and spray chamber) are required. In addition, it can  
31 remarkably reduce the analysis time by combining the sample decomposition and analyte  
32 atomization and excitation into a single step.

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34 Fig. 1 shows the number of the publications on slurry nebulization into plasmas published  
35 from 1996 to 2013 and reflects its popularity as an alternative to conventional aqueous sample  
36 introduction. It is interesting to see that the number of published papers has not varied drastically  
37 over the past 20 years, although the number of publication has decreased in the last several years.  
38 Slurry nebulization techniques have been successfully applied in many different fields such as  
39 bioorganic and inorganic materials analysis, and many of the early difficulties in achieving a  
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3 homogeneous and stable dispersion have been overcome. It has also shown advantages such as  
4 simple implementation, low cost, the requirement for only slight instrument modification, and  
5 easy calibration using aqueous solutions in a way similar to solution nebulization. The aim of  
6 this article is to survey the current status of slurry nebulization in ICP with a special focus on the  
7 work on advanced ceramic materials in latest twenty years. The outlook for this method is also  
8 discussed.  
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## 13 14 **2. Slurry Preparation**

15 The most critical step for successful analysis of advanced ceramic materials using ICP is  
16 probably the slurry sample preparation, which can determine the analysis time, trueness,  
17 precision, and reliability of the analysis results. The ideal slurry sample should be highly stable  
18 and homogeneous, and the slurry particle size should be sufficiently small.<sup>15</sup> In general, the  
19 principal steps in sample preparation include sample grinding, dispersion reagent addition, and  
20 characterization of the slurry homogeneity and stability. These steps are each discussed in the  
21 present work.  
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### 28 **2.1 Grinding methods**

29 The particle size is the limiting parameter for efficient slurry nebulization, and the maximum  
30 particle size of the slurry must be such that any single solid particle can occupy an aerosol  
31 droplet. To achieve a particle size distribution that would yield results similar to those of an  
32 equivalent equimolar aqueous solution, a wide range of grinding techniques have been employed.  
33 Technically, the slurry particles have to be sufficiently small (down to the nanometre scale) to be  
34 able to be analysed using ICP. Even if the ceramic materials are in a powder form, they may still  
35 require grinding to meet the requirements for slurry nebulization. Ebdon *et al.*<sup>16</sup> reported that  
36 during slurry nebulization, the particle size in a slurry for ceramic analysis should be smaller  
37 than 5–10  $\mu\text{m}$ . Their rigorous study showed that the slurry particles larger than 5  $\mu\text{m}$  (in some  
38 reports 2  $\mu\text{m}$ ) could not reach the plasma, which resulted in signal loss. Wang *et al.*<sup>18</sup> suggested  
39 that the particle size in some ceramic analyses should be reduced to the submicron level, because  
40 otherwise there would be a negative deviation in the analytical results due to the incomplete  
41 evaporation of the slurry particles.  
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52 Most ceramics have a high hardness and are therefore difficult to grind. The attrition materials  
53 used in mills and grinders such as zirconium and tungsten carbide have a Mohs value  
54 comparable to those of most ceramics, which can lead to poor grinding efficiency and easy  
55 contamination if a long grinding period is used. A wide range of grinding methods has been used  
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3 for slurry preparation for direct analysis in ICP. The main methods utilized for ceramic powders  
4 reported in the literature are the bottle and bead method, vibration mill, and puck-type grinder  
5 with tungsten carbide utensils. Table 1 lists the various mills and size reduction methods that  
6 have been used to date, with varying degrees of success, to grind a wide range of materials. The  
7 obtained particle size ranges are also tabulated. The choice of grinding or milling agent is also  
8 important and will affect the analytical accuracy. In particular, the grinding material should be  
9 harder than the material being ground and should not contain elements that will interfere with  
10 the analysis.

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16 The bottle and bead method<sup>19-21</sup> has been widely used to prepared slurries from a wide variety  
17 of ceramic materials, as shown by Table 1. Approximately 0.1 to 1 g of sample is weighed into  
18 30 mL polypropylene bottles, and then 10 g of zirconia beads and an aqueous dispersing agent  
19 are added. Afterward, the bottles are sealed and shaken on a mechanical laboratory shaker for the  
20 amount of time required to produce a slurry in the appropriate size range, often overnight for  
21 convenience. The resultant finely ground slurry can then be transferred to a calibrated flask  
22 through a coarse sieve that retains the grinding material and diluted to the desired volume with a  
23 dispersant. This technique is inexpensive and offers the advantage of grinding a number of  
24 samples simultaneously, depending on the type of flask shaker used.

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Vibration mills<sup>19,22,23</sup> consist of a torus-shaped or cylindrical shaped shell. The solid is  
contained in the shell, together with the beads for grinding, and vibrated. The grinding material  
and sample collide with each other and the shell, resulting in the breakdown of solid particles.  
The mills can be run dry or wet, and steel, agate, and tungsten carbide can be used as grinding  
materials.

A puck-type grinder reduces the particle size through the action of a spinning puck and a ring  
inside a grinding container.<sup>24,25</sup> The grinding container may be made with materials resistant to  
abrasion, e.g., hardened steel, tungsten carbide, agate, or alumina ceramic. Alumina ceramic in  
particular resistant to abrasion and lightweight, as it is almost pure aluminium oxide with trace  
amounts of silicon, calcium, and magnesium, although it is brittle.

Isoyama *et al.*<sup>26</sup> investigated the use of an ultrasonic grinding technique for direct analysis of  
slurries by ICP-OES. In this method, a sample block was ground with the same sample pestle  
fixed on an ultrasonic horn along with a sample coarsely crushed in water for 10 min. In result,  
10 to 20 mg of the finely ground sample (0.5 to 0.7 mm of mean particle size) was obtained.  
Reduced contamination from the grinding tools was achieved even though very fine samples of  
less than 1 mm were obtained for sintered ZrO<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> ceramic materials.

Recently, the jet-milling method has been used to further decrease the particle size.<sup>22</sup> In



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3 general, jet milling systems mainly consist of a grinding chamber and a classifying chamber.  
4 After entering the grinding chamber, the raw particles are accelerated by means of a jet stream of  
5 carrier gas, and extremely high-energy mechanical impact collision occurs among particles and  
6 between particles and rigid obstacles. As a result, the powder particles can be efficiently  
7 smashed and refined. After grinding, the particles are carried out by the carrier gas and flow up  
8 to the classifier chamber with a rotating rotor, where classification takes place.

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11 However, advanced ceramic materials with high measure of hardness (MOH) values may still  
12 be difficult to grind. For example, the attrition materials used in mills and grinders such as  
13 zirconia and tungsten carbide may have comparable MOH values, resulting in poor grinding  
14 efficiencies and even contamination if long periods are employed for size reduction. In general,  
15 the ceramic sample should be ground for particle size reduction. Several points should be  
16 considered in particle size reduction step: the type of mill and grinding methods, milling  
17 container, milling speed, milling time, and the desired particle size and particle size  
18 distribution.<sup>28</sup>

## 27 28 **2.2 Dispersion of the slurry**

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30 Preparing slurries in an aqueous solution alone is unsuitable for the majority of samples owing to  
31 flocculation effects, which result in rapid sedimentation of the finely powdered material. It is  
32 therefore essential to prepare a stable and homogeneous slurry that will yield a stable,  
33 homogeneous aerosol for introduction into the plasma to obtain accurate and precise analytical  
34 results. This is achieved by employing stabilizing agents, which are commonly termed  
35 dispersants or surfactants.

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41 Slurry stabilization can be described by the DLVO (Derjaguin and Landau, Verwey and  
42 Overbeek) theory.<sup>29,30</sup> In this approach, the electrostatic interactions between the charged  
43 particles caused by their surface electrical potential, the so called zeta potential, are calculated.  
44 The effects of van der Waals attractions and electrostatic repulsions due to the double layer of  
45 counter ions are then both considered. Slurry flocculation occurs when the van der Waals  
46 attraction overweighs the electrostatic repulsion. This slurry stabilization mechanism can be  
47 influenced by the surface properties of the sample powder to be analysed. Therefore, some  
48 preparation conditions, such as the medium pH and the type and amount of dispersants, should  
49 be experimentally optimized in order to prepare a stable and homogeneous slurry.

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61 Many reports<sup>15,18,26,31,32</sup> have discussed the effects of the pH of aqueous ceramic suspensions  
62 on colloidal stability as well as the measurement precision in slurry nebulization ICP-OES  
63 analysis. Farinas *et al.*<sup>15</sup> published an excellent study regarding the colloidal stability of ceramic



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3 suspensions for slurry nebulization in ICP-OES. They reported a case study using alumina  
4 slurries with different pH conditions and stabilization agents. The paper discusses the different  
5 types of stabilizing additives available for ceramic suspensions and also their best uses. The  
6 authors explain the electrostatic stabilization mechanism in detail and conclude that in a ceramic  
7 suspension,  $H^+$  and  $OH^-$  ions determine the potential. In the case of  $Al_2O_3$ , the charge on the  
8 surface of any slurry particle is negative, resulting in the attraction of protons to the surface and  
9 creating a proton concentration gradient from the particle surface to the liquid. Conversely, the  
10  $OH^-$  ion concentration is decreased near the surface of the particle and increases with distance  
11 from the surface. A charged layer, known as the electrical double layer, is formed because of the  
12 potential gradient. According to the Nernst equation, the surface potential of the particle can be  
13 changed by changing the concentration of the potential-determining ion, and the relative amount  
14 of ions adsorbed on the surface of the slurry particle will change as a result.

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23 If the surface of the solid in the preferred liquid carrier is lyophobic, the powder will be  
24 difficult to disperse, whereas if it is lyophilic, the powder will disperse easily. To obtain a  
25 suitable dispersion of lyophobic particles,<sup>7</sup> stabilization agents are added to wet the surface, so  
26 that the particles become lyophilic. There have been few in-depth studies regarding the  
27 stabilizing effect of dispersants. At a particular concentration of the potential-determining ion,  
28 the positive and negative surface activities will be equal, and the overall potential at the surface  
29 of the particle will be zero. In this scenario, no double layer will exist, and agglomeration of the  
30 particles will occur at a concentration known as the isoelectric point (IEP). The addition of a  
31 dispersant causes the zeta potential to change and ultimately acts to stabilize the slurry. Triton  
32 X-100 was found not to stabilize alumina slurries, and no real dispersion was achieved with  
33 Kodak photoflow in glycerol. PKV-5088 was unable to prevent sedimentation of alumina despite  
34 being commonly used to disperse non-oxide ceramics. Three dispersants, Dolapix PC-33,  
35 Darvan-C, and Darvan-7, were found to produce the desired dispersion of an alumina slurry. The  
36 intensity and precision of the ICP-OES measurement were found to be directly related to the  
37 stability offered by the dispersant system. Wang *et al.* prepared an SiC slurry with polyethylene  
38 imine (PEI),<sup>19</sup> AlN suspensions with polyacrylate amine ( $NH_4PAA$ ) or PEI,<sup>18</sup> a  $TiO_2$  slurry with  
39  $NH_4PAA$ ,<sup>24</sup> and TiN suspensions with  $NH_4PAA$  or PEI.<sup>25</sup> Before nebulization, each slurry was  
40 agitated in an ultrasonic bath for 15 min to ensure stable dispersion.

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The authors found that the dispersant may yield the required stability through an electrostatic  
stabilizing mechanism by (i) controlling the pH (using potential determining ions), (ii) using  
inorganic electrolytes, or (iii) using a polyelectrolyte whose long-chain polymers adsorb on the  
surfaces of the particles and prevent contact between them. Various dispersants used in slurry

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3 preparations are summarized in Table 2, including glycerol, sodium hexametaphosphate applied  
4 alone or mixed with monoisopropanolamine, NH<sub>4</sub>PAA, and PEI.  
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### 7 8 **2.3 Slurry concentration**

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10 In direct sample-introduction systems, the slurry concentration is important because it strongly  
11 affects the overall sensitivity. Very dilute slurries may cause a decrease in precision, whereas at  
12 high slurry concentrations the plasma stability and nebulization efficiency of the sample may be  
13 significantly reduced. The upper limit of the concentration depends on the sample matrix being  
14 analysed.  
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17 Zachariadis *et al.*<sup>43,44</sup> conducted some preliminary experiments using a cyclonic spray  
18 chamber combined with a Babington-type nebulizer. Slurry solutions with concentrations up to  
19 10% (m/v) were tested. They found that slurry solutions with concentrations up to 2.5% (m/v)  
20 can be easily aspirated without any significant influence of the concentration on the plasma or  
21 baseline stability.  
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26 The effects of the ceramic suspension viscosity should also be taken into consideration for  
27 very concentrated slurries, because viscosity is known to influence the aspiration rate and thus to  
28 alter the nebulization efficiency. As a general rule, concentrated ceramic suspensions have very  
29 poor rheological properties, especially for nanosized powders. The slurry cannot flow when the  
30 solid content reaches 30% (m/v) at any pH value. Wang *et al.*<sup>45</sup> prepared a unique SiC slurry  
31 sample containing 30 wt% solid in order to detect the trace elements in SiC powders. However,  
32 the viscosity of the suspension was only 40 mPa·s after the dispersant was added. Thus, the  
33 flowability was significantly improved to near that of an aqueous solution, and a linear signal  
34 enhancement was observed up to a slurry concentration of approximately 20% (m/v).  
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41 From these results, it can be concluded that the effects of viscosity and non-Newtonian  
42 rheological phenomena should be considered for very concentrated slurries. In addition, whereas  
43 conventional acid dissolution procedures produce solutions with a 1% sample content, the ability  
44 to use slurries with a sample content of 20% or more has clearly yielded major advantages in  
45 trace analysis.  
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### 50 51 **3. Slurry characterization**

52 In the direct analysis of a slurry, it is necessary to control the stability and homogeneity of the  
53 slurry to produce a reliable aerosol for introduction into the plasma and thus provide precise and  
54 accurate analytical results. Furthermore, even more caution is needed for the analysis of ceramic  
55 powders using techniques for slurry nebulization into plasma, because the transport and  
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3 nebulization efficiencies are strongly influenced by the stability and homogeneity of the aqueous  
4 slurries under these conditions.  
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### 7 8 **3.1 Particle size and its distribution**

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10 The stability and homogeneity of the slurry are influenced by the particle size. Recently, various  
11 methods have been developed to measure the particle size and its distribution in slurries, and the  
12 method adopted for measurement of the particle size generally depends on the available  
13 instrumentation.  
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16 Photosedimentometry combines photoelectric measurement and gravitational settling of  
17 particles. A narrow beam of light is passed through the slurry and onto a photocell. As particles  
18 begin to fall and leave the light beam, they are replaced by particles settling from above. The  
19 attenuation of the beam of light is directly related to the surface area of the particles in the light  
20 beam, and from this relationship, the particle size distribution can be determined. The method is  
21 simple and the instrument is inexpensive, and thus photosedimentometry is widely employed for  
22 particle determination.<sup>18,19,24</sup>  
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28 For more sensitive analysis, SEM is commonly used to measure the mean particle size in a  
29 slurry, as well as the dispersion stability of the slurry.<sup>18,19,25</sup> The particle size and distribution of  
30 AlN can be observed on the SEM screen (Fig. 3). The dispersing effect of the PEI can also be  
31 verified by comparing the SEM micrographs of sediments. Large flocs are observed in the  
32 sediments of suspensions without PEI, while particles of the sediments with 2 wt% PEI are well  
33 dispersed and not associated with each other.<sup>18</sup> SEM has a large depth of focus and can yield  
34 significant information about the size, shape, and surface characteristics of the particles.  
35 However, only information on spatially confined areas can be obtained.  
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41 Laser diffraction methods<sup>18,19,46</sup> can also be used to assess the dispersion efficiency of a slurry.  
42 Diffraction of light, which can be thought of as the bending of light waves, occurs at the surfaces  
43 of the particles because of slight differences in the path lengths of the light waves created upon  
44 interaction with the particle surfaces. Laser diffraction methods are rapid, do not require the use  
45 of dispersants, and can be employed with a range of slurry concentrations. However, the  
46 instrumentation required is relatively expensive. Isoyama *et al.*<sup>26</sup> measured the mean particle  
47 sizes of ground sample powders using a laser particle size analyser and SEM images. The mean  
48 diameter of the sample obtained by grinding the 0.25–0.5 mm coarsely crushed sample fraction  
49 was less than 1 mm, and the particle size distribution curve was extremely sharp.  
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### 57 58 **3.2 Slurry stability**

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3 As discussed above in section 2.2 regarding the dispersion of the slurry, the stability of the  
4 slurry can be theoretically explained and characterized by the zeta potential, which is a measure  
5 of the electric field resulting from the particle surface charge. As mentioned, the zeta potential  
6 can be altered by the pH and/or surfactants, and the zeta potential becomes zero, meaning that  
7 the surface charge disappears, at a specific pH value called the IEP. At the IEP, the suspended  
8 particles would agglomerate rapidly owing to the absence of the repulsive force. Thus, the  
9 greater the absolute value of the zeta potential, the more stable the slurry is. Wang *et al.*<sup>19</sup> studied  
10 the relationship between the zeta potential of an SiC slurry and pH, and Broekaert *et al.*<sup>36</sup>  
11 analysed similar relationships for ceramic materials (Al<sub>2</sub>O<sub>3</sub>, SiC, and ZrO<sub>2</sub>) by ICP-OES. The  
12 refractory materials were added in water and sonicated during 10 min in an ultrasound bath. A  
13 surfactant was not needed to stabilize the slurry, because the ultrasonic agitation guaranteed  
14 homogenization.  
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23 The additives adsorbed onto the particle surface create a large energy barrier to prevent the  
24 colloid particles from flocculating. Because each individual material has different surface  
25 properties, as a rule, an appropriate dispersant must be selected on a case-by-case basis.<sup>18,23-25</sup>  
26 Slurries of the same ceramic material with different particle sizes usually require different  
27 amounts of dispersant. Nanosized powders have larger specific surface areas than submicron  
28 powders and thus require more additive to disperse the particles.<sup>18,23-25,45</sup> Clearly, the  
29 concentration of a dispersant must be controlled, because excess dispersant is detrimental to the  
30 slurry stability and can cause coagulation.<sup>18</sup>  
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36 Wang *et al.*<sup>19,25</sup> studied the stability of slurries by measuring the relationship between  
37 ICP-OES signal intensity and time. The submicron SiC slurry with no dispersant at the IEP (zeta  
38 potential is zero) displayed unstable properties. The normalized line intensities of the matrix Si  
39 and the Al and Fe impurities obviously decreased with time. In contrast, the slurry with 2.0 wt%  
40 PEI as a dispersant, the zeta potential of which was far from zero, displayed relatively stable  
41 signal intensities.<sup>19</sup>  
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### 47 **3.3 Viscosity determination**

48 The viscosity influences both the aspiration rate and the nebulization process. Good fluidity of  
49 the slurry, which implies a low viscosity, improves the nebulization efficiency. However, there  
50 has been little research on the viscosity in terms of its influence on the nebulization. Farinas *et*  
51 *al.*<sup>15</sup> reported on the rheology of ceramic slurries and discussed the basic concept of rheology.  
52 Wang *et al.*<sup>45</sup> prepared a slurry for viscosity measurement with the help of ultrasonic stirring,  
53 followed by magnetic stirring for 20 min. The viscosity was determined using a cylindrical  
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3 measurement system on a rotational viscometer. The viscosity measurements for all slurries were  
4 conducted at 25 °C within 30 min after preparation. To eliminate the artefacts from different  
5 treatments during the filling procedure, the samples were presheared for 3 min, followed by 5  
6 min of rest. The viscosity can be kept very low even when the SiC slurry concentration is as high  
7 as 30% (m/v) when 2.0% PEI is added and the medium has pH 4. In fact, such a  
8 high-concentration SiC slurry can be normally nebulized for ICP-OES analysis without any  
9 problem.  
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#### 15 16 **4. Slurry sample introduction system**

17 As mentioned previously, because of the similarity between the rheological properties of  
18 suspensions and solutions, slurry nebulization in plasmas is simple to achieve with only a few  
19 instrumentation modifications in the nebulizer and spray chamber.  
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22 Although there are many similarities, there exist some differences in rheological properties  
23 between suspensions and solutions. The primary difference is that the suspensions contain  
24 particles. Theoretically, the solution sample can be divided infinitely in the nebulization process,  
25 whereas the slurry particles cannot be divided. Therefore, the sizes of aerosol droplets must be  
26 larger than the slurry particles, and clogging will occur when a conventional concentric  
27 pneumatic nebulizer is employed. Thus, the most critical consideration in the design of a  
28 nebulizer for slurry nebulization is to prevent clogging. Fig. 2 shows the different nebulizers  
29 used for the slurry nebulization into plasma.  
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36 Many nebulizers suited to slurry samples were reported in the early articles on this topic. The  
37 most popular designs for handling slurries are based on the V-groove Babington-type  
38 nebulizer.<sup>33,46,47,48</sup> The slurry sample flows along a V-groove that has a small orifice through  
39 which the carrier gas blows out at high flow rates. This causes the sample to shatter into an  
40 aerosol of small droplets. The high pressure of the argon gas ensures efficient nebulization and  
41 prevents blocking of the small orifice.  
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46 The Ebdon nebulizer<sup>49-51</sup> is a commonly used V-groove variation of the Babington nebulizer.  
47 It has been shown to exhibit excellent performance for slurry nebulization and has a robust  
48 one-piece design. The diameter of the gas orifice is 0.2 mm to allow a high-velocity gas flow to  
49 be used for nebulization.  
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52 However, in recent years, there have been few reports on nebulizer designs for slurry  
53 nebulization, probably because many commercial nebulizers are available in the market. Three  
54 different nebulizers were tested by Kollander *et al.*,<sup>52</sup> including glass expansion concentric  
55 (GEC), cross-flow, and Burgener T2100 nebulizers. They all performed roughly equally well in  
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3 terms of the quantitative results obtained for the slurries, but the GEC nebulizer yielded better  
4 sensitivity (larger slope of the calibration curve) for all elements. The cross-flow and Burgener  
5 T2100 nebulizers each yielded about the same sensitivity when the same instrument settings  
6 were used. In particular, the GEC nebulizer yielded about 1.4 to 2.5 times higher sensitivity than  
7 the Burgener T2100 nebulizer. The GEC nebulizer is known to create a very fine mist with small  
8 droplets, whereas the Burgener nebulizer allows larger droplets to form because of its wider  
9 inner diameter. These larger droplets are partly filtered by the spray chamber, and consequently  
10 less analyte reaches the plasma, resulting in a lower sensitivity. In addition, a slightly lower Mg  
11 content was observed when the cross-flow nebulizer was used, which indicates a higher matrix  
12 load in the plasma than is obtained with either the GEC or Burgener nebulizer. Increases in  
13 plasma load usually lead to decreases in temperature and thus decreases in sensitivity due to  
14 reduced atomization and excitation.

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16 The spray chamber has the same importance as the nebulizer in slurry nebulization. After  
17 passing through the nebulizer outlet, the aerosol droplets are separated based on their different  
18 sizes. Most large droplets are removed to drain, while a small fraction of droplets are carried to  
19 the plasma torch. According to the Ebdon's study,<sup>16</sup> only approximately 1%–2% of the samples  
20 can reach the plasma.

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22 Ebdon *et al.*<sup>46</sup> reported that good recovery efficiency could be obtained by using a  
23 laboratory-constructed double-pass spray chamber made of glass. The double-pass design is  
24 used to eliminate the considerable dead volume in a single-pass spray chamber, which improved  
25 the recovery efficiency for a sewage sludge analysis. There are also many chamber designs  
26 available for slurry nebulization. Williams *et al.*<sup>53</sup> used a single-pass water-cooled spray chamber  
27 worked at 13 °C. The cold spray chamber reduced the noise by lowering the vapour loading in  
28 the plasma and therefore improved the precision. Just recently, a commercial reduced-volume  
29 Sturman–Masters-type spray chamber made of poly(tetrafluoroethylene) has come on the market  
30 and has been used by many researchers.<sup>18,19,22-25,45,46</sup>

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32 Zachariadis *et al.*<sup>43,44</sup> examined different configurations of the spray chamber and nebulizer  
33 for direct aspiration of slurry samples into the plasma. As mentioned above in the discussion of  
34 slurry concentration in section 2.3, they found that slurry solutions with concentrations up to  
35 2.5% (m/v) can be easily aspirated without significant influence of the concentration on the  
36 plasma or baseline stability by using a Babington-type nebulizer combined with a cyclonic spray  
37 chamber. The combination of a Scott-spray chamber and a cross-flow nebulizer had a lower  
38 concentration tolerance: slurries containing only up to 1.0% (m/v) powder could be sufficiently  
39 aspirated. However, the combination of a cross-flow nebulizer with a double-pass spray chamber  
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exhibited better performance in terms of overall precision.

## 5. Optimization

Procedures for the optimization of the operating variables are increasingly being used to enhance the sensitivity and performance of the analysis of slurries using plasma. Some slurries may be difficult to volatilize and atomize, which could be overcome through the optimization of the plasma conditions.

Zaray *et al.*<sup>39</sup> used three-dimensional plots to identify the optimum rf power (0.8 kW) and aerosol gas pressure (2 bar) for the plasma for determination of Al, Ca, Fe, and Mg in silicon nitride powder. At the optimum conditions for the elemental analyses, the atomization efficiencies ranged from 77% to 100%.

Gervais and Salin<sup>38</sup> optimized their procedure for the ICP-OES analysis of slurries using a heated sample-introduction system. Simplex optimization was performed to optimize the viewing height and temperature. The signal-to-blank ratio for six elements was used as the criterion of merit, and a variation of 5% between the five best vertices was used as an indicator of the optimum being found. The simplex method was also used for the optimization of the forward powder, sample uptake rate, nebulizer gas pressure, plasma gas flow rate, and viewing height for the determination of Al in ZrO<sub>2</sub> slurries.

The thermodynamic properties of the ICP itself can be altered by modifying one or all three of the gas flows used (i.e., N<sub>2</sub>, He, or O<sub>2</sub>). The excitation temperature has been measured to be around 5500 K for a 1.5 kW Ar plasma, but it increases to 6800 K upon addition of 5% O<sub>2</sub>. Identical mixed gas plasmas have been used to enhance the sensitivity of ICP-OES analysis by eliminating particle size and sample composition effects.

Xhoffer *et al.*<sup>54</sup> investigated the effect of the addition of N<sub>2</sub> and O<sub>2</sub> to the plasma gas flow on the exhaust particles during the analysis of silicon carbide by ICP-OES. Both N<sub>2</sub> and O<sub>2</sub> were found to have no effect on the shape or morphology of the submicron aerosols. However, O<sub>2</sub> appeared to change the chemical composition of the exhaust SiC particles, whereas N<sub>2</sub> did not.

## 6. Calibration techniques

The calibration of the instrument is the most difficult step in the ICP-OES analysis of ceramic slurries. The major problems caused by the presence of solid particles during the direct analysis of slurries by ICP are as follows: (i) the analyte in the slurry must be physically transported into the plasma with the same efficiency as the solution, and (ii) particle-plasma interactions give rise to interference effects that are influenced by the size and chemical composition of the solid



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3 particles.

### 4 5 6 **6.1 Use of standard slurries**

7 Calibration with slurries prepared from standard reference materials is the ideal method.<sup>19,55</sup>  
8 Standard suspensions are prepared by homogenizing a known amount of the reference materials.  
9 However, the use of standard slurries for calibration is problematic because few ceramic  
10 standard reference materials meet the requirements for routine ceramic analysis. Furthermore,  
11 the standard reference material should be strictly matched with the analytical sample in terms of  
12 composition, densities of elements, and particle size distribution, which is often difficult.  
13 Nevertheless, Fernandez *et al.*<sup>55</sup> used a reference slag for the determination of a range of  
14 elements. A range of geological standard reference materials have also been made from materials  
15 including SiO<sub>2</sub>-rich granites, marine sediments, and coals.  
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### 24 **6.2 Use of aqueous standards**

25 For a slurry to be efficiently nebulized, vaporized, and atomized, the particle size distribution is  
26 crucial. When the introduced slurry is completely desolvated by evaporation and the mean  
27 particle size less than 5 μm,<sup>56</sup> simple aqueous standards can be used to establish the calibration  
28 curves, because the atomization process experienced by the slurry particles is the same as that  
29 experienced by aerosols. Wang *et al.* showed that nanosized TiO<sub>2</sub>,<sup>24</sup> AlN,<sup>18</sup> TiN,<sup>25</sup> and SiC<sup>45</sup>  
30 ceramics can meet the above requirements, so aqueous standards can be used for calibration.  
31 Totland *et al.*<sup>57</sup> used aqueous calibration standards for the analysis of geological samples by  
32 ICP-MS.  
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### 40 **6.3 Use of standard additions**

41 In the slurry technique, simple aqueous standards can be used for the standard addition method  
42 of calibration. However, this technique assumes identical transport characteristics between  
43 solutions and slurries. It was employed by Lobinski<sup>42</sup> for the ICP-OES analysis of ZrO<sub>2</sub>.  
44 Similarly, Zaray<sup>39</sup> analysed silicon nitride powder by using the standard addition method along  
45 with a multi-element aqueous stock solution. Marjanovic *et al.*<sup>58</sup> proposed a simplified  
46 generalized standard addition method (GSAM) for the analysis of cement, gypsum, and basic  
47 slag using ICP-OES with slurry nebulization. The GSAM is an extension to the conventional  
48 standard addition method in which both the sample mass and the added amount of standard  
49 solution are varied. Furthermore, Santos *et al.*<sup>21</sup> direct analysis of clay and refractory materials  
50 slurries by ICP-OES using the GSAM.  
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#### 6.4 Use of an internal standard

The analytical trueness and precision have also been markedly improved by employing an internal standard such as yttrium and scandium in ICP-OES. In ICP-MS, aluminium has been used with success.<sup>59</sup> Ebdon *et al.*<sup>49</sup> reported the use of two internal standards in ICP-MS analyses, rhodium for a semiquantitative analysis and indium for a fully quantitative study. The internal standard can be used to compensate for reduced transport efficiency and inefficient atomization, particularly in the analysis of refractory elements in geological samples and ceramics. However, it is important that any internal standard used in aqueous solution is compatible with the selected dispersant.

The intrinsic internal standard (IIS) method is efficient for correcting for differences in transport and evaporation efficiencies between the slurry and the aerosol.<sup>16</sup> In particular, for the analysis of ceramic samples with large particles (micron or submicron sizes) or high boiling points, the IIS method must normally be used to obtain a correction factor for calibration. The matrix element, for instance, the titanium in TiN, can be selected as the IIS.<sup>25</sup> The correction factor is calculated by measuring the spectral signals of the IIS element and calculating the element concentrations in both the aerosol and slurry.

#### 6.5 Use of empirical correction factors

Alternative calibration methods have also been used for the analysis of slurries to improve the analytical accuracy. For example, empirical correction factors have been employed<sup>34</sup> when the intensity of an elemental line for a slurry analysis is lower than that obtained for the equivalent solution. Conventional internal standardization will not suffice in this case, because of the differences in transport and dissociation behaviour between the analytes in the slurry particles and in the internal standard. Instead, a series of slurries can be analysed using both dissolution and slurry nebulization, and thus correction values can be computed empirically and later applied to unknown slurries. However, such a method relies on similar behaviour among all the slurries investigated.

### 7. Fundamental studies

At present, several fundamental studies on the transport and evaporation behaviour of slurry particles made of refractory materials in plasma have been reported. Wang *et al.*<sup>60,61</sup> investigated the changes in excitation temperature and electron density of the plasma when TiO<sub>2</sub> 5 wt% slurry was nebulized in an ICP torch. No obvious changes in the excitation temperature or electron density were observed in the torch tunnel compared to those obtained for aqueous nebulization

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3 for introduction into the plasma.  
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### 5 6 **7.1 Plasma parameters**

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8 Slurry vaporization depends on the fundamental properties of plasmas, so a complete  
9 understanding of the ICP is desirable. Several studies on the fundamental ICP parameters for  
10 solution nebulization have been reported.<sup>62-64</sup> However, to the best of our knowledge, there are  
11 few studies regarding the plasma parameters for slurry introduction. Both the excitation  
12 temperature and electron density in the plasma channel have been reported for TiO<sub>2</sub> 5 wt% slurry  
13 nebulization.<sup>60,61</sup> The electron density of the plasma for titanium slurry nebulization was  
14 determined using the Stark broadening method for the H<sub>β</sub> line (486.11 nm) and was found to be  
15 the same as for solution nebulization, near 10<sup>15</sup> cm<sup>-3</sup>. Similarly, the excitation temperatures were  
16 almost the same for both slurry and solution nebulization (~5000–6000 K). Thus, it can be  
17 concluded that no significant differences in the plasma parameters can be identified between the  
18 slurry and aqueous solution nebulization.  
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### 26 27 **7.2 Transport behaviour of slurry particles**

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29 The sizes of the slurry particles transported to the plasma have been examined by some research  
30 groups. It was found that only the particles smaller than 10 μm could contribute to the analyte  
31 emission signal.<sup>16</sup> Very small particles preferentially migrate to the ICP torch through selective  
32 transport due to the order-sorting of the spray chamber and gravity effects. In order to measure  
33 the upper size limit of the particles transported to the torch and to distinguish the effects related  
34 to the transport behaviour of a particular material from those caused by evaporation, the particle  
35 size distribution of the original slurry and that after pneumatic nebulization have to be  
36 determined. The experiments conducted by Wang *et al.*<sup>65</sup> revealed that the particle size  
37 distribution of the powder collected at the torch outlet was significantly different from that of the  
38 original powder. Their results were in good agreement with those reported by Ebdon.<sup>67</sup> Currently,  
39 it is only clear that particles with sizes smaller than 10 μm can reach the plasma.  
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### 48 49 **7.3 Evaporation behaviour of slurry particles**

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51 Complete evaporation is difficult to achieve in the plasma for most advanced ceramic materials  
52 because of their extremely stable thermal properties, and thus the measured results can be lower  
53 than the real values. Therefore, an understanding of the evaporation behaviour of slurry particles  
54 in the plasma is highly desirable. Merten *et al.*<sup>67</sup> proposed and developed a model for estimating  
55 the evaporation behaviour. Their model predictions were verified by experimental results,  
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3 particularly for Al<sub>2</sub>O<sub>3</sub>. However, for SiC, the model predictions deviated from the experimental  
4 values, indicating that the evaporation process is still not completely understood. Lim *et al.*<sup>68</sup>  
5 investigated the vaporization process of SiO<sub>2</sub> particles in an ICP both theoretically and  
6 experimentally. The Si I line emission intensities were measured as a function of observation  
7 height for three SiO<sub>2</sub> slurry samples with different particle sizes. Under the assumption that the  
8 height was related to the vaporization time, the heat-transfer and mass-transfer evaporation  
9 models proposed by Hieftje and co-workers<sup>69</sup> were adopted to understand the vaporization  
10 mechanism. The simulation results confirmed that SiO<sub>2</sub> particles with sizes in the range of  
11 0.3–2.6 nm were completely vaporized in atmospheric-pressure ICP through a  
12 heat-transfer-controlled mechanism rather than a Knudsen-effect-corrected  
13 heat-transfer-controlled or mass-transfer-controlled mechanism. The upper size limit for the  
14 particles that can be evaporated completely in the plasma was also determined. Wang *et al.*<sup>23</sup>  
15 studied the relationship between the percentage of Al<sub>2</sub>O<sub>3</sub> slurry powders vaporized and the  
16 particle size. The maximum Al<sub>2</sub>O<sub>3</sub> particle size for complete vaporization in the plasma is about  
17 7 μm. The vaporization efficiency decreased with increasing particle size, and only about 46% of  
18 the particle mass was vaporized for a particle size of 10 μm.  
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## 30 31 **8. Applications**

32 Some recent applications of advanced ceramics slurry nebulization introduction into plasma for  
33 trace element determination are listed in Table 3. These applications are listed alphabetically for  
34 sample type, and the elements determined, the analytical technique used, and calibration  
35 techniques are given. A wide range of applications to various advanced ceramic materials is  
36 clearly evident, showing that slurry nebulization is applicable widely for the analysis of various  
37 ceramics.  
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## 43 44 **9. Conclusions**

45 Many reports have shown that up to now, slurry nebulization into the plasma for ICP  
46 spectrometry is the most mature method among solid sample introduction techniques, because  
47 only a simple modification of the nebulizer is needed.  
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50 Ceramic samples should be ground into micron-sized or even submicron-sized particles and  
51 prepared as a slurry with an appropriate amount of a stabilization agent. In theory, addition of  
52 the stabilization agent increases the zeta potential between the particles in the chosen medium to  
53 improve the slurry stability. A zeta potential of zero means that the slurry particles will  
54 agglomerate, and thus the slurry is unstable.  
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Furthermore, a suitable calibration method should be carefully selected. Establishing a series of standard materials with the same composition and particle size distribution would be ideal, but it is almost impossible. The use of aqueous solution standards seems attractive, but it is suitable only when the slurry particles completely evaporate in the plasma. An intrinsic internal standard method can be used to derive a correction coefficient only when the matrix and all analytes evaporate with nearly the same efficiency.

Because of this lack of a standard procedure, new methods will become necessary as new advanced ceramics are developed. In particular, the evaporation behaviours of different ceramics with different particle sizes in the plasma should be studied to successfully apply slurry nebulization introduction.

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Table 1 Grinding methods used in the preparation of slurries for introduction into plasma

Grinding method	Sample	Grinding time	Resulting particle size ( $\mu\text{m}$ )	Ref.
Bottle and bead	Silicon carbide	24 h	~7.9	19
	Magnesium niobate	2 h	0.6	20
	Clay and refractory	30 min	<37	21
Vibration mill	Boron carbide	3 min	1.9	22
	Silicon carbide	2 min	~1.5	19
	Aluminium oxide	3 min	0.5	23
Puck-type grinder	Titanium dioxide	2 min	<15	24
	Titanium nitride	2 min	~4.9	25
Jet milling	Boron carbide	/	0.5	22
Ultrasonic grinding method	Zirconium oxide-	10 min	0.5–0.7	26
	Aluminium oxide			
Grinding mill	Silicon carbide	10 min	< 38	27

Table 2 Dispersants used to stabilize slurries for nebulization into plasmas

Dispersant	Concentration	Sample	Ref.
Ammonia	0.35% v/v	Kaolin clay	33
Darvan-7	0.5% m/m	Ceramics	15
Darvan-C	0.5% m/m	Ceramics	15
Dolapix PC-33	0.5% m/m	Ceramics	15
Glycerine+0.5 M HCl	40%	Zirconium oxide	34
Glycerol+Kodak photoflow	40% v/v	Ceramic, geological and refractory materials	35
	2% v/v		
Sodium hexametaphosphate	0.1% m/v	Refractory samples, sulphide ore	35
Sodium hexametaphosphate+	0.1% m/v	Titanium dioxide	36
monoisopropanolamine			
Tetrasodium pyrophosphate	0.1% m/v	Silicon nitride	37

	Triton X-100	0.01% v/v	Sediments	38
			Silicon carbide	27
	HCl	1% m/m	Silicon nitride	39
	NH <sub>4</sub> PAA	0.5% m/m	Aluminium oxide	23
	PEI	0.5% m/m	Silicon carbide	19
	NH <sub>4</sub> PAA	2% m/m	titanium nitride	25
	PEI	0.8% m/m for $\mu\text{m}$ size	Aluminium nitride	18
		2% m/m for nm size		
	NH <sub>4</sub> PAA	0.5% m/m for $\mu\text{m}$ size	Titanium dioxide	24
		1.5% m/m for nm size		
	None		Silicon nitride	40
			Aluminium oxide	41
			Silicon carbide	41
			Zirconium oxide	42
			Boron carbide	22

Table 3 Recent applications of slurry sampling ICP-OES for trace element determination

Sample	Analyte	Detection technique	Calibration	Ref
Aluminium nitride	Cr, Cd, Cu, Fe, Mg, Mn, Ni, P, Si, Ti, Y, Zr	ICP-OES	Aqueous standards, calibration curve method	18
Aluminium oxide	Cr, Cu, Fe, V	ETV-ICP-OES	Aqueous standards, standard addition method	70
Aluminium oxide	Cu, Fe, Ga	ETV- ICP-MS	Aqueous standards, standard addition method	71
Aluminium oxide	Ca, Fe, Ga, Mg, Mn, Na, Ni, V	ETV-ICP-OES	Aqueous standards, standard addition method	72
Aluminium oxide	Cr, Cu, Ga, Fe, Mg, Mn, Na, V, Zn	ETV-ICP-MS	Aqueous standards, standard addition method	73

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3				Aqueous standards, standard	
4	Aluminium oxide	Ca, Fe, Ga, Mg, Mn, Na, Ni, V	ETV-ICP-MS	addition method	74
5					
6	Boron carbide	Al, Ca, Cr, Cu, Fe, Mg, Mn, Na, Ni, Si, Ti, V, Zr	ICP-OES	Aqueous standards,	
7				calibration curve method	22
8					
9	Boron carbide	Al, B, Ca, Fe, Ti	ETV-ICP-OES/MS	Aqueous standards,	
10				calibration curve method	75
11	Boron nitride	Al, Ca, Cr, Cu, Fe, Mg, Mn, Si, Ti, Zr	ETV-ICP-OES	Aqueous standards, calibration	
12				curve method	76
13	Ceramic powder	Al, B, Na, Mg, Ca, Ti, V, Cr, Mn, Fe, Co,	ICP-OES/MS	Aqueous standards, calibration	
14		Ni, Cu, Zn, Ga, Zr, Ba, La, Ce		curve method	41
15					
16	Clay and refractory materials	Al, Ca, Fe, K, Mg, P, Si, Ti	ICP-OES	Aqueous standards,	
17				calibration curve method	77
18					
19	Clay and refractory materials	Al, Ca, Fe, K, Mg, Na, P, Si, Ti,	ICP-OES	Aqueous standards,	
20				standard addition method	21
21					
22					
23	Magnesium niobate	Ba, Ca, Cr, Cu, Fe, Mn, Ni, Pb	ICP-OES	Aqueous standards, calibration	
24				curve method	20
25					
26	Niobium(V) oxide	Cu, Cr, Mn, Ni, Ta, Ti, W	ETV-ICP-MS	Aqueous standards, standard addition	
27				method	78
28					
29	Silicon carbide	Al, Ca, Cr, Cu, Fe, Mg, Mn, Ni, Ti, V	ICP-OES	Aqueous standards, calibration	
30				curve method	45
31					
32	Silicon carbide	Al, Ca, Cr, Cu, Fe, Mg, Mn, Ni, Ti	ICP-OES	Solid standards, standard	
33				addition method	19
34					
35	Silicon carbide	Al	ETV-ICP-OES	Solid standards, standard	
36				addition method	79
37					
38	Silicon carbide	B, Mo, Ti, Zr	ETV-ICP-OES	Aqueous standards, standard	
39				addition method	80
40					
41	silicon carbide	Al, Cr, Cu, Fe, V	ETV-ICP-OES	Aqueous standards, both standard	81
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1				addition method and calibration	
2				curve method	
3					
4	Silicon carbide, graphite	Al, Ag, As, Bi, Ca, Co, Cr, Cu, Fe, Ga, K, Li, Mg, Na, Ni, Pb	ETV-ICP-OES	Aqueous standards, calibration curve method	82
5					
6	Silicon nitride powders	Cu, Cr, Ti, Al, Y	ETV-ICP-OES	Aqueous standards, calibration curve method	83, 84
7					
8	Silicon nitride ultra-fine powder	Ca, W, Co, Al, Fe, Mg, Na	ICP-OES	Aqueous standards, calibration curve method	85
9					
10	Titanium dioxide	Al, Ca, Cr, Fe, Mg, P, Pb, V, K, Si, Nb	ICP-OES	Aqueous standards, calibration curve method	24, 86
11					
12	Titanium dioxide	Cr, Cu, Fe, V, Y	ETV-ICP-OES	Aqueous standards, standard addition method	87, 88
13					
14	Titanium dioxide	As, Cd, Hg, Pb, Sb, Zn	ETV- ICP-MS	Aqueous standards, both standard addition method and calibration curve method	89
15					
16	Titanium nitride	Ca, Cr, Fe, Mg, Ni, Si, Ti, Zr	ICP-OES	Aqueous standards, calibration curve method	25
17					
18	Titanium nitride	Ca, Cr, Fe, Mg, Ni, Si, Zr, Ti	ICP-OES	Aqueous standards, calibration curve method using intrinsic internal standard	25
19					
20	Zirconium oxide	La, Eu, Y	ETV-ICP-OES	Aqueous standards, calibration curve method	90
21					
22	Zirconium oxide- Aluminium oxide	Zr, Al, Hf, Y, Ca, Fe, Mg, Mn, Ti	ICP-OES	Aqueous standards, calibration curve method	26
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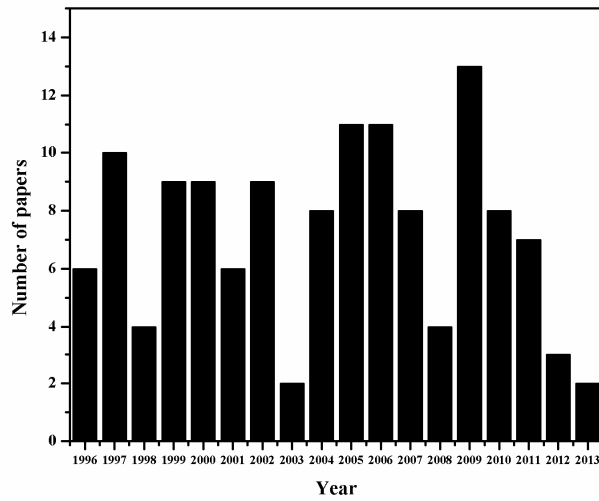
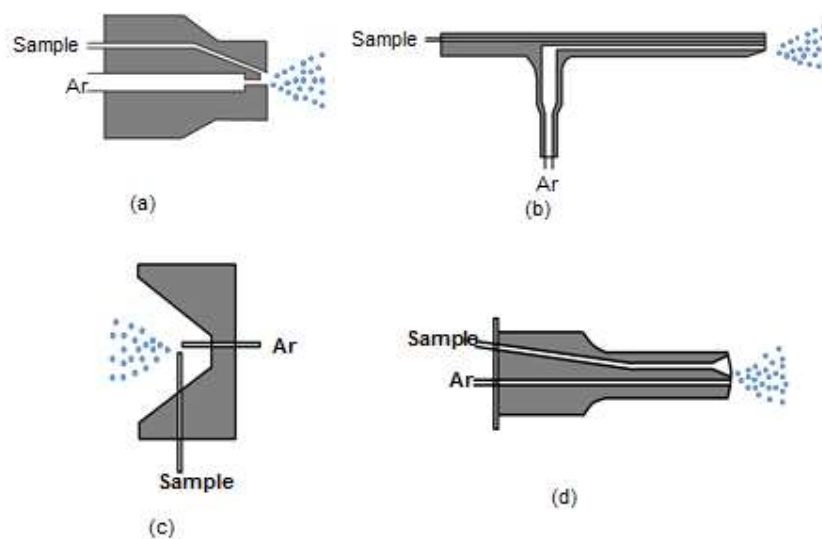


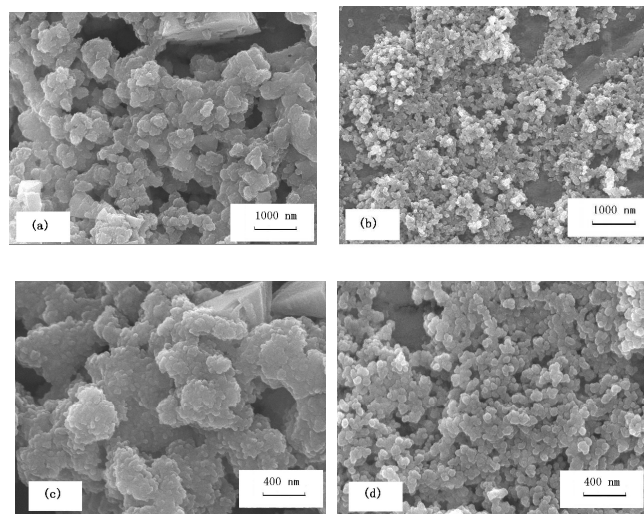
Fig.1 Number of papers published per year from 1996 to 2013 on slurry nebulization into plasma.

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**Fig. 2** Structural schematics of several nebulization devices: (a) V-groove Babington-type, (b) glass expansion concentric type, (c) cross-flow type, and (d) Burgener type nebulizers.





**Fig. 3** SEM micrographs of sediments from a 2 wt.% AlN nanoparticle suspension at pH 6.0 (a) without dispersants at  $\times 20000$ , (b) with dispersants at  $\times 20000$ , (c) without dispersants at  $\times 50000$ , and (d) with dispersants at  $\times 50000$ .  
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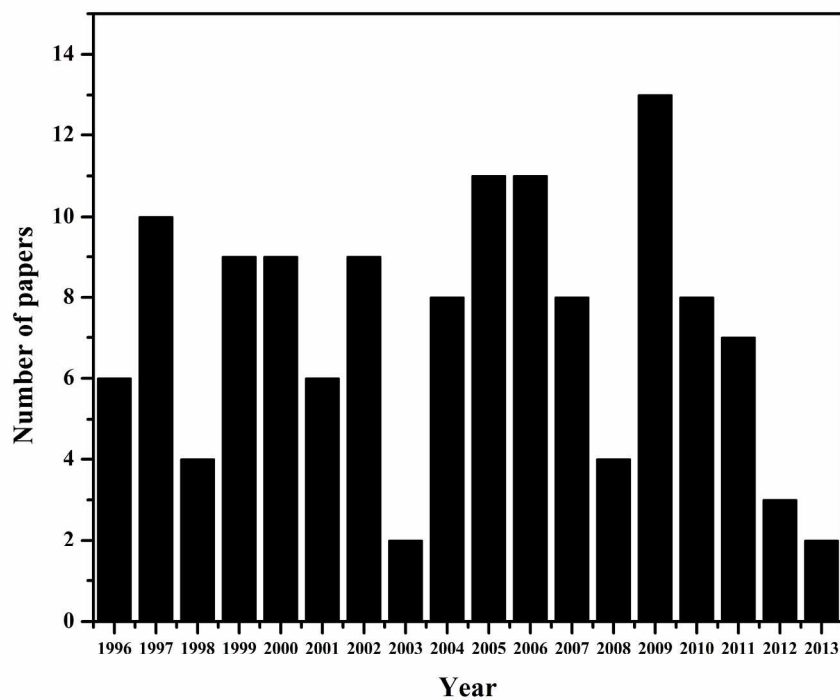


Fig.1 Number of papers published per year from 1996 to 2013 on slurry nebulization into plasma.  
218x187mm (300 x 300 DPI)

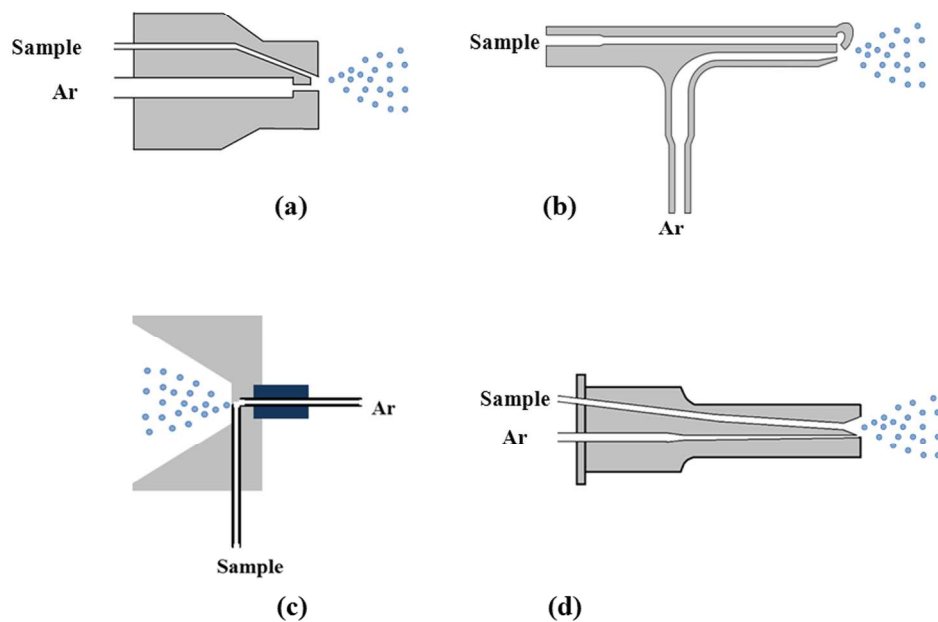


Fig.2 Structural schematics of several nebulization devices: (a) V-groove Babington-type, (b) glass expansion concentric type, (c) cross-flow type, and (d) Burgener type nebulizers.  
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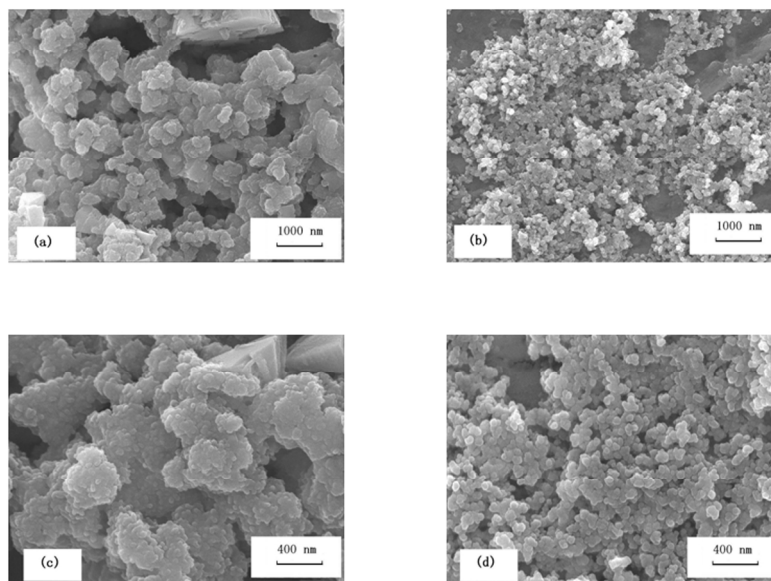


Fig.3 SEM micrographs of sediments from a 2 wt.% AlN nanoparticle suspension at pH 6.0 (a) without dispersants at  $\times 20000$ , (b) with dispersants at  $\times 20000$ , (c) without dispersants at  $\times 50000$ , and (d) with dispersants at  $\times 50000$ . (Reprinted with permission from ref. 18. Copyright 2005, Royal Society of Chemistry)

190x142mm (300 x 300 DPI)