

Polymer Chemistry

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ARTICLE

All-conjugated cationic copolythiophene “rod-rod” block copolyelectrolytes: synthesis, optical properties and solvent-dependent assembly

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Amphiphilic diblock copolythiophenes were synthesised by an efficient two-step synthetic strategy. The diblock copolyelectrolytes were obtained *via* the quasi-living Kumada catalyst-transfer polycondensation followed by quaternisation of the bromohexyl side chains of one of the monomer constituents into N-methylimidazolium, pyridinium, trimethylammonium or trimethylphosphonium units. The effects of the nature of the charged group on the thermal properties were investigated by Rapid Heat-Cool (RHC) calorimetry measurements. The solvent-driven assembly of these block copolyelectrolytes in chloroform (CHCl₃), water, methanol (MeOH), water:MeOH mixtures and in subsequently prepared thin films was investigated using a combination of photoluminescence, scattering and microscopic techniques. The rigid rod-structure of the block copolyelectrolytes lead to the formation of core-shell cylindrical aggregates in solution, with features determined by the nature of the solvent. AFM studies revealed that the aggregates formed in solution can be transferred into thin films allowing for the reliable control of the self-organisation process and the resulting nanoscale architecture.

Introduction

Conjugated polyelectrolytes (CPEs) are defined as polymers with an extended π -conjugated backbone and ionic pendant groups, which combine the organic semiconductor properties and charge-mediated characteristics of polyelectrolytes in a single functional material.¹ As such, CPEs constitute a promising class of materials for application in optoelectronic devices, chemo- and biosensors, and biological imaging.² However, since device performance depends on both the optoelectronic properties and nanoscale morphology of the polymer (which are intrinsically linked), understanding the parameters and physical processes influencing the polymer morphology and its interdependent optoelectronic properties is of crucial importance.³

Recently, self-assembly strategies have emerged as an elegant approach for the design and fabrication of reproducible nanoscale aggregates from conjugated polyelectrolyte building blocks.⁴ For example, co-assembly with oppositely charged species in solution (e.g. ionic surfactants,⁵ metal ions⁶ and complexes/clusters,⁷ DNA⁸) is facilitated *via* electrostatic interactions with the charged side-chains, leading to the

formation of extended networks exhibiting structural and morphological organisation across multiple length scales.

Block copolymers may also provide an alternative interesting route to control the self-assembly into nanostructured morphologies.⁹ However, the rodlike nature of most conjugated polymers complicates the self-assembly of corresponding block copolymers through an interplay of nanophase separation and bulk crystallisation.^{9a} Nonetheless, self-organisation at the nanoscale into lamellar/fibrillar morphology can be obtained in all-conjugated rod-rod block copolymers.¹⁰ The self-assembly of such copolymers into stable nanostructures of different shape or morphology is dictated by both the chain stiffness and the solubility of the individual blocks.¹¹ In this respect, the combination of covalently-bound ionic and neutral blocks can be used to introduce solubility gradients across the copolymer structure and subsequent self-assembly into organised nanostructures can be induced by solvent-mediation.¹²

Scherf *et al.* have in particular reported that all-conjugated cationic “rod-rod” block copolyelectrolytes containing two electronically-different conjugated blocks (namely, polyfluorene and polythiophene blocks) exhibit solvent-

mediated self-assembly in mixtures of selective and non-selective solvents, such as water-methanol (MeOH) and water-THF, which provides a convenient method of modulating the photoluminescence properties *via* segregation of the two blocks.¹² Control of the nanomorphology of the self-assembled aggregates (*e.g.* vesicles, rods, *etc.*) may be obtained through judicious selection of the concentration, solvent or molecular weight of the polymer.¹² Similar changes in nanophase morphology were also observed in all-conjugated cationic diblock copolymers composed of two polyfluorene blocks (a neutral block bound to a cationic block) in protic or non-protic solvents (MeOH *vs.* THF).¹³

Maes *et al.* have recently reported a family of ionic polythiophene-based homopolymers and random copolymers functionalised with imidazolium-terminated side chains at different comonomer ratios.¹⁴ However, the analogous all-conjugated cationic diblock polythiophene copolymers have been scarcely studied. This is somewhat surprising on two counts. Firstly, oligo- and polythiophenes are well-known to undergo temperature- and solvent-induced conformational changes which result in thermo- or solvato-chromatic responses in their optical properties (absorbance, fluorescence).¹⁵ Secondly, poly(3-hexylthiophene) (P3HT) is by far the most commonly used electron donor material in organic photovoltaics (OPV)¹⁶ and conjugated polyelectrolytes have also found application as interfacial layers in bulk heterojunction OPV devices.¹⁷ Significant efforts have been dedicated to the synthesis of sophisticated polythiophene derivatives, with the aim of improved morphological control and enhanced thermal stability.¹⁸ Amphiphilic diblock polythiophene copolymers are thus extremely attractive in this respect as they should afford concomitant tailoring of both the nanoscale morphology and the optical properties *via* solvent-induced self-assembly.

Control of the molecular weight, and in particular, the requirement for a narrow molecular weight distribution, are also important parameters to be considered when designing new diblock copolymers for the purpose of self-assembly.^{9a,f} Moreover, the molecular weight of ionic polythiophenes has been identified as a crucial parameter influencing the performance of OPV devices.^{18a} In this respect, the Kumada Catalyst Transfer Polycondensation (KTCP) method, which enables *chain-growth* polymerisation and leads to the synthesis of conjugated polymers with relatively high molecular weight, narrow polydispersity and minimised termination side reactions, is a tool of choice to obtain ordered nanostructures.¹⁹ This method has been successfully applied to the synthesis of well-defined polyfluorenes,²⁰ polypyrroles²¹ and polycarbazoles.²² Moreover, the “living” chain growth mechanism of the KTCP reaction might afford straightforward access to regioregular block copolymers through a one-pot procedure.^{19c,23}

Herein, we report on the synthesis of amphiphilic “rod-rod” block copolyelectrolytes composed of two polythiophenes blocks (a neutral block bound to a cationic block) bearing either N-methylimidazolium, pyridinium, trimethylammonium or

trimethylphosphonium groups prepared by using the KTCP method. The effects of the nature of the charged group on the optical and thermal properties are investigated. The aggregation properties of the CPEs in selective and non-selective solvent mixtures (H₂O-MeOH) are evaluated using a combination of dynamic light scattering (DLS), small-angle neutron scattering (SANS) and atomic force microscopy (AFM), with the goal of identifying key structure-property relationships in this novel class of materials.

Experimental

Materials and characterisation methods

All reactions were carried out under argon or nitrogen using standard high-vacuum and Schlenk techniques. 2-bromo-3-hexyl-5-iodo-thiophene and 2,5-dibromo-3-bromohexylthiophene were prepared according to literature methods.^{10a,24} Dry THF was obtained by distilling over CaH₂, then Na/benzophenone. Dry toluene was obtained by distilling over Na. Chemicals were obtained from Alfa-Aesar, Sigma-Aldrich and Acros and used as received. All NMR spectra were acquired with a Bruker Avance 300 (¹H 300.13 MHz, ¹³C {¹H} 75.48 MHz and ³¹P {¹H} 121.49 MHz) using the solvent as the chemical shift standard, except for ³¹P {¹H} NMR, where the chemical shifts are relative to 85% H₃PO₄ in D₂O. All chemical shifts and coupling constants are reported in ppm and Hz, respectively. Elemental analyses were performed with a Thermofinigan Flash EA 1112 instrument and expressed in terms of the weight percent ratio of nitrogen to sulphur atoms. Number-averaged (M_n), weight-averaged (M_w) molecular weights and the molecular weight distribution (PDI) of **P3HT-*b*-P3HTBr** were measured using size exclusion chromatography (SEC) on a Polymer Laboratories liquid chromatograph equipped with a PL-DG802 degasser, an isocratic HPLC pump LC 1120 (flow rate = 1 mL min⁻¹), a Marathon autosampler (loop volume = 200 μL, solution conc. = 1 mg mL⁻¹), a PL-DRI refractive index detector and three columns: a PL gel 10 μm guard column and two PL gel Mixed-B 10 μm columns (linear columns for separation of MWPS ranging from 500 to 10⁶ daltons). The eluent used was THF at a flow rate of 1 mL min⁻¹ at 40°C. Polystyrene standards were used to calibrate the SEC. Matrix-assisted laser desorption/ionisation time-of-flight (MALDI-ToF) mass spectra were recorded using a Water QToF Premier mass equipped with a nitrogen laser, operating at 337 nm with a maximum output of 500 J m⁻² delivered to the sample in 4 ns pulses at 20 Hz repeating rate. Mass analyses were performed in reflection mode at a resolution of about 10,000. All samples were analyzed using (DCTB) trans-2-[3-(4-tertbutylphenyl)-2-methylprop-2-enylidene] malononitrile. This matrix was prepared as a 40 mg mL⁻¹ solution in CHCl₃.²⁵ The matrix solution (1 μL) was applied to a stainless steel target and air dried. Polymer samples were dissolved in CHCl₃ to obtain 1 mg mL⁻¹ solutions. 1 μL aliquots of these solutions were applied onto the target area already bearing the matrix crystals, and air

dried. For the recording of the single-stage MS spectra, the quadrupole (rf-only mode) was set to pass ions from m/z 500 to m/z 15,000, and all ions were transmitted into the pusher region of the time-of-flight analyser where they were mass analyzed with 1 s integration time. Rapid Heat-Cool Calorimetry (RHC) experiments were performed on a prototype RHC from TA Instruments, equipped with liquid nitrogen cooling and specifically designed for operation at high scanning rates.^{26,27} RHC measurements were performed at 500 K min⁻¹ in aluminium non-hermetic crucibles, using neon (6 mL min⁻¹) as a purge gas. For comparison with block copolymers, pure CPE homopolymers were synthesized from a P3HTBr precursor ($M_n = 14900$, $M_w = 20700$, PDI = 1.39) following a recently reported procedure.⁸ UV/Vis absorption and fluorescence spectra were recorded at room temperature on a Shimadzu UV2401 PC UV/Vis scanning spectrometer and a Fluorolog-3 (Horiba Jobin Yvon) spectrophotometer, respectively. Emission and excitation spectra were corrected for the wavelength response of the system using correction factors supplied by the manufacturer. Fluorescence quantum yields (Φ_F) were measured using tetraphenylporphyrin (TPP) in toluene as a standard ($\Phi_F = 0.11$).²⁸ Dynamic light scattering measurements were performed using a Zetasizer Nano series nano-ZS (Malvern Instruments, U.K.) in the back-scattering configuration (175°). Cumulant analysis of the correlation function was used to determine the mean hydrodynamic diameter (Z-average) and the polydispersity of each sample.²⁹ AFM measurements were performed using an Asylum Research MFP-3D™ instrument mounted on an anti-vibration plinth, in the tapping mode at room temperature under ambient conditions. The silicon cantilevers used were 125 μm in length and had a resonance frequency of ~ 150 kHz. All raw AFM images were visualised and analysed using the Gwyddion 2.31 software. SANS was carried out on the Sans2d small-angle diffractometer at the ISIS Pulsed Neutron Source (STFC Rutherford Appleton Laboratory, Didcot, U.K.).³⁰ A simultaneous q -range of ~ 0.004 – 0.6 \AA^{-1} was achieved utilising an incident wavelength range of 1.75–16.5 \AA separated by time-of-flight and employing an instrument set up of $L_1 = L_2 = 4$ m, with the 1 m² detector offset vertically 150 mm and sideways 150 mm. $q = (4\pi/\lambda)\sin(\theta/2)$ where λ is the wavelength and θ the scattering angle. Samples were prepared in deuterated solvents, to provide good neutron scattering contrast. Each raw scattering data set was corrected for the detector efficiencies, sample transmission and background scattering and converted to scattering cross-section data ($\partial\Sigma/\partial\Omega$ vs. q) using the instrument-specific software.³¹ These data were placed on an absolute scale (cm⁻¹) using the scattering from a standard sample (a solid blend of hydrogenated and perdeuterated polystyrene) in accordance with established procedures.³² The scattering functions were fit using non-linear least-squares analysis to a Core-Shell-Cylinder model using the SasView programme.³³ Full details of the models and the fitting procedure can be found in the Supporting Information.

Synthesis

Poly[(3-hexylthiophene)-2,5-diyl]-block-poly[3-(6-bromo-hexyl)-thiophene-2,5-diyl] block copolymer (P3HT-*b*-P3HTBr). Two round-bottomed flasks (100 mL) were dried by heating under reduced pressure and cooled to room temperature. 2-Bromo-3-hexyl-5-iodothiophene (1.22 g, 3.3 mmol) was dried by three successive azeotropic distillations with toluene, and then dry THF (15 mL) was added. A 2 mol.L⁻¹ solution of *i*-PrMgCl in THF (1.65 mL) was added *via* a syringe, and the mixture was stirred at 0°C for 30 min (solution A). Solution A was added in one portion to Ni(dppp)Cl₂ catalyst (59 mg) in THF (10 mL). In the other flask, 2.7 mmol of 2,5-dibromo-3-bromohexylthiophene (1.09 g), previously dried by azeotropic distillation, was reacted with *i*-PrMgCl (1.35 mL) in the same manner (solution B). After stirring the reaction mixture for 3 h, solution B was added to the previous mixture *via* a syringe, and the resulting solution was stirred overnight. The reaction was quenched quickly by pouring HCl aq (5 M) into the solution and stirring for 0.5 h. Then, the mixture was precipitated in cold MeOH and filtered. The product was washed well with MeOH and hexane to afford a red solid. **P3HT-*b*-P3HTBr.** Yield: 0.99 g (82%). ¹H NMR (CDCl₃): $\delta = 0.90$ (t, 3H, CH₃, ³J_{H-H} = 6.8 Hz), 1.62 (m, 16H, CH₂), 2.80 (t, 4H, CH₂-Th, ³J_{H-H} = 7.9 Hz), 3.42 (t, 2H, CH₂-Br, ³J_{H-H} = 6.7 Hz), 6.98 (s, 2H, Th) ppm. ¹³C{¹H} NMR (CDCl₃): 14.3, 22.8, 28.1, 28.8, 29.4, 29.6, 30.5, 30.7, 31.8, 32.9, 34.0, 128.8, 130.7, 130.8, 139.8, 140.0. UV-Vis (CHCl₃): $\lambda_{\text{max}} = 450$ nm; SEC (THF, PS standards): $M_n = 11000$ g mol⁻¹, $M_w = 12300$ g mol⁻¹, PDI = 1.12.

General procedure for the synthesis of P3HT-*b*-P3HTIm and P3HT-*b*-P3HTPy block copolyelectrolytes. Block copolymer P3HT-*b*-P3HTBr (150 mg) was allowed to react with 1-methylimidazole (10 mL) or pyridine (10 mL) in refluxing CHCl₃ (10 mL) for 2 d. After cooling to room temperature, the mixture was poured into Et₂O to precipitate the polyelectrolytes P3HT-*b*-P3HTIm and P3HT-*b*-P3HTPy. The crude polymers obtained were repeatedly washed with diethyl ether to remove excess of 1-methylimidazole or pyridine, and then dried under vacuum at 40°C. The solid polymers P3HT-*b*-P3HTIm and P3HT-*b*-P3HTPy were further purified in refluxing diethyl ether using a Soxhlet apparatus overnight and finally, dried under vacuum at 40°C.

P3HT-*b*-P3HTIm. Yield: 80%, ¹H NMR (CDCl₃): $\delta = 0.89$ (br.), 1.19–1.47 (br.), 1.69 (br.), 1.90 (br.), 2.78 (br.), 3.19 (br.), 4.05 (br.), 4.34 (br.), 6.97 (br.), 7.74 (br.), 9.92 (br.) ppm. ¹³C{¹H} NMR (CDCl₃): $\delta = 14.2, 22.8, 26.3, 29.1, 29.4, 29.6, 30.5, 30.6, 31.8, 36.9, 50.1, 122.8, 124.1, 128.7, 130.6, 133.8, 136.7, 140.0$ ppm. UV-Vis (CHCl₃): $\lambda_{\text{max}} = 452$ nm; Anal. calcd.: N/S 0.86 Found 0.86.

P3HT-*b*-P3HTPy. Yield: 81%, ¹H NMR (CDCl₃): $\delta = 0.91$ (br.), 1.27–1.50 (br.), 1.70 (br.), 2.04 (br.), 2.80 (br.), 4.00 (br.), 6.96 (br.), 8.16 (br.), 8.60 (br.), 9.50 (br.) ppm, ¹³C{¹H} NMR (CDCl₃): $\delta = 15.4, 23.8, 27.1, 30.2, 30.5, 30.7, 31.7, 32.8, 33.2, 62.9, 129.8, 131.7, 134.9, 141.1, 146.3, 147.0$ ppm. UV-Vis (CHCl₃): $\lambda_{\text{max}} = 455$ nm; Anal. calcd.: N/S 0.43 Found 0.41.

General procedure for the synthesis of P3HT-*b*-P3HTNMe₃ and P3HT-*b*-P3HTPMe₃ block copolyelectrolytes. Block copolymer **P3HT-*b*-P3HTBr** (150 mg) was introduced into a 100 mL two-necked flask equipped for stirring and dissolved in CHCl₃ (15 mL). 12 mL of a solution of trimethylamine or trimethylphosphine (1.0 M in THF) were added and the mixture was stirred at 40°C for 48 h. After cooling to room temperature, the solvent mixture was evaporated and the residue was dissolved in the minimum amount of CHCl₃ and poured into Et₂O to precipitate the polyelectrolytes **P3HT-*b*-P3HTNMe₃** or **P3HT-*b*-P3HTPMe₃**. The crude products obtained were repeatedly washed with diethyl ether and dried under vacuum at 40°C. **P3HT-*b*-P3HTNMe₃** and **P3HT-*b*-P3HTPMe₃** were further purified with refluxing diethyl ether using a Soxhlet apparatus overnight and finally, dried under vacuum at 40°C.

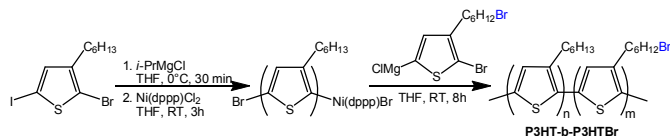
P3HT-*b*-P3HTNMe₃. Yield: 79%, ¹H NMR (CDCl₃): δ = 0.90 (br.), 1.15-1.48 (m, br.), 1.71 (m, br.), 2.80 (br.), 3.35 (br.), 3.64 (br.) 6.95 (br.) ppm, ¹³C{¹H} NMR (CDCl₃): 14.2, 22.8, 29.4, 29.6, 30.6, 31.8, 66.5, 128.7, 130.6, 133.8, 134.2, 140.0 ppm. UV-Vis (CHCl₃): λ_{max} = 452 nm; Anal. calcd.: N/S 0.43 Found 0.40.

P3HT-*b*-P3HTPMe₃. Yield: 80%, ¹H NMR (CDCl₃): δ = 0.90 (br.), 1.24-1.48 (br.), 1.70 (br.), 1.91 (br.), 2.12 (br.), 2.56 (br.), 2.80 (br.), 6.97 (br.) ppm. ¹³C{¹H} NMR (CDCl₃): δ = 10.0, 10.3, 15.2, 22.8, 23.6, 23.8, 30.4, 30.2, 30.7, 31.4, 31.7, 32.7, 32.9, 129.7, 131.7, 134.8, 141.0 ppm. ³¹P{¹H} NMR (CDCl₃): δ = 27.0 (s, 1P) ppm. UV-Vis (CHCl₃): λ_{max} = 455 nm.

Results and discussion

Polymer Synthesis

The cationic diblock copolythiophenes were synthesised by a two-step procedure, similar to the analogous random copolymers and homopolymers.^{8,14} In the first step, the preparation of the regioregular head-to-tail bromide-bearing copolythiophene (**P3HT-*b*-P3HTBr**) in a 50/50 block ratio is achieved by using KTCP polymerisation (Scheme 1).^{24,34}



Scheme 1. Synthetic route applied toward the **P3HT-*b*-P3HTBr** precursor block copolymer.

The **P3HT** block was first synthesised by polymerisation of 2-bromo-3-hexyl-5-iodothiophene with the Ni catalyst to obtain the end-living **P3HT** polymer. Subsequently, a new portion of activated 2,5-dibromo-3-bromohexylthiophene monomer was added to the reacting solution to obtain the final diblock copolymer **P3HT-*b*-P3HTBr**. After the polymerisation, the reaction mixture was quenched with 5 M HCl, following the procedure described by Yokozawa *et al.*³⁵ This quenching procedure prevents the polymer chains from undergoing dimerisation, in order to maintain the narrow polydispersity

while efficiently removing inorganic impurities from the product.

Figure 1 shows the SEC profiles of the **P3HT** block synthesised in the first step and the final diblock copolymer **P3HT-*b*-P3HTBr** after the second-stage polymerisation. Compared to the elution profile of the **P3HT** block obtained after the first step, the peak of the final diblock copolythiophene shifts to shorter elution volume (corresponding to a higher molecular weight), whilst maintaining a single peak, indicating the ‘living’ character and the formation of the diblock copolymer. A number-averaged molecular weight (M_n) of 11000 g mol⁻¹ and a polydispersity index (M_w/M_n) < 1.3 were obtained from the monomodal trace of the **P3HT-*b*-P3HTBr** copolymer in the SEC analysis.

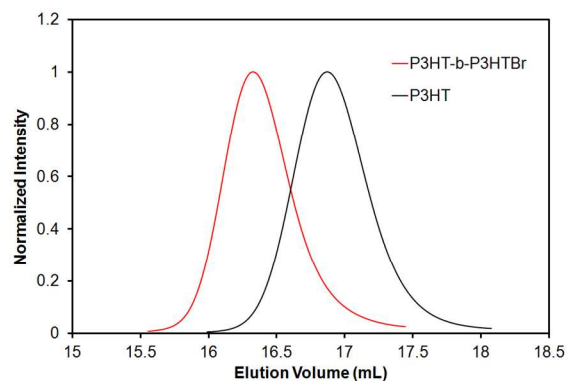
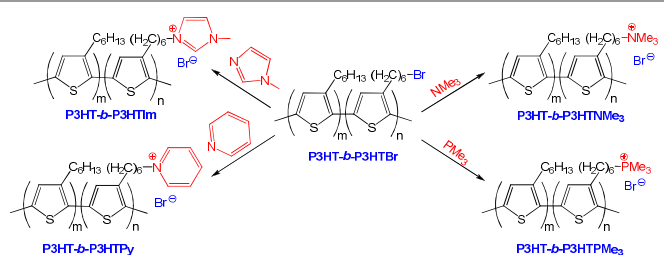


Figure 1. SEC profiles obtained during the synthesis of the **P3HT-*b*-P3HTBr** block copolymer.

MALDI-ToF analysis of **P3HT-*b*-P3HTBr** confirmed the covalent association of 3-bromohexylthiophene and 3-hexylthiophene. Indeed, the presence of m/z 166 and m/z 244 mass differences confirmed the presence of both monomer units inside the polymer chains (see Figure S1 in Supporting Information). However, the exact nature of the end-groups cannot be determined since, for instance, H(3HT)_x-(3HTBr)_y-Br is an isomer of H(3HT)_{x-1}-(3HTBr)_{y+1}-H and Br(3HT)_{x+1}-(3HTBr)_{y-1}-Br. An absolute averaged molecular weight of 7980 g mol⁻¹ was determined by MALDI. Such difference between the molecular weight calculated by SEC and MALDI is not unusual since GPC usually overestimates the molecular weight of rigid-rod poly(3-alkylthiophenes) by a factor of 1.2-2.3.^{20,36} The composition of the diblock copolythiophene was determined from the ¹H NMR spectra (Figure S2). The peaks observed at δ 0.90 and 3.42 ppm have been assigned to the resonance of the terminal methyl groups in the **P3HT** block and the methylene groups adjacent to the bromine atoms in the **P3HTBr** block, respectively. From the integration of these two peaks, the molar ratios of the **P3HT** and **P3HTBr** segments was 58:42, which is very close to the feed molar ratio of 55:45. The final step towards the diblock copolyelectrolytes **P3HT-*b*-CPE** involved the quaternization of the **P3HT-*b*-P3HTBr** precursor with 1-methyl-imidazole, pyridine, trimethylamine or trimethylphosphine (Scheme 2).



Scheme 2. Synthesis of the polyelectrolyte diblock copolymers **P3HT-*b*-CPE** (CPE = P3HTIm, P3HTPy, P3HTNMe₃, P3HTPMe₃).

The efficiency of the conversion was monitored by ¹H NMR spectroscopy by following the shift of the terminal methylene group α of bromine at ~ 3.4 ppm (in CDCl₃) for the precursor copolymer. Based on ¹H NMR spectroscopy, the quaternisation appears to be quantitative. It was also confirmed by the N/S ratio values obtained from elemental analysis for the ionic diblock copolymers containing cationic nitrogen side groups which are very close to the expected values calculated on the basis of the composition of the precursor diblock copolymer.

Thermal properties

In Rapid Heat-Cool (RHC) calorimetry measurements, an important endothermal effect can be observed during the first heating of all four materials, indicating the presence of a significant amount of water or solvent that evaporates upon heating. Such an effect is no longer present during the second heating, where a glass transition temperature (T_g) significantly above room temperature is observed for the four materials. The endothermal effect is appearing again when the copolymers are stored in the non-hermetic RHC crucibles at ambient conditions, proving water vapour absorption and indicating the materials' hydrophilicity.

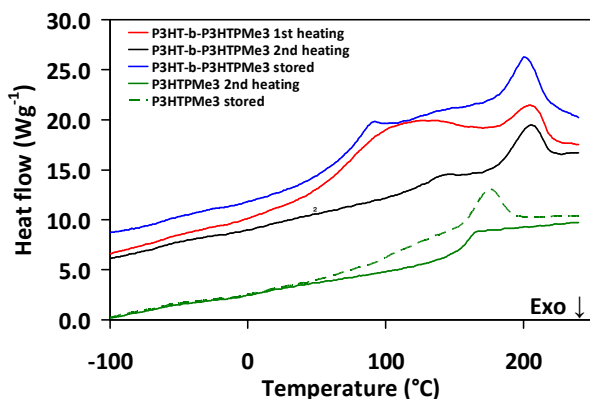


Figure 2. RHC thermogram of **P3HT-*b*-P3HTPMe₃**, showing the first heating (red) and the second heating (black). After storage at ambient conditions a curve similar to that of the first heating is found (blue). Thermograms of the second heating and after storage at ambient conditions are also included for pure **P3HTPMe₃** for comparison (dashed and solid green). The curves were shifted vertically for clarity.

Figure 2 shows this behavior for the **P3HT-*b*-P3HTPMe₃** material. As this effect is also present in the pure **P3HTPMe₃**

polymer, it can be concluded that it is caused by the CPE block and can be linked to the bromide counterion.¹⁴ This is an important finding for potential applications, as water vapour absorption will drastically lower the T_g of this block if the material is not used under dry conditions.³⁷ When comparing the second heating runs of the different **P3HT-*b*-CPE** polymers, a melting peak is observed at a similar temperature (around 205°C) for all materials except **P3HT-*b*-P3HTNMe₃**. These melting transitions can be attributed to the **P3HT** block, which is known to be a fast-crystallising conjugated polymer. A clear T_g is also always present, caused by the **CPE** block, as can be seen by comparing with the pure CPE materials. The T_g and the melting point (T_m) and enthalpy (ΔH_m) of the **P3HT** block are summarised for the four copolymers in Table 1.

Table 1. Glass transition of the CPE block and melting temperature and enthalpy of the **P3HT** block for all **P3HT-*b*-CPE** materials measured by RHC during the second heating run.

Polymer	T_g (°C)	T_m (°C)/ ΔH_m (J.g ⁻¹)
P3HT-<i>b</i>-P3HTIm	57	206/13.1
P3HT-<i>b</i>-P3HTPy	75	204/10.2
P3HT-<i>b</i>-P3HTNMe₃	159*	177/7.3
P3HT-<i>b</i>-P3HTPMe₃	136	205/9.2

* Due to the overlap with the melting peak of the **P3HT** block, the T_g of the **P3HTNMe₃** block could not be determined in the block copolymer; instead, the value of the pure **P3HTNMe₃** polymer is mentioned.

The thermal transitions observed by RHC during the second heating run are a confirmation of the diblock copolymer architecture of the synthesized materials. The nature of the CPE block can modify its T_g by approximately 100°C, ranging from 57°C up to 159°C. The thermogram of **P3HT-*b*-P3HTNMe₃** (Figure 3) clearly shows that the **P3HT** melting transition differs for this material.

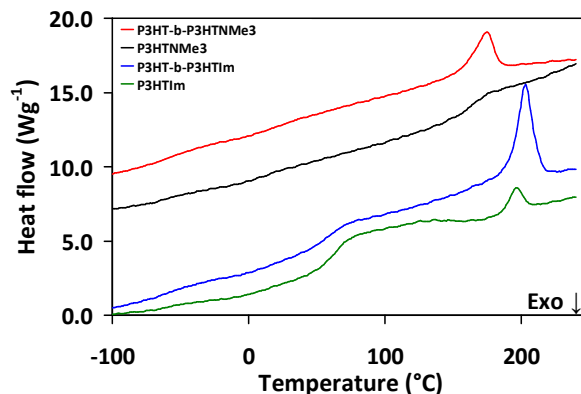


Figure 3. RHC thermogram of **P3HT-*b*-P3HTNMe₃** (red) and **P3HT-*b*-P3HTIm** (blue), showing the second heating. For comparison the second heating of the **P3HTNMe₃** (black) and **P3HTIm** (green) pure polymers is shown. The curves were shifted vertically for clarity.

In this case, the T_g of the **P3HTNMe₃** block overlaps with the **P3HT** melting peak, and this high T_g value around 159°C is

expected to strongly inhibit the crystallization process of the **P3HT** block. Indeed, in the temperature region of **P3HT** crystallization during cooling, the **P3HTNMe₃** block can impose severe mobility restrictions by a concurrent vitrification. This results in a lower melting temperature and enthalpy of the **P3HT** block. On the contrary, the other **P3HT-*b*-CPE** copolymers do not show this behaviour, such as e.g. **P3HT-*b*-P3HTIm** (see Figure 3). Interestingly, the pure **P3HTIm** polymer seems to be somewhat crystalline, showing a small melting peak at about 196°C (in close vicinity of the **P3HT** melting). This explains the slightly higher melting enthalpy found for the **P3HT-*b*-P3HTIm** copolymer, when compared to the other materials. It should be noted that the thermograms of Figure 3 and the values of Table 1 are indicative and demonstrate the trends of the different **P3HT-*b*-CPE** polymers during the second heating for the applied RHC conditions. The position of T_g and T_m and the melting enthalpy ΔH_m strongly depend on all experimental conditions, *i.e.* the synthesis procedure (sequence of block synthesis, precise block length control and resulting PDI of blocks, regioregularity of **P3HT**, nature of end-groups), the purifications and the eventual drying (vacuum at 40°C) and storage. Moreover, the solvent dependent assembly of the block copolyelectrolytes and the persistence of the solvent-induced nanomorphology after removal of the solvent are probably important for the thermal properties and their stability as well (see further in ‘Aggregation behaviour’ as studied by DLS, SANS, AFM). As an example, the thermal properties of **P3HT-*b*-P3HTIm** are still evolving in subsequent cool-heat cycles after the second heating; T_g is raised from 57°C to above 70°C after 10 cycles, while T_m is decreasing from 206°C towards 180°C in combination with a lowered ΔH_m from 13.1 J g⁻¹ to 8.5 J g⁻¹. These findings suggest further changes in the nanomorphology as a result of thermal cycling, leading to a loss of perfection of the **P3HT** crystals, possibly with a concomitant loss of a minor amount of residual solvent from the ionic block. Similar effects are noticed for the other **P3HT-*b*-CPE** polymers.

Solution behaviour

The optical properties of the cationic **P3HT-*b*-CPE** copolymers in solution have been studied by UV/Vis absorption and photoluminescence (PL) spectroscopy. The neutral diblock copolymer **P3HT-*b*-P3HTBr** exhibits a broad absorption band with a maximum (λ_{max}) centred at 450 nm in chloroform (CHCl₃), which, as expected, is similar to the parent **P3HT**. After functionalization with the imidazolium, pyridinium, trimethylammonium or trimethylphosphonium moieties, the **P3HT-*b*-CPE** polymers become also soluble in more polar solvents such as water and MeOH due to the presence of the hydrophilic pendant ionic side groups which overcomes the hydrophobic π - π stacking interactions between the polythiophene backbones. We note that **P3HT-*b*-P3HTPy** is somewhat less soluble in these solvents than its **P3HT-*b*-CPE** counterparts, indicating the somewhat reduced hydrophilic (or increased hydrophobic) character of the **P3HTPy** block, as

previously reported by Scherf *et al.* for the polyfluorene (PF2/6)-polythiophene diblock copolyelectrolyte analogue PF2/6-P3HTPy.^{12a} **P3HT-*b*-CPE** copolymers exhibit solvatochromatic behaviour, appearing dark red/brown in CHCl₃ and dark violet in methanolic and aqueous solution (Fig. 4a and Fig. S13-S15). This solvatochromatism is reflected in the corresponding UV/Vis absorption spectra, as shown in Fig. 4b for **P3HT-*b*-P3HTPMe₃** where λ_{max} red-shifts from ~450 nm in CHCl₃, to ~520 nm in H₂O and MeOH, with the additional emergence of a vibronic structure in the latter solvents.

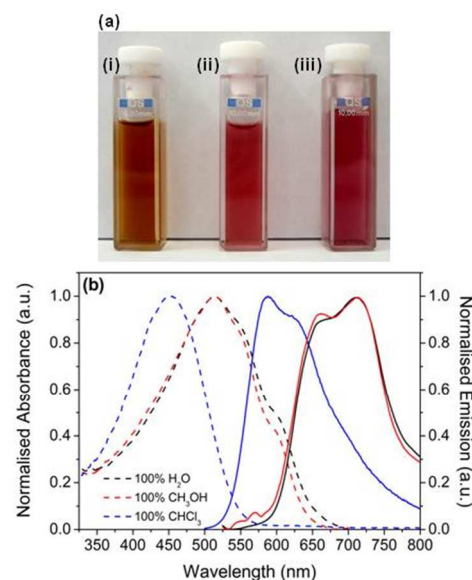


Figure 4. (a) **P3HT-*b*-P3HTPMe₃** in (i) CHCl₃, (ii) MeOH and (iii) H₂O. (b) Corresponding normalised UV/Vis absorption and emission spectra for **P3HT-*b*-P3HTPMe₃** in 100% CHCl₃ (blue), 100% H₂O (black) and 100% MeOH (red).

P3HT has previously been reported to self-assemble into a lamellar-like structure built from interchain π - π stacking interactions between the planarised aromatic backbones of the polymer chains.³⁸ This supramolecular organisation is accompanied by a bathochromic shift in the absorption spectrum and vibronic structure at longer wavelengths, very similar to the behaviour observed here for **P3HT-*b*-CPEs** in MeOH and water. Water is expected to be a non-solvent for the hydrophobic, neutral **P3HT** block, with only low to moderate solubility anticipated in MeOH. We propose that the significant bathochromic shift is indicative of increased aggregation of the hydrophobic **P3HT** block in these poor solvents. The concomitant appearance of vibronic structure further suggests increased confinement and/or planarisation of one or both of the blocks, thereby hindering free rotation of thiophene units along the polymer backbone. The **P3HT-*b*-CPE** copolymers are weakly fluorescent in solution and the emission spectra exhibit similar solvatochromatic trends (Fig. 4b). A significant red-shift in the emission maximum (λ_{em}) from 588 nm to 711 nm is observed for **P3HT-*b*-P3HTPMe₃** upon changing the solvent from CHCl₃ to MeOH or H₂O, which is again accompanied by increased vibronic structure assigned to the

vibronic progression of the C=C stretching mode ($\Delta E \approx 0.15$ eV).³⁹ The optical properties are similar for all copolymers in the **P3HT-*b*-CPE** series and absorption and photoluminescence maxima in each solvent are collected in Table 2. The use of solvent mixtures has been shown to be very effective for both breaking up the aggregates formed by homopolymer CPEs in aqueous solution⁴⁰ and controlling segregation and self-organisation of diblock copolymers.^{12,13}

Polymer	Absorption ($\lambda_{\text{max}}/\text{nm}$)			Emission ($\lambda_{\text{max}}/\text{nm}$)		
	CHCl ₃	MeOH	H ₂ O	CHCl ₃	MeOH	H ₂ O
P3HT-<i>b</i>-P3HTIm	452	519	522	586	729	729
		556		625		
		608				
P3HT-<i>b</i>-P3HTPy	455	517	520	580	721	723
		556		619		
		607				
P3HT-<i>b</i>-P3HTNMe₃	452	518	518	592	721	722
		556		631		
		607				
P3HT-<i>b</i>-P3HTPMe₃	455	516	516	588	707	710
		556		627		
		607				

Table 2. Absorption and photoluminescence maxima of the **P3HT-*b*-CPE** diblock copolymers in various solvents. (PL excitation wavelength: 523 nm)

The UV/Vis absorption and PL properties of **P3HT-*b*-CPE** copolymers were thus further examined in MeOH/H₂O solvent mixtures from 0-100% (v/v), as illustrated in Fig. 5 for **P3HT-*b*-P3HTPy** and in Fig. S16-S18 for other **P3HT-*b*-CPEs**. The gradual increase in the volume of MeOH added to an aqueous solution of **P3HT-*b*-P3HTPy** is accompanied by a concomitant increase in the absorbance, which reaches its maximum value at 50% (v/v) MeOH/H₂O; above this, the absorbance drops off once again (Fig. 5a, inset). A similar trend is observed in the corresponding PL spectrum, with the emission intensity again reaching its upper limit at 50% (v/v) MeOH/H₂O (Fig. 5b). Notably, no shift in the absorption or emission maximum wavelength was observed for any of the solvent mixtures studied.

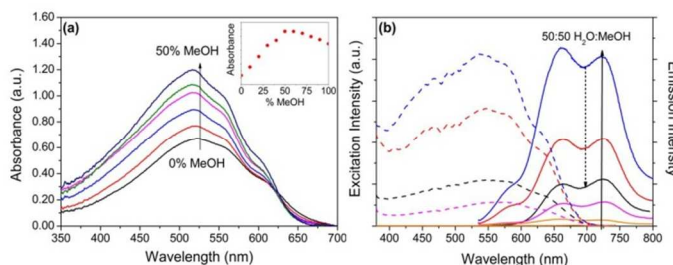


Figure 5. (a) UV/Vis absorption spectra of **P3HT-*b*-P3HTPy** in H₂O/MeOH solvent mixtures. Inset: Absorbance at 522 nm (λ_{max}) as a function of Vol% MeOH in H₂O/MeOH mixtures. (b) Selected excitation ($\lambda_{\text{em}} = 730$ nm) and PL ($\lambda_{\text{ex}} = 515$ nm) spectra of **P3HT-*b*-P3HTPy** in 100% MeOH (orange line), 70% MeOH (violet line), 50% MeOH (blue line), 30% MeOH (red line) and 0% MeOH (black line).

Fluorescence quantum yields (Φ_F) were determined to quantify the extent of the fluorescence enhancement in each solvent mixture (Fig. 6a). For **P3HT-*b*-P3HTPy**, **P3HT-*b*-P3HTNMe₃** and **P3HT-*b*-P3HTPMe₃**, $\Phi_F \sim 0.5\%$ in 100% MeOH and $\Phi_F \sim 1.0-1.5\%$ in 100% H₂O were obtained. As the ratio of solvents in the mixture is varied, Φ_F gradually increases, reaching a maximum of $\Phi_F \sim 5.5\%$ in 50% (v/v) MeOH/H₂O. **P3HT-*b*-P3HTIm** exhibits a slightly different trend; Φ_F remains constant at $\sim 1.0-1.5\%$ across the range 100% MeOH to 20:80% (v/v) MeOH/H₂O, before doubling to $\Phi_F \sim 3\%$ in 100% H₂O. The increase in the fluorescence quantum yield in the mixed solvent system suggests improved solubilisation and at least, partial isolation of polymer chains in one or both the blocks.

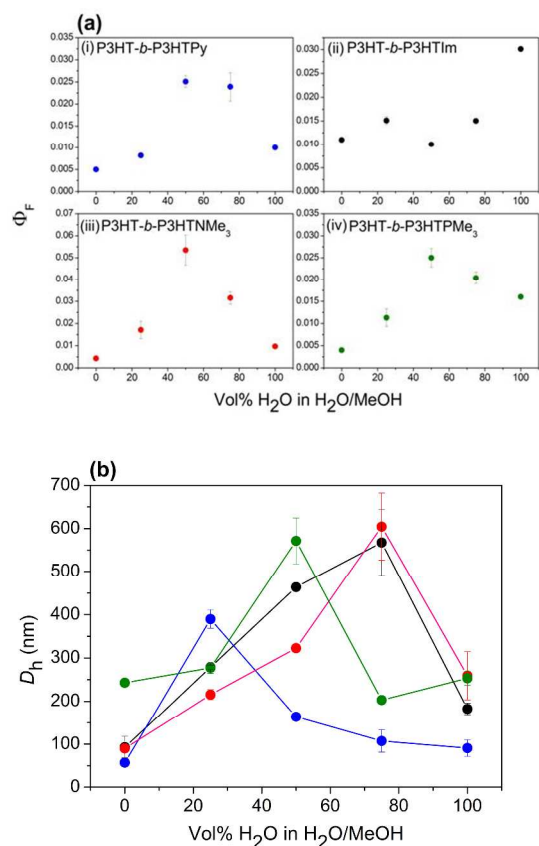


Figure 6. (a) Fluorescence quantum yields (Φ_F) and (b) hydrodynamic diameter (D_h , Z-average) of **P3HT-*b*-CPEs** in H₂O/MeOH solvent mixtures (0.35 $\mu\text{g mL}^{-1}$). Legend: **P3HT-*b*-P3HTPy** (blue circles), **P3HT-*b*-P3HTIm** (black circles), **P3HT-*b*-P3HTNMe₃** (red circles) and **P3HT-*b*-P3HTPMe₃** (green circles).

Aggregation Behaviour

The aggregation behaviour of **P3HT-*b*-CPEs** in solution and thin films has been further investigated by dynamic light scattering (DLS), small-angle neutron scattering (SANS) and atomic force microscopy (AFM). DLS analysis was initially performed to determine the average hydrodynamic diameter (D_h) of **P3HT-*b*-CPE** aggregates in MeOH/H₂O mixtures (Fig. 6b). Samples typically exhibited a monomodal distribution of

low polydispersity in all solvent mixtures investigated. The smallest aggregates were obtained in 100% MeOH (~ 100 nm for **P3HT-*b*-P3HTNMe₃**, **P3HT-*b*-P3HTIm**, **P3HT-*b*-P3HTPy** and ~ 250 nm for **P3HT-*b*-P3HTPMe₃**), with slightly larger aggregates found in 100% H₂O (~ 100 nm for **P3HT-*b*-P3HTPy**, ~ 200 – 260 nm for **P3HT-*b*-P3HTIm** and **P3HT-*b*-P3HTPMe₃**, and ~ 300 nm **P3HT-*b*-P3HTNMe₃**). The largest D_h values (~ 400 – 600 nm) were consistently obtained in ~ 40 – 60% (v/v) MeOH/H₂O, which is consistent with increased solubility of both the neutral **P3HT** block and the charged **CPE** block in the mixed solvent systems, resulting in swelling of the aggregate structure due to associated solvent molecules.

To obtain deeper insight into the particle shape and nanoscale organisation in solution, small-angle neutron scattering (SANS) studies were performed on **P3HT-*b*-P3HTIm** and **P3HT-*b*-P3HTPy** in deuterated solution. Figure 7 plots the SANS data for **P3HT-*b*-P3HTPy** in D₂O, *d*₄-MeOD and their 50:50 (v/v) mixture (corresponding plots for **P3HT-*b*-P3HTIm** are shown in Figure S22 in Supporting Information). All three solvents yield similar scattering profiles with a shoulder at $q = 0.02 \text{ \AA}^{-1}$ and an upturn at $q = 0.008 \text{ \AA}^{-1}$; however they do differ in the gradient of their slopes. At high q ($q > 0.07 \text{ \AA}^{-1}$), the SANS response stems from the internal structure of the aggregate. The excess scatter for some samples at $q \sim 0.1$ – 0.2 \AA^{-1} is likely due to internal structure within the aggregates, perhaps due to repulsive interactions between similarly charged polymer chains.^{5a-c} The upturn in $I(q)$ at $q < 0.008 \text{ \AA}^{-1}$ is indicative of particle aggregation. The power law scaling for the intermediate q region ($0.02 < q < 0.07$) varies from $q^{-4.2}$ in *d*₄-MeOD to $q^{-5.21}$ in D₂O. The low q region ($q < 0.02$) decays as $\sim q^{-5/3}$, which is typical of scattering from either cylindrical aggregates or from individual chains.⁴¹ The observation window of these SANS experiments ranged from 2–160 nm, which is well beyond the isolated chain length (22.7–25.4 nm). Unique fits to the SANS data were not possible; however all those chosen (Table S1 in Supplementary Information) have absolute SANS intensities consistent with the known sample concentrations ($\sim 1\%$ vol. dry material). All the fits show that the aggregates must contain upwards of 30% solvent, without any significant “dry core”. Better fits were obtained with a core plus shell model with a diffuse outer shell containing 85–95% solvent. The exception to the latter were the samples in pure MeOD, where a thin shell (of ~ 2 nm) seems to have slightly less solvent ($\sim 65\%$ vol.) than the core of the aggregate ($\sim 80\%$ vol. solvent). The SANS from the aggregates in MeOD are also more consistent with an elongated, rod-like structure, whilst the others are better fit as discs. The disc cases would also be equally well fitted as core/shell ellipsoids or as rather polydisperse core/shell spheres. There is no clear evidence in the SANS patterns of a sharp core/shell boundary, so a continuous radial variation in solvent composition seems likely, though this would be more difficult to model. Aggregation numbers are likely in the range ~ 100 to ~ 300 copolymer molecules.

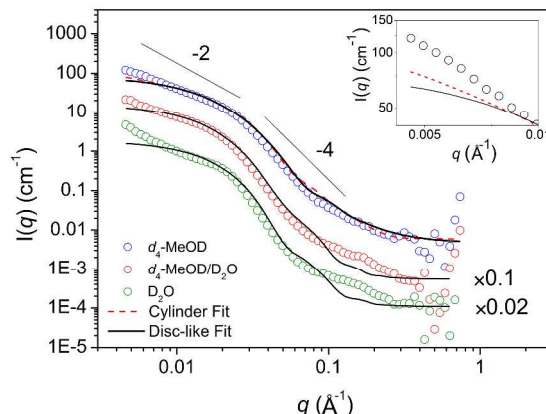


Figure 7. SANS data for **P3HT-*b*-P3HTPy** in *d*₄-MeOD, 50:50 (v/v) *d*₄-MeOD/D₂O and D₂O (10 mg mL⁻¹). For clarity, the *d*₄-MeOD/D₂O and D₂O data have been offset by 0.1 and 0.02, respectively. Straight lines show -2 and -4 decays for comparison. The solid black lines show the disc-like fits obtained using the Core-Shell-Cylinder model for each solvent mixture. Only **P3HT-*b*-P3HTPy** in *d*₄-MeOD was found to be better fit by rods at low q , as shown by the red dashed line in the inset.

The SANS data ($0.0046 < q < 0.6$) in each solvent were fitted to a Core-Shell-Cylinder model using a non-linear least-squares method, including polydispersity and q -resolution smearing (fits shown as solid lines in Figure 7).³³ For **P3HT-*b*-P3HTPy** in *d*₄-MeOD, depending on the starting imputed values of length and radius, the fits were found to converge to two slightly differing shapes: (1) a rod with core length, L_{core} , of ~ 80 nm, core radius, r_{core} , of ~ 6 nm and shell thickness, T_{shell} , of ~ 2 nm; (2) a disc with $L_{\text{core}} \sim 75$ nm, $r_{\text{core}} \sim 30$ nm and $T_{\text{shell}} \sim 1$ nm. The rod-like core-shell aggregate (1) resulted in an improved fit at low q (Figure 7, inset). The SANS data for **P3HT-*b*-P3HTPy** in 50:50 (v/v) *d*₄-MeOD/D₂O were also best modelled by a disc-like core-shell aggregate, revealing a wet core with $L_{\text{core}} \sim 11$ nm and $r_{\text{core}} \sim 7.5$ nm, and a thick solvated shell ($T_{\text{shell}} \sim 9.5$ nm). Fitting the SANS data of **P3HT-*b*-P3HTPy** in D₂O similarly yielded a disc with a solvated core ($L_{\text{core}} \sim 9$ nm, $r_{\text{core}} \sim 6$ nm) and a very wet shell of $T_{\text{shell}} \sim 8$ nm. The shell thickness corresponds to the fully extended length of the charged **P3HTPy** block. Therefore it is likely that the drier core corresponds to the aggregated neutral **P3HT** blocks. We note that reasonable fits to the SANS data in D₂O and 50:50 (v/v) *d*₄-MeOD/D₂O could also be obtained with a Core-Shell sphere model, which yielded aggregates of similar dimensions to those described above (Table S2 in Supplementary Information). Similar fits were obtained for **P3HT-*b*-P3HTIm** and can be found with the tabulated summaries of the fitting data for both CPEs in the Supporting Information.

The scattering length densities (SLDs) of the neutral and charged CPE blocks are both $\sim 1.1 \times 10^{-6} \text{ \AA}^{-2}$. Therefore, they can only be distinguished by neutron scattering when one block is substantially more solvated than the other. The SANS data fits for **P3HT-*b*-P3HTPy** in D₂O suggest the presence of aggregates with a highly swollen shell. Since D₂O is expected to be a selective solvent for the charged CPE block, we propose

that the aggregate core consists predominantly of the neutral **P3HT** blocks, surrounded by a highly solvated CPE shell. Since D_2O is a non-solvent for **P3HT**, the hydrophobic effect results in tight packing of the **P3HT** chains within the aggregate core to minimise interaction with the solvent. In 50:50 (v/v) d_4 -MeOD/ D_2O , the core-shell aggregate structure is retained, but the solvent mixture results in improved solvation of both blocks, leading to an increase in both the core radius and shell thickness. In MeOD, **P3HT-*b*-P3HTPy** forms elongated core-shell cylindrical structures, with a solvated core and a drier shell. On the basis of the Hildebrand solubility parameters, MeOH is a relatively poor solvent for **P3HT** ($\delta \sim 19.1 \text{ MPa}^{1/2}$ for **P3HT** and $\delta \sim 29.7 \text{ MPa}^{1/2}$ for MeOH, respectively⁴²) and should be a better solvent for the charged CPE block. We propose that in MeOD, **P3HT-*b*-P3HTPy** forms core-shell aggregates in which the **P3HT** blocks predominantly occupy the core, but cannot exclude partial mixing of the two blocks across both domains. In the future we would like to deuterate the side chains of the **P3HT** blocks in order to obtain unequivocal evidence for the location of the blocks from SANS.

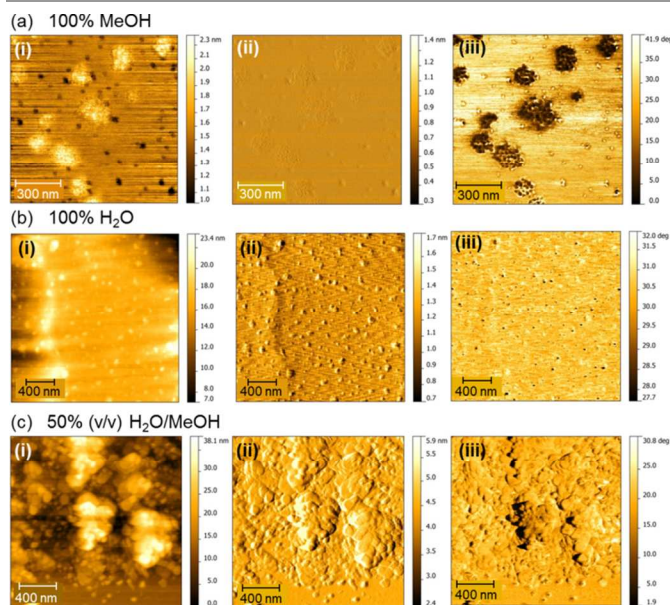


Figure 8. AFM tapping mode images of **P3HT-*b*-P3HTIm** films drop-cast from (a) 100% MeOH ($7 \mu\text{g mL}^{-1}$), (b) 100% H_2O ($3.5 \mu\text{g mL}^{-1}$) and (c) 50% (v/v) $H_2O/MeOH$ ($3.5 \mu\text{g mL}^{-1}$) onto freshly cleaved mica: (i) Height, (ii) amplitude and (iii) phase images, respectively.

Atomic force microscopy was used to determine if the structured aggregates formed in solution could be transferred as thin films with the retention of their nanoscale organisation, which is a key requirement for potential device applications. Solutions of **P3HT-*b*-P3HTIm** at low concentrations ($3.5\text{--}7.0 \mu\text{g mL}^{-1}$) drop cast onto mica substrates reveal the occurrence of spheroids whose surface morphology and size critically depend on the deposition solvent. In MeOH, a honeycomb-like texture is observed on the aggregate surface (Figure 8a), with an average diameter of $128 \pm 28 \text{ nm}$. The surface texture may be indicative of the fusion of several smaller structures into

larger aggregates. In water, slightly larger, isolated aggregates are observed (average diameter $\sim 147 \pm 28 \text{ nm}$), whose surface morphology is much smoother (Figure 8b). In 50% (v/v) MeOH/ H_2O , amorphous films formed from extended networks of much smaller, disc-like clusters, 50–80 nm in diameter, are observed (Figure 8c). Similar trends are observed for all **P3HT-*b*-CPEs** (Figure S20–21 in Supporting Information). It should be noted that the SANS data indicate that the core-shell aggregates formed in solution are highly solvated; in contrast the corresponding thin film samples were dried prior to AFM imaging. It is conceivable that removal of the solvent during drying may lead to partial collapse or reorganisation of the aggregate structure due to fluctuations in the solvent gradient.

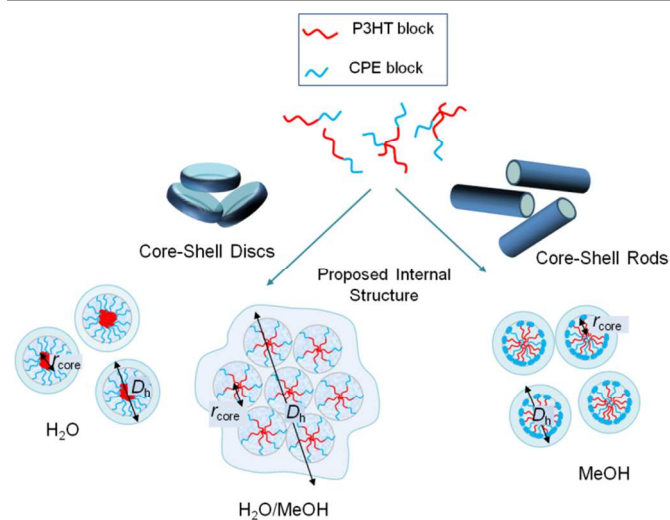


Figure 9. Schematic representation of the proposed internal aggregate structures for **P3HT-*b*-CPE** in H_2O , MeOH and $H_2O/MeOH$ mixtures. L_{core} , r_{core} and T_{shell} are the core length, core radius and shell thickness obtained from SANS data fitting, respectively. D_h is the hydrodynamic diameter measured by DLS. We note that the precise arrangement of the **P3HT** and CPE block domains in the aggregates formed in MeOH is still undetermined.

Figure 9 shows a schematic representation of the proposed aggregate structures adopted by **P3HT-*b*-CPEs** in H_2O , MeOH and $H_2O/MeOH$ mixtures. The smallest aggregates are formed in pure methanol (DLS, SANS, AFM) and exhibit a core-shell cylindrical structure comprised of a long, “wet” core, protected by a “dry” thin shell. MeOH is a poor solvent for **P3HT**, which is anticipated to predominantly occupy the aggregate core; however we cannot exclude some mixing of the **P3HT** and CPE blocks across both the core and shell domains. The small aggregate size results in increased chain proximity and interchain exciton migration, resulting in the lowest Φ_F values being obtained in this solvent. Slightly larger aggregates are obtained in pure H_2O (DLS, AFM, SANS) and consist of a drier core of aggregated, neutral **P3HT** blocks and a wet shell of well-dissolved CPE chains. The largest aggregates are formed in MeOH/ H_2O mixtures (DLS, SANS), in which the core and shell are both well-solvated. This is in excellent agreement with the corresponding increase in the Φ_F values observed in MeOH/ H_2O mixtures. Moreover, due to the larger solubility of the entire CPE chain, we expect lower aggregation

numbers for each structure, thereby reducing electrostatic repulsion between the highly-solvated, individual chains from which each aggregate is comprised. Reduced electrostatic repulsion may result in long-range assembly of multiple aggregates, which may explain the large D_h obtained by DLS and the textured extended sheets (rather than individual clusters) observed by AFM.

Conclusions

Amphiphilic rod-rod diblock copolymers containing a hydrophobic **P3HT** block and a hydrophilic cationic **P3HT** block bearing different side groups were synthesized in an efficient way by using a Kumada catalyst-transfer polycondensation. In contrast to the neutral parent polymer **P3HT-*b*-P3HTBr**, the incorporation of cationic terminal groups renders **P3HT-*b*-CPEs** soluble in highly polar solvents. The thermal properties of these copolymers indicate a clear signature of the diblock architecture by the presence of a T_g caused by the CPE block and a T_m corresponding to the **P3HT** block. The nature of the CPE block drastically affects the T_g , and to a lesser extent the melting transition of the **P3HT** block, except in the case of **P3HT-*b*-P3HTNMe₃**. The UV/Vis absorption and photoluminescence properties are solvent-dependent, resulting in a significant bathochromic shift in λ_{max} , increased vibronic structure and a decrease in the fluorescence quantum yield in pure MeOH and H₂O, which is attributed to aggregate formation. Fluorescence quantum yields are enhanced in H₂O/MeOH mixtures, which are accompanied by an analogous trend in the Z-average particle diameter determined by DLS. Small-angle neutron scattering studies have revealed that **P3HT-*b*-CPEs** form core-shell cylindrical aggregates in solution, with the relative thickness and “wetness” of the core and the shell determined by the solvent system. In H₂O/MeOH mixtures, we propose that enhanced solvation of both the core and shell results in aggregate swelling, which promotes disaggregation of localised polymer chains, thereby increasing Φ_F . AFM studies reveal that the aggregates formed in solution can be transferred as thin films.

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Notes and references

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† Electronic Supplementary Information (ESI) available: ¹H, ¹³C{¹H} NMR spectra of the precursor block copolymers and the ionic block copolythiophenes, ³¹P{¹H} NMR spectra of **P3HT-*b*-P3HTPMe₃**. MALDI-TOF spectra for **P3HT-*b*-P3HTBr**. UV-Vis. spectra of **P3HT-*b*-CPE**. Absorption, excitation and PL spectra of **P3HT-*b*-CPE** in H₂O/MeOH mixtures. Correlogram and size distribution obtained by DLS of **P3HT-*b*-P3HTPMe₃** in MeOH. SANS data analysis for **P3HT-*b*-CPEs**. AFM images of **P3HT-*b*-CPEs** drop-cast from MeOH, H₂O or H₂O/mixture. See DOI: 10.1039/b000000x/

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Table of Contents

The optical, thermal properties and solvent-dependent assembly of amphiphilic all-conjugated cationic copolythiophene block copolyelectrolytes are investigated.

