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ARTICLE TYPE

# Fabric-based flexible electrode with multi-walled carbon nanotubes@Ni network structure as a novel anode for hydrogen peroxide electrooxidation

Dongming Zhang, Dianxue Cao, Ke Ye, Yang Xu, Kui Cheng, Jinling Yin, Guiling Wang\*

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A simple method involving dyeing and electrodeposition is introduced to fabricate a three-dimensional Ni@Multi-walled carbon nanotubes flexible electrode on wearable fabric. The as-prepared Ni@multi-walled carbon nanotubes/Fabric (Ni@MWNTs/Fabric) electrode is characterized by scanning electron microscope and X-ray diffraction spectrometer. The catalytic activity of the Ni@MWNTs/Fabric electrode for hydrogen peroxide electrooxidation is tested by means of cyclic voltammetry and chronoamperometry. Such a three dimensional hybrid electrode structure allows a large specific surface area and a large mass loading, which lead to a high areal current density of 720 mA cm<sup>-2</sup> at 0.5 V in 2 mol dm<sup>-3</sup> NaOH and 2.5 mol dm<sup>-3</sup> hydrogen peroxide. The electrode shows great promise as the anode of direct peroxide fuel cell due to its flexible, wearable, and environmentally friendly.

## 1. Introduction

Direct peroxide-peroxide fuel cell (DPPFC) has attracted much attention due to its low cost, compact, easy operation, workable without air, providing both power and oxygen, and so on [1-7]. As the anode fuel of DPPFC, the electrooxidation of hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>) has fast kinetics and involve no poisoned species [1-13], which allows it a promising and environmentally friendly fuel. So the study of the electrocatalysts for H<sub>2</sub>O<sub>2</sub> electrooxidation has become a hot topic in recent years. Several types of catalysts for the electrooxidation of H<sub>2</sub>O<sub>2</sub> have been developed. It is reported that previous metals, such as platinum [4, 8-10], palladium [4, 5], gold [4-5, 11], silver [12] and a combination of these [5] exhibit effective catalytic performance, but the high prices and the limited resource restrict their extensive utilization. In order to cut down the use amount of noble metals and reduce the cost of electrode, transition metals, such as Ni, Co [4, 6, 7], are employed to catalyze the H<sub>2</sub>O<sub>2</sub> electrooxidation. However, they suffer a drawback of low catalytic activity. So, developing the transition catalysts, which can make H<sub>2</sub>O<sub>2</sub> electrooxidation more effective, is necessary and meaningful. An excellent substrate is one of the decisive factors that determine a catalytic activity of an electrode. And then, the structure, conductivity, mechanical stability and price are the considerations that judge whether the collector is a promising material to apply to our daily life and industrial production. Metallic materials (e.g. aluminum foil, titanium plate, copper sheet) [14-19] are the most widely used electrode material due to its high electronic conductivity. Nowadays, carbon materials (e.g. carbon cloth, carbon paper) [5-6, 20-22] are usually as the collector due to their deformability. Recently, three dimensional

(3D) structure electrodes (e.g. nickel foam, copper foam, titanium foam, sponge and some other foams and nano wires) [7, 23-30] have become a research hotspot owing to their excellent mass transport property, high catalyst utilization and capability of achieving large catalytic surface areas. Nowadays, energy storage devices based on conductive paper [31] or textile [32] with carbon nanotubes (CNTs) has attracted much attention. However, few of them have been employed as a flexible electrode substrate for DPPFC.

Fabrics, which are the raw material of clothes, towels, bed sheets, cotton pads, is indispensable in our daily life. In general, these fabrics are abandoned after the use of a period of time, which will lead to a waste of resource and environmental pollution. Almost all of the people ignore this phenomenon.

In this work, we fabricated a novel MWNTs/Fabric substrate by a simply “dipping and drying” method. The fabric is the cotton pads disused by a girl. MWNTs were wrapped around the fabric by Van der Waals forces and the MWNTs/Fabric exhibits a unique 3D network structure compared with the carbon cloth and carbon paper. Nano-scaled Ni particles were electrodeposited on the surface of the MWNTs/Fabric. The Ni@MWNTs/Fabric electrode saves the metal resources of the electrode material compared with the conventional metal electrode and shows a remarkably high catalytic activity for H<sub>2</sub>O<sub>2</sub> electrooxidation. Besides, the MWNTs/Fabric is much cheaper than the carbon cloth and the utilization of the discarded fabrics reduces environmental pollution.

## 2. Experimental

### Fabrication:

The MWNTs/Fabric substrate is prepared by self-assembling of

MWNTs' layer on a fabric. The fabric is the disused cotton pads, which is commercially made up of polyester fibers and has a high hygroscopicity as a normal cloth (HAOLING Daily necessity & Cosmetics Co. Ltd.). In detail, a piece of fabric were first washed  
 5 by acetone and ethanol for several times and then dried at 373.15 K in a vacuum oven for 2 hours prior to use. Then the cleaned fabric was dipped into a ultrasonic uniform suspension that contains 0.13 g MWNTs (> 50 nm in outer diameter and 10-20  
 10  $\mu\text{m}$  in length, Shenzhen Nanotech Port Co. Ltd.), 0.5 g sodium dodecyl benzene sulfonate (SDBS) and 50 mL ultrapure water (Milli-Q, 18 M $\Omega$  cm). After dipping for 30 s, the fabric was removed from the suspension and dried at 373.15 K for 2 hours. The dip-dry cycle was repeated 5 times to obtain the  
 15 large amount of ultrapure water to remove SDBS. The Ni@MWNTs/Fabric electrode was prepared by electrodeposition of Ni on the surface of the MNWNTs/Fabric, which was carried out in a standard three-electrode electrochemical cell using a computerized potentiostat (Autolab PGSTAT302, Eco Chemie)  
 20 controlled by GPES software. A piece of MWNTs/Fabric substrate ( $10 \times 10$  mm) was employed as the working electrode, a platinum foil as the counter electrode, and a saturated Ag/AgCl, KCl electrode as the reference electrode. The electrodeposition were carried out at a constant current of 5, 10, 15 and 20 mA cm $^{-2}$   
 25 in 2 mol dm $^{-3}$  NH $_4$ Cl and 0.1 mol dm $^{-3}$  NiCl $_2$ , respectively. The electrodeposition time was kept constant at 4 hours.

#### Materials characterization:

The morphology of the electrodes was determined using a scanning electron microscope (SEM, JEOL JSM-6480). The  
 30 structure was analyzed by a powder X-ray diffractometer (XRD, Rigaku TTR- III) equipped with Cu K $\alpha$  radiation ( $\lambda=0.15406$  nm).

#### Electrochemical measurements:

The electrochemical oxidation of H $_2$ O $_2$  was measured by cyclic  
 35 voltammetry and chronoamperometry in the same three-electrode electrochemical cell using NaOH as the electrolyte and the 1cm $^2$  Ni@MWNTs/Fabric electrode. All solutions were made with analytical grade chemical reagents and ultra-pure water (Milli-Q 18 MU cm). All potentials were referred to the saturated  
 40 Ag/AgCl, KCl reference electrode.

### 3. Results and discussion

The fabrication process of the Ni@MWNTs/Fabric electrode via dyeing and electrodeposition is illustrated in Fig. 1. The original fabric is pure white, and after dyeing of MWNTs and  
 45 electrodeposition of Ni, the fabric becomes black and green, respectively. The MWNTs/Fabric substrate is about 0.0079 g and the mass reached to 0.0155, 0.0228, 0.0272 and 0.0304 g after 4 h electrodeposition at the constant current densities of 5, 10, 15 and 20 mA cm $^{-2}$ , respectively. The SEM images of the bare  
 50 MWNTs/Fabric and the Ni@MWNTs/Fabric electrode prepared at different electrodeposition current density (5, 10, 15, 20 mA cm $^{-2}$ ) are shown in Fig. 2. At low magnification, it is obvious that the MWNTs/Fabric substrate, constituted by fabric lines (Fig. 2a), exhibits a 3D-network structure. The diameter of the single  
 55 fabric liner is around 15  $\mu\text{m}$ . At high magnification, we found

that the single fabric line is comprised by some detailed-lines (Fig. 2b). The fabric line shows a plush stuffed surface, which indicated that the MWNTs were coated in the surface. The existence of MWNTs ensures good electrical conductivity (A  
 60 sheet resistance of 1 $\Omega$ /square, measured by four points probe technique) of the MWNTs/Fabric substrate. Besides, the H $_2$ O $_2$  can be absorbed by the fabric, which ensures a full contact between the H $_2$ O $_2$  and Ni during the reaction process. The surface of the fabric lines are encapsulated by the Ni after the  
 65 electrodeposition process (Fig. 2c, e, g, i). In order to compare the Ni particles in the surface of the fabric line at different electrodeposition current densities (5, 10, 15, 20 mA cm $^{-2}$ ), the detail of the four kinds electrodes surfaces are shown by Fig. 2d, f, h and j. As depicted in the figures, the diameters of the Ni  
 70 particles prepared at 5 and 10 mA cm $^{-2}$  are around 1100 nm and the diameter is over 1200 nm when the electrodeposition current density reached to 20 mA cm $^{-2}$ . However, the diameters of the Ni particles prepared at 15 mA cm $^{-2}$  are only around 700 nm. First, the diameter of the Ni nanoparticles is determined by the  
 75 overpotential. When the overpotential increases, the diameter of the Ni nanoparticles will decrease. With the increase of the current density, the overpotential increases. So the diameter of the Ni nanoparticles electrodeposited in 15 mA cm $^{-2}$  is smaller than them at 5 and 10 mA cm $^{-2}$ . However, when the  
 80 electrodeposition is over high (20 mA cm $^{-2}$ ), the electric quantity is higher than the other conditions and the electrodeposition amount of the metallic Ni is more than the other conditions (5, 10, 15 mA cm $^{-2}$ ), which is further demonstrated by the electrodeposition mass of the Ni particles (Fig. 1). Second, the  
 85 metallic Ni is ferromagnetic and the massive Ni particles will aggregate at 20 mA cm $^{-2}$ , which lead to a bigger diameter of Ni particles. So the Ni electrodes prepared at 15 mA cm $^{-2}$  exhibits the largest specific surface area. This will remarkably increase the activity of the Ni owing to the highest surface energy than  
 90 others.

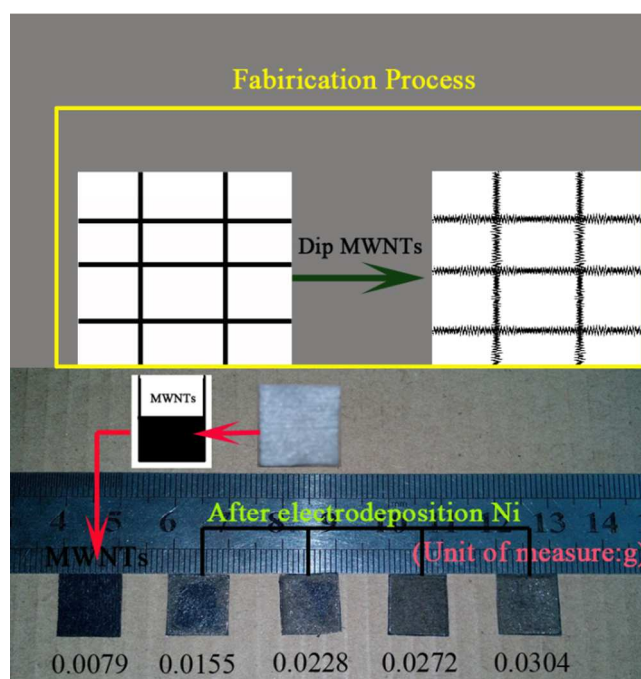


Fig. 1 Fabrication process of the Ni@MWNTs/Fabric electrode.

The X-ray diffraction patterns of the Ni@MWNTs/Fabric electrodes are shown in Fig. 3. The scan range was set from  $10^\circ$  to  $85^\circ$ . It is obvious that the diffraction peaks between  $15^\circ$  to  $30^\circ$  are the substrate peaks. According to the JCPDS card No. 04-8450, the (1 1 1), (2 0 0), (2 2 0) diffractions of cubic Ni are located at  $2\theta=44^\circ, 52^\circ, 76^\circ$ , respectively. It demonstrates that Ni presents in a metallic state of oxides or hydroxides. The intensity of the three Ni peak increases with increase of the electrodeposition current density, so it indicates that the content of the Ni in the Ni@MWNTs/Fabric increases.

The  $\text{H}_2\text{O}_2$  electrooxidation activities on the Ni@MWNTs/Fabric electrode with different electrodeposition current density are shown in Fig. 4. Fig. 4a to d represents the electrodeposition current densities range from 5 to  $20 \text{ mA cm}^{-2}$ . All of the open circuit potentials (OCP) of the Ni@MWNTs/Fabric electrode in  $\text{H}_2\text{O}_2$  solution are around  $-0.16 \text{ V}$ . The Ni@MWNTs/Fabric electrode exhibits a strong oxidation peak at around  $0.43 \text{ V}$  and an obvious reduction peak at around  $0.23 \text{ V}$  in blank NaOH solution, which is attributed to the redox reactions of  $\text{Ni}^{2+}/\text{Ni}^{3+}$  according to the previous report [4, 6-7]. The surface of the metallic Ni will be partly oxidized to  $\text{Ni}^{2+}$  in a solution with strong oxidation and the conversion of  $\text{Ni}^{2+}/\text{Ni}^{3+}$  will occur at high potential. Besides, the existence of  $\text{Ni}^{2+}$  helps the negative valence oxygen in  $\text{H}_2\text{O}_2$  to diffuse to the surface of the Ni electrode. Compared with the electrooxidation behavior in bare NaOH solution, the current density at the oxidation peak increases with the increase of the  $\text{H}_2\text{O}_2$  solution, which demonstrated that the oxidation of  $\text{H}_2\text{O}_2$  is controlled by diffusion at low  $\text{H}_2\text{O}_2$  concentration. When the  $\text{H}_2\text{O}_2$  concentration is higher than  $1.0 \text{ mol dm}^{-3}$ , the oxidation peak of  $\text{H}_2\text{O}_2$  disappeared. The highest current density at  $0.5 \text{ V}$  as a function of the deposition current density is shown in fig. 4e. Clearly, the highest oxidation current density reached to  $340, 630, 720$  and  $640 \text{ mA cm}^{-2}$  at  $0.5 \text{ V}$  in  $2.0, 3.5, 2.5$  and  $3.0 \text{ mol dm}^{-3} \text{ H}_2\text{O}_2$  solution on the four electrodes. But if the  $\text{H}_2\text{O}_2$  concentration increases further, the current density will decrease.

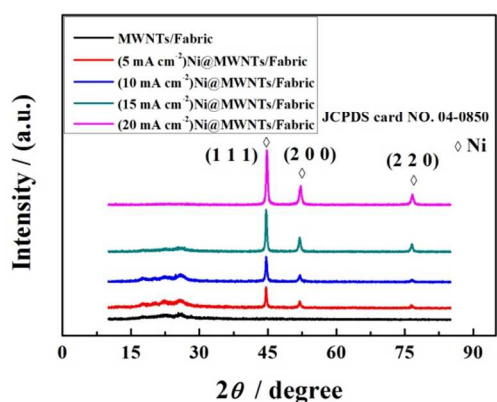


Fig. 3 X-ray diffraction patterns of the Ni@MWNTs/Fabric electrodes.

The schematic diagram depicting the mechanism for the  $\text{H}_2\text{O}_2$  electrooxidation on the Ni@MWNTs/Fabric electrode is illustrated by Fig. 5. The catalytic performance of electrooxidation of  $\text{H}_2\text{O}_2$  on the electrode is effect seriously by  $\text{H}_2\text{O}_2$  concentration. When  $\text{H}_2\text{O}_2$  concentration is low, there are

adequate Ni particles acted as the catalytic sites for  $\text{H}_2\text{O}_2$  electrooxidation, the O-O bond can be fully contacted with Ni (style I in Fig. 5) and convert to  $\text{O}_2$  and release electrons [eq.1] (2-8). With increasing the  $\text{H}_2\text{O}_2$  concentration, all of the catalytic sites in Ni particles will be occupied by O-O bond and some surplus  $\text{H}_2\text{O}_2$  can't be electro-catalyzed (style II in Fig. 5). However, when the  $\text{H}_2\text{O}_2$  concentration is over high, most of the O-O bond in  $\text{H}_2\text{O}_2$  can't come into contact with Ni, which will cause a large number of hydrolysis of  $\text{H}_2\text{O}_2$  [eq.2] (style III in Fig. 5) (2-5), and the released  $\text{O}_2$  will be adsorbed on the surface of the Ni particles and inhibit the contact between the Ni and  $\text{H}_2\text{O}_2$ . Besides, the released  $\text{O}_2$  may reduce the electronic conductivity of the reaction solution, which will also lead a poor performance.

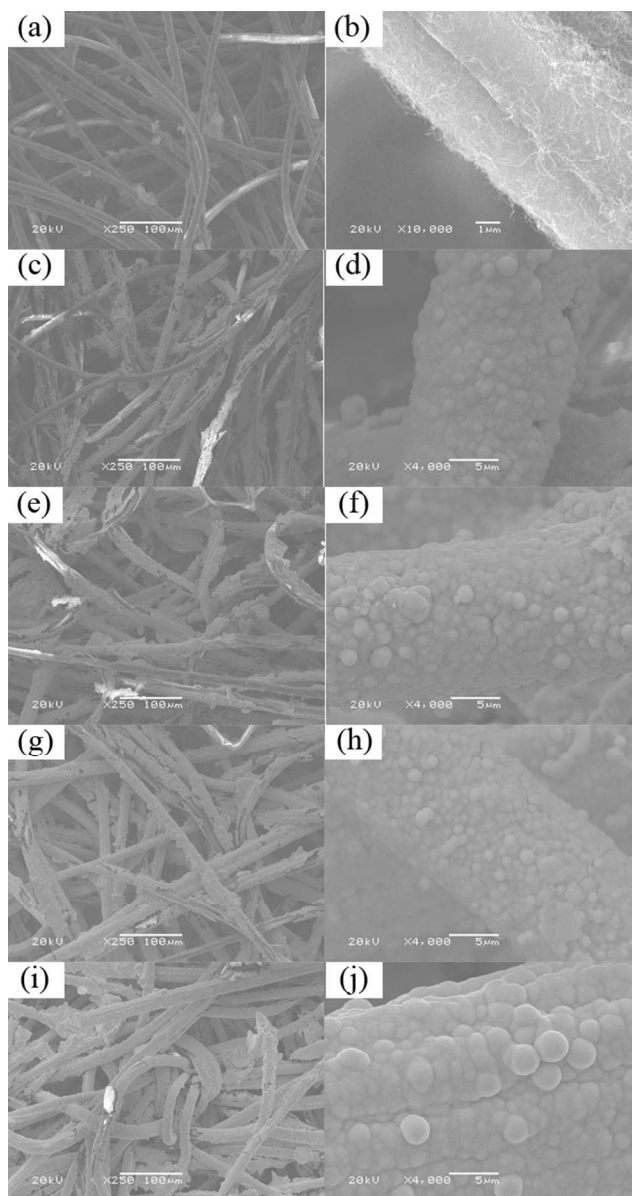
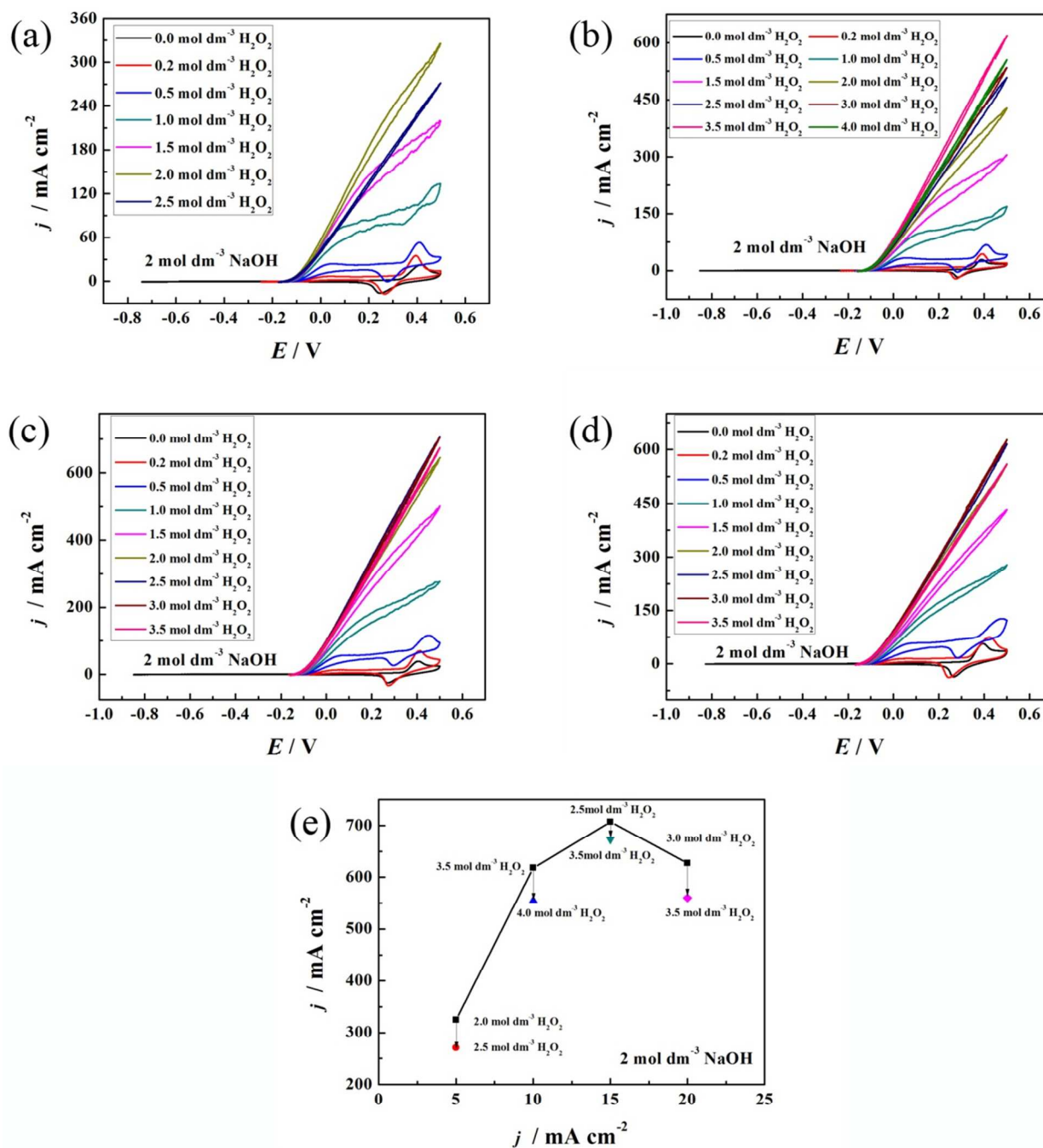


Fig. 2 SEM images of the bare MWNTs/Fabric (a, b) and the Ni@MWNTs/Fabric electrode prepared at different electrodeposition current density 5 (c, d), 10 (e, f), 15 (g, h), 20 (i, j)  $\text{mA cm}^{-2}$

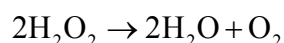
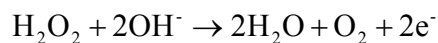
When the concentration of the alkali liquor was fixed, the  $\text{H}_2\text{O}_2$  at high concentration ( $0.25\text{--}2.0 \text{ mol dm}^{-3}$ ) almost have none influence for the electrooxidation of  $\text{H}_2\text{O}_2$  on the Ni-carbon fiber

[6] and Ni-Ni foam [7] electrodes. However, the catalytic performance of the  $\text{H}_2\text{O}_2$  electrooxidation is remarkably affected by the  $\text{H}_2\text{O}_2$  concentration ( $0.20\sim 2.5 \text{ mol dm}^{-3}$ ) on the Ni@MWNTs/Fabric electrode, which is vital to change discharge current to satisfy industrial requirements. So as the fuel of the

underwater and space power source, the  $\text{H}_2\text{O}_2$  at high and different concentration not only can provide large amounts of oxygen for breathing and offer different discharge current, both of which is more favorable than our previous work [6-7].



10 **Fig. 4**  $\text{H}_2\text{O}_2$  electrooxidation activities on the Ni@MWNTs/Fabric electrode with different electrodeposition current density a)  $5 \text{ mA cm}^{-2}$ , b)  $10 \text{ mA cm}^{-2}$ , c)  $15 \text{ mA cm}^{-2}$ , d)  $20 \text{ mA cm}^{-2}$ , Scan rate:  $50 \text{ mV s}^{-1}$ . The highest current density at  $0.5 \text{ V}$  as a function of the deposition current density (e)



(1) Among the four electrodes, the highest catalytic performance is the electrode prepared at a constant electrodeposition current of  $15 \text{ mA cm}^{-2}$  (Fig. 4e), so we discussed the specific catalytic activity of  $\text{H}_2\text{O}_2$  electrooxidation on this electrode in the following work.

The effects of NaOH concentration are discussed in Fig. 6. The concentration of  $\text{H}_2\text{O}_2$  was fixed at  $1.0 \text{ mol dm}^{-3}$  and the concentration of NaOH varies from 1 to  $6 \text{ mol dm}^{-3}$ . All of the onset potentials are around  $-0.16 \text{ V}$ . At low NaOH concentrations ( $1$  and  $3 \text{ mol dm}^{-3}$ ), the current density reached to around  $270 \text{ mA cm}^{-2}$  and there is no obvious reduction peak in the CV curves. As contrast, the reduction peaks gradually shifted

to the negative direction when the NaOH concentration increases from 3 to  $6 \text{ mol dm}^{-3}$ . Clearly, the current densities decrease significantly when the concentration of NaOH is higher than  $3 \text{ mol dm}^{-3}$ . The over high NaOH concentration reduces the  $\text{H}_2\text{O}_2$  concentration that diffuse to the surface of the electrode. Besides, it is obvious that the MWNTs/Fabric has no catalytic activity for the  $\text{H}_2\text{O}_2$  electrooxidation.

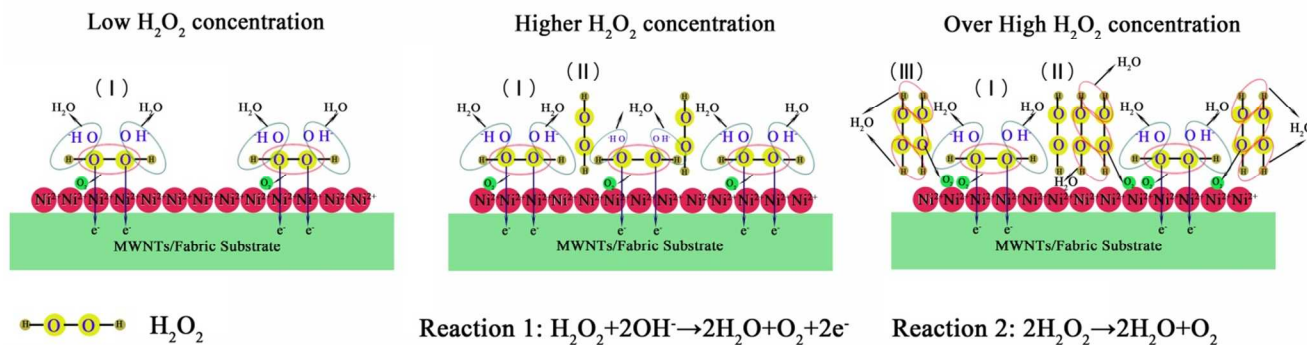


Fig. 5 Schematic diagram depicting the mechanism for the  $\text{H}_2\text{O}_2$  electrooxidation on the Ni@MWNTs/Fabric electrode.

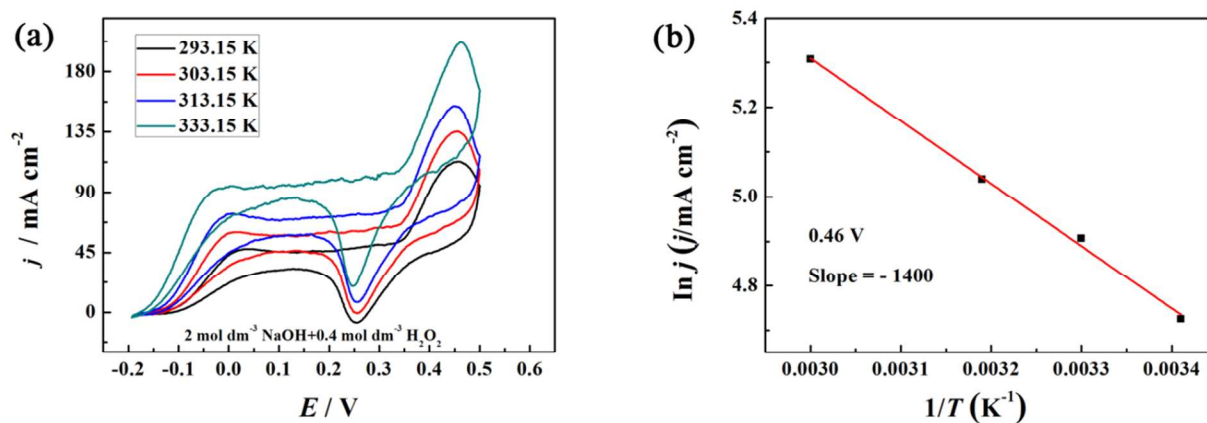
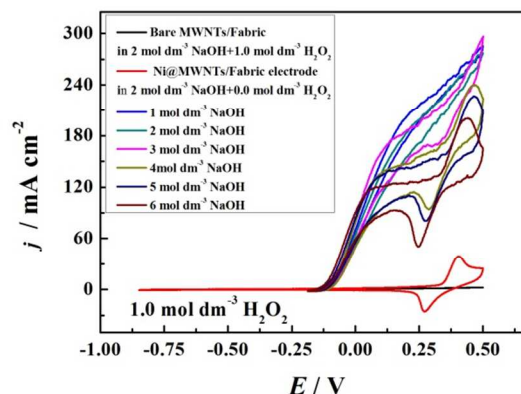


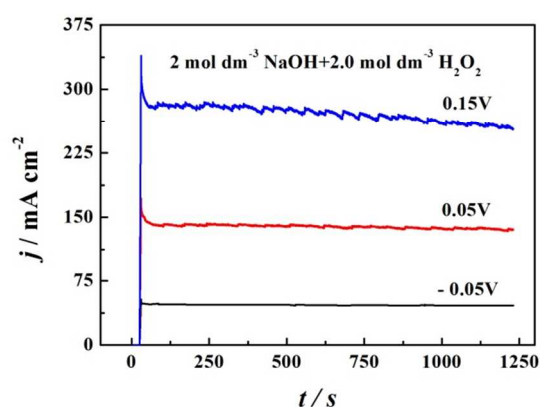
Fig. 8 The cyclic voltammograms of  $\text{H}_2\text{O}_2$  electrooxidation on the Ni@MWNTs/Fabric electrode in  $2.0 \text{ mol dm}^{-3}$  NaOH and  $0.4 \text{ mol dm}^{-3}$   $\text{H}_2\text{O}_2$  at different temperatures, Scan rate:  $50 \text{ mV s}^{-1}$  (a); Arrhenius plot of the current densities at  $0.46 \text{ V}$  for  $\text{H}_2\text{O}_2$  electrooxidation on the Ni@MWNTs/Fabric electrode (b).

The stability of Ni@MWNTs/Fabric electrode for  $\text{H}_2\text{O}_2$  electrooxidation was investigated by chronoamperometric measurements in  $2 \text{ mol dm}^{-3}$  NaOH and  $2.0 \text{ mol dm}^{-3}$   $\text{H}_2\text{O}_2$ . The result is shown in Fig. 7. The potentials were selected in the region according to the polarization curves in Fig. 4. At low potentials ( $-0.05 \text{ V}$  and  $0.05 \text{ V}$ ), the oxidation current density reached to steady-state quickly after about  $50 \text{ s}$  and kept stable during the  $20 \text{ min}$  test period. However, when the constant potential was applied at  $0.15 \text{ V}$ , the CA curve gets rough and the current density decreased slightly during the reaction process. This phenomenon may be caused by the large amount of oxygen bubbles in the surface of electrode and the consumption of the  $\text{H}_2\text{O}_2$  at high oxidation potential [7].



**Fig. 6** The cyclic voltammograms for the Ni@MWNTs/Fabric electrode in  $x \text{ mol dm}^{-3} \text{ NaOH}$  ( $x=1, 2, 3, 4, 5, 6$ ) and  $1.0 \text{ mol dm}^{-3} \text{ H}_2\text{O}_2$ ; the bare MWNTs/Fabric in  $2 \text{ mol dm}^{-3} \text{ NaOH}$  and  $1.0 \text{ mol dm}^{-3} \text{ H}_2\text{O}_2$  and the Ni@MWNTs/Fabric electrode in  $2 \text{ mol dm}^{-3} \text{ NaOH}$  and  $0.0 \text{ mol dm}^{-3} \text{ H}_2\text{O}_2$  are also shown for comparison, Scan rate:  $50 \text{ mV s}^{-1}$ .

The reaction temperature has great influence on the electrochemical behavior of  $\text{H}_2\text{O}_2$  electrooxidation. It can be seen from Fig. 8, the onset potentials move a little more negatively and the oxidation current densities increase obviously with increasing the temperature, respectively (Fig. 8a). The activation energy for the electrooxidation of  $\text{H}_2\text{O}_2$  on Ni@MWNTs/Fabric electrode was calculated to be  $-11.64 \text{ KJ mol}^{-1}$  obtained from the Arrhenius relationship (Eq. 3) [7]. Where  $j$  is the current density;  $T$  is the thermodynamic temperature;  $R$  is the molar gas constant; and  $E_a$  is the activation energy. The logarithm of peak current densities ( $\ln j$ ) at  $0.46 \text{ V}$  were plotted against the reciprocal of absolute temperatures ( $1/T$ ) (Fig. 8b).



**Fig. 7** Chronoamperometric curves for  $\text{H}_2\text{O}_2$  electrooxidation at different potentials in  $2.0 \text{ mol dm}^{-3} \text{ NaOH}$  and  $2.0 \text{ mol dm}^{-3} \text{ H}_2\text{O}_2$ .

$$\frac{\partial \ln j}{\partial T} = \frac{E_a}{RT^2} \quad (3)$$

#### 4. Conclusions

In summary, a novel Ni@MWNTs/Fabric electrode, assembled by simple directly infiltrating and electrodeposition methods, is employed to catalyze the electrooxidation of  $\text{H}_2\text{O}_2$ . The Ni@MWNTs/Fabric electrode exhibits a 3D network structure and allows a large mass loading of Ni particles. The oxidation current density reached to  $720 \text{ mA cm}^{-2}$  in  $2.0 \text{ mol dm}^{-3} \text{ NaOH}$  and  $2.5 \text{ mol dm}^{-3} \text{ H}_2\text{O}_2$  at  $0.5 \text{ V}$ . The activation energy for the electrooxidation of  $\text{H}_2\text{O}_2$  on Ni@MWNTs/Fabric electrode was calculated to be  $-11.64 \text{ KJ mol}^{-1}$ .

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#### Notes and references

Key Laboratory of Superlight Materials and Surface Technology of Ministry of Education, College of Materials Science and Chemical Engineering, Harbin Engineering University, Harbin, 150001, P.R.China. E-mail: wangguiling@hrbeu.edu.cn; Fax: +010-86-451-82589036; Tel: +010-86-451-82589036

- [1] S.A. Mousavi Shaegh, N.-T. Nguyen, S.M. Mousavi Ehteshami, S.H. Chan, A membraneless hydrogen peroxide fuel cell using prussian blue as cathode material, *Energy & Environmental Science* 5 (2012) 8225.
- [2] Y. Yamada, S. Yoshida, T. Honda, S. Fukuzumi, Protonated iron-phthalocyanine complex used for cathode material of a hydrogen peroxide fuel cell operated under acidic conditions, *Energy & Environmental Science* 4 (2011) 2822.
- [3] Y. Yamada, Y. Fukunishi, S.-i. Yamazaki, S. Fukuzumi, Hydrogen peroxide as sustainable fuel: electrocatalysts for production with a solar cell and decomposition with a fuel cell, *Chemical Communications* 46 (2010) 7334.
- [4] S.-i. Yamazaki, Z. Siroma, H. Senoh, T. Ioroi, N. Fujiwara, K. Yasuda, A fuel cell with selective electrocatalysts using hydrogen peroxide as both an electron acceptor and a fuel, *Journal of Power Sources* 178 (2008) 20.
- [5] F. Yang, K. Cheng, T. Wu, Y. Zhang, J. Yin, G. Wang, D. Cao, Dendritic palladium decorated with gold by potential pulse electrodeposition: Enhanced electrocatalytic activity for  $\text{H}_2\text{O}_2$  electroreduction and electrooxidation, *Electrochimica Acta* 99 (2013) 54.
- [6] F. Yang, K. Cheng, X. Xiao, J. Yin, G. Wang, D. Cao, Nickel and cobalt electrodeposited on carbon fiber cloth as the anode of direct hydrogen peroxide fuel cell, *Journal of Power Sources* 245 (2014) 89.
- [7] F. Yang, K. Cheng, X. Xue, J. Yin, G. Wang, D. Cao, Three-dimensional porous Ni film electrodeposited on Ni foam: High performance and low-cost catalytic electrode for  $\text{H}_2\text{O}_2$  electrooxidation in KOH solution, *Electrochimica Acta*, 107 (2013) 194.
- [8] S.B. Hall, E.A. Khudaish, A.L. Hart, Electrochemical oxidation of hydrogen peroxide at platinum electrodes. Part I. An adsorption-controlled mechanism, *Electrochimica Acta* 43 (1997) 579.
- [9] S.B. Hall, E.A. Khudaish, A.L. Hart, Electrochemical oxidation of hydrogen peroxide at platinum electrodes. Part II: effect of potential, *Electrochimica Acta* 43 (1998) 2015.
- [10] S.B. Hall, E.A. Khudaish, A.L. Hart, Electrochemical oxidation of hydrogen peroxide at platinum electrodes. Part III: Effect of temperature, *Electrochimica Acta* 44 (1999) 2455.
- [11] M. Gerlache, Z. Senturk, G. Quarin, J.-M. Kauffmann, Electrochemical Behavior of  $\text{H}_2\text{O}_2$  on Gold, *Electroanalysis* 9 (1997) 1088.
- [12] M. Honda, T. Kodera, H. Kita, On the electrochemical behavior of  $\text{H}_2\text{O}_2$  at Ag in alkaline solution, *Electrochimica Acta* 28 (1983) 727.
- [13] A. Aytaç, M. Gürbüz, A.E. Sanli, Electrooxidation of hydrogen peroxide and sodium borohydride on Ni deposited carbon fiber electrode for alkaline fuel cells, *International Journal of Hydrogen Energy* 36 (2011) 10013.
- [14] C. Iwakura, Y. Fukumoto, H. Inoue, S. Ohashi, S. Kobayashi, H. Tada, M. Abe, Electrochemical characterization of various metal foils as a current collector of positive electrode for rechargeable lithium batteries, *Journal of Power Sources* 68 (1997) 301.
- [15] H.-C. Wu, E. Lee, N.-L. Wu, T.R. Jow, Effects of current collectors on power performance of  $\text{Li}_4\text{Ti}_5\text{O}_{12}$  anode for Li-ion Battery, *Journal of Power Sources* 197 (2012) 301.
- [16] S.-L. Sin, L.-L. Teo, K.-S. Tan, C.-Y. Chan, ESCA and TOF-SIMS study on oxidised and reduced polypyrrole-poly(4-styrenesulfonate) free-standing films synthesized on Ti electrodes, *Electrochemistry Communications* 2 (2000) 685.
- [17] G. Wang, Y. Lin, Performance characteristics of dye-sensitized solar cells with counter electrode based on NiP-plated glass and titanium plate, *Current Applied Physics* 9 (2009) 1005.
- [18] G. Fonder, C. Volcke, B. Csoka, J. Delhalle, Z. Mekhalif, Electrochemical and spectroscopic study of  $\text{C}_{12}\text{H}_{25}\text{X}$  molecules adsorption on copper sheets, X (-SH, -S-S-, -SeH and -Se-Se-), *Electrochimica Acta* 55 (2010) 1557.

- [19] X.H. Huang, X.H. Xia, Y.F. Yuan, F. Zhou, Porous ZnO nanosheets grown on copper substrates as anodes for lithium ion Batteries, *Electrochimica Acta* 56 (2011) 4960.
- [20] A. Perrard, L. Retailleau, R. Berjoan, J.-P. Joly, Liquid phase oxidation kinetics of an excellulose activated carbon cloth by NaOCl, *Carbon* 50 (2012) 2226.
- [21] M.-C. Tsai, T.-K. Yeh, Z.-Y. Juang, C.-H. Tsai, Physical and electrochemical characterization of platinum and platinum–ruthenium treated carbon nanotubes directly grown on carbon cloth, *Carbon* 45 (2007) 383.
- [22] Q. Si, M. Kawakubo, M. Matsui, T. Horiba, O. Yamamoto, Y. Takeda, N. Seki, N. Imanishi, Silicon-carbon composite dispersed in a carbon paper substrate for solid polymer lithium-ion batteries, *Journal of Power Sources* 248 (2014) 1275.
- [23] N. Feng, D. Hu, P. Wang, X. Sun, X. Li, D. He, Physical Chemistry Chemical Physics, 15 (2013) 9924.
- [24] K. Ji, C. Xu, H. Zhao, Z. Dai, Electrodeposited lead-foam grids on copper-foam substrates as positive current collectors for lead-acid batteries, *Journal of Power Sources* 248 (2014) 307.
- [25] X. Niu, Y. Li, J. Tang, Y. Hu, H. Zhao, M. Lan, Electrochemical sensing interfaces with tunable porosity for nonenzymatic glucose detection: A Cu foam case, *Biosensors and Bioelectronics* 51 (2014) 22.
- [26] Z. Bi, M.P. Paranthaman, P.A. Menchhofer, R.R. Dehoff, C.A. Bridges, M. Chi, B. Guo, X.-G. Sun, S. Dai, Self-organized amorphous TiO<sub>2</sub> nanotube arrays on porous Ti foam for rechargeable lithium and sodium ion batteries, *Journal of Power Sources* 222 (2013) 461.
- [27] W. Chen, R.B. Rakhi, L. Hu, X. Xie, Y. Cui, H.N. Alshareef, High-performance nanostructured supercapacitors on a sponge, *Nano Letters* 11 (2011) 5165.
- [28] W. Chen, R.B. Rakhi, H.N. Alshareef, High energy density supercapacitors using macroporous kitchen sponges, *Journal of Materials Chemistry*, 22 (2012) 14394.
- [29] Fu, G. Qi, R. Sahore, R. Sougrat, F.J. DiSalvo, E.P. Giannelis, *Physical Chemistry Chemical Physics*, 15 (2013) 19134-19137.
- [30] L. Hu, H.S. Kim, J.-Y. Lee, P. Peumans, Y. Cui, Scalable coating and properties of transparent, flexible, silver nanowire electrodes, *ACS Nano* 4 (2010) 2955.
- [31] Q. Cheng, Z. Song, T. Ma, B.B. Smith, R. Tang, H. Yu, H. Jiang, C.K. Chan, *Nano Letters*, 13 (2013) 4969.
- [32] L. Hu, M. Pasta, F.L. Mantia, L. Cui, S. Jeong, H.D. Deshazer, J.W. Choi, S.M. Han, Y. Cui, *Nano Letters*, 10 (2010) 708.