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## ARTICLE

# Selectively Self-assembling Graphene Nanoribbons with Shaped Iron Oxide Nanoparticles

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Nanoflower and nanoplate-shaped iron oxide nanoparticles (FeNPs) were attached onto graphene nanoribbons (GNRs). Raman spectra of FeNP-GNRs reveal red-shifted G-bands along with greatly reduced D-band intensity, indicating a surface-smooth and electron-doped assembly on GNR. Distinctive decorative patterns are associated with unique sizes, shapes, and crystalline of FeNPs. FeNP-nanoflower is predominately attached around the edge of the GNRs; whereas multilayer GNRs are sandwich-stacked with FeNP-nanoplates. With the attachment of IgG antibodies, FeNP-GNRs/IgG depict distinguishable aggregation features, which have potential advantages as biosensors to target tumor cells with over-expressed folic acid.

## Introduction

Iron oxide nanoparticles (FeNPs) are candidates for biomedical applications such as imaging and therapy.<sup>1,2</sup> FeNPs have been used for industrial separations and catalysis owing to their high magnetic susceptibility.<sup>3</sup> In addition, the size, morphology and shapes of FeNPs impact on their magnetic properties and biocompatibility.<sup>4-6</sup> Compared to nanospheres, non-spherical FeNPs have advantages in biomedical applications, particularly with improved circulation time of blood.<sup>7</sup> Furthermore, FeNPs with tunable size, shape, and surface properties are desirable for biomedical application, because they have different spin-lattice ( $T_1$ ) and spin-spin ( $T_2$ ) relaxation time contrast, and thus, they are useful for magnetic resonance imaging (MRI) or hyperthermia cancer treatment.<sup>1,2,8,9</sup> Recently, FeNPs of nanoplate and nanoflower were prepared for potential biomedical application through the decomposition of iron oleate complex *via* the ‘heat-up’ method.<sup>4-6</sup>

Modification of graphene nanoribbons (GNR) on the molecular level with metal nanoparticles is an example of a ‘bottom-up’ fabrication route toward nanoscale devices.<sup>10-13</sup> One-atom thickness and planar graphene structure can be employed for effective donor/acceptor interfaces of charge transfer between metal nanoparticles and GNRs.<sup>10-11</sup> Moreover, functionalized graphene has attractive physicochemical properties, for example, biocompatible for drug carrier, good physiological stability and low cytotoxicity. Disparate functional groups on graphene surfaces (such as carboxylic or hydroxyl) can be utilized for both covalent and noncovalent functionalization graphene with nanoparticles.<sup>14-16</sup> Such functional groups can carry specific biomolecule moieties like protein and DNA, which helps make nanohybrid dispersible and biocompatible in a physiological environment.<sup>14,15</sup>

However, there are a paucity of reports regarding the decoration of FeNPs onto graphene nanoribbons.<sup>8,16-21</sup> For instance, coating FeNPs with few-layer graphene ensures the long-term stability of the core of FeNPs under biological environments.<sup>8</sup> Formation of composites of dextran-coated Fe<sub>3</sub>O<sub>4</sub> nanoparticles and graphene oxide (Fe<sub>3</sub>O<sub>4</sub>-

GO) leads to prospective application in magnetic resonance imaging (MRI) as  $T_2$ -weighted contrast agent.<sup>16</sup> The electrochemical performance of Fe<sub>3</sub>O<sub>4</sub> nanoparticle/graphene oxide nanocomposites has been evaluated for lithium ion batteries and electrochemical capacitor.<sup>17-21</sup> GNRs can be synthesized by longitudinal unzipping multi-walled carbon nanotubes (MWNT) by oxidation.<sup>22</sup> The oxidation process inevitably results in hole-doped GNR, which is typically in a semiconducting state.<sup>21,22</sup> Although efforts have been made toward reduction of oxygen on GNR surface by hydrazine chemical treatment, it remains a challenging task to effectively control oxidation process.

## Experimental

### Materials

The GNR was obtained by unzipping multiwall carbon nanotubes by oxidation<sup>22</sup> (MWCNTs, Lot# MRCMW10, 60 nm diameter and 5-6  $\mu$ m length, MER cooperation, Tucson AZ 85706 with purity > 90%). The iron oxide nanoflowers (~20 nm) and nanoplates (size ~18 nm, 3 nm thick) in octadecane were obtained through the decomposition of iron oleate *via* the ‘heat-up’ method.<sup>6</sup> The solvents used for experiments, such as chloroform and hexane were purchased with purity of 99.9 % from Sigma-Aldrich Company. IgG antibody (P-17, SC-34665, 36 KDa) and anti-rabbit green fluorescence protein (GFP) was also purchased from Sigma-Aldrich Company.

### Forming GNR-FeNPs nanocomposite

FeNPs solution (1ml) of nanoflowers or nanoplates in octadecane (13 mg/ml) was added with 10 mg graphene nanoribbon in 100 ml chloroform, which was previously sonicated for 5 min. The mixture of GNRs and FeNPs was then stirred vigorously for 1 hour and sonicated again for an extra 10 min. After sonication, the mixture was kept stirring for 2 days at the room temperature. To separate products from the mixture, the solution was filtered through 0.5  $\mu$ m PTFE membrane by Millipore® filtering system to remove unattached FeNPs in the chloroform. The black filter cakes on the

membrane were washed with 200 ml hexane and 200 ml chloroform to remove the residue of FeNPs on the surfaces of GNR. The PTFE membrane of filter cake was then sonicated in 50 ml hexane for 10 minutes again to form a black dispersion solution. The dispersed hexane solution contained the final nanocomposite products and GNRs that are not attached with FeNPs. Such mixture of hexane solution was then concentrated to 25 ml volume and transferred into a tall 25 ml glass cylinder. The bottom of cylinder was surrounded with a strong iron magnet. The solution was kept precipitated for 12 hours under magnetic field. The unattached GNRs of the upper-layer were decanted from the solution. The nanocomposite of FeNPs-GNR was left at the bottom of glass cylinder. After washing with hexane and decanting several times, the products were purified and dried under vacuum. GNR/FeNP-nanoflower (11.2 mg) or nanoplates (14.6 mg) were obtained.

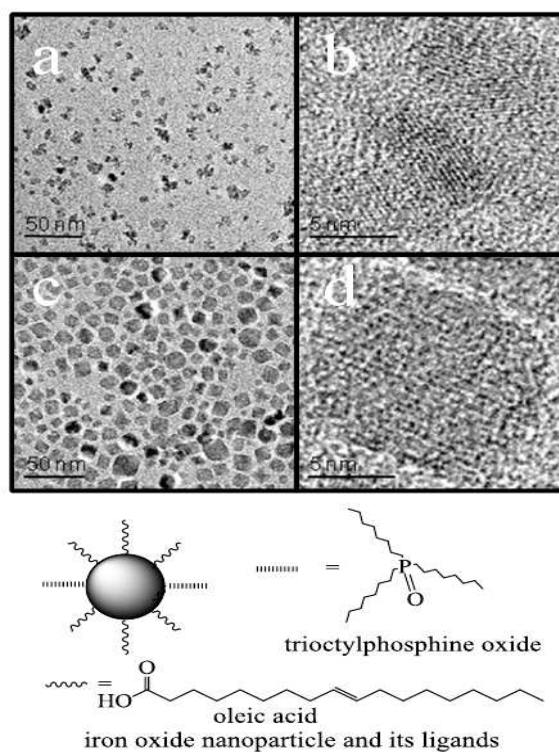
### Characterization

AFM was performed using a Veeco Thermo-microscope CP Research AFM instrument in non-contacted tapping mode with a silicon tip. The GNR/FeNP AFM image was obtained at room temperature in air on SiO<sub>2</sub> plate surfaces. UV-vis spectra were recorded using a Beckman DU640 spec-trophotometer with THF. Raman spectroscopy was performed on a Thermo Scientific Nicolet Almega XR Micro and Macro-Raman Analysis System. The laser source was a wavelength of 488 nm. IR studies were carried out on a Perkin Elmer FTIR Spectrum 65. The size and morphology of FeNPs were examined on a FEI Tecnai F-20 transmission electron microscope (TEM). High-resolution scanning electron microscopy (SEM) images were recorded using a Zeiss SEM Ultra 60 scanning electron microscope. Cyclic voltammogram (CV) was measured in chloroform on PARSTAT® 2263 made by Princeton Applied Research Inc with a working electrode of Solid Electrode (Working Electrode Area is 1.000 cm<sup>2</sup>) and reference electrode of Ag, AgCl/NaCl (saturated) (0.194 V).

Anti-rabbit green fluorescence protein (GFP, 0.2 mg/ml) was dropped onto and allowed to attach to the slides for 10 min, which have IgG antibody (2 × 10<sup>-6</sup> ug/ml) coated FeNP-graphene; and then washed with dionized water to remove unattached protein from the surface for fluorescence microscopic imaging. The fluorescence microscope of Zeiss axioimager Z1 attached with Apotome attachment was used for the acquisition of digital multi-channel fluorescence images using wide-field, structured illumination filter cubes for GFP.

### First-Principles Calculations

Since the GNRs synthesized from unzipping MWNTs have width more than 200 nm, the edge effect of ribbons can be neglected. For this reason, we considered graphene instead of GNRs. Our first-principles calculations were based on general gradient approximation (GGA) with the exchange correlation of Perdew–Burke–Ernzerhof (PBE) parametrization.<sup>35</sup> We employed the dispersion correction using the Tkatchenko-Scheffler (TS) scheme,<sup>36</sup> which exploits the relationship between polarizability and volume. The TS dispersion correction takes into account the relative variation in dispersion coefficients of various atomic bonding. Weighting values are extracted from the high-quality *ab initio* database with atomic volumes derived from partitioning of the self-consistent electronic density. A 5 × 5 cell in the plane direction with a vacuum space of 17.6 Å normal to the graphene plane was used. A kinetic energy change of 3 × 10<sup>-4</sup> eV in the double numerical orbital basis was sufficient to converge with the integration of the charge density. The optimization of the atomic positions proceeds until the change in energy is less than 1 × 10<sup>-5</sup> eV per cell.



**Figure 1.** The scheme of iron nanoparticles structure and TEM images of (a) FeNP-nanoplates; (b) HRTEM of FeNP-nanoplates; (c) FeNP-nanoflowers; and (d) HRTEM of FeNP-nanoflowers.

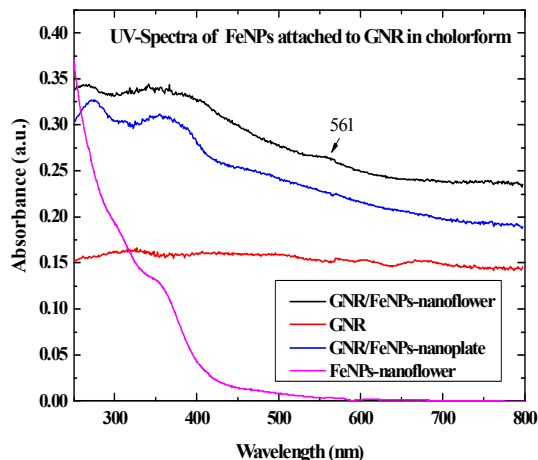
## RESULTS AND DISCUSSION

We found that different crystalline nanostructures of iron nanoparticles lead to distinctive self-assembling motifs onto GNRs. FeNP-nanoflowers are mainly attached around the fringes of GNRs, while FeNP-nanoplate have the tendency to aggregate on the surface of GNRs. As a result, by controlling the nanoshapes of FeNPs, we can select the decoration positions of GNR surfaces to form well-ordered FeNP/GNR nanohybrids. Such a position preference of decoration FeNPs onto GNRs is poised to find applications in cancer treatment<sup>8,9,16</sup> and material application<sup>37-39</sup>. In our study, position control of FeNPs on GNR surfaces will benefit the aggregation preference of IgE antibody with nanohybrid. It influences FeNP-GNR properties as IgG biosensor.

### UV-vis Spectra

Seen from Figure 2, FeNP spectrum on itself demonstrated a shoulder peak at 358 nm of iron *d-d* transitions and the characteristic ligand-to-metal charge transfer (LMCT) below 300 nm.<sup>30</sup> GNRs have a weak shoulder peak around 310 nm attributed to the  $\pi-\pi^*$  transitions of the carbonyl group formed by oxidation.<sup>31</sup> After graphene decoration with FeNP-nanoplate, a peak appeared at 272 nm, which was attributed to  $\pi-\pi^*$  transition of graphene C=C double bond. FeNP-nanoplates were attachment onto GNR surface was conformed by the peak of GNR/FeNP-nanoplates at 357 nm that is attributed to FeNP *d-d* transition. The corresponding peaks of GNR/FeNPs-nanoflower are located at 264 and 353 nm, respectively. A shoulder peak at 561 nm is attributed to iron *d-d* transitions of pair excitations ( ${}^6A_1 \rightarrow {}^4T_1$ ).<sup>23, 24</sup> As seen from UV-vis (Figures 2) spectra, There exists a slight shift of both *d-d* and  $\pi-\pi^*$  transition peaks, and intensity variation of characterized GNR

absorbance peaks, implying interaction between GNR and FeNPs. Comparing UV-vis spectra of GNR/FeNP-nanoflowers with GNR/FeNP-nanoplates, there is little difference in the peak associated with the FeNPs'  $d-d$  transition, while there is evident blueshift of the graphene  $\pi-\pi^*$  peaks for nanoflowers, indicating different assembly patterns.



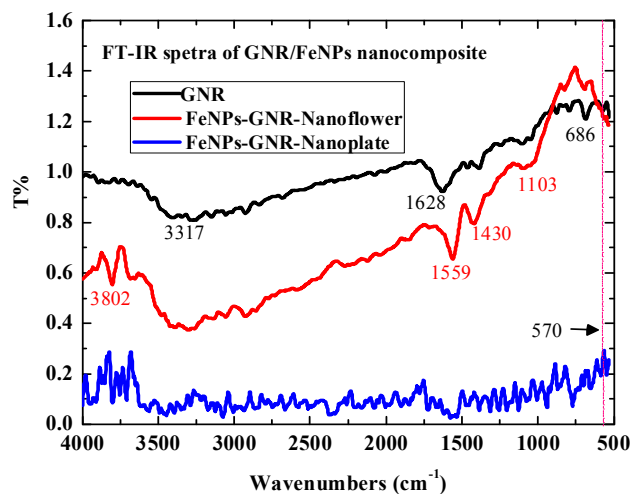
**Figure 2.** UV-vis spectra of GNRs decorated with FeNP-nanoflower and nanoplates.

### FT-IR Spectra

As seen from the FT-IR spectra (Figure 3), different nano-shaped FeNPs show distinctive interaction patterns as in their observed spectra. GNR/FeNP-nanoflower shows a new narrow peak at  $3802\text{ cm}^{-1}$  that is the hydroxyl stretching groups of oleic acid on the GNR plane. The typical vibration bands of C-H at  $1559$ ,  $1430$  and  $1103\text{ cm}^{-1}$  belong to the aliphatic and hydrophobic chain of FeNP ligands. In addition, there is enhanced vibration intensity at  $570\text{ cm}^{-1}$ , consistent with Fe-O vibrations of the  $\text{Fe}_3\text{O}_4$  structure. FT-IR spectra further confirmed the interaction between iron oxide nanoparticles and GNR surfaces (Figure 3). The peaks at  $3300$ ,  $1628$ ,  $1069$ , and  $686\text{ cm}^{-1}$  in the GNR FT-IR spectrum are attributed to hydroxyl stretching groups on the plane, carboxyl groups located at the edges of sheets, the stretching vibration of alkoxy C-O, and aromatic ring out-plane vibration, respectively.<sup>24</sup> The characteristic peaks of GNR (at  $3300$ ,  $1628$ , and  $1069\text{ cm}^{-1}$ ) are dormant in the spectra of FeNPs/GNR, which implies that carboxyl groups on the fringes and plane of GNR interact preferably with iron oxide nanoparticles.<sup>35</sup> The much-weakened intensity of these peaks is reminiscent of the strong interaction between GNR and FeNPs (Figure 3). With different shapes of iron oxide nanoparticles attaching to GNRs, there exist characteristic vibration peaks in the corresponding spectra. GNR/FeNP-nanoflower shows characteristic graphene peaks; whereas, the spectra of GNR/FeNP-nanoplate are so weak that vibration peaks virtually depict the baseline. It indicates the strong interaction between GNR and FeNP-nanoplate. The multilayer stacking of FeNP-nanoplates on graphene surfaces caused its FT-IR peak intensity weak.

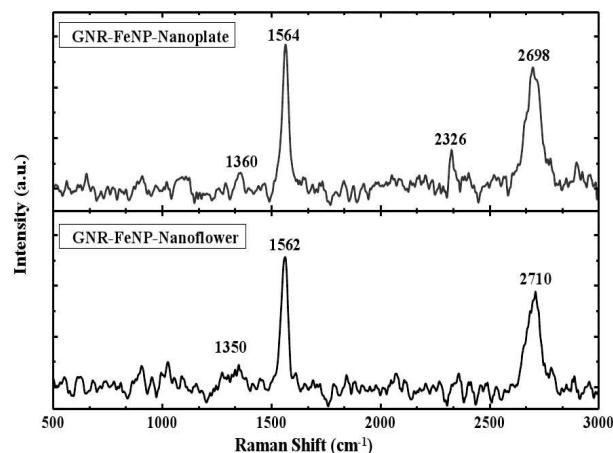
The decoration of FeNPs onto GNR surfaces<sup>37-39</sup> influences characteristic modes of Raman vibrations of defects and the  $sp^2$  graphene network, corresponding to the D- and G-bands, respectively.<sup>21,22</sup> Shown in Figure 4 are the Raman spectra of

GNR/FeNPs-nanoplate and nanoflower.<sup>18-19</sup> Utilizing the unzipping method, oxygen has been introduced on the edges and cause point defects in the lattice structure, which is reminiscent of blue-shifts in Raman G- and D-bands.<sup>19</sup> Due to the tight surface contact between FeNP and GNRs, intrinsic defects of GNRs were covered with FeNPs. Functional groups in GNR defects, such as carboxylic and hydroxyl groups, turn to bond with the iron (II/III) oxide core of nanoparticles; therefore, GNR surfaces become much smoother after decoration with nanoparticles.



**Figure 3.** FT-IR spectra of GNRs decorated with FeNP-nanoflowers or nanoplates and pristine GNR.

### Raman Spectroscopy



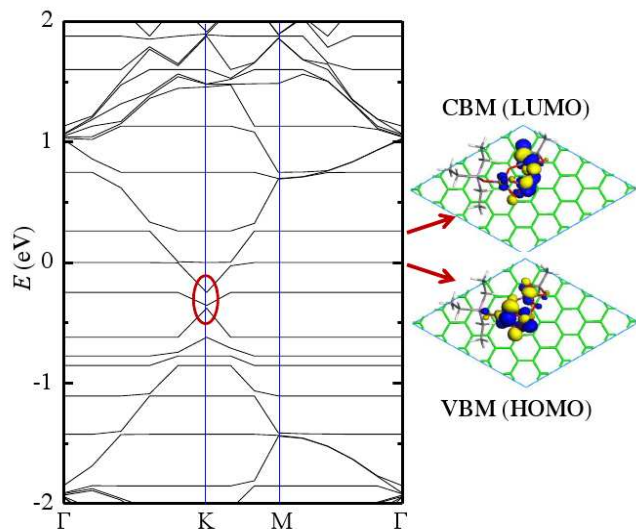
**Figure 4.** Raman spectra of FeNP-nanoplate/nanoflower on GNRs with a laser wavelength excited at  $488\text{ nm}$ .

The longitudinal unzipped GNRs have the characteristic Raman peaks of graphene oxide G-band at  $1603\text{ cm}^{-1}$ , D-band at  $1354\text{ cm}^{-1}$ , and overtone 2D-band in  $2600-3000\text{ cm}^{-1}$ , respectively.<sup>12,13</sup> In the case of FeNPs/GNR composites (Figure 4), the Raman shift of G-band is close to the value of pristine graphene at  $1570\text{ cm}^{-1}$ , which is about  $30\text{ cm}^{-1}$  red-shift from the  $1600\text{ cm}^{-1}$  G-band of oxidized GNR.<sup>22</sup> This implies that the self-assembly of FeNPs onto oxidative GNRs is a selective decoration process. FeNPs tends to interact strongly with the oxygen function groups on GNRs. As a result,

FeNP/GNR nanohybrids are converted to electron-doped graphene characteristics, which is a stark contrast to the blue-shift for oxidative GNRs.<sup>21</sup> The red-shift of G-band also indicates a strong interaction of GNR and FeNPs. For GNR/FeNP-nanoplate, the splitting 2D overtone peaks are located at 2326 and 2698  $\text{cm}^{-1}$ . The peak at 2326  $\text{cm}^{-1}$  is characteristic of multilayer graphite sheets.<sup>11,12</sup> Thus, this is referred to as “multilayer peak”. The multilayer sandwich-stacking of GNRs with FeNP-nanoplates leads to the Raman peak at 2326  $\text{cm}^{-1}$ , which serves as the characteristic signal of assembly pattern.

The D-bands, on the other hand, entail the defects in the assembly. It is worth noting that the unzipped GNRs have nearly equal intensity for G and D bands. By contrast, the suppression of D-band intensities (Figure 4) indicates smooth surfaces of FeNP/GNRs, since the D-band intensity is closely associated with defect vibrations. While the reduction of the D-band is typical for self assembly nanostructures, the substantial reduction of the D-band in FeNP/GNR is remarkable, which strongly implies a site-selective decoration of the FeNPs, particularly for nanoflower-shaped FeNPs. As such, the FeNP/GNRs serve as a desired defect-free substrate for nanoscale hybrids.

### First-Principles Calculations



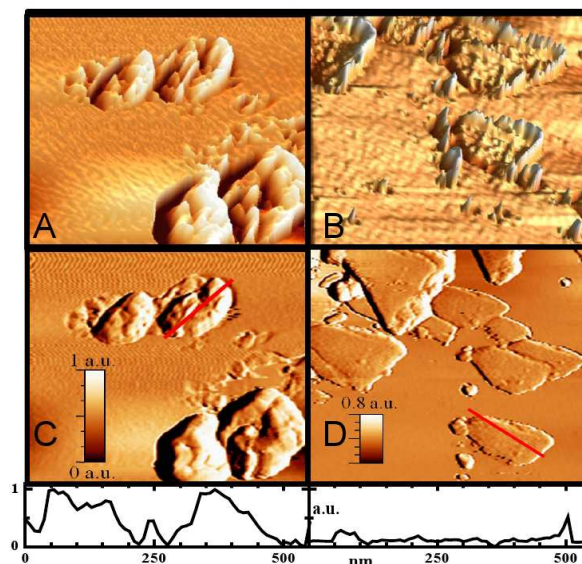
**Figure 5.** Calculated band structure of FeNP/GNR with iron oxide ligand, along with isosurface plots of the valence band maximum (VBM) and conduction band minimum (CBM).

To facilitate an in-depth understanding of the electron doping behavior, we performed first-principles calculations of electronic characteristics for the iron oxide ligand on graphene. The characteristic band structure uncovers down-ward shift of the linear dispersed  $\pi$  and  $\pi^*$  bands at the Dirac point, which is reminiscent of electron doping (Figure 5). As is characteristic to noncovalent functionalized graphene, there exist two types of bands. One type of band refers to flat bands associated with the iron oxide ligand, and the other type of bands is dispersive originated from graphene. In the absence of graphene, electrons in the HOMO and LUMO levels are localized. However, the interaction with graphene yields profound level hybridization between the dispersed bands and the flat bands, which result in delocalization of the corresponding HOMO- and LUMO-derived bands. The calculated band structure reveals that the iron oxide acts as charge donor, which is manifested by about 0.4 eV

downward shift of the Dirac level crossing (highlighted by the red circles in Figure 5)

### Atomic Force Spectroscopy

AFM images uncovered different assembling patterns based on the size and shape of the nanoparticles. The GNRs were unzipped using an oxidative chemistry technique.<sup>25</sup> During the unzipping process, oxidative nucleation sites were created, leading to oxygen functional groups in the  $sp^2$  network, which were located predominantly on the edges of the ribbon. The assembly-induced interactions between GNR and FeNPs promote charge transfer from iron metal center to GNR. There exist electrostatic interaction between iron ions (II/III) and carboxylic groups at the surface defect sites and fringes of GNRs. The long chain ligands (TOPO) of FeNPs have hydrophobic interaction with GNR surfaces, which stabilizes the structure of FeNP/GNR nanocomposites. The shape of the FeNP-nanoplate is regular and has a larger planar surface than that of FeNP-nanoflower. Therefore, the nanoplate tends to overlap and cover the GNR surface, resulting in multilayer nanocomposite GNR/FeNP-nanoplate. The 2D image of GNR/FeNP-nanoplate depicts that FeNP-nanoplates mainly aggregate onto surface of GNR with height of 24-32 nm (Figure 6a). The typical height from peaks to valleys is around 30-40 nm (Figure 6c). For the topography of GNR/FeNP-nanoflower, FeNP-nanoflowers are predominantly distributed around the GNR fringes with height of 8-16 nm (Figure 6b). The 3D topography of GNR/FeNP-nanoflower closely resembles a down-warping basin. The typical depth of a basin is around 4-8 nm (Figure 6d).

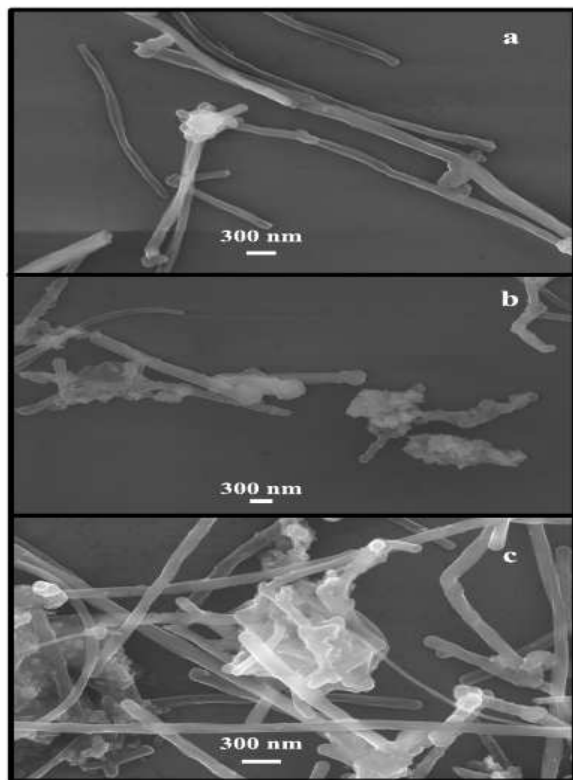


**Figure 6.** (a) 3D AFM images of FeNP/GNR-nanoplates and (b) 3D AFM images of GNR / FeNP-nanoflower; (c) 2D AFM images and topologic line analysis of FeNP/GNR-nanoplates; and (d) 2D AFM images and topologic line analysis of GNR/FeNP-nanoflower (1 a.u. = 40 nm/div).

### Scanning Electronic Spectroscopy

The MWNTs as received have a typical diameter of  $\sim 60$  nm before unzipping. The unzipped tubes have widths around 200 nm, based on high resolution SEM images (Figure 7a), confirming that the tubes were fully unzipped into ribbons. The ribbons have a height of  $\sim 0.50$ – $0.75$  nm indicating single layer ribbons. The stacking of

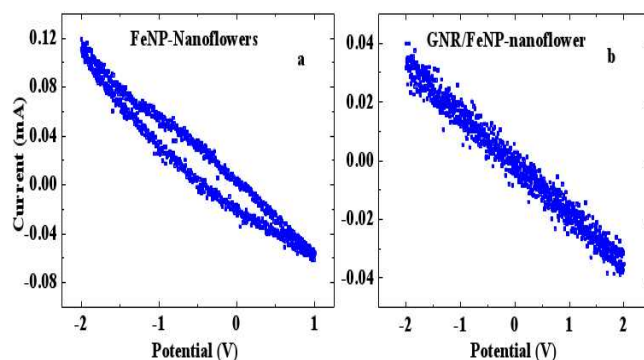
GNRs can be seen in Figures 4b and 4c. The height ranges from 2 to 8 nm, indicating multi-layer stacking of GNRs (AFM, Figure 6d).<sup>26</sup> The SEM of Figure 7b image further confirms that FeNP-nanoflowers are attached to the surfaces and edges of GNRs; whereas Figure 4c shows that there exists multiple aggregation of GNRs onto the surface of FeNP-nanoplates. The distinctive assembly patterns for nanoflowers and nanoplates suggest electronic and biosensor applications that take advantage of these features.



**Figure 7.** The SEM image of GNRs and GNRs decoration with iron oxide nanoparticles: (a) GNRs (scale bar: 300 nm, 30 KX/10 KV); (b) FeNP-nanoflower on GNRs (scale bar: 300 nm, 30 KX/10 KV); and (c) FeNP-nanoflower on GNRs (scale bar: 300 nm, 40 KX/10 KV).

#### Cyclic Voltammogram

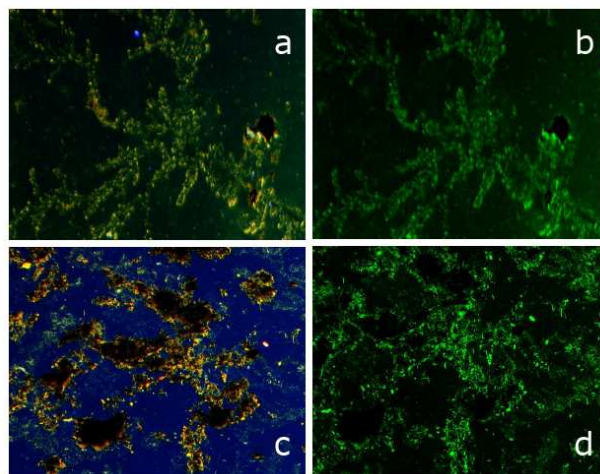
The cyclic voltammogram behavior of FeNP-nanoflower in chloroform (1 mg/ml) was studied (Figure 8). The redox couple in minus 2.0~1.0V regions is  $\text{Fe}^{3+}/\text{Fe}^{2+}$  of iron oxide, presumably the  $\gamma\text{-Fe}_2\text{O}_3/\text{Fe}_3\text{O}_4$ . When the sample of GNR/FeNP-nanoflower in chloroform (1mg/ml) was measured under the same experimental conditions, there is no such redox processing observed (Figure 8b), and the electrochemical behavior was different from that of graphene.<sup>33</sup> As an electronic scavenger, GNRs inhibit redox processing of FeNPs on their surface. FeNP-nanoplate showed the same electrochemical behavior as FeNP-nanoflower when decorated onto GNRs. Redox stability is crucial for GNR-FeNP biological applications as an effective biosensor.<sup>8</sup> The oxidation process leads to dysfunction FeNP-GNR binding with proteins and antibody, along with oxidation. Remarkably, it was found that the nanohybrid accomplishes redox stability, which is shown in cyclic voltammogram (CV) (Figures 9). The redox stability of FeNP-GNR in chloroform has an advantage to prepare a stable biosensor.



**Figure 8.** Cyclic voltammetry of FeNP-nanoflower and its attachment onto GNRs. (a) FeNPs-nanoflower, (b) GNR/FeNP-nanoflower. Scanning rate is 25.00 mV/s. working electrode is solid electrode with its area: 1.000 cm<sup>2</sup>. Reference electrode is the electrode is made of saturated Ag, AgCl/KCl (0.197 V).

#### Fluorescence Microscopy

To explore the feasibility of FeNP/GNRs as a biosensor, IgG antibodies were coated onto surfaces of nanocomposite. IgG antibodies have a high affinity for folic acid receptors.<sup>27</sup> Folic acid receptors have elevated cell membrane expression in aggressively growing cancer cells.<sup>27</sup> In fact, different types of cancer cells can be detected *via* folic acid receptor.<sup>28</sup> Utilizing the FeNP/GNR nanocomposite with IgG antibody allows us to create an effective sensor for the detection of metastasized cancer cell in the blood.<sup>29</sup> Figure 5 shows the fluorescence microscope images of GNR/FeNP-nanoplate and nanoflower with IgG antibodies attached and stained with green fluorescent protein (GFP). GFP staining readily indicates the location of IgG antibodies illuminated by green light. It is worth pointing out that the assembly is pivotal to stable biosensing applications.



**Figure 9.** Fluorescence microscopic images of GNRs/FeNP -IgG antibodies dyed with GFP protein illuminated by green fluorescent isothiocyanate (495nm): (a) and (b) for GNR/FeNP-nanoflower -IgG antibodies dyed with GFP protein; (c) and (d) for GNR / FeNP-nanoplate -IgG antibodies dyed with GFP protein.

GNR/FeNP-nanoplate-IgG was tagged with anti-goat GFP and fixed onto a glass slide. Figures 9a and 9c display microscope image with a compiled fluorescence illuminated by red, green, and blue light.

Fluorescence image further confirms that IgG mainly aggregates to the edges of GNR because IgG prefers attaching to oxidative defects. The GNR/FeNP-nanoflowers have fewer IgG antibodies and form well-organized “fern-leaves” (Figure 9b). The FeNP-nanoplate/GNR has abundant IgG antibodies on the edge (Figure 9c) due to its larger surfaces. With profound attachment of IgG, the GNR/FeNP-nanoplate has promising applications in metastasized cancer cell detection with over-expressed folic acid receptors. Further studies are warranted to explore this hybrid nanomaterial as a cancer diagnostics tool.

## Conclusions

Various sizes and shapes of FeNPs lead to different self-assembly motifs on the GNR surfaces, which is important for material application.<sup>37-39</sup> Due to the irregular shape at nanoscale, FeNP-nanoflower distribute mainly on the edges of GNRs. In contrast, FeNP-nanoplates attach to the surface of the GNR. A ramification of the distinctive patterns is that the FeNP-nanoflower has weaker interaction with GNR than the FeNP-nanoplate counterpart. 2D-band splitting of Raman spectra in 2600-3000 cm<sup>-1</sup> region demonstrates a multilayer sandwich nanostructure of GNR/FeNP-nanoplate. The extracted Raman spectra also show a red-shift in the G-band and apparent disappearance of the D-band, signifying that the resultant composite is electron doped, along with smooth composite surfaces. The self-assembly is particularly interesting in that GNR with *a priori* defects can retain the chemical properties of an oxidation-free ribbon. The redox stability shown in the cyclic voltammogram, (CV) permits the utilization of the nanohybrids for novel sensors. The present study also shows that IgG antibodies can be attached to GNR/FeNP nanocomposites, which can be tailored to detect cancer cells with over-expression of folic acids.

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## Notes and references

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1. M. Mahmoudi, S. Sant, B. Wang, S. Laurent and T. Sen, *Adv. Drug Deliv. Rev.*, 2011, **63**, 24-46.
2. G. Vallejo-Fernandez, O. Whear, A. G. Roca, S. Hussain, J. Timmis, V. Patel and K. O'Grady, *J. Phys. D: Appl. Phys.*, 2013, **46**, 312001.
3. A.-H. Lu, E.L. Salabas and F. Schueth, *Angew. Chem., Intl. Ed.*, 2007, **46**, 1222-1244.
4. A. Shavel and L. M. Liz-Marzan, *Shape Phys. Chem. Chem. Phys.*, 2009, **11**, 3762-3766.
5. S. Palchoudhury, Y. Xu, A. Rushdi and Y. Bao. *Chem. Commun.*, 2012, **48**, 10499–10501.
6. S. Palchoudhury, W. An, Y. Xu, Y. Qin, Z. Zhang, N. Chopra, R. A. Holler, C.H. Turner and Y. Bao, *Nano Lett.*, 2011 **11**, 1141-1146.

7. P. P. Karmali, Y. Chao, J.-H. Park, M. J. Sailor, E. Ruoslahti, S.C. Esener and D. Simberg, *Molec. Pharm.*, 2012, **9**, 539-545.
8. R. G. Mendes, A. Bachmatiuk, A. A. El-Gendy, S. Melkhanova, R. Klingeler, B. Buechner and M. H. Ruemmel, *J. Phys. Chem. C.*, 2012, **116**, 23749-23756.
9. H. Chen, X. Wu, H. Duan, A. Y. Wang, L. Wang, M. Zhang, and H. Mao, *ACS Adv. Mat. Interface*, 2009, **1**, 2134-2140.
10. A. K. Geim, *Science*, 2009, **324**, 1530-1534.
11. A. K. Geim and K. S. Novoselov, *Nat. Mater.*, 2007, **6**, 183-191.
12. Y. Wei, J. Wu, H. Yin, X. Shi, R. Yang, and M. Dresselhaus, *Nat. Mater.*, 2012, **11**, 759-763.
13. J. -U. Park, S.W. Nam, M. -S. Lee and C.M. Lieber, *Nat. Mater.* 2012, **11**,120-125.
14. Z. Liu, J. T. Robinson, X. Sun and H. Dai, *J. Am. Chem. Soc.*, 2008, **130**, 10876-10877.
15. L. Zhang, J. Xia, Q. Zhao, L. Liu and Z. Zhang, *Small*, 2010, **6**, 537–544.
16. W. Chen, P. Yi, Y. Zhang, L. Zhang, Z. Deng and Z. Zhang, *ACS Appl. Mater. Interfaces.*, 2011, **3**, 4085-4091.
17. G.-W. Zhou, J. Wang, P. Gao, X. Yang, Y.-S. He, X.-Z. Liao, J. Yang and Z.-F. Ma, *Indus. & Eng. Chem. Res.*, 2013, **52**, 1197-1204.
18. H. He and C. Gao, *ACS Appl. Mater. Interfaces.*, 2010, **2**, 3201-3210.
19. L. Ji, Z. Tan, T. R. Kuykendall, S. Aloni, S. Xun, E. Lin, V. Battaglia and Zhang, *Phys. Chem. Chem. Phys.*, 2011, **13**, 7170-7177.
20. D. Chen, W. Wei, R. Wang, J. Zhu and L. Guo, *New J. Chem.*, 2012, **36**, 1589-1595.
21. D. R. Reuven, H. Li, I. I. Harruna and X.-Q. Wang, *J. Mater. Chem.*, 2012, **22**, 15689-15694.
22. D. V. Kosynkin, A. L. Higginbotham, A. Sinitkii, J. R. Lomeda, A. Dimiev, B. K. Price and J.M. Tour, *Nature*, 2009, **458**, 872-876.
23. M. -Y. Liao, P.-S. Lai, H.-P. Yu, H. -P. Lin and C. -C. Huang, *Chem. Commun.*, 2012, **48**, 5319-532.
24. Y. Jin, S. Huang, M. Zhang, M. Jia, and D. Hu, *Appl. Surf. Sci.*, 2013, **268**, 541– 546.
25. A. G. Roca, J. F. Marco, M. D. P. Morales and C. J. Serna, *J. Phys. Chem. C*, 2007, **111**, 18577-18584.
26. I. Horcas, R. Fernández, J. M. Gómez-Rodríguez, J. Colchero, J. Gómez-Herrero and A. M. Baro *Rev. Sci. Instrum.*, 2007, **78**, 013705-013708.
27. I. G. Campbell, T. A. Jones, W. D. Foulkes and J. Trowsdale, *Cancer Res.*, 1991, **51**, 5329–5338.
28. M. Wu, W. Gunning and M. Ratnam *Cancer Epidemiology, Biomarkers & Prevention*, 1999, **8**, 775–783.
29. S. Hashira, S. Okitsu-Negishi and K. Yoshino, *Pediatr. Int.*, 2000, **42**, 337–342.
30. C. Fang, N. Bhattarai, C. Sun and M. Zhang, *Small*, 2009, **5**, 1637-1641.
31. L. M. Bronstein, X. Huang, J. Retrum, A. Schmucker, P. Maren, B. D. Stein and B. Dragnea, *Chem. Mater.*, 2007, **19**, 3624-3632.
32. M.-L. Chen, L.-M. Shen, S. Chen, H. Wang, X.-W. Chen and J.-H. Wang, *J. Mater. Chem., B* 2013, **1**, 2582-2589.
33. M. Pumera, *Chem. Soc. Rev.*, 2010, **39**, 4146-4157.
34. Y. Jin, S. Huang, M. Zhang, M. Jia, and D. Hu, *Appl. Surf. Sci.*, **2013**, **268**, 541-546.
35. J. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865-3868.

36. A. Tkatchenko and M. Scheffler, *Phys. Rev. Lett.*, 2009, **102**, 073005.
37. R. J. Fullerton, D. L. P. Cole, K. D. Behler, S. Das, F. Irin, D. Parviz, M. N. F. Hoque, Z. Fan and M. J. Green, *Carbon*, 2014, **72**, 192-199 .
38. T. Sharifi, E. Gracia-Espino, H. R. Barzegar, X. Jia, F. Nitze, G. Hu, P. Nordblad, C. W. Tai and T. Wågberg , *Nat. Commun.* 2013, **4**, 2319.
39. H. He and C. Gao, *ACS Appl. Mater. Interfaces.*, 2010, **2**, 3201-3210.