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1           **Vacuum-assisted layer-by-layer electrospun membranes: antibacterial and**  
2                               **antioxidative applications**

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23 **Abstract:** Layer-by-layer assembled films have been exploited for functional  
24 materials. Tannic acid with previously confirmed antibacterial and antioxidant  
25 potential was deposited on cellulose nanofibrous mats. LbL assembly technique  
26 allowed sufficient binding of TA and AgNPs-Lys to the supporting substrate via  
27 hydrogen bond and electrostatic interaction. The properties and morphology of the  
28 AgNPs-Lys/TA multilayer assembly membranes were characterized by X-ray  
29 photoelectron spectroscopy (XPS), Fourier transform infrared spectra (FT-IR),  
30 wide-angle X-ray diffraction (XRD), and scanning electron microscopy (SEM). The  
31 antibacterial and antioxidant activities were examined as well. The hybrid composite  
32 films have potential application in food packing and wound dressing, and tissue  
33 engineering, etc.

34 **Keywords:** Silver nanoparticles, Tannic acid, Antibacterial, Antioxidant,  
35 Electrospinning, Layer-by-layer

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## 45 **1. Introduction**

46 During the last years, materials based on one-dimensional nanostructures, such as  
47 nanofibers, nanotubes, nanowires, have created a subject of substantial interest due to  
48 their unique properties [1, 2]. And electrospinning is efficient and straightforward  
49 method of producing ultrafine fibers with micro- to nano-meter range diameters and  
50 with controlled surface morphology [3, 4]. Because of their high specific surface area,  
51 high porosity, small pore size and 3D structure, the nanofibrous materials may find  
52 diverse applications, for example, from electronics and military clothing to cosmetics,  
53 pharmacy and medicine [5-7].

54 In order to develop electrospun nanofibers as useful nanobiomaterials, surface of them  
55 have been functionalized by various surface modification technique [8], such as  
56 surface graft polymerization [9, 10], co-electrospinning [11], plasma treatment [12],  
57 wet chemical method [13]. A versatile surface modification method that allows  
58 surface coating with thickness from a few nano to several micrometers through  
59 precise control has been realized by layer-by-layer (LbL) polyelectrolyte multilayer  
60 assembly [14,15].

61 Layer-by-layer multilayer membrane fabrication is an attractive fabrication strategy  
62 because of the large variety of charged polymers that can be utilized in LbL assembly.  
63 Additionally, the membrane structure can be readily tailored by altering the polymer  
64 deposition conditions. This technique has been widely used to fabricate thin films  
65 from polymer pairs with complementary functional groups because of its advantages  
66 over other methods [16].

67 Lysozyme (Lys) is a ubiquitous antibacterial enzyme against many food spoilage and  
68 pathogenic microorganisms by damaging the cell walls of bacteria [17, 18]. It consists  
69 of 129 amino acid residues with free carboxylic groups, amino groups and four  
70 disulfide bonds, and has been used to prepare Au nanoparticles (AuNPs), Ag  
71 nanoparticles (AgNPs), Au nanoclusters (AuNCs) and Ag nanoclusters (AgNCs)  
72 [19-21]. The AuNCs-Lys can target notorious pathogenic bacteria, including *E. coli*  
73 and *S. aureus* [22]. Moreover, tannic acid (TA) is a glucoside of gallic acid polymer  
74 with multiple phenolic hydroxyl groups that is found in many plants [23]. It is an  
75 attractive molecule known to have antitumor, antibacterial, and antioxidant activity  
76 [16], as well as reported interactions with proteins [24]. Because of the high pKa  
77 value of TA of ca. 8.5, its association through hydrogen bonding is expected to occur  
78 at neutral pH values [25]. So it has recently been incorporated in hydrogen-bonded  
79 LbL films at physiologic pH [24].

80 In the current study, the LbL films were fabricated from TA and AgNPs-Lys. LbL  
81 assembly technique allowed sufficient binding of TA and AgNPs-Lys to the  
82 supporting substrate via hydrogen bond and electrostatic interaction. The properties  
83 and morphology of the TA/AgNPs-Lys multilayer assembly membranes were  
84 characterized and the antibacterial and antioxidant activities were examined. The  
85 hybrid composite films have potential application in food packing and wound  
86 dressing.

## 87 **Experimental**

### 88 **Chemicals and materials**

89 Cellulose acetate (CA, *Mn* 30,000) was purchased from Sigma–Aldrich Co., USA.  
90 Hen egg white lysozyme and tannic acid were all obtained from Sinopharm Chemical  
91 Reagents Co., Ltd. (Shanghai, China). The other reagents were analytical grade  
92 purchased from China National Pharmaceutical Group Industry Corporation Ltd.. All  
93 aqueous solutions were prepared using purified water with a resistance of 18.2 MΩcm.  
94 *Escherichia.coli* (*E.coli*) and *Staphylococcus aureus* (*S.aureus*) were obtained from  
95 China Center for Type Culture 140 Collection, Wuhan University (Wuhan, China).

#### 96 **Synthesis of AgNPs-Lys**

97 AgNPs-Lys were prepared using a procedure modified from that used by Mathew et al  
98 and Zhou [19, 26]. In a typical synthesis, 1 mL of 10 mM silver nitrate solution was  
99 added to 75 mg Lys powder in 5 mL distilled water solution with vigorous stirring at  
100 room temperature. The mixture solution was left to incubate for 5 minutes under  
101 vigorous stirring. Then about 0.3mL NaOH solution (1 M) was added followed by  
102 0.48 mL NaBH<sub>4</sub> solution (10 mM) drop-wise until the solution turns from colorless to  
103 reddish brown, indicating the formation of silver nanoparticles. The product of  
104 different stages were characterized by ultraviolet-visible spectrum (UV-vis) and FT-IR.  
105 The hydrodynamic diameter measured using dynamic light scattering (DLS).

#### 106 **Fabrication of template nanofibers**

107 The CA electrospun nanofibrous membranes were fabricated using a set of homemade  
108 electrospinning setup, which contained a high voltage supply (DW-P303-1ACD8,  
109 Tianjin Dongwen Co., China), a syringe pump (LSP02-1B, Baoding Longer Precision  
110 Pump Co., Ltd., China) and a grounded rotary collector. Nanofibrous CA mats were

111 fabricated by modified Ding's method [27]. 2 g CA was dissolved into 8 g acetone/N,  
112 N-dimethyl acetamide (DMAc) (2:1, W/W) mixed solvent and stirred to obtain  
113 homogeneous solution. Then it was loaded into a plastic syringe, which was driven by  
114 a syringe pump. The applied voltage was 16 kV and the tip-to-collector distance was  
115 20 cm. The ambient temperature and relative humidity were maintained at 25°C and  
116 45%, respectively. The prepared fibrous mats were dried at 80°C in vacuum for 24 h  
117 to remove the trace solvent. Hydrolysis of the CA mats was performed in alkaline  
118 aqueous solution at ambient temperature for 7 days following the previous report [28].

#### 119 **Formation of nanocomposite films on template nanofibers**

120 The bilayer film was then deposited, by adding AgNPs@ Lys (1 mg/mL lysozyme, pH  
121 7.4, in 0.01 M PBS) followed by tannic acid (1 mg/mL, pH 7.4, in 0.01 M PBS) each  
122 for 50mL. Then, the solution was suction-filtered through the nanofibrous mats.  
123 Following each deposition step, the mats wash with 50 mL 0.01 M PBS (pH 7.4) [24,  
124 29]. The water was suction-filtered through the nanofibrous mats as well. Here,  
125 (AgNPs-Lys/TA)<sub>n</sub> was used as a formula to label the LbL structured films, where n  
126 was the number of the AgNPs-Lys/TA bilayers. The outermost layer was Lys  
127 composite when n equaled to 5.5 and 10.5. The LbL films coated fibrous mats were  
128 dried at 40 °C for 2 h under vacuum prior to further characterizations.

#### 129 **Characterization**

130 The morphology characterization of the composite membranes was performed using  
131 scanning electron microscopy (SEM) (S-4800, Hitachi Ltd., Japan). The diameters of  
132 the fibers were measured using Nano measure 1.2.5. Fourier transform infrared

133 (FT-IR) spectra were acquired on a Nicolet170-SX instrument (Thermo Nicolet Ltd.,  
134 USA) in the wavenumber range of 4000-400  $\text{cm}^{-1}$ . X-ray photoelectron spectroscopy  
135 (XPS) was conducted on an axis ultra DLD apparatus (Kratos, U.K.). X-ray  
136 diffraction (XRD) was carried out using a diffract meter type D/max-rA (Rigaku Co.,  
137 Japan) with Cu target and Ka radiation ( $\lambda= 0.154 \text{ nm}$ ).

#### 138 **In vitro antibacterial activity assay**

139 The inhibition zone test was used to study the bacterial inhibition activity of  
140 nanofibrous mats. Gram-negative *E. coli* and gram-positive *S. aureus* were selected as  
141 representative microorganism and cultivated in culture medium in an incubator.  
142 Unmodified cellulose mats were used as negative control. The testing mats were cut  
143 into round disks with a diameter of 6 mm, sterilized under an ultraviolet radiation  
144 lamp for 30 min. One hundred micro-liters of  $5.0\text{-}10.0\times 10^5$  cfu/mL *E. coli* or  
145  $5.0\text{-}10.0\times 10^5$  cfu/mL *S. aureus* bacteria levitation liquid was placed onto  
146 pre-autoclave sterilized meat-peptone broth and coated uniformly, respectively. Then  
147 the prepared mats were tiled on the surface of meat-peptone broth to cling to the  
148 bacteria levitation liquid. After incubated at 37 °C in an air-bathing thermostat shaker  
149 with a rotating speed of 120 r/min for 24h, the bacteria inhibition zones were  
150 measured by a micrometer with a tolerance of one millimeter. All of the experiments  
151 were conducted in triplicate with data reported as mean  $\pm$  standard deviation.

#### 152 **In vitro antioxidant activity assay**

153 The antioxidant activity of nanocomposite films were measured according to the  
154 DPPH method with minor modification [30, 31]. Briefly, scavenging activity assay



155 was carried out by recording the absorbance of DPPH solution (100  $\mu\text{M}$ ) at 517 nm in  
156 the presence of the nanofibrous mats above at room temperature with a UV-vis  
157 spectrophotometer. The free radical scavenging potency of the nanofibrous mats were  
158 expressed as the percentage of DPPH that was decreased in comparison with that of  
159 the control condition after 30 min preservation in the dark.

## 160 **Results and discussion**

### 161 **Preparation and characterization of AgNPs-Lys**

162 The as-prepared AgNPs-Lys was characterized by UV-visible absorption as indicated  
163 in Fig. 1A. It can be observed that the UV-visible absorption spectrum of AgNPs-Lys  
164 exhibited a peak at 420 nm due to surface plasmon resonance of Ag nanoparticles.  
165 And the hydrodynamic diameter of AgNPs-Lys measured using DLS was  
166  $6.76 \pm 0.44 \text{ nm}$ . In addition, the surface chemistry of AgNPs-Lys was evaluated using  
167 FT-IR. The FT-IR spectra as shown in Fig. 1B showed that there was no S-H  
168 stretching band for Lys only contains four disulfide bonds without a free hydrosulfide  
169 group. However, an S-H stretching band around  $2485 \text{ cm}^{-1}$  appeared after the natural  
170 Lys was incubated in a solution of pH about 12, as the alkali could cleave the  
171 disulfide bonds of Lys [19]. The peak at  $2485 \text{ cm}^{-1}$  almost disappeared after the  
172 formation of AgNPs-Lys, indicating that Lys was modified on the surface of AgNPs  
173 through Ag-S interactions.

### 174 **Surface morphology analysis AgNPs-Lys/TA nanofibrous membranes**

175 In order to investigate the effect of AgNPs-Lys and TA deposition on the morphology  
176 of the cellulose nanofibrous mats, the SEM images of the composite fibrous mats were

177 taken. The representative scanning electron microscopy (SEM) image of nanofibrous  
178 mats shown in Fig. 2a revealed randomly oriented 3D nonwoven membranes with an  
179 average diameter of 600 nm. And the cellulose nanofibers exhibited cylindrical shape  
180 and was continuous and long without any defects (Fig. 2a and a'). To study the impact  
181 of the number of coating bilayers on the formation of composite films, the cellulose  
182 fibers were coated with various bilayers of AgNPs-Lys and TA.

183 Not only the diameter but also the morphology of all composite mats changed  
184 obviously caused by the deposited of AgNPs-Lys and TA on the surface of nanofibers.

185 After LbL coating process, the nanofibers showed a much higher surface roughness  
186 on each fiber compared with the smooth surface of the cellulose (Fig. 2a-e, a'-e').

187 There are many granules on the surface of the fibers, which was attributed to the  
188 interaction between Lys and TA. Moreover, with the increase of the bilayer number,

189 some junctions among the cellulose fibers can be seen, which caused by the  
190 aggregation of AgNPs-Lys and TA. As displayed in Fig. 2d', we can see that after the

191 coating, a AgNPs-Lys/TA shell layer was visible around the cellulose nanofibers.

192 These images visually confirmed that AgNPs-Lys and TA were successfully  
193 assembled onto the surface of the fibers.

#### 194 **Surface composition analysis of AgNP-Lys/TA nanofibrous mats**

195 To further confirm the depositon of AgNP@Lys and TA on the LbL films, XPS scan  
196 were performed to verify the surface chemical composition of the composite

197 nanofibrous mats. Fig. 3A displays the survey scan spectrum of AgNPs-Lys/TA

198 composite mat, in which C 1s, O 1s, N 1s, S 2p, and Ag 3d core-levels exist obviously.

199 As shown in Figure 3B, the C 1s core-level photoelectron spectrum can be curved into  
200 three peak components located at 284.6 eV, 286.0 eV, and 288.2 eV, which are  
201 assigned to C-C, C-O, and C=O or O-C=O group from TA or Lys [32]. Moreover, the  
202 N 1s spectrum had one peak centered at 399.8 eV which was characteristic of  
203 pyridinic nitrogen ( $sp^2$  hybridization), and it was associated with the assignment of the  
204 binding energy of C-N covalent bonds [33]. As all of cellulose, Lys and TA contained  
205 C and O, the presence of them cannot certify that TA was deposited on the surface of  
206 the fibers successfully. To further demonstrate LbL coating process in every layer, the  
207 ratio of C/O was measured (Table 1). As is well-known, TA was rich in oxygen,  
208 43.27%, much higher than in Lys. From the obtained atomic concentration of the high  
209 resolution scans (C 1s and O 1s), the C/O ratio of  $(AgNP@Lys/TA)_{10}$  and  
210  $(AgNP@Lys/TA)_{10.5}$  were obtained to be 2.22 and 3.02, respectively. In addition, the  
211 S 2p signal at *ca.* 163 eV (Fig. 3E) implied the presence of sulfur species on the  
212 surface of composite nanofibers. As shown in Fig. 2F, two peaks at 368.2 eV and  
213 374.2 eV were observed in the Ag 3d XPS spectra of  $(AgNP@Lys/TA)_{10}$  and  
214  $(AgNP@Lys/TA)_{10.5}$  nanofibrous mats corresponding to Ag 3d<sub>5/2</sub> and Ag 3d<sub>3/2</sub>,  
215 respectively. These results are in good agreement with the results of Zhou et al. [19].  
216 The presence of signal at 368.2 eV revealed that Ag<sup>+</sup> exist in the complex [34]. The  
217 Moreover, with the increase of particle size, the binding energy would have a slight  
218 blue shift to about 367.7 eV [34]. For the core-level spectra of  $(AgNP@Lys/TA)_{10}$   
219 and  $(AgNP@Lys/TA)_{10.5}$ , the intensity of the peaks have significant difference, which  
220 was attributed to the difference of elemental composition.

**221 FT-IR spectra of AgNPs-Lys/TA nanofibrous mats**

222 The successful assembly of the AgNPs-Lys and TA to cellulose nanofibers can also be  
223 illustrated by the FT-IR spectra. As shown in Figure 4, the broad band in the region of  
224 about 3500~3100  $\text{cm}^{-1}$  was attributed to the free O-H stretching vibration of hydroxyl  
225 groups in cellulose molecules (Fig. 4g) [35]. The absorption bands at 2898  $\text{cm}^{-1}$  was  
226 assigned to the C-H stretching. The emergence of a peak at 1637 $\text{cm}^{-1}$  corresponding  
227 to -OH bending of the nanofibrous mats [36]. The band at 1066 $\text{cm}^{-1}$  was assigned to  
228 the stretching of C-O, asymmetric stretching of C-O-C bond of the glycosidic linkage  
229 and pyranose ring of cellulose were observed at around 1238, 1164 and 1052 $\text{cm}^{-1}$ ,  
230 respectively. And the 897 $\text{cm}^{-1}$  was attributed to the C<sub>1</sub>-H deformation vibrations of  
231 cellulose [37]. The two common bands of amino group observed at 1654  $\text{cm}^{-1}$  and  
232 1540  $\text{cm}^{-1}$  belongs to the amide I and amide II peaks, which confirmed that Lys was  
233 assembly on the surface of cellulose films successfully (Fig. 4a, c-f). For the  
234 composite nanofibrous mats, the observed increase in intensity of the peak positioned  
235 at 1716  $\text{cm}^{-1}$  as a shoulder of the amide I or -OH bending can be due to the carbonyl  
236 CO vibration of the TA ester bond (Fig. 4c-f). The 1616  $\text{cm}^{-1}$  and 1533  $\text{cm}^{-1}$  are  
237 attributed to C=C stretching vibrations of aromatic ring and carbon chain, respectively.  
238 The peak at 1448  $\text{cm}^{-1}$  is associated with C-O-H in plane bend of hydroxyl group in  
239 TA. The band at 1323  $\text{cm}^{-1}$  and 1203  $\text{cm}^{-1}$  can be attributed to C-O stretch of the acid  
240 group in TA and C-O stretch in polyols, respectively (Fig. 4b) [38]. The absorption  
241 band centered at 1032  $\text{cm}^{-1}$  is associated with the C-O-C bending mode [39]. The  
242 weak absorption band at 876  $\text{cm}^{-1}$  is assigned to O-H out of plane bending mode of

243 the acid group. The band at  $758\text{ cm}^{-1}$  can be related to the C-H out plane bend of  
244 phenyl group (Fig. 4b) [39].

#### 245 **Crystalline property of AgNPs-Lys/TA nanofibrous mats**

246 The XRD patterns of samples are presented in Figure 5. From Figure 5a, we can see a  
247 broad peak ranged from about  $15^\circ$  to  $32^\circ$  corresponding to amorphous region of TA.

248 The diffractogram of cellulose nanofibers consisted of a peak at  $12.2$ ,  $20.1$ , and  $21.8^\circ$   
249 corresponding to typical cellulose crystal. After the coating, the peak at  $21.8^\circ$   
250 corresponding to cellulose crystal increased in both Lys/TA and AgNPs-Lys/TA  
251 nanofibrous mats, which can be attributed to the deposition of TA. In addition,  
252 (AgNPs-Lys/TA)<sub>10.5</sub> had a well defined characteristic diffraction peak at  $38.5^\circ$   
253 corresponding to (111) plane of face centered cubic crystal structure of silver  
254 revealing the presence of silver nanoparticles [40].

#### 255 **Free radical scavenging activity of AgNPs-Lys/TA nanofibrous mats using DPPH**

256 Radical scavenging activities are very important due to the deleterious role of free  
257 radicals in foods and in biological systems [41]. To confirm the bioactivity, the DPPH  
258 scavenging assay was employed to determine the radical-scavenging ability of LbL  
259 coating nanofibrous mats (Fig. 6). The method can evaluate the antiradical power of  
260 an antioxidant by measuring of a decrease in the absorbance of DPPH $\cdot$  at  $517\text{ nm}$ ,  
261 which was accompanied by a colour change from purple to yellow. The deposition of  
262 AgNPs-Lys and TA on the fibers enhanced the antioxidant activity of composite  
263 nanofibrous mats compared to the cellulose nanofibrous mats. The antioxidant  
264 capacity of these membranes increased with the increase of the number of bilayer.

265 The scavenging rate of (AgNPs-Lys/TA)<sub>5</sub> and (AgNPs-Lys/TA)<sub>10</sub> were 70% and 82%,  
266 respectively. However, the antioxidant capacity decreased dramatically compared to  
267 (AgNPs-Lys/TA)<sub>5</sub> and (AgNPs-Lys/TA)<sub>10</sub> when the outmost component was  
268 AgNPs-Lys ((AgNPs-Lys/TA)<sub>5.5</sub> and (AgNPs-Lys/TA)<sub>10.5</sub>). Since the presence of  
269 AgNPs-Lys on the outmost layer has a certain stereo-hindrance effect caused the  
270 decrease of the antioxidant capacity. We also investigated the relationship between  
271 scavenging rate and reaction time. The ratio of C<sub>t</sub> to C<sub>0</sub> were obtained from the  
272 relative intensity ratios of the respective absorbance (A<sub>t</sub>/A<sub>0</sub>) at 517 nm. The rate  
273 constants *k* estimated directly from the slopes were 0.107min<sup>-1</sup>, 0.044min<sup>-1</sup>,  
274 0.244min<sup>-1</sup>, and 0.094min<sup>-1</sup> for (AgNPs-Lys/TA)<sub>5</sub>, (AgNPs-Lys/TA)<sub>5.5</sub>,  
275 (AgNPs-Lys/TA)<sub>10</sub>, (AgNPs-Lys/TA)<sub>10.5</sub>, respectively (Fig. 6B). The result was  
276 consistent with the result in Fig. 6A. As can be seen, the composite nanofibrous mats  
277 exhibited a good antioxidant performance.

#### 278 **Antibacterial property of AgNPs-Lys/TA nanofibrous mats**

279 Both Lys and TA are ubiquitous antibacterial agent against many food spoilage and  
280 pathogenic microorganisms [8, 42]. We further examined the antibacterial activity of  
281 the cellulose nanofibrous mats and LbL coating mats against the Gram-negative  
282 bacteria (*E. coli*) and the Gram-positive bacteria (*S. aureus*). The antibacterial  
283 bioactivity of cellulose nanofibers and composite nanofibers with different bilayer  
284 numbers was also evaluated for comparison (Fig. 7). Obviously, as-prepared cellulose  
285 mats hardly displayed bacterial inhibition zones at all the studied time points. In  
286 contrast, an obvious bacterial inhibition zones on the composite nanofibrous mats can

287 be seen. For *S. aureus*, the antibacterial effect enhanced with the increase of the  
288 number of bilayer, which due to the fact that both TA and Lys have good antibacterial  
289 activity against *S. aureus*. However, Lys has a relatively weak antibacterial activity for  
290 Gram-negative bacillus because of the protection of lipopolysaccharide layer  
291 surrounding their outmost membrane [8]. So the composite nanofibrous mats with  
292 AgNPs-Lys on the outmost layer exhibit weaker antibacterial effect against *E. coli*.

### 293 **Conclusions**

294 AgNPs-Lys/TA multilayer nanofibrous mats were fabricated using electrospinning  
295 and electrostatic LbL assembly technique. The films were formulated based on  
296 interactions between Lys and TA at physiologic pH. The deposition of AgNPs-Lys and  
297 TA on the surface of cellulose mats was characterized by XPS, XRD, and FT-IR.  
298 Compared with the smooth surface of the cellulose, the morphology of composite  
299 nanofibrous mats became highly rough with increasing deposition layer. Moreover, all  
300 the composite nanofibrous mats examined were found to possess good  
301 DPPH-scavenging activity. Besides, the microbial inhibition assay demonstrated that  
302 the AgNPs-Lys/TA composite mats had good antibacterial effects. The antioxidant  
303 activity and antibacterials activity of the composite nanofibrous mats endows the  
304 materials with great potential application in the areas of food packing, tissue  
305 engineering, wound dressing, etc.

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419 **Figure captions:**

420 **Figure 1.** UV-vis spectra of Lys, Lys+OH<sup>-</sup>, and AgNPs-Lys (A); FT-IR spectra of Lys  
421 (a), Lys+OH<sup>-</sup> (b), and AgNPs-Lys(c) (B).

422 **Figure 2.** SEM images of (a-e): cellulose nanofibrous mats, (AgNPs-Lys/TA)<sub>5</sub>,  
423 (AgNPs-Lys/TA)<sub>5.5</sub>, (AgNPs-Lys/TA)<sub>10</sub>, and (AgNPs-Lys/TA)<sub>10.5</sub>. Image (a'-e')  
424 showed high magnification images of a-e, respectively. The right column reveals the  
425 diameter distribution histograms of the nanofibrous mats.

426 **Figure 3.** XPS survey spectra of (AgNPs-Lys/TA)<sub>10</sub> (A-a) and (AgNPs-Lys/TA)<sub>10.5</sub>  
427 (A-b); (B-F) core-level spectra of C 1s, O 1s, N 1s, S 2p, and Ag 3d (left:  
428 (AgNPs-Lys/TA)<sub>10</sub>, right: (AgNPs-Lys/TA)<sub>10.5</sub>)

429 **Figure 4.** FT-IR spectra of (a-g): Lys, TA, (AgNPs-Lys/TA)<sub>5</sub>, (AgNPs-Lys/TA)<sub>5.5</sub>,  
430 (AgNPs-Lys/TA)<sub>10</sub>, (AgNPs-Lys/TA)<sub>10.5</sub> and cellulose nanofibrous mats.

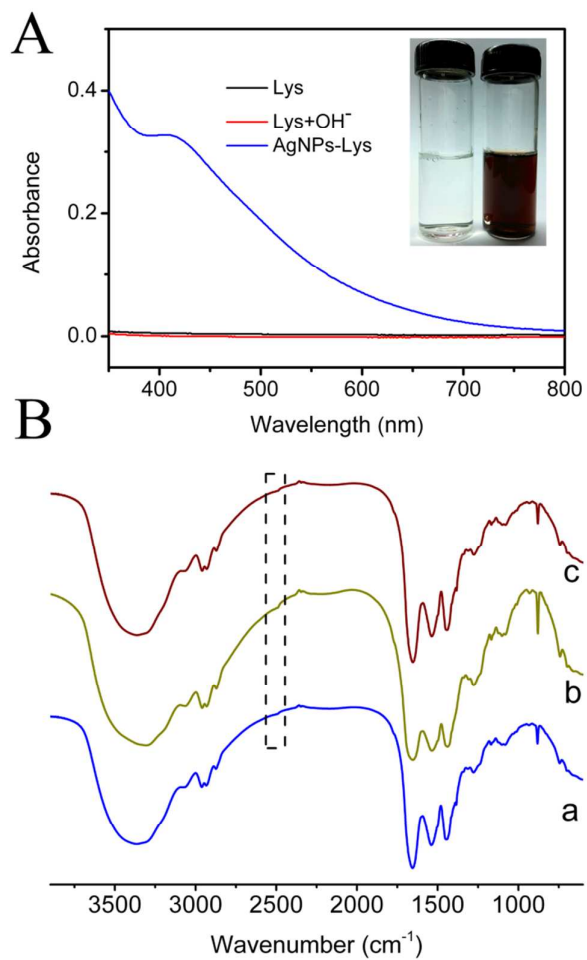
431 **Figure 5 .** XRD patterns of TA, cellulose nanofibrous mat, (Lys/TA)<sub>10</sub>,  
432 (AgNPs-Lys/TA)<sub>10</sub> (a-d).

433 **Figure 6.** Radical scavenging activities of cellulose nanofibrous mats and composite  
434 nanofibrous mats (A); plots of  $\ln(C_t/C_0)$  versus reaction time for 5 bilayer, 5.5bilayer,  
435 10bilayer, and 10.5 bilayer.  $C_t$  and  $C_0$  are the concentrations of DPPH<sup>·</sup> at the  
436 beginning and at time t, respectively (B).

437 **Figure 7.** Antimicrobial activities against *E. coli* and *S. aureus* of fibrous cellulose  
438 mats (control) and composite nanofibrous mats, error bars represent standard  
439 deviation (SD) for  $n=3$ .

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441 Figure 1



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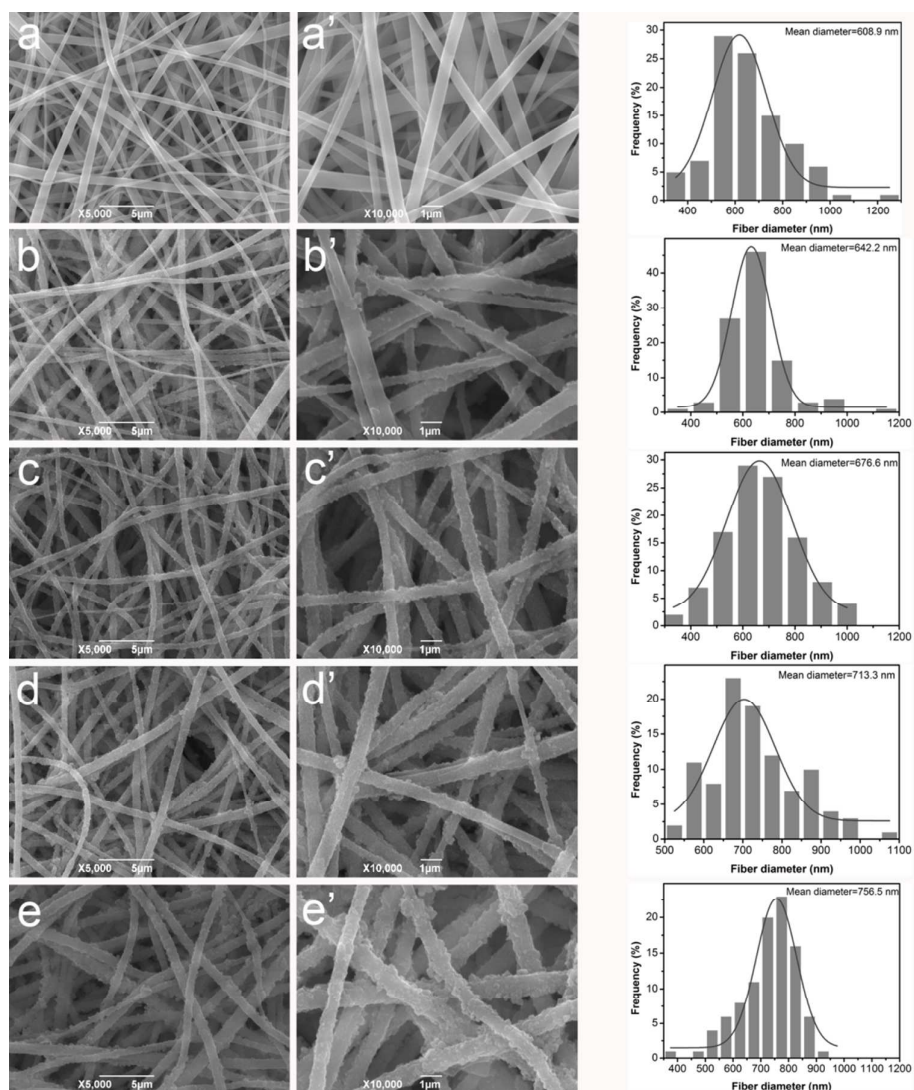
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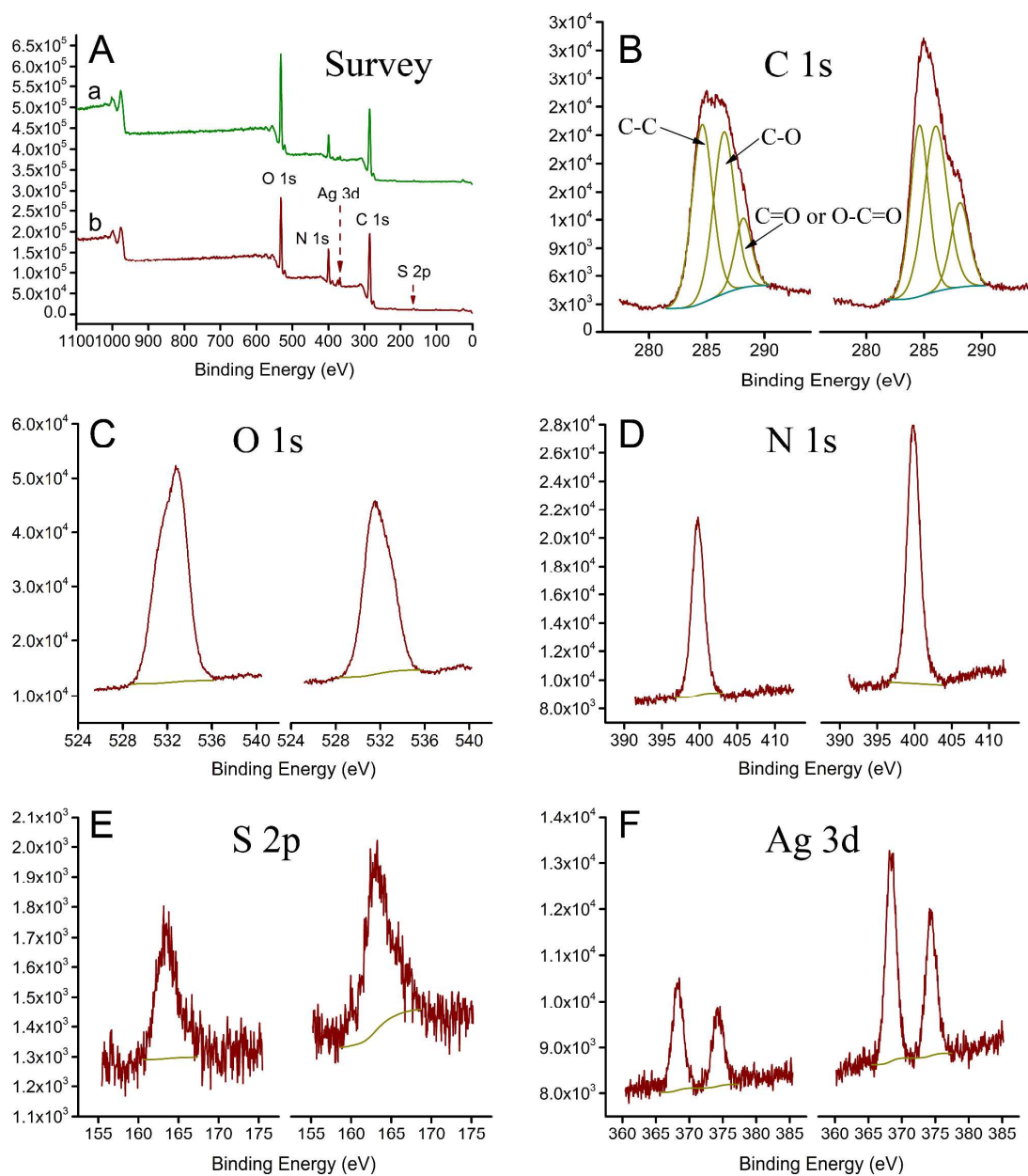
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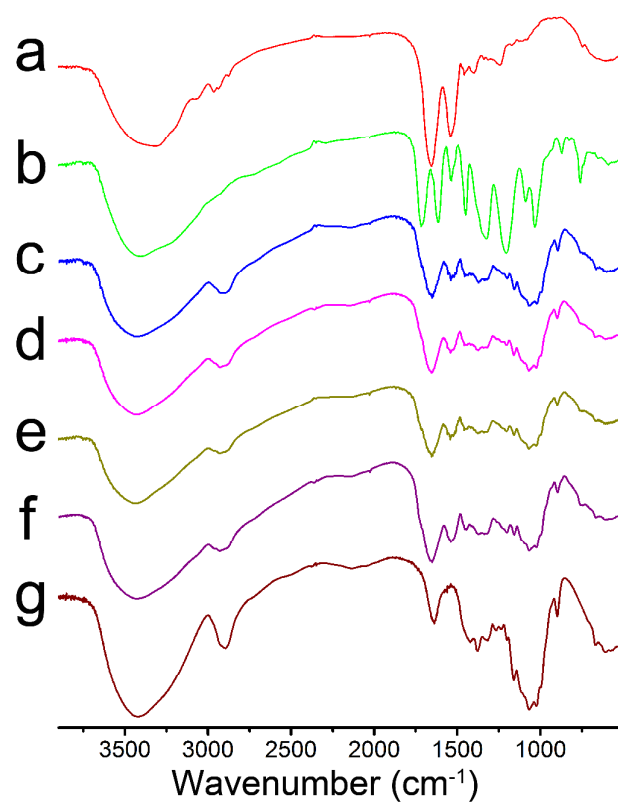
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468 Figure 4



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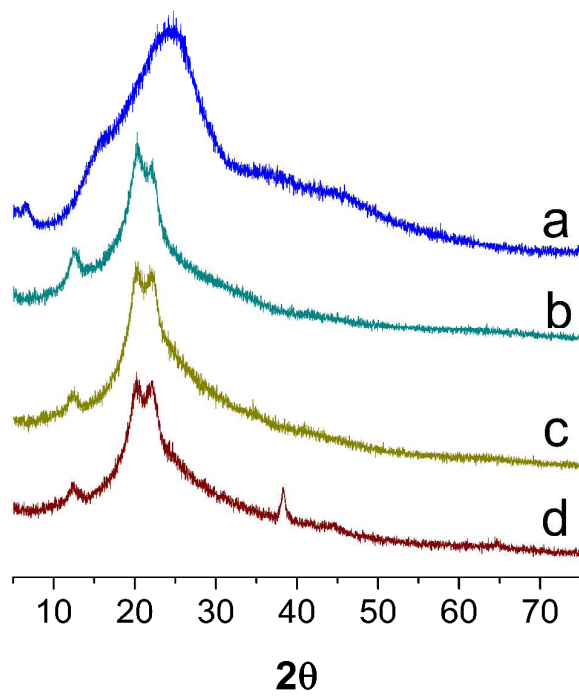
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481 Figure 5



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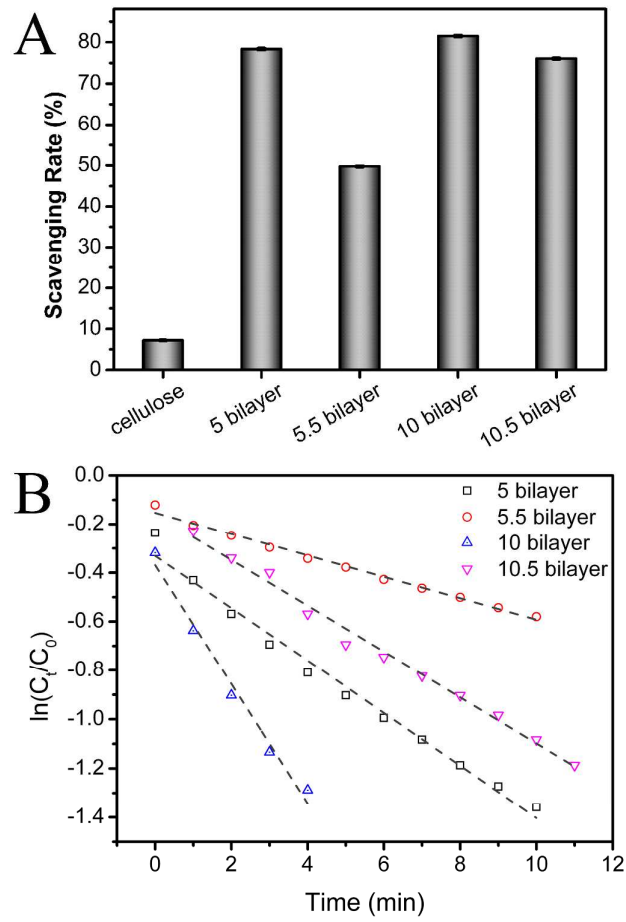
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495 Figure 6



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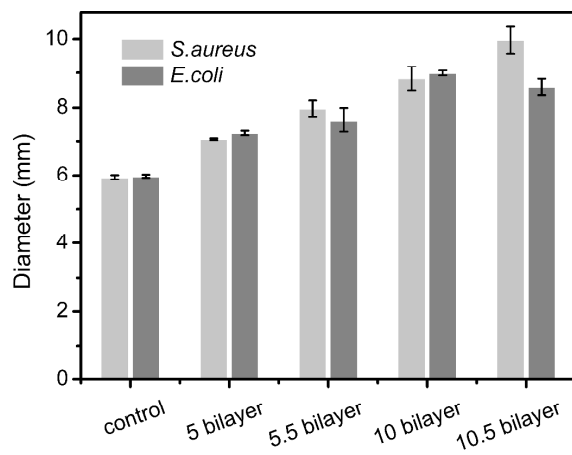
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506 Figure 7



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523 **Table 1.** Element composition and content on the surface of cellulose mats and

524 (AgNPs-Lys/TA)<sub>10</sub>, and (AgNPs-Lys/TA)<sub>10.5</sub>

Nanofibrous mats	C	O	N	S	Ag
cellulose	45.99	54.01	--	--	--
10 bilayer	62.06	27.91	9.2	0.52	0.31
10.5 bilayer	63.24	20.97	17.37	0.87	0.54

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