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## ARTICLE

## 3D controllable preparation of composite CuO/TiO<sub>2</sub> nanofibers

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The preparation and research of nanometer metal oxide has been drawn much attention for the special structure and excellent chemical properties especially titania and titanium dioxide composite nanomaterials. A three-dimensional (3D) multiple-unit semiconductor CuO/TiO<sub>2</sub> film was successfully synthesized by the combination of electrospinning and hydrothermal reaction. The samples were characterized by Differential Thermal Analyzer (TG-DTA) X-ray Diffraction (XRD), Transmission Electron Microscopy (TEM), Scanning Electron Microscope (SEM) which indicated the diameter of the TiO<sub>2</sub> fibers were about < 100 nm; the CuO nanoparticles or nanoplates were growing aslant on the surface of TiO<sub>2</sub> fibers, which led to the porous morphology for the hierarchical heterostructure. The photocatalytic property of the samples was tested via rhodamine B degradation and it showed the obvious catalytic activity for the Simulation of pollutants. Furthermore, the photocatalytic mechanism about the CuO/TiO<sub>2</sub> nanofiber was expounded at the end of paper.

### Introduction

It was in 1972 since the scientist of Japan fujishima discovered the performance of TiO<sub>2</sub> in photocatalytic hydrogen production,<sup>1</sup> it has attracted lots of attention of scientific researchers in the world widely; so the TiO<sub>2</sub> has been applied in a variety of fields from now on, such as photocatalytic degradation,<sup>2</sup> the solar cell,<sup>3</sup> gas sensor,<sup>4</sup> biosensor<sup>5</sup> and other fields in industrial catalysis. As a result of TiO<sub>2</sub> in the photocatalytic degradation of organic pollutants<sup>6</sup> in the process of high activity, non-toxic, chemically stability, refractory, long-term light corrosion and chemical corrosion resistance,<sup>7-9</sup> it shows bright prospect in photocatalytic aspects. Although the good prospect of application in photocatalysis, there are also some defects in the quantum efficiency<sup>10</sup>, Sunlight utilization,<sup>11,12</sup> recycling and photon-generated carrier problems. At the same time, the TiO<sub>2</sub> structure, size and composition will play the great role in its photocatalytic performance.<sup>13-19</sup> So in order to make full use of green pollution-free solar energy and improve the photocatalytic activity of TiO<sub>2</sub>, it usually need some decorations, such as the surface of the precious metal deposition, ion doping, dye sensitizing, coupled semiconductor. Coupled Semiconductor is a kind of effective method to improve the property of TiO<sub>2</sub> in photocatalysis.<sup>20</sup> It is universally acknowledged that the CuO in the nanoscale will appear great catalytic activity<sup>21</sup> for the low band gap. Because of the nanostructure and physical property of CuO, it is also widely used as gas sensor<sup>22,23</sup> and antibacterial materials.<sup>24</sup>

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Based on the methods of predecessor,<sup>25</sup> we have firstly synthesized the 3-D composite nanometer CuO/TiO<sub>2</sub> hierarchical heterostructure which is nanoporous on the surface of the fibers and easy to be recycled with combination of electrospinning technique and hydrothermal synthesis method. Before the thesis writing, we have done a lot of work in controlling the morphology of the CuO doping on the TiO<sub>2</sub> fibers surface by adjusting the reaction temperature, alkali source, copper ion concentration and other experiment parameters. The main factors which affect the morphology of the CuO/TiO<sub>2</sub> films were alkali source and the reaction temperature. In the process of photocatalytic degradation for rhodamine B (RhB) which is typical model organic pollutants, the as-prepared showed an obvious degradation effect for it. So it is beneficial exploration for the S-S type semiconductor compounds.

### Experimental section

#### Materials and methods

First, the TiO<sub>2</sub> sol was prepared by 5.00 g of tetrabutyl titanate (Ti(OBu)<sub>4</sub> 97 %, Sigma-Aldrich) with a mixture of 23mL of methyl alcohol (99.5 % Beijing chemical works) and 0.3 mL of acetic acid (99.5 % Beijing chemical works). Next, 1.40 g Polyvinylpyrrolidone (PVP Mw~1 300 000, Sigma-Aldrich) was added to the TiO<sub>2</sub> sol solution. The precursor mixture was stirred for 12 hours at room temperature to attain Uniform, transparent, light green solution required for electrospinning and then the solution was loaded into a syringe. The feeding rate of the solution in the syringe was controlled at 1.0 mL h<sup>-1</sup>. The voltage applied to the needle of the syringe was 12 kV and the distance from the tip of the needle and the aluminum foil collector was 13~15 cm at a fume hood in air with the

humidity of below the 40 %. After the obtained non-woven film was calcined at 520 °C in 4 hours. Then the electro-spinning TiO<sub>2</sub> fibrous film was put into an autoclave containing Cu(Ac)<sub>2</sub> – C<sub>6</sub>H<sub>12</sub>N<sub>4</sub> – H<sub>2</sub>O (CT1 sample), Cu(Ac)<sub>2</sub> – NH<sub>3</sub>•H<sub>2</sub>O – H<sub>2</sub>O (CT2 sample) solution respectively and the concentrations of Cu<sup>2+</sup>, C<sub>6</sub>H<sub>12</sub>N<sub>4</sub> and NH<sub>3</sub>•H<sub>2</sub>O were 0.1 M, 1.2 M and 13.33 M (Room temperature about 25 wt %). The hydrothermal reaction was carried out at 120 °C for 10 hours. The as-prepared products would be easily collected and then washed with deionized water and ethanol absolute at last.

### Characterization

The X-ray diffraction (XRD) patterns of the samples were measured on D/MAX 2250 V diffractometer (Rigaku Japan), using Cu K $\alpha$  ( $\lambda=0.15418$  nm) radiation under 40 kV, 30 mA and scanning over the range of 20°–80°. The morphologies and microstructures of as-prepared samples were characterized by the Scanning Electron Microscope (SEM, SHIMADZU X-550) and High Resolution Transmission Electron Microscopy (HRTEM, Tecnai G2). UV-vis absorption spectra were recorded on an UV-vis spectrophotometer (Hitachi U-3010). The as-spun PVP/TiO<sub>2</sub> composite nanofibers were subjected to Thermogravimetric Differential Thermal Analysis (TG-DTA, HCT-3).

### Photocatalytic experiments

Photocatalytic performance of the as-prepared samples was evaluated by the degradation of rhodamine B. The photo-catalytic degradation of RhB in the liquid phase was conducted in an ordinary glass of 250 ml with the condensing equipment at the room temperature. The glass reactor was illuminated by the wideband lamp bulb (25 W Philips TL/05) with a predominant wavelength of 365 nm; The vertical distance between the glass reactor and the lamp was 15cm; Then 100 ml (10 mg L<sup>-1</sup>) RhB solution was mixed with 0.10 g sample; prior to irradiation, the mixed solution was magnetically stirred in the dark for 30 minutes to ensure the establishment of an adsorption/desorption equilibrium between the CuO/TiO<sub>2</sub> films and the RhB dye. An aliquot (3 mL) of the solution was taken at half an hour interval during the experiment and tested on the Varian UV-vis spectrophotometer (Hitachi U-3010). As a comparison, an RhB photolysis process without CuO/TiO<sub>2</sub> films was also performed under the same conditions. After every assay, the analyzed aliquot was poured back into the glass reactor immediately to ensure a roughly equivalent volume of solution. Each catalytic experiment lasted for three hours in all. The change of RhB absorbance was used to monitor the extent of reaction at given irradiation time intervals.

## Results and discussion

### Characterization of CuO/TiO<sub>2</sub> nanofibers

The results of simultaneous TG and DTA analyses of the as-spun TiO<sub>2</sub>/PVP composite nanofibers are shown in Fig.1. The whole process is divided into three regions. The range of 30 °C–150 °C is the first part and there is an inconspicuous endothermic peak at 75 °C in DTA curve, which indicates the loss of moisture about 10 % from the composite fibers surface and residual moisture in internal fibers. The second part is 275 °C to 400 °C from DTA curve and there are two apparent exothermic peaks at 315 °C and 390 °C, which represent the decomposition of branched chain of PVP and the process of the formation for anatase phase respectively.<sup>26</sup> In this section, Weightlessness rate is about 30.4 %. From 400 °C to 520 °C

is the last part, the weight loss ratio about 20.4 %, which is mainly for the degradation of main chain of PVP until 520 °C. Therefore, the obtained non-woven film was calcined to form TiO<sub>2</sub> fibrous film at 520 °C.

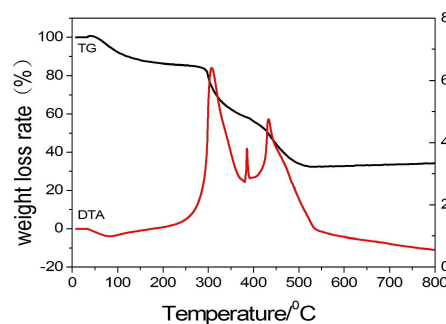


Fig.1 Thermal gravimetry– differential thermal analysis (TG-DTA) curves of PVP/TiO<sub>2</sub>

The hierarchical heterostructure of the CuO/TiO<sub>2</sub> sample was shown in Figure.2C; the SEM image Fig.2A shows that the TiO<sub>2</sub> fiber surface was smooth, which indicates that TiO<sub>2</sub> was uniformly dispersed in the PVP/TiO<sub>2</sub> mixture. The statistical tables on the Fig.2A and Fig.2C respectively for diameter of the statistics of TiO<sub>2</sub> fibers and CT1 of CuO/TiO<sub>2</sub> fibers showed that the diameter of TiO<sub>2</sub> fibers increased due to the adjunction of the nanostructural CuO after the hydrothermal treatment. Therefore, we observed the SEM Fig.2A and Fig.2C images and tested the diameter of 100 fibers before and after hydrothermal treatment respectively. The statistical graph revealed that the distribution of TiO<sub>2</sub> fibers diameter mainly ranged from 20~110nm amount to 72 %, 93 nm in average and there were 80 % of the CuO /TiO<sub>2</sub> fibers which were 300~450 nm, 381 nm in average. The CuO layer thickness was about 125 nm to 165 nm and we could control the thickness of CuO by adjusting the concentration of copper ions.

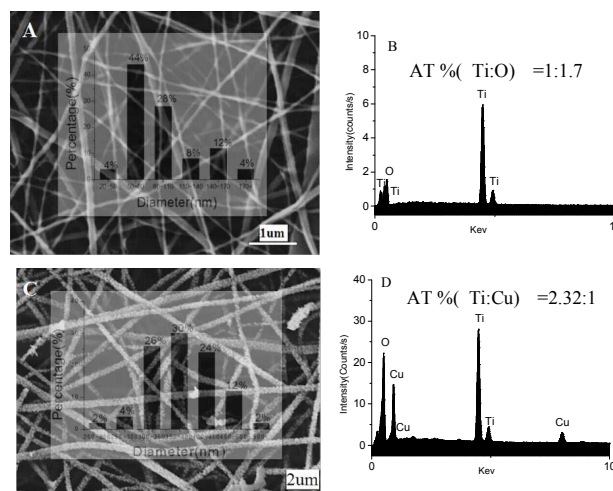


Fig.2 (A) The SEM images of TiO<sub>2</sub> nanofibers and statistics for diameter of the nanofibers TiO<sub>2</sub> fibers. (B) The EDS microanalysis on selected areas of TiO<sub>2</sub> nanofibers. (C) The

SEM of CuO/TiO<sub>2</sub> composite nanofibers heterostructure. (D) The EDS microanalysis on selected CuO/TiO<sub>2</sub> heterostructure.

The SEM image of low (Fig.3f) and high (Fig.3g) magnification show the high density of secondary CuO nanoparticles grown on the primary TiO<sub>2</sub> fibrous substrates. As we can see from Fig.3g that the CuO/TiO<sub>2</sub> fibers are full of pores on the surface, which may serve as transport paths for RhB molecules and utilize light sufficiently. So it is important for the property of photocatalysis. The SEM image Fig.3a shows the morphology of CT2; the CuO shaped in spiny grew on the surface of TiO<sub>2</sub> nanofibers following the TEM picture (Fig.3c, Fig.3d).

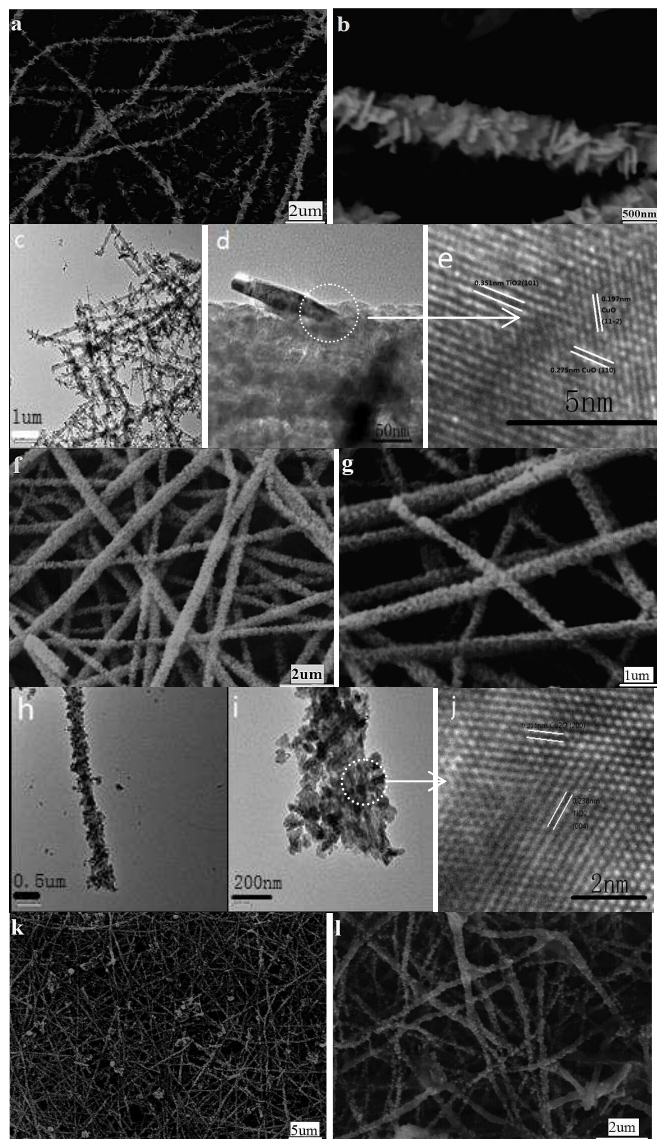


Fig.3 The SEM image (a) detailed view on CuO/TiO<sub>2</sub> hierarchical nanofibers of CT2 and a single nanofiber (b). (c) The TEM image of the CT2 and (d) a single fiber. (e) The HRTEM image for the details of CT1. (f) The SEM image of the overall morphology of the CT1 sample. (g) The SEM image of the CT1 detailed view on CuO/TiO<sub>2</sub> nanofibers. (h) The TEM image of single fiber of CT1. (i) TEM of the detail CT1. (j) The HRTEM image of CT1. (k), (l) SEM images

of CT1 (Five times after using).

Fig.3f and Fig.3g are the SEM images of CT1; the plenty of nanoparticles surround the TiO<sub>2</sub> fibers uniformly. The big difference in appearance between CT1 and CT2 is the different growth process in crystal growth. Ti-OH and Cu-OH would band together and leave a molecule of H<sub>2</sub>O on the interfacial TiO<sub>2</sub> fibers. However, the CuO crystal in the process of growth would continue to grow in the same crystal form of CuO nanoparticles which grew on the TiO<sub>2</sub> fibers initially; thus formed the "section-section" structure of bamboo shoots (Fig.3b). The CT1 was used hexamine as the alkali source. Since the mixed solution was heated, hexamine would decompose into NH<sub>3</sub> and CH<sub>2</sub>O and then the Cu<sup>2+</sup> would be reduced into Cu<sup>+</sup> by CH<sub>2</sub>O. In other words, the size of CuO crystal would increase if the Cu<sup>+</sup> was not formed. Because the two different lattices of CuO and Cu<sub>2</sub>O would restrain the growth of the crystal, therefore the morphology of composite fibers are filled with nanoparticles of CuO and little Cu<sub>2</sub>O. The XRD pattern could prove the results; the diffraction peaks in curve c (Fig.4) is the CT2 sample XRD pattern which shows the better crystallinity than that of CT1 in the curve b (Fig.4). The images of Fig.3k and Fig.3l were low and high magnification CuO/TiO<sub>2</sub> heterojunctions after five times reuse and the morphology still kept in good.

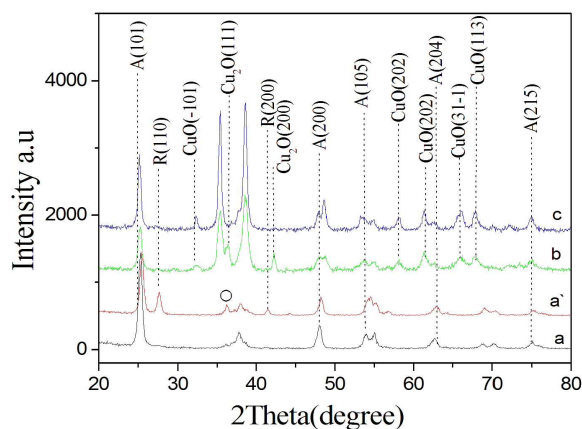
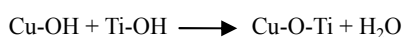
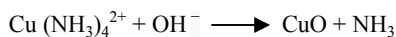
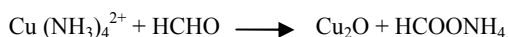
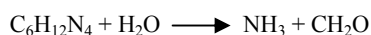


Fig.4 The XRD patterns of the as-synthesized products: (a) TiO<sub>2</sub> fibrous film. (a') TiO<sub>2</sub> films after hydrothermal treatment. (b) CT1 hierarchical heterostructure. (c) The XRD curve of CT2

The phase and crystal structure of the sintering electrospun TiO<sub>2</sub> nanofibers and CuO/TiO<sub>2</sub> composite fibers were characterized by X-ray diffraction (XRD). Curve a (Fig.4a) showed the XRD pattern of pure TiO<sub>2</sub> nanofibers and the lattice parameters were  $a = 3.785 \text{ \AA}$  and  $c = 9.514 \text{ \AA}$ , which are close to the reported values<sup>26</sup> of the anatase phase. However, all the diffraction peaks of the TiO<sub>2</sub> sample can be indexed as well-crystallized tetragonal TiO<sub>2</sub> (JCPDS # 21-1272). It is important to note that the samples analyzed in the present work showed the presence of rutile (Fig.4a') accompanying anatase after hydrothermal process, because the anatase would rearrange and form rutile under hydrothermal conditions.<sup>27</sup> There was a peak of rutile about 36° (marking) in the curve a', but not in the curve b and c for the copper ions might blocking the transition of anatase.<sup>28</sup> Fig.4b is the XRD patterns of CT1, which indicates that the Cu<sub>2</sub>O is mixed with the CuO/TiO<sub>2</sub> composite fibers. The  $2\theta = 42.30^\circ$  is the typical diffraction peak of Cu<sub>2</sub>O and the results could

be verified in the HRTEM of Fig.3j. The Fig.4c pattern which was the XRD pattern of CT2, which showed the crystallinity were better than CT1 for the sharp peaks.

The different morphology of the CuO on the fiber is mainly decided by alkali source. The hexamine would decompose into ammonia and formaldehyde gradually with the increase of temperature. Then the ammonia would combine the  $\text{Cu}^{2+}$  turning into  $\text{Cu}(\text{NH}_3)_4^{2+}$ . Small part of  $\text{Cu}(\text{NH}_3)_4^{2+}$  would be reduced to  $\text{Cu}_2\text{O}$  for the existence of formaldehyde in the alkaline environment. Meanwhile, most  $\text{Cu}(\text{NH}_3)_4^{2+}$  react with  $\text{OH}^-$  in aqueous solution and turn into CuO with the high temperature. As the large specific surface area of the  $\text{TiO}_2$  nanofibers, it provides lots of active sites for the crystal growth of CuO. Moreover, as a result of the existence of the Ti-OH groups<sup>30</sup>, it contributed to the formation of the bounds Ti-O-Cu, so the CuO would nucleate at the surface easily. According to the Oswald ripening mechanism<sup>31</sup>, the seed crystal of CuO will get bigger following the increasing number of small crystal.



### Photocatalytic oxidation of RhB

RhB is a chemically stable and poorly biodegradable dye for the difficulty to break chemical bonds and it is main contaminant in Water pollution of the environment. To prove the photocatalytic activity of CuO/ $\text{TiO}_2$  films for the three-dimensional hierarchical heterostructure, we set up two sets of controlled trials using RhB as the degradation material. The blank test (Figure.5A) (without catalyst) under UV-vis light illuminations exhibited little photolysis about 10 % photodegradation efficiency for RhB in 3h. By contrast, when the CuO/ $\text{TiO}_2$  films were used, the adsorption peak of the RhB at around 554 nm underwent fairly large decrease with irradiation for 3h about 95 %. We also investigated the photodegradation behaviors of RhB under continuous photocatalysis of the mixture of CuO and  $\text{TiO}_2$  films (Single factor control variable method) according to the molar ratio in the EDS date of CT1 (Figure.2D). As we can see from the Figure.5B and Figure.5C, the CuO/ $\text{TiO}_2$  composite fibers exhibited much higher photocatalytic activity than the mixture of CuO and  $\text{TiO}_2$  fiber. In addition, the catalyst we got had good reuse capacity; after five times reuse, the degradation ratio of RhB maintains 91.42 % and the ratio of recycled was still as high as 93 %. With the increase number of reuse, some CuO come off from the  $\text{TiO}_2$  fibers (Fig.3k, Fig.3l) and a small amount of the loss when recycled were the main reasons for the slightly lower photocatalytic efficiency than before.

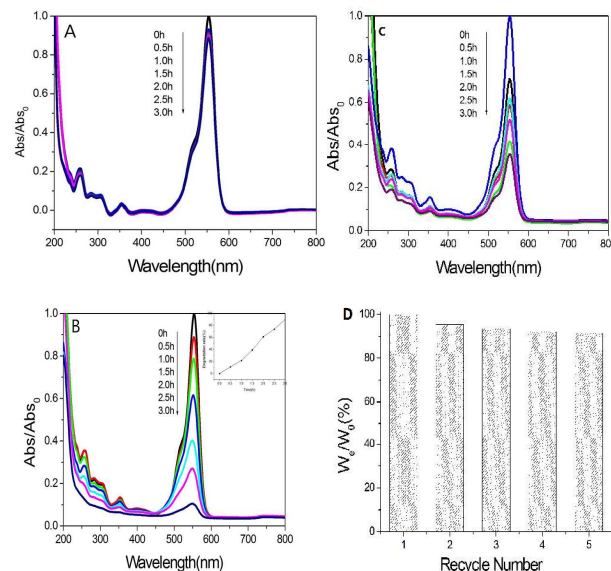
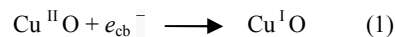
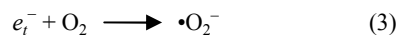


Fig.5 Curves of photocatalytic degradation of RhB under UV-light: (A) blank test. (B) In the presence of CuO/ $\text{TiO}_2$  hierarchical heterostructure. (C) The test for the mixture of CuO and  $\text{TiO}_2$  fibers. (D) The diagram of degradation ratio of RhB versus reuse times (The  $W_e$  indicates the degradation ratio of reuse times and  $W_0$  indicates the first degradation ratio of RhB).

Comparing to the two results of Photocatalytic degradation experiments (Fig.5B, Fig.5C), the composite CuO/ $\text{TiO}_2$  nanofibers showed higher Photocatalytic activity, therefore the CuO attaching on the surface of  $\text{TiO}_2$  fibers played a great role in the process of degradation of RhB. The band edge potentials of  $\text{Cu}^{2+}/\text{Cu}^0$  and  $\text{Cu}^{2+}/\text{Cu}^+$  (+0.34V and +0.17V respectively)<sup>32</sup> are located below the conduction band edge of  $\text{TiO}_2$  (-0.2V).<sup>33</sup> When the heterojunction is irradiated by UV-light, the nano  $\text{Cu}^{\text{II}}\text{O}$  could capture the electron from the surface of  $\text{TiO}_2$  fibers. Since the  $\text{Cu}^{2+}$  ion has the unfilled 3d orbits ( $3d^94s^0$  configuration), the reduction of  $\text{Cu}^{2+}$  is thermodynamically feasible (formula 1). The holes on the  $\text{TiO}_2$  would be more and react with  $\text{H}_2\text{O}$  and produce radical species  $\cdot\text{OH}$  followed by formula 2.



The electrons transfer to the nano-CuO and the charge will accumulate on the surface, therefore it must be consumed; some electrons would recombine with the photogenerated hole from the  $\text{TiO}_2$  and react with oxygen to produce the  $\cdot\text{OH}$  radicals after many transitions<sup>34</sup> (formula 3-6). The  $\text{Cu}^{\text{I}}\text{O}$  will be oxidized by the radical species which are generating from the reaction of electrons and oxygen.<sup>35</sup> Hence the  $\text{Cu}^{\text{I}}\text{O}$  would transform into  $\text{Cu}^{\text{II}}\text{O}$  after the process of electron transfer. Ultimately, the recombination rate of electron hole pair should be reduced effectively. After a series of reactions, these reactive oxygen species ( $\cdot\text{O}_2^-$  and  $\cdot\text{OH}$ ) diffusing in the solution would attack the chromophoric structure of RhB, leading to the cycloreversion of the RhB compounds according to the degradation curves (Fig.5B).<sup>36</sup>



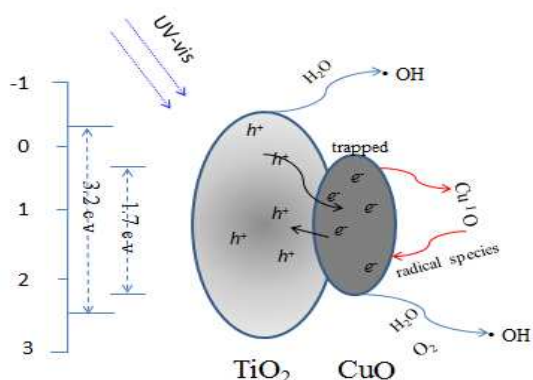
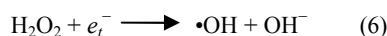
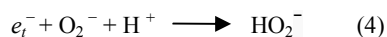


Fig.6 The mechanism of photocatalytic reactions on semiconductor photocatalyst

## Conclusions

The 3D CuO/TiO<sub>2</sub> films were synthesized via a combination of an electrospinning method and a hydrothermal reaction. The morphologies of secondary nano-CuO appeared with different shapes by adjusting the experimental parameters. Moreover, the photocatalytic performance test and the results showed the composite CuO/TiO<sub>2</sub> fibers had better photocatalytic property than the mixture of CuO and TiO<sub>2</sub> films under same conditions. So the heterojunction of CuO/TiO<sub>2</sub> may possess great potential in the photocatalysts, sensors and other fields after further research.

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