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1 **Synthesis and characterization of SiO₂/polyaniline/Ag core-shell particles and studies of**
2 **their electrical and hemolytic properties: Multifunctional core-shell particles**

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10 **Abstract**

11 Three layers conducting core-shell nanocomposites particles composed of SiO₂/Polyaniline
12 (PAni)/Ag were prepared in the presence of silicon dioxide (SiO₂) in aqueous solution containing
13 sodium dodecyl benzenesulfonate (SDBS) as a surfactant. SiO₂ nanoparticles were coated by
14 PAni, which results in the formation of core-shell nanocomposites. Silver nanoparticles
15 synthesized by citrate reduction method. Ag nanoparticles could be electrostatically attracted on
16 the surface of SiO₂/PAni nanocomposites, leading to formation of SiO₂/PAni/Ag
17 nanocomposites with core-shell structure. The products were characterized by Fourier transform
18 infrared (FT-IR) spectroscopy, X-ray diffraction (XRD), scanning electron microscopy (SEM),
19 transmission electron microscopy (TEM), current-voltage (I-V) analysis and cyclic voltammetry
20 (CV). The resultant nanocomposites have good biological property.

21 **Key words:** Ag nanoparticles, SiO₂ nanoparticles, Nanocomposites, conducting polymer,
22 Electrical Conductivity.

23 **Introduction**

24 In recent years conducting polymers with multifunctional core-shell structures have received
25 considerable interest because of their conductivity, magneto-rheological response, electronic
26 properties, mechanical properties and easy processing.¹⁻⁵ Conducting polymers such as poly (p-
27 phenylene sulfide), polyaniline (PAni), polyacetylene, polythiophene, poly (p-phenylene
28 vinylene), polypyrrole (PPy) etc, always have been the center of attraction due to their
29 conducting nature and found to be suitable for sensors, actuators, light emitting diodes,
30 capacitors, battery electrodes etc.⁶⁻⁸ Amongst conducting polymers, PAni shows some great
31 promises due to its excellent environmental and chemical stability.⁹⁻¹⁰ PAni and its composites
32 have been extensively applied for their electrorheological fluids.¹¹⁻¹² The core-shell structure is
33 particularly interesting because the core can be imitative to make the particle such as fluorescent
34 or magnetic, independent of the shell properties. The core-shell structures between silica sphere
35 and conducting polymer have attracted attention.¹³⁻¹⁴ A thin layer of conducting PAni reduces
36 the cost and enhances surface area which is favorable for application in catalysis, controlled
37 delivery and in photographic industry. The uses of silica as core have some advantages like high
38 colloidal stability and easily controllable particle size.¹⁵

39 Silica nanoparticles are used in enormous products ranging from cosmetic to construction
40 materials. It has a potential application in the drug delivery, biosensors.¹⁶⁻¹⁷ The use of
41 nanoparticles with core-shell or multilayer morphology allows one to obtain complex
42 compositional and structural patterns in the ultimate nano composite material. Generally silica
43 NPs are in colloidal form or agglomerated dry powder. However, simply PAni undergoes rapid
44 degradation in performance upon repetitive cycles of charging and discharging due to its
45 swelling and shrinkage.¹⁸⁻²⁰ In order to improve these limitations, the incorporation of silica with
46 PAni has been proved to reinforce the stability.

47 Silver nanoparticles have potential used in health care delivery due to its antimicrobial and
48 wound healing properties.²¹ Silver salts or metallic silver have been used to prevent bacterial
49 infections associated with medical devices, such as wound dressings, catheters, and orthopedic
50 and cardiovascular implants, with different degrees of clinical efficacy. Due to its high surface to
51 volume ratios silver nanoparticles have high reactivity and hence it provides a new platform for
52 use in both consumer and biomedical applications.²¹⁻²²

53 Now-a-days, three layer core-shell nanocomposites have been gaining much attention. Three-
54 layered core-shell structures with Au-modified PPy-coated Ag nanocomposites through the
55 mixing of Au colloidal solution with PPy/Ag were reported. Au/PPy/Ag nanocomposites were
56 immobilized on the surface of a glassy carbon electrode and utilized to construct a dopamine
57 biosensor. The synthesized biosensor could detect dopamine at its low concentration in the
58 presence of 5000 times concentration of ascorbic acid at neutral environment.⁶ A tri layer core
59 shell nanostructure where silica as core with Fe₃O₄ and Au as inner, outer shells, respectively
60 were prepared. These core-shell magnetic nanoparticles have excellent DNA binding properties
61 as well as magnetic properties.²³ Synthesis of silica-PAni core-shell nanoparticles by in-situ
62 polymerization was reported. The synthesized silica-PAni core-shell nanoparticles showed pH-
63 responsive redox reversibility and relatively high electrical conductivity.²⁴ Fe₃O₄/PPy/Au core-
64 shell nanoparticles with excellent electrocatalytic properties with good magnetism were reported.
65 The resultant nanocomposites easily controlled by an external magnetic field but also have the
66 good conductivity and excellent electrochemical and catalytic properties of PPy and Au
67 nanoparticles. Furthermore, the nanocomposites showed excellent electrocatalytic activities to
68 bio-species such as ascorbic acid.²⁵ Three layer core-shell nanocomposites with NiO/PPy/Ag as
69 the core, inner shell and outer shell was also reported. The synthesized nanocomposites provide

70 good electrical conductivity with the increase of NiO nanoparticles content. The core-shell
71 nanocomposites showed an enhanced bactericidal effect and hence it can be used in water
72 purification technology.²⁶ Highly conductive and crystalline PANi onto silica nanoparticles with
73 core-shell morphology were prepared by using self-stabilized dispersion polymerization method.
74 ²⁷ The conductivity of the core-shell nanoparticles depends upon the diameter of the particles. As
75 the diameter of the silica-PANi core-shell nanoparticles decreased from 130 to 18 nm, the
76 electrical conductivity of the core-shell nanoparticles was found to be increased from 16.4 to
77 25.6 Scm⁻¹.

78 In this work, a facile approach has been developed for the fabrication of SiO₂/PANi/Ag core-shell
79 nanoparticles. The SiO₂ nanoparticles were synthesized by hydrolysis of TEOS followed by
80 thermal decomposition and it is used as a core for the synthesis of core-shell nanoparticles. PANi
81 shell was formed over the surface of SiO₂ nanoparticles by *in-situ* polymerization of positively
82 charged anilinium ions on the negatively charged of the silica nanoparticle through electrostatic
83 forces. Finally SiO₂/PANi/Ag nanocomposites were formed by the electrostatic interaction
84 between the Ag nanoparticles and PANi chain and thereby we obtain three layer SiO₂/PANi/Ag
85 core-shell structure. The synthesized nanocomposites showed good electrical conductivity as
86 well as the hemolytic property.

87 **Experimental Section**

88 **Materials**

89 Tetra ethoxy silane, aniline, sodium dodecylbenzenesulfonate (SDBS) were purchased from
90 Sigma Aldrich. AgNO₃, HCl, (NH₄)₂S₂O₈, tri-sodium citrate dehydrates, acetonitrile, NH₄OH,
91 KCl were purchased from Merk, India. All the chemicals were of analytical grade. Deionized,

92 distilled water and purified ethanol were used as dispersion medium for all the chemical
93 preparation as well as for washing.

94 **Method**

95 **Preparation of Silicon dioxide (SiO₂) nanoparticles**

96 SiO₂ nanoparticles were prepared by the hydrolysis of tetra ethoxy silane (TEOS).²⁸ In a typical
97 synthesis process, TEOS (6 mL) and ethanol (40 mL) were mixed under low frequency
98 ultrasound. Hydrolysis of TEOS is performed by the addition of about 1 mL of distilled water.
99 About 2.55 mL of NH₄OH solution was added to the reaction medium and the mixture was
100 sonicated for 2 h. This results the gelation in the reaction medium. The gel was centrifuged and
101 properly washed with a mixture of ethanol and water to remove the unwanted impurities. The
102 samples were freeze dried for about 12 h. The collected sample was heated in the muffle furnace
103 at about 600°C for 2 h. Then it was allowed to cool to room temperature and collect the resultant
104 mass.

105 **Preparation of silver (Ag) nanoparticles**

106 A colloidal stable dispersion of Ag nanoparticles was synthesized by citrate reduction method.²⁹⁻
107 ³⁰ 50 mL 0.001 M AgNO₃ solution in distilled water was taken in a conical flask and heating up
108 it with constant stirring to boiling. 5 mL of 1% tri-sodium citrate solution was added into it. The
109 solution was turned into light yellow in color. This color results the formation of Ag
110 nanoparticles. Stirring was continued until the solution became cool down to room temperature.

111 **Preparation of SiO₂/PAni core-shell nano composites**

112 PAni was prepared by using chemical oxidation of aniline monomer in the acidic solution of a
113 non-anionic surfactant. SiO₂/PAni core-shell nano composites were prepared by in-situ
114 polymerization technique with slight modification, where SiO₂ is the core and the PAni is the

115 shell of the composites.³¹ Freshly prepared silica nanoparticles were dispersed in an aqueous
116 solution of SDS (0.05 g, 30 mL water). The mixture was sonicated for about 30 min. After that,
117 it was transferred into a solution composed of 4.50 g of aniline, 7.50 mL of 1 M HCl, 150 mL of
118 water at room temperature and stirred it for 1 h. 1.15 g of $(\text{NH}_4)_2\text{S}_2\text{O}_8$ (APS) was dissolved in 50
119 mL of distilled water. The APS solution was transferred to the reaction mixture and the reaction
120 was allowed to proceed for 24 h at room temperature without disturbing. The resultant products
121 were separated by filtration followed by washing with distilled water and then methanol for
122 several times. The products were dried at 60°C in an oven for 24 h. Different core-shell particles
123 of SiO_2/PAni were prepared by changing the amounts of SiO_2 nanoparticles keeping other
124 ingredients constant.

125 **Preparation of $\text{SiO}_2/\text{PAni}/\text{Ag}$ three layered nanocomposites**

126 Ag nanoparticles were adsorbed on PAni coated SiO_2 nanocomposites by electrostatic
127 interaction. SiO_2/PAni nanocomposites were added into the Ag nanoparticles colloidal solution at
128 a concentration of 2.50 mg/mL under stirring condition and the stirring was continued for 48 h.
129 Ag nanoparticles attracted on the surface of the SiO_2/PAni nanocomposites, which leads to the
130 formation of $\text{SiO}_2/\text{PAni}/\text{Ag}$ core-shell nanocomposites. The light yellow color of the Ag nano
131 colloidal solution turned colorless with the increasing time of mixing. Finally, the resultant
132 product was separated from the solution and dried under vacuum.

133 **Characterization**

134 **Fourier transforms infrared (FTIR) spectroscopy**

135 The FTIR spectrum of the composites was recorded in Nicolet Impact-410 IR spectrometer in
136 KBr medium at room temperature in the range of 400-4000 cm^{-1} .

137 **Scanning Electron Microscopy (SEM)**

138 SEM micrographs of the core-shell particles were taken with a Jeol-JSM-6390L V scanning
139 electron microscope. Composite samples were sputter coated with platinum thickness of 200Å.

140 **Transmission Electron Microscopy (TEM)**

141 The size and morphologies of the products were observed with a JEOL JEM 2100 transmission
142 electron microscope at an acceleration voltage of 200 kV. In the TEM measurement, the samples
143 were prepared by dropping highly diluted composites on the carbon coated copper grid and dried
144 in a vacuum oven at room temperature.

145 **Thermo gravimetric analysis (TGA)**

146 Thermo gravimetric analysis of the products was studied in a Shimadzu TA50 thermal analyzer.
147 The heating was done under nitrogen atmosphere at a heating rate of 5°C /min in the range of 25-
148 700°C.

149 **Electrical conductivity**

150 Pellets of composite samples were made by using a compression molding machine with
151 hydraulic pressure. High pressure was applied to the sample to get hard round shaped pellet (1.5
152 cm diameter, 2 mm breadth) which will be used in measuring conductivity.

153 The electrical conductivity of SiO₂/PAni and SiO₂/PAni/ Ag composites were measured by using
154 four probe techniques. I-V characteristic was studied by Keithley 2400 sourcemeter at the room
155 temperature in the frequency range 102-106 Hz. The voltage was applied to measure the current
156 passing through the sample.

157 Cyclic Voltammetry (CV) measurement was performed on an electrochemical work station
158 Sycopel AEW2-10 with an Ag/AgCl reference electrode, a platinum wire as a counter electrode
159 and PAni coated SiO₂ core-shell composite and PAni-Ag coated SiO₂ core-shell films (1, 3, 5, 7
160 and 10% SiO₂ with respect to the monomer) on ITO coated glass as working electrode. The

161 electrochemical characteristics of the composite sample were investigated by cyclic voltametric
162 scanning at a scan rate of 50 mVs^{-1} . A solution of 0.1 M KCl prepared in 10 ml acetonitrile was
163 used as supporting electrolyte.

164 **Method for Haemolysis Assay**

165 According to the method described by Zhu et al., the haemolytic assay was performed with little
166 modification to check the lysis of the RBC membrane by the $\text{SiO}_2/\text{PAni}/\text{Ag}$ nanoparticles.³¹ For
167 that the blood was collected in a tube containing 4% sodium citrate and centrifuged at 3000 rpm
168 at 4°C for 15 min. The supernatant was aspirated and the erythrocytes were washed with
169 phosphate saline buffer (PBS, pH 7.4) for three times. After washing, 5% packed erythrocytes
170 were gently resuspended with PBS. Different concentrations of the nanoparticles (0.5 mg/mL, 1
171 mg/mL, 2.5 mg/mL, 5 mg/mL, 10 mg/mL, and 20 mg/mL) were mixed gently with the
172 haematocrit in the ratio of 95: 5 (V: V). 1% Triton X100 was used as the positive control as it
173 can lyse the RBC cells and PBS as the negative control. The mixtures were incubated for 1 h at
174 37°C . After the termination of the incubation period, the cells were placed in an ice bath for 1
175 min and then centrifuged at 3000 rpm for 10 min at 4°C . The measures of hemolysis were
176 observed by taking absorbance 540 nm.

177 **Result and Discussions**

178 **FT-IR analysis**

179 The FT-IR spectrum of SiO_2 nanoparticle, Ag nanoparticles, $\text{SiO}_2/\text{PAni}-\text{Ag}$ nanocomposites are
180 shown in the Fig. 1. (a), (b), (c). In the FT-IR analysis of SiO_2 nanoparticles, [Fig. 1(a)], there are
181 three absorption bands are seen in the region of $1500\text{-}450 \text{ cm}^{-1}$. A characteristic absorption band
182 is observed at 475 cm^{-1} , due to racking mode of the Si-O-Si group. An absorption band at 806
183 cm^{-1} is due to the symmetric stretching of Si-O-Si bond. A sharp band at 1112 cm^{-1} corresponds

184 to asymmetric stretching of Si-O-Si bond. These absorption bands indicate the formation of SiO₂
185 nanoparticles. The absorption bands at 1169, 1268 and 1637 cm⁻¹ indicates the formation of Ag
186 nanoparticles [Fig. 1(b)]. The 1308 cm⁻¹ band is assigned to the C-N stretch of a secondary
187 aromatic amine whereas, in the region of 1010-1170 cm⁻¹, the aromatic C-H in-plane bending
188 modes are usually observed [Fig. 1(c)]. Out-of-plane deformations of C-H on 1, 4-disubstituted
189 rings are located in the region of 800-880 cm⁻¹. The region 900-700 cm⁻¹ corresponds to the
190 aromatic ring and out-of-plane deformation vibrations.

191 **Thermal stability**

192 To investigate the thermal stability of PANi, SiO₂/PANi, SiO₂/PANi-Ag core-shell
193 nanocomposites, thermo-gravimetric analysis were performed [Fig. 2]. From the analysis it is
194 seen that every sample undergoes two stage weight losses. The first weight loss at about 40-
195 100°C is due to loss of residual moisture from the polymer matrix. PANi is stable upto 250°C.
196 But core-shell composites particles degradation occurs at 300°C. Core-shell composite particles
197 degrade at higher temperature. The residual weight for SiO₂/PANi (5%), SiO₂-PANi (7%) are
198 found to be 48.41 and 54.05 respectively at 600°C. The residual weight for SiO₂/PANi/Ag (5%),
199 SiO₂-PANi/Ag (7%) are found to be 61.94 and 62.10 respectively at 600°C. TG-analysis reveals
200 that there is an improvement in the thermal stability of the PANi as well as SiO₂/PANi-Ag
201 composites due to the incorporation of SiO₂ and Ag nanoparticles into the polymer matrixes.

202 **X-ray diffraction (XRD) analysis**

203 XRD analysis was performed to investigate the crystalline or amorphous nature of the
204 nanoparticles and the nanocomposites. XRD patterns of Ag nanoparticles, SiO₂ nanoparticles,
205 PANi and SiO₂/PANi-Ag core-shell nanocomposites are shown in the [Fig. 3(a), (b), (c) (d)].
206 XRD peak for Ag nanoparticles are observed at 2θ angle 38° (111), 44° (200) and 64° (220) [Fig.

207 3(a)]. Pure SiO₂ nanoparticles give a XRD peak at 2θ angle 21.2° [Fig. 3(b)]. These
208 characteristic peaks indicate the formation of SiO₂ nanoparticles. Pure PANi is amorphous is
209 nature and hence it gives a broad XRD peak with 2θ angle in between 19.45 to 30.40 [Fig. 3(c)].
210 The XRD peaks at 38° (111), 44° (200) and 64° (220) for SiO₂/PANi-Ag nanocomposites
211 indicates the successful incorporation of Ag into the SiO₂/PANi core-shell nanocomposites [Fig.
212 3(d)].

213 Scanning Electron Microscope (SEM) analysis

214 To determine the surface morphology of the nanoparticles and the nanocomposites, SEM was
215 performed. Fig. 4(a) shows the SEM image of the SiO₂ nanoparticles. The SiO₂ nanoparticles
216 were found to be the nearly uniform shape and the particles are found in the range from 30-40
217 nm. SEM image of PANi [Fig. 4(b)] display the tubular structure with outer diameter 55-60 nm.
218 SEM image [Fig. 4(c)] of SiO₂/PANi shows that the outer diameter of the SiO₂/PANi (size 80-90
219 nm) is greater than that of pure SiO₂ nanoparticle (55-60 nm). This information proves the
220 successful incorporation of PANi onto SiO₂ nanoparticles. The dotted spot in the SEM image of
221 SiO₂/PANi is may be due to the presence of SiO₂ nanoparticles in the composites. The SEM
222 image of the SiO₂/PANi/Ag is given in the [Fig. 4(d)]. The diameter of this composite is more
223 than that of the SiO₂/PANi. This implies the successful attachment of the Ag nanoparticles on the
224 surface of the SiO₂/PANi.

225 Energy Dispersive X-ray (EDX) analysis

226 To identify the presence of the elements, the core-shell nanocomposites were characterized by
227 EDX-analysis. [Fig. 5] represented EDX image of SiO₂, SiO₂/PANi and SiO₂/PANi-Ag core-shell
228 nanocomposites. Presence of only Si and O peak in the Fig. 5(a) indicates the formation of pure
229 SiO₂ nanoparticles. The presence of Si and O peak in the EDX graph [Fig. 5(b)] indicates the

230 presence of SiO₂ in SiO₂/PAni nanocomposites. Presence of C and N peaks indicates the
231 successful formation of PAni over SiO₂ nanoparticles. The existence of Ag peak [Fig. 5(c)]
232 indicated the attachment of Ag nanoparticles on SiO₂/PAni surface.

233 **Transmission Electron Microscope (TEM) analysis:**

234 Fig. 6 represents the TEM images of SiO₂ nanoparticles, Ag nanoparticles, SiO₂-PAni core-shell
235 nanocomposites, SiO₂/PAni-Ag core-shell nanocomposites. In the Fig. 6(a), the synthesized SiO₂
236 nanoparticles are found to be spherical and the average size of the nanoparticles is found to be
237 about 40 nm. Fig. 6(b) represents the TEM image of Ag nanoparticles and the average particles
238 size of the particles is found to be 10-15 nm. TEM image [Fig. 6(c)] of SiO₂-PAni
239 nanocomposites, shows core-shell structure. It is seen that dark spots with diameter about 40-45
240 nm coated by light shaded shell structure. This result confirms the successful coating of PAni on
241 SiO₂ nanoparticles and resulting the formation of SiO₂-PAni nanocomposites with core-shell
242 morphology. The dotted spots on the surface of SiO₂/PAni-Ag nanocomposites [Fig. 6(d)]
243 indicate the successful attachment of Ag nanoparticles on the surface of SiO₂/PAni
244 nanocomposites.

245 **Electrical behavior**

246 **Current-Voltage (I-V) relationship**

247 The current-voltage relationships for the SiO₂/PAni-Ag composites with different Ag content are
248 performed to study the conducting behavior of the composites and it is shown in the Fig. 7. Fig.
249 7 denotes the I-V relationship of PAni as well as the core-shell nanocomposites with various
250 compositions of Ag at room temperature. From the I-V analysis of the core-shell nanocomposites
251 exhibit a non-ohmic behavior as V/I is not proportional and it indicates that the core-shell
252 nanocomposites particles behave as a Schottky junction. This exponentially behavior of the I-V

253 curves can be explained with the help of conducting mechanism of PANi. In case of conducting
254 polymers such as PANi, charge conduction is carried by the formation of polarons (electrons) and
255 bipolarons (holes) in addition to free charge carriers. Whenever the applied voltage increases, the
256 formation of polarons and the bipolarons increases quickly which results the rapid increase in
257 current in comparison to the applied voltage and hence results an exponential I-V curve. With
258 increase of Ag nanoparticles content in the PANi matrix, the exponential behaviour of the I-V
259 curves gradually decreases. This is because, interactions takes place between Ag nanoparticles
260 and the PANi. Hence with the increase of Ag nanoparticles content into the shell phase, the
261 potential difference (in Volt) increases exponentially with the applied current (in mA). Hence, it
262 may be concluded that the synthesized core-shell nanocomposites are of semiconducting nature.

263 **Electrochemical property**

264 To investigate the redox properties of the nanocomposites, Cyclic-Voltammetry (CV) analysis
265 was conducted by varying SiO₂ nanoparticles in the composites. Fig. 8 represents the CV graphs
266 of SiO₂/PANi nanocomposites with different SiO₂ nanoparticles content (0, 1, 3, 5, 7 and 10%
267 respectively). The shapes of the CV graphs of the nanocomposites were found to be different
268 from the PANi. The electrochemical band-gap of the nanocomposites was calculated by using the
269 following formulae:

$$270 \quad \text{HOMO} = - [\varphi_{\text{onset}}^{\text{ox}} + 4.71] \text{ (eV)} \quad (1)$$

$$271 \quad \text{LUMO} = - [\varphi_{\text{onset}}^{\text{red}} + 4.71] \text{ (eV)} \quad (2)$$

$$272 \quad E_{\text{ec}}^{\text{g}} = [\varphi_{\text{onset}}^{\text{ox}} - \varphi_{\text{onset}}^{\text{red}}] \text{ (eV)} \quad (3)$$

273 where the units $\varphi_{\text{onset}}^{\text{ox}}$ (onset oxidation potential) and $\varphi_{\text{onset}}^{\text{red}}$ (onset reduction potential)
274 are V vs Ag/AgCl. The electrochemical band gap of PANi was found to be higher than that of
275 SiO₂/PANi [Table-1]. The electrochemical band gaps of SiO₂/PANi are found to decrease from

276 2.01 eV to 1.32 eV with the increasing amount of SiO₂ in the core-shell nanocomposites. The
277 peak currents for the composite were higher compared with those for PANi. The decrease of
278 electrochemical band gaps were also observed [Fig. 9] after the deposition of Ag nanoparticles
279 in the surface of PANi shell and it was listed in the Table-2. This enhancement in current may be
280 attributed to the special nanostructure of this composite. The change in electronic band structure
281 manifested as new mid-gap state being created and thereby decreasing of electrochemical band
282 gap of the composites.

283 **Charge capacity:**

284 To investigate the charge capacity of the nanocomposites, CV analysis was performed upto 100
285 repeated cycles. Fig. 10 represents the CV graph of the nanocomposites up to 100th cycle. The
286 area under the CV peaks could be integrated to produce charge capacity of the nanocomposites,
287 which gives information about the electro activity of the nanocomposites. From the analysis it
288 clearly revealed that cathodic peaks and anodic peaks were approximately symmetrical above
289 each other with least separation. The charge capacity of the core-shell nanocomposites did not
290 get reduced even after the repeated cycles. From this analysis we can conclude that the final
291 material can be used in rechargeable battery material.

292 **DC Electrical conductivity**

293 The DC electrical conductivities of the synthesized core-shell nanocomposites with
294 varying the amount of SiO₂ nanoparticles content in the core phase were measured at the room
295 temperature (25°C) and it tabulated in the Table 3. From the conductivity measurements it is
296 seen that the conductivity value of the SiO₂/PANi, SiO₂/PANi-Ag core-shell nanocomposites are
297 found to be higher than that of the pure PANi. It gradually increases with the increasing the
298 amount of SiO₂ nanoparticles in the core. The DC-electrical conductivity value of the core-shell

299 nanocomposites vary in the range from $0.15 \times 10^{-3} \text{ Scm}^{-1}$ to $5.88 \times 10^{-3} \text{ Scm}^{-1}$. The conductivity
300 value of the core-shell nanocomposites changes rapidly with the deposition of Ag nanoparticles.
301 This increase in conductivity of the core-shell nanocomposites may be due to the increased
302 compactness of the composite material as the growing polymer chains are supported on the SiO_2
303 nanoparticles. Thus, on incorporation of SiO_2 particles into the polymer, which restricts the
304 twisting of the polymer backbone away from its planarity and hence the polymer chains become
305 more ordered. This results the linking between the polymer chains is increasingly improved and
306 at the same time it provides a higher DC electrical conductivity of the PANi/ SiO_2 core-shell
307 nanocomposites compared with pure PANi.

308 **Hemolytic assay:**

309 The lysis of the RBC membrane was investigated by RBC haemolytic assay to observe
310 hemolytic of the Ag. The percentage of the haemolytic activity was observed very less in
311 comparison to the positive control [Fig. 11]. The inhibition percentage of hemolysis is equal to
312 the negative control when the cells treated with the nanoparticles of concentration upto 5 mg/mL.
313 But it showed a little hemolysis activity when the cells treated with 10 mg/mL and more than
314 that. When the cells treated with the different amounts of the nanoparticles, the absorbance at
315 540 nm is less in the nanoparticle treated cells compared to the cells treated with Triton X-100.
316 According to the literature, the absorptivity at 540 nm is observed due to the presence of
317 hemiglobincyanide.³² Since the absorbance is less in nanoparticles treated cells, it implies that
318 the lysis of the RBC is less and equal to the negative control due to which the hemoglobin
319 molecules could not come out from the cells. But in case of Triton X-100 treated cells, though it
320 can rupture the cell membrane, the hemoglobin molecules can come out from the cells due to

321 which the absorbance is observed more. Hence it can be considered that the nanoparticles have
322 the hemolytic property upto optimum level.

323 **Conclusions**

324 At summary, we have successfully synthesized the multicomponent SiO₂/PAni/Ag
325 nanocomposites with core-shell morphology, where SiO₂ as the core which is coated with
326 conducting polymer, PAni as the inner shell and Ag nanoparticles are attached on the surface of
327 PAni. UV, FTIR, XRD, SEM and TEM analysis confirmed the formation of SiO₂ and Ag
328 nanoparticles. FT-IR and XRD analysis confirmed the formation of PAni, SiO₂/PAni and
329 SiO₂/PAni/Ag. From the TG-analysis, we have concluded that the nanocomposites have
330 potentially stable to heat and it is increases with the incorporation of SiO₂ nanoparticles into the
331 polymer matrixes. From the CV analysis it is seen that with the incorporation of SiO₂ the
332 electrical property enhances. The nanocomposites are stable under cyclic redox potential upto
333 100 cycles and hence it can be used in re-chargeable battery material. From the hemolytic
334 analysis it can be confirmed that the final core-shell nanocomposites, i.e. SiO₂/PAni/Ag is shows
335 the hemolytic property upto an optimum level. This simple approach could provide an
336 opportunity to design unique core-shell nanostructures with different functionality.

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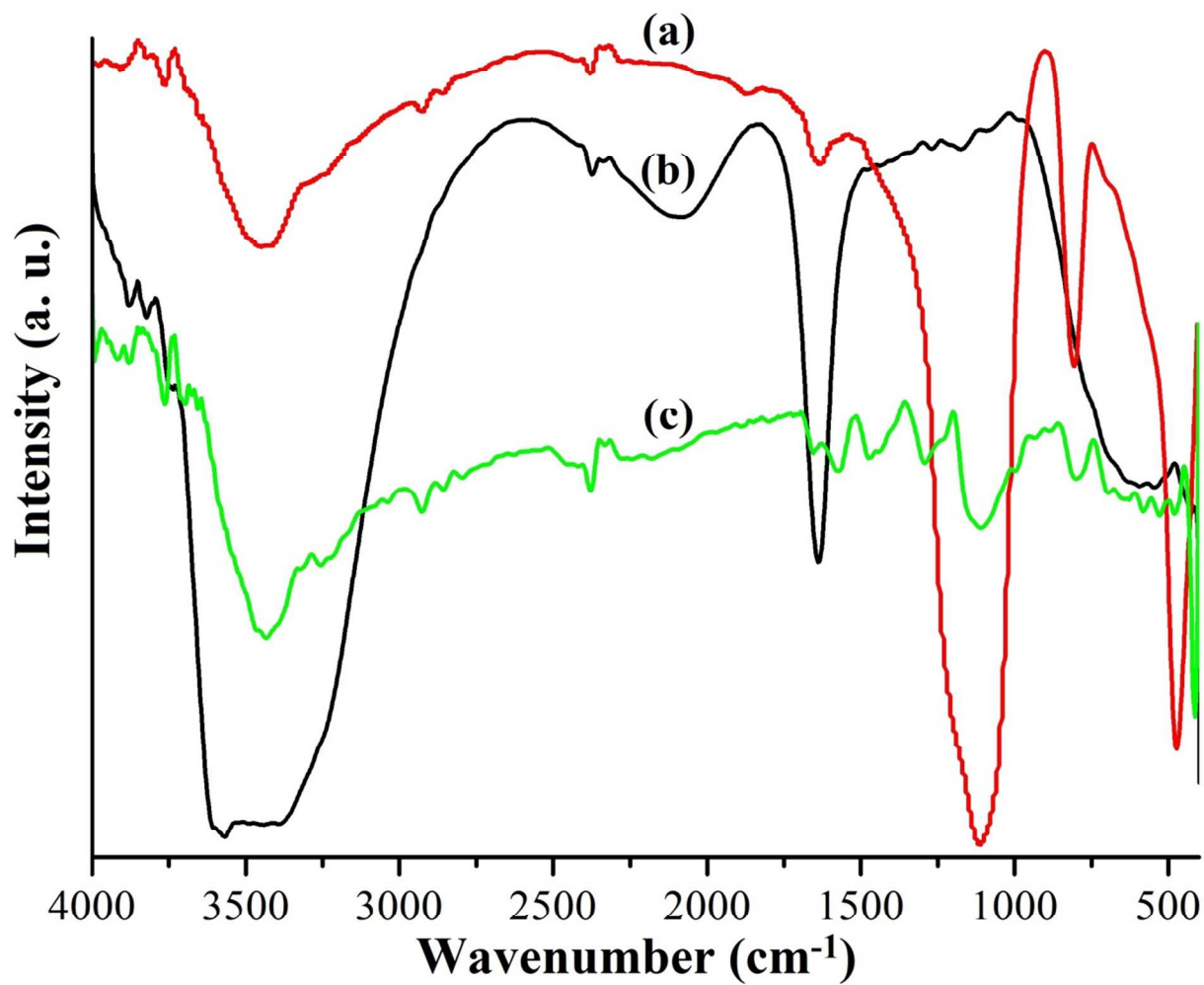
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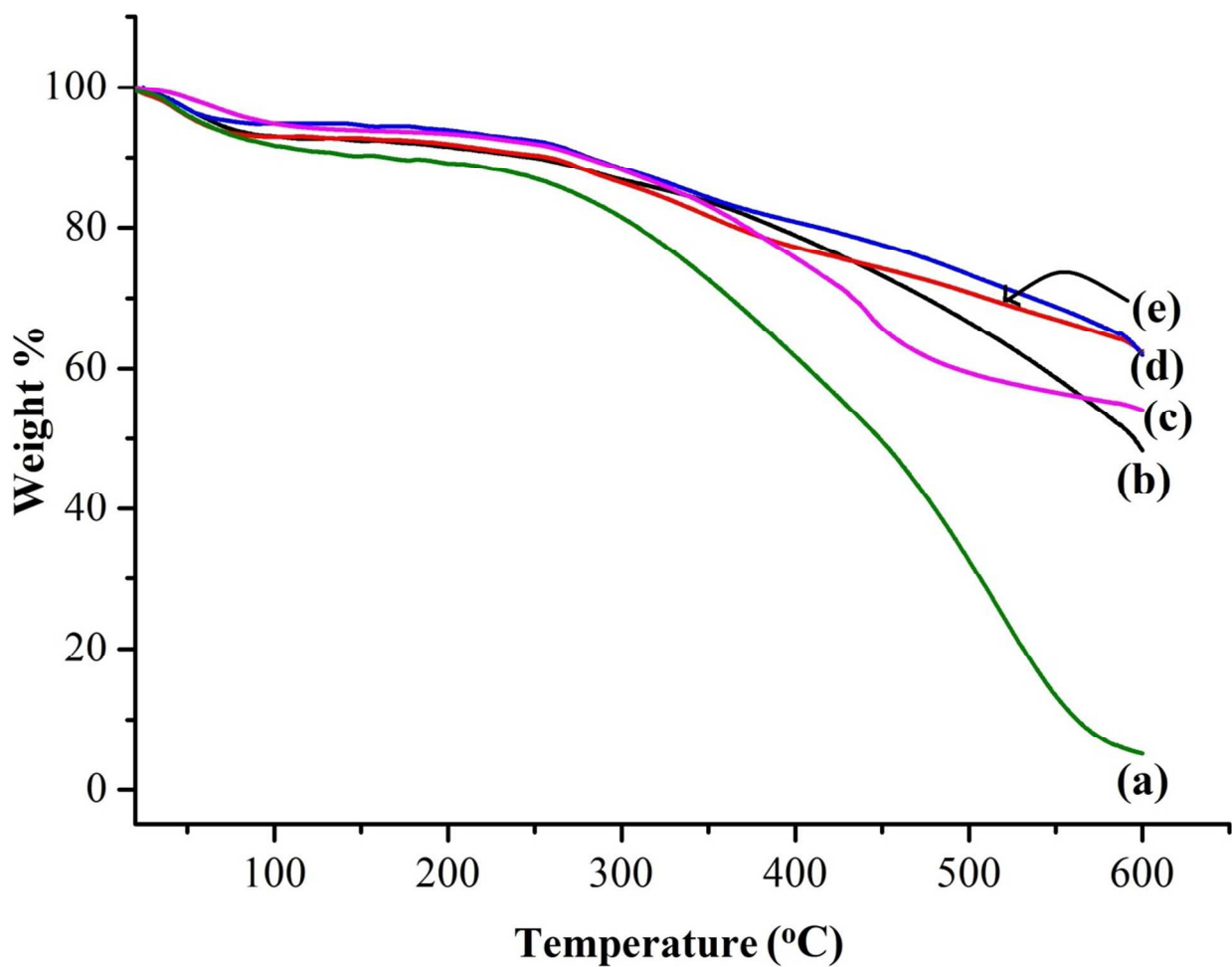
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391 **Fig. 1** The FT-IR spectrum of (a) SiO₂ nanoparticle, (b) Ag nanoparticles, (c) SiO₂/PANI-

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Ag nanocomposites.

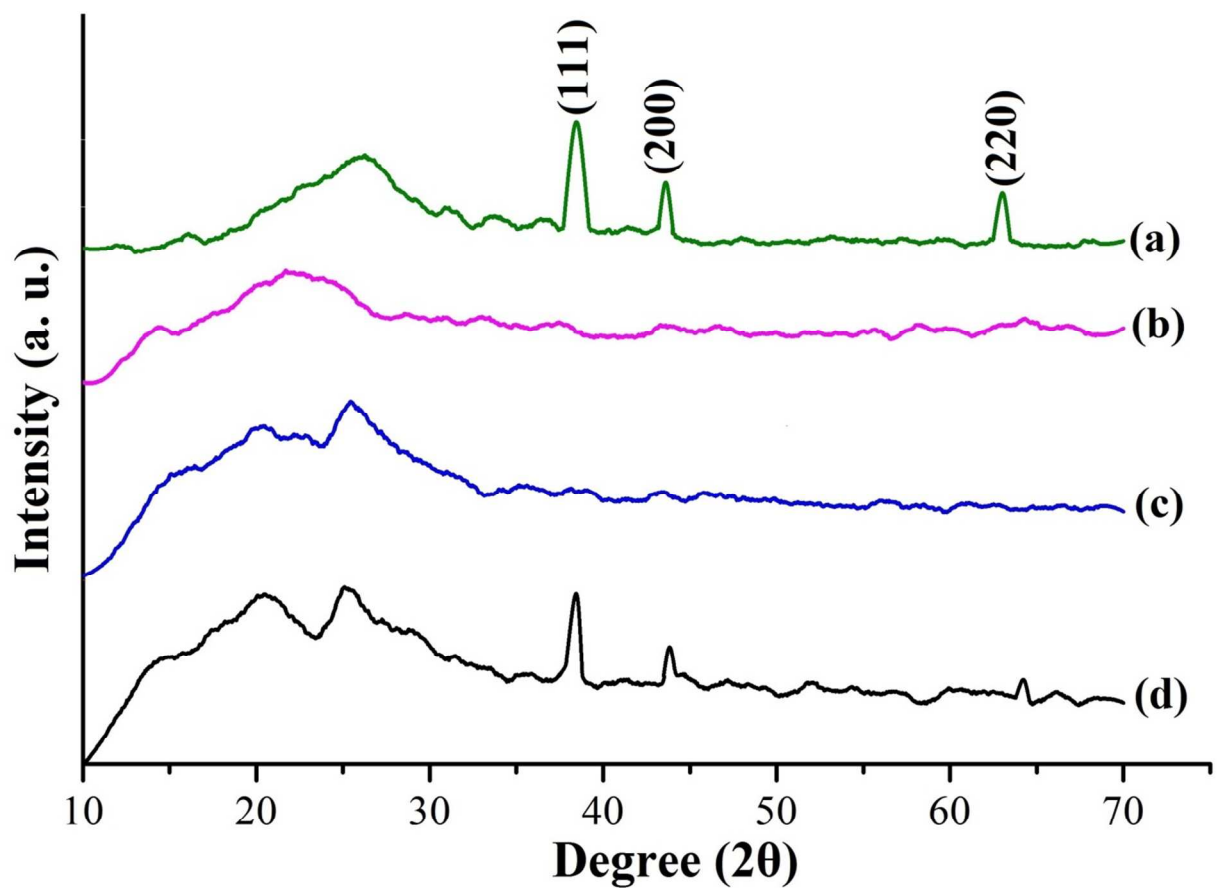


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Fig. 2 TGA analysis of (a) PANi, (b) SiO₂/PAni (5%), (c) SiO₂/PAni (7%), (d) SiO₂/PAni-Ag (5%), (e) SiO₂/PAni-Ag (7%).



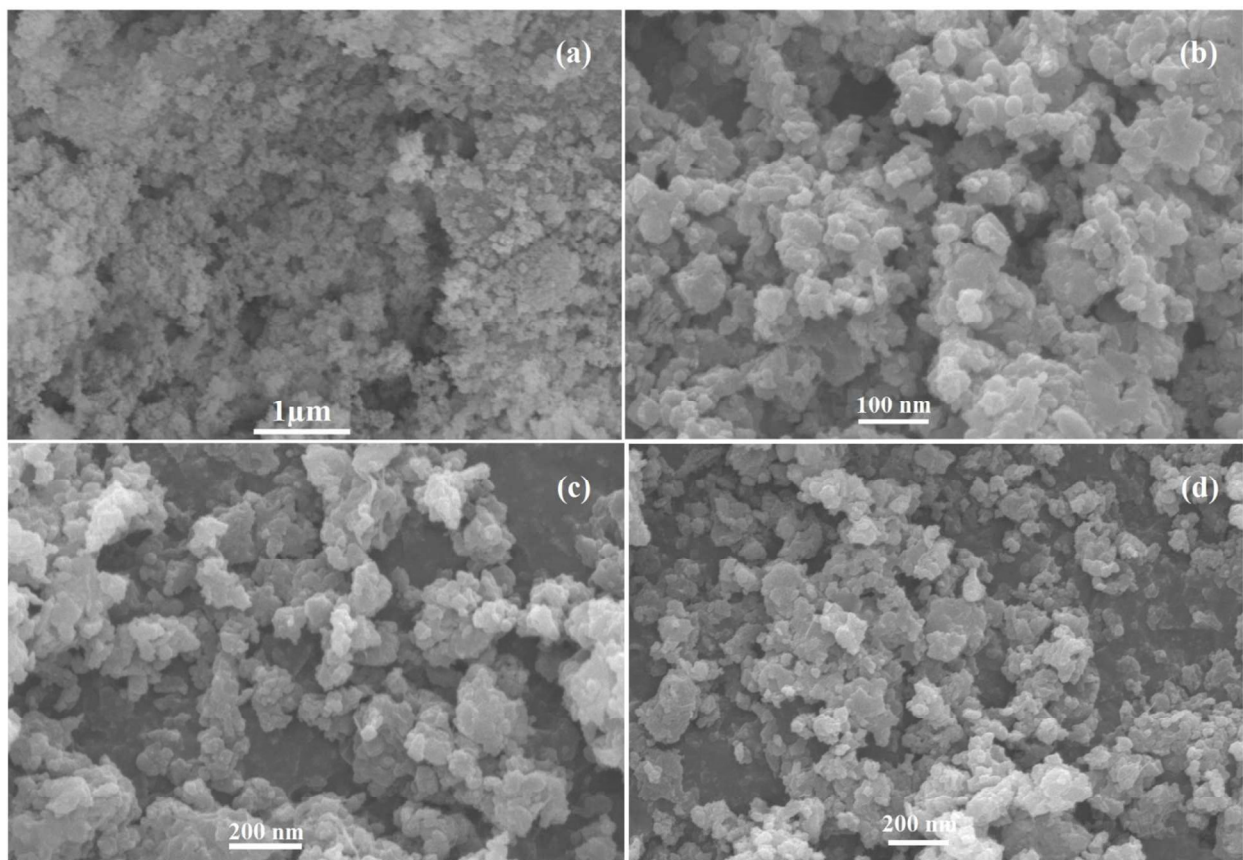
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Fig. 3 XRD patterns of (a) Ag nanoparticles, (b) SiO₂ nanoparticles, (c) PANi, (d) SiO₂/PANI-Ag core-shell nanocomposites.

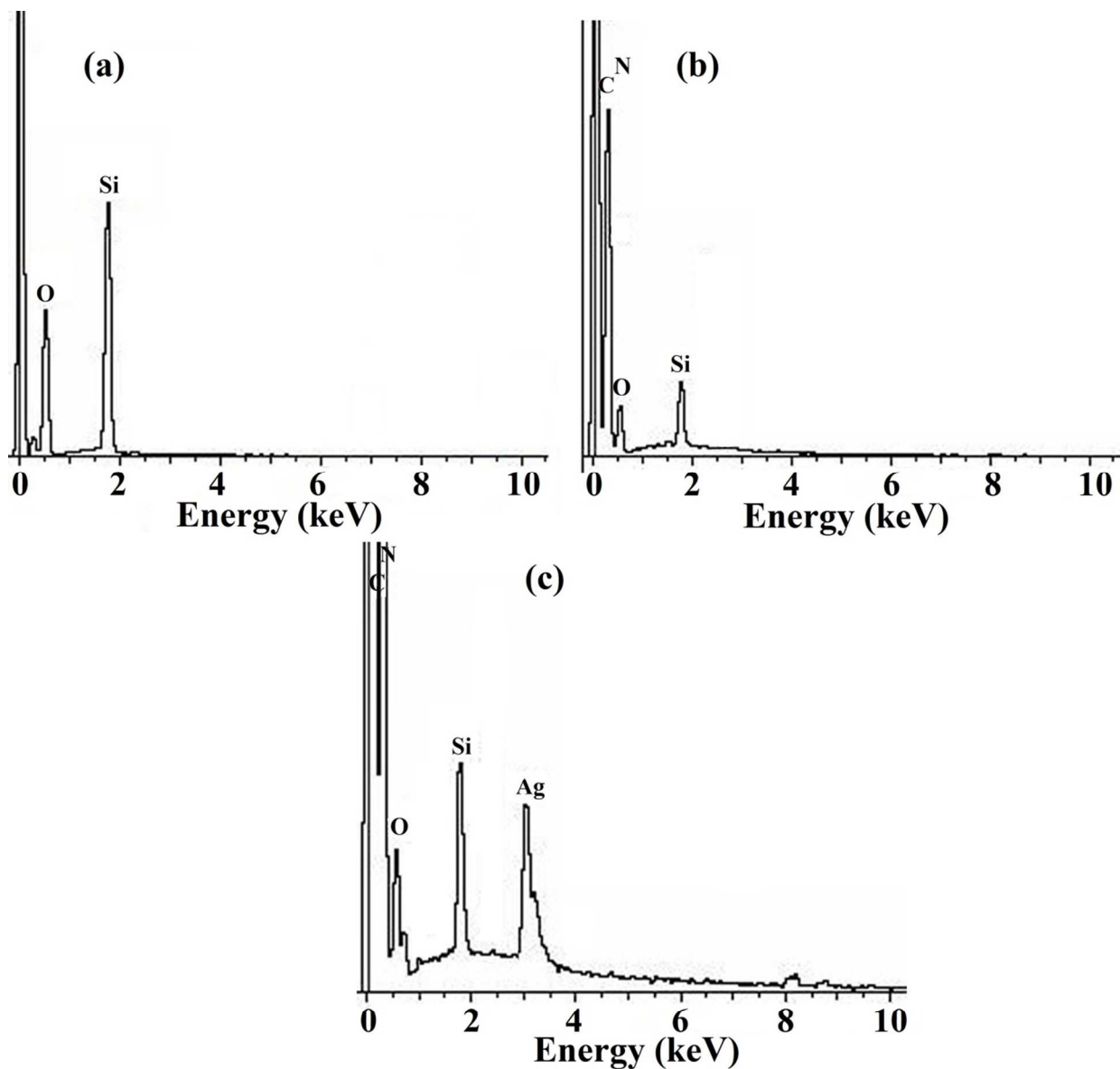


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Fig. 4 SEM image of (a) SiO₂ nanoparticles, (b) PANi, (c) SiO₂/PANi core-shell nanocomposites, (d) SiO₂/PANi-Ag nanocomposites.

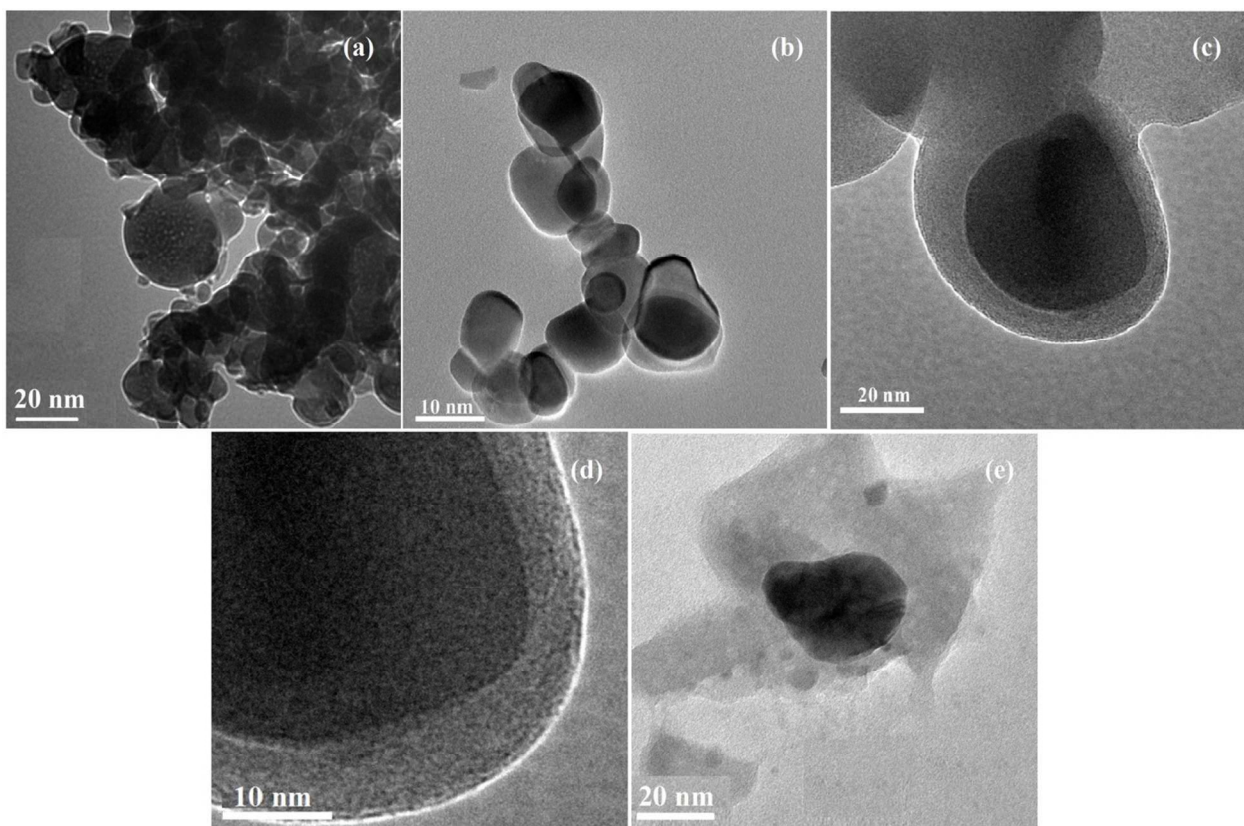


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404 **Fig. 5** EDX image of (a) SiO₂, (b) SiO₂/PAni core-shell and (b) SiO₂/PAni-Ag core-shell

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nanocomposites.



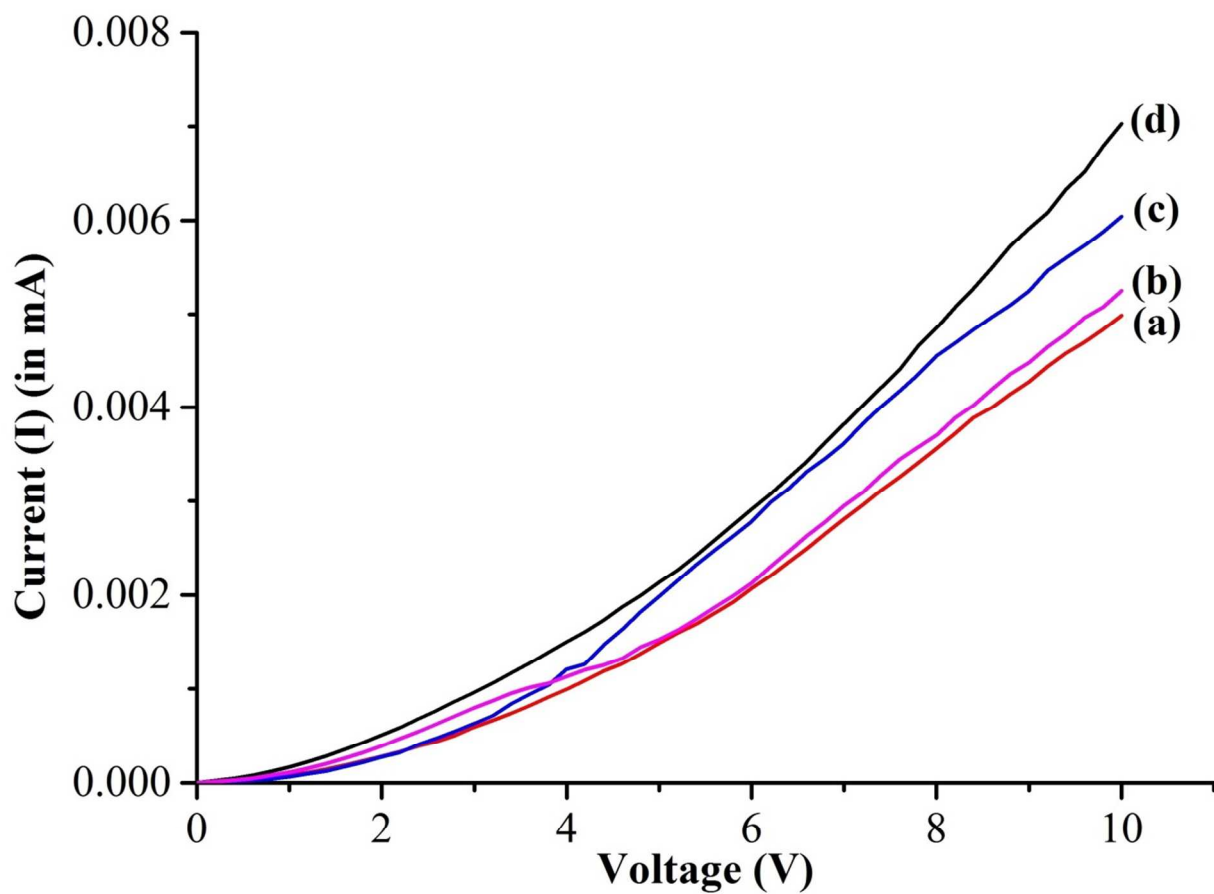
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Fig. 6 TEM images of (a) SiO₂ nanoparticles, (b) Ag nanoparticles, (c) SiO₂-PANI core-shell nanocomposites, (d) SiO₂-PANI core-shell nanocomposites at higher magnification, (e) SiO₂/PANI-Ag core-shell nanocomposites.

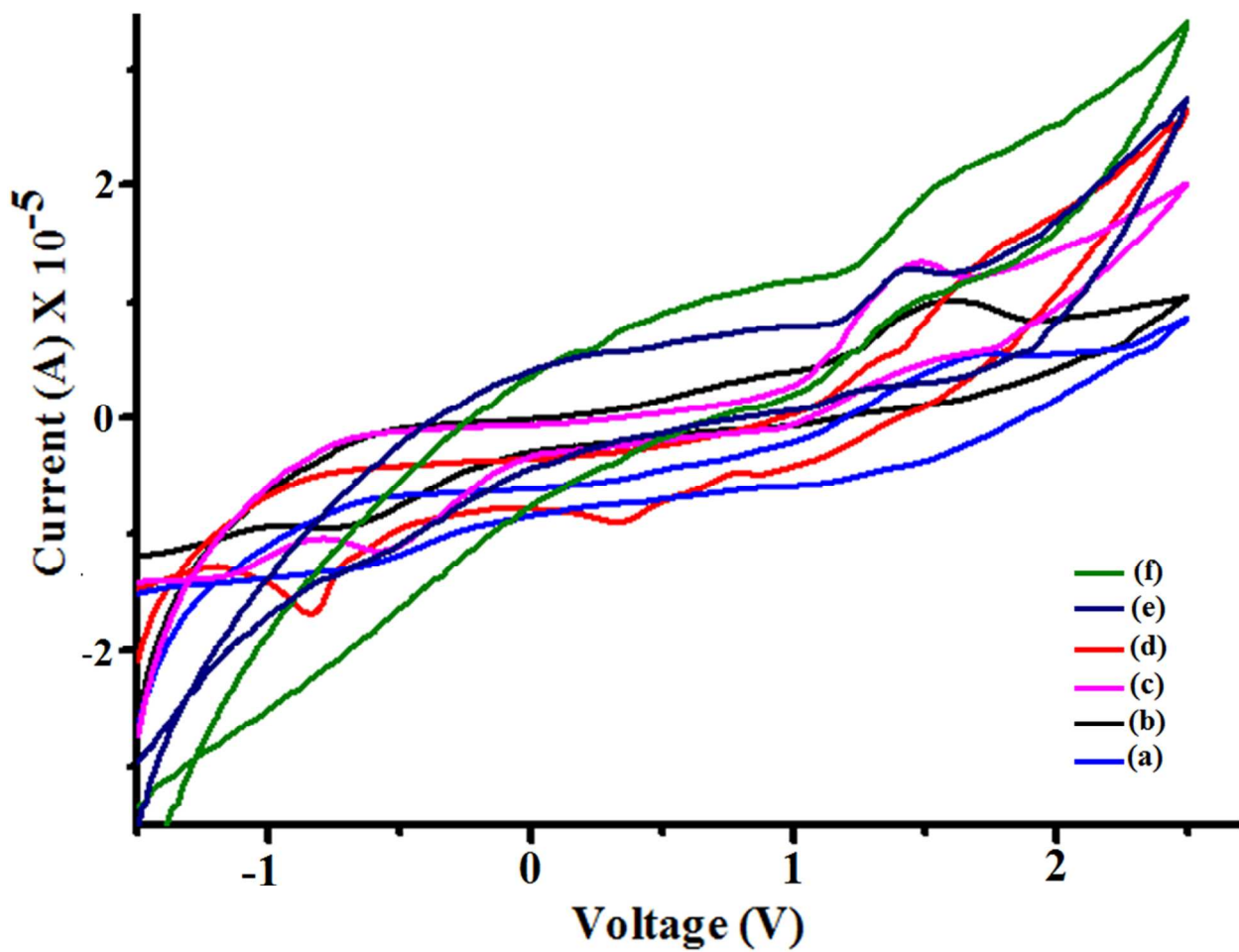


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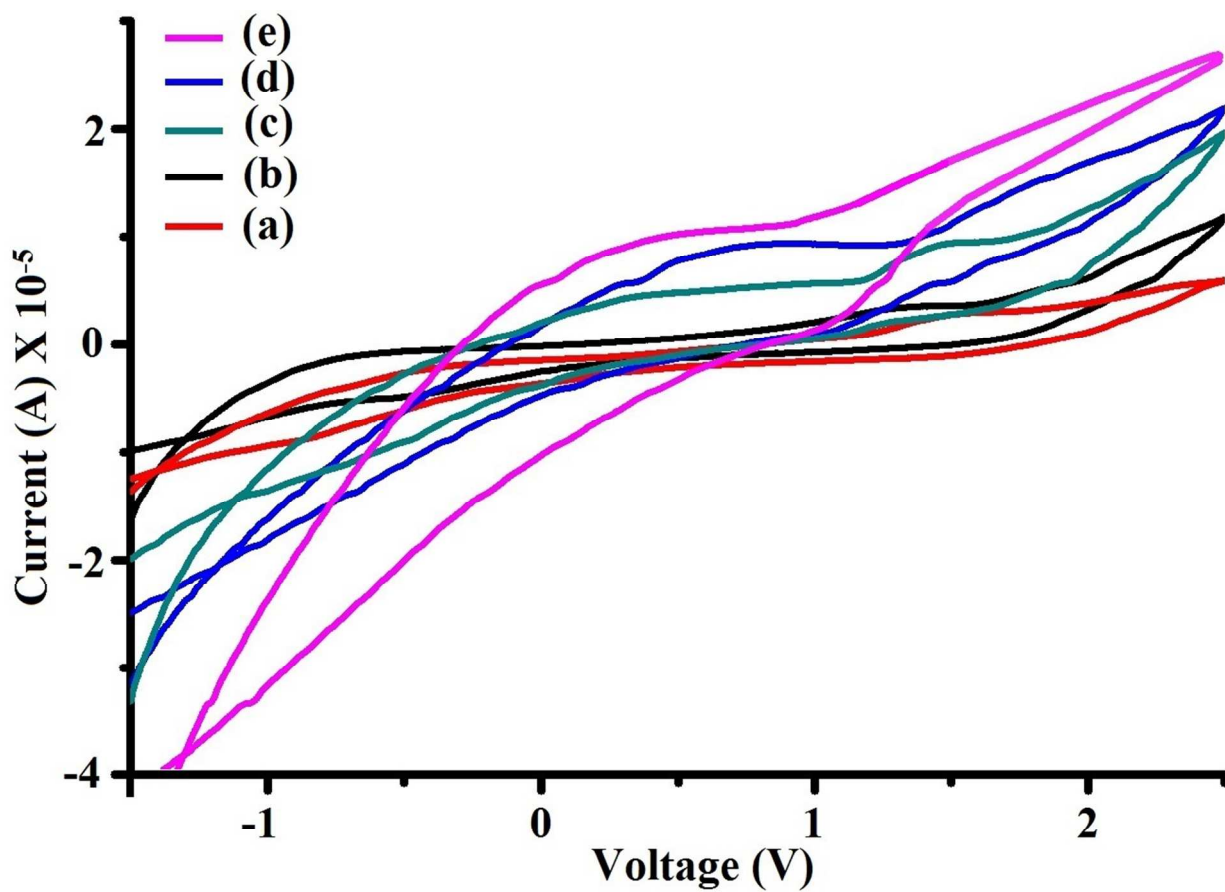
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Fig. 7 The current-voltage relationships for (a) PANi, (b) SiO₂/PAni-Ag (3%), (c) SiO₂/PAni-Ag (5%), (d) SiO₂/PAni-Ag (7%).



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414 **Fig. 8** C-V graphs of (a) PANi, (b) SiO₂/PANi (1%) (c) SiO₂/PANi (3%), (d) SiO₂/PANi415 (5%), (e) SiO₂/PANi (7%), (f) SiO₂/PANi (10%) core-shell nanocomposites.



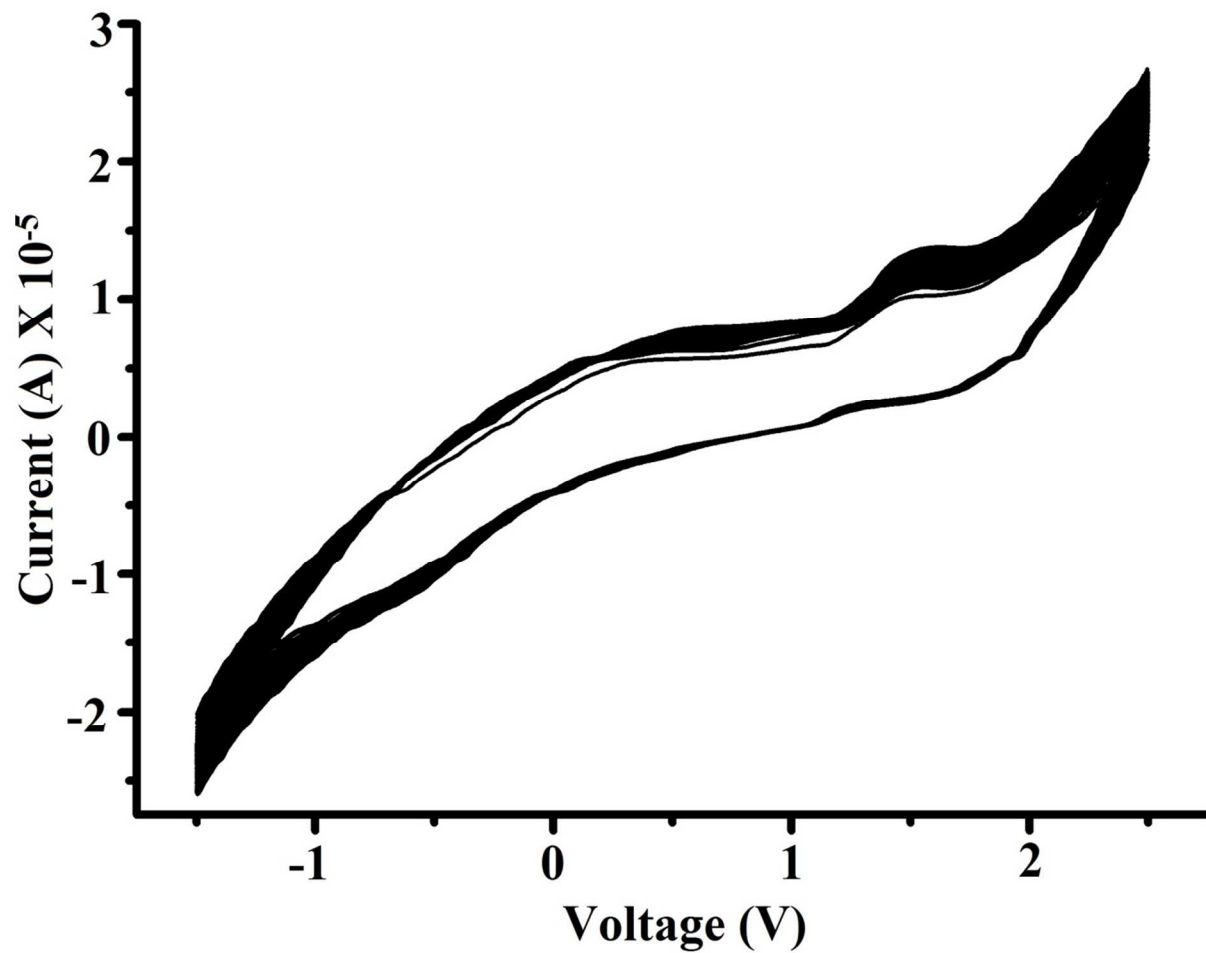
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417 **Fig. 9** C-V graphs of (a) SiO₂/PAni-Ag (1%) (b) SiO₂/PAni-Ag (3%), (c) SiO₂/PAni-Ag

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(5%), (d) SiO₂/PAni-Ag (7%), (e) SiO₂/PAni-Ag (10%) core-shell nanocomposites.

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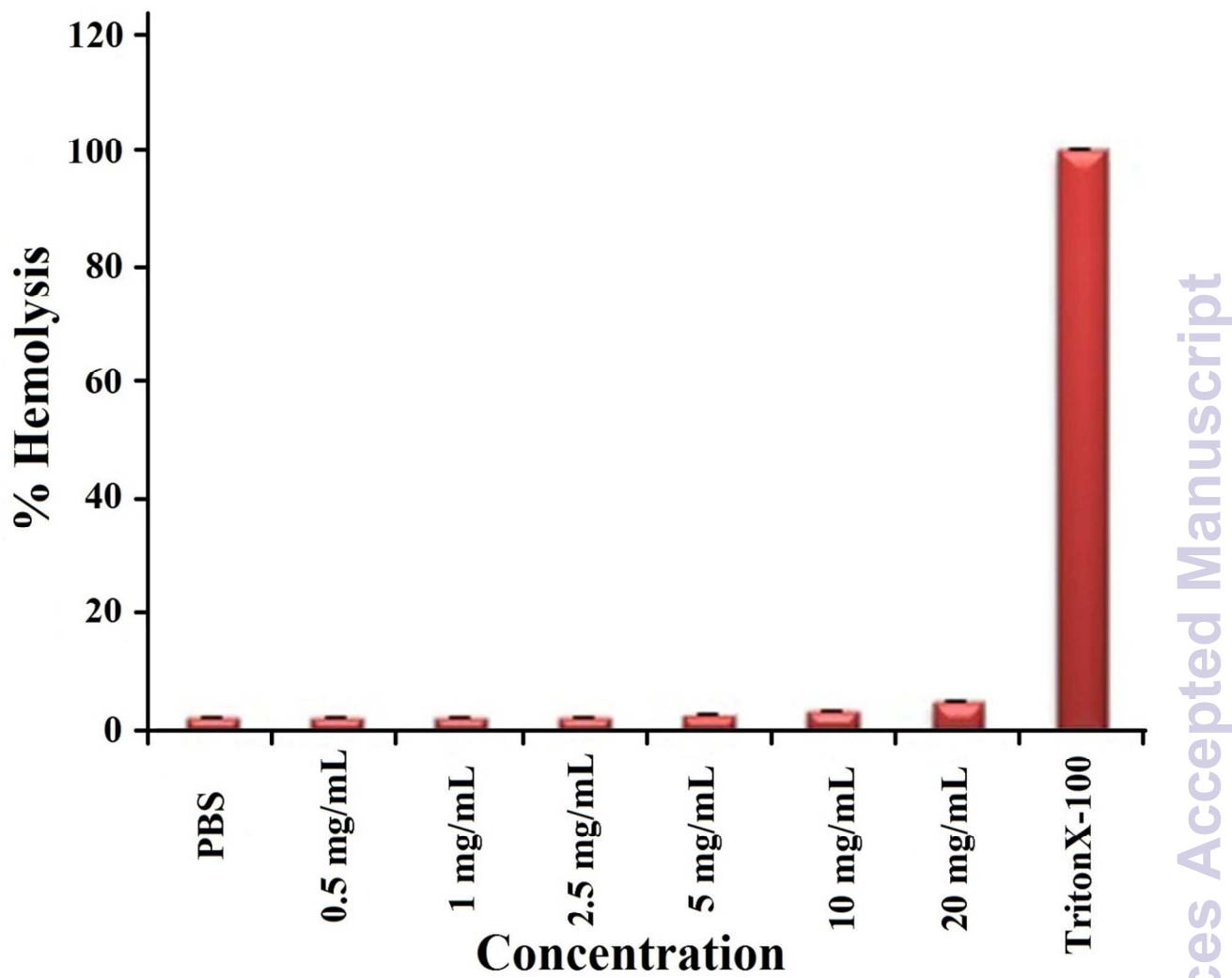


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Fig. 10 C-V analysis of SiO₂/PAni-Ag core-shell nanocomposites upto 100 repeated cycles.



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Fig. 11 Haemolytic activity of SiO₂/PAni-Ag core-shell nanocomposites.

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431 **Table 1.** Electrochemical data of the PANi and SiO₂/PANi nanocomposites, E_{HOMO} and E_{LUMO}
 432 signify the highest occupied molecular orbital (HOMO) energy level and lowest unoccupied
 433 molecular orbital (LUMO) energy level respectively for the PANi and SiO₂/PANi
 434 nanocomposites.

Sample	$\phi_{\text{onset}}^{\text{ox}}/E_{\text{HOMO}}$	$\phi_{\text{onset}}^{\text{red}}/E_{\text{LUMO}}$	E _g ^{eV} (eV)
PAni	1.62/-6.33	-0.38/-4.32	2.01
SiO ₂ /PANi			
1%	1.11/-5.82	-0.81/-3.90	1.92
3%	1.14/-5.85	-0.36/-4.35	1.50
5%	0.94/-5.65	-0.49/-4.22	1.43
7%	1.22/-5.93	-0.11/-4.6	1.33
10%	1.30/-6.01	-0.02/-4.73	1.32

435 ^a onset oxidation potential

436 ^b onset reduction potential

437 ^c electrochemical band gap

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447 **Table 2.** Electrochemical data of SiO₂/PAni-Ag nanocomposites, E_{HOMO} and E_{LUMO} signify the
 448 highest occupied molecular orbital (HOMO) energy level and lowest unoccupied molecular
 449 orbital (LUMO) energy level respectively for SiO₂/PAni nanocomposites.

Sample	$\phi_{\text{onset}}^{\text{ox}}/E_{\text{HOMO}}$	$\phi_{\text{onset}}^{\text{red}}/E_{\text{LUMO}}$	E _g ^{eV} (eV)
SiO ₂ /PAni-Ag			
1%	1.76/-6.47	-0.12/-4.59	1.88
3%	1.76/-6.47	0.29/-4.99	1.47
5%	1.20/-5.91	-0.19/-4.90	1.43
7%	1.16/-5.87	-0.14/-4.57	1.30
10%	1.35/-6.06	-0.09/-4.80	1.26

450 ^a onset oxidation potential

451 ^b onset reduction potential

452 ^c electrochemical band gap

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463 **Table 3.** DC-electrical conductivity of PANi, SiO₂/PANi and SiO₂/PANi-Ag nanocomposites with
 464 varying the amount of SiO₂, Ag content.

Core-shell nanocomposites	SiO ₂ -content (%)	Resistivity, ρ (Ωcm) ($\times 10^3$)	Conductivity, σ (Scm^{-1}) ($\times 10^{-3}$)
PAni	0	6.61	0.15
SiO ₂ /PANi	1	2.06	0.48
SiO ₂ /PANi-Ag	1	0.89	1.12
SiO ₂ /PANi	3	0.51	1.96
SiO ₂ /PANi-Ag	3	0.44	2.27
SiO ₂ /PANi	5	0.49	2.04
SiO ₂ /PANi-Ag	5	0.43	2.32
SiO ₂ /PANi	7	0.24	4.16
SiO ₂ /PANi-Ag	7	0.19	5.26
SiO ₂ /PANi	10	0.21	4.76
SiO ₂ /PANi-Ag	10	0.17	5.88

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