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## Environmentally Benign, Recyclable *nano* Hollandite and Metal Intercalated *nano* Hollandites for Hydrogen Sulfide removal

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Hollandite, a 2x2 tunnel structure, octahedral molecular sieve (OMS-2) with Ba<sup>2+</sup> as counter cation was prepared and characterized by PXRD, IR, SEM, EDX, HRTEM, XPS and TG-DTA techniques. HRTEM of the hollandite showed the particles to be nanorods (< 20 nm) and the SAED confirmed its crystalline nature. Seven metal intercalated hollandites were prepared by ion exchange process. Metal intercalated hollandites were analyzed by ICP-OES. The metal ion intercalation was in the range of 1.09 to 2.99% by weight. Order of increasing concentrations of divalent metal ions on intercalation is : Co < Zn < Ni < Cu < Cd < Pb. The observed order is an indication of the exchanging ability of the divalent ion with Ba<sup>2+</sup> ion of the hollandite in aqueous medium. Thermal analysis of H<sub>2</sub>S passed hollandite showed the hollandite to be a superior scrubber over birnessite. Maximum scrubbing of H<sub>2</sub>S (27%) was observed for copper intercalated hollandite and a minimum (22%) for zinc intercalated hollandite. The involvement of copper in scrubbing is important because of the Cu<sup>2+</sup>---SH<sub>n</sub>, a typical soft-soft interaction. Overall H<sub>2</sub>S scrubbing ability followed the order: birnessite (18%) < hollandite (20%) < metal ion intercalated hollandite (27%) and the scrubbing efficiency is remarkable for a dry scrubber under laboratory conditions. Hydrogen sulfide scrubbed hollandite showed relatively poor recyclability compared to intercalated material. Intercalated metal ion increases structural integrity, stabilizes the active metal against reduction and increases the scrubbing ability. The scrubber is recyclable by a simple heating process. The scrubber is

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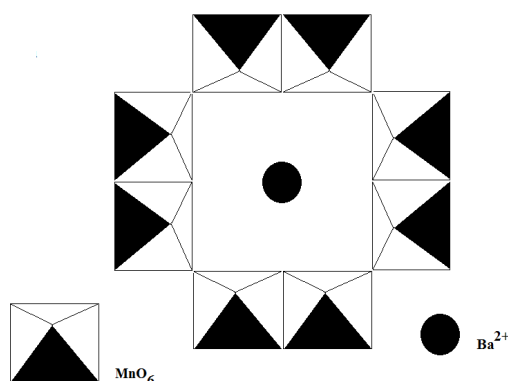
benign to nature as it's a synthetic analogue of naturally occurring marine nodule type of material.

**Key words:** Hollandite; intercalation; hydrogen sulfide; XPS; HRTEM; TG-DTA.

## 1. Introduction

Marine nodule type manganese oxides have been intensively investigated during the past few decades because of their economic values and potential applications. Synthetic analogues of the manganese oxides are used as ion sieves, molecular sieves and catalysts.  $\text{MnO}_2$  is used as a cathode material in lithium batteries. The material possesses excellent cation-exchange, electrochemical and magnetic properties.<sup>1-4</sup> Manganese dioxide shows two important structural variations: (i) layer structure and (ii) tunnel structure. Best example of a layer structured manganous oxide is birnessite, which is a two dimensional layered structure that contains edge-shared  $\text{MnO}_6$  octahedra with cations and water molecules occupying the interlayer region.<sup>5</sup> The tunnel structured are, pyrolusite (1×1), ramsdellite (1×2), spinel (1×3), hollandite (2×2), romanecchite (2×3), and todorokite (3×3).<sup>6</sup> The tunnel structures are formed by edge and corner shared  $\text{MnO}_6$  octahedra which are also known as Octahedral molecular sieve (OMS) materials.<sup>7-14</sup> Manganese oxide octahedral molecular sieve materials have straight channels inside the structure with varying pore diameters. OMS-2 material consists of two sheets of  $\text{MnO}_6$  octahedra joined at edges to form the 2x2 tunnel structure. One of the unique properties of the OMS-2 type of materials is the presence of both,  $\text{Mn}^{3+}$  and  $\text{Mn}^{4+}$  ions in the manganese oxide framework. The average oxidation state of manganese in the OMS-type of materials is in the range of 3.40 to 3.99, depending on the  $\text{Mn}^{4+}/\text{Mn}^{3+}$  ratio. Cryptomelane-type (2x2) manganese oxide (OMS-2) materials are a group of important OMS materials because of their use in catalysis, separations, chemical sensors, and batteries. The pore diameter of the tunnel is about 0.46 nm. The tunnels (2x2) are partially filled with uni- or divalent cations

such as  $\text{Ba}^{2+}$  (hollandite),  $\text{K}^+$  (cryptomelane),  $\text{Na}^+$  (manjiroite), and  $\text{Pb}^{2+}$  (coronadite).<sup>15</sup> Reflux method was mostly used to prepare bulk OMS-2 materials.<sup>16,17</sup> Other metal cations can be introduced into the tunnel by post ion exchange<sup>18-20</sup> or into the framework by



substitution during synthesis.<sup>21,22</sup> Substitution in framework sites of manganese octahedral molecular sieves by other metals is possible by doping the original reactant solutions with cations.<sup>23,24</sup> The doping amounts of foreign cations are limited because these cations are not easy to incorporate into the structures. Coal gasification and chemical industries generate large amounts of hydrogen sulfide gas which is responsible for a serious environmental problem.  $\text{H}_2\text{S}$  gas is very corrosive, and highly toxic for the living organisms even at relatively low concentration. Furthermore, it is a constituent of sulphate aerosol which decreases droplet radius in cloud. The increase of aerosol concentration weakens rain fall and hydrological cycle in polluted cloud. Several commercial techniques are available for the removal of  $\text{H}_2\text{S}$ , including aqueous  $\text{NaCl}$ , iron-based sorbents, activated carbon,  $\text{FeOOH}$ ,  $\text{Fe}_2\text{O}_3$ , metal oxides, sewage sludge, and aqueous solutions. However, the efficiency and the environment friendliness of the scrubbers need improvement. Various types of manganous oxides have been tested as scrubbers.<sup>25-27</sup> Therefore, there is a need for research on efficient and economical methods for the removal of  $\text{H}_2\text{S}$  which is inimical to the environment.<sup>28,29</sup> Use of geochemically benign

manganous oxide variants as scrubbers of environmentally highly toxic hydrogen sulfide is doubly beneficial. CO<sub>2</sub> scrubbing by birnessite has been reported from our laboratory.<sup>30</sup> In this paper we report the scrubbing of H<sub>2</sub>S by hollandite and metal intercalated hollandites.

## 2. Experimental

### 2.1. Preparation of Hollandite

Ba(CH<sub>3</sub>COO)<sub>2</sub> (3.19 g) was added to 100 mL of 0.3 M Mn(CH<sub>3</sub>COO)<sub>2</sub> solution, followed by 100 mL of 0.2 M KMnO<sub>4</sub> solution (All the reagents are Sigma-Aldrich > 99 % purity). The solution was refluxed at 100 °C for 24 h, and the product was filtered, washed, and dried at 80 °C for 24 hours.

### 2.2. Intercalation of metal ions into hollandite

Metal ions such as Fe<sup>2+</sup>, Co<sup>2+</sup>, Ni<sup>2+</sup>, Cu<sup>2+</sup>, Zn<sup>2+</sup>, Cd<sup>2+</sup> and Pb<sup>2+</sup> (Iron(II) and nickel(II) sulfates and other five metal(II) chlorides (Sigma-Aldrich > 99 % purity) have been intercalated into hollandite by the following ion exchange process. Hollandite (1 g) was added with 50 ml of 1M M<sup>2+</sup> solution in 100 ml Erlenmeyer flask, and kept for 24 hours. Then the solution was stirred vigorously with a magnetic stirrer for about 24 hours and filtered, washed with distilled water (10mL aliquot) for five times, and dried at 60 °C in an air oven. The obtained sample was dried and powdered. The hollandite and the intercalated samples were analyzed by inductively coupled plasma / optical emission spectrometry (ICP-OES) technique and the results are shown in the Table 1. The metal ion intercalation has been in the range of 1.09 to 2.99%. The results of three different intercalations agreed within +/- 0.1%. The order of concentrations of divalent metal ions on intercalation is : Co < Zn < Ni < Cu < Cd < Pb. The observed order is an indication of the exchanging ability of the particular divalent ion with Ba<sup>2+</sup> ion of the hollandite in aqueous medium.

### 2.3. H<sub>2</sub>S scrubbing process

FeS ( $4.0 \times 10^{-4}$  kg) was taken in a round bottom flask. A solution of H<sub>2</sub>SO<sub>4</sub> (4N) was added slowly through a separating funnel to the round bottom flask. The released H<sub>2</sub>S gas was passed through a tightly packed MnO<sub>2</sub> ( $5.0 \times 10^{-4}$  kg) in a glass tube of 0.5cm diameter. The outlet was connected to a solution containing Pb(NO<sub>3</sub>)<sub>2</sub> to precipitate excess H<sub>2</sub>S as PbS. The reaction was carried out in a fume cupboard. A blank run was carried out without the scrubber and the precipitate of PbS obtained was used as the standard to evaluate the scrubbing ability.

#### 2.4. Characterization

The hollandite, metal intercalated hollandite and H<sub>2</sub>S scrubbed hollandite were characterized by powder X-ray diffraction (PXRD), infrared spectroscopy (IR), Scanning electron microscopy (SEM), Energy dispersive X-ray spectroscopy (EDX), high resolution transmission electron microscopy (HRTEM), X-ray photo electron spectroscopy (XPS) and thermo gravimetric and differential thermal analysis (TG-DTA) techniques. Powder X-ray diffraction patterns were recorded with a BRUKER-D8, X-ray diffractometer using Cu-K $\alpha$  radiation (Step time = 30.6, Voltage = 40 kV, current = 30 mA, range of  $2\theta = 5.0$  to  $80.0^\circ$ ). Avatar Nicolet 360 FT-IR spectrophotometer (range: 4000 and  $450\text{ cm}^{-1}$ , 120 scans at a resolution of  $1.0\text{ cm}^{-1}$ ) was used for recording the infrared spectra of the complexes. Spectra were taken as potassium bromide discs of the compounds (A KBr disk of the compound will contain about 1 to 2% of the active compound. In a typical disk preparation,  $\sim 4 \times 10^{-6}$  kg of the active compound was thoroughly ground to a fine powder with a pestle & mortar and then subsequently,  $\sim 4 \times 10^{-3}$  kg of KBr (spectrometric grade, Sigma-Aldrich) from the oven was added and ground to a fine powder. Thin disk was pressed from the powder with a hydraulic pellet press. Thermogravimetric experiments were carried out in nitrogen atmosphere using alumina as reference with NETZSCH STA 449F3 instrument. The heating rate of the furnace was fixed at  $10^\circ\text{C}$  per minute and the samples were heated up to  $700^\circ\text{C}$  by taking approximately  $1 \times 10^{-5}$  kg of the sample in a

platinum crucible for each thermogravimetric experiment. Samples for electron spectroscopy were treated with dry absolute alcohol for dehydrating the samples. Scanning electron micrographs of the samples were recorded with JSM-6390 VERSION 1.0 & FEI QUANTA FEG 200 - high resolution scanning electron microscope (Resolution of secondary electron image (SEI) 3.0 nm at an accelerating voltage 30 kV; 8.0 nm at accelerating voltage 3 kV, Magnification x5 (WD 48 mm) to x300,000 (WD 8 mm)) HRTEM was obtained by employing Tecnai F20, with an accelerating voltage of 200kV, PTP Resolution: 0.24 nm and information limit: 0.14 nm. Metal ion concentrations were determined by PERKIN ELMER OPTIMA 5300 DV ICP-OES. About  $20 \times 10^{-5}$  kg of the sample was accurately weighed and dissolved in HCl-HNO<sub>3</sub> (3:1v/v) and made up to 50 mL in a standard measuring flask for ICP-OES analysis. A calibrant was used for each element under analysis.

### 3. RESULTS AND DISCUSSION

#### 3.1. Characterization of hollandite and metal intercalated hollandites

The hollandite is denoted as BaH. The metal ion intercalated hollandites are denoted as FeH, CoH, NiH, CuH, ZnH, CdH, PbH. On intercalation of Fe<sup>2+</sup> in to hollandite, the sample changed its color from grey black to yellow, indicating its decomposition.

PXRD pattern for hollandite is shown in Figure 1a. The pattern shows characteristic signals at 28.4, 28.7, 37.40 and 41.5° and matched with the JCPDS: 78-0962 corresponding to Ba-MnO<sub>2</sub> hollandite.<sup>24</sup> Figure 1 b, c, d, e, f, g and h correspond to PXRD patterns of FeH, CoH, NiH, CuH, ZnH, CdH, PbH respectively. Fe intercalate of hollandite showed a different PXRD pattern altogether indicating the decomposed hollandite.

Representative scanning electron micrographs of the metal intercalated hollandites, CuH, ZnH, CdH and PbH are shown in Figure 2a - d respectively. The Cu, Zn, Cd intercalated samples are of identical morphology and they appeared as microfibrils. The Pb intercalated hollandite showed the particles to be of spherical shape. EDX spectra of CuH, ZnH, CdH and PbH are shown in Figure 3a - d which confirmed the presence of Mn, Ba, and the corresponding intercalated metal ions. Representative HRTEM and Selected area electron diffraction (SAED) of the hollandites are shown in Figure 4. TEM showed the particles to be nanorods ( $< 20$  nm). Figure 4d showed the SAED spectrum of the hollandite which indicated its crystalline nature.

Representative XPS spectrum of PbH is shown in Figure 5. A survey spectrum is shown in Figure 5 a. Presence of Mn, Ba, Pb and oxygen apart from carbon is evident from the survey scan. Figure 5 b shows the presence of Mn  $2p_{1/2}$  signal at 654.17 eV which can be attributed to a mixture of Mn<sup>4+</sup> and Mn<sup>3+</sup> species.<sup>31</sup> Figure 5 c shows the presence of barium  $3d_{5/2}$  signal at 779.68 eV in agreement with those reported earlier (<http://srdata.nist.gov/xps/Default.aspx>). The corresponding signal of barium in the free state is 781.02 eV indicating marginal increase in positive charge on barium. Figure 5 d depicts the presence of Pb<sup>2+</sup>  $4f_7$  signal at 139.35 (138.03) eV and the values in parentheses correspond to the free ion binding energies. The observed values indicate minor changes in the electronic environments of the cations. However, XPS unambiguously established the intercalation of the metal ions in to the hollandite framework.

Figure 6 a - h show TG-DTA for BaH, FeH, CoH, NiH, CuH, ZnH, CdH and PbH respectively. DTA shows a clear exothermic reaction around 340°C in all the compounds (red curve) due to initial absorption of nitrogen gas in to the tunnels and a concomitant steep loss of mass was observed in the TG curve (blue curve) as the first step of thermal decomposition. The second step was a flat range without any conspicuous loss of mass up to 500 °C. In the temperature range of 550 - 700 °C, a further loss of ~ 2% mass was

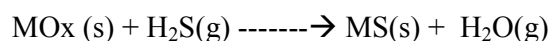


observed due to a loss of lattice trapped gas and carbon. The total mass loss up to 700 °C was approximately 18 % which is far higher than the loss observed in birnessite in a similar thermal decay. The observation shows that the 2×2 tunnelled hollandite retains more water molecules in the lattice than the birnessite.<sup>30</sup>

Figure 7 a, b depicts the representative infrared spectra (IR) of the hollandite and copper intercalated hollandites. The broad band at 3405 cm<sup>-1</sup> is due to stretching vibrations of interlayer water molecules. The band at 1617 cm<sup>-1</sup> is assigned to the bending vibration of H<sub>2</sub>O and structural OH groups. Bands around 734(sh), 553, 558 (split bands) cm<sup>-1</sup> are characteristic bands of manganese oxides with tunnel structure<sup>30</sup>. The observation signifies the stabilization of the tunnel structure after metal ion intercalation. IR of copper intercalated sample shows four medium intensity bands at 513, 602, 622 and 630 cm<sup>-1</sup> indicating Cu-O interactions. The band around 734 cm<sup>-1</sup> is retained in the intercalated spectrum suggesting the existence of the parent structure.

### 3.2. Characterization of H<sub>2</sub>S passed hollandite and metal intercalated hollandite

Dark brown colored metal intercalated hollandites on passing H<sub>2</sub>S turned yellow and on aging, the yellow color faded out. The most intense yellow colour was observed in the case of hollandite and retained the colour for a longer duration compared to other metallated hollandites. The yellow colored FeH sample on passing H<sub>2</sub>S turned black indicating the formation of ferrous sulfide. The color changes confirmed the absorption of H<sub>2</sub>S gas by the material and the concomitant chemical change. A typical sulphidation reaction takes place as follows:



Presence of additional metal ion increases structural integrity and stabilizes the active metal against reduction and increases the scrubbing ability. Figure 8 a-h shows the PXRD patterns for H<sub>2</sub>S passed BaH, FeH, CoH, NiH, CuH, ZnH, CdH, and PbH respectively. PXRD patterns excepting that of iron intercalate were identical. In general, PXRD patterns

of H<sub>2</sub>S passed hollandite and metal intercalated hollandites were different. Examination of the PXRD patterns with the JCPDS data bank revealed the presence of a mixture of BaS, MnS and the intercalated metal sulfide. For the Co, Ni, Cu, Pb, Zn, Cd intercalated hollandites, the major products on H<sub>2</sub>S scrubbing are BaS (24.1, 28.0, 39.9°; JCPDS: 750896), MnS (27.6, 45.9, 54.4°; JCPDS: 894953), MnS<sub>2</sub>, the hauerite (29.2, 41.8, 49.5°; JCPDS: 762052) and the free hollandite as evident from the PXRD patterns. Iron intercalated hollandite on passing H<sub>2</sub>S, showed the formation of a mixture of BaS (24.1, 28.0, 39.9°; JCPDS: 750896), Fe<sub>3</sub>S<sub>2</sub> (25.9, 33.2, 52.0; JCPDS: 370475).

Figure 9 shows SEM images of H<sub>2</sub>S passed metal intercalated hollandites. A comparison of the SEM of the metal intercalated images with those of the H<sub>2</sub>S passed materials show a clear change in morphology. However, nano wires of unreacted material were also visible on the surface. Figure 10 a-d displays EDX spectra of H<sub>2</sub>S passed CuH, ZnH, CdH and PbH respectively. The spectra confirmed the presence of S, Mn, Ba, and the presence of intercalated divalent ions of Cu, Zn, Cd and Pb in the sample other than oxygen and carbon. The peak intensity of sulphur represents the extent of absorption of H<sub>2</sub>S gas by the material. HRTEM of the H<sub>2</sub>S passed hollandite (Figure 11) showed spherical particles of varying homogeneity smeared over the nanorods. The unreacted hollandite appeared as nano rods and the small percentage of spherical particles are the metal sulfides. The SAED showed the highly crystalline nature of the scrubbed hollandite lattice.

Figure 12 shows the XPS spectrum of H<sub>2</sub>S passed BaH. A survey spectrum showed (Figure 12 a) the presence of Mn, Ba, S and oxygen apart from carbon. Figure 12 b showed the presence of Mn 2p<sub>1/2</sub> signal at 655.45 eV which is higher than the binding energy of BaH indicating the anion surcharged atmosphere around the cations after scrubbing. The binding energy is still close to a formal oxidation state of ~+3.5 for manganese. Figure 12 c shows the presence of barium 3d<sub>5/2</sub> signal at 783.25 eV which is a

significant increase compared to that in BaH and is attributable to the increased anionic population around the cations after scrubbing. Figure 12 d shows the presence of signal due to sulphur 2p at 165.06 eV which confirmed the scrubbing of H<sub>2</sub>S gas by hollandite. Figure 13 is a representative XPS spectrum of cobalt intercalated hollandite. Mn 2p<sub>3/2</sub> and 2p<sub>1/2</sub> signals appear almost unaltered at 640.25 and 651.98 eV with reference to free MnO<sub>2</sub>. Cobalt 2p<sub>3/2</sub> signal appears at 780.87eV significantly higher than the BE observed for CoS<sub>2</sub> / CoS.<sup>32</sup> The observation indicates the involvement of cobalt in scrubbing H<sub>2</sub>S. Table 2 gives the important binding energies associated with the XPS emissions. Binding energies associated with Mn 2p<sub>3/2</sub> electrons showed a width of 640.25-643.73 eV and the 2p<sub>1/2</sub> electrons 650.94-655.45eV. The Ba 3d<sub>5/2</sub> binding energies are observed in the range of 779.12 to 783.25eV. The maximum binding energy change was observed for hollandite after scrubbing. Other than hollandite, among the metal intercalated, cadmium and lead intercalates showed higher binding energies. The increased binding energies reflect an increased anionic environment of OH<sup>-</sup>, SH<sup>-</sup>, H<sub>2</sub>S and H<sub>2</sub>O in the solid.

A plot of binding energies of Ba 3d<sub>5/2</sub> and Mn 2p<sub>1/2</sub> electrons against the intercalated 'd<sup>n</sup>' electron configuration is shown in Figure 14. The d electron configuration of manganese was taken as 3.5 as its formal oxidation state is known to be between +3 and +4 in hollandite. For both the Ba 3d<sub>5/2</sub> and Mn 2p<sub>1/2</sub> electrons the binding energies show a maximum for hollandite. Ba 3d<sub>5/2</sub> electron binding energy shows a minimum for nickel (d<sup>8</sup>) intercalate and shows a second maximum for copper (d<sup>9</sup>). The involvement of copper in scrubbing is important because of the fact that Cu<sup>2+</sup> ion is a typical soft acid. Similarly, sulfur is a typical soft base. Therefore, the interaction between the two species, Cu<sup>2+</sup>---SH<sub>2</sub> is highly favoured due to a typical 'soft-soft' interaction. Mn 2p<sub>1/2</sub> electron binding energy shows a second maximum for copper intercalate as well. A plot of sulphur 2p electron binding energies against the d electron configurations is shown in Figure 15. The sulphur 2p electron binding energy variation depicted in the figure is found to show a maximum

for nickel intercalates ( $d^8$ ). The observed trends clearly establish the involvement of intercalated metal in the scrubbing process. Increased binding energies of the electrons of different cations unequivocally prove the fact that the residual positive charge on them increase. The observed binding energies of the electrons of the cations reflect an environment of predominantly of  $\text{OH}^-$  and  $\text{SH}^-$  anions and  $\text{H}_2\text{S}$  and *in situ* generated  $\text{H}_2\text{O}$  molecules. The conclusion is supported by reports of large binding energies associated with the cations of highly ionic salts such as  $\text{BaH}_2$ ,  $\text{Ni}(\text{OH})_2$ ,  $\text{Ni}(\text{NO}_3)_2$  and  $\text{NiCl}_2$ .<sup>33-38</sup> Cadmium and lead intercalates showed larger binding energy increases compared to other first row transition elements. However, there was no pronounced difference in scrubbing ability of the heavier cation intercalated hollandites. The XPS investigation establishes the involvement of intercalated metal ions in the scrubbing process. A highly surcharged atmosphere of anions around the cations in the solid material is also evident from the analysis. Formation of metal sulfides and existence of strong ionic interactions must be responsible for the increased binding energies of the electrons.

Figures 16 a-h show TG-DTA for  $\text{H}_2\text{S}$  passed  $\text{BaH}$ ,  $\text{FeH}$ ,  $\text{CoH}$ ,  $\text{NiH}$ ,  $\text{CuH}$ ,  $\text{ZnH}$ ,  $\text{CdH}$  and  $\text{PbH}$  respectively. An abrupt weight loss was observed at  $120^\circ\text{C}$  as an exothermic decomposition followed by another dramatic steep loss of mass at  $515^\circ\text{C}$ . The second thermal decay was a clear exoergic process. An increased mass loss observed due to the scrubbed  $\text{H}_2\text{S}$  and the  $\text{H}_2\text{O}$  generated as a by product of absorption of  $\text{H}_2\text{S}$ . In general,  $\text{MnO}_2$  is stable up to  $400^\circ\text{C}$ , in the present analysis 40% of mass loss was observed clearly indicating the loss of trapped  $\text{H}_2\text{O}$  and  $\text{H}_2\text{S}$ .

Figure 17 shows the IR spectrum of  $\text{H}_2\text{S}$  passed hollandite. Spectrum shows well resolved bands at  $507$  and  $616\text{ cm}^{-1}$  suggesting the change in Mn-O vibrations on  $\text{H}_2\text{S}$  scrubbing. A weak band at  $435\text{ cm}^{-1}$  indicates Mn-S stretching vibration.

### 3.3. Recyclability and efficiency

Grey coloured hollandite and the metal intercalated hollandites on H<sub>2</sub>S scrubbing turned yellow, on aging, in air the colour faded gradually. Maximum scrubbing of H<sub>2</sub>S (27%) was observed for zinc intercalated hollandite and a minimum (22%) for lead intercalated hollandite. Present investigation showed the following overall H<sub>2</sub>S scrubbing ability: birnessite (18%) < hollandite (20%) < metal ion intercalated hollandite (27%), the efficiencies being accurate to 1% for three experimental determinations for mass increase after scrubbing process. The scrubbing efficiency is remarkable for a dry scrubber under laboratory conditions. The material was stable below 400 °C. The scrubbing ability of the scrubber was restored on heating the used material at 110 °C for an hour. The major products on H<sub>2</sub>S scrubbing are BaS (24.1, 28.0, 39.9°; JCPDS: 750896), MnS (27.6, 45.9, 54.4°; JCPDS: 894953), MnS<sub>2</sub>, the hauerite (29.2, 41.8, 49.5°; JCPDS: 762052) as evident from the PXRD patterns. The intercalated metals are present only to a maximum of 2.99 % (Pb) by weight. The intercalated metal sulfides have not been identified by powder XRD indicating their relative miniscule percentage after scrubbing. Recycling of the scrubber indicated only a variation of 1% in their scrubbing ability after three cycles. The observation is a clear indication of the fact that the formation of metal sulfides did not affect the recyclability of the scrubbers. The metal intercalation process is aimed at improving the interaction between the scrubber and the runoff to be scrubbed at the molecular level. In addition, the metal intercalation process increases the structural integrity of the scrubber. Once the scrubber is intercalated with a metal ion, it reaches a saturation point beyond which further use of this material to scrub metal ions from waste waters will be relatively less.

#### 4. Conclusions

Hollandite was prepared and characterized by various physical techniques. XPS survey scan showed the presence of Mn, Ba. BE specific scans showed the presence of Mn 2p<sub>1/2</sub> signal at 654.2 eV and Mn 2p<sub>3/2</sub> at 642.6eV which can be attributed to a mixture of

Mn<sup>4+</sup> and Mn<sup>3+</sup> species. HRTEM of the hollandite showed it to be nanorods (< 20 nm) and the SAED showed the partial crystalline nature of the sample. On H<sub>2</sub>S scrubbing, the products were analyzed by PXRD, FTIR. The PXRD pattern accounted qualitatively for a mixture of S and MnS in birnessite and BaS, S and MnS in hollandite. Thermal analysis on hollandite after the scrubbing process revealed an increased mass loss is due to the scrubbed H<sub>2</sub>S and the concomitant release of H<sub>2</sub>O as a product of sulphidation. The observation contrasted well with the thermal analysis pattern of birnessite which showed a much less loss of mass. The metal ion intercalation analyzed by ICP-OES showed the range of metal ion as 1.09 to 2.99%. The results of three different intercalations agreed within +/- 0.1%. The order of concentrations of divalent metal ions on intercalation is: Co < Zn < Ni < Cu < Cd < Pb. The observed order is an indication of the exchanging ability of the particular divalent ion with Ba<sup>2+</sup> ion of the hollandite in aqueous medium.

HRTEM images of the H<sub>2</sub>S passed intercalated hollandites showed the presence of spherical nanosized particles of the sulfides smeared over the unreacted hollandite nanorods. EDX indicates the presence of Ba, Mn and S in the H<sub>2</sub>S passed hollandite. All the intercalated hollandites showed mass losses ranging from 10-22% during thermal analysis. The initial phase of decomposition extended up to 400 °C. The decompositions are exothermic in nature. Thermal analysis of the hollandite showed a discrete steep decay at 340 °C and the total loss of mass was ~48%. The mass of precipitated PbS during the H<sub>2</sub>S scrubbing study showed a minimum for zinc intercalated hollandite (22%) and a maximum (27%) for lead intercalated hollandite. The present investigation showed the following overall H<sub>2</sub>S scrubbing ability: birnessite (18%) < hollandite (20%) < metal ion intercalated hollandite (27%) and the scrubbing efficiency is remarkable for a dry scrubber under laboratory conditions. Presence of additional metal ion increases pore structure integrity, reduces the probability of stable metal sulfide formation and stabilizes the active metal against reduction and improves the recyclability.

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Table 1 ICP-OES analytical data for hollandite and the intercalated hollandites

S.No		Weight in %		
		Ba	Mn	'M'
1	BaH	14.40	34.21	-
2	FeH*	4.95	0.10	27.45
3	CoH	14.50	31.19	1.09
4	NiH	17.18	45.10	1.83
5	CuH	11.00	26.91	2.05
6	ZnH	13.85	39.78	1.29
7	CdH	16.74	49.92	2.82
8	PbH	7.84	24.41	2.99

\* Formation of  $\text{Fe}_2(\text{SO}_4)_3$  on reaction with  $\text{MnO}_2$

Table 2 XPS binding energies (eV)

	Compound	Ba 3d <sub>5/2</sub>	Mn 2p <sub>3/2</sub>	Mn 2p <sub>1/2</sub>	<sup>s</sup> M 2p <sub>3/2</sub>	S 2p
1	BaH	779.68	642.57	654.17	---	---
2	H <sub>2</sub> S passed BaH <sup>#</sup>	783.25	643.60	655.45	---	165.06
3	H <sub>2</sub> S passed CoH	780.87	640.25	651.98	780.87	165.08
4	H <sub>2</sub> S passed NiH	779.12	640.96	652.87	851.32	166.15
5	H <sub>2</sub> S passed CuH	779.99	641.42	653.94	932.67	164.10
6	H <sub>2</sub> S passed ZnH	779.34	639.24	650.94	1022.79	164.06
7	H <sub>2</sub> S passed CdH	781.96	642.14	653.51	407.78 <sup>†</sup>	166.04
8	H <sub>2</sub> S passed PbH	782.09	643.73	655.04	139.35 <sup>*</sup>	165.53

<sup>s</sup>M = Co, Ni, Cu, Zn, Cd, Pb

<sup>#</sup> Turned dark yellow on passing H<sub>2</sub>S and retained the colour for the longest duration

<sup>†</sup> 3d<sub>5/2</sub>

<sup>\*</sup> Appeared as a single band for 4f<sub>7/2</sub> and 4f<sub>5/2</sub>

**Captions for figures**

- Figure 1 PXRD of (a) BaH, (b) FeH, (c) CoH, (d) NiH, (e) CuH, (f) ZnH, (g) CdH, (h) PbH
- Figure 2 SE micrographs of metal intercalated hollandite (a) CuH, (b) ZnH, (c) CdH, (d) PbH
- Figure 3 EDX spectra of metal intercalated hollandite (a) CuH, (b) ZnH, (c) CdH, (d) PbH
- Figure 4 HRTEM of BaH (a) 20nm, (b) 50nm, (c)100nm, and (d) SAED of BaH
- Figure 5 XPS survey spectrum of PbH (a) survey spectrum, (b) Mn, (c) Ba, (d) Pb
- Figure 6 TG and DTA curves of sample (a) BaH, (b) FeH, (c) CoH, (d) NiH, (e) CuH, (f) ZnH, (g) CdH (h) PbH
- Figure 7 FTIR spectra of (a) BaH, (b) CuH
- Figure 8 PXRD of H<sub>2</sub>S passed (a) BaH, (b) FeH, (c) CoH, (d) NiH, (e) CuH, (f) ZnH, (g) CdH, (h) PbH
- Figure 9 SEM images of H<sub>2</sub>S passed metal intercalated hollandite samples (a) CuH, (b) ZnH, (c) CdH, (d) PbH
- Figure 10 EDX spectrum of metal intercalated hollandite sample (a) CuH, (b) ZnH, (c) CdH, (d) PbH
- Figure 11 HRTEM image of H<sub>2</sub>S passed BaH (a) 20nm, (b) 50nm, (c) 100nm, and (d) SAED spectrum of BaH
- Figure 12 XPS of H<sub>2</sub>S passed BaH (a) survey spectrum of H<sub>2</sub>S BaH (b) Mn, (c) Ba, (d) S
- Figure 13 XPS spectrum of cobalt interacted hollandite on H<sub>2</sub>S scrubbing
- Figure 14 Ba 3d<sub>5/2</sub> and Mn2p<sub>1/2</sub> binding energy variations with intercalated d electron configuration

- Figure 15 S 2p binding energy variations with intercalated d electron configuration
- Figure 16 TG and DTA curves of H<sub>2</sub>S passed samples (a) BaH, (b) FeH, (c) CoH, (d) NiH, (e) CuH, (f) ZnH, (g) CdH, (h) PbH
- Figure 17 FTIR spectrum of H<sub>2</sub>S passed BaH

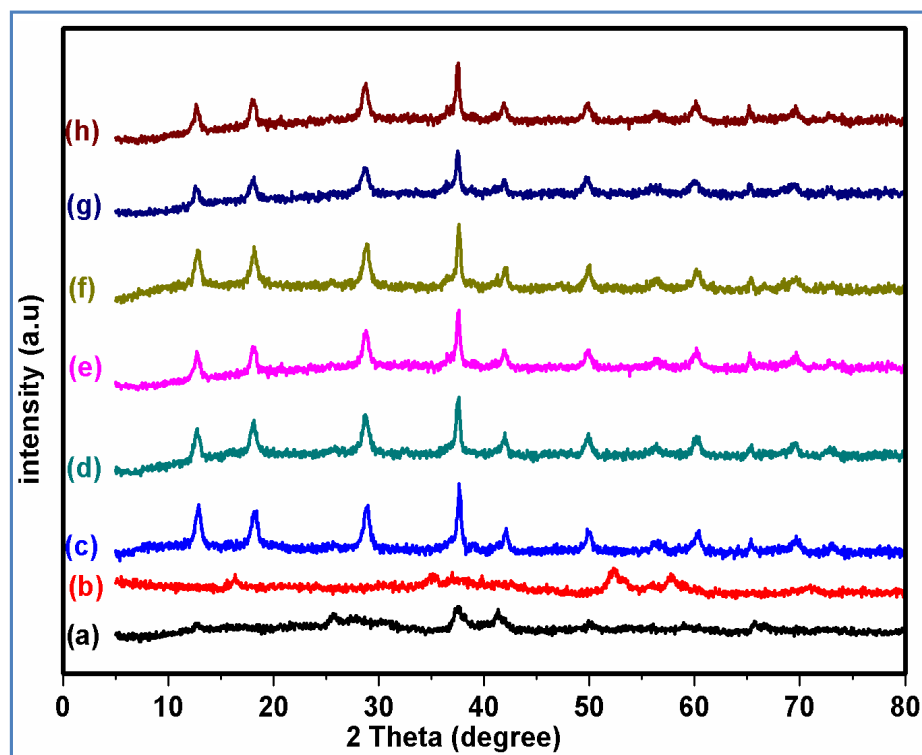


Figure 1

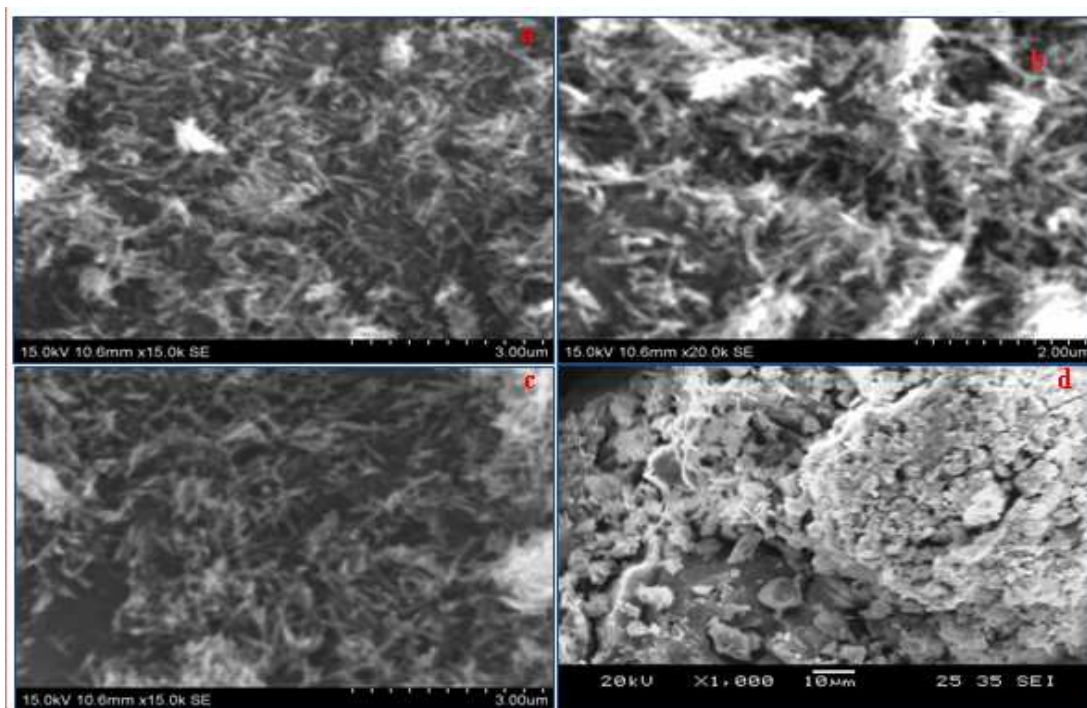


Figure 2



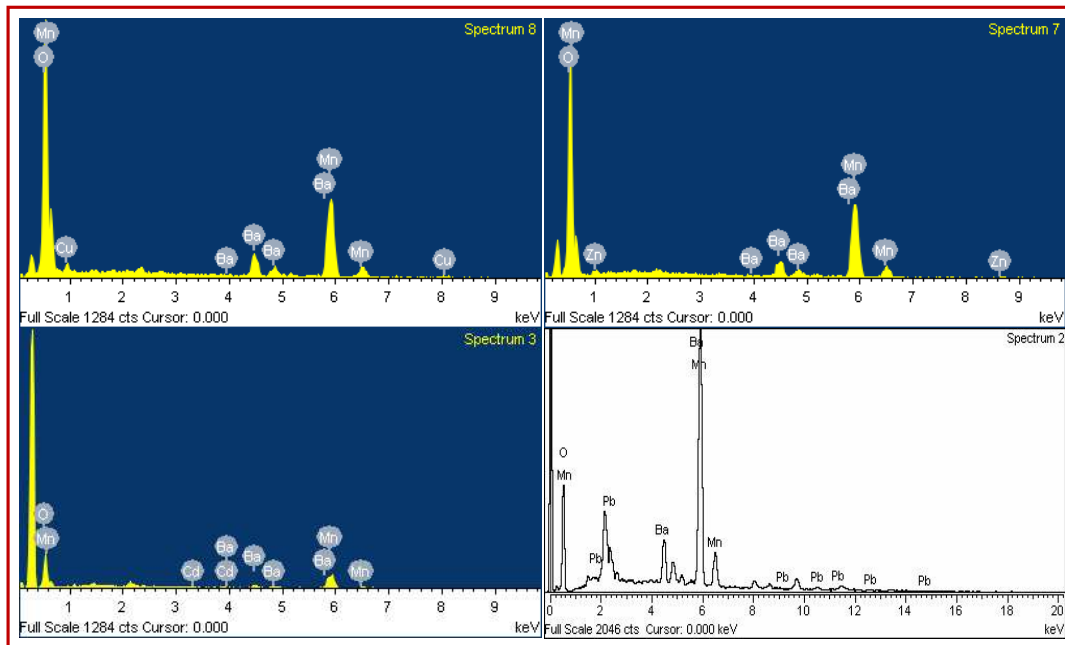


Figure 3

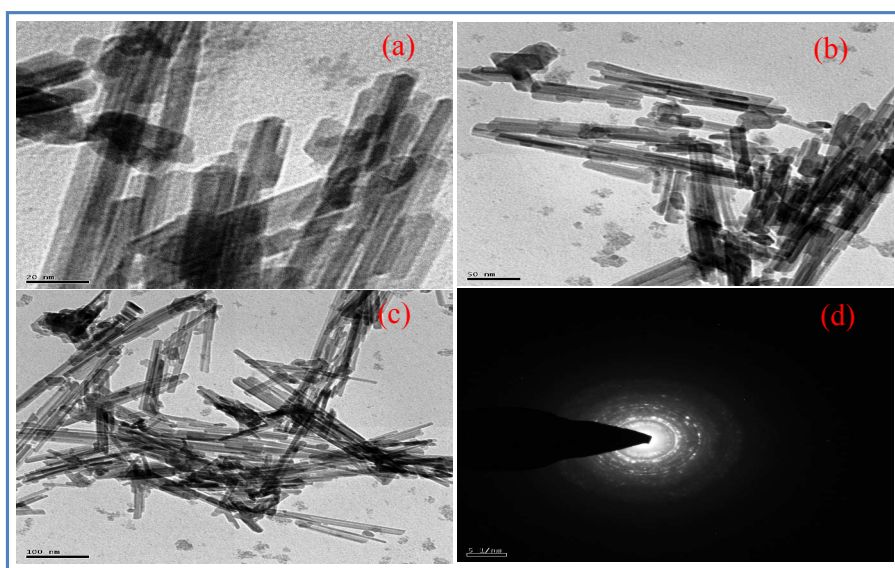


Figure 4

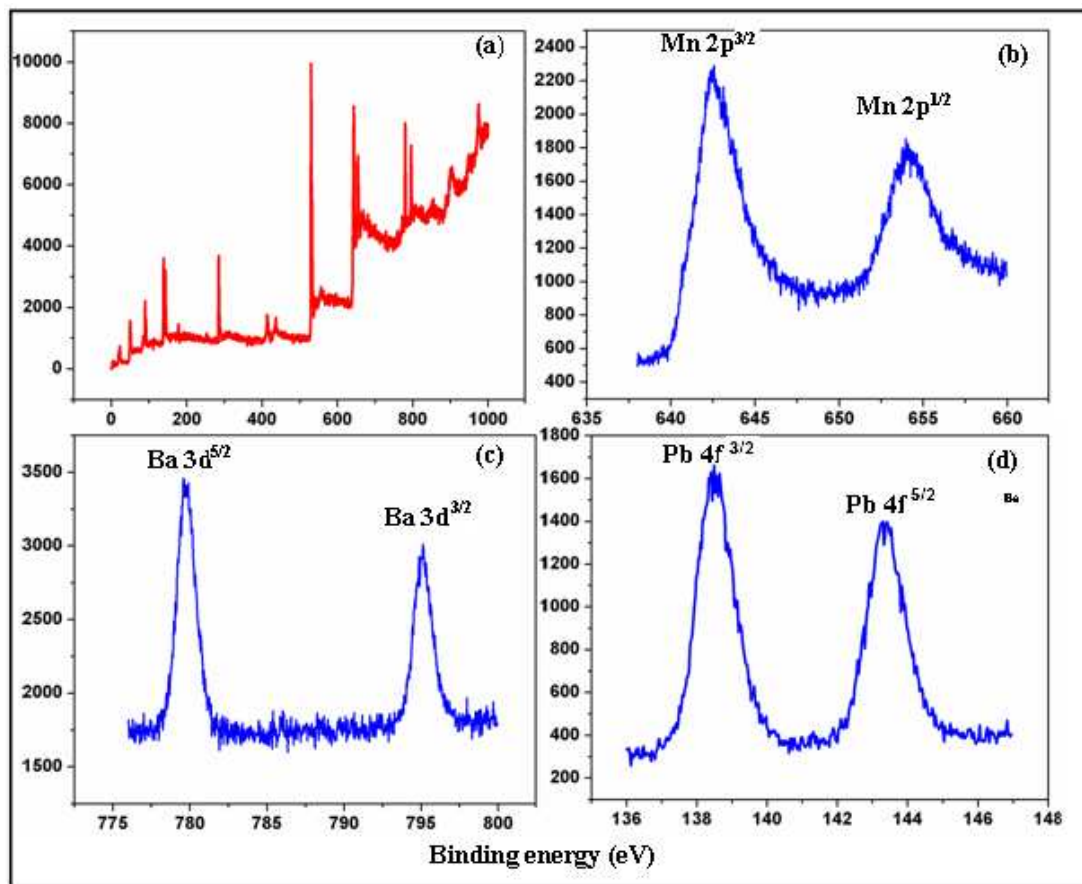


Figure 5

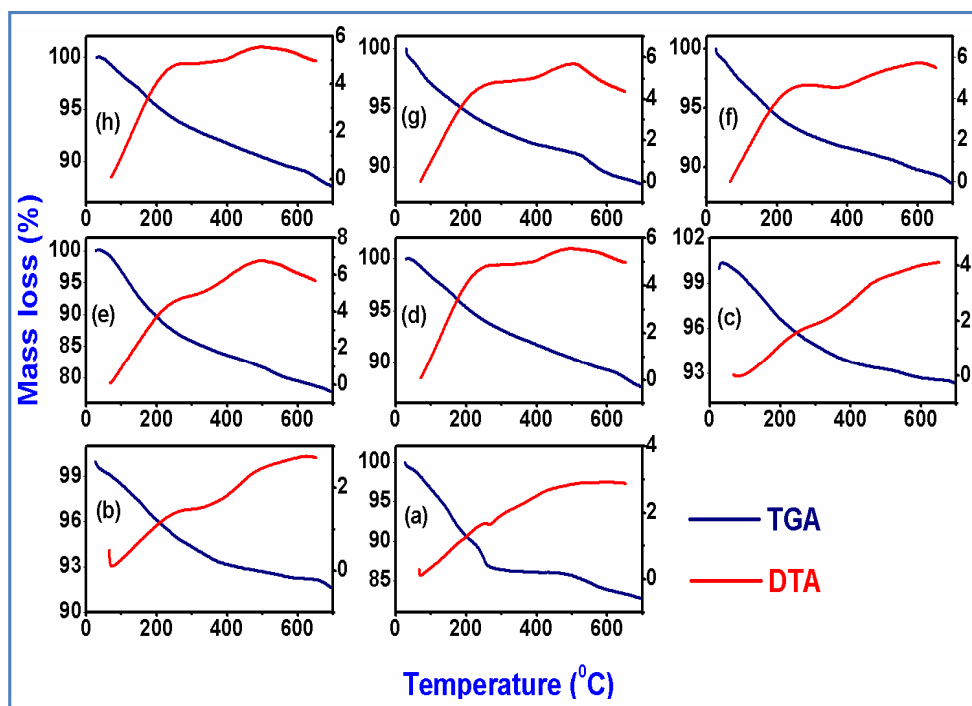


Figure 6

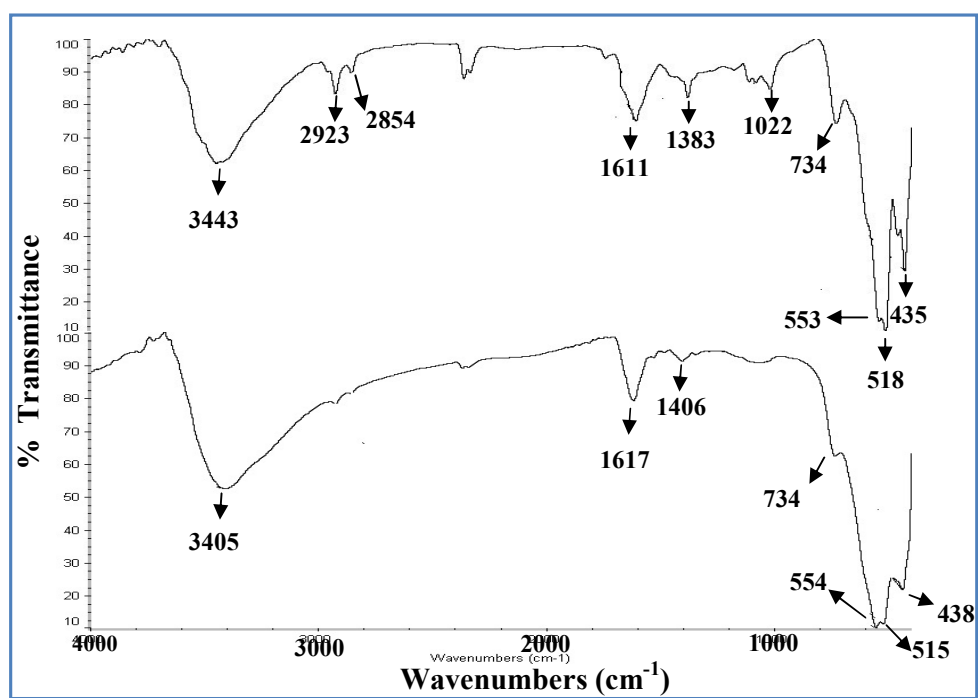


Figure 7

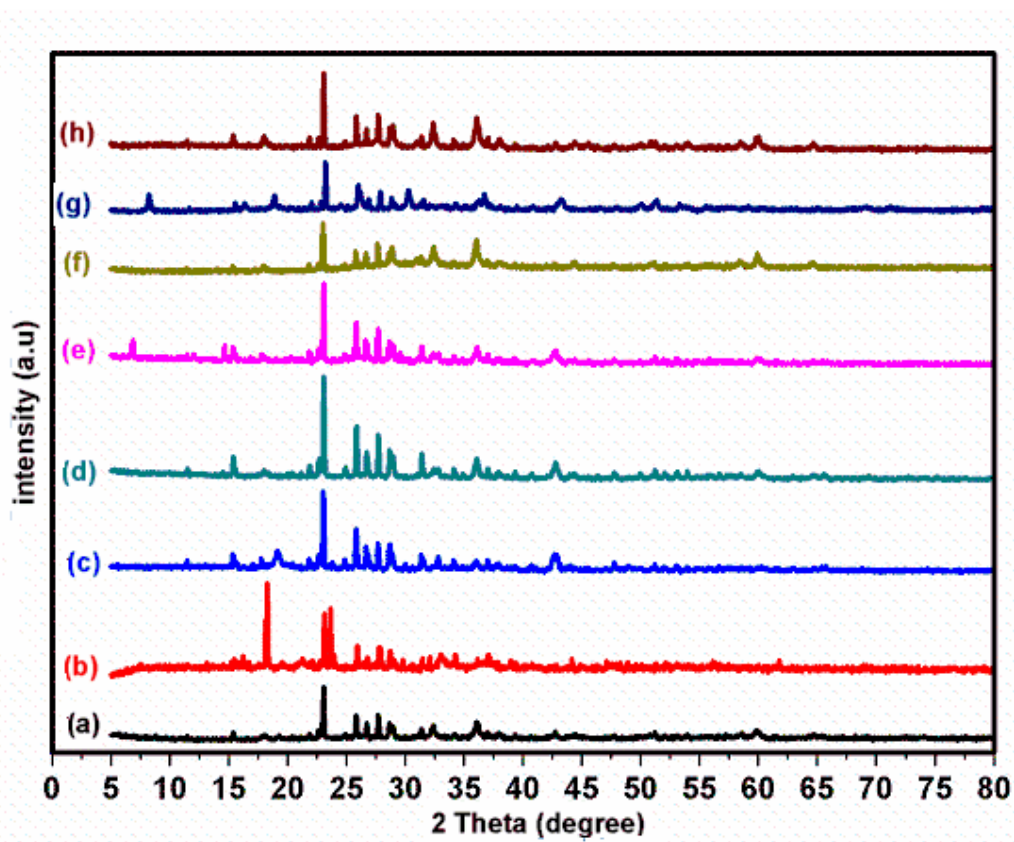


Figure 8

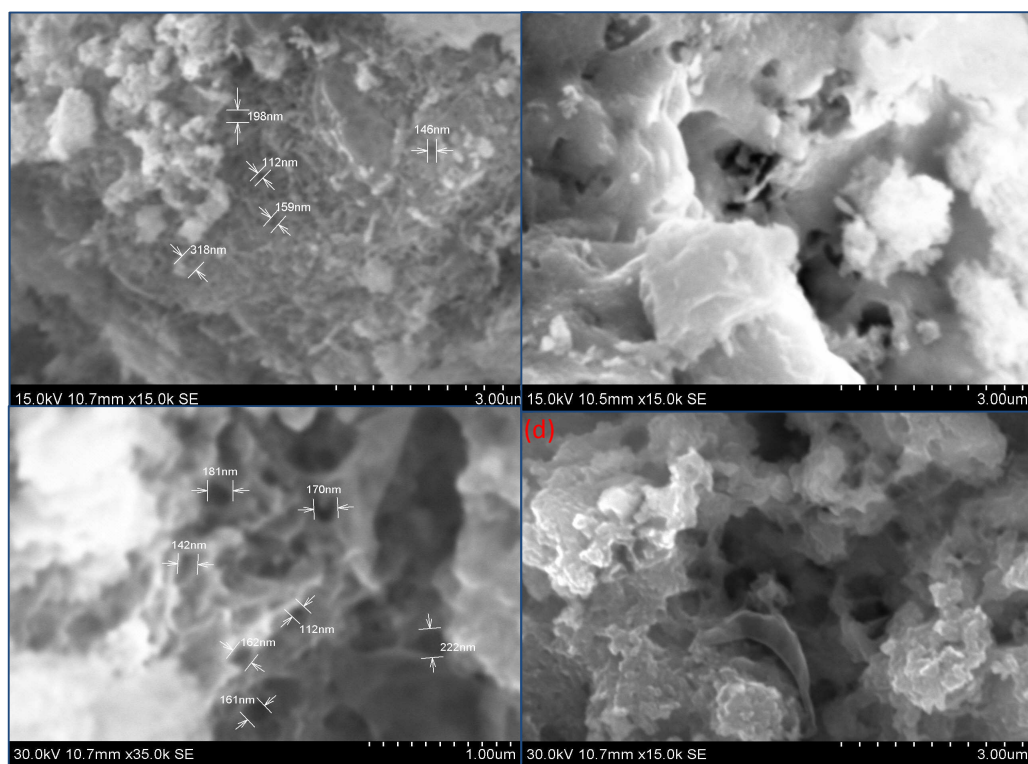


Figure 9

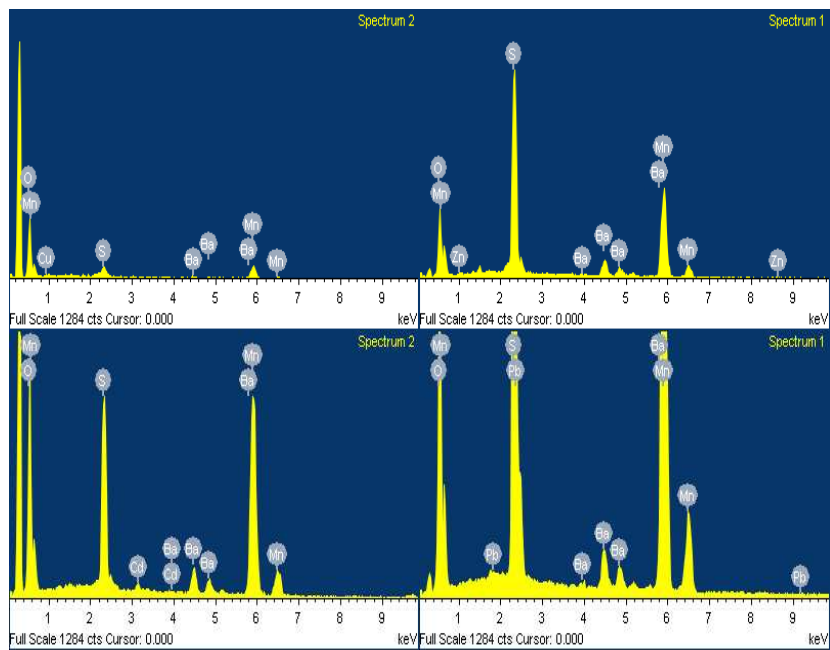


Figure 10



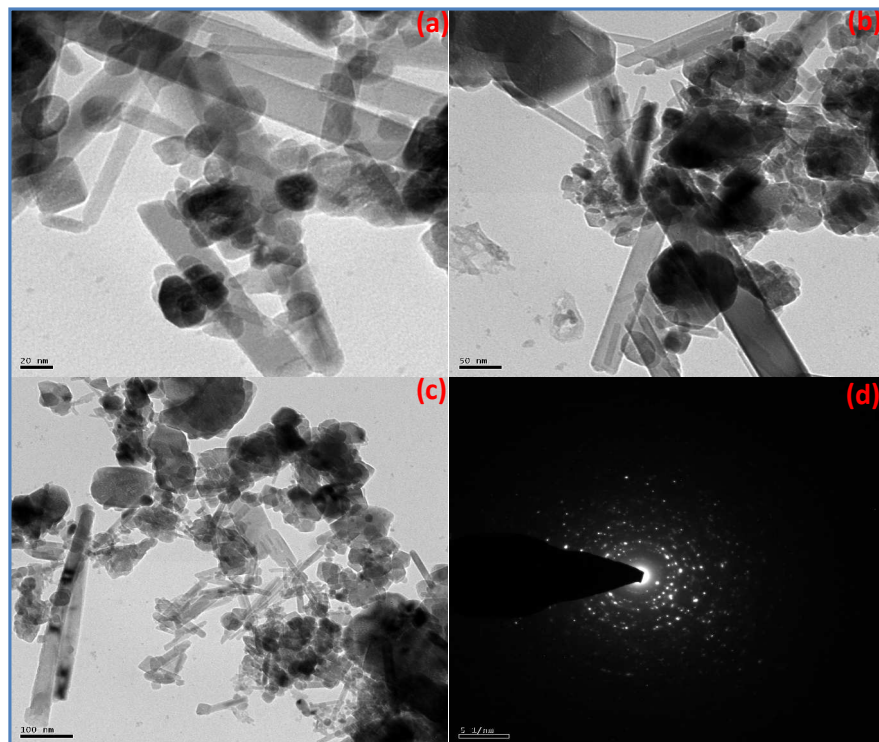


Figure 11

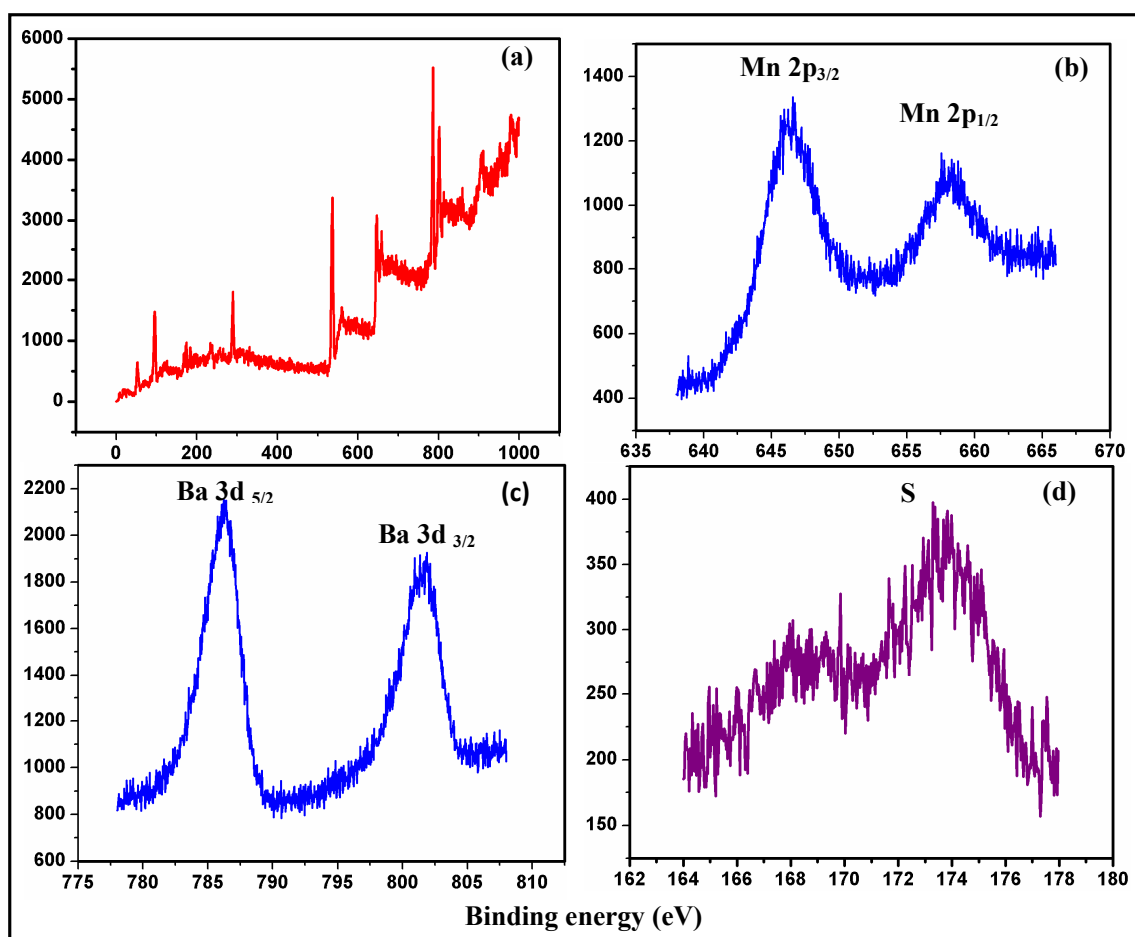


Figure 12

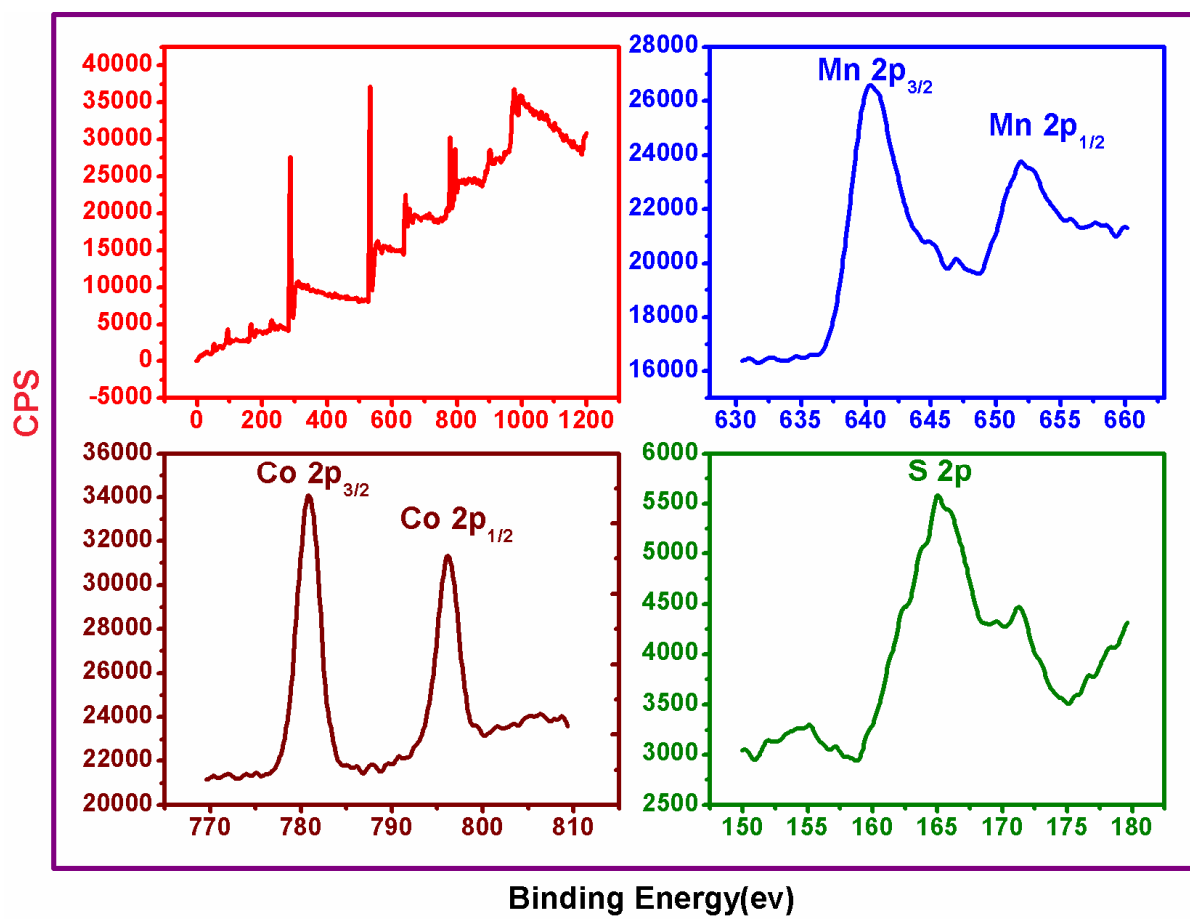


Figure 13

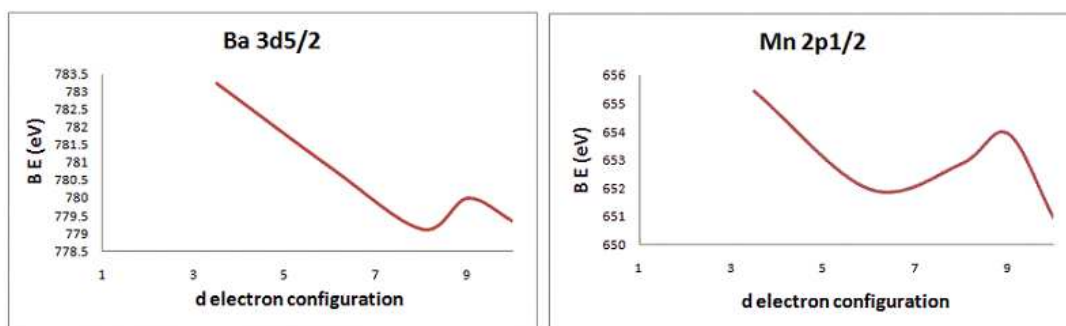


Figure 14

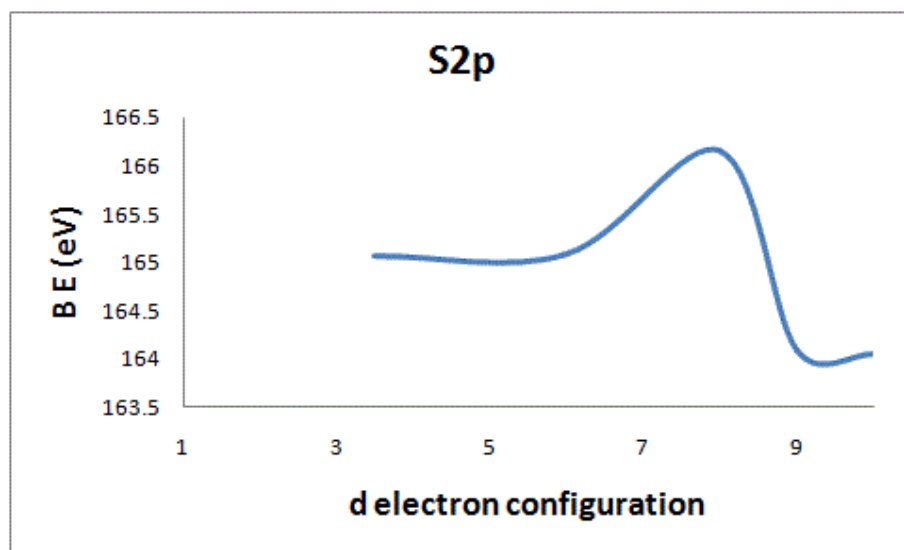


Figure 15

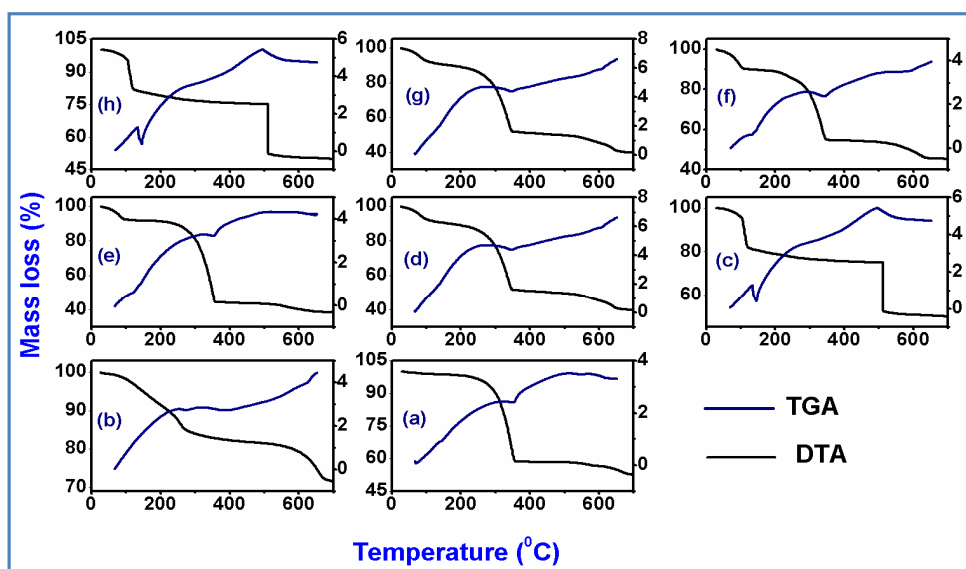


Figure 16

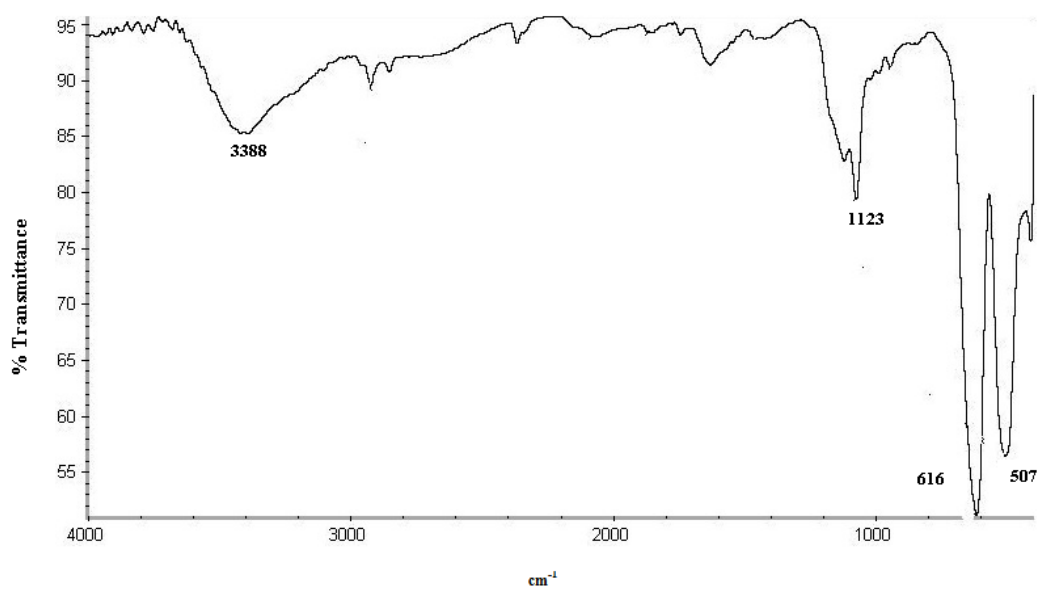


Figure 17