Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/softmatter

1	Fine structures of self-assembled beta-cyclodextrin/Pluronic in dilute and
2	dense systems: a small angle x-ray scattering study
3	
4	Kuo-Chih Shih ^a , Chi-Yen Li ^b , Wen-Hsien Li ^{b**} and Hsi-Mei Lai ^a *
5	
6	^a Department of Agricultural Chemistry, National Taiwan University,
7	No. 1, Roosevelt Rd., Sec. 4, Taipei 10617, Taiwan.
8	^b Department of Physics, National Central University,
9	No. 300, Jhongda Rd., Jhongli, 32001, Taiwan.
10	
11	*Correspondence to:
12	Dr. Hsi-Mei Lai (E-mail: hmlai@ntu.edu.tw)
13	Telephone: +886-2-33664816
14	Fax: +886-2-23633123
15	**Co-Correspondence to:
16	Dr. Wen-Hsien Li (E-mail: <u>whli@phy.ncu.edu.tw</u>)
17	Telephone: +886-3-4227151 ext 65335
18	

19 Abstract

20 The evolution of the fine structures of self-assembled polypseudorotaxane (PPR) 21 in Pluronic (PL F108) solutions containing dilute to dense beta-cyclodextrin (β -CD) 22 was illustrated for the first time by small angle x-ray scattering (SAXS). Dense β -CD 23 $(\sim 19 \text{ w/v})$ was found feasible to be dispersed in 24% citric acid solution. Five 24 percentage of PL F108 formed cylindrical micelles of 1 nm in radius and 8 nm in 25 length in the presence of 24% citric acid through the dehydration of citric acid and 26 citrate. PPR was formed through the host-guest interaction between PL F108 and 27 β -CD. In dilute β -CD system (1%), the single chains of PPR with separated β -CD 28 stacks on PL F108 were formed. The numbers of β -CD in each stack increased from 1 29 to 4 with increasing β -CD concentration to 9%. In dense β -CD system, PPR 30 condensed to correlated structures majorly composed of two unit blocks through the 31 hydrogen bonds between PPRs. Two distinguishable correlated domains with 32 correlation lengths of 50 nm (marked α -phase) and 46 nm (marked β -phase) along the 33 chains, but without fine periodic structure within each individual domain, were 34 identified in the 10% β -CD solution. Periodic stacking of β -CD in the domains 35 developed in the 12% solution. As β -CD concentration increased from 12 to 19%, the 36 correlated heights of α and β phases shrank from 41 and 32 nm into 30 and 10 nm, 37 respectively. There were 48 β -CDs that stabilized on each PL F108 chain in the 19% 38 β -CD system, which is in a good agreement with the stoichiometry.

39

40 Keywords: self-assembly, polypseudorotaxane, Pluronic, β-CD, SAXS,

Soft Matter Accepted Manuscript

41 Introduction

42 Cyclodextrins (CDs) are alpha-1,4 linked cyclic oligosaccharides, mainly consisting of 6, 7 or 8 alpha-D-glucopyranose units, known as alpha-, beta- or gamma-CD, 43 respectively.¹ The hydrophilicity of the outer rims and the relative hydrophobicity 44 45 inside cavities render CDs to form stable inclusion complexes (ICs) with organic molecules, such as flavours, pigments, vitamins and drugs.^{2, 3} In addition, the 46 formation of sequential ICs of several CD molecules within one polymer chain leads 47 to a unique supramolecule, namely the polypseudorotaxane (PPR).⁴ The driving 48 49 forces of this self-assembling process include the hydrophobic interaction between 50 polymers and CD cavities, the hydrogen-bonding among the rims of neighbouring CD 51 on the same polymer chain, and the subsequent loss of PPR solubility.⁵

52 Pluronic[®] (PL) is a triblock copolymer with a configuration of PEG-PPG-PEG. It 53 is a nonionic surfactant and has been widely used in pharmaceutics on taking the advantage of its self-assembling ability.⁶ At its critical micelle concentration (CMC), 54 55 the hydrophobic interaction among PPG segments and the phase separation between 56 PEG and PPG blocks attributed to their various hydrophobicity in the solution induce the micellization.¹ However, the self-assembled structure of PL in solution can vary 57 not only with its concentration,⁷ but also be affected by the presence of organic or 58 inorganic salt, such as potassium chloride,^{8, 9} sodium chloride,¹⁰⁻¹² sodium 59 carbonate,¹³ and salicylic acid.¹⁴ 60

61 When introducing CDs into PL, several CDs can thread onto the PPG or PEG 62 segments of the PL to form PPRs. The interaction between CDs and PL blocks has 63 been studied, through which various nano-structures have been identified in the 64 solutions containing CD and PL. These include dissolute¹⁵⁻¹⁷ or growing micelle,¹⁸

hollow sphere,¹⁹ rod-like¹⁶ and platelet structures^{18, 20} that have all been characterized
by using small angle x-ray/neutron scattering (SAXS/SANS).

67 Restricted by the low solubility of native β -CD (1.85 g/100 mL water at 25°C), modified β -CD was commonly used in investigating the self-assembled structures of 68 69 PPR. Previous studies of β -CD based PPR were focused either on low concentration of native β -CD or on high concentration of modified β -CD.^{16, 21} It has been realized 70 71 that the solubility of β -CD can be effectively enhanced by using organic acids, such as citric acid, lactic acid and malic acid.^{22, 23} Studies made on the nanostructures of PPRs 72 73 at high native β -CD concentrations are then possible in an aqueous system containing 74 organic acids. In fact, β -CD can be evenly dispersed in citric acid solution, with a solubility that is 10 times higher than in pure water.²⁴ The role of citric acid is to break 75 the hydrogen bonds among β -CD molecules.²⁵ 76

77 In this study, the evolution of nanostructure of β -CD/PL in citric acid solution 78 containing 5% PL F108 and various β -CD concentrations (0~19%) were investigated 79 by SAXS measurements. SAXS measurement is capable in identify the correlated 80 structure in the nanometer scale and is suitable for examining the self-assembling 81 structure of PL F108 itself and β -CD/PL systems. Analysing the SAXS patterns with 82 appropriate models allow quantitative extractions of PL F108 and β -CD based PPR 83 structures in citric acid solution, which provide detailed insights of these 84 self-assembled supramolecules.

85

86 **Experimental section**

87 Materials

- 88 PL F108 (EG₁₃₃-PG₄₉-EG₁₃₃, Mw 14600 g/mol) was purchased from BASF (Florham
- 89 Park, New York, USA). β -CD (purity \geq 97%) was obtained from Sigma-Aldrich (St.
- 90 Louis, MO, USA). All chemicals used in the present study were analytical grade.

91 Preparation of PL F108 and β-CD/PL samples

92 A β -CD stock solution (20 w/v% β -CD) was prepared with 24% citric acid solution. 93 Before mixing with the β -CD stock solution, an adequate amount of PL F108 pellet 94 was dissolved with 24% citric acid solution as well. Results of our measurements 95 show that the 24% citric acid solution can effectively separate β -CD molecules, 96 resulting in a clear and well dispersed solution. Below which, the aggregations of 97 β -CD are able to be detected by vision. The depletion of flocculation due to the 98 decrease of dispersibility had been reported when a high concentration of citric acid is used.^{26,27} In order to minimize the scattering background contributed by citric acids, 99 100 the 24% citric acid solution was chosen as the solvent in this study. The β -CD/PL 101 mixtures contained various concentrations of β -CD (0-19%) and 5% PL F108 in 24% 102 citric acid solutions. The mixtures were stored at $6 \pm 2^{\circ}$ C for 12 h and then at 25°C 103 for an additional 24 h before taking SAXS measurements. This process amounts to a 104 total setting time of 36 h for the β -CD/PL in the solution. The concentrations of PL 105 F108 and β -CD were used to mark the samples. As an example, 5-19 indicates the 106 sample contains 19 w/v% β -CD and 5 w/v% PL F108 in the 24% citric acid solution.

107

108 Small angle X-ray scattering (SAXS) measurement

The SAXS experiment was performed on the beamline 23A1, operated in the
transmission mode, at National Synchrotron Radiation Research Center (NSRRC),
Taiwan. The sample was loaded in a standard solution cell for the measurement,
sealed with kapton films. The data were collected using photons of incident energy 12

113 keV (incident wavelength $\lambda = 1.033$ Å), with a sample-to-detector distance of ~3 114 meters. This setup provides a *q*-range from 0.006 to 0.28 Å⁻¹ available for 115 measurements, where scattering vector $q = 4\pi \sin\theta / \lambda$ and θ is half of the scattering 116 angle.

117 SAXS data analysis

118 It has been known that the radius of gyration (R_{e}) of the scattering object in solution can be obtained from Guinier analysis,²⁸ and the pair distance distribution function 119 (PDDF) can be extracted from the indirect Fourier transform (IFT) analysis²⁹ of the 120 121 scattering profile. The PDDF was obtained by employing the GNOM program in the ATSAS suite, which was developed by the BIO-SAXS group at EMBL-Hamburg.³⁰ 122 The model fitting method suggested by Yeh et al.³¹ was used to separate the form 123 124 factor P(q) from the structure factor S(q) that could appear in the scattering pattern I(q)= P(q)S(q). The form factor contains the intensity scattered from the shape of the 125 126 object; whereas the structure factor links to the diffraction intensities from the 127 periodic structure of the correlated domains. Two scattering profiles that link to form factor were considered: one models the low $q (< -0.01 \text{ Å}^{-1})$ scattering, while the other 128 129 describes the high q scattering. The form factor was analyzed employing the IGOR 130 Pro 6.11 program package provided by NIST (National Institute of Standard and Technology) Center for Neutron Research (NIST, Gaithersburg, MD, USA).³² The 131 132 structure factor, on the other hand, was analyzed by fitting the diffraction peaks to the 133 diffraction profiles of finite-sized domains. It is known that the width of a diffraction 134 peak reflect the spatial extension of the periodicity, which corresponds to the size of 135 the correlated domain in the present cases. The line profile of a diffraction peak from 136 a size-dispersed supramolecule assembly was obtained by combining the diffraction 137 profiles contributed from each individual domain in the assembly. The line profile of

138	the structure factor was used to extract size distribution of the supramolecules in the
139	solution. Details of the fitting processes will be described in the following sections.
140	

141 **Results and discussion**

142 Qualitative analysis

143 SAXS measurements were carried out on the mixtures containing various 144 concentrations of β -CD (0~19%) in 5% PL F108 that was prepared with 24% citric 145 acid_(aq). The scattering patterns were used to extract the micelle conformation and size 146 as well as the stacking of polypseudorotaxane formed by ICs of β -CD and PL F108. 147 The scattering patterns in Fig. 1 show clearly that the structures of PL F108 and β -CD 148 changed significantly with the β -CD concentration. Interestingly, the scattering 149 patterns can be divided into two groups according to their profiles. The samples, such 150 as 5-0, 5-1 and 5-9, generate only scattering from the form factor are designated as the 151 dilute systems; whereas those, such as 5-10, 5-12, 5-14, 5-16 and 5-19, producing 152 diffraction peaks as well are designated as the dense systems.

The scattering patterns in the low-q regime (0.006 to 0.01 Å⁻¹) of the dilute 153 systems can be expressed by the power law $I(q) = Aq^{-\alpha}$, with an exponent α around 2.5 154 155 (Fig. 1, 5-0, 5-1 and 5-9), signalling the existence of a large-scale structure in the 156 dilute solutions. It is known that the mass and surface fractals will give rise to values 157 of $1 \sim 3$ and $3 \sim 4$, respectively, for the scattering exponent. The low-q scattering with 158 $\alpha \sim 2.5$ observed in the dilute systems reflects the scattering contributed mainly from 159 the structure of the mass fractal. We believe that this large-scale structure is 160 contributed from the PL F108 in the sample 5-0, and from the PL F108 as well as 161 from the PPR in the samples 5-1 and 5-9. On the other hand, the scattering exponent 162 in the high-q regime of the dilute systems was found to be near 1, reflecting the

Soft Matter Accepted Manuscript

163 scattering from elongated cylindrical objects. It is interesting to note that the onset of high-q scattering regime shifts from at $q \sim 0.07$ Å⁻¹ for the sample 5-0 to $q \sim 0.16$ Å⁻¹ 164 165 for the sample 5-9, indicating the size of the cylindrical objects was reduced upon 166 incorporating β -CD into the mixtures.

167 The appearances of diffraction peaks in the scattering patterns of the dense β -CD 168 systems show that the interactions among the β -CDs have become strong enough to form correlated structures. As an example, the broad peak at q = 0.013 Å⁻¹ in the 5-10 169 170 pattern (Fig. 1, 5-10) corresponds to a correlated size of 50 nm for the β -CD. The 171 diffraction peaks shift gradually to larger q positions in the denser systems, indicating 172 the correlated domains condensed into smaller sizes. Higher order diffraction peaks 173 are clearly revealed in the samples with a β -CD concentration higher than 12%. In addition, a q^{-2} scattering profile was observed in the low-q regime of the dense 174 175 systems, revealing scattering from two-dimensional (2D) geometry of lamellar. 176 Interestingly, similar high-q scattering profiles were observed for the samples with β -CD concentration higher than 9%, showing the conformation of the small entities in 177 178 the system has stabilized.

Guinier analysis of the dilute systems 179

Guinier region can be identified in the scattering profiles of samples 5-0, 5-1 and 5-9, 180 expressed as $I(q) \propto \exp(\frac{-R_g^2 q^2}{3})$, where R_g is the gyration radius of the scattering 181 object.²⁸ The absolute value of the slope in the $Ln\{I(q)\}$ versus q^2 plot (Guinier plot) 182 gives the value of $R_g^2/3$. The R_g values are 25.3, 7.14, and 7.75 nm for samples 5-0, 183 184 5-1 and 5-9, respectively. It is interesting to point out that a significant reduction (by a factor of ~3) in the aggregated sizes of PL F108 to PPR. Unfortunately, Guinier 185 186 analysis is not possible to perform on the dense systems, where the low-q scattering

187 profile is severely disturbed by the appearance of diffraction peaks.

Indirect Fourier Transform (IFT) analysis

189 The IFT analysis is a model independent method to obtain information on the spatial 190 distribution of mass density in the solution. This analysis gives the pair distance 191 distribution function, p(r) known as.

192
$$p(r) = \frac{1}{2\pi^2} \int_0^\infty I(q) qr \sin(qr) dq$$
 (1)

193 where q is the magnitude of the scattering vector. This p(r) gives the probability to 194 find out a mass point at a distance r away from a selective point within the scattering 195 object, weighed by the difference in scattering length density (SLD) between these 196 two points. In the case of a rod-like object, the PDDF can be determined by $p(r) = \gamma$ $_{c}(r) \times r$, where $\gamma_{c}(r)$ is the characteristic function of the thickness. Fig. 2 shows the p(r)197 198 obtained for the dilute samples. All of three p(r) distributions show a symmetric, 199 Gaussian-like peak with a maximum at r = 0.5, 0.45, and 0.38 nm for the samples 5-0, 200 5-1, and 5-9, respectively. The shape of p(r) displays a fixed SLD for the rod-like cylinder object over the entire cross-section.^{29, 33, 34} It is known that PL formed 201 core-shell micelle with a hydrophobic PPG core and a hydrophilic corona.³⁵ For 202 sample 5-0 the cross-section of the rod-like micelle is composed of a PPG core with a 203 SLD of $\rho_{PPG} = 9.47 \times 10^{-6} \text{ Å}^{-2}$ and a PEG corona with a SLD of $\rho_{PEG} = 1.11 \times 10^{-6} \text{ Å}^{-2}$ 204 10⁻⁵Å^{-2,35} Only a cylindrical object could be identified with the weak scattering 205 206 contrast between PPG core and PEG corona. For samples 5-1 and 5-9 the PPR is composed of a β -CD shell ($\rho_{\beta-CD} = 1.3 \times 10^{-5} \text{ Å}^{-2}$) and a PL F108 core ($\rho_{PLF108} = 1.07$ 207 × 10⁻⁵ Å⁻²). The scattering contrast between β-CD and 24% citric acid solvent (ρ_{sol} = 208 $1.03 \times 10^{-5} \text{ Å}^{-2}$) is much better than that between PL F108 and the solvent. We believe 209 210 that the scattering profiles of samples 5-1 and 5-9 were contributed by the rod-like PPRs composed of numerous β-CDs. These results suggest that cylindrical objects, 211

Soft Matter Accepted Manuscript

but not core-shell cylinder ones, should be incorporated in describing the scattering

213 profile.

214 Model fitting

215 Dilute systems (0, 1 and 9% β -CD)

216 As mentioned above, the scattering patterns of dilute systems can be modeled by a 217 low-q power law and a high-q cylindrical conformation scattering. Hence, the expression for scattering profile should contains scattering from large-scale network³⁶ 218 that can be formulated using the Debye-Anderson-Brumberger (DAB) model³⁷ and 219 220 from the non-interacting cylinders in the solutions. Fig. 3 shows the observed (open 221 squares) and calculated (solid line) scattering profiles of sample 5-0. The 222 contributions from the cylinders (dashed line) and from the DAB model (dotted line) 223 are displayed in Fig. 3 as well. The DAB model may be expressed as

224
$$I(q) = \frac{A\xi^2}{\left[1 + \left(q\xi^2\right)\right]^2},$$
 (2)

where ξ is the characteristic length of the domain size and *A* is a scale factor. The form factor of cylinder can be expressed as

227
$$P_{cyl}(q) = A \frac{1}{\pi r^2 L} \int_{0}^{\pi/2} f^2(q, \alpha) \sin(\alpha) d\alpha , \qquad (3)$$

228 where
$$f(q,\alpha) = 2(\rho_{cyl} - \rho_{sol})\pi r^2 L j_0 (qL/2\cos\alpha) \frac{j_1(qr\sin\alpha)}{qr\sin\alpha}$$
, $j_0(x) = \sin(x)/x$, $j_1(x)$

is the first order Bassel function, α is the angle between the axis of the cylinder and the scattering vector *q*, *r* and *L* are the radius and the length of cylinder, respectively.

231
$$\rho_{cyl} = 1.07 \times 10^{-5} \text{ Å}^{-2}$$
 and $\rho_{sol} = 1.03 \times 10^{-5} \text{ Å}^{-2}$ are the SLD of PL F108 and 24% citric

acid solution, respectively. The fit allows extracting r and L of the cylinder.

233 Dense systems (10, 12, 14, 16 and 19% β-CD)

The structure factor
$$S(q)$$
 could be extracted by separating the form factor $P(q)$ from
the scattering pattern $I(q)$. Taking the pattern of sample 5-19 (Fig. 4) as an example,
the form factor contributed from the lamellar structure was expressed as

237
$$P_{lam}(q) = \frac{2\Delta\rho^2}{q^2} \left[1 - \cos(q\delta) e^{-q^2\sigma^2/2} \right],$$
 (4)

238 where $\Delta \rho$ is the scattering contrast between the lamellar structure (β -CD) and the 239 solvent (24% citric acid solution), σ is the mean variation of the layer thickness, δ is 240 the thickness of lamellar structure. On the other hand, the scattering contributed from 241 the cylinder structure (revealed more clearly in the high-q regime) of the dense systems is complicated by the diffraction peaks (Fig. 1). The form factor associated 242 243 with the cylinders of sample 5-9 was then adopted for that of dense systems. This is 244 justified since the scattering from the cylinders has stabilized in systems with a β -CD 245 concentration higher than 9%, as can be seen in Fig. 1. The scattering contributed from form factor was thus expressed as the linear combination of $P_{lam}(q)$ and $P_{cyl}(q)$ 246 with a cut-off wave vector³⁸ q_0 in weighting the two contributions, 247

248
$$P(q) = \operatorname{A} \exp\left(-\frac{q}{q_0}\right)^{5.33} \times P_{lam}(q) + \operatorname{B} \exp\left[1 - \exp\left(-\frac{q}{q_0}\right)^{5.33}\right] \times P_{cyl}(q),$$
 (5)

where *A* and *B* are the intensity scale factors. In this form the lamellar form factor dominates in the low-q ($q < q_0$) regime, while the cylinder form factor dominates in high-q's.

252 Structure factor is expressed using the Debye-Scherrer equation with Gaussian253 peak profiles as

254
$$I_{hkl}(\theta) = \frac{1}{(w/2)\sqrt{2\pi}} \exp\left[\frac{-(\theta - \theta_c)^2}{2(w/2)^2}\right],$$
 (6)

where θ is the half of the scattering angle, θ_c is the peak position, and *w* is the full width at half maximum of the peak profile. The pattern of sample 5-19 was thus separated into three contributions of from the cylinders in the mixture (open triangles, marked $P_{cyl}(q)$ in Fig. 4), from the lamellar conformation (open circles, marked $P_{lam}(q)$ in Fig. 4), and from periodically correlated β -CD (dotted lines in Fig. 4).

260

261 Effect of β-CD concentration on PPR structure

262 Nanostructure of β-CD/PL F108 in dilute systems

263 In this section, we focus on analyzing the structure of the cylinders in the mixture. Fig. 264 5 shows the observed and calculated patterns of samples 5-0, 5-1 and 5-9, where no diffraction peaks are revealed yet. A rod-like cylinder with $r \sim 1$ nm and $L \sim 8$ nm 265 were obtained in the sample 5-0. It is known that R_g of an elongated object has been 266 calculated to be $\left(\frac{r^2}{2} + \frac{L^2}{12}\right)^{\frac{1}{2}}$ and it was calculated to be 24.7 nm for 5-0 sample. This 267 is in good agreement with the R_g value (25.3 nm) extracted from the Guinier analysis. 268 269 This cylindrical object is understood to be the micelle formed from PL F108 270 originated from the hydrophobic interaction between the PPO segments in the mixture. 271 Cylindrical micelle has been observed in PL F108 in 0.5 M sodium carbonate aqueous solution.¹³ Cylindrical micelle of other PL series has also been observed in inorganic 272 salt⁸⁻¹² and in butan-1-ol.³⁹ In the present systems, citric acid dissociates 1 proton in 273 274 24% citric acid solution with a pH value of 1.5. We believed that the citric acid and its 275 conjugated salt, known as citrate, are responsible for the dehydration process to form 276 cylindrical PL F108 micelles. Similar phenomenon has also been observed in a separated study.¹⁴ 277

278 It is known that characteristic length ξ can be used to describe the large-scale 279 network structure in the mixture of PL F108 and citric acid using the DAB model.³⁶ A 12

characteristic length of 28.3 nm for the network in sample 5-0 is identified. This
network is formed from the entanglement of the unassociated PL F108 unimer in the
mixture. The citric acid is believed to play a crucial role in the formation of the
extended network, initiated by the development of hydrogen bonds between citric acid
and PL F108.

Introducing β -CD into PL F108 drives the onset of q^{-1} behavior in the scattering 285 pattern to appear at larger q (Fig. 5, 5-0 and 5-1), indicating the formation of smaller 286 cylindrical objects. These objects are the rod-like PPR³⁸ formed through the 287 288 host-guest interaction between β -CD and PL F108. In sample 5-1, the molar ratio 289 between PG and β -CD is 19.6, the PL micelle will not be fully de-assembled. We 290 believe that both PL F108 micelle and rod-like PPR co-exists in the mixture, since the scattering from micelle vanished once PG/B-CD was below 3.1.¹⁶ However, only the 291 292 scattering from rod-like PPR were identified in the scattering pattern of sample 5-1. 293 As mentioned above the scattering contrast between 24% citric acid solution (1.03 \times 10^{-5} Å⁻²) and β -CD (1.3 × 10^{-5} Å⁻²) is known to be considerably better than that of 294 between 24% citric acid solution and PL F108 ($\sim 1.07 \times 10^{-5} \text{ Å}^{-2}$).⁴⁰ It is clear that the 295 scattering pattern in the q > 0.1 Å⁻¹ regime is dominated by the scattering from β -CD, 296 and reflects the morphology of the stacking of β -CD on PL F108.⁴¹ Table 1 lists the 297 298 fitted parameters for the samples 5-0, 5-1 and 5-9. The correlated β -CD block was 299 found to be of 0.6 nm in radius and 1.2 nm in length, revealing that single chains of 300 PPR with a stacking of 1 or 2 correlated β -CDs in each unit block were formed in these samples. We note that the radius and length of one β -CD are known¹ to be 0.77 301 302 and 0.79 nm, respectively.

The scattering patterns in the q < 0.1 Å⁻¹ regime of samples 5-1 and 5-9 are nearly identical (Fig. 4). The differences in the two patterns are, however, seen at q > 0.1 Å⁻¹

305 reflect the structure of the rod-like PPRs are indeed different. The length of PPR 306 increased from 1.2 nm for sample 5-1 to 2.4 nm for sample 5-9 (Table 1), showing the 307 number of stacked β -CD has increased to 3 or 4 β -CD in each correlated block. In the 308 mixture containing α -CD and PEO homopolymer in H₂O, the rod-like PPR self-assembled with 7 CDs per stacking block.⁴¹ The number of CD stacked in each 309 310 block on PL F108 depends on the strength of inter-CD hydrogen bonding (head-to-head or tail-to-tail).^{41, 42} It is known that β -CDs can be dispersed in citric 311 312 acid solution through the hydrogen bonds among the polar groups (-COOH and -OH) of citric acid and the -OH groups of β -CD located on its outside rim.²⁵ Therefore, the 313 314 interaction of inter-CD hydrogen bonds in citric acid solution is less effective than those in H₂O. This results in a less number of CDs that are stacked in the present 315 mixtures. In sample 5-9 of molar ratio PG/β -CD = 2, a PPG segment⁴³ that is fully 316 317 covered by β -CD can be expected. Thus, a rod-like PPR with a length of ~20 nm can 318 be expected in sample 5-9. Interestingly, only cylinders with a length of 2.4 nm were 319 observed in sample 5-9. This suggests that there are numerous separated short PPR 320 instead of forming fewer long rod-like PPRs in 24 w/v% citric acid solution. We 321 hence propose that the β -CDs not only thread onto the PPG segments but also on the 322 PEG segments in considerable portions. The formation of β -CD inclusion complex 323 with PPG homopolymer in the presence of citrate ions has been observed in a previous study.⁴⁴ β-CD has been found to form inclusion complex, with noticeable 324 amount of β-CDs, being reside on PEG segments of PL P85.⁴⁵ It has been reported 325 326 that the binding constant of naproxen with β -CD in the presence of 1% PEG would be 327 reduced that was attributed to the formation of PEG-β-CD inclusion complexes; 328 therefore, the amount of free CD available for the inclusion with naproxen was reduced.46 329

330 Through DAB model calculations, the domain size in each sample was identified. 331 These results are summarized in Table 1. A domain size of 28.3 nm was obtained for 332 PL F108 (sample 5-0). The domain size decreases to 18.5 nm in sample 5-1, and 333 further to 15.5 nm in sample 5-9. The driving force for a shorter characteristic length 334 in higher β -CD concentration is believed to be due to the stronger inter-CD 335 interaction in higher β -CD environment in the PPR chains and among the rims of the neighboring CD.⁵ In our previous work²⁴, similar results were observed in the sample 336 337 containing a higher molar ratio of PG/β -CD = 17.6, where PPR gels formed with the 338 assistance of cross-link PPR crystallines resulting from the attractive interaction in 339 neighboring PPRs.

340 Nano structure of β-CD/PL F108 in dense systems

341 Fig. 6 shows the observed and calculated scattering patterns of the dense samples. 342 Through this analysis, the structure factor is effectively extracted from the scattering 343 intensity for further analysis. The diffraction peaks were mainly contributed from the 344 periodic structure of β -CD, since the scattering contrast of β -CD is significantly 345 higher than that of PL F108. The isolated diffraction pattern of sample 5-10 is shown 346 in Fig. 7, where two diffraction peaks can be identified. The intense one corresponds 347 to a domain size of 50 nm (d = $2\pi/q$), marked as α -phase with $c_{\alpha} = 50$ nm; whereas the 348 less intense one corresponds to a domain size of 96 nm, marked as β -phase with c_{β} = 349 96 nm. It is known that the width of a diffraction peak reflect the spatial extension of 350 the lattice periodicity, which could be used to extract the stacking height of the 351 correlated β -CD. The line profile of a diffraction peak from a size-dispersed assembly 352 can be obtained by combining the diffraction profiles contributed from each 353 correlated grain in the assembly. The solid curves in Fig. 7 indicates the calculated 354 diffraction profile with the size distributions of the two phases shown in the inset to

Fig. 7, assuming the sizes consist of multiple stacking of the c_{α} or c_{β} unit blocks. It is interesting to find that most of the correlated β -CDs in both phases interconnected into stacking of two unit blocks of stacking heights $d_{\alpha} = 96$ nm and $d_{\beta} = 188$ nm. Moreover, the intensity of the α -phase is considerably higher (3.2 times more intense), showing that most of the correlated β -CDs condensed in the α -phase. There is no higher order reflections that appear in the diffraction pattern, showing the condensed β -CDs within the unit blocks are not packed into periodic structure yet.

362 Two sets of the diffraction peaks with different intensity could be identified in the 363 samples 5-12 and 5-14. The more intense one, that is the α -phase, shrinks to a shorter spatial period of $c_{\alpha} = 41$ nm, while the β -phase shrinks to $c_{\beta} = 32$ nm (Fig. 8). This is 364 365 understandable that the correlated β -CDs will condense into a more compact structure 366 in a solution with higher β -CD concentration. The peak widths of the β -phase are 367 much broader than those of the α -phase, indicating that α and β phases are originated 368 from separated structures. In addition, the size analysis based on the diffraction 369 profiles giving 94% of the α -phase and 95% of the β -phase are composed of two unit 370 blocks in sample 5-14, as shown in the inset in Fig. 8. Higher order diffraction peaks 371 of up to the (003) reflection associated with both phases are revealed in the diffraction 372 pattern, indicating the β -CDs in the unit blocks have built up periodic structures to 373 form shorter periodicities. It is known that the total diffraction intensity is directly 374 proportional to the mass of the phase that generates the diffraction pattern. The 375 intensity ratio between that from the α -phase and from the β -phase can be used to 376 calculate the mass ratio of the two phases in the solution. A mass ratio of m_{α}/m_{β} = 377 73/27 was obtained for sample 5-14, when taking the diffraction intensity is 378 dominated by the mass of the unit block. This can be justified since the main 379 difference between the α -phase and β -phase is the numbers of β -CD that are

404

Soft Matter

380	correlated into the unit block. The α -phase shrinks into a periodicity of $c_{\alpha} = 30$ nm in
381	samples 5-16 and 5-19. Higher order diffraction peaks of up to the (008) reflection are
382	revealed, but no sign of the existence of the (009) reflection in the measured
383	diffraction pattern, as shown in Fig. 9. This suggested that a long-range ordering
384	structure composed of β -CD was formed. The appearance of (008) reflection in the
385	diffraction pattern indicates that there are at least 16 correlated β -CDs condensed in
386	each 30 nm unit block, which corresponds to a separation of $(30/16)-0.8=1.075$ nm
387	between the neighboring β -CDs in the correlated structure. The other possible
388	structure is that there are 24 correlated β -CDs in each 30 nm unit block, which then
389	gives a separation of (30/24)-0.8=0.45 nm between the β -CDs. The α -phase in sample
390	5-19 composes of two unit blocks, that is, at least 32 (16×2) and 48 (24×2) β -CDs
391	could be identified on one polymer chain, respectively. In addition, the molar ratio of
392	PG/ β -CD = 1 for sample 5-19 which represented at least 49 β -CDs on a polymer chain,
393	assuming β -CD distributed equally in each polymer chain. We, hence, conclude that
394	the short separation of 0.45 nm between the β -CDs is likely to occur in sample 5-19.
395	A schematic representation of the evolution of the ordered structure of β -CDs in
396	PPR with β -CD concentration is illustrated in Scheme 1. In sample 5-10, the β -CDs
397	loosely aggregate into alternative stacking of a dense segment of 50 nm in length and
398	a dilute segment of 46 nm long (Scheme 1 (a)). The mass density of the dilute
399	segment is ~31% that of the dense one. On the average, the β -CD coated PPR chains
400	bend in every two alternative stacking. This structure generates two diffraction peaks
401	at periodicities of 50 and 50+46=96 nm, but no higher order peaks. In samples 5-12
402	and 5-14, the interaction among β -CDs is apparently strong enough for the aggregated
403	β -CDs not only condense into a shorter separation, but also correlate into a periodic

structure. The dense segment condenses into a spatial extension of 40 nm, while the

dilute one shrinks to 32 nm long (Scheme 2(b)). It is the two correlated structures that generate the two series of diffraction peaks. In the denser mixtures of samples 5-16 and 5-19, there are 24 correlated β -CDs that packed into a unit segment of 30 nm long, but no significant amount of correlated β -CD that appears in between the unit segment (Scheme 1 (c)).

As mentioned above, the scattering pattern of sample 5-9 can incorporated directly in the fitting process to model the high q scattering for samples in the dense β -CD system. Two possibilities can contribute to this result. First, these rod-like PPRs are the building blocks of the lamellae. Second, these rod-like PPRs does not condensed into the lamellar structure but remain in the 'freely dispersed' form in the solution. Based on the fact that the periodic structure composed of 24 β -CDs, the later scheme is likely to occur in the present systems.

417

418 Schematic representation of β -CD/PL F108 in citric acid

419 solution

420 A schematic representation of the nano-structure in the mixtures with or without 421 β -CD is shown in scheme 1. PL F108 forms cylindrical micelle in 24 w/v% citric acid 422 solution [Scheme 2(a)]. With the addition of $1\% \beta$ -CD, a rod-like PPR is formed with 423 $1\sim 2$ CDs that are aggregated in each threaded β -CD stack [Scheme 2(b)]. For the 424 system with 9% β -CD in the mixture, the number of CD in the β -CD block increases 425 to 3~4 units, and the characteristic length decreases, which we believe is triggered by 426 the formation of hydrogen bonds between β -CDs [Scheme 2(c)]. In addition, β -CD 427 can thread not only on the PPG segment but also on the PEG segment of the PL F108 428 chain. Increasing β -CD concentration further to 19%, the aggregated structure

429	develops into a periodic arrangement, with 24 β -CDs condensed into a unit block
430	[Scheme 2(d)]. Moreover, small but noticeable amount of PPR single chains remain in
431	the 'freely dispersed' form surrounding the ordered structure.

432

433 **Conclusion**

434 Investigated by small angle x-ray scattering measurements, the self-assembled 435 structures of PL F108 in 24% citric acid solution containing 0 to 19% β -CD were 436 revealed. PL F108 forms cylindrical micelles in 24% citric acid solution, originated 437 from the dehydration effect of citric acid and citrate. The threaded β -CDs formed 438 numerous separated short PPR not only on the PPG segment but on the PEG segment 439 instead of long single chain PPR in the dilute systems. The attractive interactions 440 among neighboring β -CDs on the single PL F108 chain and among the PPR single 441 chains, presumably through the hydrogen bonding, result in shorter characteristic 442 length. In the dense systems, the attractive interactions drive the β -CDs to pack into a 443 periodic structure. The size of the periodic structure decreased with increasing β -CD 444 concentration in the mixture. Furthermore, the number of the β -CDs on single PL 445 F108 chain has also been revealed.

446

447 Acknowledgements

This work was supported by the Grants NSC 98-2313-B-002-050-MY3 and
102-2313-B-002-056-MY3 from the National Science Council, Taipei, Taiwan. We
thank Dr. Ya-Sen Sun of the Department of Chemical and Materials Engineering,
National Central University, Taiwan and Dr. U-Ser Jeng of the NSRRC for
discussion.

454 **References**

- J. Szejtli, in *Chemical and functional properties of food saccharides*, ed. P.
 Tomasik, CRC Press, Boca Raton, 2004, ch. 17, pp. 259-277.
- 457 2. H. Hashimoto, *in Comprehensive supramolecular chemistry*, ed. J. M. Lehn,
 458 Pergamon, Oxford, 1996, vol. 3, ch. 16, pp 483-502.
- 459 3. L. S. Szente, J., in *Comprehensive supramolecular chemistry*, ed. J. M. Lehn,
 460 Pergamon, Oxford, 1996, vol. 3, ch.17, pp 503-514.
- 461 4. A. Harada and M. Kamachi, *Macromolecules*, 1990, 23, 2821-2823.
- 462 5. S. Loethen, J. M. Kim and D. H. Thompson, *Polym. Rev.*, 2007, 47, 383-418.
- 463 6. C. L. He, S. W. Kim and D. S. Lee, J. Control. Release., 2008, 127, 189-207.
- 464 7. G. E. Newby, I. W. Hamley, S. M. King, C. M. Martin and N. J. Terrill, J.
 465 *Colloid. Interf. Sci.*, 2009, **329**, 54-61.
- 466 8. K. Patel, P. Bahadur, C. Guo, J. H. Ma, H. Z. Liu and K. Nakashima, *J. Disper.*467 *Sci. Technol.*, 2008, **29**, 748-755.
- 468 9. K. Patel, P. Bahadur, C. Guo, J. H. Ma, H. Z. Liu, Y. Yamashita, A. Khanal and K.
 469 Nakashima, *European Polymer Journal*, 2007, 43, 1699-1708.
- 470 10. R. Ganguly, V. K. Aswal and P. A. Hassan, J. Colloid. Interface. Sci., 2007, 315,
 471 693-700.
- 472 11. R. Ganguly, M. Kumbhakar and V. K. Aswal, J. Phys. Chem. B, 2009, 113,
 473 9441-9446.
- 474 12. Y. Kadam, R. Ganguly, M. Kumbhakar, V. K. Aswal, P. A. Hassan and P.
 475 Bahadur, J. Phys. Chem. B, 2009, 113, 16296-16302.
- 476 13. L. Fan, M. Degen, N. Grupido, S. Bendle and P. Pennartz, *Mat. Sci. Eng.*477 *A-Struct.*, 2010, **528**, 127-136.
- 478 14. P. Parekh, R. Ganguly, V. K. Aswal and P. Bahadur, *Soft Matter*, 2012, **8**, 5864.
- 479 15. J. Joseph, C. A. Dreiss, T. Cosgrove and J. S. Pedersen, *Langmuir*, 2007, 23, 460-466.
- 481 16. C. A. Dreiss, E. Nwabunwanne, R. Liu and N. J. Brooks, *Soft Matter*, 2009, 5, 1888-1896.

483	17.	M. Valero and C. A. Dreiss, <i>Langmuir</i> , 2010, 26 , 10561-10571.
484 485	18.	C. Perry, P. Hebraud, V. Gernigon, C. Brochon, A. Lapp, P. Lindner and G. Schlatter, <i>Soft Matter</i> , 2011, 7, 3502-3512.
486 487	19.	J. Qin, X. W. Meng, B. J. Li, W. Ha, X. Q. Yu and S. Zhang, J. Colloid. Interf. Sci., 2010, 350 , 447-452.
488 489	20.	J. Huang, Z. Zhou, M. Wei, Y. Chen and P. R. Chang, J. Appl. Polym. Sci., 2008, 107, 409-417.
490	21.	M. Valero and C. A. Dreiss, <i>Langmuir</i> , 2010, 26 , 10561-10571.
491 492	22.	J. Szejtli, in <i>Comprehensive supramolecular chemistry</i> , ed. J. M. Lehn, Pergamon Press, Oxford, 1996, vol. 3, pp. 5-40.
493	23.	M. Taghvaei and G. H. Stewart, Anal. Chem., 1991, 63, 1902-1904.
494	24.	W. Y. Kuo and H. M. Lai, Polymer, 2011, 52, 3389-3395.
495 496	25.	I. V. Terekhova, O. V. Kulikov, R. S. Kumeev, M. Y. Nikiforov and G. A. Al'per, <i>Russ. J. Coord. Chem.</i> , 2005, 31 , 218-220.
497	26.	S. X. Ji and J. Y. Walz, J. Phys. Chem. B, 2013, 117, 16602-16609.
498 499	27.	Y. C. Liang, G. Gillies, H. Patel, L. Matia-Merino, A. Q. Ye and M. Golding, <i>Food Hydrocolloid</i> , 2014, 36 , 245-255.
500	28.	A. Guinier, Ann. Phys. (Paris), 1939, 12, 161-237.
501 502 503	29.	M. A. J. Gillissen, M. M. E. Koenigs, J. J. H. Spiering, J. A. J. M. Vekemans, A. R. A. Palmans, I. K. Voets and E. W. Meijer, <i>J. Am. Chem. Soc.</i> , 2014, 136 , 336-343.
504	30.	D. I. Svergun, J. Appl. Crystallogr., 1992, 25, 495-503.
505 506	31.	S. W. Yeh, K. H. Wei, Y. S. Sun, U. S. Jeng and K. S. Liang, <i>Macromolecules</i> , 2005, 38 , 6559-6565.
507	32.	S. R. Kline, J. Appl. Crystallogr., 2006, 39, 895-900.
508 509	33.	L. K. Shrestha, O. Glatter and K. Aramaki, J. Phys. Chem. B, 2009, 113, 6290-6298.
510 511	34.	J. M. Franklin, L. N. Surampudi, H. S. Ashbaugh and D. C. Pozzo, <i>Langmuir</i> , 2012, 28 , 12593-12600.
		21

- 512 35. S. Manet, A. Lecchi, M. Imperor-Clerc, V. Zholobenko, D. Durand, C. L. P.
 513 Oliveira, J. S. Pedersen, I. Grillo, F. Meneau and C. Rochas, *J. Phys. Chem. B*,
 514 2011, 115, 11318-11329.
- 515 36. H. C. Liao, C. S. Tsao, T. H. Lin, C. M. Chuang, C. Y. Chen, U. S. Jeng, C. H.
 516 Su, Y. F. Chen and W. F. Su, *J. Am. Chem. Soc.*, 2011, 133, 13064-13073.
- 517 37. P. Debye, H. R. Anderson and H. Brumberger, J. Appl. Phys., 1957, 28, 679-683.
- 518 38. C. Travelet, P. Hebraud, C. Perry, C. Brochon, G. Hadziioannou, A. Lapp and G.
 519 Schlatter, *Macromolecules*, 2010, 43, 1915-1921.
- 520 39. J. Dey, S. Kumar, S. Nath, R. Ganguly, V. K. Aswal and K. Ismail, *J. Colloid.*521 *Interf. Sci.*, 2014, 415, 95-102.
- 522 40. S. Manet, A. Lecchi, M. Imperor-Clerc, V. Zholobenko, D. Durand, C. L.
 523 Oliveira, J. S. Pedersen, I. Grillo, F. Meneau and C. Rochas, *J. Phys. Chem. B*,
 524 2011, 115, 11318-11329.
- 525 41. T. E. Girardeau, T. J. Zhao, J. Leisen, H. W. Beckham and D. G. Bucknall,
 526 *Macromolecules*, 2005, 38, 2261-2270.
- 42. A. Becheri, P. Lo Nostro, B. W. Ninham and P. Baglioni, *J. Phys. Chem. B*, 2003,
 107, 3979-3987.
- 529 43. A. Harada, Coordin. Chem. Rev., 1996, 148, 115-133.
- 530 44. P. Lo Nostro, J. R. Lopes, B. W. Ninham and P. Baglioni, *J. Phys. Chem. B*, 2002,
 531 106, 2166-2174.
- 532 45. H. Fujita, T. Ooya and N. Yui, *Macromol. Chem. Phys.*, 1999, 200, 706-713.
- 533 46. M. Valero, C. Carrillo and L. J. Rodriguez, Int. J. Pharm., 2003, 265, 141-149.
- 534

535

536 Table. 1 Fit parameters for the DAB and cylindrical object model applied to SAXS

Sample	length, L	Radius, R	β -CD per stack, N	Characteristic length,				
	(nm)	(nm)	(number)	ξ				
				(nm)				
5-0	8.20	1.02		28.3				
5-1	1.23	0.62	1-2	18.3				
5-9	2.37	0.64	3-4	15.5				
538								
539								
540								

537 data for 5% PL F108 with 0, 1 and 9% $\beta\text{-CD}$ addition.

541

Soft Matter Accepted Manuscript

542	Figure captions
543	Figure 1. Overview of the SAXS patterns of the eight samples studied. For clarity of
544	presentation, the zero point of the intensity scale has been shifted to fit
545	individual pattern.
546 547	Figure 2. Cross-sectional PDDFs for samples 5-0, 5-1 and 5-9.
548	
549	Figure 3. Observed (open squares) and calculated (solid line) SAXS patterns of the
550	sample 5-0. The dotted and dashed lines indicate, respectively, the
551	contributions from the large-scale network appeared in the solution
552	employing the DAB model and from the non-interacting cylinders of 1 nm
553	in radius and 8 nm in length.
554	
555	Figure 4. Observed (open squares) and calculated (solid line) SAXS patterns of the
556	sample 5-19. The scattering profile can be decomposed into two
557	components: the form factor $P(q)$ (dashed line) and the structure factor $S(q)$
558	(dotted line). The form factor $P(q)$ contains the contributions from the
559	lamellas (open circles) and cylinders (open triangles) in the sample.
560 561	Figure 5. Observed SAXS patterns of the dilute samples 5-0 (open squares), 5-1 (open
562	circles), and 5-9 (open triangles). The solid line on each set of data
563	indicates the calculated scattering profile of the corresponding sample. The
564	schematic representation of the local (high q) and global (low q) structures
565	of the samples 5-0 and 5-9 were shown as well.
566	

567	Figure 6. Observed SAXS patterns of the dense samples 5-10 (open squares), 5-12
568	(open circles), 5-14 (open triangles), 5-16 (open diamonds), and 5-19 (open
569	stars). The solid line on each set of data indicates the calculated scattering
570	profile of the corresponding sample.

Figure 7. Observed (open circles) and calculated (solid and dashed lines) structure 571 572 factors of the sample 5-10. The structure factor may be distinguished into 573 tow separated sets of reflection, marked α -phase (solid line) of periodicity 574 50 nm and β -phase (dashed line) of periodicity of 96 nm. Only the first order reflection is revealed in each phase. The intensity of the α -phase is 575 576 largely higher than that of the β -phase, showing the β -CD are mostly 577 localized into the α -phase. The insets show the size distributions of the 578 α -phase (left panel) and β -phase (right panel) obtained from the profile 579 analysis.

580

Figure 8. Observed (open circles) and calculated (solid and dashed lines) structure factors of the sample 5-14. Reflections of up to the third order are revealed in both phases, with the periodicities of the α -phase and β -phases shrink to 32 and 41 nm, respectively. The insets show the size distributions of the α -phase (left panel) and β -phase (right panel) obtained from the profile analysis.

587

Figure 9. Observed (open circles) and calculated (solid and dashed lines) structure
factors of the sample 5-19. Reflection peaks of up to the eighth order are
revealed in the α-phase, while only the first order peak of the β-phase is

591	revealed.	The	insets show	the	size	distributions	of	the	α-phase	(left	panel)

592 and β -phase (right panel) obtained from the profile analysis.

593

594



597 Figure 2



598 599 Soft Matter Accepted Manuscript





















615 Scheme captions

- 616 Scheme 1. Schematic plots of the correlated structures of β -CD in samples (a) 5-10, (b)
- 617 5-12 and 5-14, and (c) 5-16 and 5-19.
- 618 Scheme 2. Schematic plots of the structures of PL F108 and correlated β -CD in
- 619 samples (a) 5-0, (b) 5-1, (c) 5-9, and (d) 5-19.

620



635 Scheme 2



636

Graphical Abstract

