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Abstract

The evolution of the fine structures of self-assembled polypseudorotaxane (PPR) in Pluronic (PL F108) solutions containing dilute to dense beta-cyclodextrin (β-CD) was illustrated for the first time by small angle x-ray scattering (SAXS). Dense β-CD (-19 w/v\%) was found feasible to be dispersed in 24% citric acid solution. Five percentage of PL F108 formed cylindrical micelles of 1 nm in radius and 8 nm in length in the presence of 24% citric acid through the dehydration of citric acid and citrate. PPR was formed through the host-guest interaction between PL F108 and 27 β-CD. In dilute β-CD system (1%) , the single chains of PPR with separated β-CD 28 stacks on PL F108 were formed. The numbers of β -CD in each stack increased from 1 to 4 with increasing β-CD concentration to 9%. In dense β-CD system, PPR condensed to correlated structures majorly composed of two unit blocks through the hydrogen bonds between PPRs. Two distinguishable correlated domains with 32 correlation lengths of 50 nm (marked α -phase) and 46 nm (marked β -phase) along the chains, but without fine periodic structure within each individual domain, were identified in the 10% β-CD solution. Periodic stacking of β-CD in the domains developed in the 12% solution. As β-CD concentration increased from 12 to 19%, the 36 correlated heights of α and β phases shrank from 41 and 32 nm into 30 and 10 nm, 37 respectively. There were 48 β-CDs that stabilized on each PL F108 chain in the 19% β-CD system, which is in a good agreement with the stoichiometry.

Keywords: self-assembly, polypseudorotaxane, Pluronic, β-CD, SAXS,

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41 **Introduction**

42 Cyclodextrins (CDs) are alpha-1,4 linked cyclic oligosaccharides, mainly consisting 43 of 6, 7 or 8 alpha-D-glucopyranose units, known as alpha-, beta- or gamma-CD, 44 respectively.¹ The hydrophilicity of the outer rims and the relative hydrophobicity 45 inside cavities render CDs to form stable inclusion complexes (ICs) with organic 46 molecules, such as flavours, pigments, vitamins and drugs.^{2, 3} In addition, the 47 formation of sequential ICs of several CD molecules within one polymer chain leads 48 to a unique supramolecule, namely the polypseudorotaxane $(PPR)⁴$. The driving 49 forces of this self-assembling process include the hydrophobic interaction between 50 polymers and CD cavities, the hydrogen-bonding among the rims of neighbouring CD on the same polymer chain, and the subsequent loss of PPR solubility.⁵ 51

52 Pluronic® (PL) is a triblock copolymer with a configuration of PEG-PPG-PEG. It 53 is a nonionic surfactant and has been widely used in pharmaceutics on taking the 54 advantage of its self-assembling ability.⁶ At its critical micelle concentration (CMC), 55 the hydrophobic interaction among PPG segments and the phase separation between 56 PEG and PPG blocks attributed to their various hydrophobicity in the solution induce 57 the micellization.¹ However, the self-assembled structure of PL in solution can vary 58 not only with its concentration,⁷ but also be affected by the presence of organic or 59 inorganic salt, such as potassium chloride, $8, 9$ sodium chloride, $10-12$ sodium 60 carbonate, 13 and salicylic acid.¹⁴

When introducing CDs into PL, several CDs can thread onto the PPG or PEG segments of the PL to form PPRs. The interaction between CDs and PL blocks has been studied, through which various nano-structures have been identified in the 64 solutions containing CD and PL. These include dissolute¹⁵⁻¹⁷ or growing micelle,¹⁸

65 hollow sphere,¹⁹ rod-like¹⁶ and platelet structures^{18, 20} that have all been characterized 66 by using small angle x-ray/neutron scattering (SAXS/SANS).

67 Restricted by the low solubility of native β-CD (1.85 g/100 mL water at 25^oC), 68 modified β-CD was commonly used in investigating the self-assembled structures of 69 PPR. Previous studies of β-CD based PPR were focused either on low concentration 70 of native β-CD or on high concentration of modified β-CD.^{16, 21} It has been realized 71 that the solubility of β -CD can be effectively enhanced by using organic acids, such as 72 citric acid, lactic acid and malic acid.^{22, 23} Studies made on the nanostructures of PPRs 73 at high native β-CD concentrations are then possible in an aqueous system containing 74 organic acids. In fact, β-CD can be evenly dispersed in citric acid solution, with a 75 solubility that is 10 times higher than in pure water.²⁴ The role of citric acid is to break 76 the hydrogen bonds among β-CD molecules.²⁵

In this study, the evolution of nanostructure of β-CD/PL in citric acid solution 78 containing 5% PL F108 and various β-CD concentrations $(0~19%)$ were investigated by SAXS measurements. SAXS measurement is capable in identify the correlated structure in the nanometer scale and is suitable for examining the self-assembling structure of PL F108 itself and β-CD/PL systems. Analysing the SAXS patterns with appropriate models allow quantitative extractions of PL F108 and β-CD based PPR structures in citric acid solution, which provide detailed insights of these self-assembled supramolecules.

85

86 **Experimental section**

87 **Materials**

- 89 Park, New York, USA). β -CD (purity \geq 97%) was obtained from Sigma-Aldrich (St.
- Louis, MO, USA). All chemicals used in the present study were analytical grade.

Preparation of PL F108 and β-CD/PL samples

92 A β-CD stock solution (20 w/v% β-CD) was prepared with 24% citric acid solution. Before mixing with the β-CD stock solution, an adequate amount of PL F108 pellet was dissolved with 24% citric acid solution as well. Results of our measurements show that the 24% citric acid solution can effectively separate β-CD molecules, resulting in a clear and well dispersed solution. Below which, the aggregations of β-CD are able to be detected by vision. The depletion of flocculation due to the decrease of dispersibility had been reported when a high concentration of citric acid is 99 used.^{26, 27} In order to minimize the scattering background contributed by citric acids, the 24% citric acid solution was chosen as the solvent in this study. The β-CD/PL 101 mixtures contained various concentrations of β-CD (0-19%) and 5% PL F108 in 24% 102 citric acid solutions. The mixtures were stored at $6 \pm 2^{\circ}$ C for 12 h and then at 25^oC for an additional 24 h before taking SAXS measurements. This process amounts to a total setting time of 36 h for the β-CD/PL in the solution. The concentrations of PL F108 and β-CD were used to mark the samples. As an example, 5-19 indicates the sample contains 19 w/v% β-CD and 5 w/v% PL F108 in the 24% citric acid solution.

Small angle X-ray scattering (SAXS) measurement

The SAXS experiment was performed on the beamline 23A1, operated in the transmission mode, at National Synchrotron Radiation Research Center (NSRRC), Taiwan. The sample was loaded in a standard solution cell for the measurement, sealed with kapton films. The data were collected using photons of incident energy 12

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113 keV (incident wavelength $\lambda = 1.033$ Å), with a sample-to-detector distance of \sim 3 114 meters. This setup provides a *q*-range from 0.006 to 0.28 \AA ⁻¹ available for 115 measurements, where scattering vector $q = 4\pi \sin\theta / \lambda$ and θ is half of the scattering 116 angle.

117 **SAXS data analysis**

118 It has been known that the radius of gyration (R_g) of the scattering object in solution 119 can be obtained from Guinier analysis, 28 and the pair distance distribution function (PDDF) can be extracted from the indirect Fourier transform (IFT) analysis²⁹ of the 121 scattering profile. The PDDF was obtained by employing the GNOM program in the ATSAS suite, which was developed by the BIO-SAXS group at EMBL-Hamburg.³⁰ 123 The model fitting method suggested by Yeh et al.³¹ was used to separate the form 124 factor $P(q)$ from the structure factor $S(q)$ that could appear in the scattering pattern $I(q)$ 125 = $P(q)S(q)$. The form factor contains the intensity scattered from the shape of the 126 object; whereas the structure factor links to the diffraction intensities from the 127 periodic structure of the correlated domains. Two scattering profiles that link to form 128 factor were considered: one models the low $q \leq 0.01 \text{ Å}^{-1}$ scattering, while the other 129 describes the high *q* scattering. The form factor was analyzed employing the IGOR 130 Pro 6.11 program package provided by NIST (National Institute of Standard and 131 Technology) Center for Neutron Research (NIST, Gaithersburg, MD, USA).³² The 132 structure factor, on the other hand, was analyzed by fitting the diffraction peaks to the 133 diffraction profiles of finite-sized domains. It is known that the width of a diffraction 134 peak reflect the spatial extension of the periodicity, which corresponds to the size of 135 the correlated domain in the present cases. The line profile of a diffraction peak from 136 a size-dispersed supramolecule assembly was obtained by combining the diffraction 137 profiles contributed from each individual domain in the assembly. The line profile of

Results and discussion

Qualitative analysis

SAXS measurements were carried out on the mixtures containing various 144 concentrations of β-CD (0~19%) in 5% PL F108 that was prepared with 24% citric α acid_(aq). The scattering patterns were used to extract the micelle conformation and size as well as the stacking of polypseudorotaxane formed by ICs of β-CD and PL F108. The scattering patterns in Fig. 1 show clearly that the structures of PL F108 and β-CD changed significantly with the β-CD concentration. Interestingly, the scattering patterns can be divided into two groups according to their profiles. The samples, such as 5-0, 5-1 and 5-9, generate only scattering from the form factor are designated as the dilute systems; whereas those, such as 5-10, 5-12, 5-14, 5-16 and 5-19, producing diffraction peaks as well are designated as the dense systems.

The scattering patterns in the low- q regime (0.006 to 0.01 \AA^{-1}) of the dilute 154 systems can be expressed by the power law $I(q) = Aq^{-\alpha}$, with an exponent α around 2.5 (Fig. 1, 5-0, 5-1 and 5-9), signalling the existence of a large-scale structure in the dilute solutions. It is known that the mass and surface fractals will give rise to values of 1~3 and 3~4, respectively, for the scattering exponent. The low-*q* scattering with α \sim 2.5 observed in the dilute systems reflects the scattering contributed mainly from the structure of the mass fractal. We believe that this large-scale structure is contributed from the PL F108 in the sample 5-0, and from the PL F108 as well as from the PPR in the samples 5-1 and 5-9. On the other hand, the scattering exponent in the high-*q* regime of the dilute systems was found to be near 1, reflecting the

163 scattering from elongated cylindrical objects. It is interesting to note that the onset of 164 high-q scattering regime shifts from at $q \sim 0.07 \text{ Å}^{-1}$ for the sample 5-0 to $q \sim 0.16 \text{ Å}^{-1}$ 165 for the sample 5-9, indicating the size of the cylindrical objects was reduced upon 166 incorporating $β$ -CD into the mixtures.

The appearances of diffraction peaks in the scattering patterns of the dense β-CD systems show that the interactions among the β-CDs have become strong enough to form correlated structures. As an example, the broad peak at $q = 0.013 \text{ Å}^{-1}$ in the 5-10 pattern (Fig. 1, 5-10) corresponds to a correlated size of 50 nm for the β-CD. The diffraction peaks shift gradually to larger *q* positions in the denser systems, indicating the correlated domains condensed into smaller sizes. Higher order diffraction peaks are clearly revealed in the samples with a β-CD concentration higher than 12%. In 174 addition, a q^2 scattering profile was observed in the low- q regime of the dense systems, revealing scattering from two-dimensional (2D) geometry of lamellar. Interestingly, similar high-*q* scattering profiles were observed for the samples with β-CD concentration higher than 9%, showing the conformation of the small entities in the system has stabilized.

179 **Guinier analysis of the dilute systems**

180 Guinier region can be identified in the scattering profiles of samples 5-0, 5-1 and 5-9, expressed as $I(q) \propto \exp(-R_g^2 q^2)$ 181 expressed as $I(q) \propto \exp(\frac{-R_g}{q})$, where R_g is the gyration radius of the scattering

182 object.²⁸ The absolute value of the slope in the $\text{Ln}\lbrace I(q)\rbrace$ versus q^2 plot (Guinier plot) 183 gives the value of $R_g^2/3$. The R_g values are 25.3, 7.14, and 7.75 nm for samples 5-0, 184 5-1 and 5-9, respectively. It is interesting to point out that a significant reduction (by a 185 factor of \sim 3) in the aggregated sizes of PL F108 to PPR. Unfortunately, Guinier 186 analysis is not possible to perform on the dense systems, where the low-*q* scattering

187 profile is severely disturbed by the appearance of diffraction peaks.

188 **Indirect Fourier Transform (IFT) analysis**

189 The IFT analysis is a model independent method to obtain information on the spatial 190 distribution of mass density in the solution. This analysis gives the pair distance 191 distribution function, *p*(*r*) known as.

192
$$
p(r) = \frac{1}{2\pi^2} \int_0^{\infty} I(q)qr \sin(qr) \, dq
$$
 (1)

9 193 where *q* is the magnitude of the scattering vector. This *p*(*r*) gives the probability to 194 find out a mass point at a distance *r* away from a selective point within the scattering 195 object, weighed by the difference in scattering length density (SLD) between these 196 two points. In the case of a rod-like object, the PDDF can be determined by $p(r) = \gamma$ 197 *c*(*r*) × *r*, where *γ*_{*c*}(*r*) is the characteristic function of the thickness. Fig. 2 shows the *p*(*r*) 198 obtained for the dilute samples. All of three $p(r)$ distributions show a symmetric, 199 Gaussian-like peak with a maximum at *r* = 0.5, 0.45, and 0.38 nm for the samples 5-0, 200 5-1, and 5-9, respectively. The shape of $p(r)$ displays a fixed SLD for the rod-like 201 cylinder object over the entire cross-section.^{29, 33, 34} It is known that PL formed 202 core-shell micelle with a hydrophobic PPG core and a hydrophilic corona.³⁵ For 203 sample 5-0 the cross-section of the rod-like micelle is composed of a PPG core with a SLD of $ρ_{PPG} = 9.47 \times 10^{-6}$ Å⁻² and a PEG corona with a SLD of $ρ_{PEG} = 1.11 \times$ 205 10^{-5} Å^{-2 35} Only a cylindrical object could be identified with the weak scattering 206 contrast between PPG core and PEG corona. For samples 5-1 and 5-9 the PPR is composed of a β-CD shell ($ρ_{β-CD} = 1.3 \times 10^{-5}$ Å⁻²) and a PL F108 core ($ρ_{PLF108} = 1.07$ $\times 10^{-5}$ Å⁻²). The scattering contrast between β-CD and 24% citric acid solvent (ρ_{sol} = 209 1.03 \times 10⁻⁵ Å⁻²) is much better than that between PL F108 and the solvent. We believe 210 that the scattering profiles of samples 5-1 and 5-9 were contributed by the rod-like 211 PPRs composed of numerous β-CDs. These results suggest that cylindrical objects,

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212 but not core-shell cylinder ones, should be incorporated in describing the scattering

213 profile.

214 **Model fitting**

215 **Dilute systems (0, 1 and 9%** β**-CD**)

As mentioned above, the scattering patterns of dilute systems can be modeled by a low-*q* power law and a high-*q* cylindrical conformation scattering. Hence, the 218 expression for scattering profile should contains scattering from large-scale network 36 219 that can be formulated using the Debye-Anderson-Brumberger (DAB) model³⁷ and 220 from the non-interacting cylinders in the solutions. Fig. 3 shows the observed (open squares) and calculated (solid line) scattering profiles of sample 5-0. The contributions from the cylinders (dashed line) and from the DAB model (dotted line) are displayed in Fig. 3 as well. The DAB model may be expressed as

224
$$
I(q) = \frac{A\xi^2}{\left[1 + \left(q\xi^2\right)\right]^2},
$$
 (2)

225 where ξ is the characteristic length of the domain size and *A* is a scale factor. The 226 form factor of cylinder can be expressed as

$$
227 \t P_{cyl}(q) = A \frac{1}{\pi r^2 L} \int_{0}^{\pi/2} f^2(q, \alpha) \sin(\alpha) d\alpha , \t (3)
$$

228 where
$$
f(q, \alpha) = 2(\rho_{cyl} - \rho_{sol})\pi^2 Lj_0(qL/2\cos\alpha) \frac{j_1(qr\sin\alpha)}{qr\sin\alpha}
$$
, $j_0(x) = \sin(x)/x$, $j_1(x)$

229 is the first order Bassel function, α is the angle between the axis of the cylinder and 230 the scattering vector *q, r* and *L* are the radius and the length of cylinder, respectively. 231 $\rho_{cyl} = 1.07 \times 10^{-5}$ Å⁻² and $\rho_{sol} = 1.03 \times 10^{-5}$ Å⁻² are the SLD of PL F108 and 24% citric

232 acid solution, respectively. The fit allows extracting *r* and *L* of the cylinder.

233 **Dense systems (10, 12, 14, 16 and 19%** β**-CD)**

10

234 The structure factor $S(q)$ could be extracted by separating the form factor $P(q)$ from 235 the scattering pattern $I(q)$. Taking the pattern of sample 5-19 (Fig. 4) as an example, 236 the form factor contributed from the lamellar structure was expressed as 2 $a^2 \sigma^2$ $\frac{p}{2}$ | 1 – cos(q δ)e $\frac{1}{2}$ $P_{lam}(q) = \frac{2\Delta\rho^2}{q^2}\left[1-\cos(q\delta)e^{-q}\right]$ 237 $P_{lam}(q) = \frac{2\Delta\rho^2}{q^2} \left[1 - \cos(q\delta)e^{-q^2\sigma^2/2} \right],$ (4)

238 where $Δρ$ is the scattering contrast between the lamellar structure (β-CD) and the 239 solvent (24% citric acid solution), σ is the mean variation of the layer thickness, δ is 240 the thickness of lamellar structure. On the other hand, the scattering contributed from 241 the cylinder structure (revealed more clearly in the high-*q* regime) of the dense 242 systems is complicated by the diffraction peaks (Fig. 1). The form factor associated 243 with the cylinders of sample 5-9 was then adopted for that of dense systems. This is 244 justified since the scattering from the cylinders has stabilized in systems with a β -CD 245 concentration higher than 9%, as can be seen in Fig. 1. The scattering contributed 246 from form factor was thus expressed as the linear combination of $P_{lam}(q)$ and $P_{cyl}(q)$ 247 with a cut-off wave vector³⁸ q_0 in weighting the two contributions,

248
$$
P(q) = A \exp\left(-\frac{q}{q_0}\right)^{5.33} \times P_{lam}(q) + B \exp\left[1 - \exp\left(-\frac{q}{q_0}\right)^{5.33}\right] \times P_{cyl}(q)
$$
, (5)

249 where *A* and *B* are the intensity scale factors. In this form the lamellar form factor 250 dominates in the low- q ($q < q_0$) regime, while the cylinder form factor dominates in 251 high-*q*'s.

252 Structure factor is expressed using the Debye-Scherrer equation with Gaussian 253 peak profiles as

254
$$
I_{hkl}(\theta) = \frac{1}{(w/2)\sqrt{2\pi}} \exp\left[\frac{-(\theta - \theta_c)^2}{2(w/2)^2}\right],
$$
 (6)

255 where θ is the half of the scattering angle, θ_c is the peak position, and *w* is the full width at half maximum of the peak profile. The pattern of sample 5-19 was thus separated into three contributions of from the cylinders in the mixture (open triangles, marked *Pcyl*(*q*) in Fig. 4), from the lamellar conformation (open circles, marked *Plam*(*q*) in Fig. 4), and from periodically correlated β-CD (dotted lines in Fig. 4).

260

261 **Effect of** β**-CD concentration on PPR structure**

262 **Nanostructure of** β**-CD/PL F108 in dilute systems**

263 In this section, we focus on analyzing the structure of the cylinders in the mixture. Fig. 264 5 shows the observed and calculated patterns of samples 5-0, 5-1 and 5-9, where no 265 diffraction peaks are revealed yet. A rod-like cylinder with $r \sim 1$ nm and $L \sim 8$ nm 266 were obtained in the sample 5-0. It is known that R_g of an elongated object has been calculated to be $\left(\frac{r^2}{2}\right)$ $\frac{r^2}{2} + \frac{L^2}{12}$ 267 calculated to be $\left(\frac{r^2}{2} + \frac{L^2}{12}\right)^{\frac{1}{2}}$ and it was calculated to be 24.7 nm for 5-0 sample. This 268 is in good agreement with the R_g value (25.3 nm) extracted from the Guinier analysis. 269 This cylindrical object is understood to be the micelle formed from PL F108 270 originated from the hydrophobic interaction between the PPO segments in the mixture. 271 Cylindrical micelle has been observed in PL F108 in 0.5 M sodium carbonate aqueous 272 solution.¹³ Cylindrical micelle of other PL series has also been observed in inorganic 273 salt⁸⁻¹² and in butan-1-ol.³⁹ In the present systems, citric acid dissociates 1 proton in 274 24% citric acid solution with a pH value of 1.5. We believed that the citric acid and its 275 conjugated salt, known as citrate, are responsible for the dehydration process to form 276 cylindrical PL F108 micelles. Similar phenomenon has also been observed in a 277 separated study. 14

12 278 It is known that characteristic length ξ can be used to describe the large-scale 279 network structure in the mixture of PL F108 and citric acid using the DAB model.³⁶ A

characteristic length of 28.3 nm for the network in sample 5-0 is identified. This network is formed from the entanglement of the unassociated PL F108 unimer in the mixture. The citric acid is believed to play a crucial role in the formation of the extended network, initiated by the development of hydrogen bonds between citric acid and PL F108.

285 Introducing β-CD into PL F108 drives the onset of q^{-1} behavior in the scattering 286 pattern to appear at larger *q* (Fig. 5, 5-0 and 5-1), indicating the formation of smaller 287 cylindrical objects. These objects are the rod-like $PPR³⁸$ formed through the 288 host-guest interaction between β-CD and PL F108. In sample 5-1, the molar ratio 289 between PG and β-CD is 19.6, the PL micelle will not be fully de-assembled. We 290 believe that both PL F108 micelle and rod-like PPR co-exists in the mixture, since the 291 scattering from micelle vanished once PG/B -CD was below 3.1¹⁶ However, only the 292 scattering from rod-like PPR were identified in the scattering pattern of sample 5-1. 293 As mentioned above the scattering contrast between 24% citric acid solution (1.03 \times 294 10⁻⁵ Å⁻²) and β-CD (1.3 \times 10⁻⁵ Å⁻²) is known to be considerably better than that of 295 between 24% citric acid solution and PL F108 (\sim 1.07 \times 10⁻⁵ Å⁻²).⁴⁰ It is clear that the scattering pattern in the $q > 0.1$ Å⁻¹ regime is dominated by the scattering from β-CD, and reflects the morphology of the stacking of β-CD on PL F108.⁴¹ Table 1 lists the 298 fitted parameters for the samples 5-0, 5-1 and 5-9. The correlated β-CD block was 299 found to be of 0.6 nm in radius and 1.2 nm in length, revealing that single chains of 300 PPR with a stacking of 1 or 2 correlated β-CDs in each unit block were formed in 301 these samples. We note that the radius and length of one β-CD are known¹ to be 0.77 302 and 0.79 nm, respectively.

13 The scattering patterns in the $q \le 0.1$ Å⁻¹ regime of samples 5-1 and 5-9 are nearly identical (Fig. 4). The differences in the two patterns are, however, seen at $q > 0.1 \text{ Å}^{-1}$

305 reflect the structure of the rod-like PPRs are indeed different. The length of PPR 306 increased from 1.2 nm for sample 5-1 to 2.4 nm for sample 5-9 (Table 1), showing the 307 number of stacked β-CD has increased to 3 or 4 β-CD in each correlated block. In the 308 mixture containing α -CD and PEO homopolymer in H₂O, the rod-like PPR s elf-assembled with 7 CDs per stacking block.⁴¹ The number of CD stacked in each 310 block on PL F108 depends on the strength of inter-CD hydrogen bonding 311 (head-to-head or tail-to-tail).^{41, 42} It is known that β-CDs can be dispersed in citric 312 acid solution through the hydrogen bonds among the polar groups (-COOH and -OH) 313 of citric acid and the -OH groups of β -CD located on its outside rim.²⁵ Therefore, the 314 interaction of inter-CD hydrogen bonds in citric acid solution is less effective than 315 those in H2O. This results in a less number of CDs that are stacked in the present 316 mixtures. In sample 5-9 of molar ratio $PG/B-CD = 2$, a PPG segment⁴³ that is fully 317 covered by β-CD can be expected. Thus, a rod-like PPR with a length of \sim 20 nm can 318 be expected in sample 5-9. Interestingly, only cylinders with a length of 2.4 nm were 319 observed in sample 5-9. This suggests that there are numerous separated short PPR 320 instead of forming fewer long rod-like PPRs in 24 w/v% citric acid solution. We 321 hence propose that the β -CDs not only thread onto the PPG segments but also on the 322 PEG segments in considerable portions. The formation of β-CD inclusion complex 323 with PPG homopolymer in the presence of citrate ions has been observed in a 324 previous study.⁴⁴ β-CD has been found to form inclusion complex, with noticeable amount of β-CDs, being reside on PEG segments of PL P85.⁴⁵ It has been reported 326 that the binding constant of naproxen with β -CD in the presence of 1% PEG would be 327 reduced that was attributed to the formation of PEG-β-CD inclusion complexes; 328 therefore, the amount of free CD available for the inclusion with naproxen was 329 $reduced.⁴⁶$

Through DAB model calculations, the domain size in each sample was identified. These results are summarized in Table 1. A domain size of 28.3 nm was obtained for PL F108 (sample 5-0). The domain size decreases to 18.5 nm in sample 5-1, and further to 15.5 nm in sample 5-9. The driving force for a shorter characteristic length in higher β-CD concentration is believed to be due to the stronger inter-CD interaction in higher β-CD environment in the PPR chains and among the rims of the 336 neighboring CD.⁵ In our previous work²⁴, similar results were observed in the sample 337 containing a higher molar ratio of $PG/B-CD = 17.6$, where PPR gels formed with the assistance of cross-link PPR crystallines resulting from the attractive interaction in neighboring PPRs.

Nano structure of β**-CD/PL F108 in dense systems**

Fig. 6 shows the observed and calculated scattering patterns of the dense samples. Through this analysis, the structure factor is effectively extracted from the scattering intensity for further analysis. The diffraction peaks were mainly contributed from the 344 periodic structure of β -CD, since the scattering contrast of β -CD is significantly higher than that of PL F108. The isolated diffraction pattern of sample 5-10 is shown in Fig. 7, where two diffraction peaks can be identified. The intense one corresponds 347 to a domain size of 50 nm ($d = 2\pi/q$), marked as α -phase with $c_{\alpha} = 50$ nm; whereas the 348 less intense one corresponds to a domain size of 96 nm, marked as β-phase with c_6 = 96 nm. It is known that the width of a diffraction peak reflect the spatial extension of the lattice periodicity, which could be used to extract the stacking height of the 351 correlated β-CD. The line profile of a diffraction peak from a size-dispersed assembly can be obtained by combining the diffraction profiles contributed from each correlated grain in the assembly. The solid curves in Fig. 7 indicates the calculated diffraction profile with the size distributions of the two phases shown in the inset to

355 Fig. 7, assuming the sizes consist of multiple stacking of the c_{α} or c_{β} unit blocks. It is 356 interesting to find that most of the correlated β-CDs in both phases interconnected 357 into stacking of two unit blocks of stacking heights $d_{\alpha} = 96$ nm and $d_{\beta} = 188$ nm. 358 Moreover, the intensity of the α -phase is considerably higher (3.2 times more intense), 359 showing that most of the correlated β -CDs condensed in the α -phase. There is no 360 higher order reflections that appear in the diffraction pattern, showing the condensed 361 β-CDs within the unit blocks are not packed into periodic structure yet.

Two sets of the diffraction peaks with different intensity could be identified in the samples 5-12 and 5-14. The more intense one, that is the α-phase, shrinks to a shorter 364 spatial period of $c_α = 41$ nm, while the β-phase shrinks to $c_β = 32$ nm (Fig. 8). This is understandable that the correlated β-CDs will condense into a more compact structure in a solution with higher β-CD concentration. The peak widths of the β-phase are 367 much broader than those of the α -phase, indicating that α and β phases are originated from separated structures. In addition, the size analysis based on the diffraction profiles giving 94% of the α-phase and 95% of the β-phase are composed of two unit blocks in sample 5-14, as shown in the inset in Fig. 8. Higher order diffraction peaks of up to the (003) reflection associated with both phases are revealed in the diffraction 372 pattern, indicating the β -CDs in the unit blocks have built up periodic structures to form shorter periodicities. It is known that the total diffraction intensity is directly proportional to the mass of the phase that generates the diffraction pattern. The 375 intensity ratio between that from the α -phase and from the β-phase can be used to 376 calculate the mass ratio of the two phases in the solution. A mass ratio of $m_{\alpha}/m_{\beta} =$ 73/27 was obtained for sample 5-14, when taking the diffraction intensity is dominated by the mass of the unit block. This can be justified since the main 379 difference between the α-phase and β-phase is the numbers of β-CD that are

dilute one shrinks to 32 nm long (Scheme 2(b)). It is the two correlated structures that generate the two series of diffraction peaks. In the denser mixtures of samples 5-16 and 5-19, there are 24 correlated β-CDs that packed into a unit segment of 30 nm long, 408 but no significant amount of correlated β -CD that appears in between the unit segment (Scheme 1 (c)).

As mentioned above, the scattering pattern of sample 5-9 can incorporated directly in the fitting process to model the high *q* scattering for samples in the dense β-CD system. Two possibilities can contribute to this result. First, these rod-like PPRs are the building blocks of the lamellae. Second, these rod-like PPRs does not condensed into the lamellar structure but remain in the 'freely dispersed' form in the 415 solution. Based on the fact that the periodic structure composed of 24 β-CDs, the later scheme is likely to occur in the present systems.

Schematic representation of β**-CD/PL F108 in citric acid**

solution

A schematic representation of the nano-structure in the mixtures with or without β-CD is shown in scheme 1. PL F108 forms cylindrical micelle in 24 w/v% citric acid 422 solution [Scheme 2(a)]. With the addition of 1% β-CD, a rod-like PPR is formed with 423 1~2 CDs that are aggregated in each threaded β-CD stack [Scheme 2(b)]. For the system with 9% β-CD in the mixture, the number of CD in the β-CD block increases to 3~4 units, and the characteristic length decreases, which we believe is triggered by 426 the formation of hydrogen bonds between β-CDs [Scheme 2(c)]. In addition, β-CD can thread not only on the PPG segment but also on the PEG segment of the PL F108 chain. Increasing β-CD concentration further to 19%, the aggregated structure

Conclusion

Investigated by small angle x-ray scattering measurements, the self-assembled structures of PL F108 in 24% citric acid solution containing 0 to 19% β-CD were revealed. PL F108 forms cylindrical micelles in 24% citric acid solution, originated from the dehydration effect of citric acid and citrate. The threaded β-CDs formed numerous separated short PPR not only on the PPG segment but on the PEG segment instead of long single chain PPR in the dilute systems. The attractive interactions among neighboring β-CDs on the single PL F108 chain and among the PPR single chains, presumably through the hydrogen bonding, result in shorter characteristic 442 length. In the dense systems, the attractive interactions drive the β-CDs to pack into a 443 periodic structure. The size of the periodic structure decreased with increasing β-CD concentration in the mixture. Furthermore, the number of the β-CDs on single PL F108 chain has also been revealed.

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536 Table. 1 Fit parameters for the DAB and cylindrical object model applied to SAXS

537 data for 5% PL F108 with 0, 1 and 9% β-CD addition.

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Figure 7. Observed (open circles) and calculated (solid and dashed lines) structure factors of the sample 5-10. The structure factor may be distinguished into tow separated sets of reflection, marked α-phase (solid line) of periodicity 50 nm and β-phase (dashed line) of periodicity of 96 nm. Only the first order reflection is revealed in each phase. The intensity of the α-phase is largely higher than that of the β-phase, showing the β–CD are mostly localized into the α-phase. The insets show the size distributions of the α-phase (left panel) and β-phase (right panel) obtained from the profile analysis.

Figure 8. Observed (open circles) and calculated (solid and dashed lines) structure factors of the sample 5-14. Reflections of up to the third order are revealed in both phases, with the periodicities of the α-phase and β-phases shrink to 32 and 41 nm, respectively. The insets show the size distributions of the α-phase (left panel) and β-phase (right panel) obtained from the profile analysis.

Figure 9. Observed (open circles) and calculated (solid and dashed lines) structure factors of the sample 5-19. Reflection peaks of up to the eighth order are revealed in the α-phase, while only the first order peak of the β-phase is

- revealed. The insets show the size distributions of the α-phase (left panel)
- and β-phase (right panel) obtained from the profile analysis.
-

597 Figure 2

598

599

0.00 0.02 0.04 0.06 0.08

q $(\AA^{\text{-}1})$

α **(002)** α **(003)**

β β **(003) (002)**

64 96 ⁰ Grain size (nm)

5-14

DOOOOOO

1.0

1.5

dΣ**/d**

615 **Scheme captions**

- 616 Scheme 1. Schematic plots of the correlated structures of β -CD in samples (a) 5-10, (b)
- 617 5-12 and 5-14, and (c) 5-16 and 5-19.
- 618 Scheme 2. Schematic plots of the structures of PL F108 and correlated β-CD in
- 619 samples (a) 5-0, (b) 5-1, (c) 5-9, and (d) 5-19.
- 620

635 **Scheme 2**

636

Graphical Abstract

