

# Journal of Materials Chemistry C

Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

*Accepted Manuscripts* are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this *Accepted Manuscript* with the edited and formatted *Advance Article* as soon as it is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

1 **Enhanced photochromic efficiency of**  
2 **transparent and flexible nanocomposite films**  
3 **based on PEO-PPO-PEO and tungstate**  
4 **hybridization**

5 *Cong Wang, Bing-pu Zhou, Xi-ping Zeng, Ya-ying Hong, Yi-bo Gao and Wei-jia*  
6 *Wen\**

7 C. Wang, B. P. Zhou, X. P. Zeng, Y. Y. Hong, Y. B. Gao and Prof. W.J. Wen

8 Nano Science and Nano Technology Program and Department of Physics, the Hong  
9 Kong University of Science and Technology, Clear Water Bay, Kowloon, Hong  
10 Kong.

11 Tel: (+852)-23587979

12 Fax: (+852)-23581652

13 E-mail: phwen@ust.hk

14 Homepage: <http://www.phys.ust.hk/phwen/>

15  
16  
17  
18

## 1 Abstract

2  
3 Nanocomposite hybrid films were prepared by depositing tungstate on  
4 PEO-PPO-PEO (EPE) templates under acidic conditions by a one-pot self-assembly  
5 sol-gel method. Uniform and transparent films based on this precursor were fabricated  
6 and easily manipulated using a facile casting method. Upon irradiation by sunlight,  
7 the film exhibits a rapid photochromic response that was reversible at room  
8 temperature. UV-Vis analysis revealed that folding of the -CH<sub>2</sub>-O- chains in the  
9 polymer and the larger tungstate clusters increased the red shift for the adsorption of  
10 visible light. The mechanism underlying the effect of hybridization of polyethylene  
11 oxide (PEO) on the enhanced photochromic effect was characterized by NMR, Raman  
12 and FTIR. The degree of folding of the -CH<sub>2</sub>-O- polymer chains was influenced by  
13 the presence of acid, which increased the oxygen coordination of WO<sub>6</sub> and WO<sub>4</sub> in  
14 the tungstate clusters. The characterized W-O bonding peaks shifted upon EPE  
15 incorporation, consistent with changes in oxygen coordination within the material.  
16 This weak coordination between tungstate and -CH<sub>2</sub>-O- improved the intervalence  
17 charge transfer (IVCT). Oxygen from EPE may enter the oxygen sites in the O-W-O  
18 clusters to produce an unbalanced electron state, resulting in oxygen defects, which  
19 has a critical effect on the enhancement of the photochromic response. In addition,  
20 this method provides protons via this weak coordination, thereby dramatically  
21 enhancing the efficiency of the hybrid film.

## 23 1. Introduction

24 Photochromism of composite inorganic-organic materials has been the subject of  
25 intense investigation because of its potential application in optical devices, optical  
26 switches, and chemical sensors.<sup>[1]</sup> Polytungstates (PT) are particularly promising  
27 photochromic materials because of their rich topological, chemical, and physical  
28 properties and because they can be easily and inexpensively synthesized.<sup>[2]</sup> Crystalline  
29 tungstate trioxides usually exhibit good chemical and thermal stabilities with slow  
30 switching speeds. By contrast, conjugated polymers are useful for multicoloration and  
31 improving coloration efficiency and switching speed but suffer from poor stabilities.<sup>[3]</sup>  
32 To overcome the problems associated with using WO<sub>3</sub> and conjugated polymers as  
33 photochromic materials, nanocomposite hybrid PT polymers have been combined to  
34 form polyoxometalates (POMs), and their photoactivity<sup>[4]</sup>, water splitting capability<sup>[5]</sup>  
35 and use in energy-efficient windows<sup>[6]</sup> have been studied extensively.

36 Many studies have examined the optical density and reversibility of POM hybrids  
37 using various organic molecules and solvents,<sup>[7-9]</sup> but photochromic POMs remain  
38 time-consuming to produce, unstable, and difficult to fabricate on a large-scale. The  
39 optical contrast of such hybrids primarily arises from the switching of one component  
40 between different redox states, but the redox reactions that occur within the hybrid

1 tend to be destructive to the polymer. Therefore, it is very important to choose organic  
2 molecule, which not only serves as the proton donor for the coloration and  
3 stabilization of hybrids, but can also act as the electron donor for the formation of a  
4 coordination bond. Moreover, color bleaching may occur due to the high temperatures  
5 necessary to revert the films' color.<sup>[10, 11]</sup> To dynamically control solar radiation  
6 transmittance through nanocomposite materials, various "smart windows" have been  
7 developed and studied extensively. However, some smart windows require a  
8 relatively complex external power supply device to alter their transparency  
9 electrochromically<sup>[12, 13]</sup>, which is incompatible with the goal of decreasing energy  
10 consumption and slowing down heat propagation in commercial buildings.<sup>[14-16]</sup>

11 This article addresses the aforementioned issues by presenting hybrid systems  
12 based on the amphiphilic surfactant EPE and sodium tungstate to enhance  
13 photochromic efficiency. These two components are widely used, chemically stable  
14 structural materials. Using these materials, transparent gels can be easily fabricated on  
15 large areas of film by a one-pot casting method. Supporting PTs in or on uniform  
16 matrices with large surface areas can effectively increase the utility of the composites  
17 by facilitating the formation of functional tungstate species. Compared with previous  
18 studies on hybridization, EPE molecules are only used to create defect sites for PT, not  
19 provide protons. Herein, we demonstrate that this combination contributes to the  
20 unbalanced electron state of the W-O framework, which is beneficial for enhancing  
21 IVCT. The film offers rapid solar response without additional energy consumption  
22 and has the unique ability of recovering full transparency simply by consuming O<sub>2</sub>  
23 without heating, thereby exhibiting great potential for use in photochromic smart  
24 windows.

## 26 2. Experimental Section

### 27 *Photochromic material preparation:*

28  
29 The name of photochromic material was labelled as the order of surfactant (F127) +  
30 tungstate source + acid source (eg. F127+ST+H). F127STH precursor solution was  
31 synthesized by dissolving 3.3% w/v sodium tungstate (Na<sub>2</sub>WO<sub>4</sub>) into 10% w/v  
32 Pluronic F127 DI water solution under strong magnetic stirring at room temperature  
33 for 12h. After completely dissolved, pH value was adjusted to 2 by diluted sulfuric  
34 acid (5 mol/L). Preparation of F127WC precursor solution was also followed this  
35 procedure. 6% w/v ammonium paratungstate was added into 10% w/v F127 solution  
36 and the citric acid was used to adjust pH value to 2. The final film or block of  
37 composite material was fabricated using a sol-gel method by the two clear precursor  
38 solution described above. The two solutions were subjected to a separately vacuum  
39 defoaming process before being cast onto substrates, which were allowed to gelate by  
40 drying at room temperature overnight. These fabricated film samples were black-bold  
41 in text as **F127STH** and **F127WC**. The film for X-ray diffraction (XRD) test was  
42 based on F127WC precursor, whose tungstate weight ranged from 1% w/v to 6% w/v

1 (sample 1-6). Samples a-f contained same concentration of 6% w/v tungstate but 1%  
2 w/v, 2% w/v, 3% w/v, 5% w/v, 8% w/v and 10% w/v F127 concentration,  
3 respectively. The formaldehyde-containing sample was fabricated, in which 35%  
4 formaldehyde-water solution was used to replace the EPE solution. The fast-response  
5 reversible gel was fabricated by facile casting technology on commercial glass or PET  
6 substrates at room temperature. After evaporation, the films were covered with a  
7 single layer of a polyacrylamide polymer framework to enhance their mechanical  
8 properties. The resulting self-supporting nanocomposite films were easily peeled from  
9 the substrates due to the surface wettability of the PET plates. Uniform film thickness  
10 was achieved by casting the same concentration and amount of film-forming solution  
11 on substrates with the same surface area. The procedure of dip-coating method was  
12 that the film was firstly exposed to plasma 2 minutes; then it was dip-coated in an  
13 aqueous precursor for 30 seconds and then exposed to plasma for 2 minutes again after  
14 being dried in air, this process was repeated 10 times. The post-sintering of film  
15 samples were performed in air at 500°C for 2 h and the temperature was increased by  
16 10°C per minute to 500°C. All chemicals were purchased from Sigma-Aldrich and  
17 used without further purification.

18

#### 19 *Characterization:*

20

21 The chemical structures and compositions of the thin films were characterized by  
22 Fourier transform infrared (FTIR) spectroscopy (Perkin Elmer Spectrum GX), X-ray  
23 photoelectron spectroscopy (XPS, Physical Electronics PHI) and Raman spectroscopy  
24 (Renishaw Raman spectroscopy with an excitation wavelength of 540 nm). The FTIR  
25 scans were conducted on a silicon wafer from 4000 to 600  $\text{cm}^{-1}$ , with 32 scans  
26 collected for signal averaging. To compensate for surface-charging effects in the XPS  
27 scans, the binding energy level of  $\text{C}_{1s}$  was set at 285 eV for further data analysis. The  
28 structure of the hybrid thin films was characterized by XRD analysis using a Rigaku  
29 Dmax diffraction system using a  $\text{Cu K}\alpha$  source ( $\lambda=1.54187 \text{ \AA}$ ). UV-Vis transmittance  
30 spectra were measured over a wavelength range of 200 to 1100 nm. The film  
31 coloration and bleaching data were recorded under mercury lamp irradiation with an  
32 average irradiance of  $50\text{mW}/\text{cm}^2$ ; Transmittance data of coloration and bleaching  
33 process were recorded at 640 nm and 600 nm. Scanning electron microscopy (SEM)  
34 images were obtained with a JOEL 6390 scanning electron microscope. The SEM  
35 samples were prepared by sintering sample powder onto a Cu substrate.  
36 High-resolution transmission electron microscopy (HRTEM) images were obtained  
37 with a JEOL-2010 transmission electron microscope at 200 kV.

1

2 **3. Results and discussion**3 **3.1 One-pot synthesis of hybrid film**

4

5

6

Scheme 1. Synthesis of the EPE/WO<sub>3</sub> hybrid film from aqueous precursors

7

8 In this work, the synergistic effect of EPE, acid and tungstate is mainly discussed. The  
 9 self-assembled PT is prepared in the aqueous phase. Under aqueous conditions, EPE  
 10 has a high propensity for partitioning and favors a hexagonal structure as its  
 11 concentration approaches saturation.<sup>[17]</sup> The spherical structure shown in Scheme 1 is  
 12 formed by many EPE molecules. One EPE molecule contains two PEO folding tails,  
 13 which tend to form a double helical cylindrical (DNA-like) shape in water.<sup>[19]</sup> The  
 14 cylindrical structures formed by the two folding C-O structures efficiently form a  
 15 space that is occupied by tungstate anions. These structures not only provide an  
 16 interspace for PT, but also contribute to the formation of the defect sites. The  
 17 photochromic mechanism is described in the previous work<sup>[1, 18]</sup>. Light induces the  
 18 tungstate oxide generate one hole-electron pair, with the electron trapped in PT defect  
 19 site. A proton is then inserted into the PT network. Our method exploits the role of  
 20 acid ions (tridentate carboxylate or sulfate) via weak coordination to contribute to  
 21 separate the function of organic molecule for hybridization. Acidic ions also play a  
 22 crucial role in the formation of interactions between EPE and tungstate in the  
 23 high-yield coloration-bleaching process, which is dependent on the provision of  
 24 protons to form a crown-ether structure.

25

26

27

Figure 1. NMR analysis of tungstate acid with PEO-PPO-PEO

28

29 **Table 1.** Integrated values for chemical groups in <sup>1</sup>H-NMR spectra of different molecules

	EPE <sup>[c]</sup>	C	EPE+W	EPE+C	EPE+WC	EPE+H
a	1.41/1	—	1.35/1	1.13/1	1.05/1	0.95/1
b	—	1.55	—	1.68	1.66	—

30 [a] Specific value ratios of the hydrophobic groups -CH<sub>3</sub>, -CH-, and -CH<sub>2</sub>- and the  
 31 hydrophilic groups -CH<sub>2</sub>-CH<sub>2</sub>-. [b] Specific value ratios of carboxylic and hydroxylic groups.

32 [c] EPE: Pluronic F127; C: citrate; W: tungstate; and H: sulfate.

33

34 To determine how the citrate and sulfate anions influence the EPE molecule  
 35 differently at the same hydrogen ion concentration (pH value), the interaction between  
 36 the acid and the triblock template was measured by NMR (Table 1). Integration of the



1 peaks in the  $^1\text{H-NMR}$  spectra obtained in different chemical buffers demonstrated that  
2 the specific values of the functional groups of different molecules were influenced by  
3 their correlated action in aqueous solution. Resonances were observed for  $-\text{CH}_3$   
4 groups at 1.16 ppm,  $-\text{CH}-$  groups as a broad peak at 3.5 ppm,  $-\text{CH}_2-$  groups at 3.627  
5 ppm, and  $-\text{CH}_2-\text{CH}_2-$  sequences at 3.694 ppm. The hyperfine structure of the EPE  
6 molecule contained broad peaks corresponding to the  $-\text{CH}_2-\text{CH}-$  and  $-\text{CH}_3$  units of the  
7 PO groups that partially overlapped with the large peak of the  $(\text{CH}_2)_2$  units of the EO  
8 groups; thus, it can be concluded that the block copolymer molecules dissolve as  
9 monomers and that the segments of the chains can consequently move freely. The  
10 values presented in Table 1 indicate that the PPO groups tend to dissociate in the core  
11 because  $[\text{WO}_4]^{2-}$  exhibited only minor changes in its specific value in systems  
12 containing surfactant only and systems containing surfactant and acid. However, the  
13 specific values for both the EO and PO groups decreased after the addition of citrate  
14 and sulfate to the solution from initial average values of 1.41/1 and 1.35/1 to 1.13/1  
15 and 0.95/1, respectively. It can be concluded that the symmetric  $-\text{CH}_2-\text{O}-$  molecules  
16 were locked into the interspace of tungstate upon acidification, without retaining their  
17 former symmetric structure. The presence of the acidic anions provided more weak  
18 coordination opportunities for the oxygen and alkane functionalities of the PEO units  
19 interacting with metal oxide anions due to the double helical structure of the alkali  
20 metal-PEO complexes. Moreover, the decrease in the PEO integrated value upon the  
21 addition of sulfuric acid (0.95/1) is larger than that observed upon the addition of  
22 citric acid (1.1/1). According to the Hofmeister series, the sulfate anion contributes  
23 more to the destabilization and destruction of EPE's former asymmetrical state than  
24 the citrate anion does.

25 By contrast, when citric acid interacts with EPE, the integrated peak intensity of  
26 the carboxyl groups of the citrate molecule increased from 1.55 to 1.67 relative to the  
27 hydroxyl groups. This indicates that the carboxyl groups favor the asymmetric  
28 structure of the PEO-PPO-PEO molecules. The two-sided terminal  $-\text{COOH}$  also plays  
29 a role in forming mixed-metal complexes. Under certain pH conditions, a stable  
30 complex is irreversibly formed between PEO and carboxylic groups. During the  
31 process of complex formation, the dissociated carboxylic groups were influenced by  
32 neighboring undissociated PEO groups by taking in protons from the solution into the  
33 domain of the polymer chains through hydrogen bond formation. Moreover,  
34 esterification reactions occur between the alcohol group at the end of PEO and citric  
35 acid. Although this process was not dominant here, it may affect on the established  
36 edge of the citrate-tungsten clusters.

37 Based on the results of dynamic light scattering experiments, the F127 micelle  
38 size were increased upon the addition of citric or sulfuric acid to average diameters of  
39 19 nm and 20 nm from their original size of 17 nm. Acid modification does not only  
40 affect the micelles' surface but appreciably changes the size of the micelles as well.  
41 PEO-polypropyl oxide has been shown to be very surface active in aqueous solution.  
42 When the pH of solution decreases significantly, the resultant flattened conformation  
43 of PEO in water results in a majority of the molecular segments being in contact with  
44 the water interface. In this conformation, the tungstate anion and a suitable acid may

1 occupy the folded portion of PEO. As indicated by the NMR results, the sulfate anion  
2 contributes more to the destabilization of PEO than citric acid does. Thus, the  
3 crown-ether-like structure of F127 in sulfuric acid is less asymmetrical than that in  
4 citric acid. Therefore, the relatively short, small tungstate anion can more easily  
5 occupy the defects.  
6

### 7 **3.2 Mechanism of formation of the hybrid film**

8  
9 The chemical structures of the hybrid films were verified by FTIR and Raman  
10 spectroscopies. These two analytical methods are powerful tools for characterizing the  
11 structural alteration that occurs upon the introduction of surfactant and acid. Figures  
12 2(A), 2(B) and 2(C) describe EPE, tungstate and acid molecule peaks shift due to  
13 hybridization in the FTIR and Raman spectra, respectively. Figure 2 (A) tungstate (a)  
14 (b) shows the FTIR spectra of different films composed of **F127STH**, **F127WC**, F127  
15 with each tungstate source and the tungstate alone. For instance, as shown in FTIR  
16 spectra of ST series one (a), the three characteristic peaks of tungstate were present in  
17 the sodium tungstate crystal but were lost upon the addition of sulfate. However,  
18 when measured in systems containing EPE, the intensities of these peaks all increase  
19 when acid was added ( $785\text{ cm}^{-1}$  ascribed to  $\text{O}_c\text{-W-O}_c$ ;  $837\text{ cm}^{-1}$  ascribed to  $\text{O}_b\text{-W-O}_b$ ;  
20 and  $962\text{ cm}^{-1}$  ascribed to  $\text{W=O}_t$ , where  $\text{O}_t$  is the terminal oxygen,  $\text{O}_b$  is the bridged  
21 oxygen of two octahedra sharing a corner, and  $\text{O}_c$  is the bridged oxygen sharing an  
22 edge).<sup>[29]</sup> It should be mentioned that the strong absorption band at  $837\text{ cm}^{-1}$  was  
23 related to shared bonding in the O-W corner, which was lost when the pH changes.  
24 Meanwhile, the peak at  $619\text{ cm}^{-1}$  for W-O increases significantly, which may be  
25 attributed to weak coordination arising from hydrogen bonding. These results also  
26 demonstrate that  $\text{-CH}_2\text{-O-}$  was incorporated into the nanocomposite PT cluster film.  
27 Similar FTIR results were obtained for the citrate and paratungstate systems.  
28 Henceforth, the discussion of Figure 2(A) tungstate (c) (d) will focus mainly on the  
29 sulfate and sodium tungstate system Based on the Raman spectra shown in Fig. 2 (A)  
30 tungstate (c), the Raman bands of aqueous tetrahedral  $\text{WO}_4^{2-}$  (ST) were assigned to a  
31 ( $\text{W=O}$ ) symmetric stretching at  $931\text{ cm}^{-1}$  and a broad ( $\text{W-O}$ ) asymmetric stretching at  
32  $834\text{ cm}^{-1}$ . When the pH value reached 2.3 after acidifying the solution, the band at  $931\text{ cm}^{-1}$   
33 disappeared, and a wide band at  $960\text{--}969\text{ cm}^{-1}$  appeared in the Raman spectra.  
34 These changes occurred because aqueous octahedral  $\text{W}_{12}\text{O}_{39}^{6-}$  and  $\text{W}_6\text{O}_{19}^{2-}$  polyanions  
35 displayed a ( $\text{W-O}$ ) stretching mode at  $970\text{--}960\text{ cm}^{-1}$ . An additional band at higher  
36 wavenumber ( $999\text{ cm}^{-1}$ ) appears after sulfate addition; this band was attributable to  
37 the stretching modes of polyanionic species  $\text{W}_{10}\text{O}_{32}^{4-}$  and  $\text{W}_6\text{O}_{19}^{2-}$ .<sup>[21]</sup> Upon the  
38 addition of EPE, the bands at  $999\text{ cm}^{-1}$  attributed to the symmetric and asymmetric  
39 stretching vibrations of the  $\text{W=O}$  terminal group decreased in intensity and shifted to  
40  $995$  and  $1005\text{ cm}^{-1}$ , respectively, due to changes in the charge of the tungstate cluster.  
41 The  $\text{W-O}$  corner-sharing band at  $960\text{--}970\text{ cm}^{-1}$  shifted to  $980\text{ cm}^{-1}$ , characteristic of  
42 tetrahedral  $\text{WO}_4$ . The  $983\text{ cm}^{-1}$  peak in Fig. 2(C) acid (b) was also consistent with  
43 formation of tetrahedral structure tungstate cluster. The  $\text{W-O-W}$  band also clearly  
44 shifted, as indicated by the dotted line in Fig. 2 (A) tungstate (a). These changes



1 indicate that the vibrations of tungstate were disturbed in the nanocomposite film due  
2 to hydrogen-bonding interactions between the tungstate molecule and EO groups and  
3 weak coordination between the tungstate and sulfate groups. The oxygen in EPE may  
4 also contribute to bonding in the  $\text{WO}_6$  and  $\text{WO}_4$  frameworks. The special bonding  
5 observed in this study resulted in an unbalanced electron state, which is likely to  
6 manifest as a bond for one tungstate unit. In this manner, the formation of the  
7 composite increases the number of defects in the tungstate and enhances its  
8 photochromic character. Similar phenomena were observed in the paratungstate and  
9 citrate systems shown in Fig. 2 (A) tungstate (b) (d).

10 For the EPE molecule, shown in Figure 2(B) EPE (a) (b), the strong characteristic  
11 FTIR bands for the  $-\text{CH}_2-\text{O}-$  group, which were typically found at  $2800-3000\text{ cm}^{-1}$ ,  
12 was initially larger than usual owing to the enhancement caused by resonance  
13 between the transition metal oxide (W-O) and EPE structure.<sup>[20]</sup> However, the bands  
14 were lost when the pH was reduced to 2 by the addition of sulfate or citrate, indicating  
15 that a portion of the groups in the EPE molecule were intercalated into the hybrid.  
16 From this point, the EO molecule is also affected by the tungstate anion and the acid,  
17 as shown in Fig. 2 (B). The characteristic peaks of the EO chain in both the FTIR and  
18 Raman spectra (Fig. 2(B) EPE (c) (d)) increased in intensity upon the addition of  
19 tungstate and decreased upon the addition of acid due to the domination of proton  
20 interactions over the weak coordination between  $-\text{CH}_2-\text{O}-$  and tungstate.

21 The same conclusion is reached based on the changes in the acid peak shown in  
22 Fig. 2 (C). The results shown in Figure 2(C) acid (b) indicate that when tungstate was  
23 added, the spectral peaks associated with  $-\text{COOH}$  ( $1730\text{ cm}^{-1}$  and  $\text{S-O}$  ( $565\text{ cm}^{-1}$ ))  
24 shown in Fig. 2C (a) weakened. Therefore, acid can be incorporated into the hybrid  
25 structure to form a correlated hybridization composite system comprising three parts  
26 (tungstate, acid, and  $-\text{CH}_2-\text{O}-$ ).

27  
28  
29  
30 Figure 2. FTIR and Raman spectra changes of A) tungstate, B) EPE, and C) the acid molecule  
31 as a function of wavenumber in the hybrid.

32  
33  
34  
35 Figure 3. XPS analysis of the hybridization of the thin films **F127STH** (a) (c) and **F127WC**  
36 (b) (d).

37  
38 XPS was used to further confirm the chemical structures of the hybrid films and  
39 probe interfacial interactions. Fig. 3b shows photoelectron spectra obtained during  
40 hybridization, which reveal variations in  $\text{W}_{4f}$  levels. The shape of the XPS curve  
41 before hybridization is convoluted into two doublets, of which the main doublet at  
42  $35.8\text{ eV}$  was due to the  $\text{W}_{4f_{7/2}}$  orbital and the other at  $37.8\text{ eV}$  was assigned to  $\text{W}_{4f_{5/2}}$ ,  
43 corresponding to the  $\text{W}^{6+}$  oxidation state. As shown in Fig. 3a, a second doublet  
44 appeared after hybridization at a lower binding level with peaks at  $34.7\text{ eV}$  and  $36.7$

1 eV, which were assigned to  $W4f_{7/2}$  and  $W4f_{5/2}$  of the  $W^{5+}$  oxidation state, respectively.  
2 The results indicated the formation of  $W^{5+}$  species after hybridization, in agreement  
3 with the electron state changes of tungstate. The percentage of  $W^{5+}$  after hybridization  
4 was 25.1%, corresponding to  $W^{5+}/(W^{5+}+W^{6+})=1/5$ . This result illustrates that the  
5 unbalanced oxygen number is induced by hybridization. The chemical formula of the  
6 hybrid gel component can be simplified to  $H_{0.2}WO_3$ .

7 When tungstate was deposited with the EPE layer, the EPE/ $WO_3$  hybrid thin film  
8 exhibited two main characteristic peaks at 38.1 eV ( $W4f_{5/2}$ ) and 36.1 eV ( $W4f_{7/2}$ );  
9 these peaks were 0.3 eV higher than those of the  $WO_3$  film.<sup>[22, 23]</sup> The increase in  
10 binding energy was likely due to the presence of a large amount of defects caused by  
11 the weak coordination.

12 C1s spectra of **F127STH** and **F127WC** were shown in Fig. 3c and d, respectively.  
13 The lower binding energy peak at 285.0 eV was assigned to carbon bonded to  
14 hydrogen. The appearance of a second peak at 286.5 eV was assigned to C-O bonding  
15 present in weak coordination. Compared with results reported in previous work<sup>[24]</sup>,  
16 -CH<sub>2</sub>-O- bonding was greatly enhanced after hybridization.  
17

### 18 **3.3 Structural verification**

19  
20  
21 Figure 4. a) X-ray diffraction pattern of  $WO_3$  prepared from sodium tungstate and ammonium  
22 paratungstate precursors; b) XRD patterns of samples from **1-6** and **a-g** after gel drying. c)  
23 SAXRD of different films;

24  
25 To investigate the structural changes that occur upon calcination of the films,  
26 **F127WC** was analyzed by XRD and TEM (Fig. 4). In this analysis, samples  
27 fabricated with ammonium paratungstate were compared to a control sample, labeled  
28 **0**. Although these two tungstate oxides exhibited similar XRD patterns post-sintering  
29 (Figure 4a, b), individual  $WO_4^{2-}$  and the tungstate clusters formed by ammonium  
30 paratungstate underwent a different process from aqueous solution. It can be deduced  
31 that aqueous polytungstate loses its photochromic effect when the oxygen defects  
32 vanish during the sintering. PT fills space in the PEO folding framework, with  
33 multi-dentate acids also contributing to the formation of ordered nanostructures.

34 In addition, as shown in Fig. 4c, the evolution of layered structures can be  
35 observed in the SAXRD patterns. The inorganic moieties of the tungstate clusters  
36 exhibited a layered structure with d-spacings of 9.2 nm and 9.3 nm. For **F127STH**,  
37 the observed increase in the d-spacing was due to the insertion of additional -CH<sub>2</sub>-O-  
38 molecules into its inorganic layers with the continuous addition of EPE, which did not  
39 in itself exist as part of the ordered structure. By contrast, the carboxyl groups were  
40 incorporated into the hybrid film by adsorbed onto an underlying organic layer to  
41 form an acid-anion complex.

42 After sintering the film at 500°C for 2 h,  $WO_3$  crystallized from the gel and  
43 aggregated as 5  $\mu\text{m}$  particles, as shown in the SEM images in Fig. 5. The uniform  
44 distribution of  $WO_3$  was the result of self-assembly of precursor clusters. In Fig. 5a, b,

1 ball-shaped and flower-like structures were apparent due to the spherical aggregation  
2 behavior of the clusters. In addition, the nanoscale  $\text{WO}_3$  appeared as small dots on the  
3 surface of larger microparticles in Fig. 5c. These nanoscale  $\text{WO}_3$  particles grew by the  
4 Ostwald ripening process and aggregate at the surface of the  $\text{WO}_3$  microparticles. By  
5 adopting this sol-gel self-assembly approach, a uniform nanocrystal dispersion that  
6 incorporates both ligand interactions and polytungstate crystal formation can be  
7 achieved.

8  
9  
10  
11 Figure 5. SEM (a, b, c) and TEM (d-l) micrographs showing the morphology of  $\text{WO}_3$  after  
12 annealing. Image c shows an amplified section of image b. Micrographs d, e, and f show  
13 samples prepared without the addition of surfactant or acid. The samples shown in g, h, and i  
14 were prepared in the presence of acid. Micrographs j, k, and l are images of samples  
15 containing both surfactant and acid.

16  
17 The addition of surfactant or acid enhanced the nanocrystallization process,  
18 leading to a significant increase in the number of nanoparticles formed. TEM  
19 micrographs acquired after the sintering process (d-l) demonstrate that the formation  
20 of small nanocrystals was induced by the addition of surfactant (g, h, i) and citrate (j,  
21 k, l). The dark-field diffraction analysis of tungsten trioxide (f, i, l) revealed that the  
22 microstructure of the particles was consistent with those produced using oxalic acid<sup>[1]</sup>.  
23 Moreover, the XRD pattern (Figure 4b) was also consistent with the observed  
24 tendency of amorphous particles to form upon the addition of surfactant and acid  
25 because the level spacing between adjacent electron states is inversely proportional to  
26 the nanoparticle diameter. A decrease in the size of the oxide particles resulted in the  
27 broadening of the band gap between the highest occupied and lowest unoccupied  
28 electron levels. Thus, more energy is needed to induce an electron transition during  
29 the photochromic process for smaller particles (i.e., quantum-size effects), leading to  
30 a blue shift in the optical absorption maximum. As the concentrations of surfactant  
31 and acid increased, the oxide becomes more amorphous, implying the presence of  
32 more defect sites that facilitate the charge-transfer process. The effects of both  
33 surfactant and acid in aqueous solution enhance the photoinduced effects due to  
34 particle size and boundary conditions. The gel can also be further condensed into a  
35 film or block without losing its enhanced photochromic properties.

### 36 37 **3.4 Photochromic behavior of the hybrid film**

38  
39  
40 Figure 6. UV-Vis spectra of polytungstate under acidic conditions with varying concentrations  
41 of (a) F127, (b) sodium tungstate, (c) paratungstate and (d) UV-Vis spectra of before and after  
42 fully reduced **F127WC** and **F127STH** samples with their photographs nearby.

43  
44 Photochromic enhancement was illustrated by UV-Vis spectra as shown in Fig. 6. It

1 indicates that as the concentration of F127 increases in Fig. 6a, the peak  
2 corresponding to tungstate absorbance tends to red shift as the crystalline structure  
3 became more regular. The ordered composition of the tungstate anion resulting from  
4 the combination of poly- or mono- tungstate with EO had a much broader absorbance  
5 bandwidth.

6 As shown in Fig. 6b and c, the addition of tungstate anion induced a tunable shift  
7 in the transmittance wavelength. The size of the tungstate compounds formed changed  
8 as the concentration of tungstate increased, which in turn altered the size of the  
9 clusters formed. In particular, the sodium tungstate clusters formed in the presence of  
10 sulfate could be tuned to absorb in the visible light region. The blue color of the  
11 composite was created by electron hopping between W-O bonds in the cluster through  
12 IVCT. Therefore, the combination of the clusters is vital for the electromagnetic wave  
13 resonance. This combined structure is influenced by the folding of the long EO chains  
14 and the acidic anion.

15 In the surfactant and tungstate system, according to the Hofmeister series, citrate  
16 and sulfate tend to salt biomacromolecules out of solutions and increase the stability  
17 of biomacromolecules toward their folded natural state. These two well-hydrated  
18 anions interact with the hydrophilic portion of the polymer by changing the entropy of  
19 hydration water around the F127 micelle.<sup>[25]</sup>

20 The hydrated anions increase stability according to the following order:

21 citrate<sup>3-</sup>>sulfate<sup>2-</sup>>phosphate<sup>2-</sup>>F<sup>-</sup>>Cl<sup>-</sup>.

22 For the same tungstate species, citrate buffer has the greatest stabilizing effect,  
23 particularly for the relatively large paratungstate. A transparent, stable solution is a  
24 necessary precursor for fabricating flexible composite films. In aqueous solution, only  
25 the citrate and sulfate systems can achieve this photochromic effect. The interaction of  
26 the acid anion with the EO chain is vital to this process, as is the important influence  
27 the anions exert in terms of controlling the tungstate anion.

28 In Fig. 6d, the samples' different transmittance spectra revealed that they were  
29 composed of different-sized tungstate anions and acid species, resulting in different  
30 depths of blue color in each sample. The -CH<sub>2</sub>-O- chain surfactant was used to form  
31 both samples, indicating that -CH<sub>2</sub>-O- hybridization with tungstate is ubiquitous under  
32 acidic conditions. As shown in Fig. 6d, the **F127STH** sample not only contains a  
33 higher concentration of tungstate owing to its saturation with sodium tungstate but  
34 also exhibits another type of self-assembly behavior. The black color of the **F127STH**  
35 sample resulted in an increase in its temperature upon IR irradiation. Besides that,  
36 sample **F127STH**'s absorption over the entire range of the solar spectrum (including  
37 the IR region) could induce the photo-thermal coloration<sup>[26]</sup>. In the film containing  
38 PT network, extraction of the hydrogen oxide caused the IR-photochromic behavior.  
39 Hybrid film was afforded the advantages of long durability of the colored state, which  
40 was maintained for a longer lifetime.

41 In contrast to the citrate-containing sample, in the sulfate-containing sample, the  
42 color change occurred at the air-water interface during both the coloration and  
43 bleaching processes. There are two major reasons for this phenomenon. First, the  
44 paratungstate cluster anion is partitioned more fully by citrate, which suspends the

1 particles more uniformly within the system. Second, as a type of surfactant, the EPE  
2 molecule tends to condense at the air-water interface. As surfactant condensation  
3 increases, the degree of folding of  $-\text{CH}_2-\text{O}-$  intensifies, resulting in coloration of the  
4 air-water interface. Tungstate anion clusters need more compact space to form the  
5 hybridization structure shown in Scheme 1. Therefore, the film exhibits an enhanced  
6 photochromic interface because the condensed gel developed by the dip-coating  
7 method advantageously improves the photochromic behavior by increasing the  
8 folding level.

9 In addition to EPE, formaldehyde was used for  $\text{W}_{5d}-\text{O}_{2p}$  hybridization.  
10 Formaldehyde-containing films exhibited photochromic enhancement similar to that  
11 observed for F127-containing films. However, after drying under ambient conditions  
12 for 1 day, the photochromism of the formaldehyde-based films was lost due to a lack  
13 of C-O hybridization once formaldehyde degassed from the material.

14  
15  
16  
17 Figure 7. Hybrid films a) before and b) after UV irradiation; c) Dynamic optical transmittance  
18 of the **F127STH** gel

19  
20 Images of highly transparent flexible thin films are shown in Fig. 7a and b. The  
21 as-prepared films had smooth surfaces and were highly transparent and scatter-free  
22 due to the good miscibility of the tungstate anion and F127, the homogeneous  
23 distribution of tungstate in the films, and the elimination of microbubbles. In addition,  
24 oxygen defects were preserved during the fabrication process, enhancing the  
25 photochromic character of the system. Moreover, the efficiency and density of the  
26 color change were both enhanced significantly by film formation due to further  
27 condensation of the EO chains in the gel.<sup>[27]</sup>

28 The appearance of dark-blue coloring occurred immediately upon irradiation of  
29 this sample under a  $50 \text{ mW/cm}^2$  UV lamp for 1 second, this efficiency was much  
30 higher than that of neat  $\text{WO}_3$  or other hybrid films. The dynamic contrast of the  
31 hybrid gel was approximately 80%. With respect to the switching kinetics, bleaching  
32 was observed exposed to a sufficient amount of  $\text{O}_2$  under ambient conditions<sup>[28]</sup>. The  
33 reversible changes process for large-scale thin film precursor in the transmittance of  
34 the gel exposed to a series of optical excitation cycles are shown in Figure 7c. There  
35 was no obvious difference in the maximum and minimum transmittances  
36 corresponding to the color-bleaching process (recorded at 600 nm and 640 nm,  
37 respectively) after many cycles. The hybrid gel exhibited remarkably improved  
38 stability, retaining 98% of its original contrast after 10 cycles. The enhanced stability  
39 of the hybrid can be attributed to the fact that defect sites caused by hybridization may  
40 not involve the oxidation or reduction of tungstate. During the coloration process, an  
41 electron-hole pair is generated in hybrid  $\text{H}_{0.2}\text{WO}_3$  by solar irradiation: the hole splits a  
42 water molecule, and the electron reduces  $\text{W}^{6+}$  to  $\text{W}^{5+}$ , and a proton is intercalated  
43 following the intercalation of the electron. During the bleaching process, the electron  
44 reacts with oxygen molecules to form reactive oxygen species. Thus, it is observed

1 that the EPE molecule is only involved in electron state displacement, not in any  
2 reduction or redox process. Therefore, the reversible and sustainable  
3 photo-electrochemical behaviors observed are fast, steady and reproducible.

#### 4 **4. Conclusions**

5 In summary, hybrid films consisting of EPE and tungstate were readily prepared using  
6 a one-pot self-assembly sol-gel method. The method provides a simple and efficient  
7 means of producing hybrid films over a large area. Moreover, the oxygen from EPE  
8 hybridizes the  $W_{5d}-O_{2p}$  electron state and generates defect sites in the W-O framework,  
9 enhancing the photochromic efficiency of the film. The fabrication and morphology  
10 of  $WO_3$  prepared from isopolytungstate also indicate that the formation of amorphous  
11 colloids is enhanced by the addition of acids and surfactants, which is beneficial for  
12 enhanced coloration. The characterization of films demonstrated that the observed  
13 photochromism is the result of electron transfer from  $W^{6+}$  to  $W^{5+}$  in the colloidal  
14  $H_{0.2}WO_3$  framework. Importantly, this method for enhancing the photochromic  
15 properties does not involve any redox or reduction processes that may destroy the  
16 polymer component of the films. Thus, the interaction between EPE and tungstate in  
17 the hybrid film not only enhances the photochromic response but also improves the  
18 durability of the hybrid film.

19

20

#### 21 REFERENCES

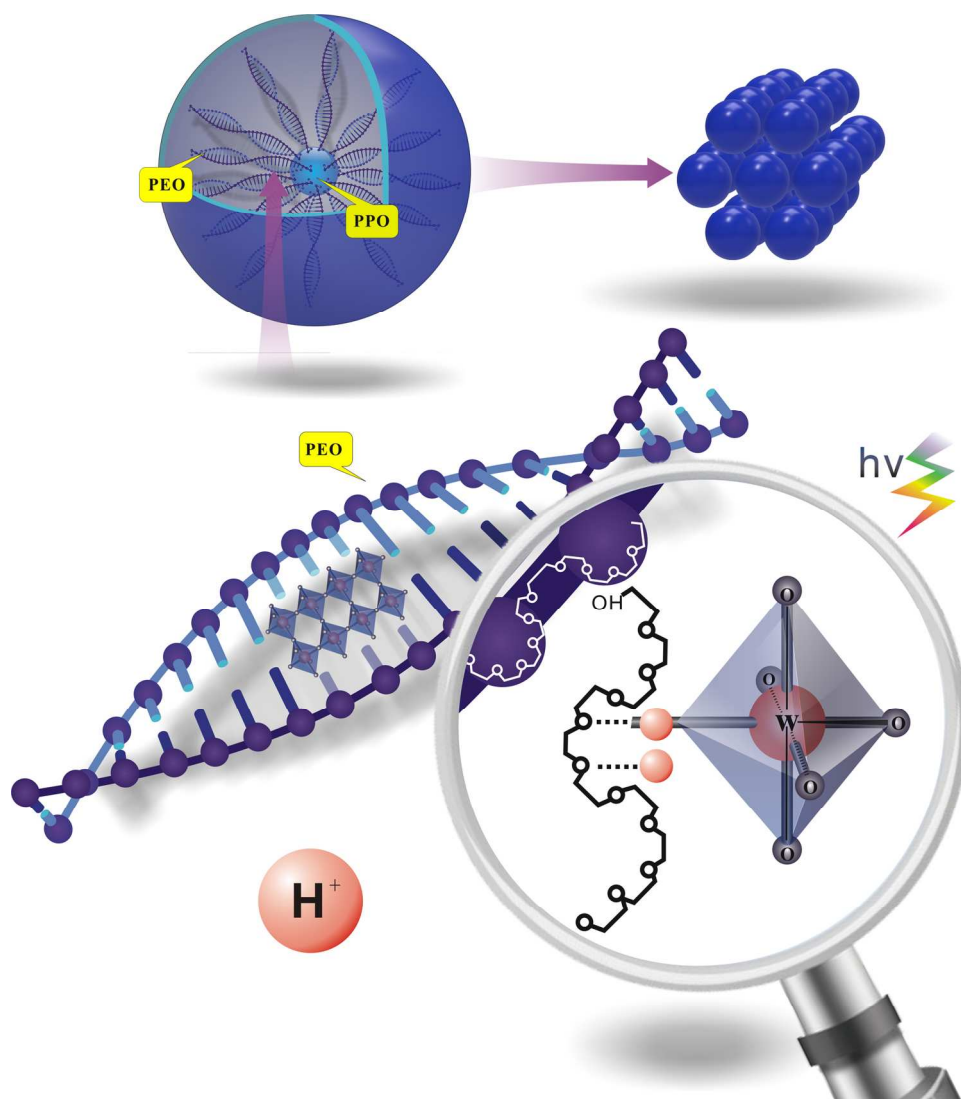
- 22 [1] (a) T. He and J. N. Yao, *Prog. Mater. Sci.*, 2006, **51**, 810; (b) *Photochromism:*  
23 *Molecules and Systems*, H. Durr and H. Bouas-Laurent, Elsevier, New York, 1990,  
24 Vol. 1. (c) G. M. Tsivgoulis and J. M. Lehn, *Angew. Chem., Int. Ed.*, 1995, **34**,  
25 1119-1122. (d) *Organic Photochromic and Thermochemical Compound*, J. C. Crano  
26 and R. J. Guglielmetti, Plenum, New York, 1999, Vol. 1. (e) M. Irie, *Chem. Rev.*,  
27 2000, **100**, 1685.
- 28 [2] a) M. T. Pope and A. Muller, *Angew Chem. Int. Ed.*, 1991, **30**, 34-36. b) A. Proust,  
29 B. Matt, R. Villanneau, G. Guillemot, P. Gouzerha and G. Izzet, *Chem. Soc. Rev.*,  
30 2012, **41**, 7605-7622. c) D. L. Long, R. Tsunashima and L. Cronin, *Angew. Chem. Int.*

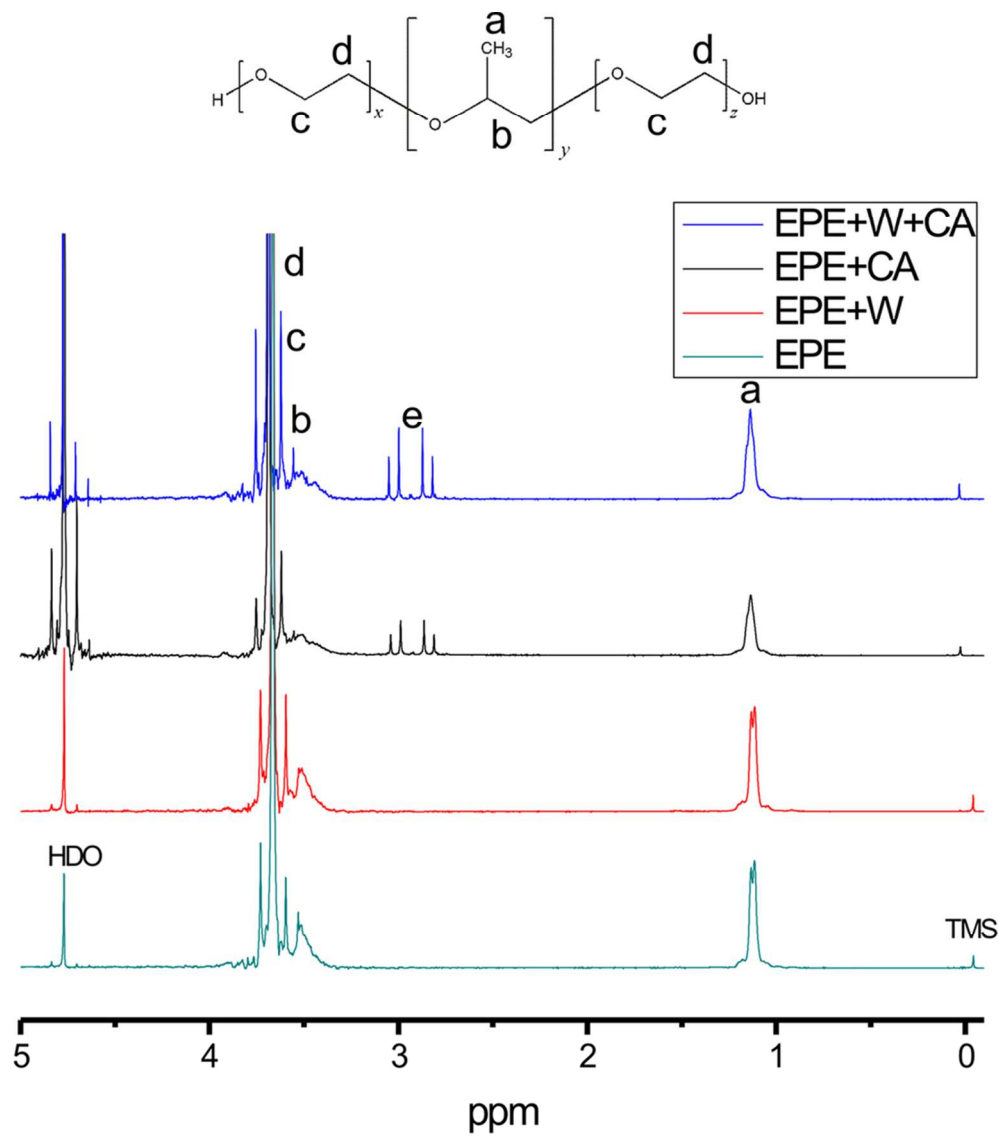


- 1 *Ed.*, 2010, **49**, 1736-1758. (d) *Polyoxometalate chemistry for nano-composite design*,
- 2 T. Yamase and M.T. Pope, Springer, New York, USA. 2002.
- 3 [3] (a) M. Gratzel, *Nature*, 2001, **414**, 338. (b) C. L. Lin, C. C. Lee and K. C. Ho, *J.*
- 4 *Electroanal. Chem.*, 2002, **524-525**, 81-89. (c) S. Vogel and R. Holze. *Electrochim.*
- 5 *Acta*, 2005, **50**, 1587-1596. (d) A. Hagfeldt and M. Gratzel, *Acc. Chem. Res.*, 2000,
- 6 **33**, 269-278. (e) P. M. Beaujuge and J. R. Reynolds, *Chem. Rev.*, 2010, **110**, 268-320.
- 7 (f) C. L. Hill, D. A. Bouchard, M. Kadkhodayan, M. M. Williamson, A. Schmidt and
- 8 E. F. Hilinski, *J. Am. Chem. Soc.*, 1988, **110**, 5471-5419.
- 9 [4] (a) A. Hiskia, A. Mylonas and E. Papaconstantinou, *Chem. Soc. Rev.*, 2001, **30**,
- 10 62-69. (b) M. Lu, B. H. Xie, J. Kang, F. C. Chen, Y. Yang and Z. H. Peng, *Chem.*
- 11 *Mater.*, 2005, **17**, 402-408. (c) Y. B. Yang, L. Xu, F. Li, X. G. Du and Z. X. Sun, *J.*
- 12 *Mater. Chem.*, 2010, **20**, 10835-10840. (d) J. M. R. Narayanam and C. R. J.
- 13 Stephenson, *Chem. Soc. Rev.*, 2011, **40**, 102-113.
- 14 [5] A. Kudo and Y. Miseki, *Chem. Soc. Rev.*, 2009, **38**, 253-278.
- 15 [6] (a) S. M. Wang, L. Liu, W. L. Chen, Z. M. Zhang, Z. M. Su and E. B. Wang, *J.*
- 16 *Mater. Chem. A*, 2013, **1**, 216-220. (b) X. Q. Gong, J. X. Li, S. Y. Chen and W. J.
- 17 Wen, *Appl. Phys. Lett.*, 2009, **95**, 251907-251911.
- 18 [7] Z. B. Han, E. B. Wang, G. Y. Luan, Y. G. Li, H. Zhang, Y. B. Duan, C. W. Hu
- 19 and N. H. Hu, *J. Mater. Chem.*, 2002, **12**, 1169-1173.
- 20 [8] X. Zhang, X. Dong and W. Yang, *Thin Solid Films*, 2006, **496**, 533-538.
- 21 [9] X. J. Luo and C. Yang, *Phys. Chem. Chem. Phys.*, 2011, **13**, 7892-7902

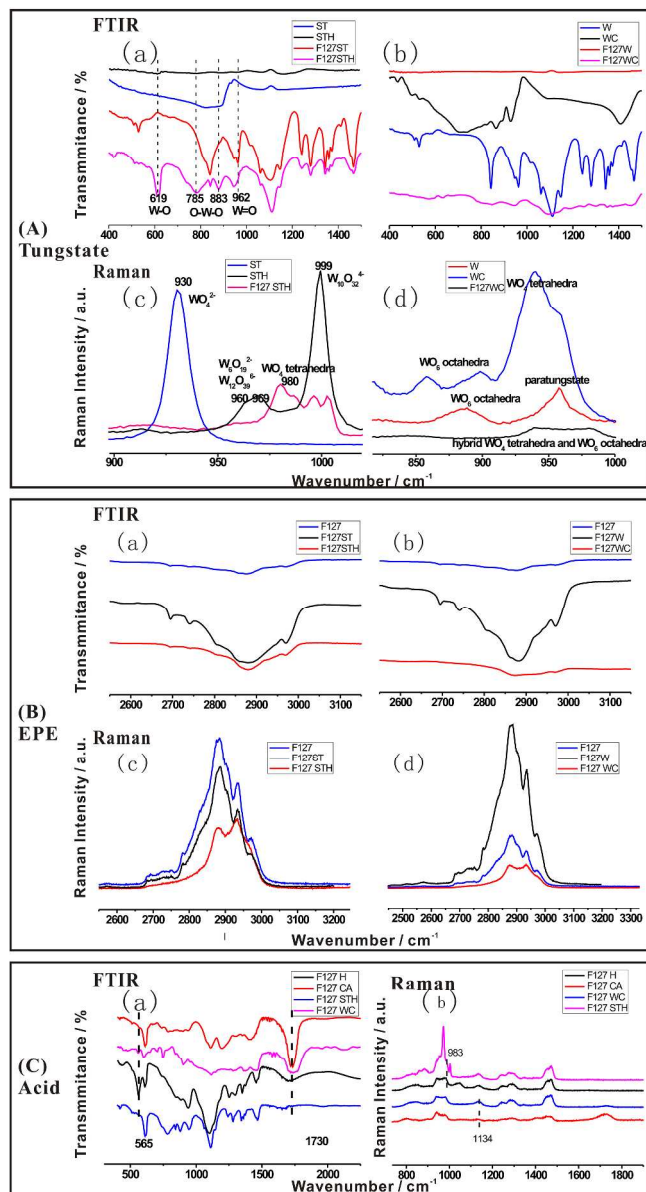
- 1 [10] Z. L. Wang, R. L. Zhang, Y. Ma, L. Zheng, A. D. Peng, H. B. Fu and J. N. Yao,  
2 *J. Mater. Chem.*, 2010, **20**, 1107–1111.
- 3 [11] G. Zhang, W. Yang and J. N. Yao, *Adv. Funct. Mater.*, 2005, **15**, 1255-1259.
- 4 [12] S. Y. Park, J. M. Lee, C. Noh and S. U. Son, *J. Mater. Chem.*, 2009, **19**, 7959–  
5 7964.
- 6 [13] D. T. Gillaspie, R. C. Tenent and A. C. Dillon, *J. Mater. Chem.*, 2010, **20**, 9585–  
7 9592.
- 8 [14] L. Pérez-Lombarda, J. Ortizb and C. Poutb, *Energy and Buildings*. 2008, **40**,  
9 394–398.
- 10 [15] A. Azens and C. G. Granqvist, *J. solid state electrochem.*, 2003, **7**, 64-68.
- 11 [16] C. G. Granqvist, *Solar Energy Materials and Solar Cells*, 2007, **91**, 1529-1598.
- 12 [17] a) T. Ikawa, K. Abe, K. Honda and E. Tsuchida, *Jour. Poly. Sci.: Poly. Chem. Ed.*  
13 1975, **13**, 7, 1505–1514. b) P. Alexandridis and T. A. Hatton, *Colloids and Surfaces A:*  
14 *Physicochemical and Engineering Aspects*, 1995, **96**, 1-46.
- 15 [18] S. S. Kanu and R. Binions, *Proc. Royal Society A*, 2010, **466**, 19-44.
- 16 [19] J. M. Parker, P. V. Wright and C. C. Lee, *Polymer*, 1981, **22**, 1305-1307.
- 17 [20] Z. G. Zhao and M. Miyauchi, *Chem. Commun.*, 2009, 2204-2206
- 18 [21] a) L. Gao, E. B. Wang, Z. H. Kang, Y. L. Song, B. D. Mao and L. Xu, *J. Phys.*  
19 *Chem. B*, 2005, **109**, 16587–16592. b) M. Picquart, S. Castro-Garcia, J. Livage, C.  
20 Julien and E. Haro-Poniatowski, *J. Sol-Gel Sci. Tech.*, 2000, **18**, 199-206. c) C.  
21 Rocchiccioli-Deltcheff, R. Thouvenot and M. Dabbabi, *Spectrochim. Acta Part A:*  
22 *Molecular Spec.* 1977, **33**, 143-153.

- 1 [22] (a) R. Gehlig and E. Salje, *J. Solid State Chem.*, 1983, **49**, 318–324.  
2 (b) B. A. D. Angelis and M. Schiavello, *J. Solid State Chem.*, 1977, **21**, 67-72.
- 3 [23] (a) M. Miyauchi, A. Kondo, D. Atarashi and E. Sakai, *J. Mater. Chem. C*, 2014,  
4 **2**, 3732-3737. (b) S. Bai, K. Zhang, L. Wang, J. Sun, R. Luo, D. Li and A. F. Chen, *J.*  
5 *Mater. Chem. A*, 2014, **2**, 7927-7934. (c) H. Ling, J. L. Lu, S. Phua, H. Liu, L. Liu, Y.  
6 Z. Huang, D. Mandler, P. S. Lee and X. H. Lu, *J. Mater. Chem. A*, 2014, **2**,  
7 2708-2717.
- 8 [24] (a) A. Sanyal, T. Bala, S. Ahmed, A. Singh, V. Piterina, T. M. McGloughlin, F.  
9 R. Laffir and K. M. Ryan, *J. Mater. Chem.*, 2009, **19**, 8974-8981 (b) T. Wang, J.  
10 Tang, X. Fan, J. Zhou, H. Xue, H. Guo and J. P. He, *Nanoscale*, 2014, **6**, 5359-5371.
- 11 [25] B. A. Deyerle and Y. Zhang, *Langmuir*, 2011, **27**, 9203–9210.
- 12 [26] Y. P. He and Y. P. Zhao, *J. Phys. Chem. C*, 2008, **112**, 61-68.
- 13 [27] W. J. Kunz, Henle and B. W. Ninham, *Curr. Opin. Colloid Interface Sci.* 2004, **9**,  
14 19–37.
- 15 [28] (a) C. Tanielian, R. Seghrouchni and C. Schweitzer, *J. Phys. Chem. A*, 2003, **107**,  
16 1102–1111. (b) C. Tanielian, *Coordi. Chem. Rev.* 1998, **178-180**, 1165-1181.
- 17



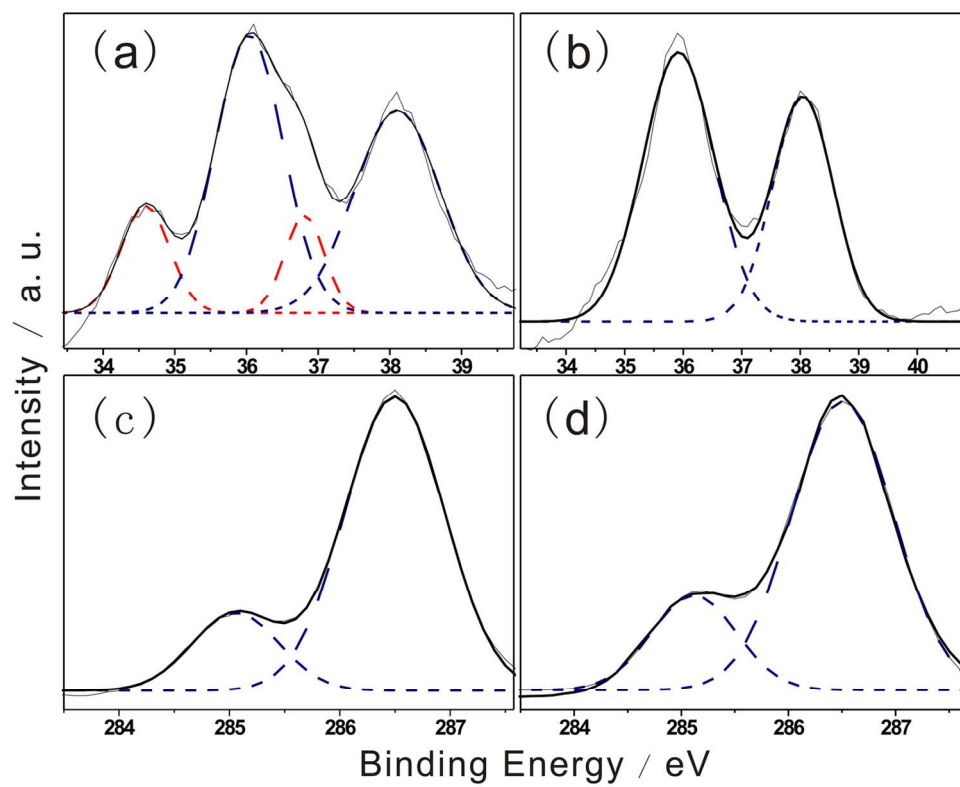


93x104mm (300 x 300 DPI)

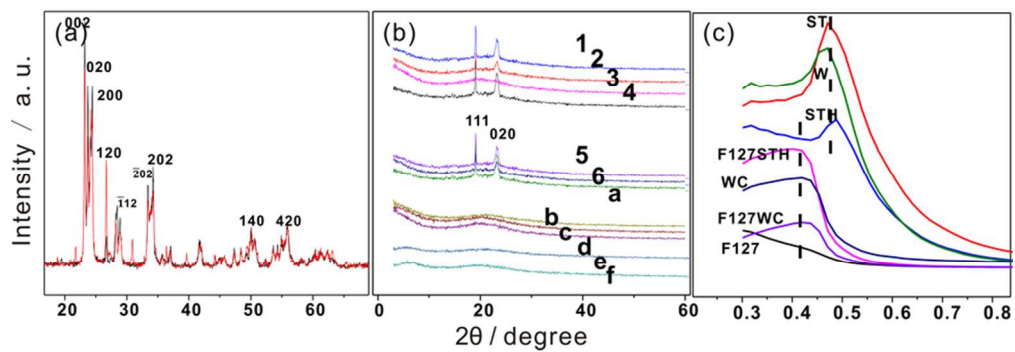


299x524mm (300 x 300 DPI)

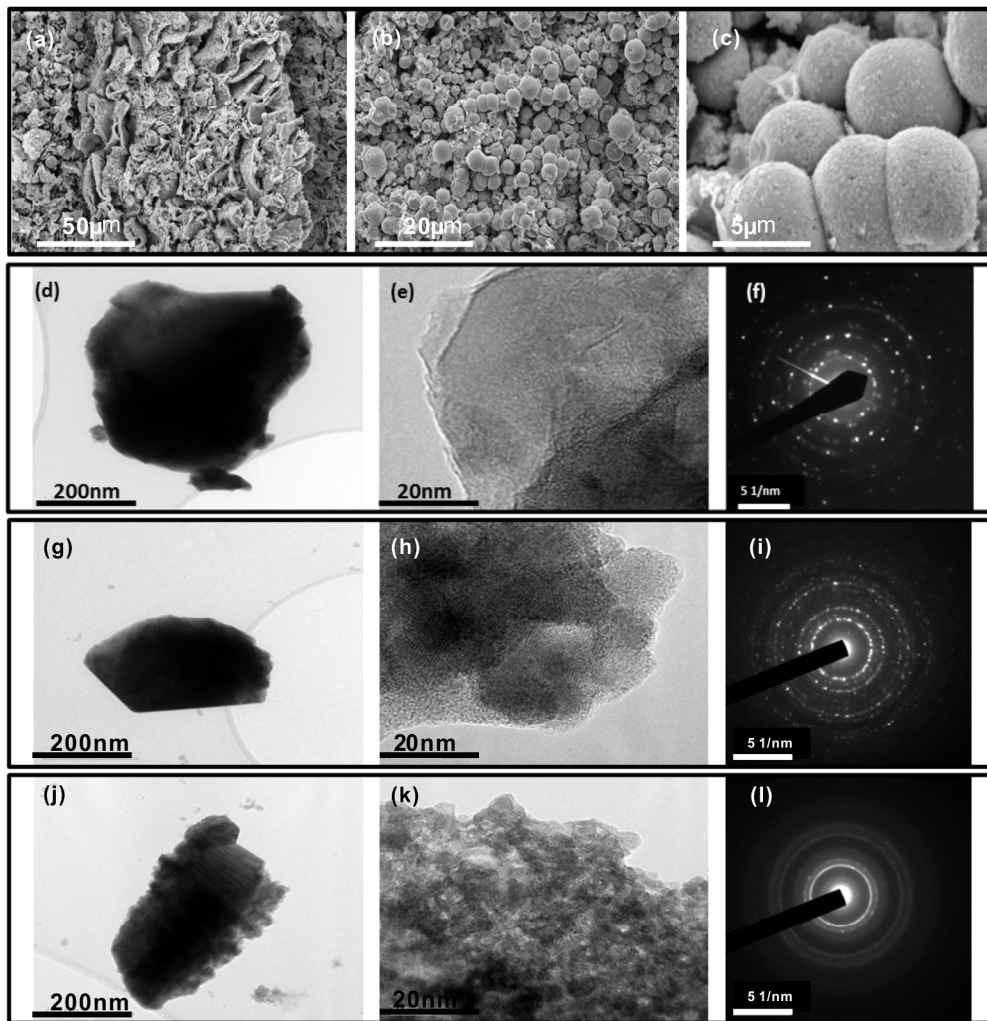




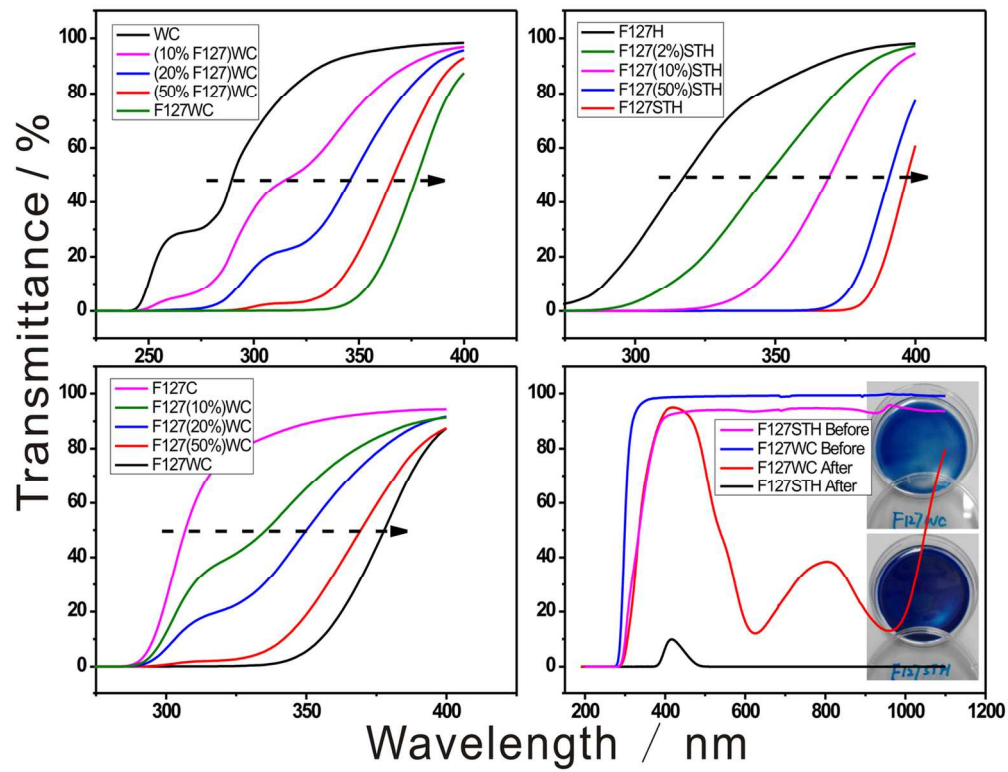
133x104mm (300 x 300 DPI)



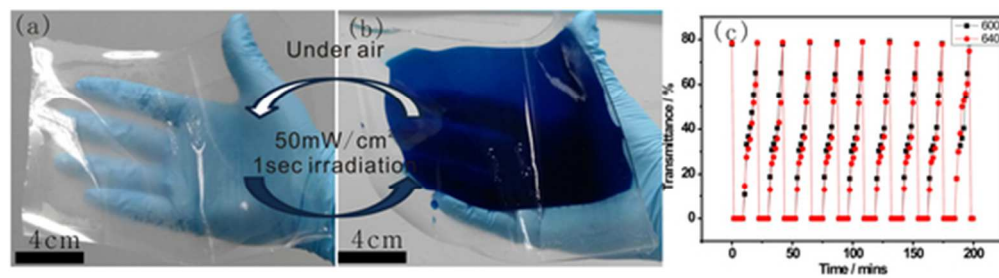
82x29mm (300 x 300 DPI)



237x243mm (300 x 300 DPI)



131x101mm (300 x 300 DPI)



46x12mm (300 x 300 DPI)

Enhanced photochromism of hybrid film synthesized by simple one-pot self-assembly method.

