# ChemComm

# Accepted Manuscript



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. We will replace this Accepted Manuscript with the edited and formatted Advance Article as soon as it is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/chemcomm

## ChemComm



## COMMUNICATION

# Spontaneous growth of ultra-thin titanium oxides shell on Ag nanowires: an electron energy loss spectroscope observation

Received 00th January 20xx, Accepted 00th January 20xx

Chi-Hang Tsai,<sup>a</sup> Shih-Yun Chen<sup>\*b</sup>, Jenn-Ming Song<sup>\*c</sup> and Alexandre Gloter<sup>d</sup>

DOI: 10.1039/x0xx00000x

www.rsc.org/

Ag nanowires with a spontaneous ultra-thin  $TiO_2$  shell (~0.5 nm) can be grown on  $TiO_2$  substrate. STEM/EELS results demonstrate that this oxygen-deficient  $TiO_2$  layer is formed through the oxidation of Ti which is released from the substrate and segregated to the nanowire surface simultaneously with crystal growth of the nanowires.

The performance of nanowires is often limited. One of the important solving methods is to synthesize composite nanowires such as core-shell nanowires to enhance functionality and usability. Recently, many studies on core-shell nanowires have been reported <sup>1-4</sup>. The purpose is to combine the materials of the core and shell to produce complementary effects or enhance their properties. With respect to the Ag-TiO<sub>2</sub> core-shell nanowires, the silver nanowires have unique properties of high surface plasmon resonance as well as good electrical conductivity, while TiO<sub>2</sub> is an ideal material as photocatalysts. As tabulated in Table 1, Cheng et  $al.^{1}$  have demonstrated the fabrication of Ag@TiO<sub>2</sub> nanowires through vapor-thermal method using poly(N-vinylpyrrolidone, PVP) as the surfactant. The diameter of the silver core and the thickness of TiO<sub>2</sub> shell are about 50 and 30 nm, respectively. Some studies have suggested to synthesize the Ag nanowires coated with amorphous  ${\rm TiO}_2$  through sol-gel or chemical reductions<sup>2,3</sup>. However, this method suffers from the sensitivity of synthesizing temperature. When the temperature is not properly controlled, the Ag@TiO2 nanowires with bristled surfaces would be produced. Moreover, Liu et al.4 used template method combined with the electrodeposition routes to synthesize Pt, Au and Ni nanowires. By manipulating the PH value of solution, the metal-TiO<sub>2</sub> core-shell nanorods could be obtained. They also indicated that low PH value benefits TiO<sub>2</sub> deposition. A small amount of Pt inoculants are needed to produce the metal-TiO<sub>2</sub> core-shell nanorods.

This study aims to develop a universal method for synthesizing ultra-long metal core-oxide shell composite nanowires. In the absence of oxide precursors, templates, inoculants and surfactants, vertically grown Ag/TiO<sub>2</sub> core-shell nanowires were synthesized on TiO<sub>2</sub> substrates using a one step process. The core-shell feature of the nanowires and the TiO<sub>2</sub> valence were observed by scanning transmission electron microscope (STEM) equipped with an electron energy loss spectroscope (EELS). The growth mechanism will thus be proposed.

Table 1 A summary of previous works on the metal-TiO $_{\rm 2}$  core- shell nanowires.

Metal@oxide	Method	Template (T) Surfactant (S)	Diameter/ Thickness/ Length	Phase	Precursor
Ag@TiO <sub>2</sub> nanowires <sup>1</sup>	vapor- thermal	PVP (S)	50 nm/ 30 nm/ 		твот
Ag@TiO <sub>2</sub> nanowiress <sup>2</sup>	sol-gel	PVP (S)	70 nm/ 10 nm/ 10-50 mm	amorphous	TTIP
Ag@TiO₂ nanowiress <sup>3</sup>	chemical reduction		40 nm/ 35 nm/ 	amorphous	твот
Pt,Au,Ni@TiO <sub>2</sub> nanowiress <sup>4</sup>	electro- deposition	AAO (T)			TiCl₃

TBOT( tetrabutyl titanate), TTIP( titanium tetraisopropoxide)

TiO<sub>2</sub> thin films were prepared via sol gel method. Si wafer substrates were dipped into gels and spun at 1000 rpm for 30 s. The solution used was prepared from isopropylalcohol (IPA), titanium isopropoxide (TTIP), and hydrogen chloride (HCl) with a volume ratio of 170:12:0.4 and then stirred for 10 min before aging at room temperature (20 °C) for 2 days. The as-synthesized TiO<sub>2</sub> films were annealed at 500 °C for 8 h in an oxygen atmosphere to obtain well-crystallized anatase TiO<sub>2</sub>. Silver nitrate (AgNO<sub>3</sub>) was selected as the precursor to grow Ag nanowires. 15  $\mu$ l of 0.05 M aqueous salt solution was dropped onto the TiO<sub>2</sub> coated substrate. Afterwards, the samples were isothermally

<sup>&</sup>lt;sup>a.</sup> Graduate Institute of Applied Science and Technology, National Taiwan University of Science and Technology, Taipei 106, Taiwan.

<sup>&</sup>lt;sup>b.</sup> Department of Materials Science and Engineering, National Taiwan University of Science and Technology, Taipei 106, Taiwan. E-mail: sychen@mail.ntust.edu.tw <sup>c</sup> Department of Materials Science and Engineering, National Chung Hsing

University, Taichung 402, Taiwan. E-mail: samsong@nchu.edu.tw

<sup>&</sup>lt;sup>d</sup> Laboratoire de Physique des Solids, Université Paris Sud 11, CNRS UMR 8502, F-91405 Orsay, France

#### COMMUNICATION

heated at 300 and 600 °C respectively for a certain period of time in air using an infrared furnace, and cooled down to ambient temperature. The structure and phase of the nanowires were characterized using a transmission electron microscope (TEM) equipped with electron energy loss spectroscope (EELS, USTEM Nion) and operated at 100 keV. The core loss was done using 0.6 eV per channel spectra dispersion and energy domain ranging from 450 to 480 eV in order to collect the Ti-L edge. In addition, inductively coupled plasma-mass spectrometry (ICP-MS) was used to determine the solubility of Ti in molten AgNO<sub>3</sub> salt.

Fig. 1(a) shows the as-synthesized Ag nanowires subjected to heating at 300 °C for 3 h. It reveals that the Ag nanowires have an average diameter and length of about 65±8 nm and 8.3  $\mu\text{m},$ respectively. The selected-area electron diffraction pattern (SADP) inserted in Fig. 1(a) confirms that the Ag nanowires are single-crystalline with [110] growth direction, and the EDS spectrum verified the purity of Ag nanowires without any detectable impurities. The EELS mapping was done to have a better understanding on the structure of the nanowires. Fig. 1(b) shows the presence of ultra-thin spontaneously-grown oxide shell enriched with Ti (0.5±0.05 nm) on the surface of the Ag nanowire. This shell layer is continuous, compact and firmly attached to the nanowire surface. The chemical composition and elemental distribution of Ag nanowires analyzed by EELS are shown in Fig. 1(c). The elemental line-scanning reveals that the Ti concentrates at the edge of the Ag signal, indicating a core-shell structure. The oxygen signal is overlapped with Ti but narrower. It locates at the outer surface region of Ti peak, which proves that the oxide layer is formed through inward diffusion of oxygen and thus oxidation of Ti.

The XRD patterns of the Ag nanowires heated at different time and temperature have been shown in Fig. 2(a). Only Ag is identified. The amount of Ti oxide layer is too small to be detected. Figs. 2 (b)(c)(d) illustrate high resolution TEM images of the corresponding Ag nanowires. The interval of lattice fringes, 0.23nm, exactly matches the interplanar distance of Ag(111). Those data all demonstrate that the cores of the nanowires are pure Ag crystals instead of silver oxides, Ag-Ti alloys or solid solutions.

Fig. 3 shows the experimentally obtained Ti L2.3-edge EELS spectra of the oxide layer in different heating conditions (300 °C-3 h, 300  $^{\circ}$ C-9 h and 600  $^{\circ}$ C-3 h) and the standard anatase TiO<sub>2</sub> as a reference. The Ti L-edge possesses two groups of peaks of Ti 2p core level consisting of  $2p_{1/2}$  (L<sub>2</sub> -edge) and  $2p_{3/2}$  levels (L<sub>3</sub> -edge). For standard TiO<sub>2</sub>, the Ti 3d band can be split into four sublevels including  $2p_{3/2}t_{2g}$  (A1),  $2p_{3/2}e_g$  (A2),  $2p_{3/2}t_{2g}$  (B1) and  $2p_{3/2}e_g$  (B2) symmetry <sup>5-8.</sup> For TiO<sub>2</sub> systems, Calvert *et al.*<sup>8</sup> reported that when the valence of Ti ions is reduced from  $Ti^{4+}$  (TiO<sub>2</sub>) to  $Ti^{3+}$  (Ti<sub>2</sub>O<sub>3</sub>), Ti Ledge peak positions shift to lower energy and the intensity of A1 peak for Ti L<sub>3</sub>-edge becomes weaker while the A<sub>2</sub> peak becomes stronger. As shown in Fig. 3, it is remarkable that no A<sub>1</sub> signal is visible in the EELS spectra of the oxide shell, but a distinct A<sub>1</sub> signal can be seen from the TiO<sub>2</sub> reference. Furthermore, the positions of A<sub>2</sub> and B<sub>2</sub> signals from all the samples are identical with the signals from  $TiO_2$  reference. These differences in  $Ti L_{2.3}$ -edge EELS

Journal Name

spectra leads us to believe that Ti ions are predominantly in between Ti<sup>3+</sup> and Ti<sup>4+</sup> oxidation states. Strictly speaking, the oxide shell should be denoted as TiO<sub>2-x</sub>.



Fig. 1 (a) TEM image and EDS spectrum of Ag nanowire synthesized on the  $TiO_2$  substrate after heating at 300 °C for 3 h (Insets: SEM image of vertically-grown Ag nanowires and SADP)(the Cu signal in EDS spectrum comes from the Cu grids), (b) EELS mapping and (c) elemental line-scans.



Fig. 2 XRD patterns (a) and high resolution TEM images of the Ag nanowire samples heated at different time and temperature: (b) 300 °C-3 h, (c) 300 °C-9 h and (d) 600 °C-3 h

Journal Name



Fig. 3 STEM-EELS of oxides layer on Ag nanowire samples heated at different time and temperature.

Fig. 4 depicts the quantitative data of the Ag core diameter and the thickness of the  $\text{TiO}_{2\text{-}x}$  shell with different heating time and temperature. As illustrated, after a prolonged heating at 300 °C from 3 h to 9 h, the thickness of TiO<sub>2-x</sub> shell remains almost same (0.5±0.05 nm for 3h and 0.52±0.06 nm for 9h). However, when the sample is heated at 600  $^{\circ}C$  for 3 h, the thickness of  $TiO_{2\text{-}x}$  shell significantly increases to 1.5±0.1 nm. Similarly, the Ag wire diameter increases from 65 nm (300  $^{\circ}$ C) to 110 nm (600  $^{\circ}$ C). Ti segregates at the nanowire surface probably through two routes, one is Ti diffusion from the  $TiO_2$  substrate through the Ag wire surface, and the other is the repelling of Ti in molten salt to the nanowire surface during the single crystal growth. In the second route, it can be speculated that Ti from TiO<sub>2</sub> substrate dissolves into molten salt (melting point of AgNO<sub>3</sub>: 212  $^{\circ}$ C) in the very early stage of heating prior to nanowire growth. The fact that thin oxide shell covers the whole nanowires as well as the pronounced effect of heating temperature compared with holding time suggests that Ti is repelled during nanowire crystallization instead of long-distance diffusion from the substrate. The solubility of Ti in molten  $\mathsf{AgNO}_3$  salt at different heating time and temperature is estimated from the ratio of TiO<sub>2-x</sub> shell thickness/Ag wire radius, and experimentally obtained by measuring the Ag and Ti concentrations in nitric acid aqueous solution using ICP. Table 2 shows the ICP results and estimated values of Ti/Ag (wt %), which corresponds with each other to some extent. As can be seen, when the sample is heated at 600  $^{\circ}$ C, the Ti solubility is two times higher than that at 300 °C. Holding time has no significant effect since the saturation has been reached.

Fig. 5 presents schematic illustration of the formation of Ag/TiO<sub>2-x</sub> core-shell nanowires. As illustrated, Ti is from the dissolution of TiO<sub>2</sub> substrate into molten silver nitrate salt. The dissolved Ti, which can be regarded as impurities due to its limited solubility in solid Ag at the heating temperature range [9], migrates to the crystal surface during the single-crystal growth of the Ag nanowires. The subsequent oxidation turns Ti into oxygendeficient TiO<sub>2-x</sub>.



Fig. 4 The thickness of spontaneous  $TiO_{2 \cdot x}$  in the samples heated at different time and temperature.

Table 2 Solubility of Ti in  $AgNO_3$  salt (given as Ti/Ag) at different time and temperature

Samples		ICP Ti /Ag (wt. %)	Estimated values Ti /Ag (wt. %)	
	300 °C-3 h	1.6	1.3	
	300 °C-9 h	1.6	1.3	
	600 °C-3 h	3.3	2.3	



Fig. 5 Schematic illustration of Ag/TiO<sub>2-x</sub> core-shell nanowires.

In summary, a one-step synthesis of Ag/TiO<sub>2</sub> core-shell nanowires was proposed in this study. Spontaneously-grown TiO<sub>2</sub> shell with a thickness of about 0.5 nm was found to cover the whole surface of Ag nanowires. Experimental results confirm that the TiO<sub>2</sub> shell is formed from the oxidation of Ti which is repelled from single-crystalline Ag during the crystallization of Ag nanowires. It is also verified that Ti is from dissolution of TiO<sub>2</sub> substrate in molten salt. The higher the heating temperature, the thicker the TiO<sub>2</sub> shell. The EELS spectra suggest that the valence of Ti ions in this spontaneous oxide layer is between Ti<sup>3+</sup> and Ti<sup>4+</sup>.

## Acknowledgements

This work was supported by Ministry of Science and Technology of R.O.C. (Contracts: NSC 102-2221-E-005-031 and MOST 103-2221-E-005-018), for which the authors are grateful.

## COMMUNICATION

## Notes and references

- 1 B. Cheng, Y. Le and J. Yu, J. Hazard. Mater., 2010, **177**, 971.
- 2 S. F. Chin, S. C. Pang and F. E. I. Dom, Mater. Lett., 2011, 65, 2673.
- 3 J. Du, J. Zhang, Z. Liu, B. Han, T. Jiang, and Y. Huang, Langmuir, 2006, **22**, 1307.
- 4 R. Liu and A. Sen, J. Am. Chem. Soc., 2012, **134**, 17505.
- 5 B. Cao, K. Suenaga, T. Okazaki and H. Shinohara , J. Phys. Chem. B, 2002, **106**, 9295.
- 6 C. N. Huang, J. S. Bow, Y. Zheng, S. Y. Chen, N. J. Ho and P. Shen, Nanoscale Res. Lett., 2010, **5**, 972.
- P. Guttmann, C. Bittencourt, S. Rehbein, P. Umek, X. Ke, G.
  V. Tendeloo, C. P. Ewels and G. Schneider, Nat. Photonics, 2012, 6, 25.
- 8 C. C. Calvert, W. M. Rainforth, D. C. Sinclair, and A. R. West, Micron, 2006, **37**, 412.
- R. H. Davies, A. T. Dinsdale, J. A. Gisby, J. A. J. Robinson, S. M. Martin, CALPHAD, 2002, 26, 229.

Journal Name

Page 4 of 4