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# Molybdenum Disulfide as a Highly Efficient Absorbent for Non-polar Gases

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Molybdenum disulfide, a kind of graphene-like two-dimensional material, has attracted great interests due to its unique properties with potential applications in electronics and sensors. Here, we perform first-principles calculations and grand canonical Monte Carlo (GCMC) simulations and show that MoS<sub>2</sub> layer is efficient to absorb non-polar gases. Compared with the popular gas sorbents (MOFs and carbon-based materials), MoS<sub>2</sub> exhibits additional advantages, including large surface to volume ratio and tunable properties. We systematically study the non-polar gases (CO<sub>2</sub> and CH<sub>4</sub>) adsorption on MoS<sub>2</sub> layer with and without vacancies. The perfect MoS<sub>2</sub> shows little or no adsorption for CO<sub>2</sub> and CH<sub>4</sub> molecules, but the MoS<sub>2</sub> with single S vacancy and double S vacancies exhibit excellent adsorption ability for CO<sub>2</sub> and CH<sub>4</sub> gases. The absorption energies are 65 kJ/mol for CO<sub>2</sub> and 47 kJ/mol for CH<sub>4</sub> (vdW-D2), respectively. An orbital coupling between the *p* orbital of CO<sub>2</sub> (or CH<sub>4</sub>) molecule and the *d* orbital of Mo atom is observed. GCMC simulation results show that MoS<sub>2</sub> with single S vacancy could absorb 42.1wt% CO<sub>2</sub> and 37.6wt% CH<sub>4</sub> under the pressure of 80 Bar at the room temperature. Our results indicate that monolayer MoS<sub>2</sub> with defects is a highly efficient absorbent for non-polar gases.

## 1. INTRODUCTION

Release of industrial and vehicle exhaust gases into atmosphere is one of the major sources of the current environmental problems. Among the exhaust gases, carbon dioxide (CO<sub>2</sub>) and methane (CH<sub>4</sub>) are the main gases that cause the warming effect. So, finding ideal sorbents for effective gas capture is a crucial issue in the energy and environmental science. Due to the inert property of non-polar CO<sub>2</sub> and

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CH<sub>4</sub> gases, most of the materials show a weak adsorption and are not sensitive to capture them. Up to now, different sorbents have been tried for CO<sub>2</sub> and CH<sub>4</sub> capture, such as MOF<sup>1-3</sup>, COF<sup>4,5</sup>, zeolites<sup>6,7</sup> and carbon-based nanomaterial<sup>8-11</sup>. For example, MOF-5, MOF-177 and zeolite 5A have been used as the sorbents for CO<sub>2</sub> and CH<sub>4</sub> adsorption, and the adsorption capacity for CH<sub>4</sub> and CO<sub>2</sub> can be as high as 22 wt% on MOF-177 at 298K and 100 Bar and 47.98 wt% on MOF-5 at 298K and 14 bar, respectively.<sup>3</sup>

As the rise of graphene, graphene-like two-dimensional (2D) materials have attracted great interests recently. Monolayer molybdenum disulfide (MoS<sub>2</sub>), a kind of transition metal dichalcogenide (TMD) material, is a well-established semiconductor with a direct band gap of 1.9 eV<sup>12</sup>. Unlike graphene, monolayer MoS<sub>2</sub> is constructed by triple atomic layers, in which Mo atoms are sandwiched between two S atoms. Due to its unique electronic, optical, and chemical properties, MoS<sub>2</sub> has been extensively investigated both experimentally<sup>13-21</sup> and theoretically.<sup>22,23</sup> It is recognized as a promising candidate for novel electronics and photonics devices. For example, monolayer MoS<sub>2</sub> has been successfully synthesized to be used as the channel material in a field-effect transistor with a high channel mobility of  $\sim 200 \text{ cm}^2\text{V}^{-1}\text{s}^{-1}$  and current on/off ratio of  $1 \times 10^8$  at the room temperature<sup>24</sup>. After that, a phototransistor based on the monolayer MoS<sub>2</sub> is fabricated by Zhang's group, and photocurrent generation and annihilation can be switched within ca. 50 ms, which exhibits a better photoresponsivity compared to the graphene-based device<sup>25</sup>.

Since possessing a high surface-to-volume ratio and flexible properties compared to graphene-based materials, MoS<sub>2</sub> layer is expected to be efficient gas sorbents or sensors. The field-effect transistors (FET) based on single- and multi-layer MoS<sub>2</sub> films have been successfully synthesized recently, which can be used as gas sensors to detect nitrous oxide (NO) with a detection of 0.8 ppm<sup>26</sup>. In addition, the

superior sensitivity for nitrous dioxide ( $\text{NO}_2$ ) has been observed in a flexible FET sensor array based on a  $\text{MoS}_2$  channel and reduced graphene oxide electrodes<sup>27</sup>. Though some experimental results indicated that  $\text{MoS}_2$  holds great promise for gas sensors, the underlying adsorption mechanism for different gases is not very clear and the theoretical studies is limited<sup>28</sup>.

Here we explore the response of monolayer  $\text{MoS}_2$  with and without vacancies to non-polar gas molecules ( $\text{CO}_2$  and  $\text{CH}_4$ ) using first-principles calculations combined with grand canonical Monte Carlo (GCMC) method. Recently, Zhou et al. have observed various structural defects in chemical vapor phase grown monolayer  $\text{MoS}_2$  by direct atomic resolution imaging, including point defects, dislocations, grain boundaries and edges<sup>29</sup>. Among these point defects, the single vacancy of S is frequently observed in experiments, which indicates its stability. These defective sites are reactive and could influence the electronic properties, so here we consider the monolayer  $\text{MoS}_2$  with point defects for the gas adsorption. Our results confirm that monolayer  $\text{MoS}_2$  with defect is a highly efficient absorbent for non-polar gases.

## 2. METHOD

All computations are performed using all-electron density functional theory (DFT) with a double numerical basis set plus polarization functions (DNP), as implemented in the DMol<sup>3</sup> program<sup>30</sup>. The generalized gradient approximation (GGA) with Perdew–Burke–Ernzerhof (PBE) exchange – correlation functional including van der Waals (vdW) corrections (DFT-D2 Grimme’s method) was adopted<sup>31</sup>. Self-consistent field (SCF) computations were performed with a convergence criterion of  $10^{-5}$  a.u. for the total energy. To make sure to get the high-quality numerical results, we chose the

real-space global orbital cutoff radius as high as 5.5 Å in all computations without symmetry constraint. A  $6 \times 6$  supercell of hexagonal MoS<sub>2</sub> unit cell was used to study the defective structures and gas adsorption. The supercell dimension for the direction perpendicular to the MoS<sub>2</sub> layer is chosen as 15 Å to avoid the interaction between neighboring layers. During the geometry optimization, the  $k$  space was sampled by the Gamma point in the Brillouin zone, and the Monkhorst–Pack grids with separation of 0.03 Å were chosen.

The adsorption energy for CO<sub>2</sub> and CH<sub>4</sub> molecules on the monolayer MoS<sub>2</sub> with and without vacancies is defined by

$$E_a = E_{gas-MoS_2} - E_{MoS_2} - E_{gas} \quad (1)$$

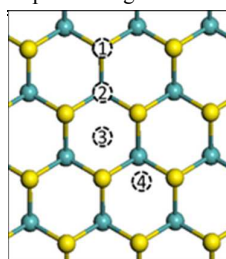
where  $E_{gas-MoS_2}$  represents the total energy of the complexes of gases and MoS<sub>2</sub> substrates,  $E_{MoS_2}$  and  $E_{gas}$  are the total energies of the MoS<sub>2</sub> substrate and the adsorbed gas molecules, respectively. A negative value of  $E_a$  indicates that the gas molecule prefers to adsorb on the substrate, while a positive value of  $E_a$  means a repulsion interaction between the gas molecule and the substrate.

The gas adsorption properties were simulated using the grand canonical Monte Carlo method, which implemented in Sorption module in Materials Studios 7.0. During GCMC simulations, the MoS<sub>2</sub> sorbent were assumed to be rigid with constraint atoms. The van der Waals (vdW) interactions between the gas molecules and the MoS<sub>2</sub> sheet are described by the Universal force field<sup>32</sup> but with the revised Lennard-Jones (LJ) parameters obtained from our first-principles calculations. We carried out a series of fixed pressure simulations at the room temperature of 298 K from 0 to 80 bar. At each pressure, the GCMC simulation consisted of  $1 \times 10^6$  steps to allow equilibration, followed by  $1 \times 10^7$  steps to sample the configuration space.

### 3. RESULTS AND DISCUSSION

#### 3.1 CO<sub>2</sub> and CH<sub>4</sub> adsorption on perfect MoS<sub>2</sub> layer

TABLE I. Adsorption energies and distances for gas molecules adsorbed on perfect MoS<sub>2</sub> layer at different sites.



Site	CO <sub>2</sub>		CH <sub>4</sub>	
	E <sub>a</sub> (kJ/mol)	d (Å)	E <sub>a</sub> (kJ/mol)	d (Å)
(1) T-S	-6.9	3.40	10.1	3.41
(2) T-Mo	-10.2	2.90	8.5	3.76
(3) Hollow	-9.6	3.16	8.6	3.78
(4) B-2S	-9.3	3.04	8.7	3.54

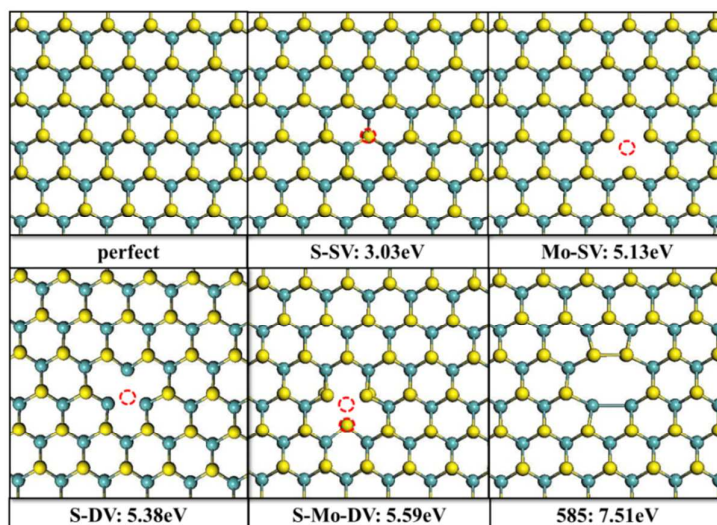
Four geometric sites have been considered for the gas molecules (CO<sub>2</sub> and CH<sub>4</sub>) adsorbed on perfect MoS<sub>2</sub> layer: 1) top S site, 2) top Mo site, 3) hollow site at the center of a hexagonal ring, and 4) bridge site between two neighboring S atoms in the same layer. TABLE I summarizes the gas adsorption energies and distances for both CO<sub>2</sub> and CH<sub>4</sub> molecules adsorbed on perfect MoS<sub>2</sub> layer at different sites. For CO<sub>2</sub>, the adsorption energies on these four sites are all negative with the similar values, which indicates that it prefers to adsorb on MoS<sub>2</sub> surface without site dependence, but the adsorption energies are very small (less than 0.1 eV). On the other hand, the values of adsorption energies for CH<sub>4</sub> are all positive, which means that there is a repulsion interaction between CH<sub>4</sub> and MoS<sub>2</sub> layer. Among these four sites, the distances between CO<sub>2</sub> (or CH<sub>4</sub>) and the Mo atom for the T-Mo site are (2.901 Å for CO<sub>2</sub> and 2.792 Å for CH<sub>4</sub>, respectively) shorter than the others (more than 3.1 Å). Therefore, our results show that the perfect MoS<sub>2</sub> is not a suitable sorbent for non-polar gas adsorption, especially for CH<sub>4</sub> molecule.

#### 3.2 CO<sub>2</sub> and CH<sub>4</sub> adsorption on MoS<sub>2</sub> with vacancies

The structural defects on MoS<sub>2</sub> layer has been observed and extensively investigated recently<sup>15, 20, 29,</sup>

<sup>33</sup>. The results show that the defective sites in MoS<sub>2</sub> layer are reactive and could influence the

electronic properties. Thus, here we propose to use the monolayer MoS<sub>2</sub> with point defects as the sorbent for the non-polar gas molecules adsorption. We considered five types of defective structures including single vacancy of missing one S (S-SV), single vacancy of missing one Mo (Mo-SV), double vacancies by missing one Mo and one S (Mo-S-DV), double vacancies by missing two S (2S-DV) and 585 defect by missing a MoS<sub>2</sub> unit. The optimized configurations are shown in FIG. 1. The defect formation energies are calculated using the Brenstone as the reservoirs at the experimental condition of S rich. From the results, the S-SV defect is found to show the lowest formation energy of 3.03 eV, while the 585 defect with missing a MoS<sub>2</sub> unit possesses the highest formation energy of 7.61 eV under S-rich environment. This is consistent with the experimental observations<sup>29</sup> and other theoretical calculations<sup>28</sup>, where S-SV is frequently observed, and other defects were only occasionally found. In the S-SV structure, three Mo atoms at the vacancy site move closer to each other and form a small triangle like structure. The neighbor Mo–Mo bond lengths ( $d_{\text{Mo-Mo}}$ ) become 3.055 Å, shorter than that in the perfect MoS<sub>2</sub> sheet (3.166 Å). Thus, the formation energy was lowered significantly due to the formation of three additional Mo–Mo bonds by removing of an S atom. And it is the same mechanism for the S-DV structure. On the other hand, in the Mo-SV, the three S atoms at the vacancy site move apart from each other and form a large triangle-like structure after optimization. And so does the S-Mo-DV structure. When one MoS<sub>2</sub> unit is missed, there will be two pentagonal rings and one octagonal ring forming the 585 defect, which possesses the highest formation energy.



**FIG. 1.** Optimized structures and formation energies of (a) perfect MoS<sub>2</sub> and MoS<sub>2</sub> with (b) S-SV, (c) Mo-SV, (d) S-DV, (e) S-Mo-DV and (f) 585 defects.

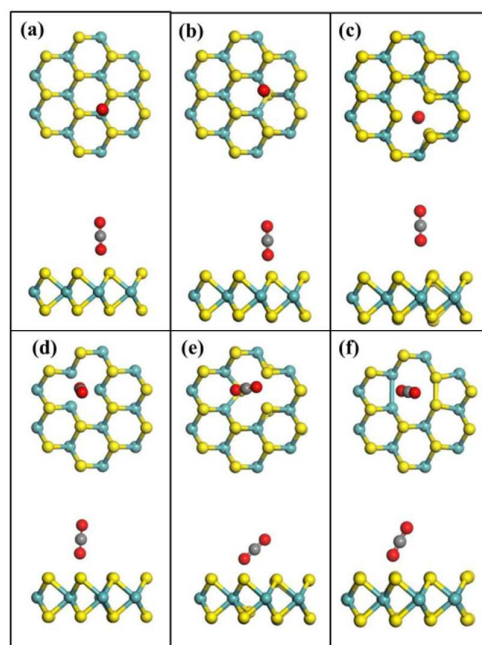
Then, we calculated the adsorption properties of CO<sub>2</sub> and CH<sub>4</sub> molecules on the MoS<sub>2</sub> layer with these vacancies. TABLE II summarizes the adsorption energies, adsorption distances and charge transfer between the adsorbed gas molecules and the MoS<sub>2</sub> layer. The adsorption distance is defined by the vertical distance between the C atom in CO<sub>2</sub> or CH<sub>4</sub> molecules and the top S-layer of the MoS<sub>2</sub> sheet. We found that the adsorption energies for CO<sub>2</sub> and CH<sub>4</sub> on the MoS<sub>2</sub> sheet with S-SV defect can be as high as -65.8 kJ/mol for CO<sub>2</sub> and -47.6 kJ/mol for CH<sub>4</sub>, respectively, which are much stronger than these gases adsorbed on some MOFs or carbon-based nanomaterials.<sup>3, 9</sup> The S-DV defect also possesses high adsorption energies close to S-SV defect, which are -64.1 kJ/mol for CO<sub>2</sub> and -45.2 kJ/mol for CH<sub>4</sub>, respectively. The adsorption distances for these two defects are around 2.5 Å, which are shorter than the other defects. For CO<sub>2</sub>, S-Mo-DV and 585 defect structures give the moderate adsorption energies of -20.5 kJ/mol and -26.9 kJ/mol, respectively, while Mo-SV structure shows the lowest adsorption energy of -9.4 kJ/mol, comparable to that on perfect MoS<sub>2</sub> layer. FIG. 2 shows the adsorption configuration of CO<sub>2</sub> adsorbed on MoS<sub>2</sub> with and without defects. There are two types of configurations for CO<sub>2</sub>, one is the linear CO<sub>2</sub> molecule perpendicular to the surface of MoS<sub>2</sub> layer, and



the other is the linear CO<sub>2</sub> molecule tilts with the O atom pointing to the S defect. For CH<sub>4</sub>, the 585 defect structure gives a low adsorption energy of -5.1 kJ/mol, while both Mo-SV and S-Mo-DV structures show the repulsive interaction with the CH<sub>4</sub> molecule as the perfect MoS<sub>2</sub> layer. From the Mulliken population analysis, the charge transfer between CO<sub>2</sub> (or CH<sub>4</sub>) molecules and the S-SV and S-DV defects are relatively larger than the others, which is further proved that the gas molecules give a stronger interacting with S-SV and S-DV defects.

**TABLE II.** Adsorption energies ( $E_a$ ), adsorption distances ( $d$ ) and the Mulliken charge transfer ( $Q$ ) between gas molecules and MoS<sub>2</sub> with and without defects.

Structures	CO <sub>2</sub>			CH <sub>4</sub>		
	$E_a$ (kJ/mol)	$d$ (Å)	$Q$ (e)	$E_a$ (kJ/mol)	$d$ (Å)	$Q$ (e)
perfect	-10.2	3.22	0	8.9	3.23	0
S-SV	-65.8	3.04	0.020	-47.6	2.58	0.041
Mo-SV	-9.4	3.24	0	8.7	3.30	0
2S-DV	-64.1	3.20	0.021	-45.2	2.50	0.058
S-Mo-DV	-20.5	2.52	0.013	10.1	3.20	0
585	-26.9	2.79	0.018	-5.1	2.75	0.020

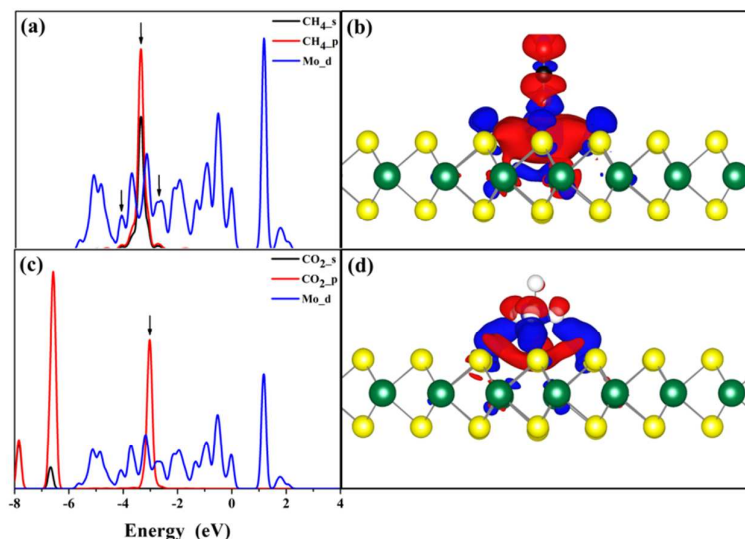


**FIG. 2.** Top view and side view for CO<sub>2</sub> molecule adsorbed on (a) perfect MoS<sub>2</sub> and MoS<sub>2</sub> with (b) S-SV, (c) Mo-SV, (d) S-DV, (e) S-Mo-DV and (f) 585 defects.

To investigate the interaction between gas molecules and the MoS<sub>2</sub> layer, we analyzed the projected electronic density of states (PDOS) for the *d* orbital of Mo atoms near the vacancy site and the *s* and *p* orbitals of gas molecules. FIG. 3 shows the PDOS and the deformation charge density for CO<sub>2</sub> and CH<sub>4</sub> molecules adsorbed on MoS<sub>2</sub> with S-SV defect. When a CO<sub>2</sub> (or CH<sub>4</sub>) molecule adsorbs on the vacancy site, there is an orbital coupling between the *p* orbital of CO<sub>2</sub> (or CH<sub>4</sub>) molecule and the *d* orbital of Mo atom. The deformation charge density is also calculated for CO<sub>2</sub> and CH<sub>4</sub> molecules adsorbed on MoS<sub>2</sub> with S-SV and S-DV defects. The differential charge ( $\Delta Q$ ) is defined as

$$\Delta Q = Q(\text{gas-MoS}_2) - Q(\text{MoS}_2) - Q(\text{gas}) \quad (2)$$

where  $Q(\text{gas-MoS}_2)$ ,  $Q(\text{MoS}_2)$  and  $Q(\text{gas})$  are the charge density for the gas-MoS<sub>2</sub> system, MoS<sub>2</sub> substrate and a gas molecule, respectively. The deformation charge density for CO<sub>2</sub> and CH<sub>4</sub> molecules adsorbed on MoS<sub>2</sub> with S-SV is plotted in Fig. 3(b)-(d), which indicates a Van der Waals interaction with polarization of the gas molecules and the MoS<sub>2</sub> substrate. The charge difference distributes in the middle region between the gas molecules (CO<sub>2</sub> and CH<sub>4</sub>) and the MoS<sub>2</sub> substrate, suggesting partly covalent character. Therefore, the interactions between the gas molecules and the MoS<sub>2</sub> with S-SV defect are mainly weak van der Waals interaction with partly covalent character. This hybrid interaction is also observed when the gas molecules adsorbed on the S-DV site of MoS<sub>2</sub> substrate, which is plotted in Fig. S1 of Supplementary Information.

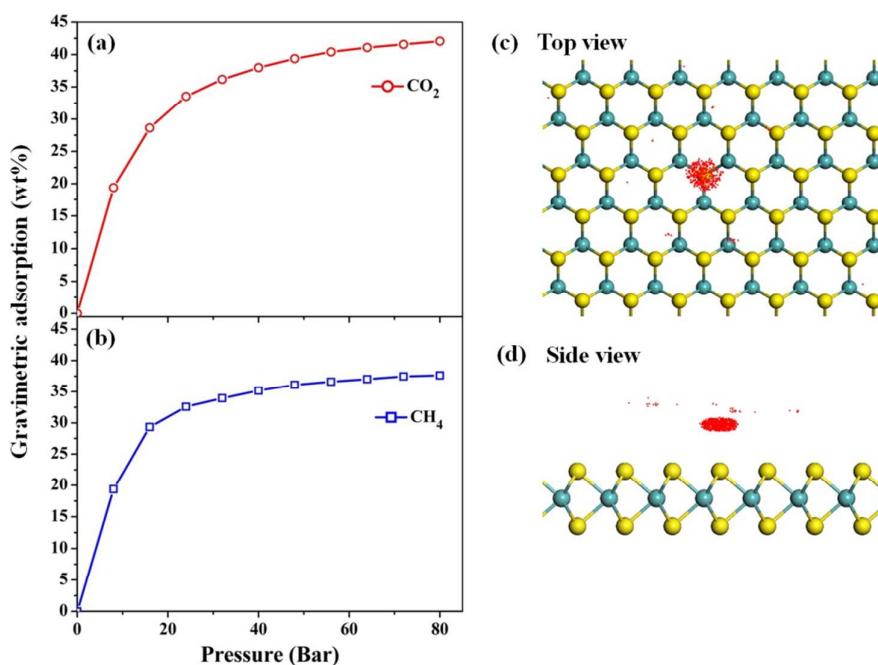


**FIG. 3.** The projected density of states (PDOS) for the  $d$  orbital of Mo and the  $s$  and  $p$  orbitals of (a) CH<sub>4</sub> and (c) CO<sub>2</sub>; (b) and (d) are the deformation charge density for CO<sub>2</sub> and CH<sub>4</sub> adsorbed on MoS<sub>2</sub> layer with S-SV defect. The increase and decrease of the electron density are colored in red and blue, respectively.

### 3.3 Gas adsorption isotherms

To determine the capability of MoS<sub>2</sub> with vacancies as absorbent for CO<sub>2</sub> / CH<sub>4</sub>, we perform GCMC (grand canonical Monte Carlo) calculations. The calculated gravimetric adsorption isotherms of the CO<sub>2</sub> and CH<sub>4</sub> molecules on MoS<sub>2</sub> layer with S-SV defect at room temperature (298 K) are plotted in FIG. 4a-b as an example. We can see that the adsorption increases with the pressure increasing and gradually approaches saturation, which indicates the stronger adsorption. The adsorption amount of CH<sub>4</sub> is much more than that of CO<sub>2</sub> on MoS<sub>2</sub> with S-SV defect, for example, 29.3wt% for CH<sub>4</sub> and 28.6wt% for CO<sub>2</sub> at the pressure of 16 bar and 35.2wt% for CH<sub>4</sub> and 38wt% for CO<sub>2</sub> at 40 bar, respectively. Finally, the adsorption amount of CH<sub>4</sub> and CO<sub>2</sub> reaches to 37.6wt% and 42.1wt% at the pressure of 80 bar and the room temperature, respectively. We also investigate the density distribution of the CH<sub>4</sub> and CO<sub>2</sub> molecules adsorbed on the MoS<sub>2</sub> with S-SV defect. FIG. 4c-d shows the top view and side view of adsorption density distribution for CH<sub>4</sub> molecules on MoS<sub>2</sub> with S-SV defect as an

example at 80 bar. Obviously, the  $\text{CH}_4$  molecules are not distributed uniformly, but congregate around the defective sites, which is consistent with our first-principles calculations. The same phenomenon is also observed for  $\text{CO}_2$  molecule (see Fig. S2 in Supplementary Information). Therefore, the  $\text{MoS}_2$  with defects can be used as an efficient absorbent for non-polar gas molecules, and the adsorption capability can be modulated by the density of defects in  $\text{MoS}_2$  layer. Since in experiments, bilayer and multilayer of  $\text{MoS}_2$  are commonly synthesized and have enough space between layers for gas molecules to access, we speculate that multilayer  $\text{MoS}_2$  can be more effective at capturing these gases and these calculations are currently in progress.



**FIG. 4.** The adsorption isotherms for (a)  $\text{CH}_4$  and (b)  $\text{CO}_2$  on monolayer  $\text{MoS}_2$  with S-SV defect at 298K; (c) and (d) are the top view and side view of adsorption density distribution for  $\text{CH}_4$  molecules on  $\text{MoS}_2$  with S-SV defect at 80 bar, respectively.

#### 4. CONCLUSION

In this work, we systematically investigated the adsorption properties for non-polar gas molecules ( $\text{CO}_2$  and  $\text{CH}_4$ ) on perfect monolayer  $\text{MoS}_2$  and  $\text{MoS}_2$  with vacancies by first-principles calculations

combined with GCMC simulations. Perfect MoS<sub>2</sub> lacks the adsorption ability for these non-polar gases, but MoS<sub>2</sub> with point defects especially the S vacancy displays good adsorption behavior with the large adsorption energies of 65 kJ/mol for CO<sub>2</sub> and 47 kJ/mol for CH<sub>4</sub>, respectively. Density of states and deformation charge density analysis show that the interactions between the gas molecules and the MoS<sub>2</sub> layer with vacancies are mainly weak van der Waals interaction with some covalent character. Our GCMC calculations predict a high adsorption amount of 42.1wt% for CO<sub>2</sub> and 37.6wt% for CH<sub>4</sub> on MoS<sub>2</sub> with single S vacancy under the pressure of 80 bar at the room temperature, respectively. Our results indicate that monolayer MoS<sub>2</sub> with defects is a highly efficient absorbent for non-polar gases.

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## REFERENCES

- 1 S. Hermes, M. K. Schroter, R. Schmid, L. Khodeir, M. Muhler, A. Tissler, R. W. Fischer and R. A. Fischer, *Angew. Chem. Int. Ed. Engl.*, 2005, **44**, 6237.
- 2 Y. Li and R. T. Yang, *Langmuir*, 2007, **23**, 12937.
- 3 D. Saha, Z. Bao, F. Jia and S. Deng, *Envir. Sci. Technol.*, 2010, **44**, 1820.
- 4 Y. Liu, D. Liu, Q. Yang, C. Zhong and J. Mi, *Ind. Eng. Chem. Res.*, 2010, **49**, 2902.
- 5 S. S. Han, H. Furukawa, O. M. Yaghi and W. A. Goddard Iii, *J. Am. Chem. Soc.*, 2008, **130**, 11580.
- 6 J. Vermesse, D. Vidal and P. Malbrunot, *Langmuir*, 1996, **12**, 4190.
- 7 R. Krishna and J. M. van Baten, *Langmuir*, 2010, **26**, 3981.
- 8 F. Schedin, A. K. Geim, S. V. Morozov, E. W. Hill, P. Blake, M. I. Katsnelson and K. S. Novoselov, *Nat. Mat.*, 2007, **6**, 652.
- 9 B. Huang, Z. Li, Z. Liu, G. Zhou, S. Hao, J. Wu, B.-L. Gu and W. Duan, *J. Phys. Chem. C*, 2008, **112**, 13442.
- 10 R. Arsat, M. Breedon, M. Shafiei, P. Spizziri, S. Gilje, R. Kaner, K. Kalantar-zadeh and W. Wlodarski, *Chem. Phys. Lett.*, 2009, **467**, 344.
- 11 L. Wang, J. Zhao, L. Wang, T. Yan, Y.-Y. Sun and S. B. Zhang, *Phys. Chem. Chem. Phys.*, 2011, **13**, 21126.
- 12 K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. Heinz, *Phys. Rev. Lett.*, 2010, **105**, 136805.
- 13 A. Castellanos-Gomez, R. Roldan, E. Cappelluti, M. Buscema, F. Guinea, H. S. van der Zant and G. A. Steele, *Nano lett.*, 2013, **13**, 5361.
- 14 H. J. Conley, B. Wang, J. I. Ziegler, R. F. Haglund, Jr., S. T. Pantelides and K. I. Bolotin, *Nano lett.*, 2013, **13**, 3626.
- 15 H.-P. Komsa, S. Kurasch, O. Lehtinen, U. Kaiser and A. V. Krasheninnikov, *Phys. Rev. B*, 2013, **88**, 035301.
- 16 Y. Li, H. Wang, L. Xie, Y. Liang, G. Hong and H. Dai, *J. Am. Chem. Soc.*, 2011, **133**, 7296.
- 17 X. Liu, T. Xu, X. Wu, Z. Zhang, J. Yu, H. Qiu, J. H. Hong, C. H. Jin, J. X. Li, X. R. Wang, L. T. Sun and W. Guo, *Nature Commun.*, 2013, **4**, 1776.
- 18 Y. Liu, L. Jiao, Q. Wu, Y. Zhao, K. Cao, H. Liu, Y. Wang and H. Yuan, *Nanoscale*, 2013, **5**, 9562.
- 19 Y. Shi, J. K. Huang, L. Jin, Y. T. Hsu, S. F. Yu, L. J. Li and H. Y. Yang, *Sci. Rep.*, 2013, **3**, 1839.
- 20 J. Xie, H. Zhang, S. Li, R. Wang, X. Sun, M. Zhou, J. Zhou, X. W. Lou and Y. Xie, *Adv. Mater.*, 2013, **25**, 5807.
- 21 C. R. Zhu, G. Wang, B. L. Liu, X. Marie, X. F. Qiao, X. Zhang, X. X. Wu, H. Fan, P. H. Tan, T. Amand and B. Urbaszek, *Phys. Rev. B*, 2013, **88**, 121301.
- 22 S. Yang, D. Li, T. Zhang, Z. Tao and J. Chen, *J. Phys. Chem. C*, 2012, **116**, 1307.
- 23 Q. Yue, S. Chang, J. Kang, X. Zhang, Z. Shao, S. Qin and J. Li, *J. Phys-Condens. Mat.*, 2012, **24**, 335501.
- 24 B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti and A. Kis, *Nature Nanotech.*, 2011, **6**, 147.
- 25 Z. Yin, H. Li, H. Li, L. Jiang, Y. Shi, Y. Sun, G. Lu, Q. Zhang, X. Chen and H. Zhang, *ACS nano*, 2011, **6**, 74.
- 26 H. Li, Z. Yin, Q. He, H. Li, X. Huang, G. Lu, D. W. H. Fam, A. I. Y. Tok, Q. Zhang and H. Zhang, *Small*, 2012, **8**, 63.
- 27 Q. He, Z. Zeng, Z. Yin, H. Li, S. Wu, X. Huang and H. Zhang, *Small*, 2012, **8**, 2994.
- 28 Q. Yue, Z. Shao, S. Chang and J. Li, *Nanoscale Res. Lett.*, 2013, **8**, 1.
- 29 W. Zhou, X. Zou, S. Najmaei, Z. Liu, Y. Shi, J. Kong, J. Lou, P. M. Ajayan, B. I. Yakobson and J. C.

- Idrobo, *Nano Lett*, 2013, **13**, 2615.
- 30 B. Delley, *Comp. Mater. Sci*, 2000, **17**, 122.
- 31 S. Grimme, *J. Comp. Chem*, 2006, **27**, 1787.
- 32 A. K. Rappé, C. J. Casewit, K. Colwell, W. Goddard Iii and W. Skiff, *J. Am. Chem. Soc*, 1992, **114**, 10024.
- 33 A. N. Enyashin, M. Bar-Sadan, L. Houben and G. Seifert, *J. Phys. Chem. C*, 2013, **117**, 10842.