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Controlled Nitrogen Insertion in Titanium Dioxide for Optimal Photocatalytic Degradation of Atrazine

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15	Abstract
16 17 18 19 20 21 22 23 24 25 26	Introducing defects in intrinsic TiO_2 structural framework with nitrogen, enhanced the photocatalytic response towards the degradation of atrazine, as compared to undoped TiO_2 . Both catalysts, prepared in an analogous manner demonstrated high crystallinity and well defined {100} facets dominant anatase, which serves as a pioneer platform for a good photocatalytic activity. The introduction of nitrogen increased the stability of the crystal structure which leads to the formation of pure active anatase phase. Although the optical response was shifted towards the visible region initiating the formation of new absorption defects and interstates energy level, the chemical state of nitrogen in doped TiO_2 controls the overall catalyst photoreactivity. In this study, it was found that surface area and degree of band gap reduction played a lesser role for photocatalysis enhancement, although it partly contributed, rather the concentration of surface charge traps and the type of structural framework formed during nitrogen incorporation. The enhancement in the photocatalytic degradation of atrazine clearly was influenced by the loading and doping nature of nitrogen dopant, which in turned, governed the type of chemical and optical properties of the final catalyst product.
27 28	Keywords: Photocatalysis, N-TiO ₂ , Doping, Visible light; Atrazine
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40 Introduction

41 Since the green revolution introduced by Nobel laureate, Normal Borlaug in 1968, the usage of synthetic chemicals in 42 maximizing crop productions increased rapidly and gave a significant impact towards decreasing the percentage of world 43 starvation. However, despite the positive changes in sustaining the global needs, the green revolution had resulted in 44 unintended serious public health and environmental issues (1). Atrazine, an s-triazine group is a type of herbicides that is 45 most commonly used in Malaysia to control the growth of broad leaf and grassy weeds (2) (Fig. 1). Due to the stable s-46 triazine ring of atrazine, it yields poor biodegradability and has great potential to leach through the soils and enters the 47 ground water system (3). According to Mc. Murray et al. (4), atrazine is repeatedly found in drinking water supplies and were 48 detected above the recommended level of 0.1 ppb throughout European countries and the United States.

The technology for water pollutant abatement which includes absorption, ultrasonic destruction, microbial, chemical oxidation and filtration are effective to a certain degree of treatment, however, lacking the necessity to degrade very low pollutant concentrations. Atrazine are usually found in parts per billion levels in water sources and can be efficiently degraded into its harmless intermediates via photocatalysis. Unlike adsorption, the hazardous pollutants are destroyed rather than being transferred to another phase (5). Photocatalysis also offers a more economical approach as there is no requirement for waste disposal, large energy usage and is less sensitive towards environmental change, i.e. pH of waste water.

55 The photocatalytic degradation of atrazine using 2.0 g/L of TiO₂ was intensively studied by Pellizzetti et al. (3) irradiated 56 under solar lamp. The photocatalytic efficiency was proven as it took only 10 minutes to degrade 2 ppb of atrazine to less 57 than 0.1 ppb. However, longer time was required for complete mineralization of the intermediates products. Following this 58 discovery, many had attempted to degrade atrazine using TiO_2 nanoparticulate. C. Minero et al. (6) evaluated the 59 photocatalytic degradation of 10 ppm atrazine using 0.2 g/L of TiO₂ suspensions and complete degradation was achieved in 60 less than 120 min, I.K Konstantinou et al. (7) observed partial degradation of 10 ppm atrazine using 0.1 g/L of TiO₂ 61 suspensions within half-lives $(t_{1/2})$ of 18.6 min, O. Zahraa et al. (8) degraded 10 ppm of atrazine using 1 g/L of TiO₂ 62 suspensions and took 200 min for complete removal of atrazine and C.A Ruslimie et al. (9) evaluated the degradation of 5 63 ppm atrazine using 0.2 g/L TiO₂ and took more than 240 min for 70.6% atrazine degradation. The photocatalytic degradation 64 of atrazine rate differs in each experiment and could be due to the different types of reactor configuration and TiO₂ synthesize 65 method. Hence, an accurate evaluation of the photodegradation process could not be done. In all of these works, complete 66 mineralization of atrazine was not observed, and a final stable end product of cyanuric acid was obtained.

67 One of the main factors that control the rate of photocatalytic degradation is the physicochemical properties of the catalyst 68 used (10-11). The application of intrinsic TiO₂ is limited due to the wide band gap which requires substantial amount of 69 photons for electron/hole excitation (12). Furthermore, the absorption efficiency of intrinsic TiO₂ covers only 4-5% of the 70 UV region, leaving the major constituents of the solar light unharnessed (13). Inducing defects into TiO_2 lattice using non-71 metal ions eliminates the drawbacks of doping using transition metals such as multiple charge carrier recombination centers 72 within the band gap due to deep localized d states (14-15). Nitrogen doped TiO₂ was initiated by Asahi et al. (14), where 73 $TiO_{2,x}N_x$ films were prepared by sputtering TiO_2 in $N_2(40\%)/Ar$ gas mixture and subsequently annealed at 550°C in N_2 gas 74 for 4 hours. The result showed the ability of $TiO_{2,x}N_x$ films to absorb lights below 500 nm, which was not observed for naked 75 TiO_2 films. Furthermore, the hydrophilicity of $TiO_{2-x}N_x$ films was enhanced, an important criteria for a good photocatalyst 76 system (16). Many works in N-TiO₂ have since been reported (13, 17-21) concluding the positive effect of nitrogen 77 incorporation into TiO₂ in terms of particles homogeneity, wider absorption of solar light, narrowing of band gap, alternative 78 pathways for electron/hole recombination, retardation of crystal phase changes and enhanced photocatalytic degradation 79 activity.

80 It is undeniable that successful preparation of N-TiO₂ photocatalyst generated successive visible-energy light conversion 81 and improved its photocatalytic response in degrading organic pollutants. However, there is still a gap in understanding the 82 factors which influences the photoactivity, especially by varying the concentration of nitrogen that enters the TiO₂ framework 83 and creating either coordinated or substitutional doping. The degree of impurity imposed by introducing different loading of 84 nitrogen was studied in this work, and was connected carefully to the photocatalytic degradation of atrazine. In addition, very 85 little work aforementioned evaluated on the photodegradation of atrazine using N-TiO₂ and was discussed in this paper, 86 followed by the degradation pathway and final formation of stable-end product of cyanuric acid.

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88 Experimental

89 Materials

90 Titanium (IV) isopropoxide (TTIP, 97%), hydrochloric acid (37%), ethanol (95%), anhydrous ethanol (99%), triethylamine 91 (99%) and atrazine (98%) were purchased from Sigma Aldrich. All chemical were used as received without further pre-92 treatment. Milli-Q deionized water was used throughout the whole experiment.

93 Synthesis of N-TiO₂

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Five (different nitrogen loading content) of N-TiO₂ photocatalyst were synthesized using modified sol-gel method (22) and titanium (IV) isopropoxide (TTIP) as the metal precursor, pre-dissolved in anhydrous ethanol. Deionized water and hydrochloric acid in the molar ratio of 15:0.5 was added drop wise to obtain a clear sol, followed by the addition of triethylamine from an initial molar ratio of 0.5 to 2.5. The sol was left to age prior to solvent evaporation at 80°C overnight. The dried sol gels were further decomposed at 500 °C for 6 hours under constant air flow rate of 20 ml/min to obtain white

99 N-TiO₂ powder. Similar method was employed for preparing un-doped TiO₂ without the addition of nitrogen precursor.

100 Characterization of N-TiO₂

101 The surface morphology and elemental composition of successful nitrogen doped in TiO₂ was observed with FESEM Quanta 102 FEI 200F and energy dispersive x-ray spectroscopy (INCA Software). CHNS elemental analyzer (Perkin Elmer, 2400 series 103 II) was also used to provide secondary information on the content of nitrogen successfully doped in TiO₂, using 104 sulfamethazine (Sigma Aldrich) as the standard. The specific BET surface area and porosity were measured using multipoint 105 nitrogen adsorption-desorption analyzer at relative pressure range of 0.01-0.90 P/Po. The crystallinity and crystal phase of the 106 samples was characterized by powder X-ray diffraction (XRD, Bruker AXS D8) using Cu Ka radiation. High resolution 107 transmission electron microscopy (HR-TEM, JEM 2100-F) was performed at an accelerating voltage of 200kV was used to 108 determine the microstructure of the catalyst. The samples were examined using 40 kV and 40 mA at a step size of 0.002 109 starting from 10° to 80°. The Raman spectra and photoluminescence (PL) measurement was performed using a LabRam 110 confocal Raman microscope. Samples were excited by the 325 nm line of a continuous He-Cd laser at room temperature. 111 Analysis of the functional group and types of chemical bonds was measured using Fourier Transformed Infra-red, FTIR 112 Bruker Vertex 80/80v from 4000 to 400 cm⁻¹. The absorption edge of the prepared photocatalyst was measured using 113 ultraviolet visible spectroscopy (Agilent, Cary 100) with a diffuse reflectance accessory. X-ray photoelectron spectroscopy 114 (XPS) surface analyses were carried out using a flat gold (Si/10nm Ti/200nm Au) as the substrate and reference. Thermo 115 Scientific K-alpha instrument with an un-monochromatized Mg Ka radiation (photon energy 1253.6 eV) source and vacuum 116 better than 10^{-9} Torr as well as spectral resolution of 0.1 eV was used for XPS studies. XPS core levels were aligned to the 117 C1s binding energy (BE) of 285 eV. The weight fractions of the crystal phase is calculated using Spurr equation of $f_A =$ 118 $1/(1+1.26 \text{ x } I_R/I_A)$ where f_A is the weight fraction of anatase, $f_R = 1 - f_A$ is the weight fraction of rutile, I_A is the intensity of 119 maximum anatase phase peak {101}, I_R is the intensity of maximum rutile phase peak {110} and 1.26 is the scattering 120 coefficient (23). For undoped TiO₂, the I_R and I_A is 5.57 and 108.44 respectively.

121 Photocatalytic Reactor and Degradation Activity

122 A custom built stirred tank photo-reactor (STR) consisting of six 250 ml quartz tube of equal dimensions were systematically 123 arranged to received equal light intensity. The source of light was Xenon lamp (150W) with a UV light block filter ($\lambda < 420$ 124 nm), positioned in the center of the reactor. The mass transfer between the photocatalyst and atrazine was enhanced via 125 turbulent mixing at 450 rpm. Undoped TiO₂ and N-TiO₂ powder was used to degrade an initial atrazine concentration of 0.5 126 mg/L. The initial molar ratio of nitrogen in doped TiO₂ were 0.5, 1.0, 1.5, 2.0 and 2.5 with an initial catalyst loading of 0.5 127 g/L. Prior to the experiment, the photocatalyst were left in the aqueous suspension in darkness for 70 minute to achieved 128 adsorption equilibrium between the catalyst and atrazine molecules. The experiment was treated in batch operation mode 129 with a constant temperature maintained at $25\pm2^{\circ}$ C. Throughout the experiments, 5.0 cm³ of liquid samples were taken at 130 various time intervals. The resultant aqueous solution was filtered prior to UV-Vis analysis (Perkin Elmer, Lambda 35). The 131 average wavelength of atrazine used for UV-Vis analysis is 222.3 nm, with calibration coefficient, r^2 of 0.9998. The 132 degradation efficiency, η was described using the equation [$\eta = (c_0-c)/c_0 \ge 100\% = (A_0-A)/A_0 \ge 100\%$]; where c_0 and c/A_0 133 and A represents atrazine concentration/ absorption intensities at the initial and after photocatalytic activity for a certain time.

134 Intermediates Identification – LC/MS/MS

135 Atrazine degradation intermediates was determined using triple quadrupole liquid chromatography mass spectroscopy, 136 LCMS/MS (Agilent 6400 Series) under electrospray ionization (ESI) and Zorbax StableBond-C18 reversed phase column 137 (4.6 mm ID x 250 mm, 5 μ m). A full scan mode (1 spectra/s) with a flow rate of 0.3 ml/min and runtime of 15 minute was 138 used to determine the MS and MS/MS data. The mobile phase included filtered deionized water and acetonitrile at 9:1 139 volume ratio and the analysis was done under isocratic elution at pH 3.0.

140 **Result and discussion**

141 1. Characterization of N-TiO₂ powders

142 **Morphology**. Nitrogen was successfully introduced in TiO_2 in the range of 8-17 wt% as shown in Table 1. The nitrogen 143 content in CHNS appeared slightly higher than by EDX, as CHNS measures the total bulk nitrogen while EDX measures the 144 nitrogen presence on the surface of N-TiO₂. In both prepared catalyst, no change of color was observed and remained as 145 white powder. This implies that the probability of coordinated nitrogen doping or chemisorbed γ -N₂ is present in TiO₂, as 146 color changes is generally related to the red shift of the absorption edge due to substitutional doping (24). The FESEM 147 images of undoped and doped TiO₂ possessed agglomerates of spherical particles with an average particle diameter of 27 nm 148 to 37 nm (Fig. 2). The EDX spectrum can be observed in Fig. 3. Although it was observed that N-precursor did not influence 149 much on the size and shape of the particles, there was a slight increment of particle size for N-TiO₂ prepared at lower 150 nitrogen loading. This could be partially due to the larger size of nitrogen atomic radius as compared to the oxygen atomic 151 radius that occurred during dopant incorporation. This observation contradicts to the work done by M. Sathish et al. (18), 152 where a significant decreased in the particle size was obtained after nitrogen doping. At higher nitrogen loading, the particles 153 agglomeration appeared less, thus homogeneity and dispersibility of the particles was enhanced. This is clearly revealed in 154 HR-TEM images (Fig. 4) where N1-TiO₂ appeared more dispersed as compared to N0-TiO₂. Both sample showed irregular 155 crystal size ranging from 11 nm to 22 nm, where larger crystallite size was observed in doped TiO₂. The BET specific surface 156 area and porosity presented in Table 2 showed less influence on the change of porosity after nitrogen doping, as compared to 157 the specific surface area. The nitrogen absorption-desorption isotherms in Fig. 5 showed strong interaction of isotherm type 158 IV and narrow distribution of uniform mesoporous pores in the range of 2 to 50 (25).

159 Crystallinity and Crystal Phase. XRD analysis was used to determine the crystallinity, crystal phase and weight fraction of 160 each phase. Undoped TiO₂ consists of two crystal phase, namely anatase and rutile with mass ratio of 94:6 (Table 2). 161 Interestingly, all doped TiO₂ showed predominant anatase phase and enhanced crystallinity under all synthesis condition. A 162 major peak corresponding to {101} anatase phase was observed at angle 25°, followed by minor peaks at 37°, 47°, 54°, 55°, 163 62° , 69° , 70° and 75° and is based on JCPDS 731764 (Fig. 6). During the sol-gel process, the growth rate of the crystal 164 exceeded the nucleation rate due to the addition of amine aqeuous solution which affects the overal pH of the solution. The 165 presence of nitrogen in TiO_2 inhibits the condensation of spiral chain of anatase TiO_6 octahedral to linear chain of rutile TiO_6 166 octahedral. This can be seen from the disappearance of rutile major plane $\{110\}$ at 27° angle after the addition of nitrogen. 167 Similar observation was reported (20) whereby the addition of nitrogen improved the thermal stability of the catalyst by 168 increasing the temperature required for rutile phase formation. It was further observed that the presence of nitrogen affects 169 the full width of half maximum (FWHM), which is an indication of crystal size changes and possible lattice distortion (Fig. 170 7). At low nitrogen loading, there was a significant narrowing of FWHM, and the crystal size increased from 23 nm to 34 nm. 171 The drastic change in the crystal size, however, did not affect the lattice spacing as shown in Table 2 (26). All crystallite size 172 was calculated using Debye-Scherer's formula (27) using the highest intensity peak of the predominant phase.

173 Chemical Composition. RAMAN spectroscopy was employed to study the crystalline structure of undoped and doped TiO₂. 174 All prepared catalyst showed significant anatase phase with major bands at 144 cm⁻¹ (E_{σ}), 395 cm⁻¹ ($B1_{\sigma}$), 515 cm⁻¹ ($A1_{\sigma}$) and 175 637 cm^{-1} (E1_o) (Fig. 8). A weak, but apparent Raman band of anatase was shown at 195 cm⁻¹ (E_o). In all doped TiO₂, no 176 remarkable rutile band at 446 cm⁻¹ (E_e) was observed and supports the single crystal phase observed in the X-ray diffracion 177 (XRD) spectrum. Although, 6 wt% of rutile phase was present in undoped TiO₂, no significant rutile band can be seen in the 178 Raman spectra. This shows that Raman spectroscopy is suitable to identify the present of TiO_2 but not for distinguishing the 179 different phases of anatase and rutile, especially at a low mass fraction. G. Yang et al. (28) prepared N-TiO₂ at different N:Ti 180 atomic ratio using solvothermal method and obtained predominant anatase bands in the raman spectra. It was observed that 181 the amount of nitrogen loaded in TiO_2 influences the size of the particles in a linear proportion. Larger particle size caused a 182 blue shift towards a lower value of band in the raman spectra. Interestingly, in this work, the raman shifting patterns was 183 similar to doped TiO₂ prepared below 11 wt% of nitrogen loading (Fig. 9). However, as nitorgen loading increased, red 184 raman shift was observed, indicating smaller particles size. This observation is in good agreement with those shown in the 185 XRD data. FTIR analysis was used to analyzed the presence of nitrogen in TiO₂ prior to doping at different loading. In Fig. 186 10a-b, similar spectra was observed in both undoped and doped TiO₂ around 2363-2360 cm⁻¹ and 700-400 cm⁻¹ which 187 corresponds to absorbed CO₂ (29), Ti-O stretching and Ti-O-Ti bridging stretching mode (30). The enlarged FTIR spectra 188 from 1700 cm⁻¹ to 1200 cm⁻¹ revealed significant differences between both catalyst (Fig. 10c). Doped TiO₂ possessed 189 multiple peaks within this region which corresponds to surfaced absorbed nitrogen at 1463-1384 cm⁻¹ (28) and lattice 190 nitrogen in the TiO₂ at 1632 cm⁻¹, 1546 cm⁻¹, 1338 cm⁻¹ and 1255 cm⁻¹ (31).

191 **Chemical State.** XPS analysis was used to determine the surface chemistry of the prepared photocatalyst. In Fig. 11, no 192 nitrogen signal was detected in undoped TiO₂. The binding energy, E_g of N1s spectra was observed at 400.4 eV and no other 193 E_g was present between 395 eV to 399 eV and above 400.4 eV. From previous work, interstitial nitrogen showed E_g around 194 ~400 to 402 eV (*13,32*), 399.6 eV (*33*) and 399.8 eV (*34*). In addition, molecularly chemisorbed γ -N₂ shows Eg around ~400 195 to 402 eV (*35*). It is still debatious on the nature of N 1s present at E_g of ~400 to 402 eV, as whether nitrogen is doped 196 interstitially or is present as chemisorbed N₂ or both. The difference of E_g and XPS intensity for each reported work were

197 influenced by the selection of dopant (20) and catalyst preparation routes (33). In this work, triethylamine was used as the

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198 source of nitrogen precursor. J. Ananpattarachai et al. (20) used similar dopant in preparing N-TiO₂ and obtained E_g at 402.5 199 eV, which was assigned to nitric oxide or nitrogen monoxide (NO) and nitrite (NO2-), corresponding to an interstitial doped 200 nitrogen in TiO₂ due to the formed localised states with a π characteristics. L. Hu et al. (35) also incorporated triethylamine as 201 N-precursor in synthesizing N-TiO₂. Interestingly, it was reported that the point of dopant introduction during the hydrolysis 202 process determines the final state of N 1s formed in TiO2. Molecularly chemisorbed N2 are generally present if the 203 crystallization process of TiO2 occurs before the introduction of nitrogen and showed N 1s Eg around 400.5 and 401.4 eV 204 while substitutited nitrogen in vice versa. As shown in Fig. 11, similar broad N1s XPS emission was also observed in other 205 reported work (37-39), however strong corelation could not be made as different preparation procedures were employed. B. 206 Viswanathan et al. (40) reviewed that low nitrogen loading in TiO₂ generally consists of Eg of N1s spectra within ~396-397 207 eV while for high nitrogen loading at ~400 eV, however no quantification was mentioned. In this work, the reported E_g of 208 N1s peak at ~400.4 eV is attributed to the interstitial nitrogen in TiO₂ matrix with possible Ti-O-N or Ti-N-O linkages or 209 chemisorbed molecular γ -N₂. For the latter, TiO₂ was doped at high nitrogen loading, however, no significant change in the 210 crystal phase and lattice spacing was observed (Table 2). This indicates the presence of chemisorbed molecular γ -N₂ on the 211 surface of TiO₂. As the amount of nitrogen doped in TiO₂ increased, the crystallinity of TiO₂ decreased. Observations made 212 from FTIR (Fig. 10) supports the presence of probable Ti-O-N or Ti-N-O linkages as clear demarcation of lattice nitrogen is 213 observed especially in the magnified FTIR spectrum from 1200 - 1700 cm⁻¹. N-H bond and NO_x are not presence in N1-TiO₂ 214 due to the absency of E_{σ} around 398.7 eV and above 403 eV respectively (33). This is further supported by the FTIR 215 spectrum in Fig. 10c as no frequency of ammonia (N-H) was presenced. The probability of N-Ti-O lattice substitution can be 216 ruled out as no Eg was observed below 397.5 eV, a characteristic of substituted nitrogen (20). Futhermore, a slight negative 217 shift of E_g in Ti $2p_{3/2}$ between undoped TiO₂ and N-TiO₂ was observed in previous work when nitrogen substitutes for 218 oxygen atom in a TiO₂ lattice (41). The change of E_a is due to the difference of electronegativity of nitrogen and oxygen atom 219 in a N-Ti-O linkage, causing partial electron transformation from N to Ti as oxygen is more electronegative than nitrogen, 220 thus increasing the electron density of Ti. However, this was not observed in this study hence the probability of substituted 221 nitrogen in TiO₂ lattice is again eliminated. Based on Ti 2p spectra, the E_g of the photoelectron peak illustrates the existence 222 of Ti^{4+} species in the TiO_2 nanostructures (42). The two deconvoluted O 1s peak remained unchanged for both undoped and 223 doped TiO₂ at 530.0 eV and 531.7 eV. The former represents metallic oxide (Ti-O) and the latter represents surface hydroxyl, 224 O_{OH} and is in agreement with the IR frequency observed in N0-TiO₂ and N1-TiO₂ at 3437.8 cm⁻¹ and 3442 cm⁻¹ respectively. 225 No absorbed H₂O at ~532.7 eV was detected in the XPS peak (43). The presence of nitrogen in TiO₂ increased the number of 226 hydroxyl sites and enhance the hydrophilicity of the catalyst. The hydroxyl peak in both FTIR (Fig. 10) and XPS (Fig. 11) 227 analysis were more significant in N1-TiO2 as compared to N0-TiO2. In agreement with T. Morikawa et al. (16); the addition 228 of nitrogen increased the wettability of the catalyst surface, thus compliments the photocatalytic activity.

229 **Optical Properties.** The UV-Vis spectra of undoped and doped TiO₂ catalyst prepared at different nitrogen loading are 230 presented in Fig. 12. The corresponding band gap was obtained using Kubelka-Munk function (34) calculated using diffuse 231 reflectance spectrum, with an indirect band gap coefficient of 0.5(44) (Table 2). A significant increased of light absorption 232 towards lower photons energy level was observed for all doped TiO2, extending the absorption coverage towards 400-550 nm 233 of visible light region. This is due to the additional impurity level created within the band gap, thus causing a shift of the 234 fermi level closer to the conduction band and consequence narrowing of band gap. This claim is further supported by Xiang 235 et al. (25), whereby from first principle density functional theory (DFT) calculation, interstitial N-precursor can induced local 236 states above the valence band and is responsible for the visible light response. According to B. Tian et al. (29), the 237 incorporation of nitrogen in the TiO₂ matrix forms a narrow N 2p band, which overlapped with O 2p orbital, promoting 238 greater electrons mobility from the valence band to the conduction band during photoexcitation. However, no continuum 239 interstates overlapping was observed in this work, as XPS data clearly shows no presence of substituted nitrogen in the TiO_2 240 lattice forming Ti-N-Ti linkages, but merely interstitial doping and chemisorbed γ -N₂. Apart from that, the formation of 241 oxygen vacancies to compensate for overall charge balance played an important role towards the narrowing of the catalyst 242 band gap as well. The formation of oxygen vacancies in doped TiO_2 compliments wider light absorptions above 500 nm (45) 243 and is in good agreement with the blue shift of Ti 2p binding energy obtained in Fig. 11. Furtherance, the work done by H. 244 Wu et al. (46) suggests that the formation of oxygen vacancy under nitrogen existence is easier to be formed as compared to 245 naked TiO₂. Despite the remarkable improvement of photonic efficiency, the fate of electron and holes created during catalyst 246 photoexcitation determines the overall photoactivity mechanism. The incorporation of nitrogen into TiO_2 shows broad visible 247 emission around 530 nm, which consequently impacts the electron/hole recombination rate (Fig. 13). At high nitrogen 248 loading, the photoluminescence intensity increases due to increasing number of oxygen vacancies and defect sites. Excessive 249 oxygen vacancies acts as electron/hole recombination centre thus reduced the amount of active radicals generated and give 250 rise to high photolumniscence intensity (29). It is observed that electron/hole recombination is inhibited at low nitrogen 251 loading due to fewer charge trapping sites, hence facilitates better photocatalytic activity. In addition, the PL peak at 450 nm 252 and 500 nm indicates defects presence in the doped TiO₂, and could act as a new adsorption centers (47) in the visible range, 253 as N 2p interstates overlapping with O 2p responsible for vis-active was not observed in this study. Furthermore, the PL 254 peaks shows a broad spectrum indicating the presence of oxygen vacancies, demonstrating not all of the nitrogen doped takes place in forming Ti-O-N or Ti-N-O linkages, but exists as amorphous nitrogen. This observation is supported by no change in the crystal structure as shown in Fig. 6.

257 Photocatalytic Activity

258 Decomposition of Atrazine under Visible Light. Prior to the photocatalytic degradation, the catalyst was left in the dark to 259 reached absorption-desorption equilibrium (Fig. 14). In this study, 70 min was sufficient to reach the absorption-desorption 260 saturation point. The surface area and porosity of the catalyst played an important role in controlling the amount of atrazine 261 absorbed, as well as the surface charge of catalyst and substrate at the present pH. Relatively, higher absorption of atrazine 262 was observed for undoped TiO_2 and highest loading of doped TiO_2 due to smaller particle size and greater porosity. (Table 263 2). However, these properties are not sufficient to justify the strength of a photocatalyst, and many other factors are involved 264 such as morphology, crystal structure, optical and electronic properties (48). In addition, the catalyst surface charge is an 265 important factor and is predominantly controlled by pH. The measured initial pH of atrazine solution is 7.5 and no further 266 adjustment of pH was done throughout the experiment. In this condition, the charge of both catalysts is presumed as 267 negatively charged as the isoelectric point of TiO_2 is from 6.25 to 6.90 (24). The attraction between the negatively charged 268 catalyst and atrazine substrate was favored, as atrazine is protonated by the presence of H⁺, thus facilitate the absorption of 269 atrazine. In return, this provide a good combination for the photocatalytic activity as the generated electron/holes pair on the 270 catalyst surface could readily reduce and oxidize absorbed atrazine into its degradation intermediates. As observed, the 271 presence of nitrogen in TiO₂ significantly improved the photocatalytic degradation activity of atrazine, when compared to 272 undoped TiO₂ (Fig. 15) under visible light irradiation. As shown in Fig. 15, the degradation rate constants are dependent on 273 the amount of nitrogen loaded onto TiO₂, however, it does not represent a mathematical linear function. The function fits first 274 order of reaction with coefficient, $R^2 = 0.98$, whereby the rate of degradation is dependent on the loading of nitrogen (Fig. 275 16). Highest removal of atrazine was achieved less than 150 min at the lowest nitrogen loading in doped TiO₂. Higher 276 nitrogen loading does not necessarily favor the photocatalytic reaction (49)due to the rapid electron/hole recombination as 277 represented in Fig. 13, although absorption of atrazine during darkness was greater. This phenomenon explained the necessity 278 of slow electron/hole recombination as well as enhanced photons absorption for a good photocatalysis process. Subsequently, 279 undoped TiO_2 showed the weakest performance as the absorption of photons in the visible region is very poor, hence less 280 light can be absorbed to generate active radicals for photodegradation.

281 Degradation Mechanisms. The intermediates for the photodegradation of atrazine were studied after 150 min of reaction 282 using 0.5 g/L doped TiO₂. Based on LCMS/MS peaks, a total of 8 degradation intermediates were observed, which enables 283 the plotting of potential degradation pathway (Fig. 17). The obtained m/z at 215.7 showed presence of atrazine even after 150 284 min of photodegradation activity and indicates incomplete removal of atrazine. The degradation pathway involved 285 dehalogenation of chlorine at position two with a hydroxyl group, oxidation of alkyl side chain, further dealkylation and 286 deamination. The formation degradation processes includes the of hvdroxvatrazine. 287 deethylatrazine,desethylhydroxyatrazine,deisopropylatrazine, deethyldeisopropylatrazine, ammelide and ammeline (3,4,6) and 288 are listed in Table 3. The degradation pathway leads to a final amino group displacement with hydroxyl and formed a stable 289 intermediate of cyanuric acid at m/z 129.1. I.K. Konstantinou et al. (5) referred the photodegradation mechanism of s-triazine 290 as photo-Kolbe decarboxylation process, whereby similar pattern formation of alcohol, aldehyde and acid derivatives were 291 observed.

292 Conclusion

293 Undoped and nitrogen doped TiO₂ were synthesized via sol gel technique using titanium (IV) isopropoxide and triethylamine 294 as the source of Ti and N precursors respectively. A single anatase phase with less particles agglomeration was observed for 295 all doped TiO₂. There was no dramatic change in the particle size of doped TiO₂ as the atomic radius of nitrogen atom is 6% 296 larger than oxygen atomic radius. Based on the FTIR and XPS data, the hydrophilicity of TiO₂ surface increased in the 297 presence of nitrogen, and is beneficial to any photocatalytic degradation of water pollutant. Nitrogen is doped interstitially 298 forming Ti-O-N or Ti-N-O linkages, and induces local states 0.23 to 0.26 eV above the valence band which was responsible 299 for the visible light response between 400 to 550 nm and effective narrowing of the band gap. In addition, no significant 300 change in the crystal structure coupled with N1s XPS data and high nitrogen loading confirms the presence of chemisorbed γ -301 N₂ as well. In all prepared photocatalyst, doped TiO₂ with the lowest nitrogen loading yielded the highest rate of atrazine 302 degradation, and superseded the performance of undoped TiO₂. At low nitrogen loading, the number of electron-hole 303 trapping sites are hindered, which drives higher concentration of active radicals for atrazine degradation and mineralization. 304 The degradation of atrazine followed first order rate of reaction, where the degradation activity is greatly influenced by 305 atrazine concentration. The degradation pathway include dehalogenation of chlorine at position two with a hydroxyl group, 306 oxidation of alkyl side chain, further dealkylation and deamination to form a stable non-toxic by product of cyanuric acid.

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Caption of Figures

Fig. 1. Atrazine molecular structure

Fig. 2. FESEM images of a) N0-TiO₂, b) N1-TiO₂, c) N3-TiO₂ and d) N5-TiO₂ taken at 50,000x magnification

Fig. 3. EDX spectrum of N0-TiO₂ and N1-TiO₂

Fig. 4. HR-TEM images of TiO_2 powders for a) N0-TiO₂, b) N1-TiO₂ and c) lattice spacing of anatase phase for N1-TiO₂

Fig. 5. Nitrogen adsorption-desorption linear isotherm (Type IV) for a) N0-TiO₂ and b) N1-TiO₂

Fig. 6. XRD spectra showing the crystal structure of a) N0-TiO₂, b) N1-TiO₂, c) N2-TiO₂, d) N3-TiO₂, e) N4-TiO₂ and f) N5-TiO₂

Fig. 7. Effect of FWHM, crystallite size and band gap alterations at different loading of nitrogen in ${\rm TiO}_2$

Fig. 8. Raman spectra of a) N0-TiO₂, b) N1-TiO₂, c) N2-TiO₂, d) N3-TiO₂, e) N4-TiO₂ and f) N5-TiO₂

Fig. 9. Raman localized spectra of a) N0-TiO₂, b) N1-TiO₂, c) N2-TiO₂, d) N3-TiO₂, e) N4-TiO₂ and f) N5-TiO₂

Fig. 10. FTIR spectra of a) N0-TiO₂, b) N1-TiO₂ and c) enlarged FTIR spectra of Fig. 10b) from 1200 cm⁻¹ to 1700 cm⁻¹

Fig. 11. XPS Spectra of N 1s, Ti 2p and O 1s of N0-TiO₂ and N1-TiO₂

Fig. 12. a)Absorption spectrum and b) localized absorption spectrum for all prepared photocatalyst at different nitrogen loading

Fig. 13. Photoluminescence spectrum for for all prepared photocatalyst at different nitrogen loading

Fig. 14. Absorption-desorption profile of atrazine in darkness for all prepared photocatalyst at different nitrogen loading

Fig. 15. Percentage removal of atrazine and its corresponding rate constant for all prepared photocatalyst at different nitrogen loading

Fig. 16. First order rate of reaction graph of TiO₂ doped with nitrogen (r^2 coefficient > 98%)

Fig. 17. Proposed degradation pathway of atrazine using TiO₂ doped with nitrogen



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Fig. 5 Nitrogen adsorption-desorption linear isotherm (Type IV) for all a) N0-TiO₂ and b) N1-TiO₂





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Fig. 9



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Fig. 16. First order rate of reaction graph of TiO₂ doped with nitrogen (r^2 coefficient > 98%)



Fig. 17. Proposed degradation pathway of a trazine using $\rm TiO_2$ doped with nitrogen HAAct and HHeaAT were absence and not detected by LC/MS/MS

Caption of Tables

- Table 1. Atomic and weight percentage of nitrogen in N-TiO₂
- Table 2. Crystal structure, crystallite size, band gap, surface area and porosity of undoped and N-TiO₂
- Table 3. Atrazine degradation intermediates by LCMS/MS

Table 1

Table 1. Atomic and weight percentage of nitrogen in N-TiO₂

^a Initial N Molar	Sample Marking, NX-	^b Amount of N doped in	^c Amount of N doped	
Concentration	TiO ₂	TiO ₂ (wt %)	in TiO ₂ (wt %)	
0.0	N0	0.00	0.00	
0.5	N1	8.71	9.05	
1.0	N2	10.32	11.03	
1.5	N3	11.29	12.00	
2.0	N4	13.99	13.71	
2.5	N5	16.56	16.99	

^a Source of N from triethylamine precursor
^b Amount of N successfully doped in TiO₂ (EDX analysis)
^c Amount of N successfully doped in TiO₂ (CHNS analysis)

Table 2

Sample	Weight fractions of phase (%)		Average crystallite	d spacing	Band gap	BET surface	Porosity
			size, µm			area	
	Anatase	Rutile		Å	eV	m^2/g	cm ³ /g
N0-TiO ₂	94	6	23.0	3.50	3.20	48.98	0.235
N1-TiO ₂	100	-	34.5	3.50	2.97	30.32	0.221
N2-TiO ₂	100	-	34.5	3.50	2.93	26.61	0.245
N3-TiO ₂	100	-	29.6	3.51	2.95	34.88	0.113
N4-TiO ₂	100	-	29.6	3.51	2.97	38.78	0.134
N5-TiO ₂	100	-	23.0	3.49	2.93	55.02	0.269

Table 2. Crystal phase, crystallite size, band gap, surface area and porosity of undoped and N-TiO₂

Table 3

Mass spectra, m/z ^a	Abbreviation	Compound
215.7	CIET	2-chloro-4-ethylamino-6-isopropylamino-s-triazine (atrazine)
197.3	HEIT	2-hydroxy-4-ethylamino-6-isopropylamino-s-triazine
197.2	HEAcT	2-hydroxy-4-ethylamino-6-acetamido-s-triazine
155.2	HEAT	2-hydroxy-4-ethylamino-6-amino-s-triazine
127.1	HAAT	2-hydroxy-4,6-diamino-s-triazine (ammeline)
169.2	HAIT	2-hydroxy-4-amino-6-isopropylamino-s-triazine
128.1	OOAT	2,6-hydroxy-4-amino-s-triazine (ammelide)
129.1	OOOT	2,4,6-triols-s-triazine (cyanuric acid)

Table 3. Atrazine degradation intermediates by LCMS/MS

^a MS analysis in +ve ion mode