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15 **ABSTRACT**

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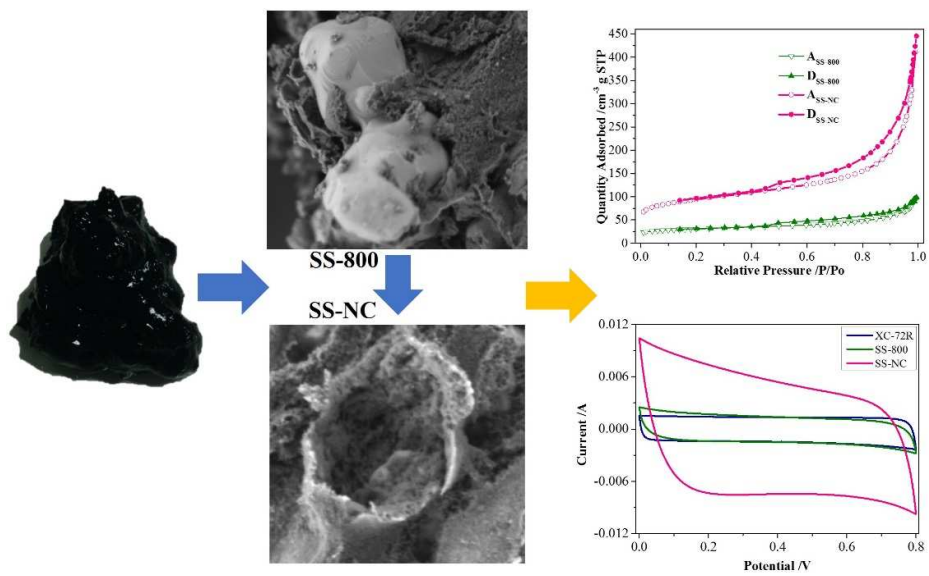
17 With the imposition of more stringent regulations governing the disposal and use of
18 sewage sludge, the need to develop more cost-effective and environmentally benign
19 re-uses of sewage sludge is of particular concern. Pyrolysis converting the sewage
20 sludge to porous carbon is an emerging technology for the disposal of huge amounts
21 of sewage sludge. In this study, a unique heteroatom-doped porous carbon was
22 prepared via the direct pyrolysis of sewage sludge with SiO₂, transition metals and
23 organic matter, the special components of sewage sludge, acted as the in-built
24 template, the graphitizing catalysts, and the ideal precursor and nature dopant,
25 respectively. The as-prepared N, O-doped porous carbon had a high specific surface
26 area, numerous heteroatoms, good electrical transport properties and a
27 meso-/macroporous composite. It exhibited favorable charge storage capacity and
28 good stability over 10000 cycles. The supercapacitor performance results from its
29 hierarchical porous structure and heteroatom doping effects, which combine electrical
30 double-layer capacitors and Faradaic contributions. Our protocol demonstrates a new
31 approach for the potentially eco-friendly benign re-use of sewage sludge and provides
32 a proven technique for synthesizing electrode materials as promising candidates for
33 electrochemical energy storage.

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TOC Art

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39 1. Introduction

40

41 Sewage sludge, which is the residue generated from the activated sludge method,
42 consists of organic material, mainly dead bacterial cells and organic pollutants, and
43 inorganic components in the form of silica, iron salts, calcium oxide, alumina,
44 magnesium oxide and a wide variety of transition metals (e.g., Cr, Co and Ni). The
45 European Environmental Agency defines it as “a future waste problem”. [1, 2] With
46 the increasing generation of sewage sludge around the world, the disposal of sewage
47 sludge has remained a thorny issue to date.[3] Recently, this issue has been further
48 aggravated by the imposition of more stringent regulations governing the disposal of
49 sewage sludge. Traditional options for sludge disposal such as combustion, landfilling
50 or ocean dumping are no longer acceptable.[4, 5] In this situation, the initiation of
51 more cost-effective, environmentally benign re-use of sewage sludge is certainly
52 necessary from an environmental standpoint.[6, 7]

53 Sludge pyrolysis, different from sludge combustion, is a proven, innovative
54 technology that can convert approximately half of the organic matter in sewage sludge
55 into useful bioenergy (oil and gas). This process can also immobilize the rest of the
56 organic and inorganic matter into a stabilized form of pyrolytic residue (biochar).[8, 9]
57 Furthermore, sludge pyrolysis is often carried out under inert atmospheric conditions,
58 which precludes the production of highly toxic dioxin-like compounds and particulate
59 matter (e.g., PM_{2.5}). The gases and oils produced in the sludge pyrolysis process can
60 be used as renewable liquid fuels and chemical feedstocks.[10, 11] However, no

61 value-added utilization has been found for biochar, which is a carbonaceous matrix
62 byproduct that is environmentally resistant.

63 Supercapacitors are energy storage devices that accumulate energy in the form of
64 electrical charge and bridge the gap between dielectric capacitors and batteries. These
65 devices are attracting considerable attention due to their high specific power, short
66 charging time and long cycle life.[12, 13] Porous carbonaceous materials have been
67 found to be promising candidates for electrical double-layer capacitors (EDLCs)
68 because they possess a large specific area, a more favorable cycle durability and a
69 high level of electrical conductivity.[14, 15] Recently, the synthesis of porous
70 carbonaceous materials derived from waste and biomass for making energy storage
71 materials has attracted considerable attention. For instance, heteroatom-doped porous
72 carbon flakes, porous carbon materials and hierarchically porous carbon nanosheets,
73 which were prepared via carbonization of human hair fibers, broad beans and waste
74 coffee grounds respectively,[16-18] were used for supercapacitor electrode materials,
75 and exhibited high specific capacitance. Compared with conventional carbon
76 precursors (e.g., wood, coal, pitch or nutshell), the cost-effectiveness and
77 environmental friendliness of waste and biomass-derived porous carbon materials
78 make them more suitable for large-scale production and for various practical
79 applications.[16-18] Although biochar produced from sludge pyrolysis holds potential
80 for carbon sequestration, the use of sewage sludge to prepare carbon nanomaterials
81 for EDLCs has not yet been achieved.

82 In this study, for the first time, we demonstrate the synthesis of

83 heteroatom-doped carbon materials for EDLCs via the direct pyrolysis of the
84 “all-in-one” precursor sewage sludge. During this pyrolysis process, SiO₂, a special
85 component of sewage sludge, acts as an in-built template that prevents agglomeration
86 and results in the formation of a unique pore size distribution. The organic matter in
87 sewage sludge exhibited as the structure-directing templates through carbonization
88 and graphitizing under the catalytic action of the transition metals. These organic
89 matter abundant in nitrogen and oxygen also serve as the source of heteroatom dopant,
90 which leads to a significant enhancement in the charge storage capacity. The
91 as-synthesized heteroatom-doped carbon material has a high specific surface area,
92 numerous heteroatoms, good electrical transport properties and a meso-/macroporous
93 composite. The electrochemical results demonstrate its favorable charge storage
94 capacity and good stability. The results of our study provide a novel route for the
95 synthesis of heteroatom-doped carbon materials for use in energy storage devices by
96 using a simple, low-cost, green process. This study therefore demonstrates a new
97 approach for the value-added re-use of sewage sludge.

98

99 **2. Experimental section**

100

101 *2.1. Synthesis of the Sewage Sludge-derived Carbon Material.* As the raw material for
102 this process, a dewatered sewage sludge sample was obtained from the Anting
103 wastewater treatment plant in Shanghai, China.[7] The obtained sludge was stored at
104 -20°C before use. All other reagents were of analytical grade and were used as

105 received unless otherwise stated. All of the water used was prepared with a
106 purification system (Hitech Instrument Co., Shanghai, China).

107 The sewage sludge-derived heteroatom-doped carbon material was synthesized
108 by a facile one-step pyrolysis process in a quartz tubular reactor (Fig. S1). In brief, the
109 lyophilized sewage sludge was first placed in the quartz tubular reactor under 100
110 ml/min of N₂ flow for 30 min to remove air. Then the quartz tubular reactor was
111 heated to 800 °C at a heating rate of 5 °C/min under a 50 ml/min of N₂ flow, and held
112 at that temperature for 2 h. After being cooled to ambient temperature in the N₂ flow,
113 a resultant black powder was obtained and designated as SS-800. The SS-800 was
114 immersed in 20 wt% HF to etch-out the SiO₂, then washed until the pH reached 7, and
115 dried at 100°C. The product of this process was the sewage sludge-derived carbon
116 nanomaterial (SS-NC).

117 *2.2. Characterization.* Scanning electron microscopy (SEM; FEI Nova-Nano,
118 The Netherlands) was used to image the morphology of the as-synthesized sewage
119 sludge-derived nanomaterials. The analysis of the surface area, pore diameter and
120 volume was carried out by using the Brunauer-Emmett-Teller (BET) method on an
121 AUTOSORB-IQ instrument (Quantachrome Co., USA). The x-ray diffraction (XRD)
122 patterns were measured with an X' Pert PRO system (Philips Co., The Netherlands) to
123 characterize the crystal structure of the nanomaterials. The functional groups of the
124 as-prepared catalysts were determined by Fourier transform infrared (FTIR) spectra
125 recorded with a VERTEX 70 FT-IR (Bruker Co., Germany) and by Raman analysis
126 with a laser Raman spectrometer (ProTT-RZRaman-B2, Enwave Optronics Inc.,

127 USA). The electronic environment of the nanomaterials was investigated by using
128 X-ray photoelectron spectroscopy (XPS, PHI-5000C, Perkin Elmer Co., USA). The
129 composition of the sewage sludge-derived carbon nanomaterials was measured via
130 energy-dispersive X-ray spectroscopy (EDX) and inductively coupled plasma
131 spectrometry (ICP, Agilent 720ES, USA) after total digestion in a microwave using a
132 mixture of $\text{HNO}_3 + \text{HCl} + \text{HF}$. The C, O and N contents of the materials were also
133 determined by a Vario EL III Elemental Analyzer (vario EL III, GmbH, Germany).

134 *2.3. Electrochemical Characterization.* All of the electrochemical tests were
135 carried out on a CHI760E electrochemical workstation (Shanghai Chenhua
136 Instruments Co., China) at room temperature in a conventional three-electrode system.
137 An aqueous solution of 0.5 M Na_2SO_4 was used as the electrolyte, a platinum wire
138 was used as the counter electrode and an Ag/AgCl electrode served as the reference
139 electrode. To prepare the work electrode, a slurry containing 80 wt% active material
140 (e.g., SS-800, SS-NC), 10 wt% carbon black (Vulcan XC-72R, surface area $254 \text{ m}^2/\text{g}$,
141 Cabot Co., USA) and 10 wt% polytetrafluoroethylene was mixed and loaded on a
142 nickel foam substrate. After being dried in an oven at 353 K for several hours, the
143 as-prepared work electrode was pressed at 15 MPa to assure good electrical contact
144 between the nickel foam substrate and the active material. The electrode was then
145 further dried in an oven at 373 K for several hours. The total mass of the active
146 materials on the nickel foam substrate was about 2~3 mg per electrode with a surface
147 area of 1.0 cm^2 .

148 Cyclic voltammetry (CV) curves were obtained in the potential range of 0-0.8 V
149 vs. Ag/AgCl by varying the scan rate from 5 to 100 mV/s. Electrochemical impedance
150 spectroscopy (EIS) was measured in a frequency range of 10^5 Hz to 0.01 Hz at open
151 circuit voltage with an alternate current amplitude of 5 mV. The EIS results were
152 fitted by ZView software according to the corresponding equivalent circuits.
153 Galvanostatic charge/discharge was carried out at 0.5-8 A/g over a voltage range of
154 0-0.8 V vs. Ag/AgCl, and the specific capacitance was calculated from the
155 galvanostatic discharge curves according to the following equation (Eq. 1):

$$C = It/m\Delta V \quad (1)$$

156 where C (F/g) is the discharge specific capacitance, I is the discharge current, Δt is the
157 discharge time consumed in the potential range of ΔV and m (g) is the weight of the
158 active materials loaded on the work electrode.

159

160 **3. Results and discussion**

161

162 *3.1. Textural Properties of the As-synthesized Nanomaterials.* The surface morphology
163 and structure of the as-synthesized sewage sludge-derived nanomaterials were
164 examined using SEM, and the typical images are shown in Figure 1. Numerous
165 particles with diameters ranging from about 20 nm to 2 μ m were identified by EDX as
166 being composed of SiO_2 , a special component of sewage sludge. These particles,
167 inlaid into the nanomaterials derived from the sewage sludge (Figs. 1A and B), acted
168 as the in-built template during the synthesized process. This template prevented

169 agglomeration and resulted in the formation of mesoporous structures combined with
170 graphitization or evaporated of the organic matters in sewage sludge during the
171 pyrolysis process (Figs. 1A and B). The highly porous structures around the SiO₂
172 particles (Fig. 1B) were formed by the partial graphitization of the organic matter as
173 the carbon precursor.[19] After being immersed in HF, all of these SiO₂ particles
174 vanished. Also, the rough mesoporous structures displayed accumulations of
175 numerous particles with nanometer-sized diameters that increased the surface area of
176 the catalyst (Figs. 1C and D). This accumulation was made further evident by the
177 increased BET surface area of the as-synthesized nanomaterial SS-NC compared with
178 that of the SS-800 (Table 1).

179 The surface area and pore structure of the as-synthesized sewage sludge-derived
180 nanomaterials were analyzed by the nitrogen adsorption-desorption method. The
181 representative results for SS-800 and SS-NC (Fig. 2A) showed characteristic type IV
182 curves (according to the IUPAC classification) with sharp capillary condensation
183 steps that are typical features of mesoporous solids.[20] The isotherms of the SS-800
184 and SS-NC all had a clear upward trend at a low relative pressure ($P/P_o < 0.4$),
185 suggesting that the as-synthesized nanomaterials were rich in micropores. The
186 appearance of a sharp upward-type H3 hysteresis loop at $P/P_o > 0.4$ implied the
187 presence of abundant mesopores in the nanomaterials[16, 21]. At high relative
188 pressures before 1.0, the curve exhibited a slight upward tendency, which could be
189 ascribed to the internal macropores that represented the accumulation of particles
190 present in the nanomaterials.

191 The pore size distributions further suggested the intrinsic hierarchical porous
192 structure of the as-synthesized nanomaterials (Fig. 2B). The peaks observed at 4 nm
193 could be interpreted as mesopores generated by the graphitization or evaporation of
194 organic matters in the sewage sludge. Compared with the SS-800, the increased broad
195 peak observed between 5 and 100 nm for the SS-NC may have corresponded to the
196 mesopores and macropores generated by the removal of SiO₂. The BET surface areas
197 of the nanomaterials significantly increased from 107.44 to 331.39 m²/g with the
198 etch-out of the in-built SiO₂ template. This increase was accompanied with an
199 expansion of the average pore size from 8.06 to 11.88 nm due to the generated
200 mesopores and macropores. This higher surface area of the SS-NC is advantageous
201 for charge storage.[16, 21]

202 *3.2. Structural Characterization of the As-synthesized Nanomaterials.* XRD was
203 used to investigate the phase structure of the as-synthesized sewage sludge-derived
204 nanomaterials (Fig. 2C). Obvious diffraction peaks at $2\theta = 20.9^\circ$ and 26.7° were
205 observed for the SS-800. These peaks corresponded to the typical crystallite structures
206 of SiO₂ (JCPDS, File No. 33-1161) that originate from sewage sludge. However, such
207 peaks disappeared in the SS-NC, indicating the total removal of the hard template.
208 This development was further confirmed by the disappearance of the characteristic
209 asymmetric stretching vibrations of the Si-O-Si (1036 cm⁻¹) peak in the FTIR (Fig.
210 S2A) and the elemental composition of the nanomaterials (Table 1). The HF washing
211 process could remove most of the unwanted elements. No Si content was detected for

212 the SS-NC, and this disappearance of Si was accompanied by an increase in the C
213 content from 31.13% to 51.14%.

214 The amount of Fe content in the sewage sludge and in the as-synthesized
215 nanomaterials was significant, while the content of other heavy metals was relatively
216 low (Fig. S2 and Table 1). The presence of heavy metals such as Fe and Cr would
217 tend to improve the degree of graphitization.[19] A well-developed peak at 22.3° and
218 a weak peak at 43.8° , which corresponded to the (002) and (100) spacing of the
219 graphene stacks, respectively,[16] could be clearly observed. The pattern of the
220 SS-NC also showed several peaks at 15.4° , 29.5° , 30.9° , 39.2° , 47.1° and 51.5° , which
221 could be attributed to the pentlandite (JCPDS, File No. 08-0090) (Fig. 2C). This
222 pattern suggested the existence of iron sulphide, which is also a type of
223 energy-storage material and may have contributed to the excellent electrochemical
224 performance.[22] It should be noted that there was still some N and O content
225 detected (Table 1), which would suggest that the as-synthesized nanomaterials were
226 N- and O-doped.

227 FTIR and Raman spectra were used to characterize the structures of the
228 as-synthesized nanomaterial surfaces (Fig. S3). Comparing the FTIR spectra of
229 SS-800 and SS-NC, it could be clearly seen that the characteristic peaks of carbon
230 nanomaterial ranging from 1250 to 1750 cm^{-1} appeared on both curves. These peaks
231 indicated that the framework structures and functional groups were analogous to each
232 other (Fig. S2A).[23] The two peaks in the Raman spectra at around 1335 and 1532
233 cm^{-1} were the characteristic D and G bands of carbon, which reflected the degree of

234 disorder and the level of graphitization in the carbon materials, respectively (Fig.
235 S2B).[16, 18, 23] These measurements suggested that the obvious G peak, which
236 accorded with the XRD results, was formed by the partial graphitization of the
237 organic materials as carbon precursors under the catalytic action of the critical toxic
238 heavy metals (e.g., Fe, Cr, Co and Ni) that were uniformly distributed in the sewage
239 sludge. The relatively higher D peak implied defects from the ideal graphitic lattice,
240 which might have been caused by the heteroatom doping effects. The significantly
241 decreased numbers of peaks in the SS-NC compared with those in the SS-800
242 indicated the removal of impurities, e.g., inorganic matter, accompanied with the
243 etching-out of the in-built SiO₂ template. This process would tend to generate
244 considerable quantities of porous structures, increase the surface areas of the
245 as-synthesized nanomaterials and expose the functional groups in the carbon wall to
246 the pore surfaces.

247 XPS analysis was used to further analyze the content and chemical state of the
248 elements in the as-synthesized nanomaterials (Fig. 3A). The binding energies were
249 calibrated with respect to the C 1s peak at 284.6 eV. Obvious C 1s and O 1s peaks and
250 fine N 1s, Fe 2p and S 2p peaks were all observed as expected. Compared with
251 SS-800, the intensity ratio of Si 2p vanished, demonstrating the etch-out of the
252 in-built SiO₂ template in accordance with the above-described results. As the
253 positions of the elemental peaks depended on the local chemical environment, high
254 resolution scans of C, O and N were performed and deconvoluted to obtain the
255 corresponding atom binding states by searching for the optimal combination of

256 Gaussian bands (Figs. 3B, C and D). In the case of the C1s XPS spectrum of SS-NC,
257 the four peaks at 282.8, 284.4, 286.0 and 288.4 eV were attributable to the C-O, C=C,
258 C-C/N and C=O species, respectively (Fig. 3B).[24] The three components of the O1s
259 XPS spectrum, with their peaks at binding energies of about 530.7, 532.1 and 534.6
260 eV, corresponded to the C=O, C-O and O=C-O species, respectively (Fig. 3C). These
261 components indicated the existence of some oxygen-enriched functionalities in the
262 carbon frameworks. Such oxygen-enriched functionalities have been reported to
263 improve wettability and to produce pseudocapacitance, resulting in better
264 supercapacitive performance. [24]

265 Sewage sludge contains essential proteins that are abundant in nitrogen. The
266 high-resolution N 1s peaks of SS-NC fit into four types of nitrogen-containing groups
267 at about 398.0, 399.7, 400.8 and 402.5 eV (Fig. 3D). The two peaks located at about
268 398.0 and 399.7 eV were attributed to pyridinic N and pyrrolic N species, which were
269 assumed to have contributed to the pseudocapacitance. The peak located at 400.8 eV
270 was attributed to quaternary N, which could enhance the electrical conductivity of
271 SS-NC.[25, 26] These results further indicated that N atoms had been successfully
272 doped into the carbonaceous nanomaterials. The introduction of nitrogen into the
273 carbonaceous nanomaterials may have imposed nitrogen-related Faradaic reactions in
274 the electrodes and changed the electron distribution of the materials, thus increasing
275 their supercapacitor performance and enhancing the wettability between the electrode
276 materials and electrolytes.

277 3.3. *Electrochemical Performance of the As-synthesized Nanomaterials as*
278 *Supercapacitor Electrodes.* Electrochemical tests were performed to explore the
279 potential applications of the as-synthesized nanomaterials as electrode materials for
280 EDLCs. The CV curves of XC-72R, SS-800 and SS-NC in 0.5 M Na₂SO₄ aqueous
281 solution at a scan rate of 50 mV/s are shown in Fig. 4A. The control XC-72R
282 exhibited regular rectangular-shaped CV curves, indicating an ideal EDLC nature of
283 the charge-discharge process. The approximately rectangular-shaped and symmetric
284 CV curves of the SS-800 and SS-NC also indicated a dominant contribution of the
285 capacitance from the EDLC. The gravimetric capacitance of the control XC-72R (as
286 calculated from its CV curve with a potential scan rate of 50 mV/s) was 21.29 F/g,
287 which was very close to the values reported in the literature[27] and the 22.03 F/g
288 exhibit by the SS-800. The SS-NC exhibited a much higher specific capacitance of
289 120.98 F/g. The significant increase of capacitance was mainly attributed to the
290 etch-out of the in-built SiO₂ template, which would not only increase the specific
291 surface area of the nanomaterials but also increase the content of C and the
292 heteroatoms, thereby promoting conductivity and wettability (Table 1).[25, 26] A
293 Faradaic hump was observed at ~0.10 V in the CV curve at low scan rate of 5 mV/s
294 (Fig. 4B), which was probably due to redox reactions of the doped heteroatoms such
295 as pyridinic N and pyrrolic N species, the oxygenated functional groups on the carbon
296 frameworks, or the remaining Fe compounds.[25, 26] As the scan rate increased, the
297 plateau current increased accordingly. The specific capacitance of the SS-NC was up
298 to 178.32 F/g at a potential scan rate of 5 mV/s, which was comparable to the

299 capacitance of many other EDLCs reported previously.[16-18] The quasi-rectangular
300 shape of the CV curve could still be maintained with little distortion even when the
301 scan rate rose to 100 mV/s (Fig. 4B), which indicated that rapid ion transport and
302 good rate capability could operate in the SS-NC. The distortion of the CV curve
303 reflected the universal characteristic that ion diffusion and transport on the EDLC
304 electrode surface are restricted at high scan rates.

305 Galvanostatic charge/discharge experiments were carried out at different current
306 densities to further explore the capacitance performance and estimate the specific
307 capacitance of the SS-NC. The appearance of shapes that were almost isosceles
308 triangles demonstrated the ideal charge and discharge characteristics for EDLCs with
309 low dynamic voltage drops and almost 100% Coulombic efficiency over a large
310 voltage range.[18, 26] The insignificant dynamic voltage drop indicated a low internal
311 series resistance, which could also be demonstrated by the EIS (Fig. 5A). The SS-NC
312 had a specific capacitance of 109.73 F/g at a current density of 0.5 A/g (Fig. 4D),
313 which was marginally higher than the previous reports for 100% carbon nanotube
314 (CNT) film (24 F/g) or pristine CNT paper (32 F/g).[18] The 55.26% decrease of the
315 capacitance with a current density increase from 0.5 to 8.0 A/g was attributed to the
316 insufficient electrolyte ion diffusion kinetics at higher operating current densities.
317 These diffusion kinetics would reduce the amounts of electrolyte ions accumulated
318 onto the electrode interfaces and result in the decrement of specific capacitance.[18,
319 24]

320 Nyquist plots of the as-synthesized nanomaterials were measured over the
321 frequency range from 10^5 Hz to 0.01 Hz and simulated by ZView software (Fig. 5A).
322 Unlike the SS-800 plot, the plot of SS-NC featured a vertical line in the
323 low-frequency region, which indicated ideal capacitive behavior. This plot was
324 composed of three distinct parts at different frequency ranges.[16, 18, 24] The real
325 impedance (Z'), which was found at very high frequencies with the imaginary
326 impedance near to zero, was the sum of the ohmic resistances (R_s) accounting for the
327 electrolyte and the electrode. The R_s of the SS-800 (4.47 Ω) was much higher than
328 that of the SS-NC (3.53 Ω), which confirmed that the promoted conductivity was due
329 to the increased content of C and the heteroatoms. The semicircular loop observed at
330 high frequencies represented the composite interfacial impedance and the expected
331 pseudocapacitance impedance that were generated from the Faradaic redox reactions
332 of the electrode material due to the content of N and O species in the carbon
333 frameworks.[16, 18] The inclined portion of the curve (of about 45°) at the middle
334 frequency (which ranged from ~ 175.8 to ~ 3.8 Hz) was ascribed to the Warburg
335 impedance, which was the level of encountered impedance to the diffusion of ions
336 from the electrolyte to the electrode surface.[16, 24] The short Warburg resistance
337 section indicated the efficient access of the electrolyte ions to the porous carbon
338 framework of the SS-NC electrode with its hierarchical pore structure. The almost
339 vertical line of the Nyquist plot represented the dominance of ideal double-layer
340 charge/discharge behavior by the as-synthesized SS-NC at low frequencies.[18, 24]
341 Furthermore, the as-synthesized SS-NC exhibited a superior capacitance retention

342 performance. A greater than 99% retention of the initial capacitance of SS-NC was
343 observed after 10000 charge/discharge cycles at a current density of 1 A/g, which
344 indicated its excellent cycling stability (Fig. 5B).

345 *3.4. Significance of This Work.* We synthesized a unique heteroatom (N, O)
346 doped porous carbon nanomaterial, SS-NC, via direct pyrolysis of the “all-in-one”
347 precursor sewage sludge. This material exhibited favorable charge storage capacity
348 with excellent stability and durability. The unique qualities of this carbon
349 nanomaterial and its favorable electrochemical performance as a supercapacitor stem
350 from the particular compositions of sewage sludge and the proper utilization of almost
351 all of the content of the sludge in synthesizing this carbon material. The organic
352 matter in sewage sludge, which is mainly composed of carbon, hydrogen, oxygen and
353 nitrogen, worked as an ideal precursor and nature dopant for the synthesis of
354 heteroatom (N, O) doped porous carbon nanomaterials. The doping of nonmetal
355 heteroatoms in the sp^2 carbon framework not only improved the electrical
356 conductivity and the wettability between the electrodes and electrolytes, but also
357 induced additional pseudocapacitance via reversible redox reactions, consequently
358 improving the capacitive performance of the SS-NC as the electrode.[16, 18, 24]

359 The SiO_2 content in sludge was used as the in-built template. The critical toxic
360 heavy metals (e.g., Fe, Cr, Co and Ni) that are uniformly distributed in sewage sludge
361 with different phases were used as the “in-built catalysts” to catalyze the partial
362 graphitization of the numerous special components of organic matter as carbon
363 precursors during the pyrolyzation process.[19] The intrinsic hierarchical pore

364 structures of the SS-NC were formed due to the combined effects of the in-built SiO₂
365 template that prevented agglomeration and the uniformly in-built heavy metal
366 catalysts that produced graphitized areas with a three-dimensional stacking order,
367 resulting in transition-metal-containing carbon materials with essentially highly
368 porous during the pyrolyzation process.[19, 28] These hierarchical pore structures
369 with their unique designs could offer abundant mesopore and macropore structures
370 that could improve the migration and diffusion of the electrolyte ions, resulting in a
371 decreased diffusion distance and high power performance.[29] The micropores within
372 the walls of the mesopores and macropores could supply a highly effective specific
373 surface area for double-layer capacitance to obtain a high specific capacitance and a
374 favorable electrochemical energy storage performance.

375 The generation of sewage sludge around the world is continually increasing and
376 now exceeds annual totals of 30 million tons in China, 10 million dry tons in the EU
377 and 5.6-7 million dry tons in the US, respectively.[2, 5, 7] In this situation, the need to
378 develop a more cost-effective and environmentally benign value-added re-use of
379 sewage sludge is of particular concern. Our protocol uses the proven pyrolysis
380 technique to convert sewage sludge into a unique porous carbon nanomaterial for the
381 sustainable development of low-cost energy storage devices. This simple and
382 easy-to-handle process is suitable for large-scale industrial production. We suggest
383 that our approach deserves particular attention not only as an eco-friendly and
384 value-added way of re-using sewage sludge, but also as a means of easy fabrication of
385 a low-cost heteroatom-doped nanocarbon composite with a favorable charge storage

386 capacity. The sludge source and property, especially the C content, do has important
387 influence on this preparation. Thus dewatered sewage sludge from domestic
388 wastewater treatment plant is more approach.

389 Although the specific capacitance of the as-synthesis SS-NC is intermediate in
390 all of the reported carbon materials,[16-18, 30] the excellent stability and durability of
391 these materials makes them promising candidates for energy storage devices in many
392 electronics products. Further studies are needed to investigate possible methods of
393 chemical activation (KOH activation and carbonized in NH_3 flow for instance),[16]
394 which are expected to enable additional increases in the specific capacitance of the
395 sewage sludge-derived carbon material by increase its specific surface areas and the N
396 content.

397

398 **4. Conclusions**

399

400 In summary, a unique heteroatom (N, O) doped porous carbon nanomaterial, SS-NC,
401 was synthesized via the direct pyrolysis of sewage sludge, with SiO_2 used as the
402 in-built template, transition metals, the critical toxic components of sewage sludge,
403 used as the graphitizing catalysts, and organic matter, which is mainly composed of
404 carbon, hydrogen, oxygen and nitrogen, worked as an ideal precursor and nature
405 dopant. The SS-NC exhibited favorable charge storage capacity, with a specific
406 capacitance of 109.73 F/g in 0.5 M Na_2SO_4 at a current density of 0.5 A/g, and
407 excellent stability and durability over 10000 charge/discharge cycles. This high

408 supercapacitor performance can be attributed to the hierarchical porous structure,
409 which provides a highly effective specific surface area for double-layer capacitance,
410 and to the heteroatom doping effects that induce additional pseudocapacitance via a
411 reversible redox reaction. Our protocol shows attractive prospects for establishing an
412 eco-friendly, value-added re-use of sewage sludge. At the same time, our protocol
413 demonstrates a proven technique for synthesizing alternative electrode materials for
414 electrochemical energy storage.

415

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420

421 **References**

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515

516 Table 1. Properties of the as-synthesized sewage sludge-derived catalysts

	SS-800	SS-NC
S_{BET} (m ² /g)	107.44	331.39
average pore size (nm) ^a	8.06	11.88
conc. (wt %)		
C ^b	31.13	51.14
N ^b	0.69	1.28
O ^c	22.40	18.46
Fe ^d	11.87	4.15
Si ^d	6.67	-
Al ^d	0.02	0.03
Mg ^d	0.06	0.40
Ca ^d	0.59	0.87
Cr ^d	0.03	0.02
Mn ^d	0.12	0.03
Ni ^d	0.08	0.04
Cu ^d	0.11	0.11

517 ^a: Calculated from the Barrett-Joyner-Halenda equation using the desorption isotherm.518 ^b: The contents of C and N were obtained by Elemental Analyze.519 ^c: The content of O was obtained by EDX.520 ^d: The contents of Si and metals were obtained by ICP.

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Figure Legends

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523

524 **Figure 1.** Textural properties of the as-synthesized catalysts. SEM images of the
525 as-prepared SS-800 (A, B) and SS-NC (C, D).

526

527 **Figure 2.** N₂ adsorption-desorption isotherms (A), pore size distributions (B) (A
528 represents absorption and D represents desorption) and (C) XRD spectrum (G
529 represents graphene and P represents pentlandite) of the SS-800 and SS-NC.

530

531 **Figure 3.** X-ray photoelectron spectra of the as-synthesized SS-800 and SS-NC (A).
532 The high resolution C1s, O1s and N1s XPS spectra of the SS-NC are shown in (B), (C)
533 and (D), respectively.

534

535 **Figure 4.** Electrochemical performance of the as-synthesized nanomaterials as
536 supercapacitor electrodes. (A) CV measurements of XC-72R, SS-800 and SS-NC in
537 0.5 M Na₂SO₄ aqueous solution over a potential range from 0 to 0.8 V at a scan rate
538 of 50 mv/s. (B) CV measurements of SS-NC at different scan rates. (C)
539 Charge-discharge curves of SS-NC at different current densities. (D) Specific
540 capacitances of SS-NC at different current densities.

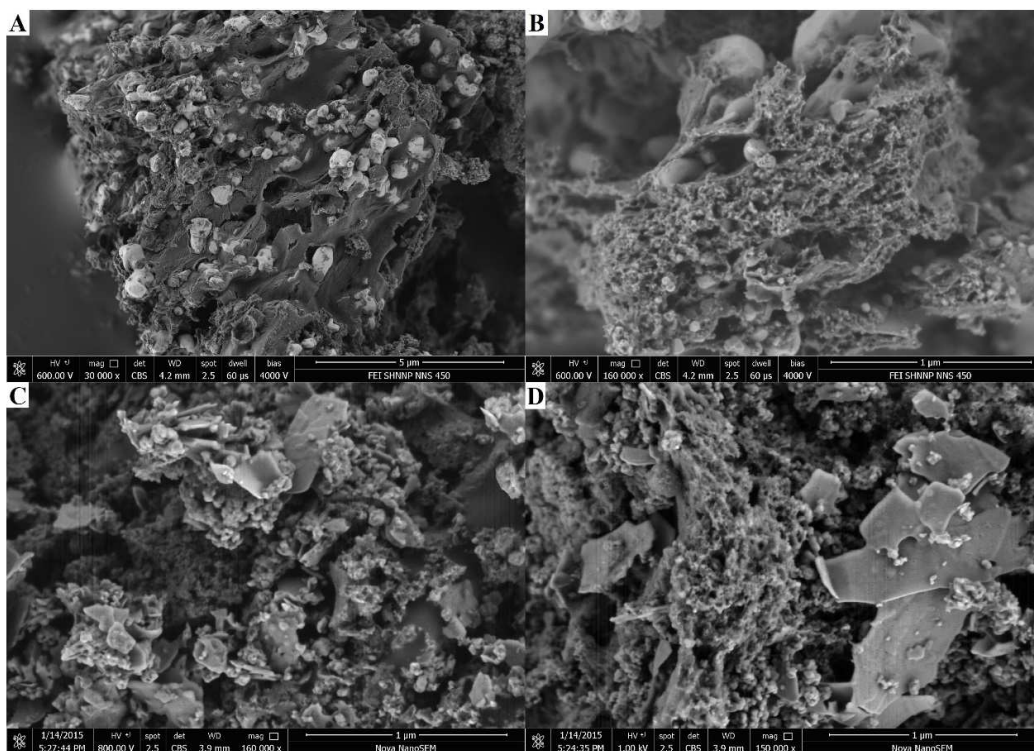
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542 **Figure 5.** (A) Partial enlargement of the Nyquist plot in the high-frequency region
543 with its entire range shown in the inset. (B) Long-term cycling performance of the

544 SS-NC during 10000 charge/discharge cycles at a current density of 1 A/g.

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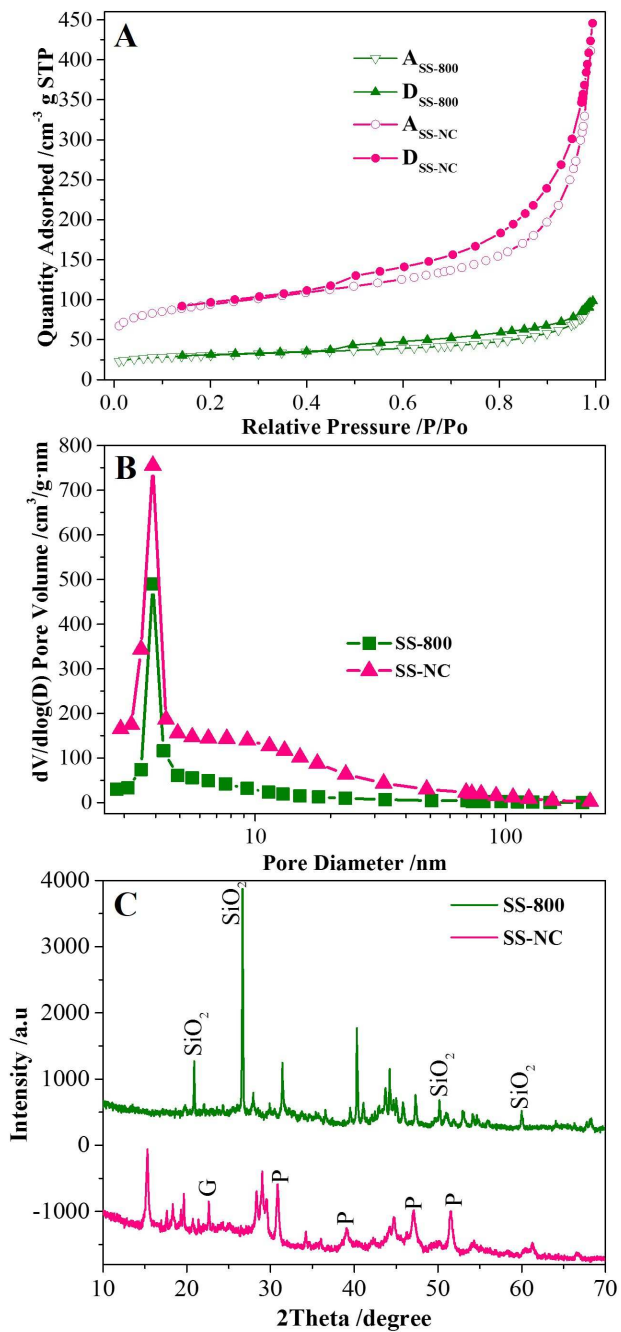
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Figure 1.

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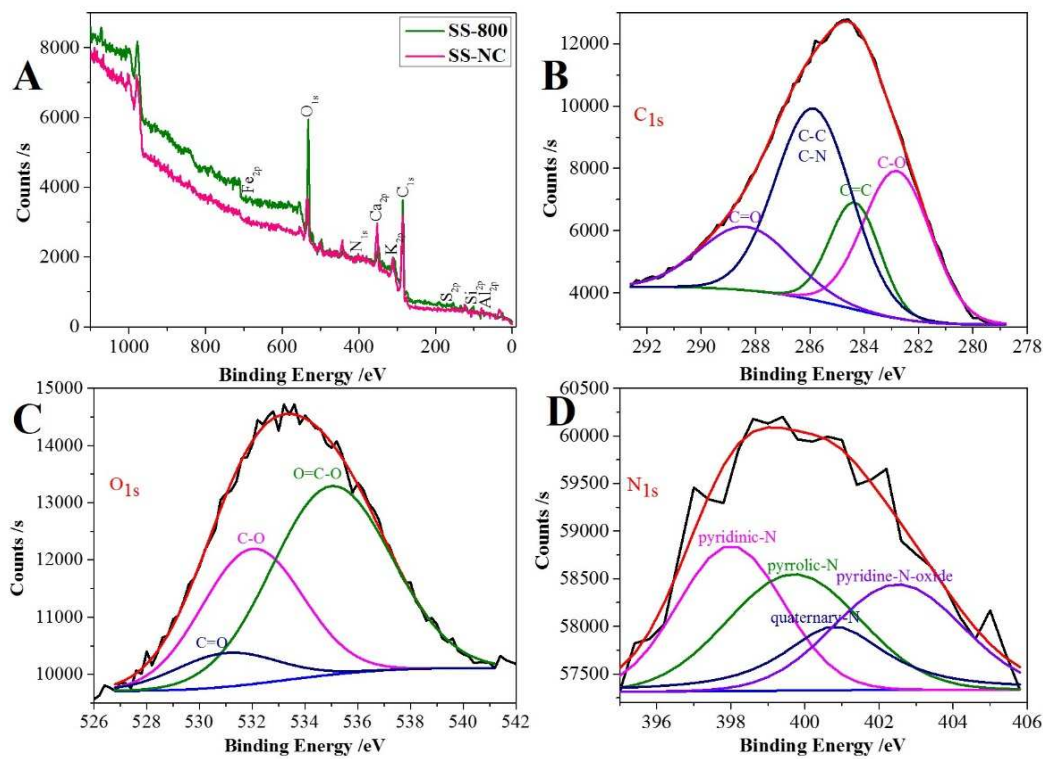
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Figure 2.

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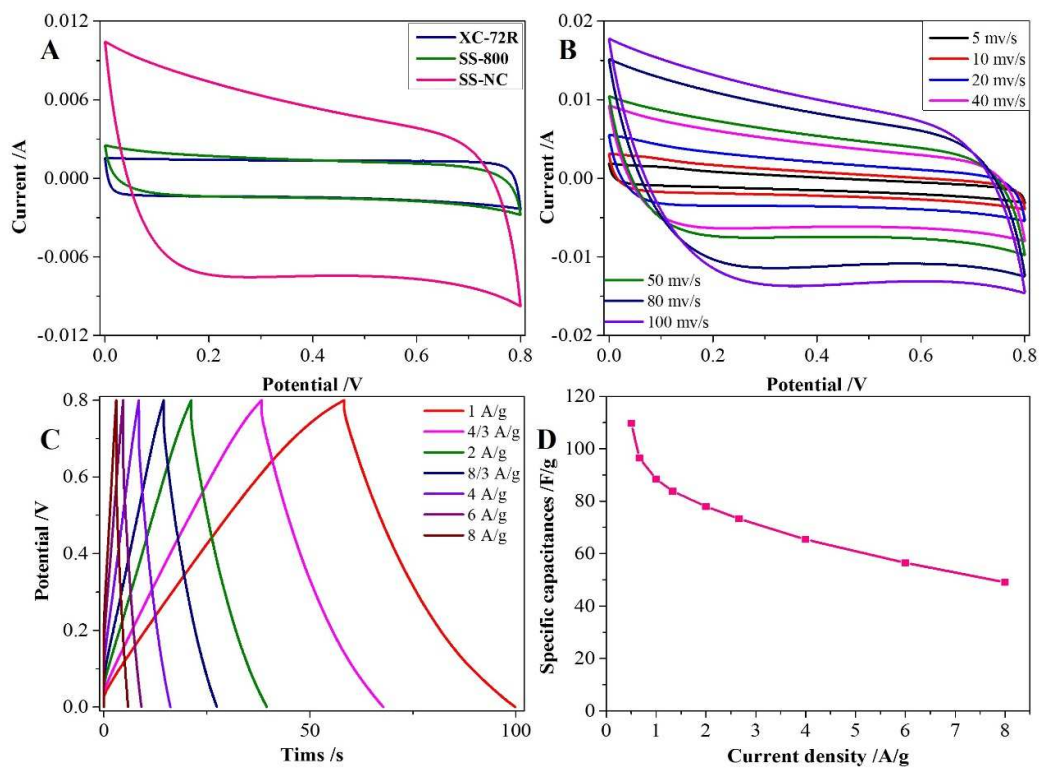
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Figure 3.

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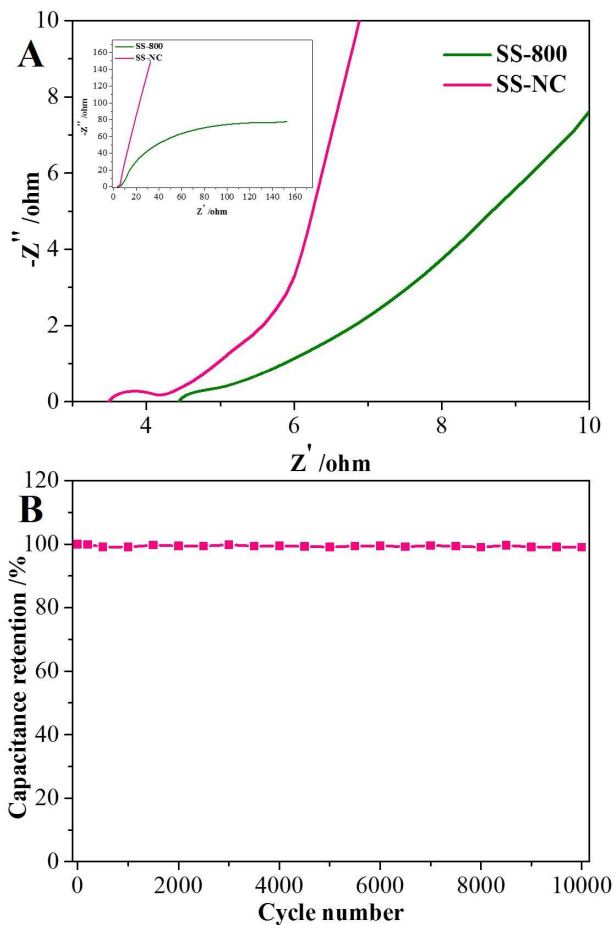
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Figure 4.

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Figure 5.

Graphical contents entry

A unique heteroatom (N, O) doped porous carbon nanomaterial with favorable charge storage capacity and excellent stability and durability was synthesized via direct pyrolysis of the “all-in-one” precursor sewage sludge.

