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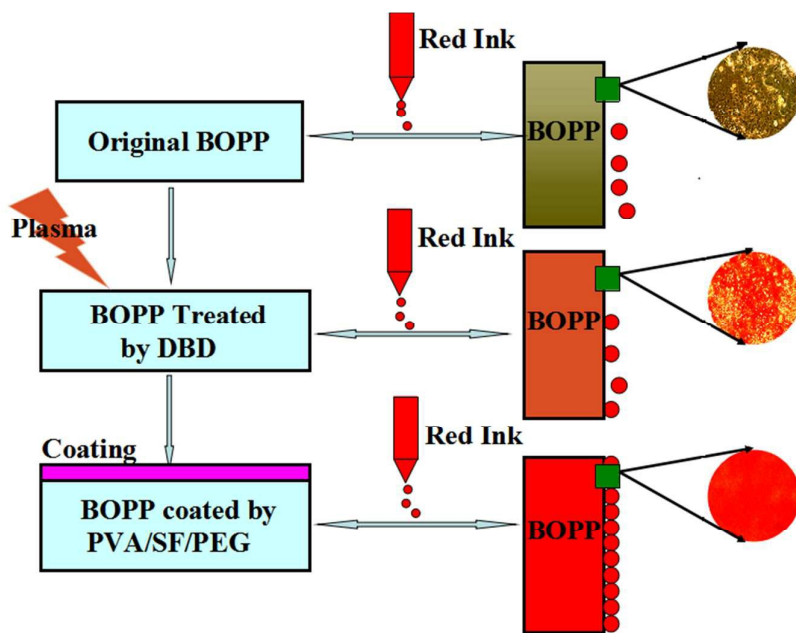
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Graphical Abstract



1 **Development of a novel protocol for the permanent hydrophilic**
2 **modification of a BOPP film for high quality printing with**
3 **water-based ink**

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13 **Abstract**

14 This paper reports the successful modification of biaxially oriented
15 polypropylene (BOPP) films to permanently enhance their hydrophilic properties for
16 potential flexible packing applications. This protocol consists of three sequential
17 processes: 1) an on-line dielectric barrier discharge (DBD) plasma pretreatment, 2) a
18 polyvinyl alcohol/silk fibroin/polyethylene glycol (PVA/SF/PEG) coating, and 3)
19 ethanol solution finishing. The optimal modification conditions included: DBD
20 plasma pretreatment for 10 seconds, coating with aqueous PVA/SF/PEG (3%/3%/1%)
21 solution, and finally, 8 minutes of treatment with 60% ethanol solution. The fully
22 modified BOPP films exhibited approximately a 16° static contact angle (SCA), a
23 near zero haze value, and an effectively 100% transmittance value under visible light
24 (400–700 nm). The atomic force microscopy (AFM) of the surface morphology of the
25 modified BOPP films showed that the surface roughness increased from 3.79 nm

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1 (untreated) to 21.10 nm (fully treated). The Fourier transfer infrared spectroscopy
2 (FT-IR) results showed that polar functional C=O groups were grafted onto the BOPP
3 film that was pretreated with the DBD plasma. Further modification of the pretreated
4 BOPP film with the PVA/SF/PEG coating significantly enhanced the density of the
5 C-O and N-H groups. The gravure printing images indicated that the adhesive
6 property of the BOPP film for water-based ink improved substantially after the
7 hydrophilic modifications.

8 **Keywords:** BOPP film; DBD plasma pretreatment; Composite coating; Ethanol
9 solidification; Gravure printing

10 **1. Introduction**

11 Biaxially oriented polypropylene (BOPP) film is widely used in the production
12 of protective coatings, pressure sensitive tape, and decorative products and in labeling
13 and printing [1]. It is also increasingly being used to replace traditional materials such
14 as glass, metal, and paper in food packing applications because it possesses better
15 flexibility, higher transparency, and adequate chemical inertness [2]. However,
16 because of its low surface free energy (SFE) (lower than 22 mJ/m² [3, 4]), BOPP film
17 has a low adhesive ability with other materials. Therefore, it is essential to increase
18 the SFE value of the hydrophobic BOPP film to meet the requirements for
19 applications in the material and printing fields. It should be noted that presently the
20 prevailing practice is to use the oily-inks to print on the hydrophobic BOPP films,
21 such practices are unfriendly to both the workers and the environment. Because of the
22 stricter regulations imposed by many countries (both developed and developing
23 countries) on printing industries to minimize environmental concerns, new protocols
24 employing water-based inks for printing are greatly desired and the hydrophilic
25 modification of the BOPP film is therefore of great interest both scientifically and
26 practically.

27 In recent years, chemical, plasmatic, and corona discharge processes have been
28 applied to increase the SFE value of BOPP film [5]. Among these approaches, the
29 corona discharge technique has gained popularity for industrial applications of BOPP

1 film because of its simplicity, suitability for continuous online operation, and cost
2 effectiveness. This technique can graft functional groups including hydroxyl, carbonyl,
3 and carboxylic polar groups onto a BOPP film surface, which will increase the SFE
4 value of the modified BOPP film [6]. However, Novak *et al.* [7] showed that the SFE
5 value of corona discharge-treated BOPP film decreases with time.

6 To mitigate such a drawback, a BOPP film treated by corona discharge with an
7 acrylic acid (AA) monomer was investigated by Liao [8] and Nanticha [6] et al.
8 Unfortunately, the AA polymer coating has a weak water resistance and is generally
9 inhomogeneous because of the filament discharge created by the corona plasma [9].
10 Additionally, while many vacuum plasma systems have been widely used to modify
11 organic films [10], they are often associated with high operational costs. Therefore, it
12 is necessary to develop a new protocol for the modification of BOPP film, especially
13 for water-based ink printing applications.

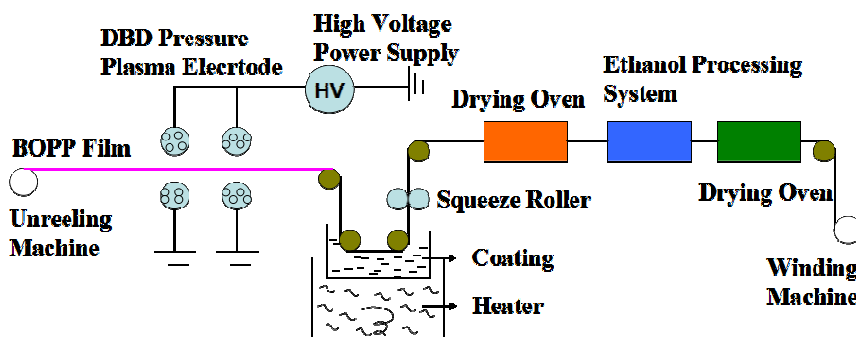
14 In this paper, a novel protocol for the BOPP film modification is reported. The
15 protocol consisted of three sequential processes: 1) the dielectric barrier discharge
16 (DBD) plasma pretreatment, 2) the polyvinyl alcohol/silk fibroin/polyethylene glycol
17 (PVA/SF/PEG) coating, and 3) ethanol solution finishing. The pretreatment of an
18 on-line DBD plasma system ensured an adequate adhesion between the BOPP film
19 and the functional coating, the PVA/SF/PEG coating yielded the desired topology, and
20 the final finishing step with 60% ethanol solution transformed the silk fibroin
21 structure in the functional coating from the alpha helix to beta folding to enhance the
22 hydrophilicity of the modified BOPP film [11].

23 **2. Experimental**

24 **2.1 The BOPP film modification system**

25 Fig. 1 illustrates the system employed for the hydrophilic modification of the
26 BOPP film, which consists of four main components: the DBD plasma pretreatment
27 apparatus, the coating device, drying system, and the ethanol treatment chamber. The
28 DBD plasma was realized by using four parallel liquid electrodes, and a quartz tube
29 with a wall thickness of 1 mm acted as the dielectric layer. The gap between the top

1 and bottom electrodes was 2 mm. An AC power source with a maximum peak voltage
 2 of 30 kV and an adjustable frequency range of 8 to 30 KHz was employed for the
 3 plasma generation. The discharge power and the treatment time of the plasma system
 4 are optimized to 60 watts and 3 s, respectively. The PET pretreatment with DBD
 5 plasma can improve the binding intensity between the PET membrane and
 6 PVA/SF/PEG film. The PVA/SF/PEG solution warmed in a water tank (40-50°C) was
 7 coated homogeneously onto the surface of the pretreated BOPP film using squeeze
 8 rollers. Subsequently, the coated BOPP film was dried in the first drying oven
 9 (60-80°C) for 120 s. Finally, The PVA/SF/PEG-coated BOPP film passed through the
 10 ethanol treatment chamber and was dried by an oven (80-120°C) for 90 s. It should be
 11 noted that the batch-type operation can be performed in the step of ethanol processing.



12
 13 Fig.1. Schematic illustration of the complete system for the BOPP film
 14 modification

15 2.2 Preparation of the PVA/ SF/PEG coating solution

16 Silk fibroin was obtained following a similar process as reported in our previous
 17 work [12]. Briefly, a degummed silk fibroin from *B. mori* was dissolved in a 9.3 M
 18 lithium bromide (LiBr) solution overnight at 37°C, followed by dialysis processing
 19 for 3 days to remove the LiBr. PVA (PVA 1797, Shanghai Petrochemical Co., Ltd.)
 20 and PEG (PEG 10000, Hangzhou Gaojing Fine Chemical Industry Co., LTD) powders
 21 were separately dissolved in deionized water at the desired concentrations (at 80°C
 22 and facilitated by a water bath) and mixed with the SF solution at ratios corresponding
 23 to the desired SF/PVA/PEG ratios. The rheological characteristic of the PVA/SF/PEG
 24 mixture was evaluated using Dynamic Rheometry (Brookfield / DV-III). In this paper,

1 unless otherwise specified, the PVA, SF, PEG, and ethanol concentrations are all in
2 weight percentages.

3 **2.3 The PVA/SF/PEG coating characterization**

4 Water static contact angle (SCA) measurements (using a Kruss Drop Shape
5 Analysis System (DSA10, Kruss GmbH, Germany)) were employed to characterize
6 the hydrophilicity of the BOPP films that were modified under various conditions. A
7 volume of 3 μL of deionized water was dropped on the surface of the modified BOPP
8 film, and the SCA value was measured with a CCD camera. Three measurements on
9 three different locations of each specimen were performed, and the average SCA
10 value was used for analysis. The surface morphology and roughness of the modified
11 BOPP film was achieved in air under ambient conditions using atomic force
12 microscopy (AFM, XE-100E, PSLA, Korea), which employs a NSC-15/Al probe
13 operated in non-contact mode. Meanwhile, the topography of modified BOPP film
14 was also characterized by a field emission scanning electron microscopy (FESEM,
15 JEOL, Japan). The chemical composition of the modified BOPP film was investigated
16 using XPS (K-Alpha, USA). The X-ray source was Al $K\alpha$ at 1486.6 eV and was
17 operated at 300 W. The tensile strength was measured as a representative mechanical
18 property of the modified BOPP film using a tensile strength instrument (XLW-B,
19 Shanghai Tianzhi Co., LTD). The haze and transmittance values of the modified
20 BOPP film were measured with an ultraviolet-visible spectrometer at 633 nm (WGT-S,
21 shanghai Jingke Co., LTD, China). Finally, the gravure printing properties of the
22 modified BOPP film were explored using a pilot-scale roll-to-roll machine (IGT/G1)
23 with a red water-based polyurethane ink (Red-1) (Wuhan Sanhe Surplus Industrial and
24 Trading Company, China).

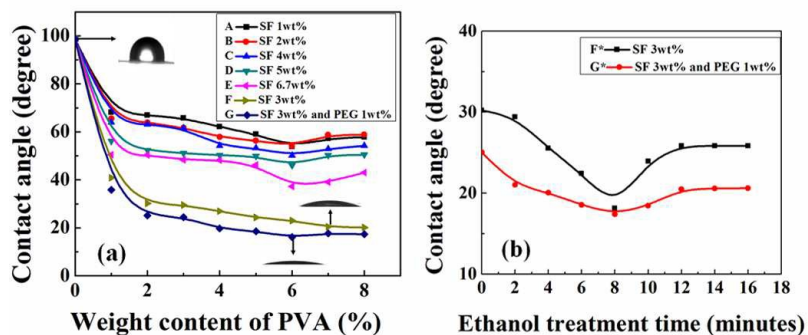
25 **3. Results and discussion**

26 **3.1 The optimal ratio of PVA/SF/PEG for modifying the BOPP film**

27 To determine the optimized PVA/SF/PEG ratio, solutions containing PVA, SF,
28 and PEG at various concentrations were prepared, and the SCA values of the modified
29 BOPP films were measured. Fig. 2 (a) shows the SCA values as a function of the PVA

1 for various SF and PEG concentrations, and the SCA value initially significantly
2 decreased with an increase in the PVA content. However, the decrease rate became
3 much smaller once the PVA concentration exceeded 3%. In contrast, for the same PVA
4 concentration, the SCA value of the modified BOPP film decreased noticeably with an
5 increase in the SF concentration (lines A to E for 1%, 2%, 4%, 5%, and 6.7%,
6 respectively), with the exception of 3%, which clearly resulted in a significantly lower
7 SCA value, as shown in Fig. 2 (a) for line F. Based on the phenomena observed in Fig.
8 2 (a) and the effect of the viscosity on the coating process, 3%/3% was selected as the
9 optimized ratio for PVA/SF.

10 As an excellent softening and dispersant agent, PEG [10] was added to negate
11 the adverse effect (the PVA/SF coating layer was brittle) of the inner stress between
12 the PVA/SF coating and the BOPP film. The experimental results, depicted by line G
13 in Fig. 2 (a), showed that adding 1% of PEG is sufficient for our purposes. The
14 optimized PVA/SF/PEG ratio was determined consequently to be 3%/3%/1%. At this
15 ratio, the SCA value was lower than 20°, which is the desired SCA value for high
16 quality printing in the packing industry. To increase the hydrophilicity of the
17 functional layer, the PVA/SF and PVA/SF/PEG coated BOPP films were further
18 treated with 60% ethanol solution, and Fig. 2 (b) shows the change in the SCA values
19 as a function of treatment time for the samples F and G as shown in Fig. 2 (a)
20 (represented as F* and G*). The data depicted in Fig. 2 (b) indicated that the SCA
21 values of F* and G* as a function of ethanol treatment time followed a “V” pattern
22 and the minimum SCA values of F* and G* lines were approximately 22° and 16°,
23 respectively, with a treatment time of 8 minutes (considered as the optimal ethanol
24 treatment time). Moreover, the treatment with 60% ethanol solution also significantly
25 improved the hydrophilicity of the modified BOPP films (discussed in later sections).



1

2 Fig. 2. SCA measurements: (a) The SCA values of the BOPP films modified
 3 under various conditions, (b) The SCA values of the BOPP film coated with
 4 PVA/SF(F*: 3%/3%) and PVA/SF/PEG (G*: 3%/3%/1%) as a function of the
 5 treatment time for the 60% ethanol solution

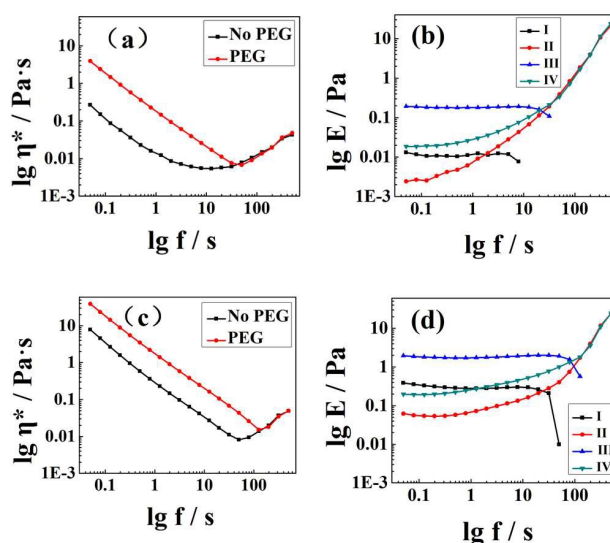
6 3.2 The rheological properties of the PVA/SF/PEG solution

7 The rheological properties of the PVA/SF/PEG solution were evaluated by
 8 studying the linear viscoelastic behaviors, as shown in Fig. 3. Figures 3 (a) and 3 (c)
 9 show the complex viscosity as a function of the angular frequency. The measured
 10 complex viscosity inversely depended on the angular frequency in a near linear
 11 manner at low frequencies for two different PVA/SF ratios with/without 1% PEG
 12 (3%/3% and 5%/3% for Fig. 3 (a) and 3 (c), respectively). The data depicted in Fig.
 13 3(a) and 3(c) also showed that the measured complex viscosity increased noticeably
 14 when the PVA concentration increased from 3% to 5%. Additionally, adding 1% PEG
 15 also significantly increased the measured complex viscosity value, which is in good
 16 agreement with Zhao [13] and Gahleitner [14].

17 Figures 3(b) and 3(d) show the values of the storage moduli (lines I and III) and
 18 the loss moduli (lines II and IV) as a function of the angular frequency for the samples
 19 depicted in Fig. 3(a) and 3(c), respectively. As observed from Fig. 3(b) and 3(d), for
 20 both samples, the storage modulus is relatively independent of the angular frequency
 21 (lower end), and the addition of 1% PEG would significantly increase the storage
 22 modulus and result in an improved elasticity.

23 However, for the loss modulus and for low angular frequencies, the loss modulus
 24 increased with increasing frequency. The addition of 1% PEG resulted in a significant

1 increase in the loss modulus. However, the data depicted in Fig. 3 (b) showed that
 2 such an increase diminished as the frequency increased up to 30 Hz (Line II for no
 3 PEG and line IV for 1% PEG, as shown in Fig. 3 (b)). A similar trend for the loss
 4 modulus was observed in Fig. 3 (d) as well. However, unlike what was depicted in Fig.
 5 3 (b), the increase in the loss modulus induced by the 1% PEG remained unchanged
 6 as a function of frequency until the frequency approached 100 Hz. The phenomena
 7 observed in Figs. 3 (b) and 3 (d) indicate that adding 1% PEG increased the flexibility
 8 and adhesive property of the functional coating.



9

10 Fig. 3. The rheological behaviors of samples with different PVA/SF/PEG ratios:
 11 (a) The PVA/SF (3%/3%) solution, (b) The storage modulus (I without PEG, III with
 12 PEG) and the loss modulus (II without PEG, IV with PEG) of the sample (a), (c) The
 13 PVA/SF (5%/3%) solution, (d) The storage modulus (I without PEG, III with PEG)
 14 and loss modulus (II without PEG, IV with PEG) of the sample (c). The PEG
 15 concentration was 1%.

16 3.3 The mechanical properties of the modified BOPP film

17 The tensile force (TF_{σ} , MPa) and the relative elongation (E_{ε} , %) at break were
 18 evaluated to determine the effect of the modification parameters on the mechanical
 19 properties of the BOPP film. Table 1 summarizes the relevant mechanical
 20 characteristics of the BOPP films modified under various conditions. The thickness,
 21 length and width of the original BOPP film were 20 μm , 100 mm and 10 mm,

1 respectively, and the thicknesses of the PVA/SF/PEG and PVA/SF coating layers were
 2 each approximately 1.5 μm . Sample I was the original BOPP film, and its tensile force
 3 at break σ (TF_σ) and its elongation at break ε (E_ε) were approximately 138.65 MPa
 4 and 9%, respectively. After the DBD plasma pretreatment for 3 seconds (for sample
 5 II), the TF_σ and E_ε values increased to 158.7 MPa and 10%, respectively (such
 6 increases were considered irrelevant and were usually neglected because of the
 7 characteristics of the DBD plasma treatment [15]). The TF_σ and E_ε values of samples
 8 III-VIII were compared with those of sample II to identify the effects of the various
 9 coating layers. As shown in Table 1, after the BOPP film was modified by the
 10 PVA/SF/PEG (of various ratios) coatings, substantial increases in both the TF_σ and E_ε
 11 values were observed, and sample VI (PVA/SF/PEG ratio of 3%/3%/1%) exhibited
 12 the best mechanical properties in terms of the TF_σ and E_ε values. These observations
 13 were consistent with the rheological property analysis in Section 3.2. Moreover, after
 14 sample VI was further treated with deionized water or 60% ethanol solution, the TF_σ
 15 and E_ε values decreased, but the ethanol solution treatment only marginally decreased.
 16 Considering the significant hydrophilicity that was induced by the 60% ethanol
 17 solution treatment, such a marginal sacrifice of the mechanical strength was
 18 warranted.

19 Table 1. Mechanical strength properties of the BOPP samples: I: The original
 20 BOPP film; II: The DBD plasma pretreated sample; III: The PVA/SF (3%/3%) coated
 21 sample; IV: Sample III treated with deionized water for 8 minutes; V: Sample III
 22 treated with 60% ethanol solution for 8 minutes; VI: The PVA/SF/PEG (3%/3%/1%)
 23 coated sample; VII: Sample VI treated with deionized water for 8 minutes; VIII:
 24 Sample VI treated with 60% ethanol solution for 8 minutes

Sample	Tensile force at break σ (MPa)	Elongation at break ε (%)
I	138.65	9.00
II	158.70	10.00
III	215.44	21.00

IV	165.40	16.00
V	210.05	20.00
VI	216.34	26.00
VII	199.05	15.00
VIII	213.32	24.00

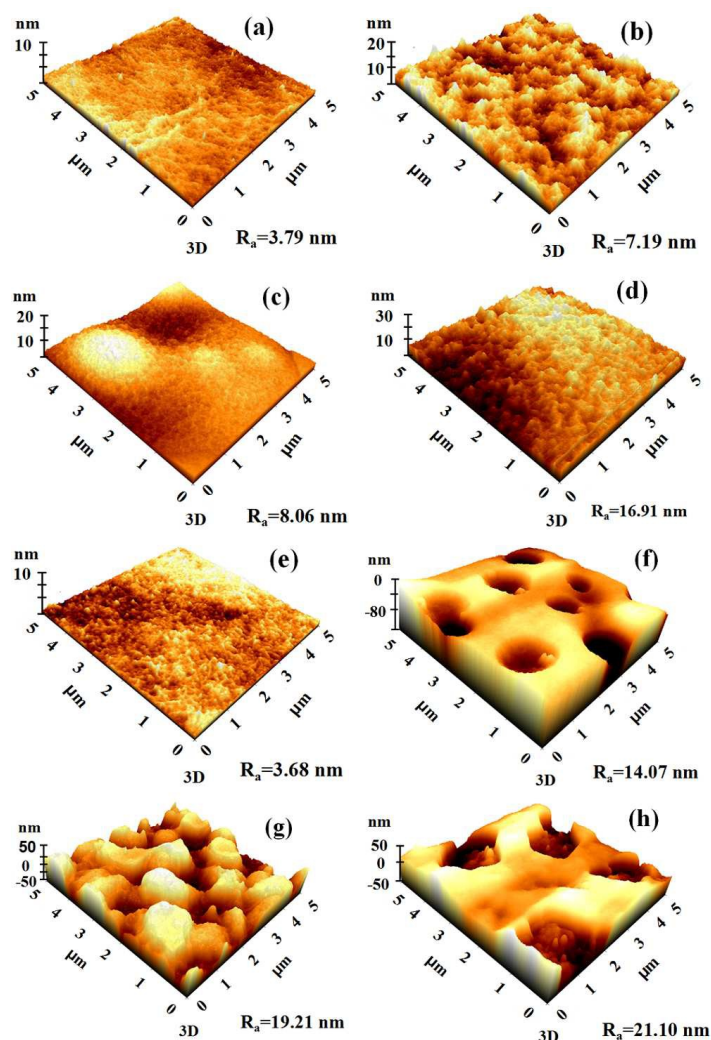
1 3.4 The morphological characteristics of the modified BOPP film

2 AFM provides both qualitative (visual) and quantitative characterization of the
3 surface morphology of films [16] and was employed in this study. Fig. 4 shows the
4 morphologies and the roughness of the BOPP film treated under different conditions.
5 Compared with the original BOPP film (Fig. 4(a)), the roughness of the BOPP film
6 pretreated by the DBD plasma increased by approximately 90% (from 3.79 to 7.19
7 nm), and the surface was visually much coarser, as shown in Fig. 4(b). This
8 observation was encouraging because a higher surface roughness often leads to a
9 significantly enhanced combined intensity due to the BOPP film and the functional
10 coating. Fig. 4(c) and 4(d) display the AFM images of the BOPP film modified with
11 PVA/SF (3%/3%) and PVA/SF/PEG (3%/3%/1%) coatings, and a homogeneous
12 distribution of the coating materials is observed in both figures. However, the
13 PVA/SF/PEG coating resulted in a significantly higher roughness of 16.91 nm (a
14 135% increase from 7.19 nm), whereas only a marginal increase in roughness (from
15 7.19 to 8.06 nm) was induced by the PVA/SF coating. The addition of 1% PEG was
16 highly beneficial.

17 Figures 4(e) and 4(f) present the effect of deionized water on the surfaces of the
18 BOPP films that were modified with PVA/SF and PVA/SF/PEG coatings. The
19 treatment with deionized water on the PVA/SF coated BOPP film resulted in a similar
20 topology with a significantly decreased surface roughness (from 8.06 to 3.68 nm, even
21 lower than the 7.19 nm before coating), as shown in Fig. 4(e). However, the treatment
22 with deionized water on the PVA/SF/PEG coated BOPP film resulted in a much
23 different topology with a slightly decreased surface roughness (from 16.91 to 14.07
24 nm), as shown in Fig. 4 (f). Isolated holes were created, and some had diameters as
25 large as 1 μm . Such a dramatic alteration in the surface topology suggests that the

1 PEG in the PVA/SF/PEG coating might have leached out into the deionized water.
2 This may be attributed to the high solubility of PEG in water. Fig. 4(f) indicates that a
3 substantial increase in the hydrophilicity of the BOPP film that was coated with the
4 PVA/SF/PEG would be required for it to meet the criteria for high quality printing
5 with water-based ink. This issue was addressed by treating the modified BOPP film
6 with the 60% ethanol solution, as discussed in section 3.1.

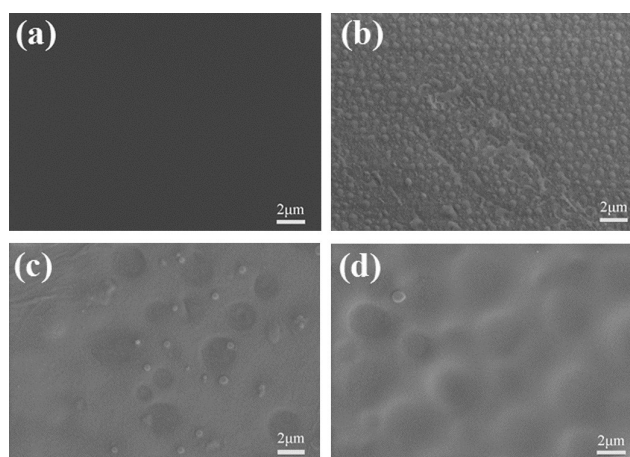
7 Figures 4(g) and 4(h) show the surface morphologies of the PVA/SF and the
8 PVA/SF/PEG coated BOPP film after treatment with the 60% ethanol solution. Some
9 nodules formed due to the ethanol in the PVA/SF coating (see Fig. 4(g)). A potential
10 explanation for this phenomenon is that the contraction of the SF micro-islands during
11 the transformation from the alpha helix to beta folding may extrude the swollen PVA
12 nodules. However, for the PVA/SF/PEG coating, the treatment with 60% ethanol
13 solution still induced the creation of micro holes, but the average depth of these holes
14 was approximately 50 nm, which is substantially lower than those created by the
15 deionized water (approximately 80 nm) (Fig. 4(h) and 4(f)). Moreover, the surface
16 roughness was further increased to 21.10 nm. Such improvements may be due to the
17 restructuring effect of ethanol on the SF, which may have partially reduced the
18 exposure of the PEG content in the PVA/SF/PEG coating.



1
 2 Fig. 4. The AFM 3D images of different samples: (a) The original BOPP film, (b)
 3 The BOPP pretreated with DBD plasma, (c) The BOPP pretreated with DBD plasma
 4 and coated with PVA/SF (3%/3%), (d) The BOPP pretreated with DBD plasma and
 5 coated with PVA/SF/PEG (3%/3%/1%), (e) The sample (c) treated with deionized
 6 water, (f) The sample (d) treated with deionized water, (g) The sample (c) treated with
 7 60% ethanol solution, (h) The sample (d) treated with 60% ethanol solution

8 In order to investigate the effect of processing protocol employed in this study on
 9 the BOPP topography, the SEM measurement was also performed. Fig. 5(a) shows the
 10 surface morphology of original BOPP film, and it was fairly smooth. After the
 11 original BOPP film was treated by the DBD plasma for 3 s, as shown in Fig. 5 (b),
 12 many micro-papilla were formed on the BOPP surface and the surface roughness was
 13 increased significantly, consistent with the AFM results presented in Fig. 4. When the

1 plasma pretreated BOPP film was coated with the PVA/SF/PEG coating and dried, as
2 shown in Fig. 5 (c), many micro-islands appeared. A definite mechanism of such an
3 island formation is yet to be determined but it may be caused by the agglomeration of
4 the SF phase in the drying process. Fig. 5 (d) shows the surface of the PVA/SF/PEG
5 coated BOPP film after the 60% ethanol solution treatment. Compared with Fig. 5 (c),
6 the boundary of islands became unclear, which indicates that the phases of PVA and
7 SF in PVA/SF/PEG film were rearranged under the function of ethanol.

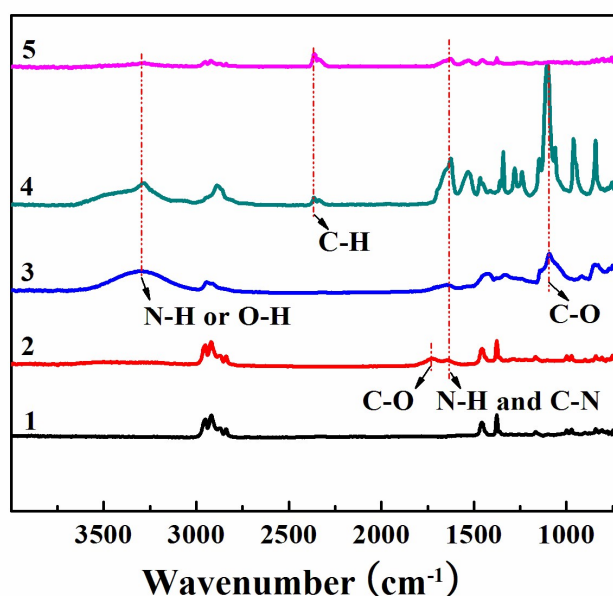


8
9 Fig. 5. The SEM images of samples: (a) The original BOPP film, (b) The BOPP
10 pretreated with DBD plasma, (c) The sample (b) coated with PVA/SF/PEG
11 (3%/3%/1%), (d) The sample (c) treated with 60% ethanol solution.

12 3.5 The chemical structure analysis on the modified BOPP film

13 A widely used method to probe the chemical constituents of thin films,
14 ATR-FTIR spectroscopy [17], was employed to characterize the chemical structures
15 of the modified BOPP films, as shown in Fig. 6. The assignments of the main
16 absorption peaks in the ATR-FTIR spectra are summarized in Table 2. Curve 1 in Fig.
17 6 is the spectrum of the original BOPP film. The characteristic peaks at 2930, 1450,
18 1375, 1168, 978 and 840 cm^{-1} that represent C-H stretching matched well with those
19 previously reported [18-20]. Curve 2 in Fig. 6 is the spectrum of the BOPP film
20 modified with the PVA/SF coating, and new peaks appeared within the range of
21 1725-1650 cm^{-1} . These new peaks were assigned to the bending of the N-H bond
22 associated with C-N and C-O stretching [21]. When the PVA/SF coated BOPP film
23 was further treated with the 60% ethanol solution, the peak at 1725 cm^{-1} disappeared,

1 as represented by curve 3. This disappearance may be due to the transformation of the
 2 SF [11]. Moreover, the intensity of the absorption bands at 3300 cm^{-1} , which
 3 represented the stretching of the N-H or O-H groups, increased substantially,
 4 indicating an increased exposure of the PVA or the amide groups to the SF. Curves 4
 5 and 5 in Fig. 6 show the structure conversion of the PVA/SF/PEG coated BOPP film
 6 before and after 60% ethanol solution treatment, respectively. All of the absorption
 7 peaks decreased. Additionally, the peak at 1080 cm^{-1} in curve 5 also disappeared. The
 8 cause for such decreases and the disappearance may be due to possible leaching of the
 9 PEG contents into the solution.



10

11 Fig. 6. The ATR-FTIR spectra of the BOPP films: Curve 1: The original BOPP
 12 film; Curve 2: The PVA/SF (3%/3%) coated film; Curve 3: Sample 2 treated with
 13 60% ethanol solution; Curve 4: The PVA/SF/PEG (3%/3%/1%) coated film; Curve 5:
 14 Sample 4 treated with 60% ethanol solution

15 Table 2 The assignments of the main absorption peaks in the infrared spectra of
 16 the BOPP samples

Absorption Peak/ cm^{-1}	Assignment	Vibration type
--------------------------------------	------------	----------------

3300	N-H	O-H	Stretching vibration	
2930	C-H		Stretching vibration	
2350	C-H		Stretching vibration	
1725 - 1650	N-H	C-N	C-O	Stretching and bending vibration
1450 - 1375	CH ₂	CH ₃	Bending vibration	
1100 - 1080	C-O		Stretching vibration	

1 Table 3 summarizes the atomic compositions of the BOPP films, as determined
 2 via the XPS analysis. Compared with the original BOPP film (only C was present), 10
 3 seconds of DBD plasma pretreatment exposed O and N and increased the O/C and
 4 N/C ratios, indicating the polar groups grafted onto the BOPP surface. These polar
 5 groups were beneficial because they increased the adhesive properties of the film
 6 surface. Moreover, after the pretreated BOPP film was modified with the
 7 PVA/SF/PEG coating, the atomic concentration of O increased significantly and the
 8 O/C ratio doubled (compared with the pretreated BOPP film). However, the treatment
 9 with 60% ethanol solution resulted in a slightly lower O/C ratio and a substantially
 10 higher N/C ratio. This may be due to the bonding of the amino functional groups of
 11 the SF on the surface due to the presence of the polar groups in ethanol. Moreover, the
 12 treatment with 60% ethanol solution may decrease the surface energy because it
 13 increased the surface roughness and the density of the -NH₂ groups as the FTIR
 14 results presented in Fig. 6. Both lead to lower surface energy, according to the
 15 Guimond theory, which suggests that the surface energy of a material is mainly
 16 determined by the surface structures and atomic types [22].

17 Table 3 Summary of the atomic compositions of the BOPP films calculated with
 18 the XPS data

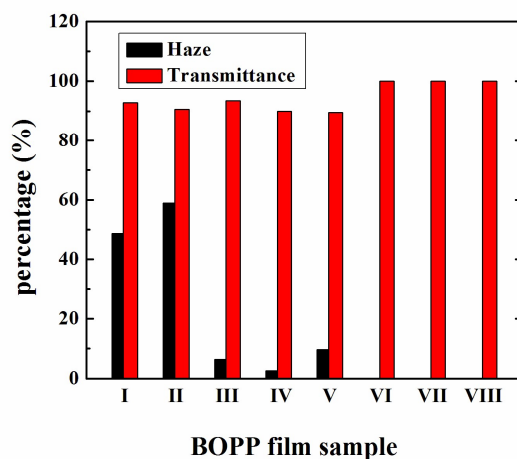
Sample	Atomic concentration (%)			Atomic ratio	
	O	N	C	O/C	N/C
Original BOPP film	0	0	100	0	0
DBD pretreated	17.27	2.19	80.54	0.21	0.03

PVA/SF/PEG coated	29	1.9	69.1	0.42	0.03
Ethanol treated (PVA/SF/PEG coated)	25.63	7.64	66.74	0.38	0.11

1 3.6 The haze and transmittance analysis

2 The haze and transmittance values are two important parameters for the packing
3 applications of BOPP films. The haze and transmittance values for the
4 original/modified BOPP films are shown in Fig. 7. For the original BOPP film
5 (sample I), the haze and transmittance values were 48.6% and 92.9%, respectively.
6 The DBD plasma pretreatment resulted in a noticeable 21.0% increase in the haze
7 value and a marginal 2.0% decrease in the transmittance value. (sample II in Fig. 7).
8 A significant decrease in the haze value was observed after the BOPP film was modified
9 with the PVA/SF (3%/3%) coating (to 6.4%, as represented by sample III in Fig. 7).
10 Further deionized water treatment led to a lower haze value (2.6%), while further
11 ethanol solution treatment yielded an increased haze value (9%) (samples IV and V,
12 respectively). However, after the BOPP film was modified with the PVA/SF/PEG
13 (3%/3%/1%) coating, the haze value was reduced to near zero, and the transmittance
14 value increased to effectively 100%. Virtually no difference in the haze and
15 transmittance values was observed after the PVA/SF/PEG coated BOPP film was
16 further treated with either deionized water or ethanol solution. The cause of the low
17 haze and high transmittance values was due to the antistatic functional groups [5] and
18 the surface roughness [23, 24].

19 Lin [25] suggested that a lower surface roughness decreases the haze and
20 increases the transmittance values of BOPP films, in contrary to our results. Our
21 experimental results showed that the BOPP film modified with the PVA/SF/PEG
22 coating had a high surface roughness, a near zero haze value, and a nearly 100%
23 transmittance value. The cause of the discrepancy has yet to be determined but may be
24 due to the hydrophilic nature of the PVA/SF/PEG coating, which generally is
25 associated with reduced haze values [26, 27].

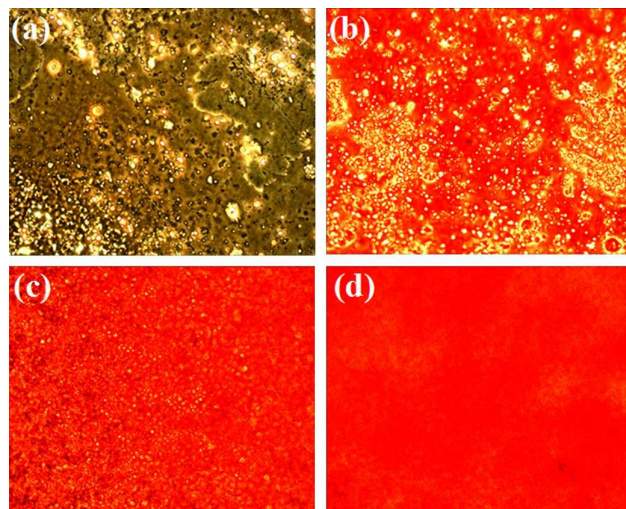


1

2 Fig. 7. The visible range haze and transmittance values of the BOPP films: I: The
3 original BOPP film; II: The DBD plasma pretreated film; III: The PVA/SF (3%/3%)
4 coated film; IV: Sample III treated with deionized water; V: Sample III treated with
5 60% ethanol solution; VI: The PVA/SF/PEG (3%/3%/1%) coated film; VII: Sample
6 VI treated with deionized water; VIII: Sample VI treated with 60% ethanol solution

7 3.7 Printability

8 Fig. 8 shows the gravure printing quality of the BOPP films that were treated
9 with different methods. The printing quality of the original BOPP film was very poor
10 (multiple ink dots were present), as shown in Fig. 8 (a). The printability increased
11 marginally when the BOPP film was pretreated with the DBD plasma. However, the
12 red ink was still not homogeneously printed on the BOPP surface, as shown in Fig. 8
13 (b). Fig. 8 (c) demonstrates the printing quality of the BOPP film modified with the
14 PVA/SF/PEG coating, and a substantially higher pixel accuracy was achieved. After
15 further treatment with 60% ethanol solution for 8 minutes, visually flawless printing
16 was achieved, as depicted by Fig. 8 (d). Fig. 8 shows that the novel protocol
17 developed in this study successfully improved the applicability of the BOPP film for
18 high quality printing. Additionally, our experimental results agree with the claim that
19 the wettability is closely associated with the ink adhesion [28, 29].



1

2 Fig. 8. Images (20 mm x 20 mm) at 100 times magnification of the red water-based
3 ink printed on the BOPP films: (a) On the original BOPP film; (b): On the DBD
4 plasma pretreated film; (c): On the PVA/SF/PEG (3%/3%/1%) coated film; (d): On
5 sample (c) treated with 60% ethanol solution

6 **4. Conclusion**

7 A novel modification protocol was established and an environmentally friendly
8 PVA/SF/PEG coating was designed to improve the printability of BOPP film, making
9 the modified BOPP film highly competitive for packing applications. The entire
10 modification process included DBD plasma pretreatment (corona discharge),
11 PVA/SF/PEG coating, and ethanol solution finishing. The 60% ethanol solution
12 finishing significantly enhanced the hydrophilicity of the BOPP films modified with
13 the PVA/SF/PEG coating. The contact angle of the finished BOPP film decreased
14 from 98.5° to 16.09° after the entire treatment processes, and the haze value decreased
15 to nearly zero, whereas the transmittance value increased to effectively 100%. The
16 surface characteristics of the modified BOPP film are highly competitive for high
17 quality printing applications. Moreover, besides enhancing the printability, the
18 PVA/SF/PEG coating also improved the mechanical strength of the BOPP films.
19 While some micro-holes occurred on the modified BOPP surface, their impact on the
20 printing quality is negligible because the micro-holes are shallow. Indeed, such
21 micro-holes may be favorable because our experimental results indicated that they

1 may enhance the combined intensity of the ink and the modified BOPP film. High
2 quality printing with water-based ink was achieved using the fully modified BOPP
3 film.

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