

# RSC Advances

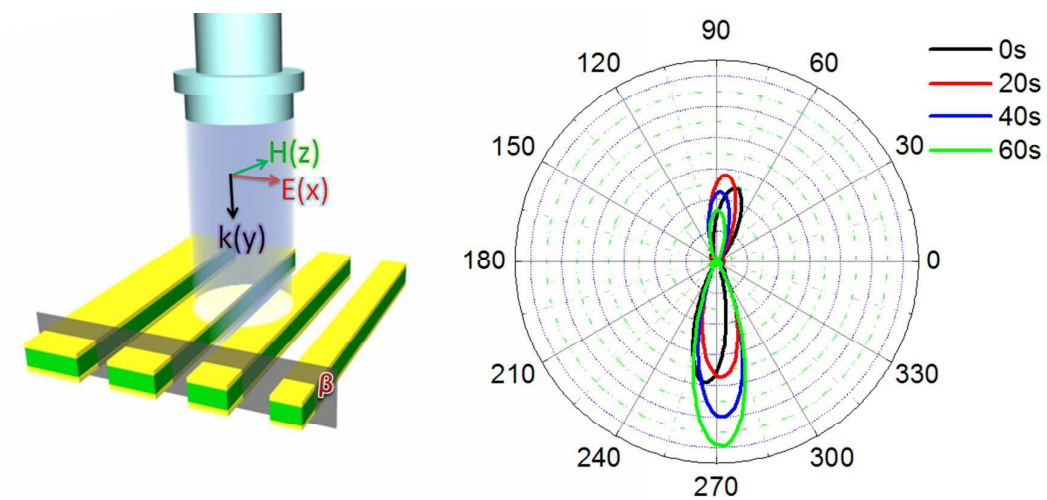


This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

*Accepted Manuscripts* are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



Achievement of continuous light-steering in an array of gradient Au/Bi<sub>2</sub>Se<sub>3</sub>/Au strips by modulating the dielectric function of Bi<sub>2</sub>Se<sub>3</sub>



Journal Name

ARTICLE

## Chemical control of continuous light-steering using an array of gradient Au/Bi<sub>2</sub>Se<sub>3</sub>/Au strips

Tun Cao,<sup>\*a</sup> Guangzhao Zheng<sup>a</sup> and Shuai Wang<sup>a</sup>Received 00th January 20xx,  
Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

www.rsc.org/

Light-steering devices have been extensively investigated for a large host of applications in defense, communications, data storage and display technologies. However, the challenge is to continuously steer the light over a wide angular range in the optical region, using actively tunable structure with a small footprint. Here, we numerically demonstrate a gradient phase-array-like Au/Bi<sub>2</sub>Se<sub>3</sub>/Au trilayers plasmonic resonators for actively steering the beam in the near-infrared (N-IR) region. The proposed device provides a continuously large beam-steering of 18° for the reflected light and 9° for the transmitted light around the resonant wavelength of 1500 nm by changing the state of Bi<sub>2</sub>Se<sub>3</sub> from amorphous to crystalline. The essence of this phenomenon lies in the variation of refractive index of Bi<sub>2</sub>Se<sub>3</sub> induced by the phase transition, which is achieved through the chemical reactions between the Bi<sub>2</sub>Se<sub>3</sub> film and AgNO<sub>3</sub> solution. The continuous beam-steering is enabled by gradually increasing the immersion time ( $t_m$ ) of the Bi<sub>2</sub>Se<sub>3</sub> into AgNO<sub>3</sub> solution. This study exploits a new research area of Bi<sub>2</sub>Se<sub>3</sub> based nano-antenna for dynamic optical routing and switching in photonic circuits.

### 1 Introduction

On-chip optical antennas array represents an enabling technology that makes fast, straightforward, and lightweight laser beam-steering with precise stabilization, random access pointing and programmable multiple simultaneous beams.<sup>1-3</sup> This would greatly impact optical communication, holographic video displays, and laser ranging and detection.<sup>4-7</sup> Such optical antennas array can steer and shape light efficiently by controlling the phase of each antenna element.<sup>8</sup> The phase control of the optical antennas array can be obtained by optimizing the geometry and location of the antenna elements, so called as passive antennas array.<sup>9-13</sup> Yet in order to achieve the required phase, designers have to carefully consider the pattern, arrangement, coupling between resonators as well as nanofabrication tolerance of the passive nano-antennas array. This makes packaging challenging and increases the complexity of the device. To solve the problem, active control of the propagation direction of beam has been extensively investigated and became the very heart of the beam-steering technology.<sup>14</sup> For example, on-chip optical phased arrays (OPAs) based on gratings have been proposed to actively steer the beam using frequency tuning.<sup>15-16</sup> Although these grating based OPAs have a good steering performance, the tunable frequency limits their applications in the field of free-space communication link and sensing.<sup>7</sup> Recently, OPAs integrated with various active elements, i.e. liquid crystals (LCs)<sup>17</sup>, active silicon<sup>14</sup>, thermally reconfigurable semiconductors<sup>18</sup> and phase changed materials (PCMs)<sup>8</sup>, have been approached to provide

an active beam-steering with low cost and simple mechanical systems. The key element of this technology is to modify the phase of OPAs by tuning the refractive index of the electro-optical material, and thus controlling direction and divergence of light at a fixed frequency. Despite the substantial progress in active OPAs, it is still a formidable challenge to continuously steer the beam over a wide angular range in the near-infrared (N-IR) region. For example, the beam-steering devices using OPAs integrated with active semiconductors are normally designed to operate at the wavelength of 10 μm<sup>14,18</sup>. However, little research has been done on actively steering the direction of the incident light at the shorter wavelength region. That is because the semiconductors do not sustain high densities of injected free electrons outside the THz frequency range, especially from the visible to N-IR regions<sup>19</sup>. Therefore, implementation of beam-steering using OPAs in the N-IR region is desirable, considering their applications in the field of optical communication. Meanwhile, the continuous beam-steering is often necessary for a rich variety of new and exciting areas i.e. automotive applications.<sup>1,6</sup> Moreover, the integration of the electrodes for tuning active dielectric materials restricts the scaling of OPAs and may be difficult to be realized by the current nanofabrication techniques. Therefore, an appealing route that is particular for active nano-antennas array is to chemically tune the optical properties of the structure without using large bulky electrodes.<sup>20-21</sup>

Topological insulator (TI), a new class of Dirac material, has attracted great interests in recent years.<sup>22-25</sup> It is a potential candidate for high performance optical devices such as photodetectors, terahertz lasers, rewriteable optical data storage, photonic circuitry that is less dependent on isolators

<sup>a</sup> Department of Biomedical Engineering, Dalian University of Technology, Dalian116024, China (P.R.C). E-mail: caotun1806@dlut.edu.cn

and slow light that is insensitive to disorder.<sup>26-27</sup>  $\text{Bi}_x\text{Sb}_{1-x}\text{Bi}_2\text{Se}_3$ , and  $\text{Bi}_2\text{Te}_3$  compounds are shown to be TIs using angle-resolved photoemission spectroscopy (ARPES),<sup>28-33</sup> where  $\text{Bi}_2\text{Se}_3$  is particularly interesting since it has a relatively large bulk band gap and simple surface state consisting of a single Dirac cone-like structure.<sup>30,34</sup> Recently, experiments have revealed that the optical dielectric constant of  $\text{Bi}_2\text{Se}_3$  can be very different in the N-IR region as transiting its state between crystalline and amorphous.<sup>35</sup> In this context, we envisage the possibility of continuously steering the light by using gradient phase-array-like metal/dielectric/metal (MDM) multilayer strips, where  $\text{Bi}_2\text{Se}_3$  is selected as the dielectric interlayer.

Here, an array of gradient Au/ $\text{Bi}_2\text{Se}_3$ /Au tri-layer strips is proposed to continuously control the propagation direction of incident light over a wide angle in the N-IR region. As two metal layers are placed closer than the surface plasmon polaritons (SPPs) attenuation length, the SPPs propagating along each of the two metal-dielectric interfaces of the MDM structures can couple to each other hence providing a very strong localization of light inside the internal dielectric layer.<sup>36</sup> By introducing  $\text{Bi}_2\text{Se}_3$  into the MDM structure as a dielectric interlayer, the tunable resonant characteristics of the structure can be realized by switching between two states of  $\text{Bi}_2\text{Se}_3$ , where the phase transition can be obtained through the chemical reactions between the  $\text{Bi}_2\text{Se}_3$  film and  $\text{AgNO}_3$  solution.<sup>35</sup> The variation in refractive index of the amorphous and crystalline state of  $\text{Bi}_2\text{Se}_3$  will change the intrinsic effective dielectric properties of the MDM structure.<sup>37</sup> Therefore, the MDM strips array integrated with  $\text{Bi}_2\text{Se}_3$  can be used as actively controlled phase shifters for beam-steering device. Moreover, it is hypothesized that a continuous beam-steering can be obtained by immersing  $\text{Bi}_2\text{Se}_3$  in  $\text{AgNO}_3$  solution for different periods of time. It is because that  $\text{Bi}_2\text{Se}_3$  film is amorphous in nature while those immersed in the  $\text{AgNO}_3$  solution are crystalline. By increasing immersion time ( $t_{im}$ ), the  $\text{Bi}_2\text{Se}_3$  dielectric layer can be gradually crystallized hence continuously steering the reflected light over a  $18^\circ$  (from  $72^\circ$  to  $90^\circ$ ) angular range and transmitted light over a  $9^\circ$  (from  $264^\circ$  to  $273^\circ$ ) angular range around the resonance wavelength of 1500 nm, respectively.

Reversibly transiting the state of  $\text{Bi}_2\text{Se}_3$  may sound challenging. To address this problem, we suggest that a future design, with improved cycleability, should include a possibility of the crystalline  $\text{Ag}_2\text{Se}$  transforming into the amorphous  $\text{Bi}_2\text{Se}_3$  through cation exchange reaction.<sup>38</sup> The low crystallization activation energy of 1.32 eV of the  $\text{Bi}_2\text{Se}_3$  may affect the stability of its amorphous phase.<sup>39</sup> However, new phase change materials, that may switch between two crystalline states without melting, such as interfacial phase change materials (IPCMs), should be used.<sup>40</sup> Although the tunable resonance of the plasmonic structure in one way limits its applications, it can be a potential candidate for a write-once device and thus still attracting much attention. For example, adjusting of the structural parameters to obtain the tunable metamaterials (MMs) has been widely studied.<sup>41-44</sup> Particularly, García-Meca et al. demonstrated that one can tune the permeability of the multilayer fishnet MMs by changing the

lattice constant of the structure.<sup>41</sup> The effective optical parameters of these MMs can only be tuned in one way since it is hard to change the geometry size of resonators in the metamaterials once they are fabricated.

Meanwhile, capillary action in the MDM strips array leads to an efficient reaction between the  $\text{Bi}_2\text{Se}_3$  film and  $\text{AgNO}_3$  solution.<sup>45</sup> Compared to the electrical controlled active beam-steering devices, our approach to demonstrate a chemically controlled optical beam-steering device can remove large bulky electrodes and thus considerably reducing size, weight and power requirements. The structure possesses a simple geometry which can be fabricated using standard photolithography patterning. Finally, it should be noted that  $\text{Bi}_2\text{Se}_3$  does not require any energy to maintain the structural state of the material. Thus, once the device has been switched it will retain the beam direction until it is switched again. This obviously makes the proposed beam-steering design interesting from a 'green technology' perspective.

## 2 Results and Discussions

### Structure and design

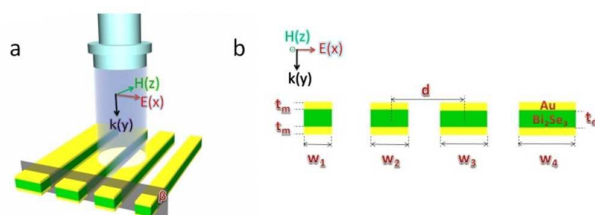
Fig. 1a shows an array of four gradient MDM multilayer resonators suspended in a vacuum. Each resonator consists of a sandwiched  $\text{Bi}_2\text{Se}_3$  dielectric interlayer between a top and bottom Au film. Au is selected as the metal due to its stability and low ohmic loss. The thicknesses of the Au/ $\text{Bi}_2\text{Se}_3$ /Au trilayers are at 20/140/20 nm. The MDM multilayer strips are assumed infinite along the  $z$  direction and  $\beta$  is a cross-section plane of the structure ( $x$ - $y$  plane). The geometry of the structure is optimized for the maximum sensitivity of the beam-steering angle to a change in the refractive index of the  $\text{Bi}_2\text{Se}_3$  layer at  $\lambda = 1500$  nm. The center to center distance between each element is  $d = 600$  nm. A finite-width MDM strip represents a plasmonic resonator, in which the counter propagating SPPs at the top and bottom metal-dielectric interfaces are efficiently reflected by the ends of the strip. It can form a resonant standing-wave pattern hence localizing the light within the dielectric interlayer when the counter-propagating SPPs interfere constructively. The resonance condition is given by

$$k_0 n_{eff} w = m \pi + \varphi \quad (1)$$

where  $k_0 = \frac{2\pi}{\lambda_0}$  is the wave vector of the incident light,  $n_{eff}$  is

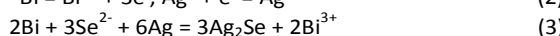
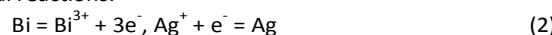
the effective refractive index of the MDM strip,  $w$  is the width of the strip,  $m$  is an integer, and  $\varphi$  is a reflection phase at the end of the strip.<sup>14,37,46</sup> For amorphous  $\text{Bi}_2\text{Se}_3$ ,  $n_{eff}$  is 3.42 at  $\lambda = 1500$  nm derived from the explicit dispersion relation in Ref. [37]. The  $w$  is then calculated to be 190 nm using Eq. (1), where  $m = 1$  for a first-order resonance. By adding a 20 nm width step on the resonance value of 190 nm, one can obtain four gradient MDM strips with different widths of  $w_1 = 190$ ,  $w_2 = 210$ ,  $w_3 = 230$  and  $w_4 = 250$  nm. These variously wide MDM resonators can come up with different phase shifts. The structure is simulated by a commercial software (Lumerical FDTD Solutions), which is based on the Finite Difference Time Domain (FDTD) method. The dielectric properties of Au as

given by Johnson & Christy are used.<sup>47</sup> The structure is excited by a plane wave source at a central wavelength of 1500 nm, propagating along the positive  $y$  direction with the electric-field vector ( $E$ ) polarized in the  $x$  direction. Perfectly match layer (PML) absorbing boundaries are applied for all directions. A uniform FDTD mesh size is adopted, the mesh size is the same along all Cartesian axes:  $\Delta x = \Delta y = 2$  nm, which is sufficient to minimize the numerical errors arising from the FDTD method. A near-to-far field transformation within Lumerical FDTD has then been used to calculate the far field radiation pattern of the structure.<sup>48-49</sup>



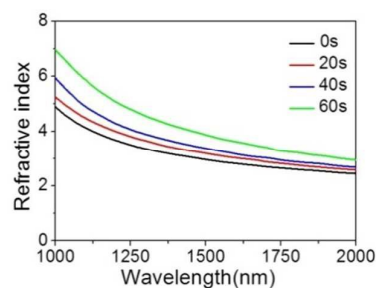
**Fig. 1** a) Schematic of an array of four MDM strip resonators consisting of a 140 nm thick  $\text{Bi}_2\text{Se}_3$  dielectric layer between two 20 nm thick Au films suspended in a vacuum; b) cross-section of array of MDM strips with different widths for active beam-steering, where  $w_1 = 190$ ,  $w_2 = 210$ ,  $w_3 = 230$ ,  $w_4 = 250$  and  $d = 600$  nm.

The  $\text{Bi}_2\text{Se}_3$  is a promising candidate to realize modulation functionality since its optical properties dramatically change in the N-IR regime for different  $t_{im}$  of the  $\text{Bi}_2\text{Se}_3$  in the  $\text{AgNO}_3$  solution. Due to an ion exchange process in the  $\text{AgNO}_3$  solution, the substitution of Bi by Ag occurs according to the following chemical reactions:



At the beginning of doping, Ag enters into the interstitial sites in the crystal lattice of the  $\text{Bi}_2\text{Se}_3$ . However, with further increase of Ag addition such that increase the  $t_{im}$ , the interstitial site occupation increases attaining saturation, thus the extra Ag atoms occupy the place of Bi sites in the  $\text{Bi}_2\text{Se}_3$  lattice. This leads to a decrease of the optical energy band gap ( $E_g$ ) of the  $\text{Bi}_2\text{Se}_3$  lattice. Namely, increasing the  $t_{im}$  leads to the decrease of the  $E_g$  and increase of the refractive index of  $\text{Bi}_2\text{Se}_3$ <sup>35</sup>. This phenomenon satisfies the definition of the crystallization of the semiconductor chalcogenide: the crystallization of chalcogenide films is accompanied by a decrease in the  $E_g$ .<sup>50-52</sup> The  $\text{Bi}_2\text{Se}_3$  dielectric interlayer can be gradually crystallized by increasing  $t_{im}$ . A sufficient  $t_{im}$  can lead to a complete phase transition between the amorphous and crystalline states. Therefore, the refractive index of  $\text{Bi}_2\text{Se}_3$  can be continuously changed as gradually increasing  $t_{im}$ . Fig. 2 shows the refractive index of amorphous  $\text{Bi}_2\text{Se}_3$  film (not immersed into the  $\text{AgNO}_3$ ,  $t_{im} = 0$  s) as well as those immersed in the  $\text{AgNO}_3$  solution for different periods of immersion time ( $t_{im} = 20, 40$  and  $60$  s), followed by annealing in an Ar atmosphere at 437 K for 1 h, are obtained from the published double-beam spectrophotometer spectroscopy data in [35].

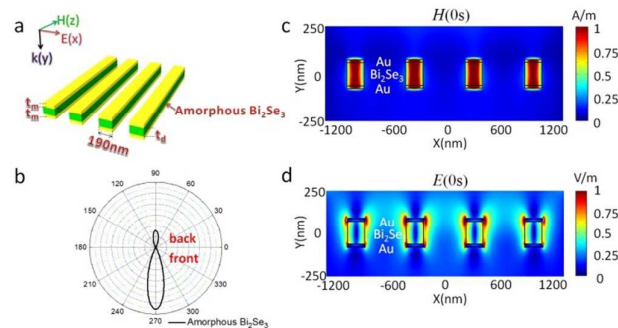
Herein, annealing the  $\text{Bi}_2\text{Se}_3$  film at a temperature of 473 K for 1 h is in aid of diffusing Ag into the  $\text{Bi}_2\text{Se}_3$  layer, in order to effectively crystallize the  $\text{Bi}_2\text{Se}_3$  film. As can be seen, in the N-IR regime  $\text{Bi}_2\text{Se}_3$  shows a pronounced variation in the refractive index during the structural transformation from amorphous ( $t_{im} = 0$  s) to crystalline ( $t_{im} = 20, 40$  and  $60$  s). With these unique properties,  $\text{Bi}_2\text{Se}_3$  is of great interest for actively tunable plasmonics and nanophotonics.



**Fig. 2** Variation of refractive index vs. wavelength, for the amorphous  $\text{Bi}_2\text{Se}_3$  film immersed in  $\text{AgNO}_3$  solution for different periods of time 0, 20, 40 and 60 s.

### Beam-steering using phase-array-like Au/ $\text{Bi}_2\text{Se}_3$ /Au multilayer strips

We firstly study the radiation characteristic of four equally wide MDM resonators with amorphous  $\text{Bi}_2\text{Se}_3$  shown in Fig. 3a. The width of MDM strip is set to  $w_1 = 190$  nm so as to satisfy the interference condition at  $\lambda = 1500$  nm given in Eq. (1) for the first-order resonance ( $m = 1$ ). Fig. 3b presents the radiation pattern of the MDM strips array under normal incidence. Half-power beam widths (HPBW) of the front lobe and back lobe are  $30^\circ$  and  $34^\circ$  respectively, where the front and back lobes are symmetrical to the  $y$  axis owing to the symmetry in the structure around the  $y$  axis. The total magnetic field intensity distribution  $H = \sqrt{|H_x|^2 + |H_y|^2 + |H_z|^2}$  and total electric field intensity distribution  $E = \sqrt{|E_x|^2 + |E_y|^2 + |E_z|^2}$  along the  $\beta$  plane at  $\lambda = 1500$  nm are shown in Fig. 3c and 3d. There is a very strong confinement for both  $E$  and  $H$  field intensity within the dielectric interlayer and the apertures between the MDM strips, indicating the excitation of the strong SPPs.

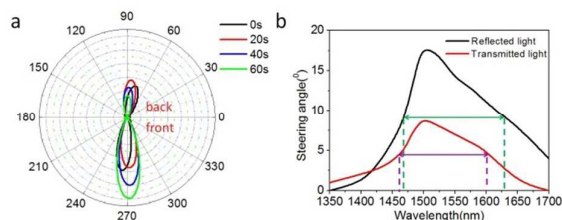


**Fig. 3** a) Four MDM strips with the equal width of 190 nm, where the dielectric layer is amorphous  $\text{Bi}_2\text{Se}_3$ ; b) radiation pattern of the

structure at  $\lambda = 1500$  nm; c) map of the normalized total magnetic field intensity distribution along the plane  $\beta$  at  $\lambda = 1500$  nm; d) map of the normalized total electric field intensity distribution along the plane  $\beta$  at  $\lambda = 1500$  nm.

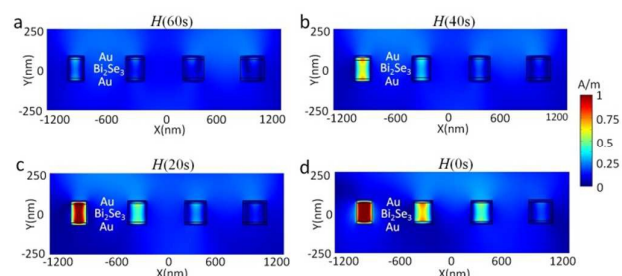
In conventional OPAs, beam is steered by modulating phase shifts between consecutive elements of the array. However, in the array of gradient MDM multilayer strips, the phase shift introduced by each resonator is controlled by changing the refractive index of the dielectric interlayer. The MDM multilayer strips with gradually increasing widths are chosen in order to introduce a different phase shift. Here, we set the width of each MDM multilayer strip as  $w_1 = 190$ ,  $w_2 = 210$ ,  $w_3 = 230$  and  $w_4 = 250$  nm, respectively. In Fig. 4a, the gradient structures with various refractive index of  $\text{Bi}_2\text{Se}_3$  (shown in Fig. 2) are simulated to investigate the effect of the phase change of  $\text{Bi}_2\text{Se}_3$  on the beam-steering angles. Compared to the amorphous non-gradient MDM multilayer strips in Fig. 3, the reflected beam (back lobe) and transmitted beam (front lobe) of the amorphous gradient structure (shown in black solid line in Fig. 4a,  $t_{im} = 0$  s) are deflected by  $18^\circ$  and  $6^\circ$  at  $\lambda = 1500$  nm, respectively. These deflection angles are due to the different phase shift introduced by increasing the width of the element. The HPBWs are  $29^\circ$  and  $34^\circ$  for the front and back lobes, respectively. Active beam-steering can be achieved by switching the phase of  $\text{Bi}_2\text{Se}_3$  dielectric interlayer from amorphous to crystalline in the gradient structure. It shows that a continuously angular steering of  $9^\circ$  from  $264^\circ$  to  $273^\circ$  for the transmitted beam (front lobe), as well as  $18^\circ$  from  $72^\circ$  to  $90^\circ$  for the reflected beam (back lobe) at  $\lambda = 1500$  nm are obtained by increasing the  $t_{im}$ . Particularly, a significant angular steering of  $18^\circ$  for the back lobe has a potential of controlling the wavefront of the reflected beam for the possible applications of free-space optical inter/intra chip interconnects.

Moreover, the HPBW only has an increment of  $2^\circ$  for the front lobe and decrement of  $1^\circ$  for the back lobe with the phase transition between amorphous and crystalline. Therefore, the HPBWs are almost independent with the refractive index variation in the  $\text{Bi}_2\text{Se}_3$ . Fig. 4b shows the beam-steering angles ( $\Delta\vartheta_{max}$ ) of both reflected and transmitted light against the central wavelength of the incident light as completely transiting the  $\text{Bi}_2\text{Se}_3$  from the amorphous ( $t_{im} = 0$  s) to crystalline ( $t_{im} = 60$  s). As can be seen, a big value of beam-steering angle of more than  $9^\circ$  ( $4.5^\circ$ ) for the reflected wave (transmitted wave) can be maintained across a full width half maximum (FWHM) of  $160$  nm ( $140$  nm) in the  $1350$  nm– $1700$  nm wavelength.



**Fig. 4** 3D FDTD simulation of (a) front lobes and back lobes of the phased-array-like MDM structure with  $w_1 = 190$  nm,  $w_2 = 210$  nm,  $w_3 = 230$  nm and  $w_4 = 250$  nm at normal incidence for different  $t_{im}$  of 0, 20, 40 and 60 s; b) the steering angles  $\Delta\vartheta_{max}$  of both reflected and transmitted light against the central wavelength of the incident light as completely transiting the  $\text{Bi}_2\text{Se}_3$  from the amorphous ( $t_{im} = 0$  s) to crystalline ( $t_{im} = 60$  s).

The mechanism of the continuously active beam-steering is based on the variation in the localization of the incident wave between the MDM multilayer strips.<sup>14</sup> Therefore, in order to observe this underlying mechanism, it is instructive to examine the patterns of the  $H$  field intensity distribution for the different phases of  $\text{Bi}_2\text{Se}_3$  along the cross sectional plane,  $\beta$ , which is shown in Fig. 1a. As can be seen in Fig. 5, the SPPs resonance indicated by  $H$  field intensity at  $\lambda = 1500$  nm gradually increases as decreasing  $t_{im}$  since the refractive index of  $\text{Bi}_2\text{Se}_3$  is decreased with  $t_{im}$  shown in Fig. 2. The  $H$  field intensity moves towards wider strips (larger  $w$ ) to satisfy Eq. (1) when  $t_{im}$  is decreased (namely, the refractive index of  $\text{Bi}_2\text{Se}_3$  is decreased), hence leading to the continuously active beam-steering; the largest deflection angle of  $18^\circ$  in the amorphous gradient structure ( $t_{im} = 0$  s) is due to the fact that SPPs resonance shifts towards wider resonators where the  $\text{Bi}_2\text{Se}_3$  exhibits the smallest refractive index at  $\lambda = 1500$  nm. Here, the  $H$  field intensities are normalized to the maximum intensities of the  $H$  field in the amorphous non-gradient structure shown in Fig. 3.



**Fig. 5** A map of the normalized total magnetic field intensity distribution ( $H$ ) along the  $\beta$  plane at  $\lambda = 1500$  nm: a)  $t_{im} = 60$  s; b)  $t_{im} = 40$  s; c)  $t_{im} = 20$  s; d)  $t_{im} = 0$  s.

## Discussion

Here, we propose a concept of chemically controlled beam-steering using an array of gradient MDM multilayer strips based on  $\text{Bi}_2\text{Se}_3$ . The strips with increasing width are chosen to introduce the different phase shift. The transmitted and reflected lights at  $\lambda = 1500$  nm can be continuously steered from  $264^\circ$  to  $273^\circ$  and  $72^\circ$  to  $90^\circ$  by switching between the amorphous and crystalline state of  $\text{Bi}_2\text{Se}_3$ , where the phase transition is achieved by immersing  $\text{Bi}_2\text{Se}_3$  film in  $\text{AgNO}_3$  solution for different periods of time. Moreover, the steering angle is stable for spectral changes with a FWHM of  $140$  nm for the transmitted beam and  $160$  nm for the reflected beam around  $\lambda = 1500$  nm. It is expected that chemically controlled tunable plasmonic antenna array based on  $\text{Bi}_2\text{Se}_3$  will promise

an enormous variety of breakthroughs in technological outcomes and lead to new nanophotonics applications.

## Acknowledgements

We acknowledge the financial support from National Natural Science Foundation of China (Grant No. 61172059 and 51302026), Ph.D Programs Foundation of Ministry of Education of China (Grant No. 20110041120015), Postdoctoral Gathering Project of Liaoning Province (Grant No. 2011921008), and The Fundamental Research for the Central University (Grant No. DUT14YQ109).

## Notes and references

- J. Sun, E. Timurdogan, A. Yaacobi, E. S. Hosseini and M. R. Watts, *Nature*, 2013, **493**, 195.
- S. B. Raghunathan, H. F. Schouten, W. Ubachs, B. Ea Kim, C. H. Gan and T. D. Visser, *Phys. Rev. Lett.*, 2013, **111**, 153901.
- D. Kwong, A. Hosseini, J. Covey, Y. Zhang, X. Xu, H. Subbaraman and R. T. Chen, *Opt. Lett.*, 2014, **39**, 941.
- S. Young and B. Schwarz, *Opt. Photonics.*, 2011, **12**, 7.
- B. Schwarz, *Nat. Photonics*, 2010, **4**, 429.
- A. Yaacobi, J. Sun, M. Moresco, G. Leake, D. Coolbaugh and M. R. Watts, *Opt. Lett.*, 2014, **39**, 4575.
- C. T. DeRose, R. D. Kekatpure, D. C. Trotter, A. Starbuck, J. R. Wendt, A. Yaacobi, M. R. Watts, U. Chettiar, N. Engheta and P. S. Davids, *Opt. Express.*, 2013, **21**, 5198.
- L. Zou, M. Cryan and M. Klemm, *Opt. Express.*, 2014, **22**, 24142.
- L. Zou, W. Withayachumnankul, C. Shah, A. Mitchell, M. Bhaskaran, S. Sriram and C. Fumeaux, *Opt. Express*, 2013, **21**, 1344.
- A. Pors, O. Albrektsen, I. P. Radko and S. I. Bozhevolnyi, *Sci. Rep.*, 2013, **3**, 2155.
- Y. Yang, W. Wang, P. Moitra, I. I. Kravchenko, D. P. Briggs and J. Valentine, *Nano Lett.*, 2014, **14**, 1394.
- Y. Yifat, M. Eitan, Z. Iluz, Y. Hanein, A. Boag and J. Scheuer, *Nano Lett.*, 2014, **14**, 2485.
- Z. H. Jiang, L. Lin, J. A. Bossard and D. H. Werner, *Opt. Express*, 2013, **21**, 31492.
- E. Battal and A. K. Okyay, *Opt. Lett.*, 2013, **38**, 983.
- K. V. Acoleyen, H. Rogier and R. Baets, *Opt. Express*, 2010, **18**, 13655.
- J. K. Doylend, M. J. R. Heck, J. T. Bovington, J. D. Peters, L. A. Coldren and J. E. Bowers, *Opt. Express*, 2011, **19**, 21595.
- D. de Ceglia, M. A. Vincenti and M. Scalora, *Opt. Lett.*, 2012, **37**, 271.
- D. C. Adams, S. Thongrattanasiri, T. Ribaudou, V. A. Podolskiy and D. Wasserman, *Appl. Phys. Lett.*, 2010, **96**, 201112.
- P. Y. Yu and M. Cardona, in *Fundamentals of semiconductors*, ed. H.E.Stanly and W.T.Rhodes. Springer, Berlin, 4th edn., 2004.
- T. Cao, S. Wang and W. X. Jiang, *RSC Adv.*, 2013, **3**, 19474.
- L. H. Shao, M. Ruther, S. Linden, S. Essig, K. Busch, and J. Weissmüller and M. Wegener, *Adv. Mater.*, 2010, **22**, 5173.
- M. Z. Hasan and C. L. Kane, *Rev. Mod. Phys.*, 2010, **82**, 3045.
- X. L. Qi and S. C. Zhang, *Rev. Mod. Phys.*, 2011, **83**, 1057.
- J. Qi, H. Liu and X. C. Xie, *Phys. Rev. B*, 2014, **89**, 155420.
- J. Y. Ou, J. K. So, G. Adamo, A. Sulaev, L. Wang and N. I. Zheludev, *Nat. Commun.*, 2014, **5**, 5139.
- L. Lu, J. D. Joannopoulos and Marin Soljačić, *Nat. Photonics*, 2014, **8**, 821.
- X. Zhang, J. Wang and S. C. Zhang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 245107.
- D. Hsieh, D. Qian, L. Wray, Y. Xia, Y. S. Hor, R. J. Cava and M. Z. Hasan, *Nature*, 2008, **452**, 970.
- D. Hsieh, *Science*, 2009, **323**, 919.
- D. Hsieh, Y. Xia, D. Qian, L. Wray, J. H. Dil, F. Meier, J. Osterwalder, L. Patthey, J. G. Checkelsky, N. P. Ong, A. V. Fedorov, H. Lin, A. Bansil, D. Grauer, Y. S. Hor, R. J. Cava and M. Z. Hasan, *Nature*, 2009, **460**, 1101.
- Y. Xia, D. Qian, D. Hsieh, L. Wray, A. Pal, H. Lin, A. Bansil, D. Grauer, Y. S. Hor, R. J. Cava and M. Z. Hasan, *Nat. Phys.*, 2009, **5**, 398.
- D. Hsieh, Y. Xia, D. Qian, L. Wray, F. Meier, J. H. Dil, J. Osterwalder, L. Patthey, A. V. Fedorov, H. Lin, A. Bansil, D. Grauer, Y. S. Hor, R. J. Cava and M. Z. Hasan, *Phys. Rev. Lett.*, 2009, **103**, 146401.
- Y. L. Chen, J. G. Analytis, J.-H. Chu, Z. K. Liu, S.-K. Mo, X. L. Qi, H. J. Zhang, D. H. Lu, X. Dai, Z. Fang, S. C. Zhang, I. R. Fisher, Z. Hussain and Z.-X. Shen, *Science*, 2009, **325**, 178.
- Z. H. Pan, E. Vescovo, A. V. Fedorov, D. Gardner, Y. S. Lee, S. Chu, G. D. Gu and T. Valla, *Phys. Rev. Lett.*, 2011, **106**, 257004.
- G. B. Sakr, I. S. Yahia, G. M. El-Komy and A. M. Salem, *Surf. Coat. Technol.*, 2011, **205**, 3553.
- R. Ortuno, C. Garcia-Meca, F. J. Rodriguez-Fortuno, J. Marti and A. Martinez, *Phys. Rev. B*, 2009, **79**, 075425.
- S. I. Bozhevolnyi and T. Sondergaard, *Opt. Express*, 2007, **15**, 10869.
- G. D. Moon, S. Ko, Y. Min, J. Zeng, Y. Xia and U. Jeong, *Nano Today*, 2011, **6**, 186.
- D. Nataraj, K. prabakar, S. K. Narayandass and D. Mangalaraj, *Cryst. Res. Technol.*, 2000, **35**, 1087.
- R. E. Simpson, P. Fons, A. V. Kolobov, T. Fukaya, M. Krbal, T. Yagi and J. Tominaga, *Nat. Nanotechnol.*, 2011, **6**, 501.
- C. García-Meca, R. Ortuño, F. J. Rodríguez-Fortuño, J. Martí and A. Martínez, *Opt. Lett.*, 2009, **34**, 1603.
- S. Zhang, W. J. Fan, K. J. Malloy, S. R. J. Brueck, N. C. Panoiu and R. M. Osgood, *Opt. Express*, 2005, **13**, 4922.
- G. Dayal and S. A. Ramakrishna, *Opt. Express*, 2012, **20**, 17503.
- J. Q. Wang, C. Z. Fan, J. N. He, P. Ding, E. J. Liang, and Q. Z. Xue, *Opt. Express*, 2013, **21**, 2236.
- M. Tamura and H. Kagata, *IEEE Trans. Microwave Theory Tech.*, 2010, **58**, 3954.
- R. Gordon, *Phys. Rev. B*, 2006, **73**, 153405.
- P. B. Johnson and R. W. Christy, *Phys. Rev. B*, 1972, **6**, 4370.
- J. D. Jackson, *Classical Electrodynamics*, Second Edition (John Wiley & Sons), 1975, 401.
- T. Cao, Y. Zou, A. M. Adawi, and M. J. Cryan, *Opt. Express*, 2014, **22**, 22699.
- M. Digiulio, D. Manno, R. Rella, P. Siciliano and A. Tepore, *Solar Energy Mater.*, 1987, **15**, 209.
- M. M. Hafiz, M. A. Abdel-Rahim and A. A. Abu-Sehly, *Physica B*, 1998, **252**, 207.
- M. M. Hafiz, O. El-Shazly and N. Kinawy, *Appl. Surf. Sci.* 2001, **171**, 231.