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Epoxy/graphene nanocomposites – processing and properties: a review

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Abstract

Graphene has recently attracted significant academic and industrial interest because of its excellent performance in mechanical, electrical and thermal applications. Graphene can significantly improve physical properties of epoxy at extremely small loading when incorporated appropriately. Herein, the structure, preparation and properties of epoxy/graphene nanocomposites are reviewed in general, along with detailed examples drawn from the key scientific literature. The modification of graphene and the utilization of these materials in the fabrication of nanocomposites with different processing methods have been explored. This review has been focused on the processing methods and mechanical, electrical, thermal, and fire retardant properties of the nanocomposites. The synergic effects of graphene and other fillers in epoxy matrix have been summarised as well.

Keywords

Epoxy; Graphene; Processing; Composites; Mechanical; Thermal; Synergic

1. Introduction

Materials play key role in every field of technology such as aeronautics, electronics, energy, health, sensors, etc¹. It is important to continuously update existing materials and develop new materials with improved properties and multi-functionalities, so they can be exploited for advanced applications². Compared to traditional composite materials, nanocomposites exhibit extraordinary properties because of the exceptionally high surface to volume ratio of the nanofiller and/or its exceptionally high aspect ratio³. Polymer nanocomposites combine the functionalities of polymer matrices, such as low cost, easy processability⁴, with the unique features of the inorganic nanoparticles such as high aspect ratio, excellent toughness and strength and other properties like electrical and thermal conductivities⁵. In the past few years, polymer nanocomposites with enhanced optical, mechanical, electrical, thermal, and fire retardant properties have been developed⁶⁻¹⁰. However, nanofillers used in these materials have strong tendency to agglomerate which would cause inhomogeneous dispersion of nanofillers in matrices¹¹, and reduction in mechanical and thermal properties of these nanocomposites. The optimum enhancement in the properties of polymer matrices can only be achieved if the nanofillers are uniformly dispersed¹². Attaining the optimum dispersion is one of the main challenges for processing of nanocomposites and therefore it is essential to review the current processing techniques used for preparing nanocomposites.

However, there has been no review article dealing specifically with epoxy/graphene nanocomposites, which is the subject of this paper. Mechanical, electrical, thermal and fire retardant properties of epoxy/graphene nanocomposites have been reviewed; besides that, processing methods and properties of nanomaterials have been correlated. Furthermore, some of the listed points have been highlighted in this paper are:

1. Summarised a new method of epoxy/graphene nanocomposites preparation - resin impregnation, which impregnating epoxy into a graphene filter cake without pre-mixing.
2. Summarised the synergic effects of graphene and other fillers in epoxy matrix.
3. Summarised the reason why thermal stability decreased with the incorporation of graphene.

To the best of our knowledge, those points haven't been seen in other reviews and we believe this review covers most of the important publications relating to the processing and properties of epoxy/graphene nanocomposites to date.

2. Epoxy

Discovered in 1936 by Dr. Castan of Switzerland and Dr. Greenlee of USA, epoxy based materials are used widely because of their superlative mechanical properties, thermal stability, solvent resistance and ease of processing¹³. Epoxies are one of the most adaptable and widely sold high performance material in the world¹⁴, some of the applications of epoxy and its nanocomposites include aerospace, automotive, marine, sports materials, construction, structures, electrical and electronic systems, biomedical devices, thermal management systems, adhesives, paints and coatings, industrial tooling and other general consumer products¹⁵. Because of its versatile nature, epoxy is replacing many conventional materials, e.g. epoxy based materials have already replaced wood in majority of the boats and various sports goods.

Epoxy resins are thermosetting polymers and defined as a molecule containing more than one epoxide groups, as shown in Fig. 1.

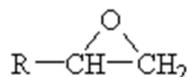


Fig. 1: Molecular Structure of epoxide group.

The curing process is a chemical reaction in which the epoxide groups in epoxy resin reacts with a hardener (curing agent) to form a highly crosslinked, three-dimensional network¹⁶. There are wide varieties of curing agents available for epoxy based materials. Depending on the chemical formulation of the hardeners, epoxy resins can be cured at temperatures range from 5 to 150°C¹⁷. However, the epoxy materials for varying engineering applications are often limited by their brittle nature and poor electrical, thermal properties¹⁸. A simple solution to overcome this problem is to modify the matrix molecular structure or add compatible fillers. For example, incorporation of inorganic nanofillers has been shown to be a very efficient strategy to increase the performance of the material¹⁹.

3. Graphene

3.1 History and properties

Since the historical observation of single layer graphene by Andre Geim and Kostya Novoselov in 2004²⁰, this atomically thin carbon sheet has received ever-increasing attention and become a rapidly rising star on the horizon of materials science²¹. For example, recently the European Commission has financed a 10-year research initiative, the European Graphene Flagship, which provides 1 billion Euro in funding and involves more than 140 academic and commercial institutions in 23 countries²².

Graphene exhibits many specific and useful properties such as large surface area (2630 m²/g)²³, excellent thermal conductivity (5000 W/m·s)²⁴, very high Young's modulus (1 Tpa)²⁵, high value of white light transmittance as to 97.7%²⁶, exceptionally high room-temperature electron mobility of

$2.5 \times 10^5 \text{ cm}^2/\text{V}\cdot\text{s}^{27}$. These fascinating properties have attracted extensive research interest in recent years with ever-increasing scientific and technological impetus.

For example, as a conductive nanomaterial, graphene can be used for printed electronics beyond conventional silicon based technologies²⁸. For energy storage, Yang *et al.*²⁹ prepared a supercapacitor with a capacitance of 200-300 F/g. Kim *et al.*³⁰ used graphene as a transparent electrode and fabricated an organic photovoltaic devices; Prasai *et al.*³¹ incorporated graphene into organic coatings significantly enhanced its corrosion resistance. Today, impressive achievements have been made at the cross-section of nanotechnology and various applications by employing the specific properties of graphene³².

3.2 Fabrication

Efforts to exfoliate graphite down to its ultimate constituent can be dated back to 1960s. Fernandez *et al.*³³ extracted millimetre-sized graphene sheets (as thin as 5 nm, about 15 layers) from graphite crystals by micromechanical exfoliation for the very first time. However, it was not until 2004, Andre Geim and Kostya Novoselov²⁰ isolated individual graphene layers by repeatedly cleaving a graphite crystal with a scotch tape to its limit. This led to the creation of a wonder two dimensional (2D) material, marking the onset of successful fabrication of graphene.

Micromechanical exfoliation, the top-down method, is a simple peeling process as shown in Fig. 2. Similarly, ultrasonication will also produce thin graphene sheets³⁴. Currently, exfoliation of bulk graphite is the most commonly used method for the mass production of small graphene sheets³⁵. This can be through direct exfoliation in a liquid, with or without the use of surfactant³⁶, or in the solid state by edge functionalization³⁷, or by first inserting a chemical species between the graphene layers in graphite to weaken their interaction and then followed by thorough exfoliation³⁸.



Fig. 2: Scotch tape method of graphene synthesis from graphite block³⁹. Reproduced with permission from ref. 39.

Bottom-up approaches have also been developed such as Chemical Vapour Deposition (CVD)⁴⁰. In typical CVD process, a substrate is exposed to volatile precursors in a reaction chamber, the precursors react and/or decompose on the substrate surface to produce the desired deposit⁴¹. For graphene production, silicon or a transition metal often serves as the substrate, the CVD chamber is vacuumed and heated, under the high temperature and the effect of catalyst, hydrocarbon gases are induced and decomposed. This process deposits a spread of carbon atoms onto the surface of the substrate, thus forming the graphene layers⁴².

Another advanced method is the chemical reduction of exfoliated graphene oxide, which is an economical and very practical approach to synthesise graphene⁴³. This process takes the advantage of π - π interactions of graphene oxide and other molecules such as hydrazine (one of the most effective reductive agents), which can effectively return graphene oxide to its original state⁴⁴. This method maintains graphene's electrical conductivity, flatness and optical properties, but it's not as same as pristine graphene and still contains some significant oxygen groups and a few irreversible lattice defects⁴⁵.

There are a number of other growth methods, some of these methods have certain advantages and should be investigated further, such as arc discharge method⁴⁶, template route method⁴⁷, electrochemical synthesis of graphene⁴⁸ and total organic synthesis of graphene⁴⁹. Many studies have been directed towards developing techniques to create single layer graphene, however, to date, scalable production of single layer graphene is still at exploration stage and there is no mature method to produce good quality graphene in mass quantity⁵⁰. In general, mechanical exfoliation, CVD, chemical reduction, epitaxial growth of graphene are among the most notable techniques in graphene production⁵¹.

3.3 Graphene oxide (GO) and functionalization of graphene

3.3.1 GO

GO is obtained from the exhaustive oxidation and exfoliation of graphite, and contains a range of oxygen functional groups with specific chemistry⁵². It is generally produced by the treatment of graphite using strong mineral acids and oxidizing agents, typically via treatment with KMnO_4 and H_2SO_4 as in the Hummers⁵³ method, or KClO_3 (or NaClO_3) and HNO_3 as in the Staudenmaier⁵⁴ or Brodie⁵⁵ methods, or some variation of these methods. There is no unambiguous model to describe the exact structure of GO because there is no single definitive analytical technique available to characterize this material. However, it is generally accepted that the carboxylic groups are mainly located at the edge, while the rest of functional groups (hydroxyl, epoxide, etc.) are present in highest concentration in the basal planes of the graphene layers⁵⁶. Fig. 3 shows a proposed structure of graphene oxide that is supported by solid-state nuclear magnetic resonance (SSNMR) experiments on ^{13}C -labeled GO.

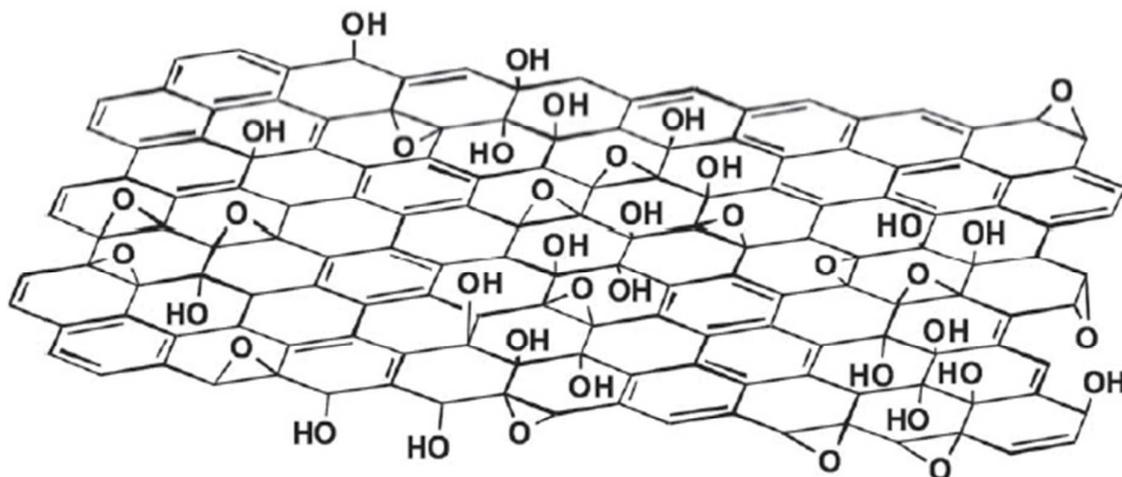


Fig. 3: A proposed schematic (Lerf-Klinowski model) of graphene oxide structure⁵⁷. Reproduced with permission from ref. 57.

The oxygen functional groups on GO surface are polar and renders GO hydrophilic. GO can be dispersed in many solvents, and particularly well in water⁵⁸. In addition, currently the most promising methods for large scale production of graphene are based on exfoliation and reduction of graphene oxide⁵⁹.

3.3.2 Functionalization of graphene

Pristine graphene is unsuitable for intercalation with large species, such as polymer chains, because graphene has a pronounced tendency to agglomerate in a polymer matrices⁶⁰. As observed for other nanofillers, the maximum improvements in final properties can be achieved when the filler is homogeneously dispersed in the matrices and the external load is efficiently transferred through strong polymer/filler interfacial interactions⁶¹. Thus, dispersion and strong interaction between graphene and matrices play important role in the performance of matrices/graphene nanocomposites⁶².

The chemical functionalization of graphene is of significant interest because it can not only improve the solubility and processability but can also enhance the interactions with organic polymers⁶³⁻⁶⁶. The functional groups attached to graphene can be small molecules⁶⁷ or long polymer chains⁶⁸, for which various functionalization approaches have been completed such as covalent and non-covalent functionalization of graphene⁶⁹.

Covalent functionalization is based on covalent linkage between graphene and other functional groups⁷⁰. The structural alteration can take place at the end of the sheets and/or on the surface⁷¹. Covalent functionalization is associated with rehybridization of one or more sp^2 carbon atoms of the carbon network into the sp^3 configuration accompanied by simultaneous loss of electronic conjugation⁷². The covalent modification of graphene can be achieved in four different ways: nucleophilic substitution, electrophilic addition, condensation, and addition⁷³. By conducting an epoxide ring-opening reaction, Yang *et al.*⁷⁴ covalently grafted 1-(3-aminopropyl)-3-methylimidazolium bromide onto the surface of graphene sheets. The modified graphene shows

enhanced solubility in water, N,N-Dimethylformamide (DMF) and dimethyl sulfoxide (DMSO) at various concentrations, formed long-term stable and homogeneous dispersions.

Non-covalent functionalization helps in networking or connecting the molecules without actually forming chemical bonds. However, this process requires the physical adsorption of suitable molecules on the graphene surface⁷⁵. This can be achieved by wrapping molecules around the graphene by forming Van der Waals bonds between functional groups and graphene, such as π - π interactions, electrostatic attraction, adsorption of surfactants and polymer wrapping⁷⁶⁻⁷⁹. For example, Song *et al.*⁸⁰ prepared epoxy/graphene nanocomposites with improved mechanical properties and thermal conductivities by non-covalent functionalization of graphene. The modified graphene shows good dispersibility in acetone, DMF, ethanol, pyridine, methanol, tetrahydrofuran (THF) and water, but only short-term stability in iso-propyl alcohol (IPA), dichlorobenzene (DCB), chloroform, dichloromethane and chlorobenzene. This short-term stability is attributed to the surface functional group 1-pyrenebutyric acid, which is not compatible with these solvents.

4. Epoxy/graphene nanocomposites

Epoxy and its composites are versatile materials for plenty of industrial fields like electrical applications, thermal applications, high-performance nanocomposites in automobiles and aerospace applications but these composites have some limitations as well. As a young rising carbon allotrope, graphene showed a new path to overcome these limitations. The exploration of property enhancement of epoxy/graphene nanocomposites is rapidly advancing as evident in Fig. 4, which shows the dramatically increase in epoxy/graphene nanocomposites research in recent years.

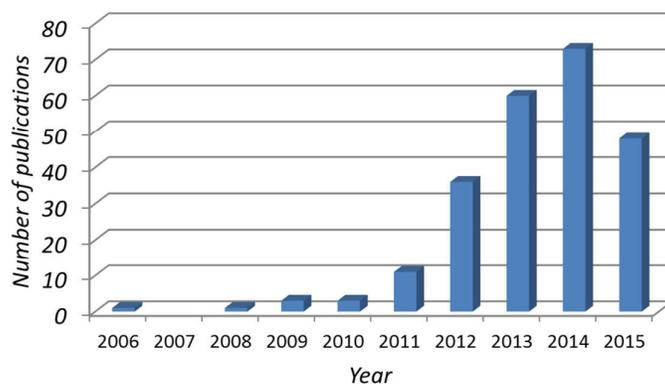


Fig 4: Number of publications returned using “graphene epoxy” as keywords searched “in title” in Web of Science (By 30/06/2015).

4.1 Processing methods

Obtaining a good distribution of the graphene-reinforcement is one of the greatest challenges in the preparation of epoxy/graphene nanocomposites. A well dispersed state ensures availability of maximum surface area of filler, which will affect the neighbouring polymer chains and, consequently, the properties of the whole nanocomposite⁸¹. For epoxy or any other matrices, dispersion significantly depends on the processing techniques. Significant research has been carried out on the manufacturing techniques for achieving a homogeneous and well-dispersed system⁸²⁻⁸⁸. The commonly used methods for epoxy/graphene nanocomposites are solution mixing, and recently, a newly emerged method as epoxy impregnation and will be discussed here.

4.1.1 Solvent processing

The simplest and most widely used method for processing epoxy/graphene nanocomposites is to take advantage of the presence of functional groups attached on the graphene surface which enable the direct dispersion of graphene in water and many organic solvents. This contributes to strong physical or chemical interaction between the functionalized graphene and polymeric matrices⁸⁹. A number of studies explain how the surface modification of graphene has been done by adding various functional groups such as amine⁹⁰, organic phosphate⁹¹, silane⁹², plasma⁹³ etc.

Functionalized graphene is normally dispersed in a suitable solvent by, for example, bath sonication, then mixed with epoxy resin, and then solvent is evaporated in a controlled condition⁹⁴. The guiding principle is to select solvents compatible with the functional groups on the surface of graphene, and the functional group should be compatible with the epoxy resin as well⁹⁵. To achieve better dispersion of functionalized graphene, many solvents have been investigated. Rafiee *et al.*⁹⁶ prepared epoxy/graphene nanocomposites by dispersing graphene platelets in acetone by tip sonication, mixed graphene/acetone solution with epoxy resin and finally acetone was removed by heating the mixture to 70°C. The prepared nanocomposites showed enhanced mechanical properties and resistance to fatigue crack growth at low graphene concentration (0.1 wt%). Fang *et al.*⁹⁷ dispersed graphene in DMF under bath sonication and modified graphene with amine, which provided a mechanical adhesion at the graphene-epoxy interface. The nanocomposites showed improved load transfer efficiency between graphene nanosheets and the matrix, accompanied by the enhanced dissipation capacity of nanocomposites for strain energy during fracture. Tang *et al.*⁹⁸ investigated the influence of reduced graphene oxide (rGO) dispersion on the mechanical properties of epoxy resin. They found that with the assistance of ball milling in ethanol solution, the blends showed higher dispersibility, which resulted in higher strength and fracture toughness of epoxy resin as well as improved glass transition temperature (T_g) and electrical conductivity. In addition, they also found that the highly dispersed rGO resulted in much more tortuous and fine river-like structures on the fractured surface. This consumes fracture energy in comparison with the poorly dispersed RGO or in other words improves the fracture toughness of the material. Chatterjee *et al.*⁹⁹ investigated the reinforcements of mechanical and thermal properties of a functionalized graphene filled epoxy nanocomposites. The amine functionalized expanded graphene nanoplatelets (GNP) were dispersed within epoxy resins using high-pressure processor followed by three roll milling in the solvent of acetone. The resulting nanocomposite exhibited significant improvements in mechanical properties and thermal conductivity indicating a favourable interaction at graphene/epoxy interface. Table 1 shows summary of representative investigations on the solvent processing.

Table 1. Different graphene dispersion method and the properties enhancement

reference	year	solvent	filler	disperse method	% increase in σ	% increase in E	% increase in K_{IC}	% increase in G_{IC}	a^x increase in κ	folds increase in λ	increase in T_g (°C)	increase in T_d (°C)
99	2014	acetone	m-GO	bath sonic + mechanical mix	18.8	42.2	85.7					
100	2014	acetone	m-GO	bath sonic + mechanical mix + ball mill	63.2	12					1.6	
101	2014	THF	m-G	mechanical mix	-11.1	21.5	103	236.1			11.7	
102	2014	THF	m-G	bath sonic + mechanical mix	28	23.6	188.3	597	10			
103	2014	acetone	m-GO	bath sonic + mechanical mix + ball mill	47.8	9.5	39	85.7				
104	2014	acetone	rGO	bath sonic + mechanical mix	46	10.9	63.3					
105	2014	acetone	m-GO	bath sonic + mechanical mix + ball mill	61.4	16.5	33				3.9	30
106	2014	DCM water	m-GO	mechanical mix	31.8						18	
81	2014	DMF	m-G	bath sonic	46.2	31.7	127.2				4.9	4.8
107	2014	DCM	m-GO	bath sonic + mechanical mix	47.3	21.7						
108	2013	acetone	G	bath sonic + mechanical mix	20.2	19.3					11.4	
109	2013	acetone	GO	bath sonic + mechanical mix	14.3	24					5.1	
110	2013	DMF	m-G	bath sonic	24.4	14.4			7		9.3	4
111	2013	acetone	G	bath sonic + mechanical mix	31.8	34.1	75.3		10		7.6	-2
112	2013	acetone	GO	bath sonic + mechanical mix		11	76.9				11	
113	2013	DCM	m-G	bath sonic	21.2	43.1			10	11		
114	2013	water	rGO	mechanical mix	468	68.7			10		19.6	19
115	2013	water acetone	m-GO	bath sonic + mechanical mix	16.5	32	19.6	8.3				
116	2013	ethanol	m-G	bath sonic + mechanical mix	47.9	103.3						
117	2013	ethanol	m-GO	mechanical mix + ball mill	57.4	8.2						
98	2013	ethanol	rGO	bath sonic + ball mill	7.5	6.1	51.7		3		11.1	
118	2012	THF acetone	m-G	bath sonic + mechanical mix	-0.23	0.267	124	292.8			12.2	
119	2012	acetone	GO	mechanical mix + 3-roll calendaring	12.3	10	60	116			1.8	
120	2012	DMF	m-G	bath sonic	97.2	11.4				-9	8	
121	2012	MEK	G							26		
122	2011	THF	m-G	bath sonic + mechanical mix	-17.1	21.5	122	205			12.4	
123	2011	ethanol	m-Gi	bath sonic + mechanical	29.5	42						

				mix								
⁹⁶	2009	acetone	G	tip sonic + shear mix	41.8	29.8	62.5	128				
¹²⁴	2009	water	rGO						8			
⁹²	2008	water ethanol	m-Gi						7	29		

Abbreviations
in table:

G: graphene

GO: graphene oxide

m-G: modified graphene

m-GO: modified graphene oxide

rGO: reduced graphene oxide

m-Gi: modified graphite

bath sonic: bath sonication

tip sonic: tip sonication

σ : tensile strength

E : elastic modulus

K_{IC} : fracture toughness

G_{IC} : energy released

a^x increase: increase in order of magnitude

κ : electrical conductivity

λ : thermal conductivity

T_g : glass transition temperature

T_d : thermal degradation temperature

As can be seen from table 1, a wide range of solvents have been used for the dispersion of graphene, such as THF, DMF, acetone, ethanol, water, dichloromethane (DCM), methyl ethyl ketone (MEK), etc. Dispersion techniques like tip sonication, bath sonication, mechanical mix, shear mix, and three roll calendaring have been widely adopted for homogenous dispersion and most of these methods showed good results.

4.1.2 Resin impregnation

This method refers to impregnation of epoxy resin into the as-prepared graphene filter cake. It has not been widely reported in the literature until recently as a method for preparing polymer nanocomposites. Im *et al.*¹²⁵ prepared a 60 wt% nanocomposite material by using this method for the very first time in 2012. They suspended GO particles in H₂O under ultrasonication and then the prepared mixture was poured into a glass mould which was placed on a silicon oxide membrane. The mixture poured into the glass mould was filtered via vacuum filtration. After filtration, the filter cake which was peeled off from the SiO₂ membrane was annealed under heating to remove the residual water. Finally, the epoxy containing the curing agent was dropped onto the filter cake and cured under heating. This method infuses epoxy resin into the graphene sheet by capillarity driven wetting force and appropriate for fabricating highly concentrated nanocomposites with reasonably high mechanical properties.

Similar approach has been used by Li *et al.*¹²⁶ to fabricate a 11.84 wt% epoxy/graphene nanocomposite. They first dispersed graphene platelets in the mixture of ethanol and water by ultrasonication and then removed the solvent by vacuum filtration. During the filtration process, self-assembly of the aligned graphene occurred (Fig. 5), after that they immersed this aligned graphene into epoxy monomer and curing agent. By this method, they prepared a nanocomposite with aligned multilayer graphene in epoxy matrix. The nanocomposite showed a thermal conductivity highly as 33.54 W/(m·K) at 90 °C. This remarkable improvement in thermal conductivity was due to the unique alignment structure formed during processing.

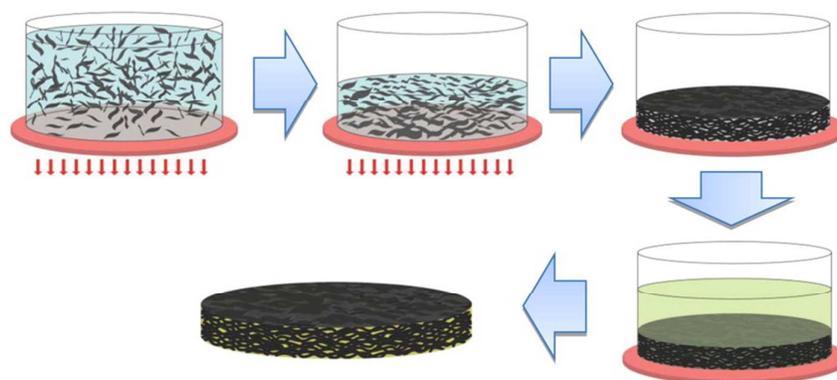


Fig. 5: Schematic diagram of the preparation of aligned epoxy/graphene¹²⁶. Reproduced with permission from ref. 126.

Low filler percentage nanocomposites could also be prepared by this method. Jia *et al.*¹²⁷ reported the preparation of a 0.1 wt% epoxy/graphene nanocomposite by impregnation of epoxy resin into a three dimensional (3D) graphene-nickel (Ni) foam via chemical vapour deposition, followed by curing of the polymer and etching of the Ni template. This nanocomposite with 0.1 wt% graphene delivered excellent fracture toughness, and the glass transition temperature increased 31°C compared to solid epoxy. More than that, they reported this 3D interconnected graphene network serves as fast channels for charge carriers, giving rise to a remarkable electrical conductivity of the nanocomposite.

4.1.3 Other methods

The most widely used method to prepare epoxy/graphene nanocomposites is through solvent processing. However, some derivative methods have also been adopted. Martin *et al.*¹²⁸ dispersed graphene in epoxy monomer by mechanical mixing. The mixture was then mixed with photo initiator and cured by UV irradiation. They reported an enhancement in thermal and mechanical properties of the nanocomposite as a result of UV curing. Similarly, Sangermano *et al.*¹²⁹ prepared UV cured epoxy/graphene nanocomposites as well and showed enhanced properties. Yu *et al.*¹³⁰ used hot press in the curing procedure to fabricate the epoxy/graphene nanocomposite which showed several folds of increments in thermal conductivity. However, dispersing graphene in epoxy matrix without using solvent is likely to be less efficient. Hsu *et al.*¹³¹ mixed graphene, epoxy monomer and curing agent all together using three roll milling at room temperature. Uniformly dispersion of graphene was hindered by the high viscosity of epoxy resin, therefore, mixing without solvent might be considered as a less effective dispersion strategy.

4.2 Properties of nanocomposites

Due to the recent developments in graphene and epoxy/graphene nanocomposites, the literature on this subject is still in its early stage but growing rapidly. However, some interesting studies have already been reported and will be discuss here to illustrate the potential of these new nanocomposites.

4.2.1 Morphology

As property enhancements strongly correlate with nanocomposite microstructure¹³², effective characterization of morphology is important to establish structure-property relationships for these materials.

Transmission electron microscope (TEM) images of these nanocomposites can provide direct observation of dispersed multilayer graphene platelets. Thicker platelets typically shows adequate contrast against the epoxy matrix to be imaged without staining, whereas single layer platelets may be difficult to observe directly by TEM¹³³. Studies on layered nanofillers based nanocomposites have suggested the existence of three general states of dispersion on short length scales: stacked, intercalated, or exfoliated, as shown in Fig. 6.

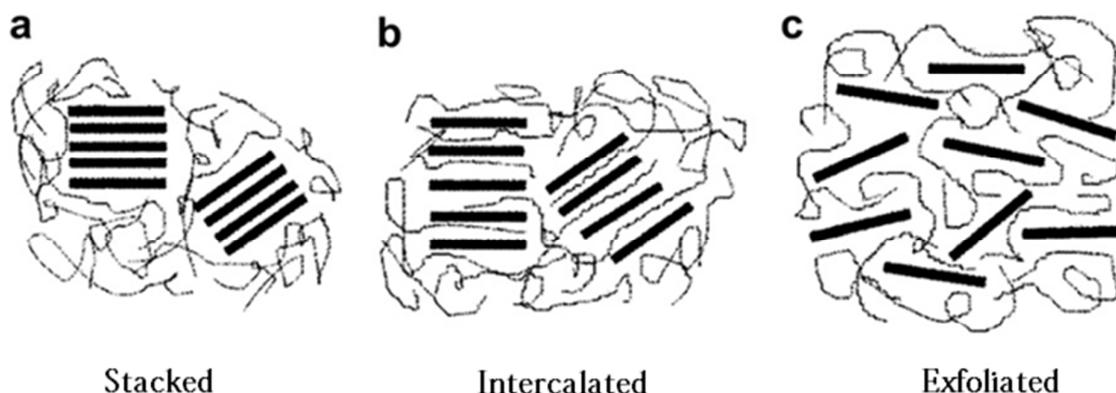


Fig. 6: Schematic showing three morphological states for layered nanofillers based nanocomposites¹³⁴: (a) stacked, (b) intercalated, (c) exfoliated. Reproduced with permission from ref.134.

TEM is the most common method for assessing the state of dispersion. Immiscibility of the phases and/or insufficient exfoliation of the graphite or graphene platelet prior to mixing with epoxy can result in large agglomerates consisting of stacked graphene sheets when observed by TEM. Fig. 7 shows an accurate measurement of the number of graphene layers in epoxy matrix.

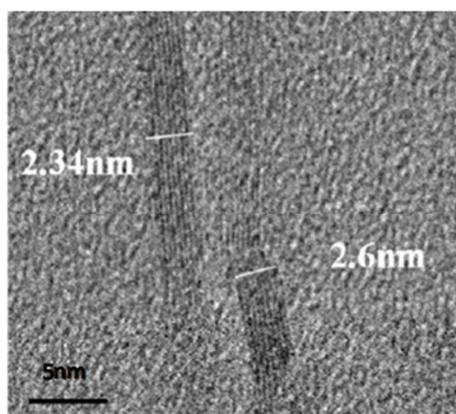


Fig. 7: TEM image of layered graphene in epoxy matrix¹³⁵. Reproduced with permission from ref. 135.

The drawback of TEM is only a small area of the material could be observed, so cross sectional analysis with scanning electron microscopy (SEM) has also been used to evaluate dispersion of graphene as well as to examine the surface for filler pull-out, which could give insight into the strength of interfacial adhesion¹³⁶. However, SEM image cannot resolve the degree of exfoliation of

the platelets and is therefore best utilized combined with TEM. Furthermore, atomic force microscope (AFM) and the corresponding height profile graph is an important technique to characterize the pristine or functionalized graphene. The AFM study could give the length and thickness of graphene sheets along with morphology (Fig. 8). For AFM study, the sample is prepared by dispersing graphene in water or solvents and drop casting on a freshly cleaved mica surface. The dried sample is then observed through the instrument.

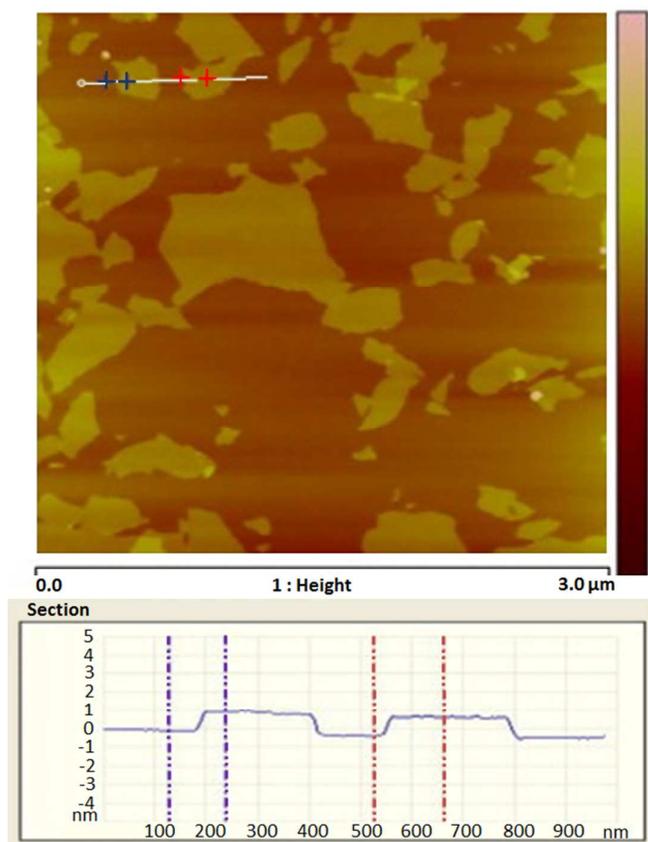


Fig. 8: AFM image and Height profile of graphene¹³⁷. Reproduced with permission from ref. 137.

4.2.2 Mechanical properties

As previously mentioned, graphene have excellent mechanical properties namely high Young's modulus, high tensile strength, low density, etc¹³⁸. These exceptional properties make graphene an ideal candidate as filler for nanocomposite materials. Most of the work on epoxy/graphene nanocomposites is aimed at exploiting the remarkable mechanical enhancement effect of the graphene coupled with the possibility to introduce further functionalities, such as electrical conductivity¹³⁹ or thermal stability¹⁴⁰.

Recently, Bortz *et al.*¹¹⁹ conducted the investigation on the mechanical properties of epoxy/graphene oxide nanocomposites. The study showed the influence of graphene oxide concentration (0.1, 0.25, 0.5 and 1 wt%) on the fracture toughness and flexural strength of nanocomposites which are presented in Fig. 9. The graphs showed that with the increase in graphene oxide concentration, the mechanical properties of nanocomposites increased as well. For example, at the concentration of 1 wt%, the nanocomposite showed more than one hundred

percent increase in G_{IC} . Qi *et al.*¹⁴¹ used thermotropic liquid crystalline epoxy to functionalize the graphene surface. The fabricated nanocomposites showed enhancement in tensile strength from 55.43 MPa to 78.96 MPa at 1 wt% accompanied by nearly one hundred percent increase in impact strength. Similarly, Liu *et al.*¹⁴² investigated the interphase of epoxy/graphene oxide and reported increase in the modulus and toughness. Fracture toughness and flexural modulus were increased with the increasing in filler concentration which indicated the significant enhancement effect of graphene in epoxy matrix.

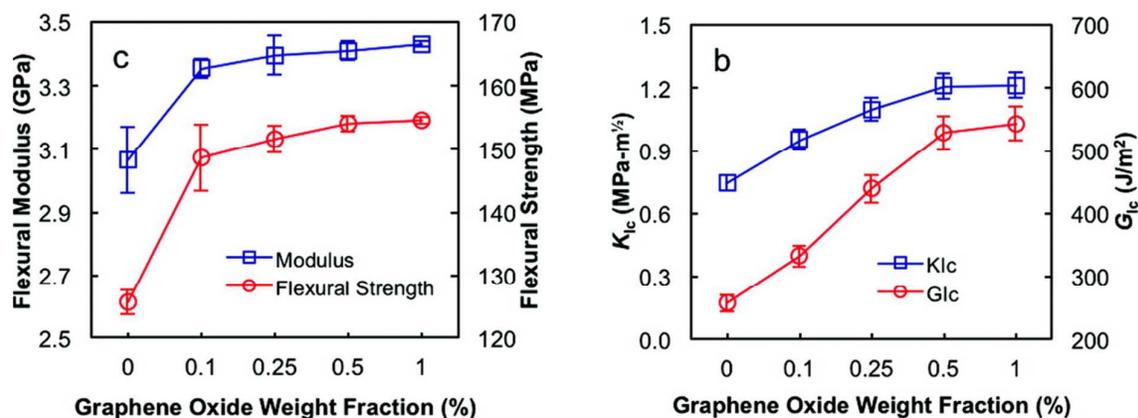


Fig. 9: Quasi-static mechanical properties of epoxy nanocomposites¹¹⁹. Reproduced with permission from ref. 119.

As discussed in the previous sections, dispersion of graphene plays a very crucial role in the preparation of epoxy/graphene nanocomposites¹⁴³, for which many different techniques have been investigated. A homogenous dispersion could give better load transfer to filler material which results into better mechanical properties for the graphene nanocomposites¹⁴⁴. For example, Li *et al.*¹¹⁵ incorporated 0.5 wt% silane functionalized graphene into epoxy matrix by mechanical mixing and bath sonication and reported 20% increase in elastic modulus and 16% increase in tensile strength as compared to neat epoxy. Similarly, Rafiee *et al.*⁹⁶ reported a significant enhancement of Young's modulus at 0.1 wt% of epoxy/graphene nanocomposites processed by shear mixing and tip sonication. For well dispersed nanocomposites, improved mechanical interlocking with polymer chains and graphene can be observed, and slipping of entrapped polymer molecules was suppressed, along with improved dispersion, tensile strength and fracture toughness. Izzuddin *et al.*¹¹⁸ reported that the presence of good adhesion between graphene and matrix were the main attributes for these increments. In order to form a strong interface, polyoxyalkylamine functionalized graphene were dispersed in epoxy matrix by bath sonication and mechanical mixing, and it was evident that the functionalization treatment increased modulus and fracture properties of the nanocomposites. In their report, the samples with 0.489 vol% functionalized graphene, showed 224% improvement in the fracture toughness as compared to the pure resin. Therefore, functionalization of graphene has significant positive effect on the mechanical properties of the epoxy nanocomposites.

4.2.3 Electrical conductivity

A number of studies, related to the electrical properties of polymer/graphene nanocomposites, have been conducted¹⁴⁵. The combination of graphene and polymer matrices offers new attractive electrical properties and novel conducting polymers. These polymers can be used for various engineering applications like, electrical conducting adhesives, antistatic coating and films, electromagnetic interference shielding materials for electronic devices, thermal interface materials, etc¹⁴⁶. These conducting nanocomposites follow the principle of percolation theory which basically explains the transition from insulator to conductor in materials. The percolation threshold is the concentration at which the electrical conductivity of an insulating polymer matrices increases dramatically. A conductive continuous network of filler is created and electrons can be transported by direct contact among nanofiller particles, beyond this concentration, the conductivity of the nanocomposite increases marginally¹⁴⁷.

The electrical conduction in a nanocomposite is due to the formation of a continuous conductive network formed by the fillers. Therefore, the aligned nanofillers have higher probabilities to percolate at lower volumetric concentrations than spherical nanofillers¹⁴⁸. Graphene becomes an ideal candidate to achieve this percolated network at low loading fractions due to its intrinsically high conductivity and the 2D structure. Wajid *et al.*¹¹⁰ reported the ultra-low electrical percolation threshold at 0.088 vol% in epoxy/graphene nanocomposites, by dispersing graphene with the assistance of tip sonication, mechanical mixing and shear mixing. Similarly, Liang *et al.*¹²⁴ also reported significant increase in the electrical conductivity by incorporating graphene in epoxy nanocomposites matrix by bath sonication and mechanical mixing. The conductivity was improved from 0.8×10^{-10} to 0.8×10^{-2} by incorporating 8 vol% reduced graphene oxide into epoxy. Such improvements are only possible when graphene are thoroughly de-bundled and homogeneously dispersed in epoxy matrix. Monti *et al.*¹⁴⁹ dispersed graphene into epoxy to study the electrical conductivities. In order to improve the dispersion, they processed the mixture by tip sonication and mechanical mixing and used different solvents such as chloroform or THF. The highest electrical conductivity was observed for sample with 3 wt% graphene. It was also shown that the thermal conductivity increased with the increase of graphene concentration.

4.2.4 Thermal conductivity

As opposed to the electrical conductivity, the thermal conductivity of epoxy/graphene nanocomposites has received less attention to date. As compared to electrical conductivity enhancements of several orders of magnitude, thermal conductivity enhancement by the carbon nanofillers is not as significant¹⁵⁰. However, several folds increment in thermal conductivity can easily be obtained, as it has been reported that the 2D shape platelets like graphene nanosheets can improve thermal conductivity more effectively than 1D rod like carbon nanotube (CNT)^{92, 151}. As given by Kapitza resistance, the transfer of thermal energy are carried out by the free electrons interaction and lattice vibration between the two contacted interface, poor coupling at the filler/polymer interfaces will significantly impact on thermal resistance¹⁵². Hence, a strong filler/polymer interface is required to achieve good thermal conductivity¹⁵³.

Veca *et al.*¹⁵¹ applied alcohol and oxidative acid treatment with the assistance of extended and vigorous sonication to thermally expanded graphite. Carbon nanosheets were found well dispersed in epoxy matrix with a thickness of less than 10 nm. The incorporation of 33 vol% carbon nanosheets

could improve the in plane thermal conductivity of epoxy nanocomposites to 80 W/(m·K). However, the across-plane thermal conductivity was found only one-tenth to one-fifth of the average in-plane value. This highly anisotropic nature resulted from the 2D structure of the graphene sheets. Wang *et al.*¹⁵⁴ reported that 5% graphite oxide (prepared via thermal expansion) increased the thermal conductivity of epoxy to over 0.8 W/(m·K) and decreased the coefficient of thermal expansion by 31.7% below T_g . Ganguli *et al.*⁹² found that 20 wt% silane functionalized thermally expanded graphite enhanced the thermal conductivity of epoxy from 0.2 to 5.8 W/(m·K). It was interesting to find that silane functionalization could form covalent bonding with epoxy and improve the interfacial heat transfer between two components by reducing acoustic impedance mismatch in the interfacial area. However, excessive functionalization also tends to reduce the intrinsic thermal conductivity of carbon materials. Fig. 10 shows a larger contribution of graphene for the thermal conductivity for epoxy matrix as compared to CNTs and carbon black.

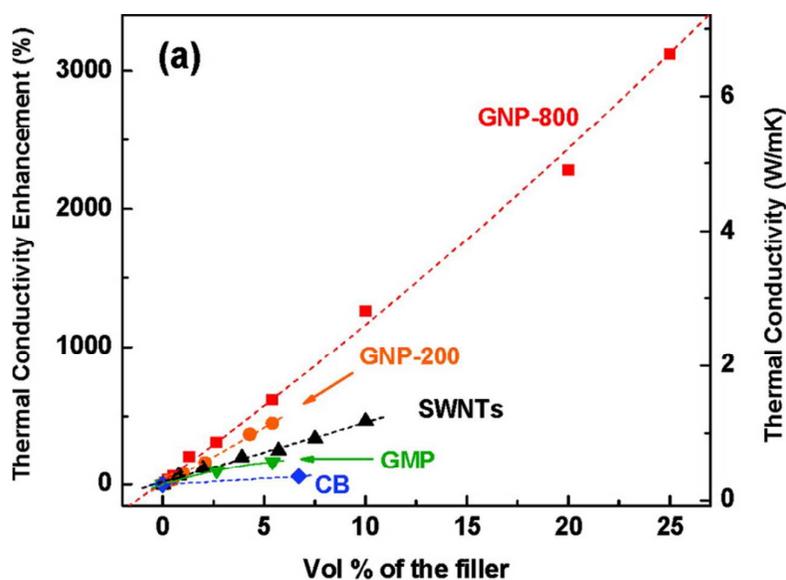


Fig. 10: Thermal conductivity enhancement of epoxy-based nanocomposites¹⁵⁵. Utilized fillers: graphitic microparticles (GMP), GNPs exfoliated at 200°C (GNP-200) and 800°C (GNP-800), carbon black (CB) and SWNTs. Reproduced with permission from ref. 155.

4.2.5 Thermal stability

Several studies have evaluated the effect of graphene on the thermal properties in many polymer matrices, such as thermal degradation temperature¹⁵⁶, glass transition temperature¹⁵⁷, melting temperature¹⁵⁸ and polymer crystallinity of the nanocomposites¹⁵⁹. However, there is no melting temperature for epoxy because of its thermosetting nature, so glass transition and thermal degradation behaviours are among the most important characters to characterize the thermal stability of epoxy nanocomposites which have been discussed in this section.

It is generally observed that graphene would enhance the T_g of epoxy matrix¹⁶⁰. This is due to the adhesion force between epoxy and graphene which reduces the mobility of epoxy chains on graphene surface. Contrarily, decrease in T_g is expected for weakly adhering fillers and unstable interfaces facilitating the chain polymer mobility, thus lowering the T_g ¹⁶¹. Li *et al.*¹⁶² reported the increase in T_g of epoxy by hindering segmental motion of polymer chains via mechanical interlocking

and hydrogen bonding with surface oxygen functionalities. Similarly, a T_g increase of 14°C in epoxy/graphene nanocomposites has been measured by Park *et al.*¹⁶³ at 1 phr (parts per hundred resin) of graphene in epoxy matrix. This is an expected outcome of the strong filler-matrix adhesion and the conformational changes of the epoxy matrix at the epoxy/graphene interface.

In contrary, a significant volume of research reported the opposite trend and will be discuss here. It has been vastly reported that graphene reduces the glass transition or thermal degradation temperature of epoxy matrix and there is no unanimous agreement for this negative trend. Galpaya *et al.*¹⁶⁴ proposed the theory that the T_g of nanocomposites depends on the balance of two effects, i.e., influence on reaction conversion and molecular confinement. Graphene sheets are stiffer than epoxy matrix which could lead to significant confinement on the polymer chains. On the other hand, graphene sheets may impede the epoxy curing reaction. The reason could be the functional groups on graphene surface reacting with the curing agent and/or epoxy resin, or graphene sheets covering the reactive sites in the resin due to its high surface area. If the latter one plays the dominant role, this would be expected to reduce the polymer cross link density and would also increase polymer chain mobility. Liao *et al.*⁷⁰ and Kim *et al.*¹²³ reported similar conclusions as well. According to them, the incorporation of graphene reduces the cross link density of the epoxy matrix, which results in the decrease of T_g . Some research groups like Saurin *et al.*¹⁶⁵, Liu *et al.*¹⁶⁶, and Guo *et al.*⁹¹ reported that graphene acts as reactive plasticizer and has a plasticizing effect on epoxy resin, thus increasing the flexibility of chain segments of the epoxy matrix. Liu *et al.*¹²⁰ prepared an epoxy/imidazole functionalized graphene nanocomposite and reported that the short molecular chains of functional group on graphene surface are flexible and would result in the overall T_g decrease. There are also some other claims, such as Liu *et al.*¹⁶⁷ incorporated edge functionalized graphene into epoxy resin and found that the T_g decreased because of the existence of graphene sheets that could result in increased flexibility of the network. Zhang *et al.*¹⁶⁸ prepared magnetic graphene reinforced epoxy nanocomposites and reported that the rigid structure of graphene nanoplatelets would cause extra enlarged free volume, which is detrimental to the thermal stability of the matrix.

For thermal decomposition temperatures(T_d), which are characterized by the maximum weight loss rate in thermogravimetry, shifts up 30°C for epoxy nanocomposites by incorporating 0.5 wt% functionalized graphene were reported¹⁰⁵. Decomposition of graphene nanocomposites is substantially slower than neat epoxy, which is attributed to restricted chain mobility of polymers near the graphene surface. Similarly, Prolongo *et al.*¹⁶⁹ reported that 0.5 wt% graphene nanoplatelet can push the thermal degradation temperature of epoxy from 377°C to 397°C. Yousefi *et al.*¹¹⁴ reported that both graphene oxide and reduced graphene oxide improved the thermal decomposition temperature of epoxy matrix. Fig. 11 shows the shift in thermal gravimetric analysis (TGA) curves to higher temperature, which means a higher thermal stability due to the incorporation of graphene.

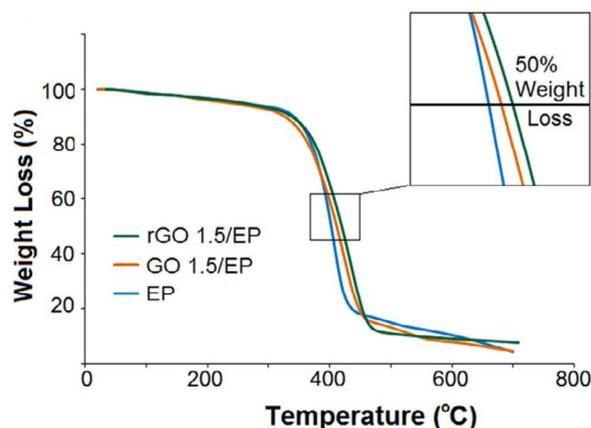


Fig. 11: TGA curves of epoxy nanocomposites containing GO and rGO¹¹⁴. Reproduced with permission from ref. 114.

Wang *et al.*¹⁷⁰ and Xin *et al.*¹⁷¹ incorporated functionalized graphene into epoxy matrix and reported decrease in T_d . This was due to the presence of thermally unstable chemicals, which on decomposition, lowered T_d as compared to monolithic epoxy. Feng *et al.*¹¹¹ used epoxy resin to modify graphene first, and then mixed with the epoxy matrix. They found that in some cases, the T_d of the nanocomposite decreased because the filler might causes defects in the polymeric networks during the curing.

In general, it is widely acknowledged that graphene could enhance the thermal stability of epoxy. However, there are still many controversies where many researchers reported a decrease in thermal properties of epoxy with the incorporation of graphene. The reason of such has not been fully explained and requires deeper understanding via extensive further research.

4.2.6 Flame retardant properties

Engineering materials are required to resist degradation during in an unlikely event of fire in many critical applications like skyscrapers, boats, or airplanes¹⁷². In fact, some studies reported that about 20% of victims of airplane crashes are killed not by the crash itself but by ensuing fires¹⁷³. Materials used in aviation should be designed to inhibit, suppress, or delay the production of flames to prevent the spread of fire. Flame retardant materials are mainly based on halogen, phosphorus, inorganic, and melamine compounds¹⁷⁴, however, among these flame retardants, only inorganic fillers are normally nontoxic¹⁷⁵. Current research on epoxy/graphene nanocomposites has been focused on improving the flame retardant properties such as ease of ignition, limiting oxygen index, rate of heat release, and the evolution of smoke and toxic gases by incorporating modified graphene, along with improving the physical properties of the epoxy matrix¹⁷⁶⁻¹⁷⁸. For example, Li *et al.*¹⁷⁹ use 2-(Diphenylphosphino)ethyltriethoxy silane modified graphene oxide, and then incorporated this modified graphene oxide into epoxy matrix. They found that the limiting oxygen index increased from 20 to 36, which means a huge transition of material's nature from flammable to non-flammable. Jiang *et al.*¹⁸⁰ prepared epoxy/graphene-ZnS nanocomposites and reported that with the incorporation of ZnS decorated graphene, the carbon monoxide production rate for the nanocomposites is much lower than that of pure epoxy along with decreased total smoke release. Wang *et al.*¹⁸¹ prepared Ni-Fe Layered Double Hydroxide (LDH) modified graphene/epoxy nanocomposites. They found that with the incorporation of 2 wt% Ni-Fe LDH modified graphene, the

time of ignition of epoxy matrix increased from 68s to 89s, the total heat release decreased from 113.1 MJ/m² to 44.2 MJ/m², and the fire growth index decreased from 13.3 kW/m²·s to 4.8 kW/m²·s. Fig. 12 shows the drastic decrease of heat release rate with the incorporation of graphene and Ni-Fe LDH modified graphene.

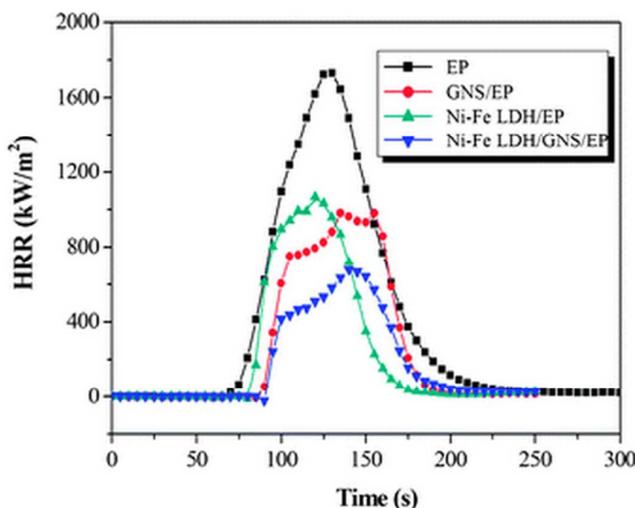


Fig. 12: Heat release rate versus time curves of epoxy and its nanocomposites¹⁸¹. Reproduced with permission from ref. 181.

Zhuo *et al.*¹⁸² proposed a flame retarding mechanism for polymer matrices when filled with graphene. According to Zhuo *et al.*¹⁸², the barrier effect of graphene plays dominant role in flame retardancy. Graphene walls make excellent gas barriers which delay the oxidative degradation of epoxy during a fire, moreover, the large surface area of graphene can induce a large amount of char which prevents the resin from suffering heat.

In general, the addition of graphene into epoxy matrix results in improving flame retardancy and thermal stability of epoxy along with improved mechanical properties. Moreover, no environmental or toxicity issues have been reported for graphene. Therefore, it can be concluded that graphene has a great potential to be one of the most promising flame retarding fillers for nanocomposites in near future.

4.2.7 Synergic effects with other fillers

Synergic effect or hybridisation means incorporation of two or more fillers together for enhanced functionality which is not possible to achieve with single filler alone. Recently, a tremendous research effort can be witnessed to generate enhanced properties by synergistically combining different fillers as reported by Inam *et al.*¹⁸³. The group fabricated multiscale epoxy composites which showed enhanced mechanical properties with the combination of carbon nanotubes and carbon fibers. Chatterjee *et al.*¹⁸⁴ found that the CNT:GNP ratio is an interesting factor influencing the properties of the epoxy based nanocomposites. At nanofiller concentration of 0.5 wt%, highest CNT content (9:1) showed marked improvement in fracture toughness of 76%. Kumar *et al.*¹⁸⁵ suggested that by bringing together two nanofillers like CNT and GNP, they could form a co-supporting network. This net-resemble structure could shield the fillers from fracture and damage during processing, while still allow full dispersion of both filler during high power sonication, thus causes improved properties. Apart from the mechanical properties, incorporation of carbon fillers in polymer matrices attained significance for the applications where enhanced thermal and electrical

conductivity were required together. Epoxy resins containing a binary mixture of GNP and single wall carbon nanotube (SWCNT) in 3:1 weight ratio have higher thermal conductivity than those reinforced with either individual fillers. Yu *et al.*¹⁸⁶ explained this synergistic effect by bridging interactions between GNP and SWCNT which can reduce the interfacial resistance for thermal conduction. Also, a remarkable synergetic effect between graphene platelets and multiwalled carbon nanotubes (MWCNTs) in improving the mechanical properties and thermal conductivity of epoxy nanocomposites was demonstrated by Yang *et al.* The tensile strength and thermal conductivity were increased by 35.4% and 146.9% respectively by using MWCNT/graphene fillers as compared to either filler for epoxy nanocomposites. They found that stacking of individual 2D graphene is effectively inhibited by introducing 1D MWCNTs. Long and tortuous MWCNTs can bridge adjacent graphene platelets and inhibit their agglomeration, resulting in a high contact area between the MWCNT/graphene structures and the polymer matrices¹⁵⁹.

In general, the exact mechanism responsible for this dramatic enhancement is not entirely understood. It is widely believed that molecular level interactions between the nanomaterials and polymer matrices play a major role. The large interface area available for such interactions clearly hold the key for the dramatic enhancement in mechanical properties¹⁸⁷. Table 2 lists some representative papers which adopt multi filler or hybridisation approach to modify the properties of epoxy matrix.

Table 2. Synergic effect of graphene and other fillers in epoxy matrix

refer	year	filler	dispersion method	% increase in σ	% increase in E	% increase in K_{IC}	% increase in κ	% increase in λ	% increase in T_g	% increase in T_d
188	2014	G + CNTs	bath sonic + mechanical mix				10			
189	2014	G + capron	mechanical mix						31	
190	2014	G + CNTs	bath Sonic	-23	-11.5		4			
191	2014	GO + carbon fiber		15.1	20.2				9	
192	2013	rGO + CNTs	3-roll calendaring + shear mix						4	
193	2013	GO + CNTs	bath sonic + mechanical mix							
171	2013	G + glass fiber	shear mix	-16.3	-8.9				-9	-16
184	2012	G + CNTs	bath sonic + 3-roll calendaring + high pressure homogenizer			78		84.2		
125	2012	GO + CNTs	mechanical mix							

159	2011	G + CNTs	bath sonic + shear mix + mechanical mix	0.9	23.1			23.8		
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Conclusions

Graphene shows great potential as filler for the next generation advanced nanocomposite materials. Numerous efforts have been made to prepare useful epoxy/graphene nanocomposites. However, the development and applicability of epoxy/graphene nanocomposites will be significantly related to the dispersion and the interfacial bonding of graphene in epoxy matrix, which are the two most critical factors to determine the performance of these novel nanocomposites. Thus, the key to prepare advanced epoxy/graphene nanocomposites is to improve the techniques for the dispersion of graphene and engineering the graphene-epoxy interface. This review provided a detailed introduction of epoxy/graphene nanocomposites and critical analyses on recent research investigations. Following conclusions can be drawn from the existing reported research:

- Graphene has significant potential for epoxy based composites. Extremely enhanced multi-functional properties can be achieved subject to homogenous dispersion and strong interfacial interactions. Chemical functionalization of graphene can also significantly improve the graphene-epoxy interfacial interactions.
- Solvent processing is the most widely adopted method to prepare epoxy/graphene nanocomposites. The high viscosity of epoxy may hinder the uniform dispersion of graphene and therefore, it is also difficult to adopt solvent-free processing approach.
- Mechanical properties, electrical conductivity, thermal conductivity, thermal stability and flame retardant properties are generally increased with the incorporation of graphene.
- Graphene could increase the glass transition and thermal degradation temperatures of epoxy nanocomposite. However, this needs to be further explored as some investigations have reported the negative trend.

In general, epoxy/graphene materials have remarkably high thermal and electrical conductivities, as well as improved mechanical strength and thermal stability. Because of these excellent properties, graphene reinforced epoxy nanocomposites possess great potential to be used in automotive, electronics, aerospace and etc. However, a lot is still required to be understood before such application can be materialised.

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