



**In-situ Constructing  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> Composites  
and Their Highly Efficient Photocatalytic Performances**

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## In-situ Constructing $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> Composites and Their Highly Efficient Photocatalytic Performances

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In this study,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were constructed by an in-situ method in one step by  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>. It indicates that  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> transformed to  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> when it was calcined with g-C<sub>3</sub>N<sub>4</sub> together. The co-existences of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>, g-C<sub>3</sub>N<sub>4</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> were proved by X-ray diffraction (XRD), Fourier transform infrared (FT-IR) spectroscopy and high-resolution transmission electron microscope (HRTEM). The energy band structures and optical properties were studied by ultraviolet-visible diffuse reflectance spectroscopy (UV-vis DRS) and Valence band X-ray photoelectron spectroscopy (VB-XPS). The in-situ formation mechanism of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites in one step was studied by the samples' surface chemical states, which were measured by X-ray photoelectron spectroscopy (XPS). The results show that the surface covered or coordinated g-C<sub>3</sub>N<sub>4</sub> and CO<sub>3</sub><sup>2-</sup> have promotional effect on the stability of  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> at room temperature. The enhanced photocatalytic activities of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were evaluated by the photocatalytic oxidation of isopropyl alcohol (IPA), and attributed to the heterojunction formation between  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>.

### Introduction

TiO<sub>2</sub> has been developed to be a famous photocatalyst since the discovery of photocatalytic splitting of water molecular on TiO<sub>2</sub> electrode in 1972<sup>1</sup>. However, its practical application is greatly hindered by its only UV light response. From the view of effectively utilizing the solar light, visible-light-responsive photocatalysts are attracting increasing interests in the recent years. Bi<sub>2</sub>O<sub>3</sub> semiconductors are a serious of visible-light-responsive photocatalysts with appropriate band gaps<sup>2</sup>. Up to now, six polymorphic forms have been found in this family, monoclinic  $\alpha$ -phase, tetragonal  $\beta$ - phase, body-centered cubic  $\gamma$ - phase, face-centered cubic  $\delta$ - phase, tetragonal  $\epsilon$ - phase, and triclinic  $\omega$ - phase<sup>3</sup>. Among them,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> were mostly studied in the past because of their respective highest thermal stability and photocatalytic activity<sup>2, 4-13</sup>.

$\alpha$ -Bi<sub>2</sub>O<sub>3</sub> is an attractive photocatalyst because of its easy preparation<sup>10</sup>, environmental friendship and thermal stability<sup>14-18</sup>. However, its photocatalytic activity has been proved to be poor due to its some unfavorable properties for photocatalysis, although it has some properties beneficial for photocatalysis, such as narrow band gap (+ 2.8 eV), high valence band potential (+ 3.13 V vs. NHE)<sup>19</sup>, etc. Firstly, its conduction band potential (+ 0.33 V vs. NHE) is too low to oxidize the surface O<sub>2</sub> to O<sub>2</sub><sup>-</sup> (- 0.33 V vs. NHE) through a fast single-electron reaction, resulting in a high recombination rate of the

photogenerated electrons and holes<sup>14, 20</sup>. Secondly, the particle size of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> prepared by usual methods is very large (dozens of micrometers), resulting in low surface areas<sup>9, 10, 21</sup>. Therefore, developing useful strategies to improve the photocatalytic activity of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> is the main research point in this field.

The construction of heterojunctions is a mostly used method to improve the photocatalytic activity in the last decades<sup>4, 6, 22, 23</sup>. g-C<sub>3</sub>N<sub>4</sub> is a typical, polymeric and graphical layered material; its photocatalytic activity is poor due to its high photocarriers' recombination rate<sup>24-26</sup>. Therefore, it is often regarded as a good candidate for the construction of heterojunctions. However, this work is one of the accidents in the heterojunction construction process of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>, while the other accident has been reported by us recently<sup>19</sup>. In this work, both morphology and crystal structure of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> have great changes in this process. Here, we will mainly discuss the crystal structure changes of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>, since the morphology changes has been discussed detailedly in another paper<sup>19</sup>.

In this study,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were successfully constructed by  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> in one step with an in situ method. Components of the prepared samples were studied by X-ray diffraction (XRD) and Fourier transform infrared (FT-IR) spectroscopy. Morphologies of the composites were recorded by field-emission scanning electron microscope (FE-SEM), transmission electron microscope (TEM), and high-resolution transmission electron microscope (HRTEM). The energy band structures and optical properties were studied by ultraviolet-visible diffuse reflectance spectroscopy (UV-vis DRS) and Valence band X-ray photoelectron spectroscopy (VB-XPS). The in-situ formation mechanism of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites in one step was studied by the samples' surface chemical states, which were measured by X-ray photoelectron spectroscopy (XPS). The

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photocatalytic activity enhancement of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were also studied by the photocatalytic oxidation of IPA, and discussed in detail in this work.

## Experimental

### 2.1 Sample preparation

The micro-rod like  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> was prepared by a precipitation method using Bi(NO<sub>3</sub>)<sub>3</sub> as the Bi source<sup>10,27</sup>. Typically, 10.78 g of Bi(NO<sub>3</sub>)<sub>3</sub>·5H<sub>2</sub>O was dissolved in a 30 mL aqueous solution of HNO<sub>3</sub> (1.5 M) at first, to avoid the hydrolyzation of Bi<sup>3+</sup> ions. Secondly, the NaOH solution (50% w/v) was added into the solution dropwise until pH > 13, under vigorous agitation. In that process, yellow precipitate was formed. Thirdly, the suspension was heated to 80 °C and kept for 2 hours, and the precipitate was subsequently collected by centrifugation and being washed with deionized water several times. After that, the resultant solid was dried at 120 °C overnight in air. Lastly, the powder was calcined at 450 °C for 5 hours.

The g-C<sub>3</sub>N<sub>4</sub> was synthesized by heating the melamine at 520 °C for 4 hours in a half-open system, according to the previous report<sup>28</sup>.

The  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were constructed according to our earlier research as follows<sup>29,30</sup>: The g-C<sub>3</sub>N<sub>4</sub> and micro-rod-like  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> with different weight fractions were thrown into 100 ml alcohol (95%), following with 3 hours' ultrasonication and 24 hours' stirring at room temperature. Before the last calcinations at 450 °C in a half-open system, the products were dried at 100 °C for 12 hours in advance. The weight ratios of g-C<sub>3</sub>N<sub>4</sub> and micro-rod  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> were represented by n<sub>1</sub> : n<sub>2</sub>. The calcination time was set to be 3 hours, 4 hours and 5 hours.

Moreover, the nano  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> were synthesized for comparison, using the Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> as a precursor<sup>31</sup>.

### 2.2 Characterizations

The crystal structures of the precursor and Bi<sub>2</sub>O<sub>3</sub> samples were determined with an X-ray diffractometer (X'pert Powder, PANalytical B. V., Netherlands) with Cu - K $\alpha$  radiation. The IR absorption spectra were obtained with a Fourier transform-infrared (FT-IR) spectrophotometer (IRPrestige-21, Shimadzu Co., Japan). Field-emission electron microscopy (FESEM) was performed to observe the morphologies on a JEOL 6701F microscope. High-resolution transmission electron microscope images were recorded with a high-resolution transmission electron microscope (HRTEM, Tecnai G2F30). The diffuse reflection spectra were measured with an integrating sphere equipped ultraviolet-visible (UV-vis) recording spectrophotometer (UV-2500PC, Shimadzu Co., Japan) using BaSO<sub>4</sub> as reference and the optical absorption spectra were converted from the diffuse reflection spectra according to the Kubelka-Munk equation. X-ray photoelectron spectroscopy (XPS) was performed on an ESCALAB 250Xi spectrometerequipped with 300W Al K $\alpha$  radiation, while all binding energies were referenced to the C1s peak (284.6 eV) of the surface adventitious carbon.

### 2.3 Photocatalytic Activity

The photocatalytic oxidations of gaseous IPA were carried out to evaluate the photocatalytic activities of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites. The experiments were carried out under the visible light irradiation (420 nm  $\leq$   $\lambda$   $\leq$  800 nm). The light source was a 300 W Xe-arc lamp (10 A imported current, focused through a 50  $\times$  50 mm<sup>2</sup>

shutter window) equipped with wavelength cut-off filters and a water filter. 50 mg of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> powder was bespread uniformly on a glass dish with an area of 9 cm<sup>2</sup>. A certain amount of gaseous IPA (~1400 ppm) was injected into the vessel and kept for 2 hour in the dark before irradiation. During the irradiation by visible light, 0.5 ml of the gas was sampled everyone 1 hour intervals. The products were analyzed with a gas chromatograph (GC-2014, Shimadzu, Japan) with a flame ionization detector (FID). Meanwhile, the g-C<sub>3</sub>N<sub>4</sub>, nano  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> were also used to photooxide the IPA for comparison under the same conditions. Furthermore, the adsorptions of IPA in dark over  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were also carried out without light irradiation.

## Results and discussion

### 3.1 Components of $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites

The components of the prepared  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were identified by XRD and FT-IR spectra.

The crystal structures of the composites were identified by XRD pattern in Fig. 1. All the samples were compared with the ICSD file of No. 41-1449 and No. 27-0053, respectively corresponding to  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub><sup>32</sup>. The results indicate that, the crystal structures obviously changed with the increase of the weight ratio of g-C<sub>3</sub>N<sub>4</sub> in the mixture of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>, as well as the calcination time. Firstly, with the increase of the weight ratio of g-C<sub>3</sub>N<sub>4</sub>, we found that the characteristic peaks of  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> become more and more strong, evidences that  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> gradually changes to  $\beta$ -Bi<sub>2</sub>O<sub>3</sub>. Then for the sample of 9.5:0.5, when the calcination time prolonged from 3 hours to 4 hours, almost no characteristic peaks of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> is observed in the XRD spectrum; when the calcination time was further prolonged to 5 hours, the characteristic peaks of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> appear again. The XRD results show that both  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> exist in the composites. However, we can't see any characteristic peaks derived from g-C<sub>3</sub>N<sub>4</sub> in the XRD spectra, so we can't get any information of g-C<sub>3</sub>N<sub>4</sub> by XRD. Therefore, the FT - IR was used to prove the existence of g-C<sub>3</sub>N<sub>4</sub> in the composite samples. As shown in Fig. 2, we can clearly see the vibration peaks (820 cm<sup>-1</sup>, 890 cm<sup>-1</sup>, 1000-1900 cm<sup>-1</sup>, 1965 cm<sup>-1</sup>, 2054 cm<sup>-1</sup>, 2141 cm<sup>-1</sup>, and 2800-3700 cm<sup>-1</sup>) derived from g-C<sub>3</sub>N<sub>4</sub> in all the composite samples.

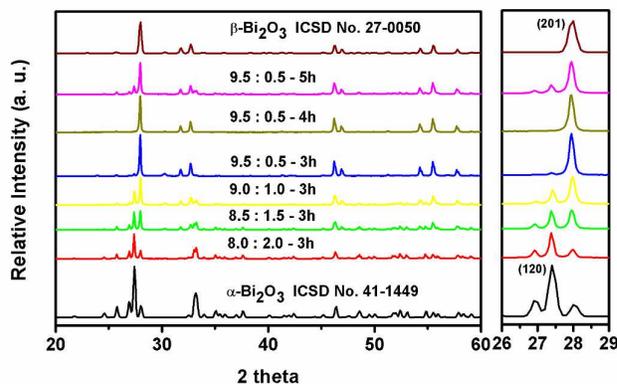
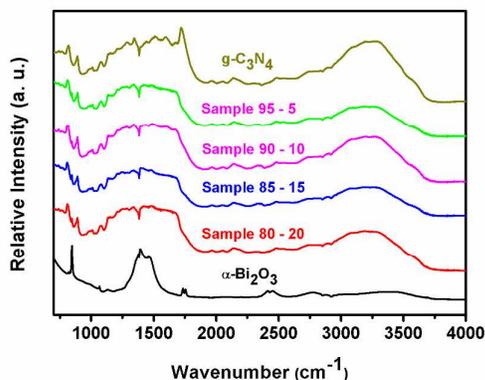


Fig. 1 XRD spectra of all the  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite samples



**Fig. 2** FT-IR absorption spectra of all the  $\alpha$ - $\text{Bi}_2\text{O}_3$ / $\text{g-C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composite samples

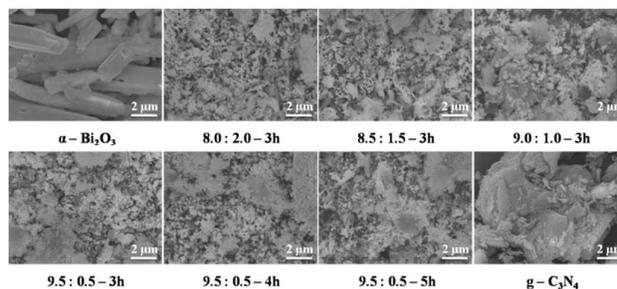
Detailedly, we also measure the practical weight ratios of the respective  $\alpha$ - $\text{Bi}_2\text{O}_3$ ,  $\beta$ - $\text{Bi}_2\text{O}_3$  and  $\text{g-C}_3\text{N}_4$ , in  $\alpha$ - $\text{Bi}_2\text{O}_3$ / $\text{g-C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composites. The weight ratios of the  $\text{g-C}_3\text{N}_4$  were measured through the mass discrepancy before and after the calcination process of every  $\alpha$ - $\text{Bi}_2\text{O}_3$ / $\text{g-C}_3\text{N}_4$  mixture samples, because the  $\text{g-C}_3\text{N}_4$  will dissolve in the presence of a second phase during the calcination processes at  $450^\circ\text{C}$ . The contents of  $\alpha$ - $\text{Bi}_2\text{O}_3$  and  $\beta$ - $\text{Bi}_2\text{O}_3$  in the sequence of as-prepared composites, were obtained by simulating the XRD pattern with standard crystal structures of  $\alpha$ - $\text{Bi}_2\text{O}_3$  and  $\beta$ - $\text{Bi}_2\text{O}_3$ , through a Maud program<sup>33</sup>. The results were displayed in Table 1.

**Table 1** Contents of the  $\alpha$ - $\text{Bi}_2\text{O}_3$ / $\text{g-C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composites

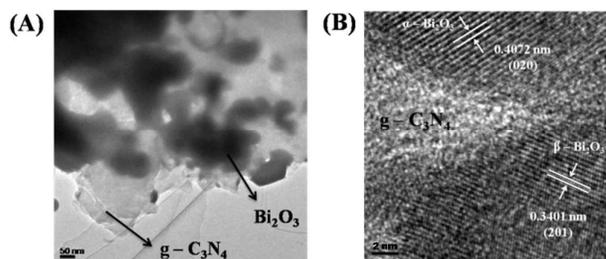
Sample	Content			
	Weight Fraction (%)	$\alpha$ - $\text{Bi}_2\text{O}_3$	$\beta$ - $\text{Bi}_2\text{O}_3$	$\text{g-C}_3\text{N}_4$
$\alpha$ - $\text{Bi}_2\text{O}_3$	100	-	-	-
$\beta$ - $\text{Bi}_2\text{O}_3$	-	100	-	-
8.0:2.0-3h	91.15	8.85	<0.100	
8.5:1.5-3h	58.20	33.74	8.060	
9.0:1.0-3h	37.86	57.24	4.900	
9.5:0.5-3h	1.350	18.46	80.19	
9.5:0.5-4h	1.150	47.41	51.44	
9.5:0.5-5h	14.16	61.84	24.00	
$\text{g-C}_3\text{N}_4$	-	-	100	

### 3.2 Micro-morphologies and structures

As we reported in the earlier research, the morphologies of  $\alpha$ - $\text{Bi}_2\text{O}_3$  was significantly changed when it was heat treated with  $\text{g-C}_3\text{N}_4$  together<sup>19</sup>. In this study, the morphologies of the composites were observed by FE-SEM, as shown in Fig. 3. We found that the morphologies of both  $\alpha$ - $\text{Bi}_2\text{O}_3$  and  $\text{g-C}_3\text{N}_4$  have huge changes. As we can see that, the particle size of  $\text{Bi}_2\text{O}_3$  decreased from micrometer to nanometer, which is the same with our previous report<sup>19</sup>. The size of  $\text{g-C}_3\text{N}_4$  also has some decrease. The mechanism of the



**Fig. 3** SEM images of all the  $\alpha$ - $\text{Bi}_2\text{O}_3$ / $\text{g-C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composite samples



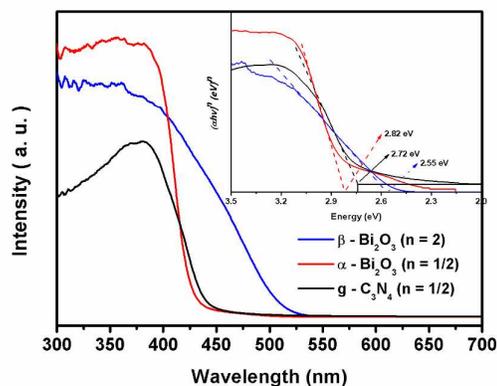
**Fig. 4** TEM (A) and HRTEM (B) images of the  $\alpha$ - $\text{Bi}_2\text{O}_3$ / $\text{g-C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composite samples (9.5 : 0.5 - 3h)

morphology changes has been explained in our earlier report<sup>19</sup>.

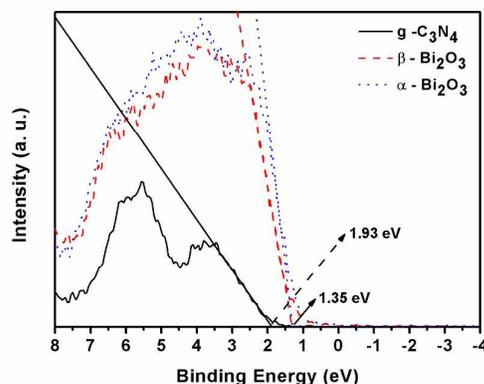
Further, layered structures of  $\text{g-C}_3\text{N}_4$  were observed in Fig. 4 (A) with TEM, and  $\text{Bi}_2\text{O}_3$  particles were packaged in the  $\text{g-C}_3\text{N}_4$  nanosheets. From the HRTEM image in Fig. 4 (B), we can clearly see the coexistence of  $\alpha$ - $\text{Bi}_2\text{O}_3$ ,  $\beta$ - $\text{Bi}_2\text{O}_3$  and  $\text{g-C}_3\text{N}_4$ . The amorphous fraction is identified to be  $\text{g-C}_3\text{N}_4$  nanosheets; the lattice space of 0.4072 nm is attributed to the (020) facet of  $\alpha$ - $\text{Bi}_2\text{O}_3$ ; the lattice space of 0.3401 nm is accordance with the (201) crystal planes of  $\beta$ - $\text{Bi}_2\text{O}_3$ . The TEM and HRTEM images indicate the co-existence of  $\alpha$ - $\text{Bi}_2\text{O}_3$ ,  $\beta$ - $\text{Bi}_2\text{O}_3$  and  $\text{g-C}_3\text{N}_4$ , as well as good contacts between any two phases of them. In addition, the SEM spectra of nano  $\alpha$ - $\text{Bi}_2\text{O}_3$  and  $\beta$ - $\text{Bi}_2\text{O}_3$  are also displayed in Fig. S1.

### 3.3 Energy band structures and optical properties

The energy band structures of  $\alpha$ - $\text{Bi}_2\text{O}_3$ ,  $\beta$ - $\text{Bi}_2\text{O}_3$  and  $\text{g-C}_3\text{N}_4$  were studied by the combinations of UV-vis DRS and VB-XPS.



**Fig. 5** UV-vis absorption spectra of the as-prepared  $\alpha$ - $\text{Bi}_2\text{O}_3$ ,  $\text{g-C}_3\text{N}_4$  and  $\beta$ - $\text{Bi}_2\text{O}_3$ ; the insets are the corresponding Tauc Plots curves



**Fig. 6** VB-XPS spectra of the as-prepared  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>, g-C<sub>3</sub>N<sub>4</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub>

The band gaps of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> were obtained from a Tauc plot spectra, which was transferred from the UV-vis DRS spectra (Fig. 5) through Tauc equation as follows<sup>18</sup>:

$$(ah\nu)^n = A(h\nu - E_g)$$

In this equation,  $\alpha$ ,  $\nu$ ,  $A$  and  $E_g$  are the absorption coefficient, light frequency, proportionality constant and band gap. The value of the index  $n$  depends on the property of the materials: for the direct band-gap semiconductors,  $n = 2$ ; for the indirect band-gap semiconductors,  $n = 1/2$ . According to the earlier literatures,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub><sup>34</sup> and g-C<sub>3</sub>N<sub>4</sub><sup>28</sup> are indirect band-gap semiconductors;  $\beta$ -Bi<sub>2</sub>O<sub>3</sub><sup>35</sup> is direct band-gap semiconductor. Therefore, the value of  $n$  is determined to be 1/2 for  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>, and 2 for  $\beta$ -Bi<sub>2</sub>O<sub>3</sub>. Based on the results, the Tauc plot spectra of  $(ah\nu)^n$  vs. photo energy is described as the inset of Fig. 5. The band gaps of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> were estimated to be 2.82 eV, 2.55 eV and 2.72 eV, from the tangent lines.

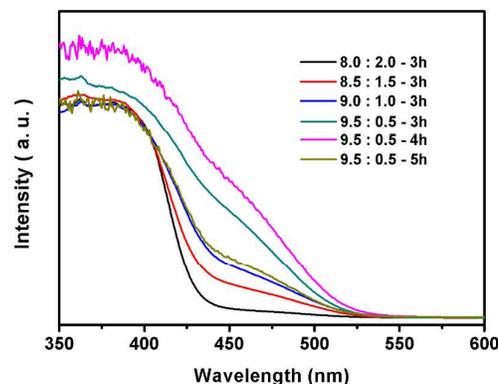
The energy differences between the Fermi levels and valence bands of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> were measured by VB-XPS (Fig. 6). As shown in Fig. 6, the energy differences between the Fermi levels and valence bands of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> are almost the same, and the value is +1.35 eV, while this value of g-C<sub>3</sub>N<sub>4</sub> is +1.93 eV. According to the literatures, the Fermi levels of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub> are respective +1.73 eV<sup>36</sup>, +0.54 eV<sup>6</sup> and -0.3 eV<sup>37</sup>.

Combing the values of band gaps, Fermi levels and the energy differences between Fermi level and valence bands, we obtained the energy band structures of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>. The specific values are displayed in Table 2.

**Table 2** Energy band structures of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>

Contents Values Sample	Band gap (eV)	Valence band (eV vs. NHE)	Conduction band (eV vs. NHE)	Fermi Level (eV vs. NHE)
$\alpha$ -Bi <sub>2</sub> O <sub>3</sub>	2.82	+ 3.08	+ 0.26	+ 1.73
g-C <sub>3</sub> N <sub>4</sub>	2.72	+ 1.63	- 1.09	+ 0.54
$\beta$ -Bi <sub>2</sub> O <sub>3</sub>	2.55	+ 1.89	- 0.66	- 0.30

The optical absorption spectra of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites were studied by UV-vis DRS spectra (Fig. 7). We found that with the weight ratio decrease of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> in  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites, the visible light absorption intensity during 420



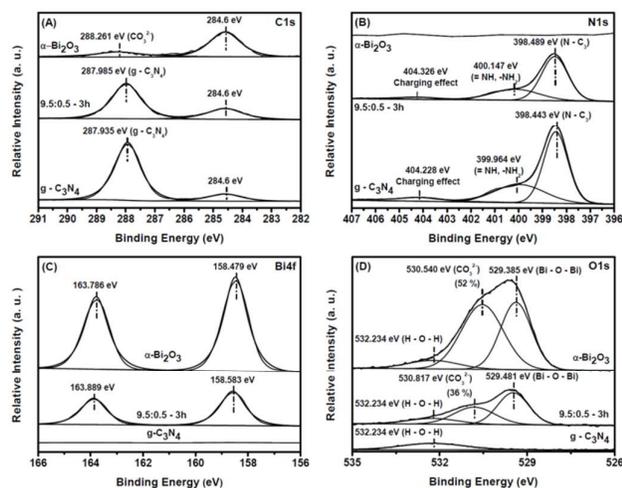
**Fig. 7** UV-vis absorption spectra of the  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite samples

nm to 550 nm has gradual increase and a red shift. According to the UV-vis DRS results, the light absorption region is greatly enhanced by the constructions of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites.

### 3.4 Formation Mechanism

As we know,  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> is not stable at room temperature, but it is formed in this work. It was reported that, the surface-coordinated organic ligands and CO<sub>3</sub><sup>2-</sup> can lower the surface energy of NaNbO<sub>3</sub> and make it stable at room temperature<sup>38</sup>. Therefore, we speculate that the CO<sub>3</sub><sup>2-</sup> and g-C<sub>3</sub>N<sub>4</sub> have the same effect for the stabilization of  $\beta$ -Bi<sub>2</sub>O<sub>3</sub> at room temperature.

In order to understand the formation mechanism of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composites and prove our speculation, we studied the surface chemical states of the composite samples, g-C<sub>3</sub>N<sub>4</sub> and  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> by XPS spectra. In Fig. 8, the high resolution XPS spectra of C1s, N1s, Bi4f and O1s were studied in detail. From Fig. 8 (A), two types of carbon were observed in all the three samples. For  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>, the two types of carbon were attributed to the surface adventitious carbon (286.4 eV) and carbon in CO<sub>3</sub><sup>2-</sup> group (288.261 eV), according to the earlier literatures<sup>39, 40</sup>. For g-C<sub>3</sub>N<sub>4</sub> and  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite, the two types of carbon were attributed to the surface adventitious carbon (286.4 eV) and g-C<sub>3</sub>N<sub>4</sub><sup>25</sup> (287.935 eV and 287.985 eV). From Fig. 8 (B), three types of nitrogen were observed in the samples of g-C<sub>3</sub>N<sub>4</sub> and  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite, and they are all derived from g-C<sub>3</sub>N<sub>4</sub>. They are respectively assigned to tertiary nitrogen [N-(C)<sub>3</sub>] groups (398.443 eV and 398.489 eV), sp<sup>2</sup>-hybridized nitrogen (=NH, -NH<sub>2</sub>) (399.964 eV and 400.147 eV), and the charging effects (404.228 eV and 404.326 eV)<sup>25</sup>. In Fig. 8 (C), the characteristic peaks of Bi<sup>3+</sup> were observed in both the  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite samples, and ascribed as Bi4f<sub>5/2</sub> (163.786 eV and 163.889 eV) and Bi4f<sub>7/2</sub> (158.479 eV and 158.583 eV)<sup>22</sup>. In Fig. 8 (D), the three peaks of oxygen in the samples of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite were attributed to Bi - O - Bi<sup>21</sup> (529.385 eV and 530.817 eV), CO<sub>3</sub><sup>2-</sup> group (530.540 eV and 529.817 eV)<sup>41</sup> and H - O - H (532.234 eV)<sup>42</sup>, while only H - O - H coordination peak of oxygen was observed in g-C<sub>3</sub>N<sub>4</sub> samples. Meanwhile, the CO<sub>3</sub><sup>2-</sup> on the surface of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/ $\beta$ -Bi<sub>2</sub>O<sub>3</sub> composite sample (52%) is higher than that of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> (36%). In addition, we also found that



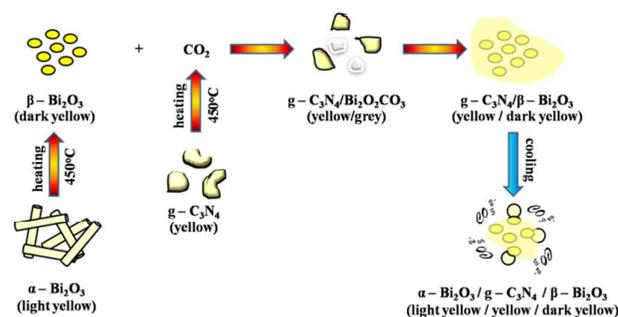
**Fig. 8** XPS spectra of (A) C1s; (B) N1s; (C) Bi4f; (D) O1s for  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub>, and g-C<sub>3</sub>N<sub>4</sub>

the binding energies of all the peaks in  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composite sample are slightly higher than those in  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> or g-C<sub>3</sub>N<sub>4</sub>, except the C1s of surface adventitious carbon and O1s of H-O-H. This is probably because of the interaction between  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>, β-Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>. The XPS results indicate the co-existence of g-C<sub>3</sub>N<sub>4</sub> [Fig. 8 (A)] and CO<sub>3</sub><sup>2-</sup> [Fig. 8 (D)] on the surface of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composites.

As a result, we proposed the formation mechanism of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composites as described in Scheme 1. Step 1:  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> transfers to β-Bi<sub>2</sub>O<sub>3</sub> at the temperature of 450 °C, while g-C<sub>3</sub>N<sub>4</sub> slowly dissolves into CO<sub>2</sub> at about 400 °C<sup>19</sup>; Step 2: The released CO<sub>2</sub> corrodes the surfaces of the microrod-like  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> by reacting with it, and produces Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub>, forming g-C<sub>3</sub>N<sub>4</sub>/Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> composite; Step 3: The Bi<sub>2</sub>O<sub>2</sub>CO<sub>3</sub> crystals further decompose into β-Bi<sub>2</sub>O<sub>3</sub> and CO<sub>2</sub> at this temperature, forming g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composites; Step 4: partial β-Bi<sub>2</sub>O<sub>3</sub> transfers to  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> during the cooling process due to the deprotection of g-C<sub>3</sub>N<sub>4</sub> and CO<sub>3</sub><sup>2-</sup>, while the other left β-Bi<sub>2</sub>O<sub>3</sub> crystals can keep its crystal structures during this process. After these 4 steps,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composites were successfully constructed. This process was partially proved by the combination of TG-DTA and XRD in our earlier research<sup>19</sup>, and supported by the XPS results in this study.

### 3.5 Photocatalytic Activity

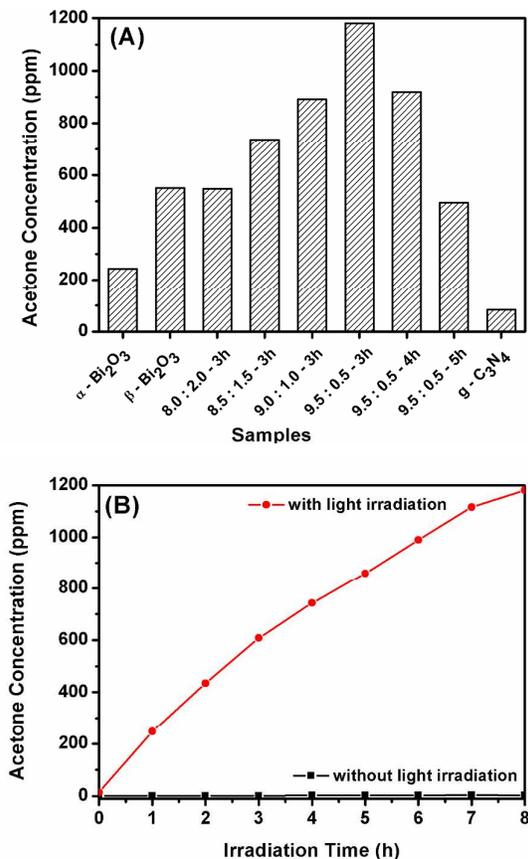
The photocatalytic activities of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composites were evaluated by the photocatalytic oxidation of gaseous IPA under visible light irradiations. The production of acetone, the unique product of this reaction, was monitored to judge the photocatalytic activities. From Fig. 9 (A), we can see that the sample of 9.5:0.5-3h shows the highest photocatalytic activity in this reaction. The production of acetone over it is almost 1200 ppm after 8 hours' irradiation, and it is almost twice more than that over β-Bi<sub>2</sub>O<sub>3</sub> (550 ppm), 5 times more than that over nano  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> (244 ppm), and 14 times more than that over g-C<sub>3</sub>N<sub>4</sub> (84 ppm). Meanwhile, other  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composite samples also shows better photocatalytic activity than the three semiconductors



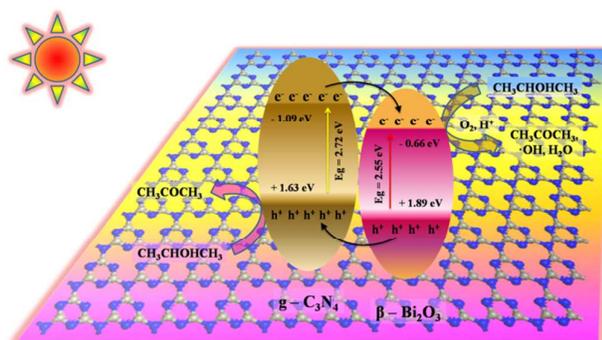
**Scheme 1** The formation mechanism of  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composite samples from  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>.

separately, except the samples of 8.0:2.0 - 3h and 9.5:0.5 -5h. Furthermore, the oxide reaction of IPA over  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composite (9.5:0.5 - 3h) in dark was also carried out for comparison. The result in Fig. 9 (B) shows that little acetone was produced under this condition.

We note that, β-Bi<sub>2</sub>O<sub>3</sub> shows the highest photocatalytic activity in the IPA oxidation among the three semiconductors,  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>, β-Bi<sub>2</sub>O<sub>3</sub> and g-C<sub>3</sub>N<sub>4</sub>. Meanwhile, the composite sample of 9.5:0.5-4h shows the highest visible light absorption (Fig. 7), and the composite sample of 9.5:0.5-5h contains the most β-Bi<sub>2</sub>O<sub>3</sub> among all the



**Fig. 9** (A) Photocatalytic oxidation of gaseous IPA over  $\alpha$ -Bi<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub>/β-Bi<sub>2</sub>O<sub>3</sub> composite samples; (B) Control experiments of the gaseous IPA oxidation



**Scheme 2** The photocatalytic oxidation mechanism of IPA over  $\alpha$ - $\text{Bi}_2\text{O}_3$ /g- $\text{C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composite samples.

composite samples. However, both of them are not the sample with the highest photocatalytic activity, but the composite sample of 9.5:0.5-3h shows the highest photocatalytic activity. In order to explain this phenomenon, we need to investigate it in another way. When we analyzed the contents of every composite sample once again (Table 2), we got the results. From Table 1, we found that, for the composite sample of 9.5:0.5-3h, the weight ratio of g- $\text{C}_3\text{N}_4$  is the most, and  $\alpha$ - $\text{Bi}_2\text{O}_3$  is almost negligible in this composite sample. So we think that g- $\text{C}_3\text{N}_4$  and  $\beta$ - $\text{Bi}_2\text{O}_3$  play the important role in the photocatalytic performance of the composite samples, and the formation of g- $\text{C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  heterojunction is the crucial factor of the high photocatalytic activity.

According to the previous report, the photooxidation of IPA to acetone undergoes two typical reaction processes as follows<sup>43-45</sup>:

- (1)  $\text{h}^+$  oxidation :  $\text{CH}_3\text{CHOHCH}_3 + \text{h}^+ \rightarrow \text{CH}_3\text{COCH}_3 + 2\text{H}^+ + \text{e}^-$
- (2)  $\text{O}_2$  oxidation :  $\text{CH}_3\text{CHOHCH}_3 + \text{e}^- + \text{O}_2 + \text{H}^+ \rightarrow \text{CH}_3\text{COCH}_3 + \text{HO}\cdot + \text{H}_2\text{O}$

Therefore, the photocatalytic reaction mechanism can be described as Scheme 2. Under the visible light irradiation, both g- $\text{C}_3\text{N}_4$  and  $\beta$ - $\text{Bi}_2\text{O}_3$  are excited and generated the photoexcited electron and hole pairs. Since the conduction band potential of g- $\text{C}_3\text{N}_4$  is more negative than that of  $\beta$ - $\text{Bi}_2\text{O}_3$ , the photogenerated electrons on the surface of g- $\text{C}_3\text{N}_4$  particles will transfer to  $\beta$ - $\text{Bi}_2\text{O}_3$ . Meanwhile the photogenerated holes on the surface of  $\beta$ - $\text{Bi}_2\text{O}_3$  will transfer to g- $\text{C}_3\text{N}_4$ , because the valence band potential of  $\beta$ - $\text{Bi}_2\text{O}_3$  is more positive than that of g- $\text{C}_3\text{N}_4$ . These two processes are both based on the well built heterojunction between g- $\text{C}_3\text{N}_4$  and  $\beta$ - $\text{Bi}_2\text{O}_3$ . Therefore, the photogenerated electrons and holes are separated effectively in the space, inducing a smaller possibility to recombine. Combining with the increased visible light absorption and decreased particle size of the sample, the photocatalytic activity is efficiently enhanced.

## Conclusion

In conclusion, this work successfully synthesized  $\alpha$ - $\text{Bi}_2\text{O}_3$ /g- $\text{C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composites in one step with an in situ method from  $\alpha$ - $\text{Bi}_2\text{O}_3$  and g- $\text{C}_3\text{N}_4$ . Based on the earlier researches, the formation mechanism of  $\alpha$ - $\text{Bi}_2\text{O}_3$ /g- $\text{C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$  composites were studied and discussed in detail. The results show that  $\text{CO}_3^{2-}$  and g- $\text{C}_3\text{N}_4$  on the surface of  $\beta$ - $\text{Bi}_2\text{O}_3$  greatly improves its thermal stability at room temperature, inducing the formation of  $\alpha$ - $\text{Bi}_2\text{O}_3$ /g- $\text{C}_3\text{N}_4$ / $\beta$ - $\text{Bi}_2\text{O}_3$

composites. More interestingly, the photocatalytic activity for the IPA oxidation is significantly enhanced by the recombination of  $\alpha$ - $\text{Bi}_2\text{O}_3$ , g- $\text{C}_3\text{N}_4$ , and  $\beta$ - $\text{Bi}_2\text{O}_3$ . According to the analysis on the practical weight ratio of everyone component, the surface heterojunction between g- $\text{C}_3\text{N}_4$  and  $\beta$ - $\text{Bi}_2\text{O}_3$  was considered to be the main reason of the photocatalytic activity enhancement. The results and discussions in this study reveal the significant effect of g- $\text{C}_3\text{N}_4$  on the crystal structures and photocatalytic activity of  $\alpha$ - $\text{Bi}_2\text{O}_3$ .

## Acknowledgements

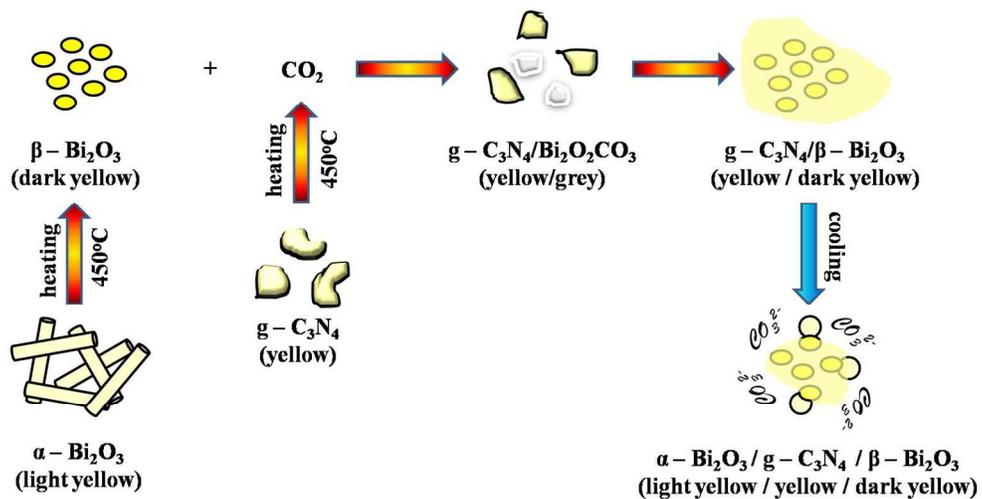
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