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1 **Efficient photo-degradation of dyes using CuWO<sub>4</sub>**  
2 **nanoparticles with electron sacrificial agents: A combination**  
3 **of experimental and theoretical exploration**

4  
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35 **Abstract**

36 CuWO<sub>4</sub> is one of the promising photocatalytic materials, responding in the visible  
37 light range, to enhance the utilization of solar energy. Here, the CuWO<sub>4</sub> nanoparticles  
38 have been synthesized via polyol-mediated synthesis method and subsequently  
39 characterized by X-ray Diffraction (XRD), scanning electron microscope (SEM),  
40 transmission electron microscope (TEM), UV-Vis Spectrophotometer combined with  
41 theoretical density functional theory (DFT) calculations. For as-prepared CuWO<sub>4</sub>  
42 samples, a strong adsorption capacity of organic pollutant MB rather than  
43 photodegradation has been observed. The first-principle calculation with  
44 Heyd–Scuseria–Ernzerhof (HSE) screened coulomb hybrid functional results indicate  
45 that localization of hybridization of O 2*p*-orbitals and Cu 3*d*-orbitals, large electron  
46 effective mass and more positive conduction band edge of CuWO<sub>4</sub> lead to low carrier  
47 mobility and thus the high recombination of excited carriers. Meanwhile, the optical  
48 absorption spectrum of experimental observation is consistent with theoretical  
49 calculation of pristine CuWO<sub>4</sub>, demonstrating few defects inhibiting light absorption.  
50 To avoid the high rate of recombination of the excited carriers, electron sacrificial  
51 agents (H<sub>2</sub>O<sub>2</sub>, Na<sub>2</sub>S<sub>2</sub>O<sub>8</sub>) are utilized to suppress the recombination. The photocatalytic  
52 activity is thus largely improved.

53

54 **Key words**

55 CuWO<sub>4</sub>, Visible light photocatalysis, Electron sacrificial agent, First-principles  
56 calculations

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## 64 Introduction:

65 With the increasing problem of environmental pollution, it is important to  
66 develop effective technology to clean waste water. Various techniques including  
67 biodegradation,<sup>1</sup> ultrafiltration,<sup>2</sup> ion exchange<sup>3</sup> and etc.<sup>4,5,6</sup> have been utilized to  
68 achieve this objective. Among these technologies, the advanced oxidation processes  
69 (AOP)<sup>7</sup> stand out as a promising way to clean water. The AOP differ from  
70 conventional physical and biological water treatment processes as the AOP generate  
71 the strong oxidant hydroxyl radicals ( $\bullet\text{OH}$ ) to degrade toxic and refractory pollutants  
72 (organic carbon) into simple and harmless inorganic molecules (carbon dioxide and  
73 water) without producing secondary pollutants. The  $\bullet\text{OH}$  generation can be initiated  
74 by primary oxidants ( $\text{H}_2\text{O}_2$  or  $\text{O}_3$ ),<sup>8</sup> energy sources (UV-light, ultrasonic)<sup>9,10</sup> or  
75 catalysts.<sup>11</sup> Semiconductor materials could absorb solar energy and excite carries to  
76 generate oxidant for further photocatalytic degradation of organic pollutants. For a  
77 superior photocatalyst, the band edges should align up with water redox levels.  
78 Specifically, the top of valence band ( $E_v$ ) is required more positive than the redox  
79 potential  $\bullet\text{OH}/\text{H}_2\text{O}$  ( $\sim 2.5$  V vs. NHE).<sup>12</sup> When solar light is incident on a  
80 semiconductor material, the photo-generated electrons and holes react with water and  
81 form oxidants ( $\text{O}_2^-$ ,  $\text{H}_2\text{O}_2$  and  $\text{O}_3$ ) and essentially produce  $\bullet\text{OH}$ .<sup>13,14</sup>

82 The semiconductor materials have been used to generate oxidants for several  
83 years<sup>15,16</sup> since the photocatalytic splitting of water was discovered on the  $\text{TiO}_2$   
84 electrodes by Fujishima and Honda in 1972.<sup>17</sup> To date, anatase  $\text{TiO}_2$  dominates the  
85 photocatalysis market owing to its low cost, non-toxicity, highly catalytic activity and  
86 chemical stability.<sup>18,19</sup> Whereas,  $\text{TiO}_2$  with a band gap of 3.2 eV displays a low  
87 efficiency ( $\sim 5\%$ ) of utilizing solar energy. In order to ultimately harvest solar energy,  
88 it is important to continue searching for visible light driven photocatalysts.<sup>20,21</sup>

89  $\text{CuWO}_4$ , as a ternary narrow band gap semiconductor ( $E_g \sim 2.2$  eV), is an ideal  
90 high-efficiency semiconductor photocatalyst due to its absorption of visible-light<sup>22,23</sup>

91 with reasonable valence band alignments with  $\bullet\text{OH}/\text{H}_2\text{O}$  energy level. Moreover, the  
92 catalytic performances could also be influenced by other factors such as crystallinity,  
93 defects and interface, which introduce various electronic structures including band  
94 tails,<sup>24</sup> defect states,<sup>25</sup> and interfaces states.<sup>26</sup> These states significantly impact on  
95 carrier mobility,<sup>27</sup> carrier's recombination,<sup>28</sup> electronic conductivity<sup>29</sup> and etc,<sup>30</sup>  
96 resulting in the variations of the catalytic activity. Without full understanding of the  
97 microscopic electronic structures, it would be difficult to further improve the  
98 photocatalytic performance. Here, we combined theoretical calculations and  
99 experimental methods to link the electronic structures and the catalytic performance  
100 of  $\text{CuWO}_4$ . Based on the DFT calculations, the catalytic activity has been improved  
101 with the presence of electron capture agents. In such a way, these findings provide  
102 insights into further promising photocatalyst design.

103 In this work,  $\text{CuWO}_4$  was prepared by polyol-mediated synthesis method. The  
104 synthesized samples are characterized by the X-ray Diffraction (XRD), scanning  
105 electron microscope (SEM), transmission electron microscopy (TEM) and UV-Vis  
106 Spectrophotometer. As-prepared powder displays a strong adsorption capacity of  
107 organic pollutant MB rather than photodegradation. First-principle calculations point  
108 out that the disadvantage of localized band edge states of hybridization of O  
109  $2p$ -orbitals and Cu  $3d$ -orbitals causes the high recombination of excited carrier. On  
110 the other hand, the experimental optical absorption spectrum of sample is consistent  
111 with theoretical calculation of pristine  $\text{CuWO}_4$  which indicates no problem for our  
112 synthesized samples to absorb light. In the presence of electronic sacrificial agent, the  
113 excited holes are survived and thus  $\text{CuWO}_4$  displayed a high photocatalytic  
114 performance to degrade the MB dye.

## 115 **Experiment and Calculation Method**

### 116 **1. Polyol Mediated Synthesis Method**

117 Copper nitrate ( $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$ , AR, purity  $\geq 99.5\%$ ) and sodium tungstate  
118 ( $\text{Na}_2\text{WO}_4 \cdot 2\text{H}_2\text{O}$ , AR, purity  $\geq 99.5\%$ ) were used to prepared  $\text{CuWO}_4$ . Copper nitrate  
119 (1 mmol) was dissolved in 50 mL diethylene glycol (DEG) and stirred by magnetic  
120 stirring at room temperature. Sodium tungstate (1 mmol) was dissolved in 1 mL  
121 deionized water and was injected into copper nitrate solution quickly with magnetic  
122 stirring. After stirring 3 minutes (min), the products were separated from these  
123 suspensions via high-speed centrifuge at 10000 rpm for 5 min and washed by  
124 sequential centrifugation/redispersion from/in ethanol for 5 times to remove  
125 redundant impurity and DEG completely, and then dried in the oven at 70 °C for 12  
126 hours. In the annealed process, temperature was kept at 500 °C for 1 hour in the air.

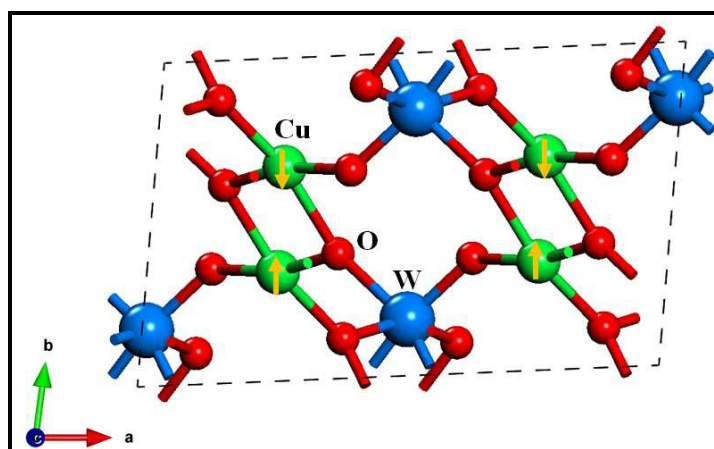
## 127 **2. Evaluation of Photocatalytic Activity**

128 The photocatalytic ability of  $\text{CuWO}_4$  was evaluated by degradation of MB. The  
129 40 mg  $\text{CuWO}_4$  powder was dispersed in methylene blue solution (100 ml, 10 mg/L) in  
130 a beaker. Subsequently, it was stirred about 30 min in dark in order to reach  
131 adsorption-desorption equilibrium. A xenon lamp without filter has been used as the  
132 light source and the light intensity remains at  $20 \text{ mW} \cdot \text{cm}^{-2}$  measured by an irradiance  
133 meter (Model: FZ-A, Beijing Normal University, China) to reduce solution  
134 evaporation. During photogradation of MB solution, the magnetic stirring was kept  
135 running. The photodegradation was quantified by monitoring the concentration of MB  
136 at its maximum of absorption (664 nm). Commercially available Degussa P25  
137 (nanoscale  $\text{TiO}_2$  powder; 80% anatase, 20% rutile; particle diameter: 25 ~ 30 nm) as a  
138 reference was used to compare with the photocatalytic performance of  $\text{CuWO}_4$ .

## 139 **3. Theoretical Model and Calculation Details**

140 The crystal structure of  $\text{CuWO}_4$  is presented in Fig. 1, containing two formula  
141 units with characteristic corner-linked  $\text{CuO}_6$  and  $\text{WO}_6$  octahedral. The Jahn-Teller  
142 effect of the  $\text{Cu}^{2+}$  cation causes a pseudo-tetragonal elongation of the  $\text{CuO}_6$  octahedral.  
143 The ab initio calculations were performed based on the density functional theory

144 (DFT), as implemented in plane-wave based code Vienna ab initio Simulation  
145 Package (VASP). The generalized gradient approximation with the  
146 Perdew-Burke-Ernzerhof version of the exchange-correlation potential was  
147 employed.<sup>31</sup> A large energy cutoff of 550 eV was adopted. Brillouin zone was  
148 sampled by the set of  $(5 \times 4 \times 5)$   $k$ -points for  $\text{CuWO}_4$  to balance calculation efficiency  
149 and accuracy. The convergence criterion for energy is chosen as  $10^{-4}$  eV and the  
150 maximum Hellmann-Feynman force acting on each atom is less than  $0.01$  eV/Å in  
151 ionic relaxation calculations. The calculated lattice constants are  $a = 4.78$  Å,  $b = 6.00$   
152 Å,  $c = 4.93$  Å;  $\alpha = 92.77^\circ$ ,  $\beta = 93.51^\circ$ ,  $\gamma = 81.48^\circ$ . Considering that GGA usually  
153 overestimates the lattice constant slightly,<sup>32,33</sup> the calculated results are in good  
154 agreement with the previous results using the GGA functional ( $a = 4.84$  Å,  $b = 6.05$  Å,  
155  $c = 4.94$  Å)<sup>34</sup> and the experimental values ( $a = 4.69$  Å,  $b = 5.83$  Å,  $c = 4.88$  Å).<sup>35</sup> For  
156 electronic structure calculations, it is well known that DFT underestimates the band  
157 gap, thus the electronic structure with the PBE relaxed structure was calculated using  
158 the Heyd–Scuseria–Ernzerhof (HSE)<sup>36</sup> hybrid functional, in which a portion (16%) of  
159 Hartree–Fock exchange was mixed with the PBE functional to produce a band gap of  
160  $\sim 2.2$  eV which is highly consistent with experimental observations.<sup>37</sup>



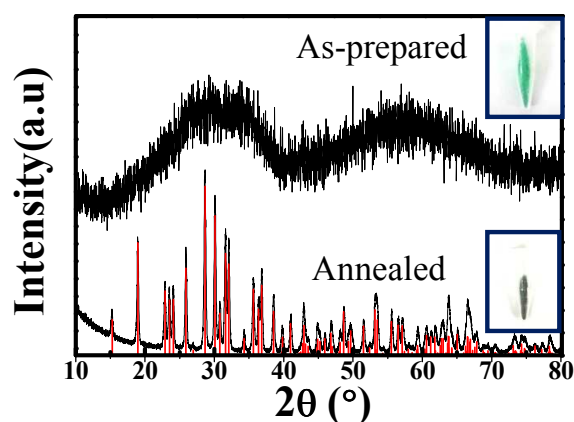
161

162 **Fig. 1** (color online) The crystal structure of  $\text{CuWO}_4$ , the green, blue and red balls  
163 represent Cu, W and O atoms, respectively. Arrows represent the spin directions of  
164 Cu.

## 165 Results and Discussion

166 Due to the high polarity of multidentate alcohols such as ethylene glycol,  
167 diethylene glycol or glycerol, polyol-mediated synthesis is considered as an effective  
168 way to control nucleation and growth of nanoparticles, stabilize the particle surface  
169 and avoid agglomeration.<sup>38,39</sup> In this work, we adopted this method to synthesize  
170  $\text{CuWO}_4$ .

171 The as-prepared  $\text{CuWO}_4$  powder shows green color (inset in Fig. 2) and the  
172 according XRD peaks (Fig. 2) are broadened which indicates the amorphous phase of  
173 the sample power. After heating the sample under 500 °C for 2 hours, the color turned  
174 into dark grey (inset in Fig. 2) and the XRD pattern displays the formation of the pure  
175 phase of  $\text{CuWO}_4$  (Fig. 2). The color change before and after sample annealing results  
176 from the typical quantum size effect.



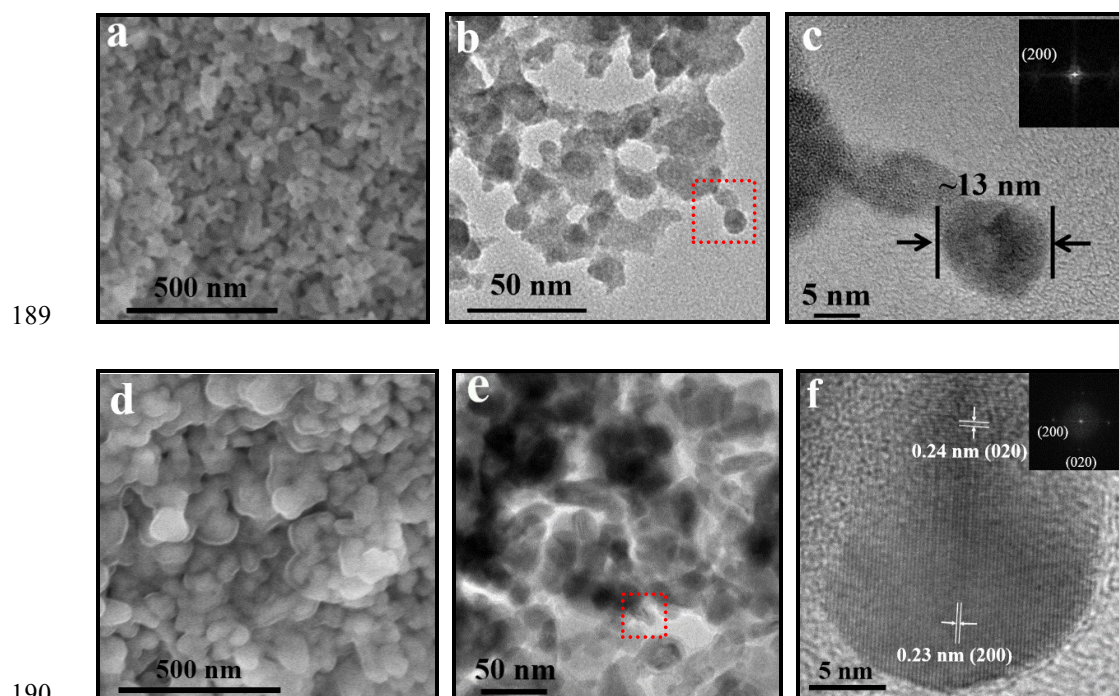
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178 **Fig. 2** (color online) XRD pattern of as-prepared and annealed samples.

179 Fig. 3(a), (b) and (c) describe morphologies of as-prepared sample. Irregular  
180 nanoparticles have been observed by SEM (Fig. 3(a)), the grain size varying from  
181 10~20 nm has been verified by TEM (Fig. 3(b), (c)) and the specific surface area is  
182 expected to be large. The selected area electron diffraction (SAED) (Fig. 3(c)) shows  
183 the blurry spots corresponding to low degree of crystallinity of as-prepared  $\text{CuWO}_4$ .



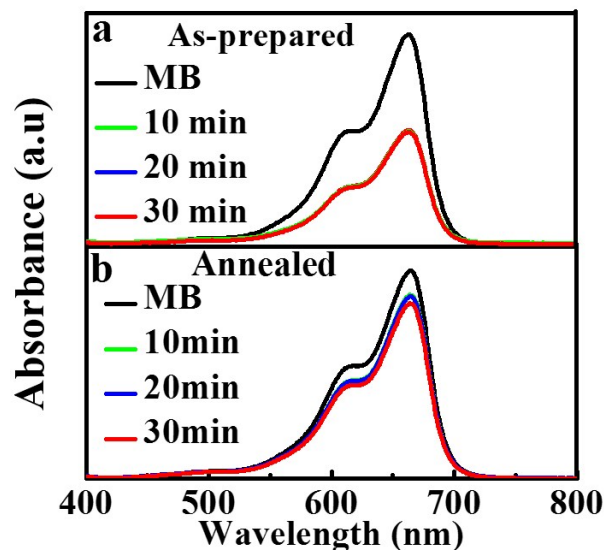
184 For the annealed sample, SEM and TEM (Fig. 3(d), (e), (f)) show that the particle size  
185 significantly grows up to  $\sim 50$  nm and nanoparticles seriously reunite due to regrowth  
186 and recrystallization during the annealed process. On the other hand, clear spots of  
187 SAED (Fig. 3(f)) also identify the improvement of crystallinity after the sample was  
188 annealed.



191 **Fig. 3** (color online) SEM, TEM images of as-prepared sample (a), (b) and annealed  
192 sample (d), (e); (c) and (f) are magnified high-resolution TEM images of red boxes in  
193 (b) and (e), respectively.

194 For the photodegradation of MB, the as-prepared powder displays a strong  
195 adsorption capacity of organic pollutant MB within the first 10 min (Fig. 4(a)). No  
196 photodegradation phenomenon has been observed during this period. When  
197 continuing increasing observing time larger than 10 min, no more adsorption is  
198 displayed since as-prepared  $\text{CuWO}_4$  has reached its adsorption equilibrium. For the  
199 annealed sample, the particle size significantly grows up to  $\sim 50$  nm and nanoparticles  
200 reunite seriously which have been observed by SEM and TEM (Fig. 3d, 3e, 3f).  
201 Consequently, the decrease of the surface areas leads to the decrease of MB

202 adsorption, as shown in Fig. 4(b). Also, the annealed sample displays no  
203 photocatalytic behavior. Little variation of the photocatalytic behaviors for the  
204 as-prepared sample and the annealed one proves that particle size is not the key to  
205 impact the catalytic activity for this case.



206

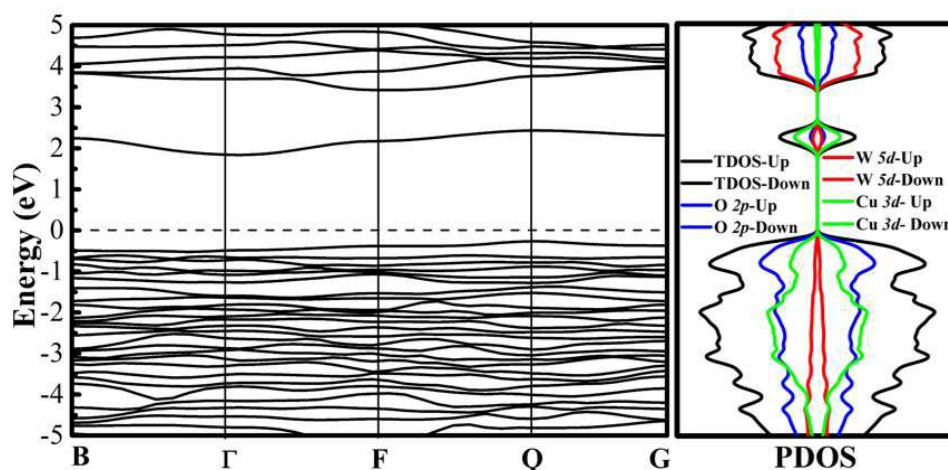
207 **Fig. 4** (color online) The adsorption of MB (a) as-prepared  $\text{CuWO}_4$  sample and (b)  
208 annealed sample.

209 Intrinsically, two critical factors would govern the photocatalytic activity, i.e.  
210 carrier conductivity and optical absorption. For the former one, it is governed by the  
211 band edge shapes of  $\text{CuWO}_4$ . The later one is influenced by light absorption spectrum.  
212 In order to access the failure mechanism of the photocatalytic performance of  $\text{CuWO}_4$ ,  
213 electronic structure and optical spectrum should be calculated.

214 With the advantage of the start-of-art supercomputer and computational  
215 algorithm, it is now feasible to access the fundamental electronic structures at the  
216 atomic level. In this work, we employed density functional theory (DFT) to explore  
217 the electronic structure and optical spectrum of the  $\text{CuWO}_4$ .

218 Fig. 5 illustrates the calculated band structure along high symmetric k-point B  
219  $(0.5, 0, 0) - \Gamma (0, 0, 0) - F (0, 0.5, 0) - Q (0, 0.5, 0.5) - G (0, 0, 0.5)$  in the first

220 Brillouin zone. The conduction band minimum (CBM) and valence band maximum  
 221 (VBM) are located at  $\Gamma$  k-point and Q k-point, respectively, indicating an indirect  
 222 band gap. The gap value is  $\sim 2.2$  eV which is highly consistent with experimental  
 223 result.<sup>37</sup> Also, in the bottom region of conduction band at  $\Gamma$  point, the band is rather  
 224 flat. By extracting the curvature of the CBM and VBM, we obtained a large the  
 225 effective mass of electron and hole ( $m_e^* = 59 m_0$ ,  $m_h^* = 57 m_0$ ,  $m_0$  is free electron  
 226 mass), resulting in the poor electron and hole conductivity, respectively.<sup>23</sup>



227

228 **Fig. 5** (color online) The band structure diagram (left panel) and the spin-dependent  
 229 density of states (right panel) for  $\text{CuWO}_4$ . Fermi level is set at zero eV in both panels.

230 Fig. 5 also shows the normalized total and projected density of states (PDOS).  
 231 The spin-up states are identical to the spin-down ones. Thus,  $\text{CuWO}_4$  presents an  
 232 anti-ferromagnetic (AFM) ground state with the magnetic moment of  $0.68 \mu_B$  for Cu  
 233 atoms and  $0.05 \mu_B$  for O atoms, being consistent with experiment observations of  $\mu_{\text{Cu}}$   
 234  $= 0.67 \mu_B$ ,  $\mu_{\text{O}} = 0.06 \mu_B$ <sup>35</sup> and PBE+U calculation of  $\mu_{\text{Cu}} = 0.74 \mu_B$ ,  $\mu_{\text{O}} = 0.07 \mu_B$ .<sup>40</sup>

235 In Fig. 5, the O-2p and Cu-3d orbitals hybrid and dominate the band edges. More  
 236 specifically, valence band maximum is mainly composed of the orbital hybridization  
 237 of oxygen 2p-orbitals and copper 3d-orbitals, leading to the more positive position of  
 238 the VBM,<sup>41</sup> and the strong oxidizing property of the excited holes. Therefore, we

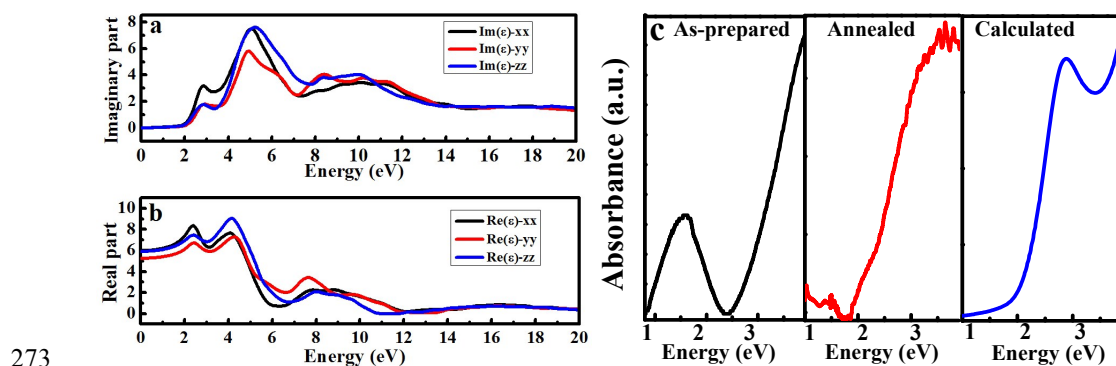
239 speculate that O and Cu could be the activity sites for photocatalysis. The excited hole  
240 either directly reacts with organic pollutant molecular to decompose it or reacts with  
241 OH<sup>-</sup> to produce strong oxidant hydroxyl radicals. Hybridization of Cu-3*d* and O-2*p*  
242 results in more positive CBM and narrow band gap which improves the visible-light  
243 absorption. Nevertheless, Hybridization of Cu-3*d* and O-2*p* also lead to more positive  
244 conduction band edge of CuWO<sub>4</sub> compared to H<sup>+</sup>/H<sub>2</sub> reduction level in solution.<sup>42</sup> In  
245 other words, there is no excited electron acceptor in this photocatalytic system, as a  
246 result, excited electrons will gather on the CuWO<sub>4</sub> surface and cause more severe  
247 surface recombination. Moreover, the localized states of hybridization of O 2*p* and Cu  
248 3*d* orbitals could lead to the lower carrier mobility, high recombination rate of excited  
249 carrier and essentially result in low efficient photocatalytic behavior.

250 In addition to the material intrinsic electronic properties, optical adsorption is  
251 another important factor to govern the photocatalytic activity. For the absorption  
252 spectrum of bulk material, it is formatted as:

$$253 \quad I(\omega) = \sqrt{2}\omega \left[ \sqrt{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)} - \varepsilon_1(\omega) \right]^{1/2}$$

254  $\varepsilon_2$  and  $\varepsilon_1$  are imaginary part and real part of dielectric function, respectively,  $\omega$  is  
255 angular frequency. In order to obtain the absorption spectrum, dielectric constant was  
256 calculated via DFT. Fig. 6 (a) and (b) show the calculated results of the imaginary part  
257 and real part of dielectric function for CuWO<sub>4</sub> along three Cartesian directions,  
258 respectively. The optical anisotropy is due to low crystal symmetry and the  
259 peculiarities in the crystal structure, i.e., the existence of the bridge-oxygen ions  
260 connecting with neighboring CuO<sub>6</sub> and WO<sub>6</sub> complexes. Based on the calculated  
261 dielectric function curves, we obtained the theoretical absorption spectrum, as shown  
262 in Fig. 6 (c). The light absorption begins about 2.0 eV, corresponding to the electron  
263 excitation from the VBM to the CBM (Fig. 6 (c)). Experimentally, the as-prepared and  
264 annealed CuWO<sub>4</sub> has been used to test the absorption spectrum by UV-Vis  
265 Spectrophotometer. For the as-prepared sample, the localized absorption peak

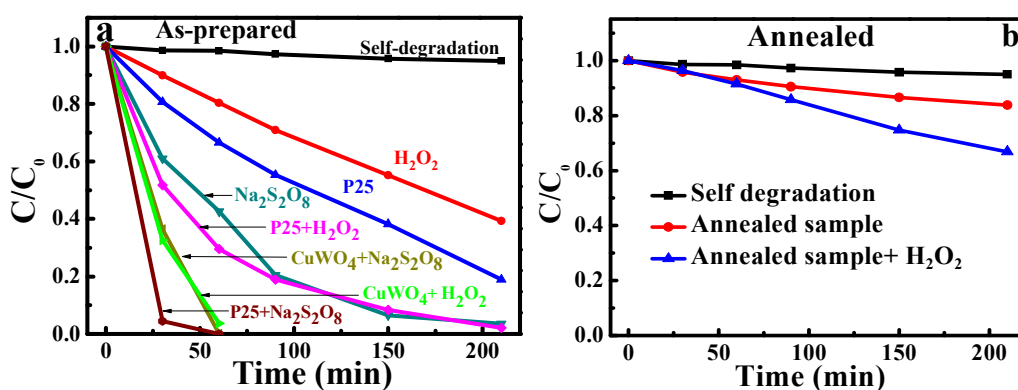
266 appeared at 1.5 eV as the subgap absorption,<sup>43</sup> which originates from bonding defects  
267 induced located states in the forbidden energy gap.<sup>44</sup> This specific subgap absorption  
268 helps to improve the visible-light absorption. Compared the theoretical and pure  
269 annealed sample absorption spectrum, it is found that they are consistent with each  
270 other.<sup>45</sup> This means the light absorption of sample is close to the theoretical value  
271 attributed to few defects and impurities hindrance light absorption.<sup>46,47</sup> In other words,  
272 our sample shows bulk-like absorption behavior without impacting by defect levels.



273  
274 **Fig. 6** (color online) The dielectric function of CuWO<sub>4</sub> calculated by DFT with HSE  
275 functional: (a) imaginary part, (b) real part; (c) The absorption spectrum of CuWO<sub>4</sub>  
276 obtained from as-prepared sample, annealed sample and theoretical simulation.

277 On the basis of the theoretical analysis, charge carrier separation would be the  
278 key to influence the photocatalytic activity. Thus, promoted separation of the excited  
279 carriers would be an effective method to achieve the photocatalytic degradation of  
280 MB. In this work, we utilized electron sacrificial agents to examine the charge  
281 separation effect on the photocatalytic performance.<sup>48</sup> To confirm the electron  
282 capture agent's boost effect on enhancing the photodegradation by as-prepared  
283 CuWO<sub>4</sub>, 1 mmol H<sub>2</sub>O<sub>2</sub> or Na<sub>2</sub>S<sub>2</sub>O<sub>8</sub> was dispersed into the MB solution to check the  
284 catalytic activity. Fig. 7 (a) indicates that individual H<sub>2</sub>O<sub>2</sub> or Na<sub>2</sub>S<sub>2</sub>O<sub>8</sub> displays a  
285 limited degraded performance. Surprisingly, in as-prepared CuWO<sub>4</sub> (40 mg)  
286 combined with 1 mmol H<sub>2</sub>O<sub>2</sub> or Na<sub>2</sub>S<sub>2</sub>O<sub>8</sub>, we found that MB decomposes rapidly and  
287 the MB concentrations reaches to zero within 60 min. (see Fig. 7 (a)). As a

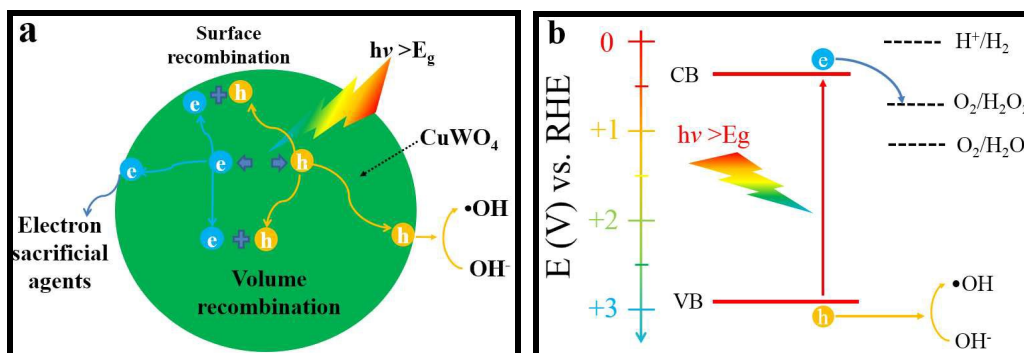
288 comparison, the as-prepared  $\text{CuWO}_4$  with electronic capture agents ( $\text{H}_2\text{O}_2$  or  $\text{Na}_2\text{S}_2\text{O}_8$ )  
 289 shows superiority over the same amount of P25 and P25 together with  $\text{H}_2\text{O}_2$  rather  
 290 than  $\text{Na}_2\text{S}_2\text{O}_8$  in weight in terms of the decomposition of MB. In any case, P25 with  
 291  $\text{Na}_2\text{S}_2\text{O}_8$  shows the highest efficiency because of the stronger oxidation of  $\text{Na}_2\text{S}_2\text{O}_8$   
 292 regards to  $\text{H}_2\text{O}_2$  (Fig.7 (a)). To examine the role of hole to decompose MB, we  
 293 utilized  $(\text{NH}_4)_2\text{C}_2\text{O}_4$  to exhaust the supply of excited holes, and we found that MB  
 294 degradation with  $(\text{NH}_4)_2\text{C}_2\text{O}_4$  is identical with the self-degradations under UV-Vis  
 295 light. In other words, hole is the key to decompose MB in our case.



296

297 **Fig. 7** (color online) The photodegradation of MB (a) as-prepared  $\text{CuWO}_4$  sample,  
 298 P25 and (b) annealed sample added electron sacrificial agents, and only electron  
 299 sacrificial agent without any catalyst.

300 In order to access the role of electron sacrificial agent to assist in the catalytic  
 301 performance of the as-prepared  $\text{CuWO}_4$ , we proposed the following schematic  
 302 diagram to address how electron sacrificial agents improve the photocatalytic  
 303 efficiency (Fig. 8(a)).



304

305 **Fig. 8** (color online) (a) Photocatalytic process in the presence of electron sacrificial  
 306 agents. (b)  $\text{CuWO}_4$  band alignments with  $\text{H}^+/\text{H}_2$ ,  $\text{O}_2/\text{H}_2\text{O}_2$ ,  $\text{O}_2/\text{H}_2\text{O}$  and  $\bullet\text{OH}$  energy  
 307 levels.

308 When the  $\text{CuWO}_4$  absorbs the photon energy larger than its band gap, the excited  
 309 electrons ( $e^-$ ) jump from the valence band to the conduction band and form  
 310 electron-hole pairs. Parts of them recombine in the as-prepared  $\text{CuWO}_4$  particles and  
 311 the others move to the surfaces (Fig. 8 (a)). Meanwhile, owing to the improper band  
 312 alignment with water redox potentials, as presented in Fig. 8 (b), severe surface  
 313 recombination occurs as discussed above. Consequently, no enough  $\text{h}^+$  combines with  
 314  $\text{OH}^-$  and forms  $\bullet\text{OH}$  because reactive time (about  $10^{-3}$  s) is much longer than the  
 315 surface recombining time (about  $10^{-12}$  s). When the electron sacrificial agent ( $\text{H}_2\text{O}_2$  or  
 316  $\text{Na}_2\text{S}_2\text{O}_8$ ) is introduced into MB solution, the excited electrons on surface are captured  
 317 by electron sacrificial agents. The carrier recombination is thus suppressed and the  
 318 excited holes have enough time to oxidize the  $\text{H}_2\text{O}$  molecules and generate the strong  
 319 oxidant  $\bullet\text{OH}$ . In addition to the role of charge separation from electron sacrificial  
 320 agents, the strong MB adsorption on the as-prepared  $\text{CuWO}_4$  surface could be another  
 321 factor to enhance the photodegradation

322 For any photocatalyst, crystallinity, electronic structure and possibly surface  
 323 charge would be crucial to determine the catalytic performance. In the annealed  
 324  $\text{CuWO}_4$  sample with diameter of  $\sim 50$  nm, even in the presence of electron sacrificial  
 325 agent, very limited catalytic activity was observed in Fig. 7(b). The limited activity  
 326 could be attributed to the lower adsorption capacity arising from the high crystallinity

327 (Fig. 4 (b)). Compared to as-prepared sample, the localized absorption peak at 1.5 eV  
328 disappeared in the absorption spectrum of annealed sample (Fig. 6 (c)), decreasing the  
329 visible-light absorption and photocatalytic efficiency. From the perspective of surface  
330 charge, as an n-type semiconductor,<sup>49</sup> band will upper bend when contacting with  
331 solution.<sup>50</sup> As a result, the exited holes tend to transfer to the nanoparticles surface,  
332 then directly react with MB or produce •OH which leads to the decomposition of MB  
333 pollutants. For the as-prepared amorphous phase, it provide more active sites for  
334 surface photogenerated carriers and prevent them from rapid recombination due to  
335 high disorders, thus promoting carrier transfer and photocatalytic reactions.<sup>51</sup> After  
336 annealing the sample, the particle size increases and thus the active sites significantly  
337 reduce. Consequently, the carrier recombination became severe, and the  
338 photocatalytic efficiency decreased.

### 339 **Conclusions**

340 In summary, we have investigated the catalytic activity of CuWO<sub>4</sub> nanoparticles  
341 in experiment, combined with the DFT electronic structures calculations. The first  
342 principle calculation results illustrate that more positive conduction band edge  
343 position than H<sup>+</sup>/H<sub>2</sub>O level, low carrier mobility and high recombination rate of  
344 CuWO<sub>4</sub> result in an effective adsorption rather than photodegradation of MB for the  
345 as-prepared sample. The annealed samples with larger particle size display even a  
346 smaller adsorption capability of MB. In the presence of electron sacrificial agents, the  
347 photocatalytic efficiency of the as-prepared CuWO<sub>4</sub> is significantly improved, due to  
348 the suppression of the combination of photo-generation carrier and the enhancement  
349 of the formation of •OH. On the other hand, the annealed sample displays rather low  
350 catalytic behavior owing to the low adsorption capability.

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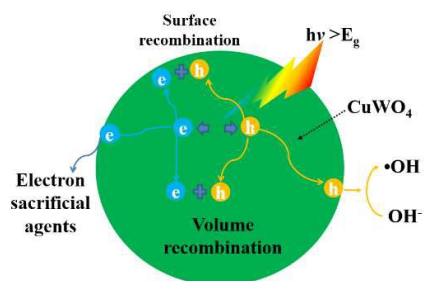
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## Table of Contents



Photocatalytic process of  $\text{CuWO}_4$  in the presence of electron sacrificial agents.