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Journal Name

ARTICLE

Crosslinkable polyesters based on monomers derived from renewable lignin

Jie Zhang,^a Chengcai Pang^a and Guolin Wu^{a,*}Received 00th January 20xx,
Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

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Crosslinked polyesters were successfully prepared using thiol-ene click chemistry. Vanillic acid, syringic acid, and p-hydroxybenzoic acid have similar aromatic structures. They can be derived from renewable resource, lignin. In this paper, they were used as raw materials for the synthesis of unsaturated thermoplastic polyesters, by polycondensation in solvent. The obtained polyesters were characterized by FTIR and NMR, and their tensile properties were tested. Compared to the other two unsaturated polyesters, the vanillic acid based unsaturated polyester (P1) was found to have the best tensile property. The latter was chosen for the preparation of the crosslinked polyester and further studies were carried out. Differential scanning calorimetry (DSC) revealed an amorphous character of P1, having a T_g of 35.8 °C. Thermogravimetric analysis (TGA) and tensile tests were conducted to study the thermal and mechanical properties of P1 and the crosslinked polyesters. The weight-average molecular weight (M_w) of P1 was found to be 73000 g mol⁻¹. The unsaturated bonds in the polymers were found to have a special feature, which improved the flexibility of the polyesters on crosslinking. Moreover, both the unsaturated polyesters and the crosslinked polyesters were found to be biodegradable.

Introduction

Today, a vast majority of commercial synthetic polymeric materials (plastic, rubber, and fiber) are obtained from fossil fuels. These fossil resources are limited and their non-degradability leads to serious environmental pollution. This stimulates investigation, aimed at developing macromolecular materials from renewable feedstock that minimize the effects, detrimental to the environment, by their usage.^[1-4] Oils from plant or animal resources, represent a promising class of raw materials for polymer synthesis, owing to their abundance, inherent biodegradability, and versatility.^[5-6] The common renewable resources reported earlier are carbohydrates,^[7-8] plant-based oils,^[9-10] lignin,^[11] etc. Amongst the different types of polymeric systems, based on renewable resources, thermosetting resins are highly-crosslinked polymers that are cured by heat, pressure, light radiation, or a combination of these energy sources, and cannot be reshaped after curing.^[12-13]

Synthesis of polyesters from renewable resources has been an appealing topic in recent years, owing to the ecological and

economical aspects.^[14] It is well known that lignin is a renewable material that has the potential to yield valuable single aromatic chemicals when strategically depolymerized, and it is the second most abundant natural raw material, only after cellulose.^[15] Vanillin, derived from lignin, has been identified as a suitable bio-based replacement for use in vinyl ester resins, mainly due to its aromatic character. The incorporation of aromatic groups into cured resins can provide structural rigidity and thermal stability by inhibiting the rotational freedom of the polymeric network.^[16-17] As an attractive bio-based monomer, vanillin has been successfully incorporated into novel polymers for use in a wide variety of commercial applications.^[18] Moreover, syringic acid and p-hydroxybenzoic acid, having structures similar to vanillin, are also potential bio-based monomers derived from lignin. Click chemistry reactions are often cited as chemical syntheses that are consistent with the goals of green chemistry.^[19] Crosslinking, brought about by UV or visible light, is a widespread and convenient approach, leading to enhancement in thermal and mechanical properties of the bio-based unsaturated polyesters.^[14] Thiol-ene chemistry has developed in the past few decades and has been identified as a 'click' method, due to its high conversions and low side products.^[20] Thiol-ene coupling of various polymeric enes has also been extensively studied.^[21] The unsaturated bond in the polyester monomer allows these polymers to be crosslinked.^[22] Crosslinking is a very important process in the formation of elastomeric films, wherein the formed network determines the applications and properties of the film.^[23-24]

^a Key Laboratory of Functional Polymer Materials, Institute of Polymer Chemistry, Nankai University, Tianjin 300071, PR China. Address here.

*To whom correspondence should be addressed at: Key Laboratory of Functional Polymer Materials, Institute of Polymer Chemistry, Nankai University, Tianjin 300071, PR China.

† Footnotes relating to the title and/or authors should appear here.

Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/x0xx00000x

The three-dimensional structures of the networks and the final properties of the films depend not only on the type and content of crosslinking agents but also on the ratio of thiol to double bond, which is a crucial parameter.^[25]

In our previous studies, several saturated polyesters using vanillic acid were synthesized.^[29-30] In the present study, three unsaturated polyesters using vanillic acid, syringic acid, and *p*-hydroxybenzoic acid have been synthesized under mild reaction conditions. 2,2'-(Ethylenedioxy) diethanethiol has been used as the crosslinker to obtain thin crosslinked films, after irradiation in solution state by UV light, in the presence of a photoinitiator, (2,2-dimethoxy-2-phenylacetophenone). They have been found to have better tensile properties. Furthermore, different strategies to obtain modified aromatic polyesters, which can be either chemically degraded under smooth conditions or biologically, are currently drawing a lot of interest.^[26-27] Most of the engineering plastics that are obtained from synthetic polymers, are non-biodegradable.^[28] In this work, the degradation of bio-based polyesters has also been investigated in different media.

Experimental section

1. Materials

Vanillic acid (4-hydroxy-3-methoxybenzoic acid, 98%), syringic acid (98%) and *p*-hydroxybenzoic acid (98%) were purchased from Shanghai Darui Fine Chemical Co. Ltd. Trans-1,4-dibromo-2-butene (98%), and 2,2'-(ethylenedioxy) diethanethiol (95%) were obtained from Heowns (Tianjin, China). Potassium carbonate (99%), 2,2'-dimethoxy-2-phenylacetophenone (DMPA, 98%), 1-methyl-2-pyrroline (99%), chloroform (99%), methanol anhydrous (99.5%), citric acid (99.5%), and trisodium citrate dehydrate (99%) were supplied by Tianjin Chemical Reagent Co. Ltd. Lipase from porcine pancreas was purchased from Sigma-Aldrich. Methanol, 1-methyl-2-pyrroline and other solvents were dried according to the standard methods in literature.^[31]

2. General instrumentation and methods

NMR spectra were recorded in CDCl₃ at 25°C, using a Bruker AC-400 NMR spectrometer. Tetramethylsilane was used as the internal standard. Fourier transform infrared spectra (FTIR) were measured at room temperature using a Bio-Rad FTS6000 spectrophotometer. Each polymer sample for FTIR was prepared by grinding the polymer adequately with KBr powder, followed by compressing the mixture to form a pellet. The molecular weights and molecular weight distributions of the obtained polymers were determined by gel permeation chromatography (GPC). THF was used as the eluent at a flow rate of 1.0 mL min⁻¹. The average molecular weights were calibrated against monodisperse polystyrene (PS) standards. Thermogravimetric analysis (TGA) was performed using a Seiko Exstar 6000 TGA quartz rod microbalance. As a general method, the polymer sample was heated from 27 to 800°C at a

heating rate of 10°C min⁻¹. The temperature at which 5% weight loss occurred and the temperature at which maximum rate of degradation took place were recorded. Differential scanning calorimetric (DSC) studies were carried out using DSC Q100 from TA instruments. Polymer samples were first heated from -50°C to 150°C and the glass transition temperature (*T_g*) was determined in the second heating run. All runs were carried out at a rate of 10°C min⁻¹. The tensile assays were conducted in triplicates on rectangular samples having the dimensions (length × width × thickness = 12 mm × 2 mm × 0.5 mm). The strain was measured by applying a ramp rate of 0.5 N min⁻¹ at 25°C. A preload force of 0.05 N and a soak time of 3 min were employed. All films for tensile tests were obtained by casting chloroform solutions having concentrations of 0.1 g mL⁻¹.

3. General procedure for the preparation of unsaturated polyesters

The unsaturated polyesters were prepared according to a procedure mentioned in the literature.^[32] Trans-1,4-dibromo-2-butene (8.56 g, 0.04 mol) was dissolved in 100 mL 1-methyl-2-pyrroline, containing vanillic acid (6.72 g, 0.04 mol) (or 0.04 mol syringic acid, or 0.04 mol *p*-hydroxybenzoic acid) in a three-neck round-bottom flask, wrapped by a foil, by stirring at 40°C. Potassium carbonate (11 g, 0.08 mol) was added to the clear solution and the mixture was refluxed for 60 hours under N₂ atmosphere. The viscous solution was diluted with THF and poured into a large excess of cold water, which was acidified by adding a few drops of concentrated HCl. The precipitated polymer was filtered and then washed repeatedly with distilled water till neutral pH, then washed with methanol and finally dried under vacuum till the product attained a constant weight. The yields of the products were more than 90%.

4. Preparation of the crosslinked thermoplastic polyesters

The crosslinked thermoplastic polyesters were obtained by mixing 1 g unsaturated vanillic acid-based polyester, containing 4.5 mmol C=C, with a specific amount of 2,2'-(ethylenedioxy) diethanethiol (0.9, 1.8, or 2.7 mmol) as the crosslinking agent, in chloroform, along with 5% DMPA as the photoinitiator. The reaction was irradiated with UV light ($\lambda = 365$ nm, 400 W). The crosslinking process was monitored by infrared spectroscopy.

5. In vitro degradation

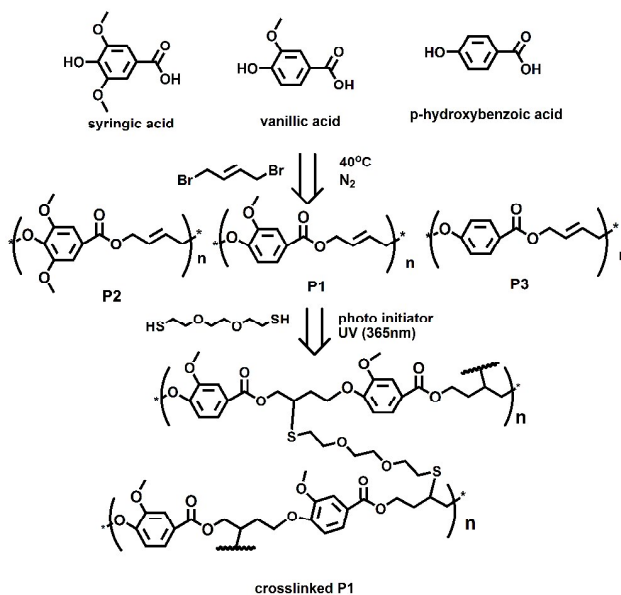
Vanillic acid-based unsaturated polyester and crosslinked vanillic acid-based polyester films, having thicknesses of 0.5 mm, were prepared by casting their chloroform solutions (C=0.1 g mL⁻¹). The films were cut into specimens, each weighing 25 mg, and dried under vacuum till their weights were constant. Rate of degradation after hydrolysis were measured by incubating the crosslinked polyester films in 20 mL of citric acid buffer solution at pH 2.5 at 80°C, with continuous shaking (60 strokes min⁻¹). After specified periods of time, the samples were rinsed thoroughly with distilled water and dried till their weights were constant, under vacuum

at 50° C, and reweighed to determine the mass losses. Meanwhile, the enzymatic degradations were carried out at 37°C in pH 7.4 PBS buffered solutions containing lipase from porcine pancreas (1 mg mL⁻¹). P1 and crosslinked P1 with 20% crosslinker were tested. The buffered enzyme solutions were replaced every 72 h to maintain the enzyme activity. A sample immersed in the phosphate solution, without lipase, was used as a control. GPC and NMR spectroscopy were used to analyze the degradation products.

Results and discussion

Synthesis of polyesters

The purpose of this work is to prepare a crosslinkable polyester chemically, and then to improve its mechanical properties by crosslinking. In the first step, vanillic acid, syringic acid, and p-hydroxybenzoic acid are used as the starting materials, each of which is reacted with trans-1,4-dibromo-2-butene, to prepare the unsaturated polyesters, P1, P2, and P3, respectively (Scheme 1). All the three raw materials are obtained from renewable resources. The GPC analyses indicate that the vanillic acid based polyester has a weight average molecular weight (M_w) of 73000 g mol⁻¹ and number average molecular weight (M_n) of 44600 g mol⁻¹, with a dispersity of 1.6. Syringic acid and p-hydroxybenzoic acid based polyesters have M_w of 12000 and 11000 g mol⁻¹, with dispersities of 1.51 and 1.47, respectively. The obtained unsaturated polyesters are characterized by ¹H-NMR, ¹³C-NMR, and FT-IR (Figures 1, 2, and 3). Figure 1 shows the ¹H-NMR spectra of the three unsaturated polyesters. All three polymers show peaks at 4.8 ppm, 4.7 ppm, and 6.0 ppm. The appearance of two singlets at 4.8 ppm and 4.7 ppm suggests the formation of the ester and ether-linkages. The triplet at 6.0-6.1 ppm is attributed to the protons on the unsaturated bond (-C=C-) in the copolymers. Figure 2 shows the ¹³C-NMR spectrum of P1. The linear polyester shows the expected main signals from the chemical structure of the repeating unit. The two methylene carbons, involved in the ether (-CH₂-O-Ar) and ester (-COOCH₂-) groupings are well separated. The signal corresponding to the unsaturated carbons (-C=C-) appears between 129.0-129.6 ppm, and the ester carbonyl (-COO-) appears at 165.9 ppm. Figure 3 shows the FT-IR spectra of the three unsaturated polyesters. The three unsaturated polyesters show similar FTIR absorption patterns. In the FTIR spectra, the stretching vibrations of the ester carbonyl, attached to the aromatic ring, appears at 1713 cm⁻¹. The absorption peaks at 1670 cm⁻¹ and 1600 cm⁻¹ are attributed to the unsaturated bond (C=C), and the aromatic rings in the polymer chains, respectively.



Scheme 1. Synthetic route of unsaturated polyesters and the crosslinked polyester networks

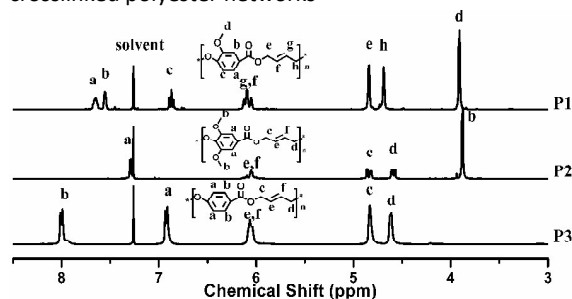


Figure 1. ¹H NMR spectra of unsaturated polyesters.

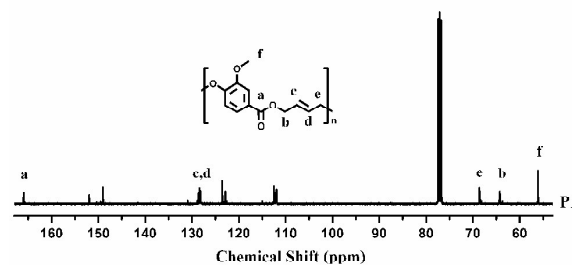


Figure 2. ¹³C NMR spectrum of P1.

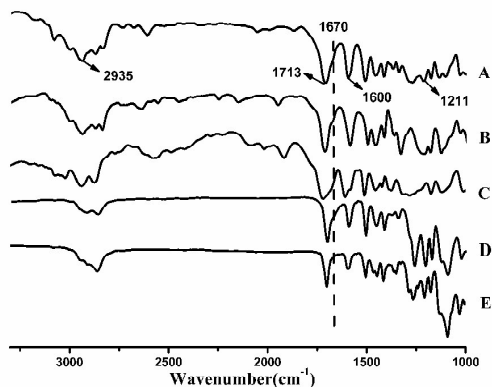


Figure 3. FTIR spectra of P1, P2, P3 and crosslinked P1 with different degree of crosslinking. A: P1; B:P2; C: P3; D: crosslinked P1 with 20% crosslinker; E: crosslinked P1 with 60% crosslinker.

It is worth noting that all the three monomers, derived from biomass, having unsymmetrical structures. These monomers have two functional groups, phenolic hydroxyl and carboxyl. Both these groups can react with *trans*-1,4-dibromo-2-butene under alkaline conditions, producing aregic polymers. For example, in the case of vanillic acid, two orientations of vanillic acid are possible along the polymeric chain, leading to two isoregic and two syndioregic dyad sequences, as illustrated in Figure 4. As it could be reasonably anticipated, four different peaks would appear in the ^{13}C -NMR spectrum, arising from the methylene adjacent to the unsaturated bond ($-\text{C}=\text{C}-$), which correspond to the four magnetically different $\text{CH}_2\text{-O}$ s (two from the iso dyads and two from the syndio dyads). Measurement of the peak areas shows that P1 is composed of 45.5% isoregic and 54.5% syndioregic dyads. Therefore, it is evident that essentially the vanillic acid units are randomly incorporated into the chain. The other two bio-based monomers show similar trends.

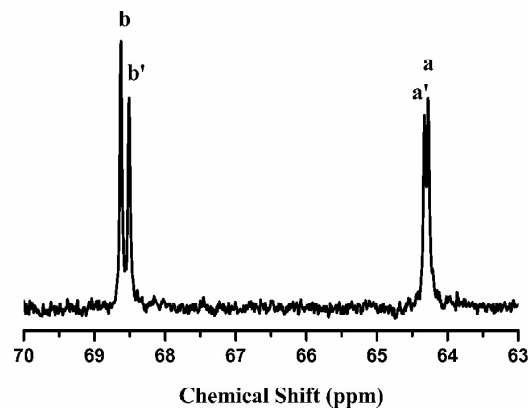
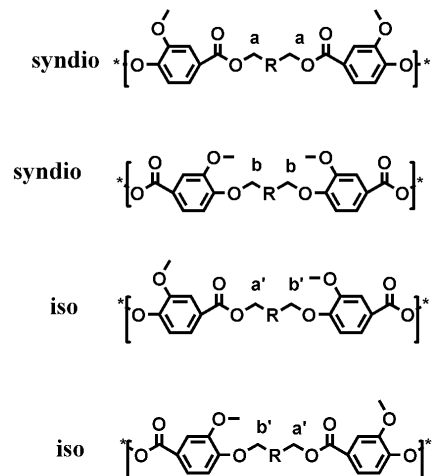


Figure 4. The O-CH₂ signals of P1 with indication of the dyads responsible for the four peaks.

Tensile properties

The mechanical properties of the three polyesters derived from renewable resources, vanillic acid, syringic acid and *p*-hydroxybenzoic acid were investigated. The recorded stress-strain curves for the obtained unsaturated polyesters are shown in Figure 5. It would not be expedient to directly compare the mechanical properties of these polymers. The unsaturation in the polyester allows the polymer to crosslink. The second step involves the preparation of the crosslinked polyesters. P1 was selected for the preparation of the crosslinked polyesters, through thiol-ene click chemistry, since it showed a better tensile property, compared to the other two unsaturated polyesters (Scheme 1). The reduction in the intensity of $\text{C}=\text{C}$ absorption peak at 1670 cm^{-1} in the FT-IR spectra of crosslinked polyesters suggests a chemical crosslinking reaction, as shown in Figure 3. The most obvious attribute of a crosslinked polyester material, reported here, is its tensile property. The stress-strain curves of the crosslinked polyesters with different degrees of crosslinking are shown in Figure 5 and the mechanical property parameters are presented in Table 1. Initially, with a concentration of crosslinker below 40%, the ultimate strength increases as the degree of crosslinking increases. However, as the concentration of crosslinker crosses 40%, the ultimate strength decreases with an increase in the degree of crosslinking. This can be explained on the basis of a change in the crosslinked structure. After crosslinking, the polyester forms a 3D network, which inhibits the activity of the polymeric chains and the deformation capacity of the materials. Therefore, the results show an increase of stress, and a decrease of strain. The strain at break for unsaturated polyester P1 is 495%, whereas the values are 242%, 290%, and 240% for the crosslinked polymers with 20%, 40%, and 60% crosslinkers, respectively. However, if the crosslink density is too high, the crosslinked polyesters generate more uneven networks. Under the influence of external force, the loose part in the network is destroyed first, which induces a decrease in the measured strength. Amongst

the samples, the crosslinked polyester with 40% crosslinker has the highest ultimate stress of 9.56 MPa. The Young's modulus for P1 is 9.03 MPa, whereas the values are 4.6, 8.57, and 13.6 MPa for crosslinked P1 with 20%, 40%, and 60% crosslinkers, respectively. After crosslinking, the polyester becomes elastomeric and the Young's modulus decreases, when the concentration of crosslinker is below 40%. When the concentration of crosslinker is 60%, the value of elongation at break is minimal; the stress improves to a certain degree and has the maximum Young's modulus. This result shows that the degree of crosslinking affects the deformation degree. It can be concluded that optimum crosslinking can improve the characteristic tensile properties.

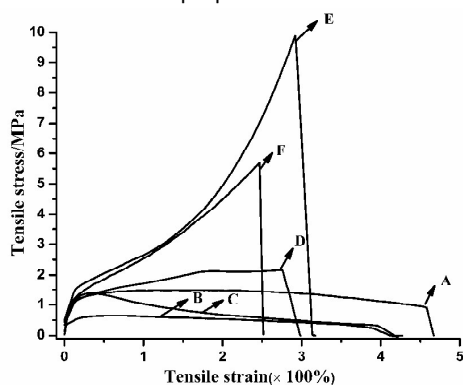


Figure 5. Stress-strain curves at 25 °C, 50 mm min⁻¹. A: P1; B: P2; C: P3; D: crosslinked P1 with 20% crosslinker; E: crosslinked P1 with 40% crosslinker; F: crosslinked P1 with 60%.

Table 1. Tensile properties of the polyesters

polymer	Young's modulus (MPa)	Ultimate strength (MPa)	Strain at break (%)
P1	9.03±0.93	1.4±0.06	495±37
crosslinked P1 with 20% crosslinker	4.6±0.48	2.13±0.04	242±33
crosslinked P1 with 40% crosslinker	8.57±1.05	9.56±0.3	291±12
crosslinked P1 with 60% crosslinker	13.6±0.9	5.56±0.11	240±21
P2	1.44±0.04	0.71±0.07	452±51
P3	3.36±0.12	1.41±0.06	461±27

Thermal properties

DSC analyses reveal the amorphous character of P1. The T_g for P1 is 35.8 °C, as shown in Figure 6. The thermal stabilities of P1 and its crosslinked materials are investigated using TGA. The data is presented in Table 2. Figure 7 shows that after crosslinking, the initial degradation temperature of the polymer decreases. The $T_{5\%}$ for P1 is 318 °C, and 180 °C and 194 °C for crosslinked P1 with 40 % and 60 % crosslinker, respectively. The crosslinked materials degrade sooner than the polyester precursors, because much weaker C-S bonds are introduced between the chains, as a result of the crosslinking reaction. This characteristic feature has been reported earlier. [33-34]

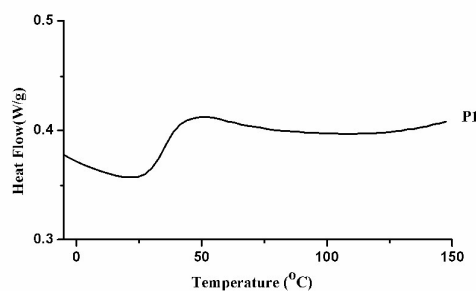


Figure 6. DSC curve of P1 at a scanning rate of 10 °C min⁻¹.

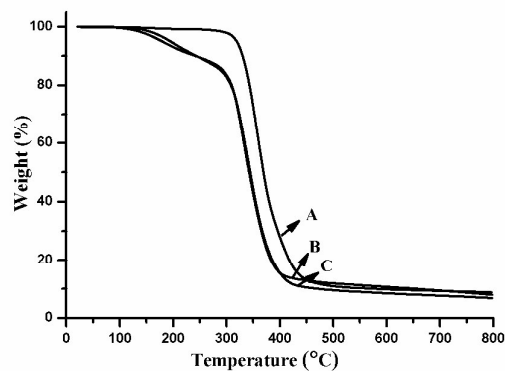


Figure 7. TGA curves of P1 with different levels of crosslinking recorded at a heating rate of 10 °C min⁻¹ between 25-800 °C under a nitrogen atmosphere.

A: P1; B: crosslinked P1 with 40% crosslinker; C: crosslinked P1 with 60% crosslinker.

Table 2. Thermal properties of P1 and crosslinked P1 with different crosslinker density obtained from TGA

polymer	$T_{5\%}^a$ (°C)	T_d^b (°C)	W^c (%)
P1	318	355	8.7
crosslinked P1 with	180	332	9.4

40% crosslinker			
crosslinked P1 with			
60% crosslinker	194	340	7.6

^a Temperature at which 5% weight loss was observed.

^b Temperature for maximum degradation rate.

^c Remaining weight at 800°C.

In vitro degradation

It is well known that normally aromatic polyesters, hydrolyze and biodegrade with difficulty.^[26] To investigate the biodegradability of bio-based polyester P1, it was incubated in pH 7.4 buffer at 37 °C, with and without porcine pancreas lipase. The changes in the weights of the samples and molecular weight of P1 at the end of the incubation time are shown in the table 3. The biodegradation was more noticeable when lipase was present in the medium. In buffered sodium phosphate solution, having pH 7.4 and lipase, P1 loses more than 50% of its initial weight after 100 days and the M_n and M_w also showed a decrease by more than 65%. In case of crosslinked P1 with 20% crosslinker, the samples were incubated at 37°C in buffered sodium phosphate solution with porcine pancreas lipase. The average time for losing 50% of its weight was about 120 days.

To test the hydrodegradability of these polyesters in a more aggressive medium, P1 and crosslinked P1 polyesters were incubated at 80°C and 20°C in citric acid buffer at pH 2.5. The sample P1 incubated at 80°C lost more than 50% only in 20 days, whereas the sample incubated at 20°C required 100 days to lose the same amount of weight. This implies that high temperature and acidic pH enhance the rate of degradation. In the case of crosslinked P1 with 60% crosslinker, when the samples were incubated at 80°C, the average time for weight loss of more than 50% was 85 days. However, in the case of crosslinked P1 with 20% crosslinker, the time required was 42 days. This implies that the extent of crosslinking can affect the rate of degradation of the crosslinked polymer. The degradation curves of unsaturated and crosslinked polyesters are shown in Figure 8.

NMR study was undertaken to have an insight of the degradation of the polyester chains at a molecular level. ¹H NMR spectra of P1 incubated in citric acid buffer at 80°C and 20°C and the one incubated in buffered sodium phosphate solution, with and without lipase, are all shown in Figure 9. Spectra of P1 in citric acid buffer at 80°C and in buffered sodium phosphate solution with lipase show appearance of a new peak at 1.6 ppm, when compared to the initial spectra. This shows that degradation of this polyester occurs mainly due to the splitting of the unsaturated ester groups in the polyester.

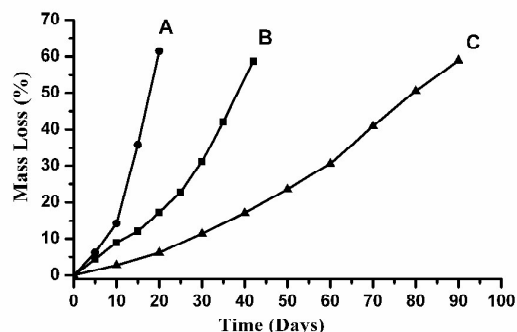


Figure 8. Degradation curves of P1 and crosslinked P1 under 80°C in citric acid buffer.

A: P1; B: crosslinked P1 with 20% crosslinker; C: crosslinked P1 with 60% crosslinker

Table 3. Degradation of P1

Buffer solution	Reaction condition	m_0 (mg)	m_e (mg)	M_w^b	M_n^b	Time (days)
pH 2.5	80°C	19.5	9.6	21566	13103	20
	20°C	25.7	12.6	22756	13831	100
pH 7.4	Include lipase (37°C)	26.4	12.5	23552	14106	100
	Without lipase (37°C)	24.7	23.5	23627	15934	100

m_0 : Initial weight of the samples; m_e : Remaining weight after degradation; M_w^b : M_w after degradation; M_n^b : M_n after degradation.

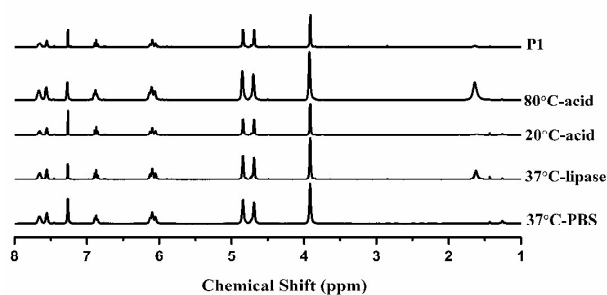


Figure 9. ¹H NMR spectra of P1 and degraded P1 at different conditions.

Conclusions

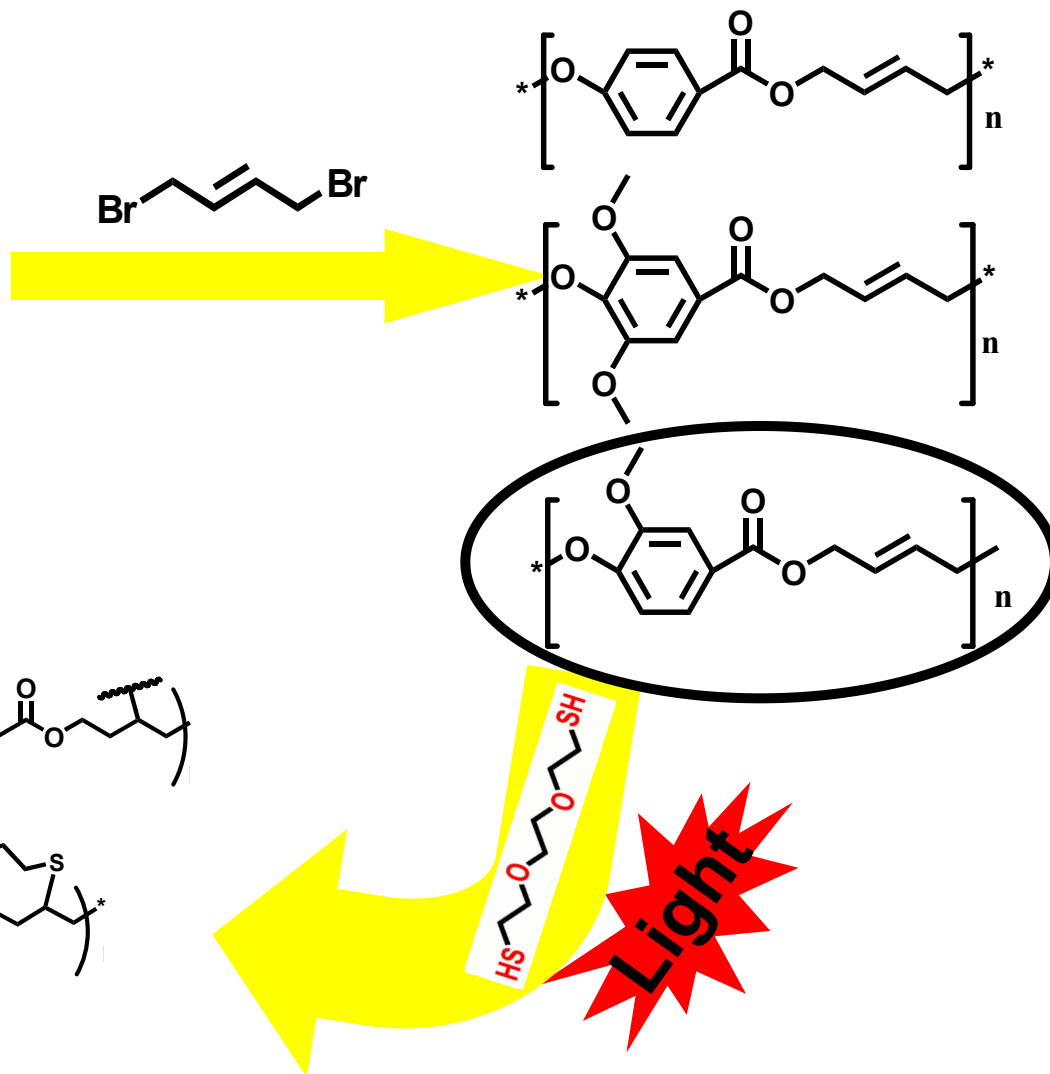
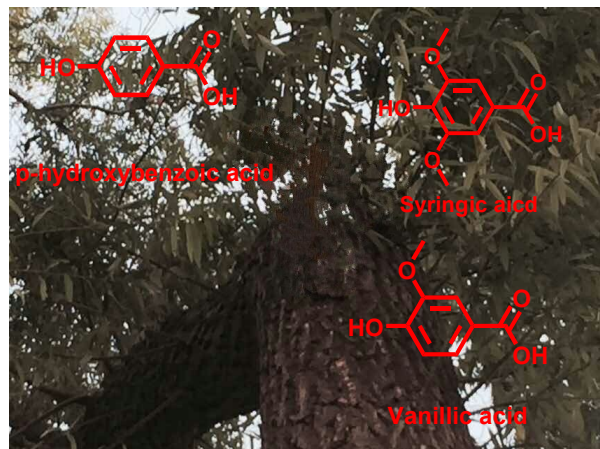
Three unsaturated polyesters were synthesized from vanillic acid, syringic acid, and p-hydroxybenzoic acid derived from lignin. The monomers had one, two and zero methoxy groups on the benzene ring, respectively. Comparison of the properties of the three unsaturated polyesters showed that the polyester synthesized from vanillic acid had the highest molecular weight and best tensile mechanical properties. A series of crosslinked elastomeric polyesters were prepared by the thiol-ene click reaction, using 2,2'-(ethylenedioxy) diethanethiol as the crosslinking agent. The experimental data and results showed that an appropriate degree of crosslinking could improve the tensile property effectively. With 40% crosslinker, the ultimate strength reached 9.56 MPa, and strain at break was 290%. The integrated mechanical properties were increased, compared to the unsaturated polyester and to that of the previous work. The crosslinked elastomeric polyesters showed steady thermal properties. They could be degraded in citric acid buffer at pH 2.5 and in buffered sodium phosphate solution, containing porcine pancreas lipase, at pH 7.4. Compared to other plastic materials widely used commercially, the polyesters synthesized from renewable materials have an enormous advantage of being biodegradable, which can decrease environmental pollution to a great extent.

Acknowledgements

This work was funded by NSFC (51203079), the Natural Science Foundation of Tianjin (14JCYBJC18100), and PCSIRT (IRT1257).

References

- Ashby, R. D.; Solaiman, D. K. Y.; Strahan, G. D.; Zhu, C.; Tappel, R. C.; Nomura, C. T., *Bioresource Technology* 2012, **118**, 272-280.
- Jiang, M.; Liu, Q.; Zhang, Q.; Ye, C.; Zhou, G., *Journal of Polymer Science Part A: Polymer Chemistry* 2012, **50** (5), 1026-1036.
- Tschan, M. J. L.; Brulé, E.; Haquette, P.; Thomas, C. M., *Polymer Chemistry* 2012, **3** (4), 836-851.
- Vilela, C.; Sousa, A. F.; Fonseca, A. C.; Serra, A. C.; Coelho, J. F. J.; Freire, C. S. R.; Silvestre, A. J. D., *Polymer Chemistry* 2014, **5** (9), 3119-3141.
- Supanchaiyamat, N.; Hunt, A. J.; Shuttleworth, P. S.; Ding, C.; Clark, J. H.; Matharu, A. S., *RSC Advances* 2014, **4** (44), 23304.
- Aouf, C.; Durand, E.; Lecomte, J.; Figueroa-Espinoza, M.-C.; Dubreucq, E.; Fulcrand, H.; Villeneuve, P., *Green Chemistry* 2014, **16** (4), 1740-1754.
- Lavilla, C.; de Iarduya, A. M.; Alla, A.; García-Martín, M. G.; Galbis, J. A.; Muñoz-Guerra, S., *Macromolecules* 2012, **45** (20), 8257-8266.
- Japu, C.; Alla, A.; Martínez de Iarduya, A.; García-Martín, M. G.; Benito, E.; Galbis, J. A.; Muñoz-Guerra, S., *Polymer Chemistry* 2012, **3** (8), 2092-2101.
- Hong, J.; Shah, B. K.; Petrović, Z. S., *European Journal of Lipid Science and Technology* 2013, **115** (1), 55-60.
- Maisonneuve, L.; Lebarbé, T.; Grau, E.; Cramail, H., *Polymer Chemistry* 2013, **4** (22), 5472-5517.
- Saito, T.; Brown, R. H.; Hunt, M. A.; Pickel, D. L.; Pickel, J. M.; Messman, J. M.; Baker, F. S.; Keller, M.; Naskar, A. K., *Green Chemistry* 2012, **14** (12), 3295-3303.
- Raquez, J. M.; Deléglise, M.; Lacrampe, M. F.; Krawczak, P., *Progress in Polymer Science* 2010, **35** (4), 487-509.
- Lebarbé, T.; Maisonneuve, L.; Nga Nguyen, T. H.; Gadenne, B.; Alfos, C.; Cramail, H., *Polymer Chemistry* 2012, **3** (10), 2842-2851.
- Lidia Jasinska.; Core E. Koning., *Journal of Polymer Science: Part A: Polymer Chemistry* 2010, **48** (13), 2885-2895
- Mialon, L.; Vanderhenst, R.; Pemba, A. G.; Miller, S. A., *Macromolecular rapid communications* 2011, **32** (17), 1386-1392.
- Stanzione Iii, J. F.; Sadler, J. M.; La Scala, J. J.; Reno, K. H.; Wool, R. P., *Green Chemistry* 2012, **14** (8), 2346-2352.
- Araújo, J. D. P.; Grande, C. A.; Rodrigues, A. E., *Chemical Engineering Research and Design* 2010, **88** (8), 1024-1032.
- Stanzione, J. F., 3rd; Sadler, J. M.; La Scala, J. J.; Wool, R. P., *Chemistry and Sustainable Chemistry* 2012, **5** (7), 1291-1297.
- Lligadas, G.; Ronda, J. C.; Galià, M.; Cádiz, V., *Journal of Polymer Science Part A: Polymer Chemistry* 2013, **51** (10), 2111-2124.
- Claudio, M.; van der Meulen, I.; Trey, S.; Jonsson, M.; Heise, A.; Johansson, M., *Journal of Polymer Science Part A: Polymer Chemistry* 2012, **50** (1), 16-24.
- Türünc, O.; Meier, M. A. R., *European Journal of Lipid Science and Technology* 2013, **115** (1), 41-54.
- Llevot, A.; Grau, E.; Carlotti, S.; Grelier, S.; Cramail, H., *Polymer Chemistry* 2015, **6** (44), 7693-7700.
- Jiang, Y.; van Ekenstein, G. O. R. A.; Woortman, A. J. J.; Loos, K., *Macromolecular Chemistry and Physics* 2014, **215** (22), 2185-2197.
- Wilbon, P. A.; Chu, F.; Tang, C., *Macromolecular rapid communications* 2013, **34** (1), 8-37.
- Desroches, M.; Caillol, S.; Auvergne, R.; Boutevin, B.; David, G., *Polymer Chemistry* 2012, **3** (2), 450-457.
- Lavilla, C.; Alla, A.; Martínez de Iarduya, A.; Benito, E.; García-Martín, M. G.; Galbis, J. A.; Muñoz-Guerra, S., *Journal of Polymer Science Part A: Polymer Chemistry* 2012, **50** (16), 3393-3406.
- Madbouly, S. A.; Schrader, J. A.; Srinivasan, G.; Liu, K.; McCabe, K. G.; Grewell, D.; Graves, W. R.; Kessler, M. R., *Green Chemistry* 2014, **16** (4), 1911-1920.
- Dong, W.; Ren, J.; Lin, L.; Shi, D.; Ni, Z.; Chen, M., *Polymer Degradation and Stability* 2012, **97** (4), 578-583.
- Pang, C.; Zhang, J.; Wu, G.; Wang, Y.; Gao, H.; Ma, J., *Polymer Chemistry* 2014, **5** (8), 2843.
- Pang, C.; Zhang, J.; Zhang, Q.; Wu, G.; Wang, Y.; Ma, J., *Polymer Chemistry* 2015, **6** (5), 797-804.
- W.L.F. Armarego.; Christina. L. L.Chai; Purification of Laboratory Chemicals *Butterworth-Heinemann Ltd. 7th Revised edition* 2013, 180-462.
- Parthiban, A.; Ming Choo, F.; Chai, C. L. L., *Polymer International* 2011, **60** (11), 1624-1628.
- R.K. Bharadwaj, A.R. Mehrabi, C. Hamilton, C. Trujillo, M. Murga, R. Fan, A. Chavira, A.K. Thompson *Polymer* 2002, **43**, 3699-3705.
- Kuo, P.-Y.; Sain, M.; Yan, N., *Green Chemistry* 2014, **16** (7), 3483-3493.



Crosslinked polyesters based on renewable resources have been successfully prepared, they have better tensile properties, and can be biodegradable.