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**Carbon nanotube decorated with nickel phosphide nanoparticles  
as efficient nanohybrid electrocatalyst for enhanced hydrogen  
evolution reaction**

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**Abstract**

Designing efficient, stable and inexpensive electrocatalysts to replace Pt-based catalysts for the hydrogen evolution reaction (HER) is highly desired in renewable energy research. In this study, we report the synthesis of nickel phosphide nanoparticles decorated on multiwall carbon nanotubes (Ni<sub>2</sub>P/CNT) by in situ thermal decomposition of nickel acetylacetonate as nickel source and trioctylphosphine as phosphorus source in oleylamine solution of CNT. As novel HER electrocatalyst, the Ni<sub>2</sub>P/CNT nanohybrid exhibits excellent electrocatalytic activity in 0.5 M H<sub>2</sub>SO<sub>4</sub> with low onset overpotential (88 mV), a small Tafel slope (53 mV·dec<sup>-1</sup>), a high exchange current density (0.0537 mA·cm<sup>-2</sup>) and good stability. It only needs overpotentials of 98 and 124 mV to attain current densities of 2 and 10 mA·cm<sup>-2</sup>, respectively. In addition, the Ni<sub>2</sub>P/CNT nanohybrid shows nearly 100% Faradaic efficiency in acid solutions. This work successfully demonstrates that the introduction of Ni<sub>2</sub>P NPs into CNT for enhanced electrocatalytic properties is feasible, and that this may open up a potential way for designing more efficient Ni<sub>2</sub>P-based catalysts for HER.

**Keywords:** Carbon nanotube; nickel phosphide nanoparticles; electrocatalyst; hydrogen evolution reaction.

## Introduction

With the increase of global energy demand and the aggravation of environmental problems, more and more attention is being paid to the production of hydrogen<sup>1</sup>. Hydrogen is a clean and efficient fuel and it has been regarded as the most promising candidate for replacing fossil fuels in the future<sup>2</sup>. Electrolysis of water is the most efficient, sustainable, and important technology to produce hydrogen<sup>3</sup>. To date, Pt-based catalysts exhibit excellent catalytic activity for the hydrogen evolution reaction (HER) with nearly zero overpotential<sup>4</sup>. However, the high price and scarcity of noble metal limits their application<sup>5</sup>. Therefore, it is important to look for low-cost electrocatalysts for large-scale industrial application.

Transition metal phosphides (TMPs), such as CoP<sup>6</sup>, MoP<sup>7</sup>, Ni<sub>2</sub>P<sup>8</sup>, Ni<sub>12</sub>P<sub>5</sub><sup>9</sup> and FeP<sup>10</sup>, have been used as effective electrocatalysts to replace the Pt-based noble metal catalyst for HER. In order to improve the electrical conductivity of the catalyst and increase the dispersion of the active phase, several kinds of carbon materials have been used as support, including carbon black, carbon cloth (CC), carbon nanotubes (CNT), carbon flakes (CF) and reduced graphene oxide (RGO). For example, Tian et al.<sup>11</sup> reported a high-activity, acid-stable 3D hydrogen-evolving cathode composed of CoP nanowire arrays on CC via low-temperature phosphidation of the Co(OH)F/CC precursor. Liu et al.<sup>12</sup> prepared a nanohybrid electrocatalyst that consisted of CNT decorated with CoP nanocrystals (NCs) by low-temperature phosphidation of a Co<sub>3</sub>O<sub>4</sub>/CNT precursor. Cui et al.<sup>13</sup> demonstrated the one-step facile preparation of MoP nanosheets supported on CF via a solid-state reaction. Jiang et al.<sup>14</sup> synthesized

phosphorous-rich FeP<sub>2</sub>/C nanohybrid via the pyrolysis of ferrocene and red phosphorus in an evacuated and sealed quartz tube at 500 °C. In contrast, reports are very rare on the design and electrocatalytic properties of novel electrocatalysts based on carbon materials decorated with nickel phosphide nanoparticles (NPs). CNT has high surface area and high electrical conductivity, and it has been extensively applied to electrode materials for enhanced electrocatalytic performance<sup>15</sup>. Previously we reported the electrocatalytic properties of nickel phosphide NCs with different phase for HER<sup>16</sup>, and we found that the Ni<sub>3</sub>P<sub>4</sub> NCs with solid structure exhibited higher catalytic activity than Ni<sub>12</sub>P<sub>5</sub> and Ni<sub>2</sub>P NCs. Herein, we describe our recent efforts in developing CNT decorated with Ni<sub>2</sub>P NPs to enhance the electrocatalytic activity via an in situ thermal decomposition approach using nickel acetylacetonate (Ni(acac)<sub>2</sub>) as nickel source and trioctylphosphine (TOP) as phosphorus source in oleylamine (OAm) solution of CNT for the first time. In our experiment, CNT was treated using concentrated nitric acid at 120 °C for 12 h because many literature reported that the acid-treated CNT was beneficial for enhanced the catalytic activity. As a novel HER electrocatalyst, the Ni<sub>2</sub>P/CNT nanohybrid exhibits excellent electrocatalytic activity in 0.5 M H<sub>2</sub>SO<sub>4</sub> with low onset overpotential (88 mV), a small Tafel slope (53 mV·dec<sup>-1</sup>), a high exchange current density (0.0537 mA·cm<sup>-2</sup>) and good stability. It only needs overpotentials of 98 and 124 mV to attain current densities of 2 and 10 mA·cm<sup>-2</sup>, respectively. In addition, the Ni<sub>2</sub>P/CNT nanohybrid shows nearly 100% Faradaic efficiency in acid solutions.

## Results and discussion

The Ni<sub>2</sub>P/CNT nanohybrid was synthesized by an one-pot in situ thermal decomposition reaction of Ni(acac)<sub>2</sub> and TOP in an OAm solution of the acid-treated CNT (Scheme 1) at 320 °C for 2 h. During this process, the Ni(acac)<sub>2</sub> was reduced to nickel clusters on CNT, then the P-C bonds of TOP broke and phosphorus atoms formed. The Ni<sub>2</sub>P/CNT nanohybrid was obtained by the reaction of nickel and phosphorus on CNT. The Ni<sub>2</sub>P/CNT nanohybrid was collected by centrifugation and subsequently washed with the mixture of hexane and ethanol several times. The final product was obtained by drying in vacuum at 60 °C for 24 h. For comparison, we also synthesized Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids by changing synthetic conditions (see the Supporting Information for experimental section). Fig. 1 shows the X-ray diffraction (XRD) patterns of CNT, Ni/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni<sub>2</sub>P/CNT. CNT (Fig.1a) shows two diffraction peaks at 26.1° and 42.6°, which can be attributed to the (002) and (101) planes of hexagonal graphite<sup>17</sup>. Ni/CNT (Fig. 1b) shows only one diffraction peak at 44.5°, which was indexed to the (111) plane of the face centered cubic (fcc) nickel (PDF#01-087-0712), indicating that the nickel nanoparticles (NPs) are highly dispersed on the CNT. In addition, the diffraction peaks of Ni<sub>12</sub>P<sub>5</sub>/CNT (Fig. 1c) at 32.6°, 35.8°, 38.4°, 40.7°, 41.6°, 44.6°, 47.1°, 49°, 54.2°, 56.1°, 68.6°, 74.5°, 79.8° and 88.9° are attributed to the (310), (301), (112), (202), (400), (330), (240), (312), (510), (501), (161), (004), (262) and (552) planes of the tetragonal structure of Ni<sub>12</sub>P<sub>5</sub> (PDF # 03-065-1623). The diffraction peaks of Ni<sub>2</sub>P/CNT (Fig. 1d) at 40.7°, 44.6°, 47.4°, 54.3°, 66.6°, 72.7°, 75.1°, 80.3° and 88.9° are attributed to the (111),

(201), (210), (300), (310), (311), (400), (401) and (321) planes of the hexagonal structure of  $\text{Ni}_2\text{P}$  (PDF # 03-065-3544). Moreover, the great suppression of the (002) plane of CNT was attributed to the loading of Ni,  $\text{Ni}_{12}\text{P}_5$  and  $\text{Ni}_2\text{P}$  NPs with high density<sup>12</sup>. These observations indicate that the CNT was successful decorated with Ni,  $\text{Ni}_{12}\text{P}_5$  and  $\text{Ni}_2\text{P}$  NPs.

The transmission electron microscopy (TEM) image of Ni/CNT (Fig. 2a) nanohybrid reveals that the nickel NPs exhibit monodisperse sphere-like morphology with an average particle size of  $6.1 \pm 1.3$  nm (Fig. S1a). The high-resolution TEM (HRTEM) image (Fig. 2b) reveals that the fringe spacings are about 2.05 Å and 1.77 Å, corresponding to the (111) and (200) lattice planes of fcc nickel. The TEM image of  $\text{Ni}_{12}\text{P}_5$ /CNT (Fig. 2c) nanohybrid reveals that the  $\text{Ni}_{12}\text{P}_5$  NPs exhibit hollow structure with an average particle size of  $19.4 \pm 2.4$  nm (Fig. S1b), and the HRTEM image (Fig. 2d) reveals that the fringe spacing is about 1.93 Å, corresponding to the (240) lattice plane of tetragonal  $\text{Ni}_{12}\text{P}_5$ . However, the highest intensity (312) lattice plane was not observed from HRTEM, which indicates that the (312) lattice plane was covered by the others planes. The hollow  $\text{Ni}_{12}\text{P}_5$  NPs form in solution from the reaction of Ni NPs with phosphorus, which is liberated from the decomposition of TOP, the obtained phosphorus atoms can diffuse into the nickel NPs via a nanoscale Kirkendall pathway. The outward diffusion rate of nickel is faster than the inward diffusion rate of phosphorus, which leads to the formation of hollow  $\text{Ni}_{12}\text{P}_5$  NPs<sup>16</sup>. The TEM image of  $\text{Ni}_2\text{P}$ /CNT (Fig. 2e) nanohybrid reveals that the  $\text{Ni}_2\text{P}$  NPs exhibit sphere-like morphology with an average particle size of  $6.1 \pm 1.3$  nm (Fig. S1c), and

the HRTEM image (Fig. 2f) reveals that the fringe spacing is about 2.21 Å, corresponding to the (111) lattice plane of hexagonal Ni<sub>2</sub>P. Furthermore, one can observe that the Ni, Ni<sub>12</sub>P<sub>5</sub> and Ni<sub>2</sub>P NPs are highly dispersed on the outer surface of CNT, as well as uniformly grown inside the CNT. The energy dispersive X-ray (EDX) spectra of the Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni<sub>2</sub>P/CNT nanohybrids indicate that the measured atomic Ni:P ratios are 2.33:1 (Fig. S2a) and 1.7:1 (Fig. S2b), respectively, which are very close to the stoichiometric ratios of 2.4:1 and 2:1 in Ni<sub>12</sub>P<sub>5</sub> and Ni<sub>2</sub>P. Fig. S3 shows the scanning TEM (STEM) image and the corresponding EDX mapping image of C, P and Ni. The results further confirms the good distribution of Ni<sub>2</sub>P NPs on the CNT. All these results strongly support the formation of Ni/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni<sub>2</sub>P/CNT nanohybrids by the in situ thermal decomposition approach.

N<sub>2</sub> adsorption-desorption (Fig. S4a) was carried out to further study the surface physical structures. The isotherm belongs to type IV, which is typical for mesoporous material. The Brunauer-Emmett-Teller (BET) specific surface area (SSA) of the as-synthesized Ni<sub>2</sub>P/CNT is 59.4 m<sup>2</sup>·g<sup>-1</sup>. However, the BET SSA of the Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrid is 51 m<sup>2</sup>·g<sup>-1</sup> (Fig. S4c), this value is smaller than that of Ni<sub>2</sub>P/CNT, which indicates that the Ni<sub>2</sub>P/CNT nanohybrid exposed more active sites. The Barrett-Joyner-Halenda (BJH) pore-size distribution of the Ni<sub>2</sub>P/CNT (Fig. S4b) and Ni<sub>12</sub>P<sub>5</sub>/CNT (Fig. S4d) show the narrow peak at 3 nm and 3.5 nm, and the average pore-size are 8.55 nm and 12.2 nm, respectively, confirming the nanoporous nature of the Ni<sub>2</sub>P/CNT and Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrids.

The chemical states of Ni, P and C in the as-synthesized Ni<sub>2</sub>P/CNT were



investigated by X-ray photoelectron spectroscopy (XPS), as shown in Fig. 3. The XPS survey spectrum (Fig. 3a) confirms the elemental composition of Ni<sub>2</sub>P/CNT with the peaks of Ni, P, C and O present. For the C 1s energy level, four peaks can be observed at 284.6, 285.4, 286.7 and 289.3 eV (Fig. 3b), which can be assigned to an sp<sup>2</sup> hybridized graphite-like carbon atom, sp<sup>3</sup> hybridized carbon atom, carbon atoms bound to one oxygen atom by a single bond and a double bond, respectively<sup>18</sup>. For the Ni 2p<sub>3/2</sub> energy level, three peaks can be observed at 853.5, 856.3 and 860.9 eV (Fig. 3c), which can be attributed to Ni in Ni<sub>2</sub>P, oxidized Ni species and the satellite of the Ni 2p<sub>3/2</sub> peak<sup>19</sup>. For the Ni 2p<sub>1/2</sub> energy level, three peaks are observed at 870.6, 874.5 and 880.2 eV (Fig. 3c), corresponding to Ni in Ni<sub>2</sub>P, oxidized Ni species and the satellite of the Ni 2p<sub>1/2</sub> peak, respectively. For the P 2p energy level, two peaks at 129.8 and 133.8 eV can be observed (Fig. 3d), which can be assigned to P in Ni<sub>2</sub>P and oxidized P species formed on the surface of Ni<sub>2</sub>P/CNT nanohybrid due to air contact<sup>20</sup>. The Ni 2p<sub>3/2</sub> peak at 853.5 eV is very close to that of Ni metal, which indicates that the Ni species in Ni<sub>2</sub>P has a very small positive charge (Ni<sup>δ+</sup>, 0<δ<2). The P 2p binding energy at 129.8 eV is less than that of elemental P (130.2 eV), which indicates that the P species in Ni<sub>2</sub>P has a very small negative charge (P<sup>δ-</sup>, 0<δ<1). Thus it can be concluded that there is an electron transfer from Ni to P<sup>21</sup>, and a small charge indicates a mainly covalent character with small ionic contribution. Furthermore, the Ni<sub>12</sub>P<sub>5</sub>/CNT has similar C 1s, Ni 2p and P 2p energy levels as Ni<sub>2</sub>P/CNT (Fig. S5), but the positive charge of Ni in Ni<sub>12</sub>P<sub>5</sub>/CNT is lower than that of Ni<sub>2</sub>P/CNT<sup>16</sup>.

The HER catalytic activity of Ni<sub>2</sub>P/CNT nanohybrid was evaluated by the

electrochemical experiment in 0.5 M H<sub>2</sub>SO<sub>4</sub> with a slow scan rate of 5 mV·s<sup>-1</sup> using a three-electrode setup with the working electrode prepared by the deposition of the nanohybrid on glassy carbon electrode (GCE). Fig. 4a shows the linear sweep voltammetry (LSV) curve of Ni<sub>2</sub>P/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub> with a scan rate of 5 mV·s<sup>-1</sup>. Bare GCE, CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT, Ni/CNT and commercial Pt/C were also studied for comparison. It can be seen that the bare GCE and CNT exhibit very small HER activity. Typically, the Pt/C catalyst exhibits the highest HER catalytic activity with nearly zero overpotential. It is surprising that the Ni<sub>2</sub>P/CNT exhibits high catalytic activity with low onset overpotential of 88 mV for HER. When the HER current density reached 2 and 10 mA·cm<sup>-2</sup>, the overpotentials were 98 and 124 mV, respectively. We also synthesized Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids as reference electrocatalysts. Ni<sub>12</sub>P<sub>5</sub>/CNT shows lower HER activity than that of Ni<sub>2</sub>P/CNT, with onset overpotential of 121 mV, but higher than that of Ni/CNT with onset overpotential of 157 mV. Generally, the loading of activity phase on the catalyst is very important in activity comparison. Therefore, the element contents of Ni and P in Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni<sub>2</sub>P/CNT nanohybrids were estimated by inductively coupled plasma-mass spectrometry (ICP-MS) analysis (Table 1). It can be seen that the element contents of Ni and P in Ni<sub>2</sub>P/CNT nanohybrid are less than that of Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrid, but the former exhibits higher catalytic activity. In other words, the percentage of CNT in Ni<sub>2</sub>P/CNT nanohybrid is higher than that of Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrid. The improvement of catalytic performance probably due to the high CNT content in Ni<sub>2</sub>P/CNT nanohybrid. Table S1 compares the HER

performance of the Ni<sub>2</sub>P/CNT nanohybrid with some reported non-noble-metal catalysts, such as CoP/CNT<sup>12</sup>, MoS<sub>2</sub>/RGO<sup>22</sup>, Mo<sub>2</sub>C/CNT<sup>23</sup>, MoP-CA<sup>24</sup>, MoS<sub>2</sub>/FTO<sup>25</sup>, MoS<sub>3</sub>/FTO<sup>26</sup>, CoSe<sub>2</sub> NP/CP<sup>27</sup>, MoN/C<sup>28</sup>, NiMoN<sub>x</sub>/C<sup>28</sup>, WS<sub>2</sub>/rGO<sup>29</sup>, CoP/Ti<sup>30</sup>, CoP/CC<sup>11</sup>, Ni<sub>2</sub>P/Ti<sup>31</sup>, NiP<sub>2</sub> NS/CC<sup>32</sup>, MoSe<sub>2</sub>/RGO<sup>33</sup>, Mo<sub>2</sub>C/XC<sup>23</sup>, MoP/CF<sup>34</sup>, Cu<sub>3</sub>P NW/CF<sup>35</sup>, FeP NA/Ti<sup>36</sup>, FeP NAs/CC<sup>37</sup>. From Table S1, one can observe that the overpotentials of the Ni<sub>2</sub>P/CNT nanohybrid compare favorably to the behavior of other non-noble-metal catalysts.

Tafel plots can be used to explain the reaction mechanism of HER. The Tafel plots (Fig. 4b) were recorded with the linear regions fitted to the Tafel equation ( $\eta = a + b \log j$ , where  $b$  is the Tafel slope and  $j$  is the current density). This yielded Tafel slopes of approximately 30, 53, 81 and 124 mV·dec<sup>-1</sup> for Pt/C, Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT respectively, which indicates that the HER rate of the Ni<sub>2</sub>P/CNT nanohybrid is faster than the rates of the Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids with the increase of potential, and the Pt/C catalyst exhibits the fastest HER rate. According to the mechanism for HER, hydrogen evolution in acidic aqueous proceeds in three steps<sup>38a</sup>. The first step is discharge reaction called Volmer reaction with a Tafel slope of 120 mV·dec<sup>-1</sup> [Equation (1)], the second step is the ion and atom reaction called Heyrovsky reaction with a slope of 40 mV·dec<sup>-1</sup> [Equation (2)], the third step is the atom combination reaction called Tafel reaction with a slope of 30 mV·dec<sup>-1</sup> [Equation (3)]. In our studies, the observed Tafel slopes of Ni<sub>2</sub>P/CNT and Ni<sub>12</sub>P<sub>5</sub>/CNT were 53 and 81 mV·dec<sup>-1</sup>, respectively, which indicates that the HER reaction took place via a fast Volmer step followed by a rate-determining Heyrovsky step<sup>38</sup>.

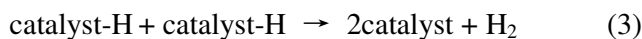
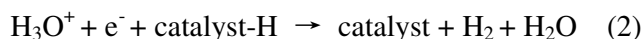
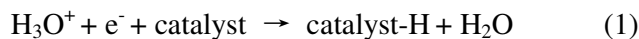


Table S2 shows the exchange current density values of the Pt/C, Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids, which were calculated by applying the extrapolation method to the Tafel plots (Fig. S6). Ni<sub>2</sub>P/CNT displayed an exchange current density of 0.0537 mA·cm<sup>-2</sup>, which further indicates that the catalytic activity of the Ni<sub>2</sub>P/CNT nanohybrid is better than that of the Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids.

In order to investigate the intrinsic catalytic activity of the Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids, the turnover frequency (TOF) for each active site was then calculated using the reported methods<sup>26</sup>. The number of active sites was determined by CVs (Fig. S7) at a scan rate of 20 mV·s<sup>-1</sup> over a range of -0.2 to 0.6 V vs. RHE in 1.0 M phosphate buffer solution (PBS, pH = 7). No apparent redox peak was observed, which indicates that this process only one electron transfer occurs, and the integrated charge over the entire potential range should be proportional to the total number of active sites. The calculated active sites number of Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT nanohybrids are 2.42 × 10<sup>-7</sup>, 8.72 × 10<sup>-8</sup> and 5.52 × 10<sup>-8</sup> mol, respectively. These values suggest that the Ni<sub>2</sub>P/CNT nanohybrid has higher catalytic active sites. Additionally, we quantified the active sites by electrochemistry method<sup>26</sup>. Fig. 4c shows the polarization curves (measured in 0.5 M H<sub>2</sub>SO<sub>4</sub>) normalized by the active

sites, which are expressed in terms of TOF. To achieve a TOF value of  $0.1 \text{ s}^{-1}$ , the  $\text{Ni}_{12}\text{P}_5/\text{CNT}$  and  $\text{Ni}/\text{CNT}$  nanohybrids need an overpotential of about 254 and 336 mV, respectively. However, the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid only need an overpotential of about 170 mV, much smaller than that required by the  $\text{Ni}_{12}\text{P}_5/\text{CNT}$  and  $\text{Ni}/\text{CNT}$  nanohybrids, which further indicates that the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid exhibits higher catalytic activity.

We further studied the stability of the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid using cyclic voltammetry (CVs) measurements for scanning 500 cycles with a scan rate of  $100 \text{ mV}\cdot\text{s}^{-1}$  in  $0.5 \text{ M H}_2\text{SO}_4$ , the result inserted in Fig. 4d. The  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid exhibited good stability in acidic solution with a slight activity loss after 500 cycles. We also found that the current density loss of  $\text{Ni}_2\text{P}/\text{CNT}$  was smaller than that of  $\text{Ni}_{12}\text{P}_5/\text{CNT}$  and  $\text{Ni}/\text{CNT}$  (Fig. S8), which further reveals the excellent catalytic performance of the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid. Furthermore, the time-dependent current density curve of the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid under static overpotential of 150 mV (Fig. 4d) suggests that the current density has a slight degradation after a long period of 40 000 s. This phenomenon can be explained as follows: 1) the consumption of  $\text{H}^+$  led to the decrease of the current density. 2) the HER was inhibited by the remaining hydrogen bubbles on the electrode surface. In addition, the typical serrate shape of the as-measured time-dependent curve was observed, which can be attributed to the accumulation and release of hydrogen bubbles<sup>39</sup>.

Electrochemical impedance spectroscopy (EIS) experiments were carried out to investigate the kinetics process of the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid in HER. Fig. 5a shows

the Nyquist plots of the Ni<sub>2</sub>P/CNT nanohybrid at various potentials. The diameters of the semicircles exhibit the potential-dependent behavior. With the increasing of potentials, the diameter of semicircles decreased gradually, which indicates that the HER kinetics process become faster at higher potential. Fig. 5b shows the Bode plots of the Ni<sub>2</sub>P/CNT nanohybrid. These spectra exhibit two-time-constant behavior, and the model equivalent circuit (Fig. S9) was used to fit the experimental data, as shown in Table S3. The value of the solution resistance ( $R_s$ ) was about 9  $\Omega$ , which is nearly potential-independent. However, the value of charge transfer resistance ( $R_{ct}$ ) is potential-dependent, and the lower  $R_{ct}$  reflects the superior electrocatalytic activity of the Ni<sub>2</sub>P/CNT nanohybrid. The  $R_{ct}$  values also can be used to fit the Tafel slope. In the case where the electron transport resistance at the catalyst/substrate interface or between catalysts is comparable to  $R_{ct}$ , the Tafel slope derived from the polarization curve might contain the contribution of electron transport resistance<sup>40</sup>. The Tafel slope of was obtained by fitting the  $\log(1/R_{ct})-\eta$  plot of the Ni<sub>2</sub>P/CNT nanohybrid, as shown in Fig. 5c. This yielded Tafel slope of about 53  $\text{mV}\cdot\text{dec}^{-1}$ , and this value is in accord with the previous Tafel slope obtained from the polarization curve.

Furthermore, we determined the Faradaic efficiency (FE) of the as-synthesized Ni<sub>2</sub>P/CNT and Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrids for hydrogen evolution according to previously reported method<sup>34</sup>. The gas generation was measured quantitatively using a calibrated pressure sensor to monitor the pressure change in the cathode compartment of a H-type electrolytic cell. Galvanostatic electrolysis was carried out with a current density of 20  $\text{mA cm}^{-2}$  in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution. The FE was calculated by comparing

the amount of measured hydrogen with calculated hydrogen (assuming 100% FE), as shown in Fig. 6. The agreement of the amount of experimentally quantified hydrogen with theoretically calculated hydrogen suggests Ni<sub>2</sub>P/CNT nanohybrid shows a FE close to 100% for hydrogen evolution. However, the Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrid only exhibits a FE of 83 % after 30 min, which further indicates that the Ni<sub>2</sub>P/CNT nanohybrid exhibits higher catalytic activity.

According to the above results, the excellent catalytic activity and stability can be attributed to the following factors: 1) the electronic effect of nickel and the ensemble effect of P. Previously we reported that the positive charge of Ni and the ensemble effect of P in pure Ni<sub>2</sub>P nanocrystals are stronger than that of pure Ni<sub>12</sub>P<sub>5</sub> nanocrystals<sup>16</sup>, and this reason also can be used to explain the present experiment result. Likewise, Ni/CNT exhibits lower activity than Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni<sub>2</sub>P/CNT may be due to the low electron valence state of nickel. All results also demonstrate that the HER activity can be improved by introducing phosphorus atom into nickel. 2) when the nickel phosphide NPs supported on CNT, the electrical conductivity of catalyst can be improved, the EIS data (Fig. 5d) shows that the Ni<sub>2</sub>P/CNT exhibits a smaller semicircle radius in the Nyquist plot than Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT, which indicates that Ni<sub>2</sub>P/CNT has the lowest charge transfer impedance. 3) the Ni<sub>2</sub>P NPs supported on the CNT have a small size and thus good dispersity, which means that more active sites are exposed. The catalytic mechanism of transition metal phosphides such as CoP<sup>41</sup>, MoP<sup>34</sup>, Cu<sub>3</sub>P<sup>35</sup> and FeP<sup>36</sup> for HER have been reported. Hydrogenase uses pendant bases proximate to the metal centers as active sites for hydrogen evolution<sup>42</sup>.

A metal complex HER catalyst also incorporates proton relays from pendant acid-base groups positioned close to the metal center where hydrogen evolution occurs<sup>43</sup>. Because Ni<sub>2</sub>P also has pendant base P in close proximity to the metal center Ni, the nickel centers ( $\delta^+$ ) and the basic phosphorous ( $\delta^-$ ) act as the hydride-acceptor and proton-acceptor centers to promote the HER<sup>44</sup>. Moreover, the phosphorous centers also could promote the formation of nickel-hydride for subsequent hydrogen evolution via electrochemical desorption<sup>45</sup>. Therefore, it is reasonable to conclude that Ni<sub>2</sub>P adopts a similar catalytic mechanism with hydrogenase and metal complex catalyst for the HER. In addition, the Ni<sub>2</sub>P/CNT exhibits superior HER activity than Ni<sub>12</sub>P<sub>5</sub>/CNT also may be due to the Ni<sub>2</sub>P/CNT are expected to offer more proton-acceptor centers<sup>32</sup>.

## Conclusions

In summary, carbon nanotubes decorated with nickel phosphide NPs have been successfully synthesized by in situ thermal decomposition using Ni(acac)<sub>2</sub> as the nickel source and TOP as the phosphorus source in OAm solution of CNT. The Ni<sub>2</sub>P/CNT nanohybrid exhibits excellent electrocatalytic activity with low onset overpotential (88 mV), a small Tafel slope (53 mV·dec<sup>-1</sup>), a high exchange current density (0.0537 mA·cm<sup>-2</sup>) and good stability. Furthermore, it only needs overpotentials of 98 and 124 mV to attain current densities of 2 and 10 mA·cm<sup>-2</sup>, respectively. Furthermore, the Ni<sub>2</sub>P/CNT nanohybrid shows nearly 100% Faradaic efficiency in acid solutions. This study shows that the synthesis of Ni<sub>2</sub>P supported on inexpensive and conductive CNT is feasible and has a great potential for the HER.



Similarly, this simple and efficient route also can be applied as a general method to design other electrocatalysts based on carbon materials decorated with transition-metal phosphides.

### Acknowledgements

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### Figure Captions

**Scheme 1.** Synthesis of Ni<sub>2</sub>P/CNT nanohybrid by the in-situ thermal decomposition.

**Fig. 1.** XRD patterns for (a) CNT, (b) Ni/CNT, (c) Ni<sub>12</sub>P<sub>5</sub>/CNT and (d) Ni<sub>2</sub>P/CNT.

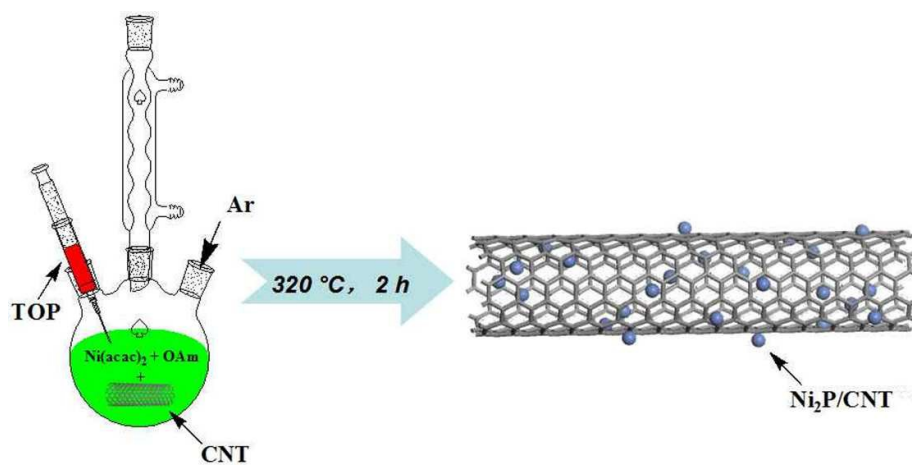
**Fig. 2.** TEM and HRTEM images of (a, b) Ni/CNT, (c, d) Ni<sub>12</sub>P<sub>5</sub>/CNT and (e, f) Ni<sub>2</sub>P/CNT.

**Fig. 3.** XPS spectra of (a) survey spectrum, (b) C 1s, (c) Ni 2p and (d) P 2p regions for Ni<sub>2</sub>P/CNT.

**Fig. 4.** (a) LSV curves of the Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT, Ni/CNT, Pt/C, CNT and bare GCE in 0.5 M H<sub>2</sub>SO<sub>4</sub> with a scan rate of 5 mV·s<sup>-1</sup>. (b) Tafel plots of the Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT, Ni/CNT and Pt/C. (c) Calculated TOFs for the Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub>. (d) Time-dependent current density curve of the Ni<sub>2</sub>P/CNT nanohybrid under static overpotential of 150 mV. Inserted is the Polarization curves of Ni<sub>2</sub>P/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub> initially and after 500 CV sweeps.

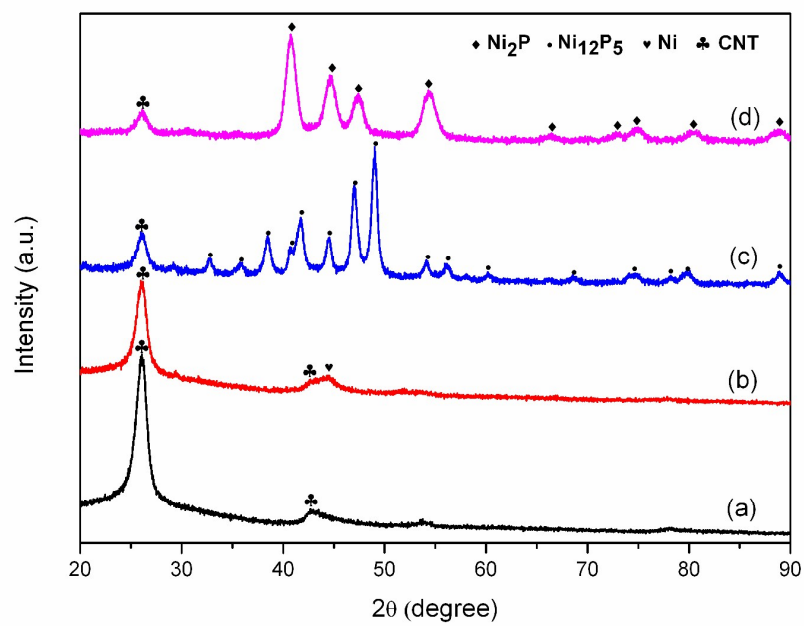
**Fig. 5.** (a) Nyquist plots and (b) Bode plots of the Ni<sub>2</sub>P/CNT nanohybrid 0.5 M H<sub>2</sub>SO<sub>4</sub>. (c) Tafel slope fitted from EIS data measured in 0.5 M H<sub>2</sub>SO<sub>4</sub>. (d) Nyquist plots of Ni<sub>2</sub>P/CNT, Ni<sub>12</sub>P<sub>5</sub>/CNT and Ni/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub> with an overpotential of 200 mV. Inserted is an expansion of the high frequency region.

**Fig. 6.** Fig. 6. The amount of H<sub>2</sub> theoretically calculated (red line) and experimentally measured versus time for Ni<sub>2</sub>P/CNT (black line) and Ni<sub>12</sub>P<sub>5</sub>/CNT (blue line) at 0.5 M H<sub>2</sub>SO<sub>4</sub> under a static current density of 20 mA cm<sup>-2</sup> for 30 min, respectively.

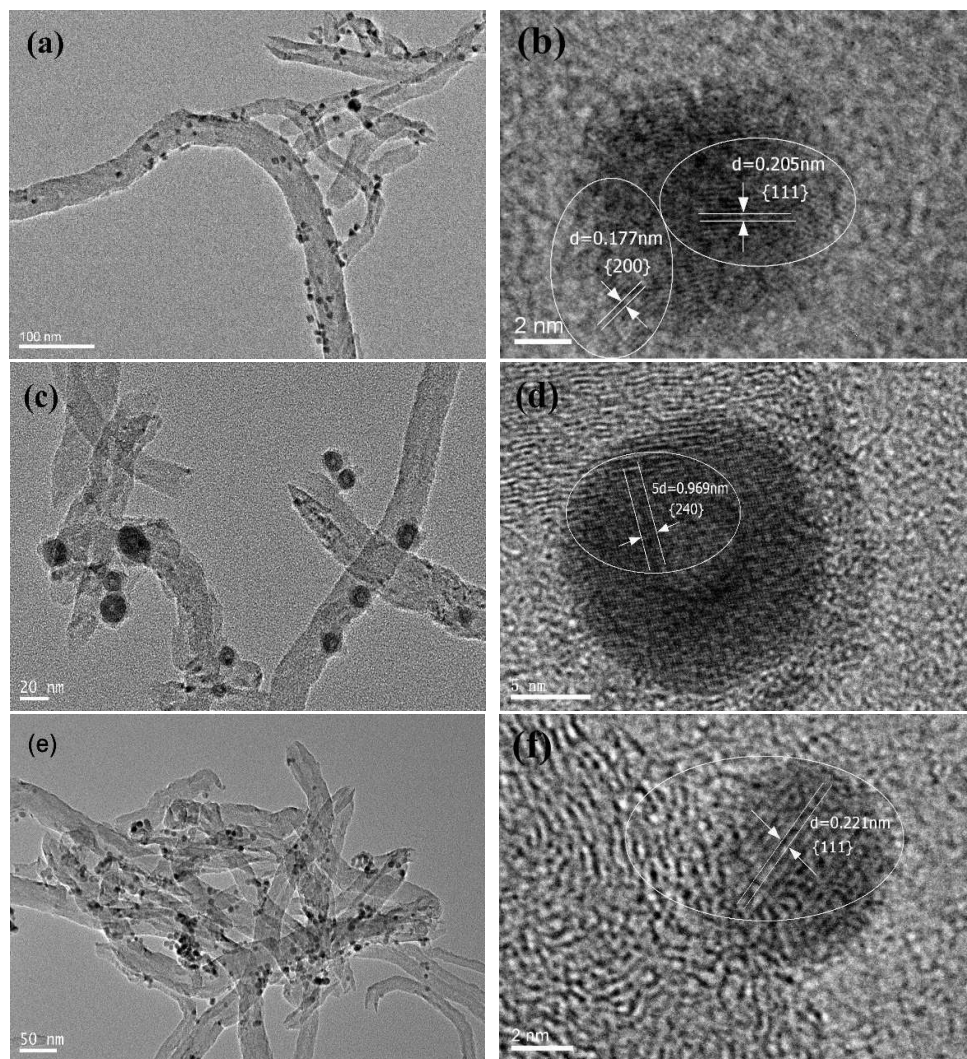


**Scheme 1.** Synthesis of Ni<sub>2</sub>P/CNT nanohybrid by the in-situ thermal decomposition.

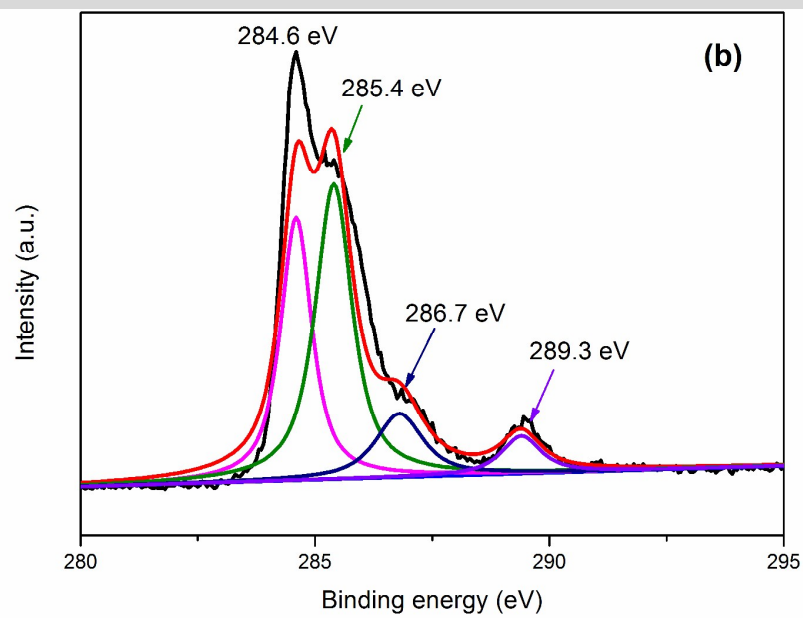
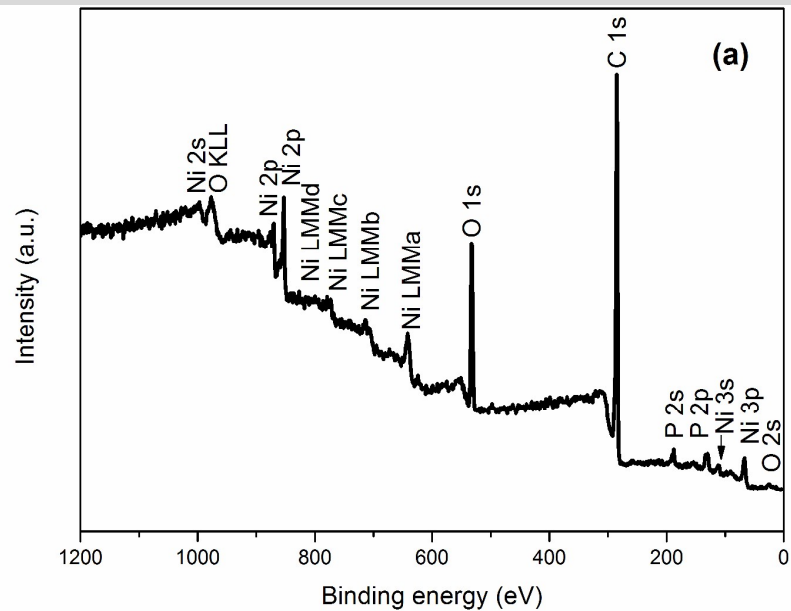


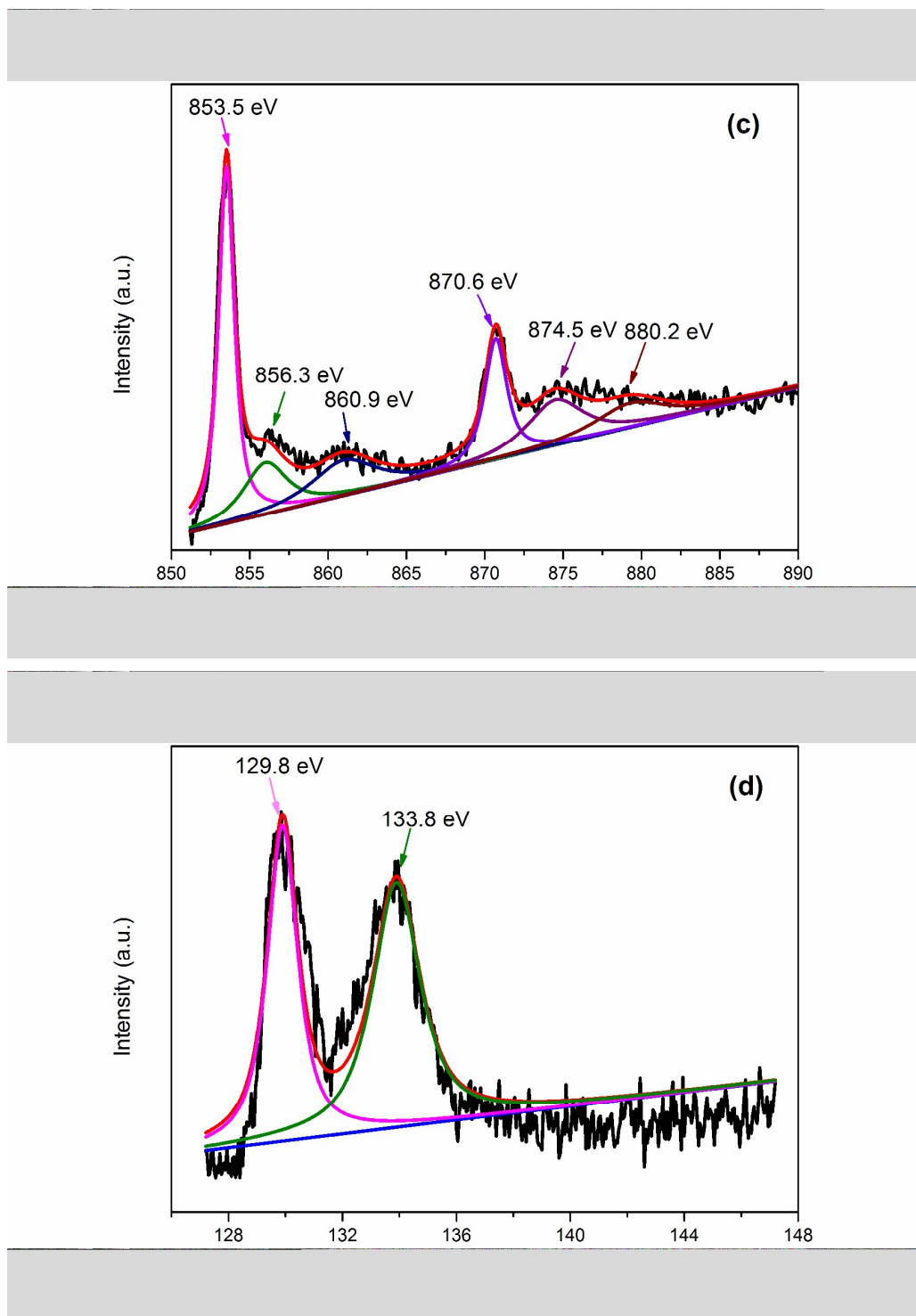


**Fig. 1.** XRD patterns for (a) CNT, (b) Ni/CNT, (c)  $\text{Ni}_{12}\text{P}_5/\text{CNT}$  and (d)  $\text{Ni}_2\text{P}/\text{CNT}$ .

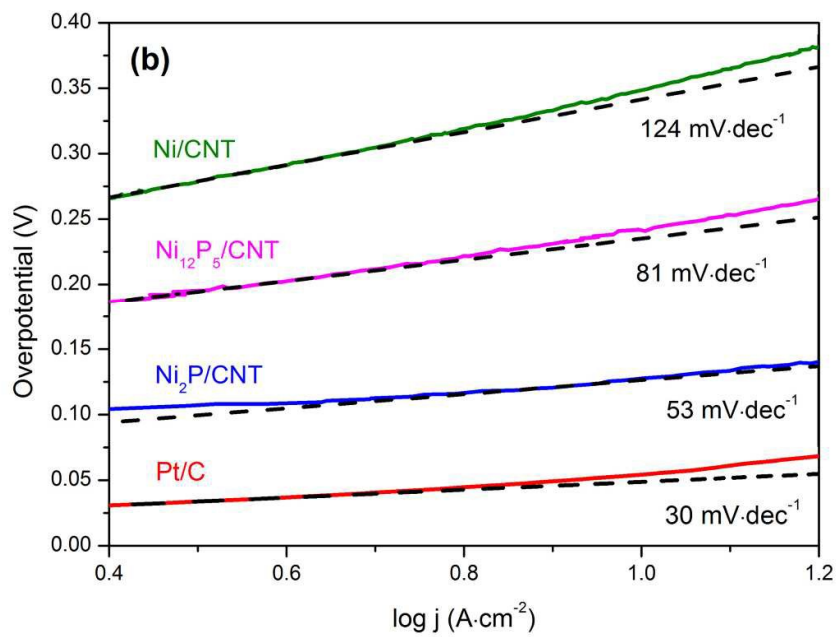
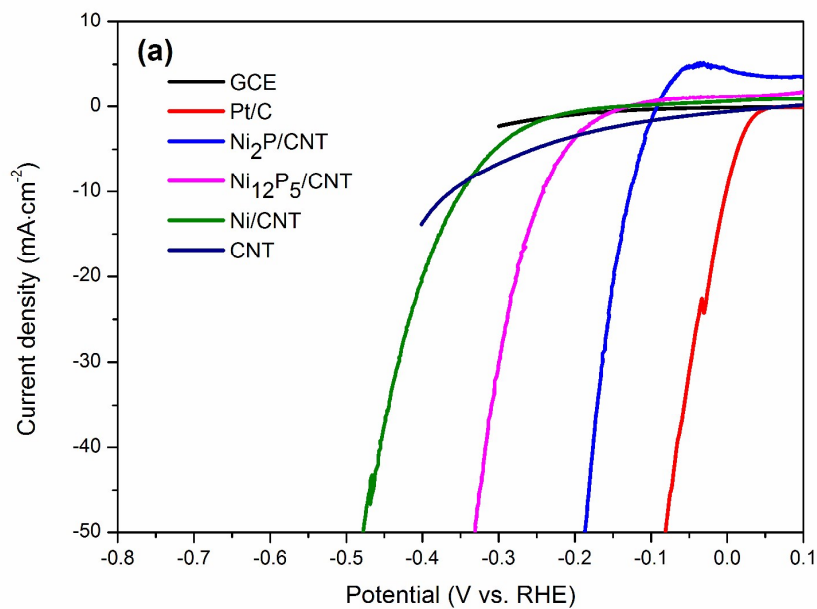


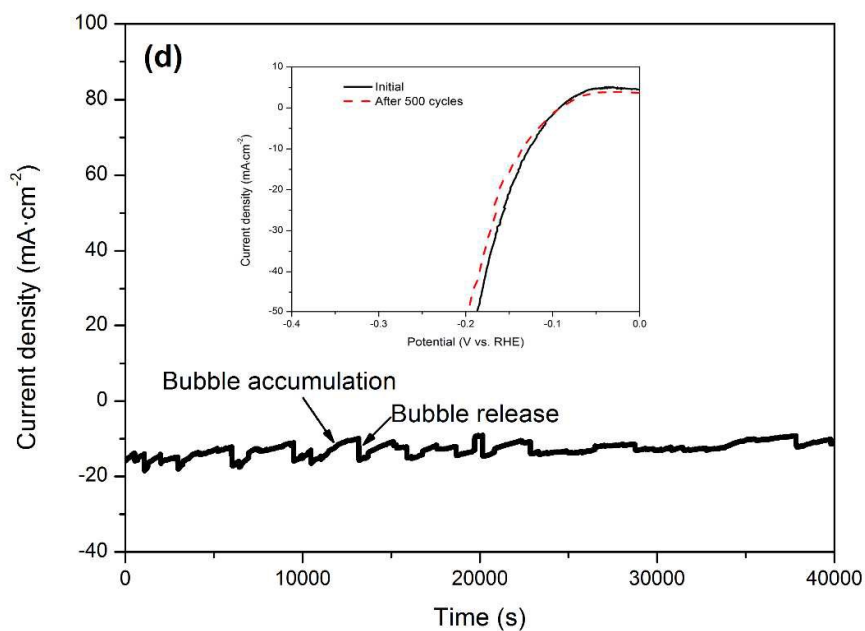
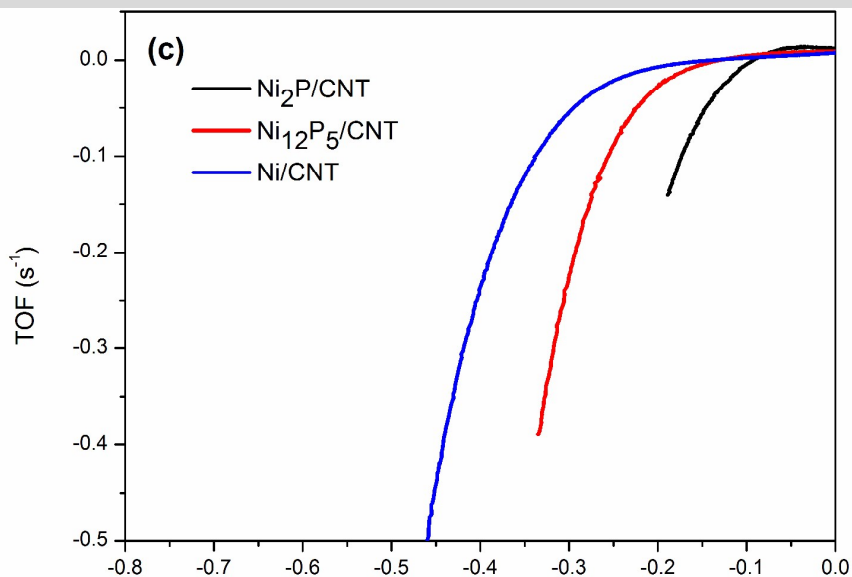
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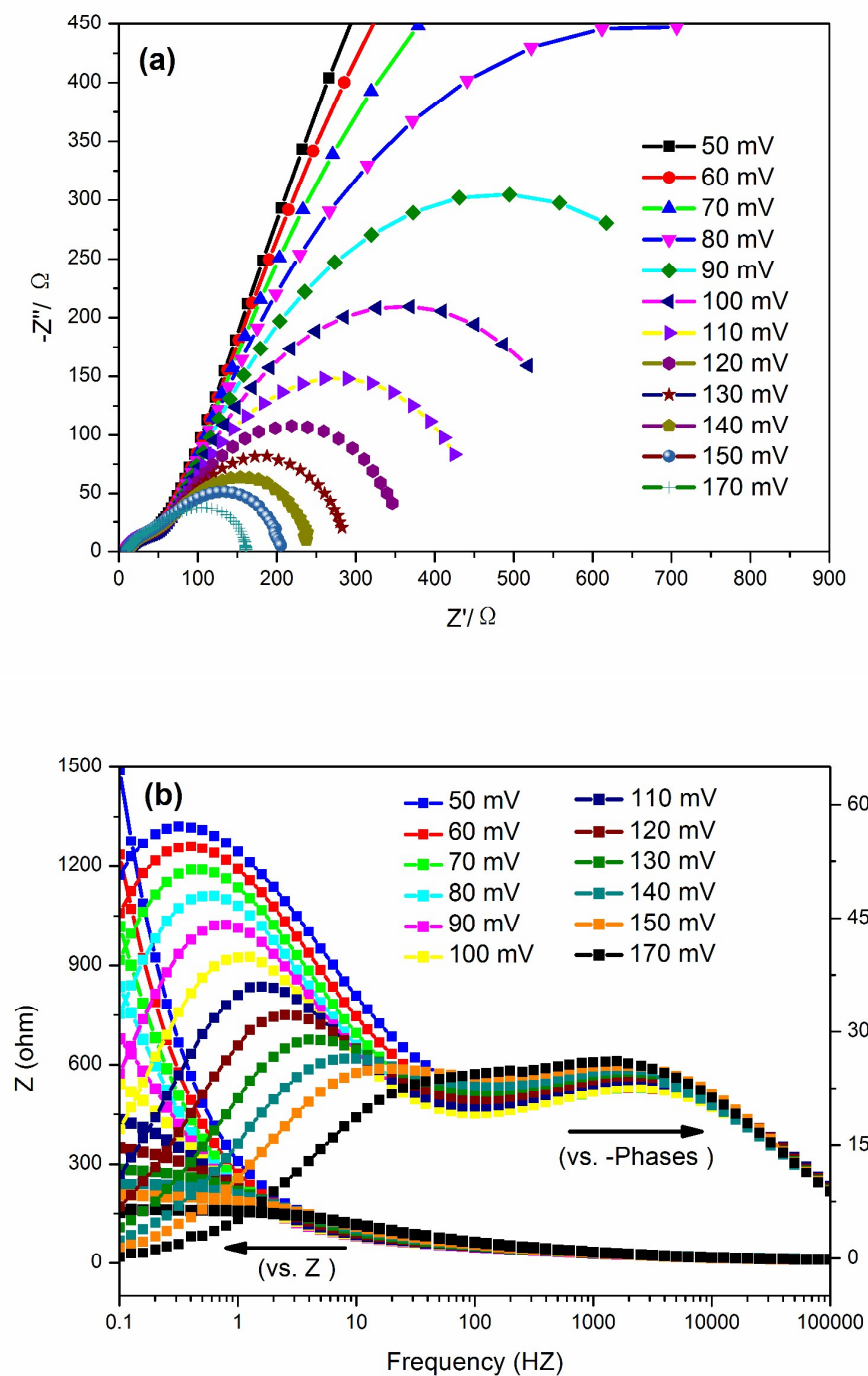
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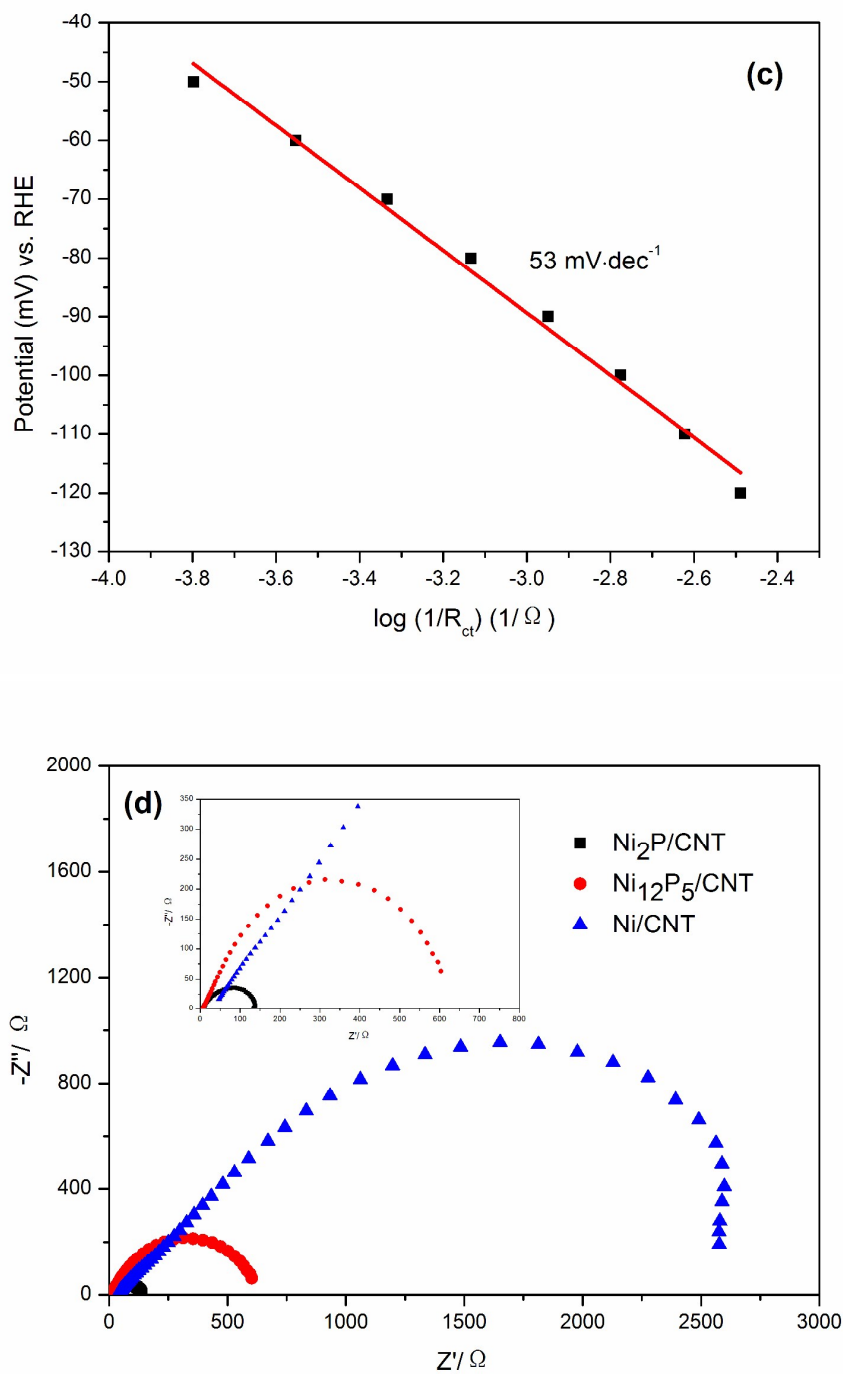




**Fig. 4.** (a) LSV curves of the  $\text{Ni}_2\text{P}/\text{CNT}$ ,  $\text{Ni}_{12}\text{P}_5/\text{CNT}$ ,  $\text{Ni}/\text{CNT}$ ,  $\text{Pt}/\text{C}$ ,  $\text{CNT}$  and bare GCE in 0.5 M  $\text{H}_2\text{SO}_4$  with a scan rate of  $5 \text{ mV}\cdot\text{s}^{-1}$ . (b) Tafel plots of the  $\text{Ni}_2\text{P}/\text{CNT}$ ,  $\text{Ni}_{12}\text{P}_5/\text{CNT}$ ,  $\text{Ni}/\text{CNT}$  and  $\text{Pt}/\text{C}$ . (c) Calculated TOFs for the  $\text{Ni}_2\text{P}/\text{CNT}$ ,  $\text{Ni}_{12}\text{P}_5/\text{CNT}$

and Ni/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub>. (d) Time-dependent current density curve of the Ni<sub>2</sub>P/CNT nanohybrid under static overpotential of 150 mV. Inserted is the Polarization curves of Ni<sub>2</sub>P/CNT in 0.5 M H<sub>2</sub>SO<sub>4</sub> initially and after 500 CV sweeps.





**Fig. 5.** (a) Nyquist plots and (b) Bode plots of the  $\text{Ni}_2\text{P}/\text{CNT}$  nanohybrid  $0.5 \text{ M H}_2\text{SO}_4$ .

(c) Tafel slope fitted from EIS data measured in  $0.5 \text{ M H}_2\text{SO}_4$ . (d) Nyquist plots of

$\text{Ni}_2\text{P}/\text{CNT}$ ,  $\text{Ni}_{12}\text{P}_5/\text{CNT}$  and  $\text{Ni}/\text{CNT}$  in  $0.5 \text{ M H}_2\text{SO}_4$  with an overpotential of  $200 \text{ mV}$ .



Inserted is an expansion of the high frequency region.

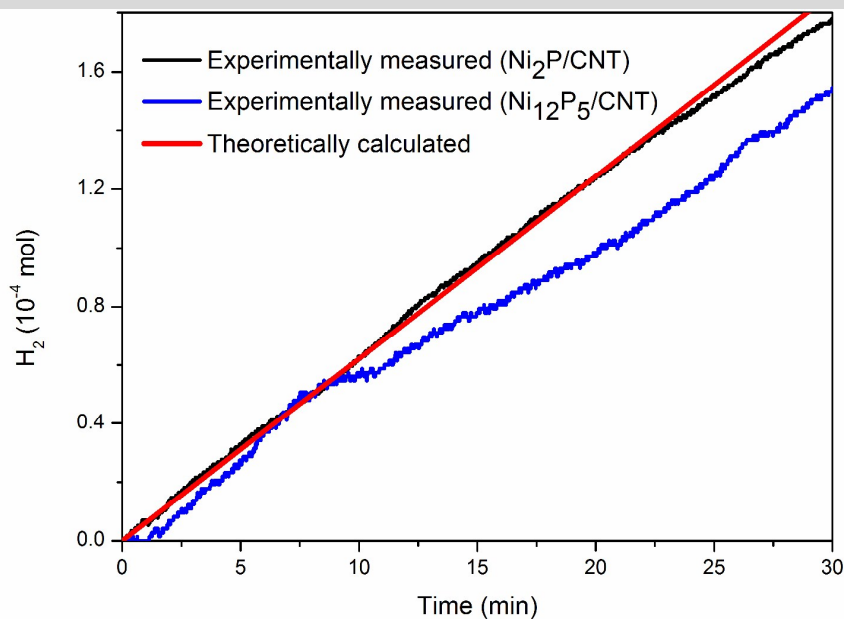
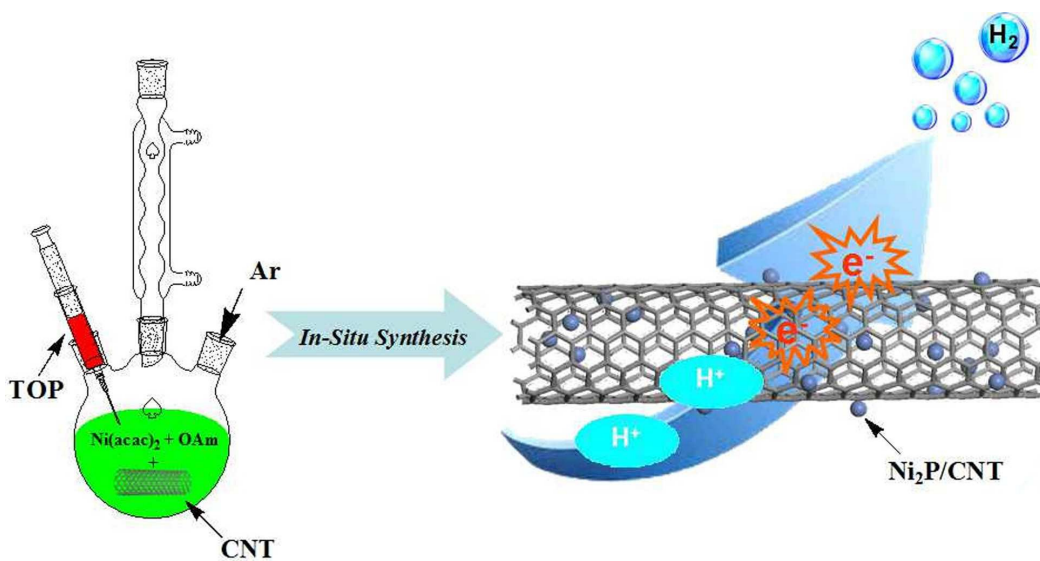


Fig. 6. The amount of H<sub>2</sub> theoretically calculated (red line) and experimentally measured versus time for Ni<sub>2</sub>P/CNT (black line) and Ni<sub>12</sub>P<sub>5</sub>/CNT (blue line) at 0.5 M H<sub>2</sub>SO<sub>4</sub> under a static current density of 20 mA cm<sup>-2</sup> for 30 min, respectively.

Table 1 ICP-MS analysis results of the as-synthesized Ni<sub>2</sub>P/CNT and Ni<sub>12</sub>P<sub>5</sub>/CNT nanohybrids.

Catalyst	Element content (Ni, ω %)	Element content (P, ω %)
Ni <sub>2</sub> P/CNT	49.07	14.76
Ni <sub>12</sub> P <sub>5</sub> /CNT	67.68	17.42

## Graphic Abstract



Nickel phosphide nanoparticles decorated on multiwall carbon nanotubes ( $\text{Ni}_2\text{P}/\text{CNT}$ ) was synthesized by in situ thermal decomposition for the first time. The  $\text{Ni}_2\text{P}/\text{CNT}$  nano hybrid exhibits high activity and stability for the hydrogen evolution reaction.