

RSC Advances



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This *Accepted Manuscript* will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the [Information for Authors](#).

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard [Terms & Conditions](#) and the [Ethical guidelines](#) still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.

1 **Precursor dissolution temperature as a size-controller in**
2 **Fe₃O₄ submicrospheres synthesis and their effect in catalytic**
3 **degradation of Rhodamine B**

4
5 Beatriz da Costa Carvalho^a, Fabiana Cristina Andrade Corbi^a, Fernando Aparecido
6 Sigoli^a and Italo Odone Mazali^{a,*}

7
8 ^a *Functional Materials Laboratory – Institute of Chemistry – University of Campinas -*
9 *UNICAMP, P.O. Box 6154, Zip Code 13083-970 - Campinas - SP- Brazil*

10
11 * Author to whom correspondence should be addressed.

12 E-mail: mazali@iqm.unicamp.br

13 Phone: +55 19 35213164 / Fax: +55 19 35213023

14
15
16
17
18

19 **Abstract**

20 Iron oxides submicrospheres have been synthesized by solvothermal method.
21 Particle sizes decrease from 700 to 100 nm as a function of dissolution temperature of
22 iron salt precursors, keeping constant the reaction temperature in the autoclave. The
23 submicrospheres particles are formed by aggregation of smaller nanoparticles,
24 nanograins. Fe₃O₄ submicrospheres show high saturation magnetization (M_s) and low
25 hysteresis (low remnant magnetization, M_r and coercivity, H_c), showing
26 superparamagnetic behavior. The size tailoring of the iron oxide particles allowed their
27 application as catalyst on photo-Fenton reaction of Rhodamine B degradation, in which
28 smaller particles showed high catalytic activity and the degradation efficiency showed
29 strong dependence on the nanograin size.

30

31

32

33

34

35

36

37 **Keywords:**

38 Iron oxide, solvothermal synthesis, dissolution of iron salt precursor, temperature effect,

39 Rhodamine

40

41 Introduction

42 The synthesis of magnetic iron oxide particles has been widely explored in the
43 last decades due to their potential application as catalysts, chemical sensors, theranostic
44 materials in biological systems, etc^{1,2,3,4}. These magnetic nanoparticles show interesting
45 properties such as high saturation magnetization and superparamagnetic behavior, i.e.,
46 the magnetic response of the particles is immediately canceled by removing the external
47 magnetic field, and thus, they can be easily used in catalysis and biomedical
48 application⁵. The magnetic properties may be adjusted in accordance with the
49 methodology of the particles preparation. Several chemical methods can be used to
50 prepare magnetic iron oxide particles such as co-precipitation⁶, microemulsions⁷,
51 sonochemical⁸, hydrothermal⁹ and solvothermal¹⁰ reactions and thermal
52 decomposition¹¹, which can provide nano/micro particles of magnetite (Fe₃O₄) with
53 controlled size and morphology. In general, such particles must provide high chemical
54 stability, uniform size distribution and facile dispersion in aqueous medium. In this
55 study the solvothermal method was chosen because it produces highly dispersed
56 particles and narrow size distribution. The reaction consists in dissolving a precursor of
57 Fe^{III} and a precipitating agent, such as sodium acetate in a particular solvent which must
58 also promote the reduction of ions Fe^{III} to Fe^{II} such as ethylene glycol, to obtain
59 (Fe^{II})(Fe^{III})₂O₄ with inverted spinel structure. In some cases surfactants are also used to
60 stabilize the particles in the process of magnetite formation at high pressure, such as
61 polyvinylpyrrolidone (PVP), polyethylene glycol (PEG).

62 It has been observed that the variation of different parameters in the
63 solvothermal synthesis may affect directly the morphology and size of nanoparticles.
64 Fan et al.¹² studied the Fe^{III} concentration effect on particle sized and observed that the
65 particle size increases as a function of the precursor concentration. Shen et al.¹³ used

66 diethylene glycol as solvent and reducing agent, and sodium citrate as growth inhibitor
67 to obtain nanoparticles from 2 to 13 nm highly stable in water ("water-soluble"). Zhu
68 and Diao¹⁴ investigated the influence of different parameters of the reaction such as the
69 concentration of precursor and precipitant agent, reaction time and autoclave
70 temperature on the morphology and size of the particles. They observed that an increase
71 in autoclave temperature leads to smaller particles size. When the concentrations of the
72 precipitant agent or Fe^{III} precursors were low, porous nanospheres with small size were
73 obtained. The authors observed that the autoclave reaction time directly affects the final
74 magnetic properties of the products. However, to the best of our knowledge, the effect
75 of dissolution temperature of iron salt precursor in the properties of iron oxides was not
76 investigated.

77 In our previous study, we showed that maghemite nanoparticles synthesized in a
78 Vycor® glass exhibit typical superparamagnetic behaviors¹⁵. In the present study, we
79 report the solvothermal synthesis of iron oxide particles using ethylene glycol as the
80 solvent and reducing agent, while sodium acetate and polyethylene glycol were used as
81 precipitating and stabilizing agents, respectively. The influence of dissolution
82 temperature of iron salt precursor in ethylene glycol was evaluated according of
83 magnetic behavior, crystal phase, morphology and particles size. Furthermore, the
84 degradation of Rhodamine B under visible light, using H₂O₂ as the oxidation reagent in
85 a photo-Fenton reaction was chosen as a model reaction to investigate the effect of
86 particle size on the catalytic activity of the Fe₃O₄

87

88 **Experimental section**

89 *Materials*

90 All chemicals were analytical grade and used as received. FeCl₃ and Rhodamine
91 B (RhB) were purchased from Sigma-Aldrich, CH₃COONa·3H₂O (NaAc), ethylene
92 glycol (EG) and polyethylene glycol 4000 (PEG) from Vetec, Hydrogen peroxide
93 (H₂O₂, 30%) from Synth.

94

95 *Synthesis of Fe₃O₄ particles with different particle size*

96 Fe₃O₄ particles were prepared via solvothermal synthesis¹⁶. Typically, 5.0 mmol
97 of FeCl₃ were dissolved in 40 mL of EG, followed by addition of 26.5 mmol of NaAc
98 and PEG (1.0 g). The mixture was stirred vigorously for 30 min at room temperature to
99 form a homogeneous solution. In order to evaluate the effect of dissolution temperature
100 of iron salt precursors (before autoclave step) on morphology and particle size, the
101 mixture solution was heated at different temperatures (25, 40, 60, 80, 100, 120 and 140
102 °C), and then transferred and sealed in a Teflon-lined stainless-steel autoclave (80 mL of
103 capacity). The autoclave was heated to 200 °C for 8 h to complete the reaction. The
104 products were collected and rinsed several times with deionized water and ethanol, then
105 dried under vacuum at 60 °C for 15 h.

106

107 *Catalytic Activity of Fe₃O₄ in heterogeneous photo-Fenton degradation of RhB*

108 The photocatalytic activities of the Fe₃O₄ particles were evaluated by photo-
109 Fenton degradation of RhB with H₂O₂ under UV-Vis irradiation (125 W Hg lamp with a
110 399 nm cut-off filter). For RhB degradation, 12 mg of Fe₃O₄ was suspended in 10 mL of
111 a 1.0×10⁻⁵ mol L⁻¹ RhB aqueous solution. Before irradiation, the RhB-catalyst
112 suspension was dispersed in the dark for 5 min. 1.24 mmol of H₂O₂ was added to the
113 solution at the beginning of the irradiation. The concentration of RhB at different

114 reaction times was determined by UV spectroscopy (Varian Cary-50
115 spectrophotometer).

116

117 *Characterization*

118 X-ray powder diffraction (XRD) was performed on Shimadzu XRD-7000
119 diffractometer, operating with CuK α radiation ($\lambda = 1.5406 \text{ \AA}$, 40 kV, 30 mA). Infrared
120 spectra (FTIR) were obtained on a Bomem MB100 Spectrometer using a KBr pellet
121 with resolution of 4 cm^{-1} . The morphology and particle size were determined by
122 Scanning Electron Microscopy (SEM) in a FEI Inspect-50 microscope and
123 Transmission Electron Microscopy (TEM) in a HR-TEM JEOL 3010 (300 kV)
124 equipment from the National Nanotechnology Laboratory (LNNano). The magnetic
125 behavior of the samples was investigated by the Vibrating Sample Magnetometer
126 technique (VSM) on a LakeShore-7400 magnetometer with a maximum magnetic field
127 equal to 2.0 Tesla. The measurements were performed at room temperature.

128

129 **Results and Discussion**

130

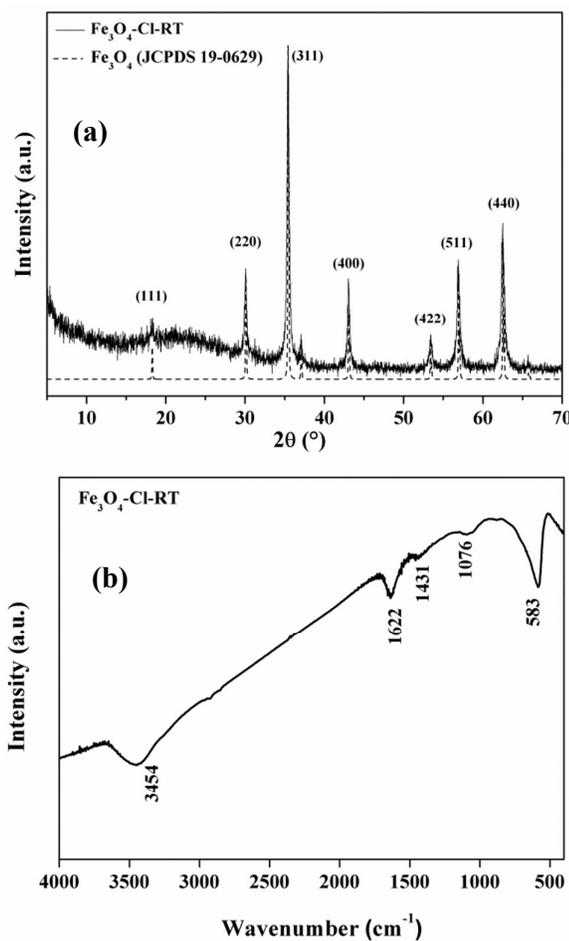
131 *Iron oxide particles using classical solvothermal synthesis: precursor dissolution at*
132 *room temperature*

133 The classical solvothermal synthesis was performed to be used as a reference.
134 The diffraction peaks were indexed (Figure 1a) and magnetite phase was identified
135 (Fe_3O_4 , JCPDS 19-0629). The crystallite size (D) was evaluated by Debye-Scherrer
136 equation¹⁷ ($D = k\lambda/\beta\cos\theta$), where k is a shape constant (equal to 0.9 assuming that the
137 crystallite is spherical), λ is the X-ray wavelength (1.5406 \AA), θ is the Bragg diffraction

138 angle (deg.), and β is the full-width at half-maximum (rad.) of the diffraction peak (311)
139 with higher intensity. To this sample, the crystallite size was calculated as 77 nm.

140 The FTIR spectrum (Figure 1b) shows a single band at 583 cm^{-1} assigned to
141 $\nu(\text{FeO})$ in magnetite. In addition, other two bands are observed with lower intensity
142 ascribed to $\delta(\text{CH}_2)$ at 1431 cm^{-1} and $\nu(\text{COC})$ at 1076 cm^{-1} . These bands indicate the
143 presence of a small amount of PEG in the sample.

144



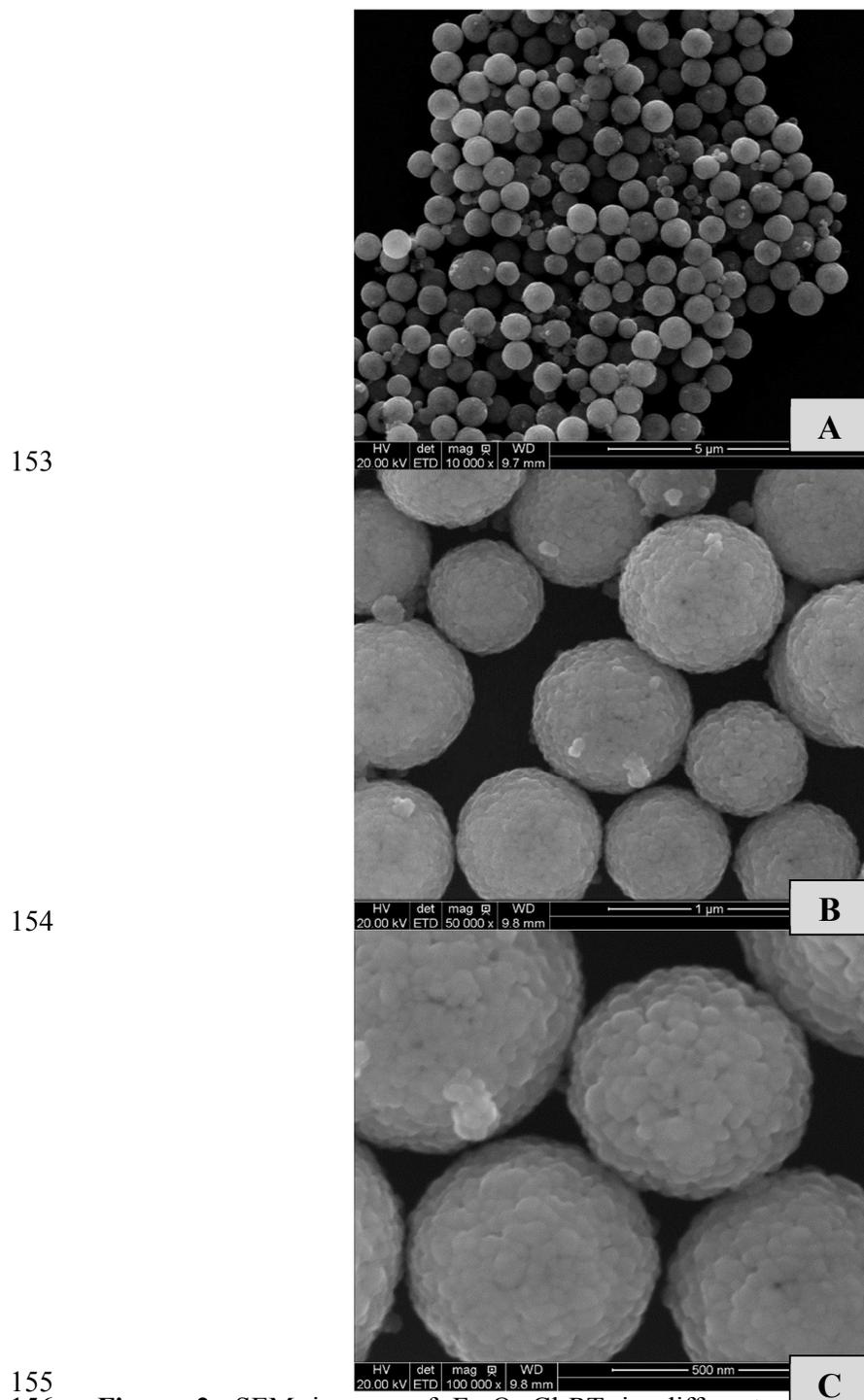
145

146 **Figure 1. (a) XRD and (b) FTIR of the $\text{Fe}_3\text{O}_4\text{-CI-RT}$ sample**

147

148 The morphology of the particles obtained with precursors dissolution at room
149 temperature was investigated by SEM, the images are shown in Figure 2. The particles
150 are spheroidal with particle size of ca. 700 nm. The particles surface is rough, as can be

151 clearly seen in the Figure 2c. The microsphere particles are formed by aggregation of
152 smaller nanograins (Figure 2c).



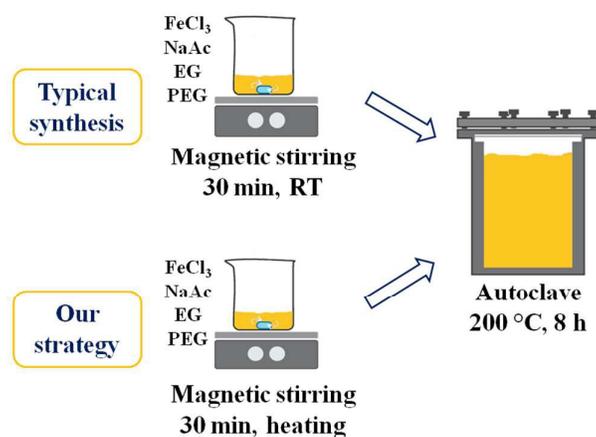
155 **Figure 2.** SEM images of Fe₃O₄-CI-RT in different magnifications revealing the
156 nanograins structures
157

158 Based on the results obtained from the magnetite particles here synthesized and
159 on the literature^{18,19,20,21}, two steps to obtain these particles are proposed. Firstly, in the
160 precursors dissolution FeCl₃, EG, NaAc and PEG (before autoclave step), the formation
161 of iron(III) acetate is observed (reddish precipitate). In the second step, which occurs in
162 the autoclave at high temperature and pressure, the partial reduction of Fe^{III} to Fe^{II} by
163 EG occurs, leading to the formation of the respective Fe^{II} and Fe^{III} hydroxides, which
164 lead to the magnetite phase. Therefore, in this paper a new strategy of particle size
165 control was used by changing the temperature of precursors dissolution before the
166 solvothermal reaction.

167

168 *Dissolution temperature of precursors before solvothermal process: size, morphology,*
169 *crystal structure and magnetic properties of Fe₃O₄*

170 Usually, the procedure of salt dissolution is carried out at room temperature, but
171 in this study the dissolution temperature was raised from 40 to 140 °C, keeping the
172 other synthesis conditions unchanged, as shown in Scheme 1.



173

174 **Scheme 1.** Experimental arrangement of the Fe₃O₄ synthesis, highlighting the step of
175 dissolution of iron salt precursors under heating. The autoclave temperature was kept at
176 200 °C for all experiments, and the change was carried out just in the step of FeCl₃,
177 NaAc, EG and PEG mixture before autoclave process.

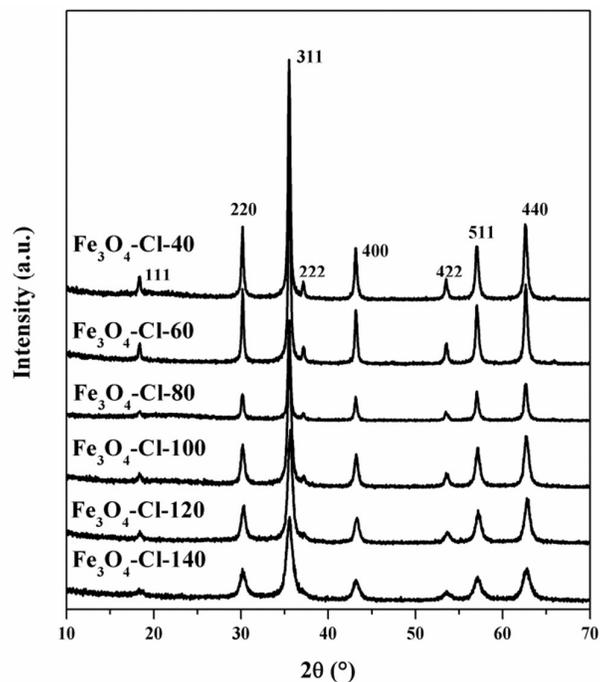
178 The XRD pattern of the samples shown in Figure 3 reveals that all diffraction
 179 peaks may be indexed to magnetite (Fe_3O_4 JCPDS 19-0629). It can be noticed that the
 180 peaks broaden as the dissolution temperature increases, indicating the formation of
 181 smaller crystals. The crystallite size was calculated by Debye-Scherrer equation and one
 182 may observe (Table 1) that their size decreases as the dissolution temperature increases
 183 SEM images show that the nanograins are equivalent to crystallite size, as can be seen
 184 in Figure 4. Particles with 700 nm formed by aggregation of 77 nm grains were
 185 obtained when the precursors dissolution was carried out at room temperature, while at
 186 140 °C the particle size is c.a.100 nm with nanograins of c.a. 13 nm. Therefore, as the
 187 dissolution temperature increases, the nanograin and particles sizes decrease. These
 188 results reveal that the dissolution temperature plays an important role on controlling the
 189 Fe_3O_4 particle sizes obtained by solvothermal process.

190

191 **Table 1.** Parameters of Fe_3O_4 microspheres prepared using different temperatures of
 192 dissolution of precursors.

Sample	Temperature dissolution of precursors / °C	Particle size / nm ^a	Crystallite size / nm ^b	Mr / emu g ⁻¹	Hc / Oe
$\text{Fe}_3\text{O}_4\text{-Cl-RT}$	25	700	77	7.2	51.8
$\text{Fe}_3\text{O}_4\text{-Cl-40}$	40	700	77	6.8	60.5
$\text{Fe}_3\text{O}_4\text{-Cl-60}$	60	600	78	6.3	67.5
$\text{Fe}_3\text{O}_4\text{-Cl-80}$	80	350	70	7.6	65.4
$\text{Fe}_3\text{O}_4\text{-Cl-100}$	100	150	28	5.5	33.2
$\text{Fe}_3\text{O}_4\text{-Cl-120}$	120	115	20	3.7	23.0
$\text{Fe}_3\text{O}_4\text{-Cl-140}$	140	100	13	1.9	9.5

193 ^aMeasured by SEM, ^bMeasured by XRD

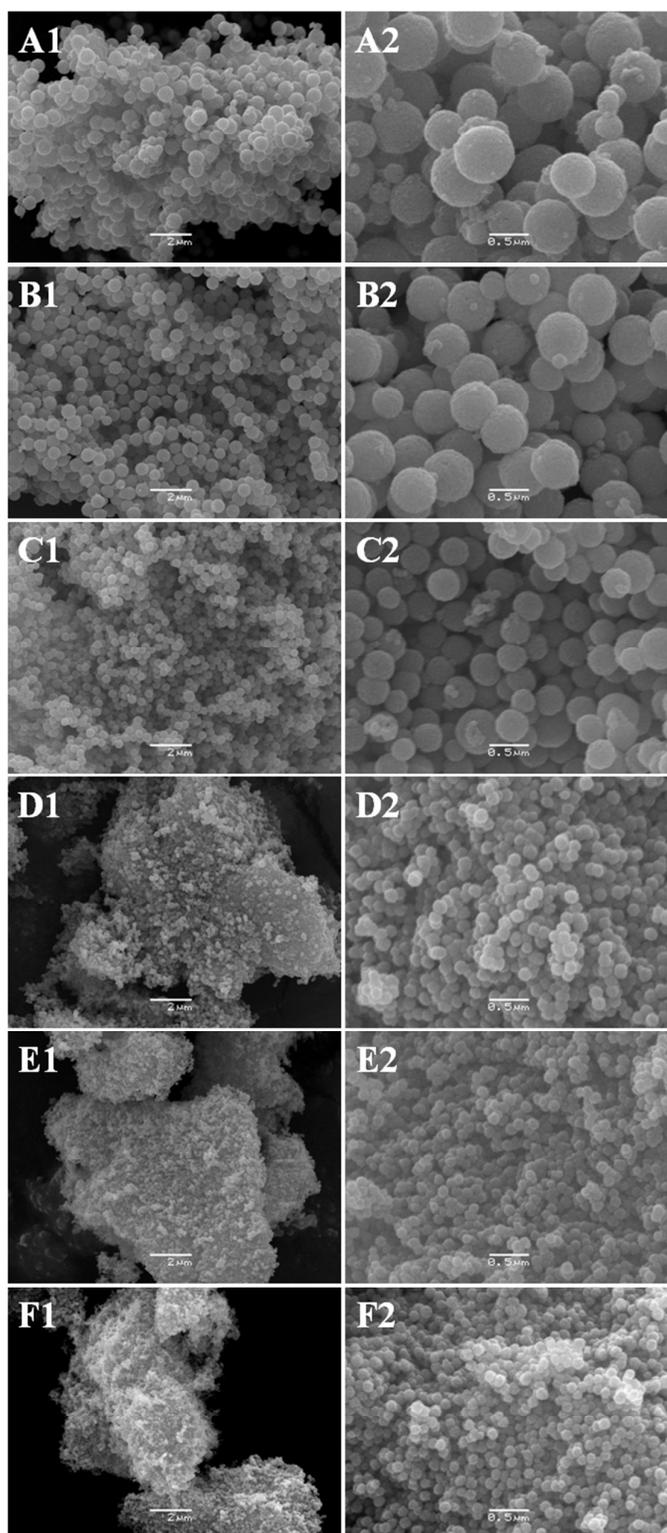


194

195 **Figure 3.** XRD pattern of Fe₃O₄-CI-T (T= 40-140 °C) samples

196

197 Some studies have showed that the initial concentration of precursors affects the
198 formation of nanocrystals and consequently the size of microspheres^{14,20}. In the present
199 study, it was noticed for the first time that the precursor dissolution temperature plays
200 direct influence on the particle and their crystallite sizes. It is known that the formation
201 of Fe₃O₄ includes nucleation of the oxide, followed by the growth of these nuclei to
202 form primary crystals (nanograins) which will organize to form larger
203 submicrospheres¹⁴. Some authors have shown that the particle size decreases as the
204 water content decreases due decrease the ratio ethylene glycol/water (EG/W)^{20,22}. It is
205 possible that evaporation of water during heating causes lower EG/W ratio explaining
206 the small size for particles whose synthesis was carried out in high temperature
207 dissolution of precursors. However, the procedure performed in reflux condenser does
208 not show change on particle size, such fact confirms the temperature influence has main
209 role in the particle size.



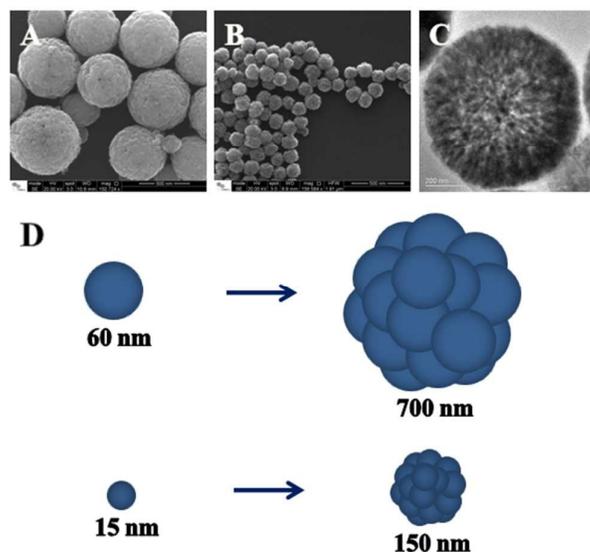
210

211 **Figure 4.** SEM images of the Fe_3O_4 synthesized in different precursor dissolution
212 temperature. **A:** 40 °C, **B:** 60 °C, **C:** 80 °C, **D:** 100 °C, **E:** 120 °C, **F:** 140 °C. On the left
213 column the bar is 2 μm. On the right column the bar is 0.5 μm for all samples.

214 The morphology and size of the particles were investigated by SEM. Figure 4
215 shows SEM images of the Fe_3O_4 microspheres synthesized at different precursor
216 dissolution temperatures. One may clearly see that the higher the dissolution
217 temperature the smaller the particle size. The particle size of the sample $\text{Fe}_3\text{O}_4\text{-Cl-40}$
218 and $\text{Fe}_3\text{O}_4\text{-Cl-60}$ (Figure 4c and 4b) do not show significant changes compared to the
219 sample obtained at room temperature $\text{Fe}_3\text{O}_4\text{-Cl-RT}$ (Figure 2). The $\text{Fe}_3\text{O}_4\text{-Cl-80}$ sample
220 (Figure 4c) has an average particle size of 350 nm, that is about the half of the magnetite
221 particle size obtained at room temperature (classic synthesis¹⁶). When the precursor
222 temperature dissolution is 100 °C, 120 °C and 140 °C (Figure 4d, 4e and 4f), the mean
223 sizes of particles are 150 nm, 115 and 100 nm, respectively, and they are about seven
224 times smaller than those obtained by classical synthesis.

225 TEM images also show that the particles are also formed by nanograins similar
226 to those obtained by classical synthesis. (see detail in Figure 5). The Fe_3O_4 particle is
227 made of aggregates of nanograins and therefore may be understood as a nanostructured
228 particle. Spheroidal particles of c.a. 700 nm are formed by nanograins of 60 nm, while
229 those with diameter of 150 nm are formed by nanograins of 15 nm, implying that the
230 precursor temperature dissolution has a direct influence on nanograin size and
231 consequently on the final particle size.

232



233

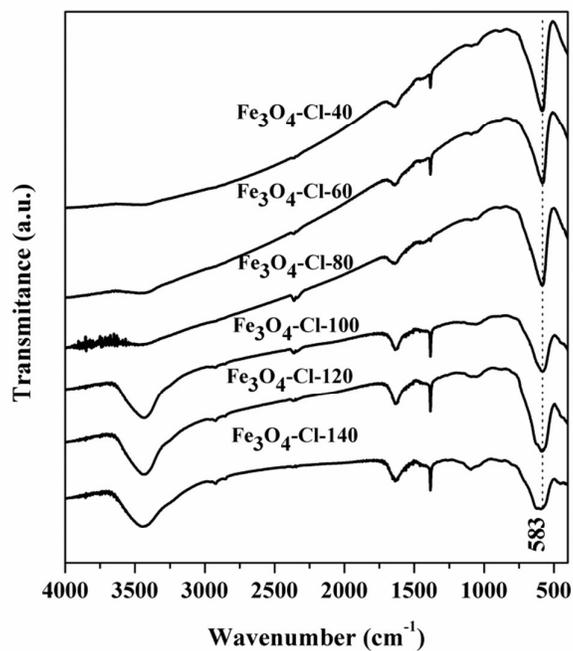
234 **Figure 5.** SEM high magnification of the Fe₃O₄ microspheres (A: Fe₃O₄-Cl-40, B:
235 Fe₃O₄-Cl-100), (C): TEM of the Fe₃O₄-Cl-40 and (D): scheme of the formation of
236 Fe₃O₄ nanostructure with different size formed from nanograins, demonstrating that
237 smaller grains grow in smaller particles

238

239 The FTIR spectra of all Fe₃O₄ samples show a band at 583 cm⁻¹ corresponding
240 to the Fe-O lattice mode of Fe₃O₄ (Figure 6). The other signals observed in the spectra
241 indicate the presence of a small amount of PEG or EG in the samples.

242 The magnetic properties of the microspheres were evaluated using a vibrating
243 sample magnetometer (VSM), measured at room temperature. The magnetic
244 measurements of Fe₃O₄-Cl-RT sample (magnetite obtained with reaction mixture at
245 room temperature) show magnetization saturation (M_s), remanent magnetization (M_r)
246 and coercivity (H_c) values of 85 emu g⁻¹, 7.2 emu g⁻¹ and 51.8 Oe, respectively. The
247 hysteresis loops shown in Figure 7 suggests the superparamagnetic behavior of the
248 sample. Furthermore, these spheres with homogeneous dispersion showed fast response
249 to the external magnetic field due to its superparamagnetic properties.

250

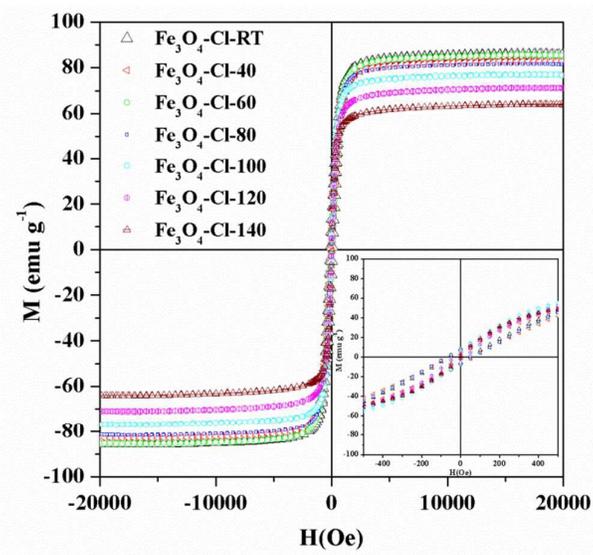


251

252 **Figure 6.** FTIR spectra of the Fe₃O₄ samples at different precursors dissolution

253 temperatures

254



255

256 **Figure 7.** Magnetic hysteresis loops of Fe₃O₄ at different temperature of precursors

257 dissolution

258

259 All samples obtained from the modified precursor dissolution temperature show
260 the same magnetization profile curve (Figure 7). The data showed in Table 1 indicate
261 that the M_s decrease while the particle size decreases

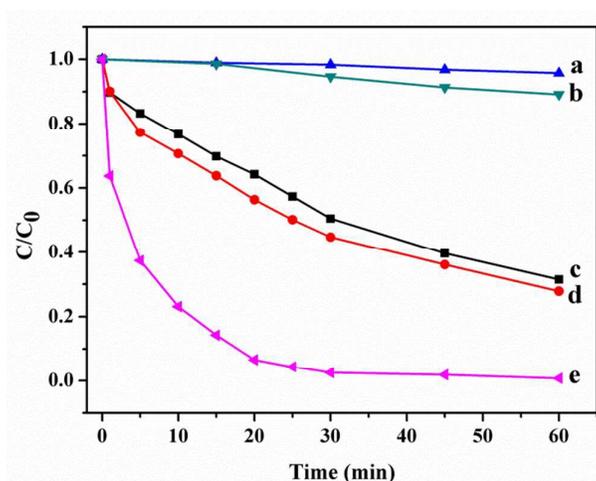
262 The decrease in M_s is due to higher contribution of surface effects, anisotropy
263 and non-stoichiometry of smaller diameters particles. Because of the spin disorder on
264 the particles surface, such particles tend to have lower M_s than the respective bulk. The
265 spin disorder is directly related to the magnetic behavior of the system^{23,24,25}. M_r and H_c
266 of the samples also decreased as the particle size decreases (Table 1), and it is known
267 that superparamagnetic particles do not exhibit M_r and H_c , being very interesting in
268 biomedical and catalytic applications due to null residual magnetization observed after
269 magnetic field removal. 700 nm magnetic particles show H_c 60.5 Oe and M_r 6.8 emu g⁻¹
270 values, being lower than reported by Fan et al¹² in nanostructured particles with 130 nm.
271 The smaller particles (100 nm) obtained in this work show the $H_c = 9.5$ and $M_r = 1.9$.
272 Thus, it can be seen that the values of H_c and H_r tend to zero as the particle sizes
273 decreases. The magnetic properties of the particles with size above critical diameter to
274 Fe_3O_4 superparamagnetic (ca. 60 nm)²⁶ can be explained due to nanograin size that
275 aggregate to form a submicrosphere (100 to 700 nm) thus the superparamagnetic
276 response, low coercivity and magnetization remanent can be assigned to the nanograins.

277

278 *Catalytic Activity of the Fe_3O_4 particles with different sizes*

279 The Fe_3O_4 particles with different sizes can be used in heterogeneous catalytic
280 degradation of organic compounds. The photo-Fenton reaction was used as model
281 reaction to study the catalytic properties of these magnetic particles. RhB was chosen as
282 organic dye to degradation reaction by H_2O_2 under UV-Vis radiation. The UV-Vis
283 degradation curves are presented on Figure S1 (Supporting Information). As shown in

284 Figure 8a, without Fe_3O_4 particles, the degradation is very low (less than 5%,
285 considered negligible), being attributed to the low oxidation potential of H_2O_2 as
286 compared to hydroxyl radicals (OH^\cdot)²⁷. When Fe_3O_4 -Cl-140 was irradiated under visible
287 radiation without H_2O_2 (Figure 8b), it was observed a little RhB degradation,
288 approximately 10%, due to adsorption of RhB in Fe_3O_4 catalyst. The effect of the
289 particles size in degradation efficiency was evaluated using 100, 350 and 700 nm Fe_3O_4
290 particles.
291



292
293 **Figure 8.** Degradation of RhB under different conditions: (a) RhB/H₂O₂/vis; (b)
294 RhB/Fe₃O₄-Cl-140/vis; (c) RhB/ Fe₃O₄-Cl-RT/ H₂O₂/vis; (d) RhB/ Fe₃O₄-Cl-80/
295 H₂O₂/vis; (e) RhB/ Fe₃O₄-Cl-140/ H₂O₂/vis

296
297 The RhB degradation of the 700 and 350 nm particles (Figure 10c and 10d,
298 respectively) is similar, around 70% in 60 min. Although these particles have different
299 sizes, there is almost no difference in their crystallite size (nanograin), as can be seen in
300 Table 1, being about 77 nm (Fe₃O₄-Cl-RT) and 70 nm (Fe₃O₄-Cl-80), and thus this
301 slight difference is not enough to promote higher catalytic activity for the 350 nm
302 microspheres. However, the degradation of RhB by H₂O₂ using 100 nm magnetite

303 particles ($\text{Fe}_3\text{O}_4\text{-Cl-140}$) under Vis irradiation reached 100% in 60 min (Figure 8e),
304 showing higher efficiency for RhB degradation. The smaller crystallite size (13 nm)
305 contributes to enhance the catalytic effect in these particles for RhB degradation
306 probably because the higher surface/volume ratio. Furthermore, the catalysts can be
307 conveniently separated by applying an external magnetic field.

308

309 **Conclusions**

310 Fe_3O_4 nanostructured spheroidal particles have been obtained by modified
311 solvothermal method. The synthesis leads to nanostructured spheres formed by
312 nanograins that aggregate to form submicroparticles. The dissolution of iron precursor
313 at different temperatures before autoclave process provided a control of the particle size
314 without changing the morphology. As the precursor dissolution temperature increases,
315 the particle size decreases, being possible to obtain particles from 100 to 700 nm formed
316 by different nanograins sizes. The particles showed superparamagnetic properties, high
317 magnetization saturation and small M_r and H_c . The precursor dissolution temperature is
318 an important synthesis parameter that can be changed to modulate the size of particles
319 depending on the need of the application. Fe_3O_4 particles showed to be highly efficient
320 in RhB degradation by photo-Fenton reaction, demonstrating the influence of crystallite
321 size on catalytic activity of the particles, with best results for 100 nm particles and 13
322 nm crystallite.

323

324 **Acknowledgements**

325 The authors would like to thank the FAPESP, CAPES and CNPq agencies for
326 financial supports. Contributions from Brazilian Nanotechnology National Laboratory
327 (LNNano, Campinas-SP) for SEM and TEM are also gratefully acknowledged. This is a

328 contribution of the National Institute of Science and Technology in Complex Functional
329 Materials (CNPq-MCT/FAPESP).

330

331 **References**

- 332 1. E. Andronescu, M. Fikai, G. Voicu, D. Fikai, M. Maganu and A. Fikai, *Journal of*
333 *Materials Science: Materials in Medicine*, 2010, **21**, 2237-2242.
- 334 2. G.-D. Fang, D. D. Dionysiou, S. R. Al-Abed and D.-M. Zhou, *Applied Catalysis B:*
335 *Environmental*, 2013, **129**, 325-332.
- 336 3. A. Ito, H. Honda and T. Kobayashi, *Cancer Immunology, Immunotherapy*, 2006,
337 **55**, 320-328.
- 338 4. H. Qiu, C. Luo, M. Sun, F. Lu, L. Fan and X. Li, *Carbon*, 2012, **50**, 4052-4060.
- 339 5. M. Mahmoudi, S. Sant, B. Wang, S. Laurent and T. Sen, *Advanced Drug Delivery*
340 *Reviews*, 2010, **63**, 24-46.
- 341 6. W. Zhang, S. Jia, Q. Wu, J. Ran, S. Wu and Y. Liu, *Materials Letters*, 2011, **65**,
342 1973-1975.
- 343 7. P. A. Dresco, V. S. Zaitsev, R. J. Gambino and B. Chu, *Langmuir*, 1999, **15**, 1945-
344 1951.
- 345 8. F. Dang, N. Enomoto, J. Hojo and K. Enpuku, *Ultrasonics Sonochemistry*, 2009,
346 **16**, 649-654.
- 347 9. T. J. Daou, G. Pourroy, S. Bégin-Colin, J. M. Grenèche, C. Ulhaq-Bouillet, P.
348 Legaré, P. Bernhardt, C. Leuvrey and G. Rogez, *Chemistry of Materials*, 2006, **18**,
349 4399-4404.
- 350 10. A. Yan, X. Liu, G. Qiu, H. Wu, R. Yi, N. Zhang and J. Xu, *Journal of Alloys and*
351 *Compounds*, 2008, **458**, 487-491.

- 352 11. D. Maity, P. Chandrasekharan, P. Pradhan, K.-H. Chuang, J.-M. Xue, S.-S. Feng
353 and J. Ding, *Journal of Materials Chemistry*, 2011, **21**, 14717-14724.
- 354 12. T. Fan, D. Pan and H. Zhang, *Industrial & Engineering Chemistry Research*, 2011,
355 **50**, 9009-9018.
- 356 13. L.-h. Shen, J.-f. Bao, D. Wang, Y.-x. Wang, Z.-w. Chen, L. Ren, X. Zhou, X.-b. Ke,
357 M. Chen and A.-q. Yang, *Nanoscale*, 2013, **5**, 2133-2141.
- 358 14. M. Zhu and G. Diao, *The Journal of Physical Chemistry C*, 2011, **115**, 18923-
359 18934.
- 360 15. D. Cangussu, W. C. Nunes, H. L. d. S. Corrêa, W. A. d. A. Macedo, M. Knobel, O.
361 L. Alves, A. G. S. Filho and I. O. Mazali, *Journal of Applied Physics*, 2009, **105**,
362 013901.
- 363 16. P. Yang, Z. Quan, Z. Hou, C. Li, X. Kang, Z. Cheng and J. Lin, *Biomaterials*, 2009,
364 **30**, 4786-4795.
- 365 17. R. Jenkins and R. L. Snyder, *Introduction to x-ray powder diffractometry*, John
366 Wiley & Sons, New-York, 1996.
- 367 18. Q. Q. Xiong, J. P. Tu, Y. Lu, J. Chen, Y. X. Yu, Y. Q. Qiao, X. L. Wang and C. D.
368 Gu, *The Journal of Physical Chemistry C*, 2012, **116**, 6495-6502.
- 369 19. S. Liu, R. Xing, F. Lu, R. K. Rana and J.-J. Zhu, *The Journal of Physical Chemistry*
370 *C*, 2009, **113**, 21042-21047.
- 371 20. M. Jean, V. Nachbaur and J.-M. Le Breton, *Journal of Alloys and Compounds*,
372 2012, **513**, 425-429.
- 373 21. L.-L. Wang and J.-S. Jiang, *Nanoscale Research Letters*, 2009, **4**, 1439-1446.
- 374 22. S.-B. Cho, J.-S. Noh, S.-J. Park, D.-Y. Lim and S.-H. Choi, *Journal of Materials*
375 *Science*, 2007, **42**, 4877-4886.

- 376 23. B. D. Cullity, C. D. Graham, *Introduction to magnetic materials*, John Wiley,
377 Hoboken, 2009.
- 378 24. A. Yan, X. Liu, G. Qiu, H. Wu, R. Yi, N. Zhang and J. Xu, *Journal of Alloys and*
379 *Compounds*, 2008, **458**, 487-491.
- 380 25. O. Moscoso-Londono, D. Muraca, L. A. S. de Oliveira, K. R. Pirota and L. M.
381 Socolovsky, *Magnetics, IEEE Transactions on*, 2013, **49**, 4551-4554.
- 382 26. S. Mörup, M. F. Hansen and C. Frandsen, in *Comprehensive Nanoscience and*
383 *Technology*, Academic Press, Amsterdam, Editon edn., 2010, pp. 437-491.
- 384 27. S. Guo, G. Zhang and J. Wang, *Journal of Colloid and Interface Science*, 2014,
385 **433**, 1-8.

