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## COMMUNICATION

# Increased CO<sub>2</sub>/N<sub>2</sub> Selectivity of PTMSP By Surface Crosslinking

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**The surface crosslinking of poly[1-(trimethylsilyl)-1-propyne] (PTMSP) membranes by dithiothreitol under thiol-ene click reaction conditions has yielded membranes having CO<sub>2</sub>/N<sub>2</sub> selectivities in excess of 30 with CO<sub>2</sub> permeances in excess of 300 GPU (gas permeation units). The simplicity of this surface crosslinking strategy together with these permeation results suggests that PTMSP that is modified in such ways could lead to useful materials for the separation of CO<sub>2</sub>/N<sub>2</sub> from flue gas and for certain other gaseous mixtures.**

The need for stabilizing the earth's greenhouse gases is considered as urgent if further changes in climate are to be avoided.<sup>1</sup> Of major concern is the CO<sub>2</sub> that is being produced from the combustion of fossil fuels at power plants.<sup>2,3</sup> Due to the climate crisis, efforts aimed at creating cost-effective ways of separating CO<sub>2</sub> from N<sub>2</sub> in flue gas have intensified.<sup>4</sup> One strategy that is being pursued is to create materials that can absorb CO<sub>2</sub>, selectively.<sup>5,6</sup> A second strategy involves the creation of polymer membranes that show high permeability towards CO<sub>2</sub> relative to N<sub>2</sub>.<sup>7-14</sup> Of these two approaches, membrane-based separations appears to be more feasible due to lower energy and capital requirements.<sup>7,11</sup>

A cost analysis for a real-world capture of CO<sub>2</sub> from flue gas that is based on polymer membranes has shown that the minimum requirements that need to be met are polymers having (i) CO<sub>2</sub>/N<sub>2</sub> selectivities that are  $\geq 30$  and (ii) CO<sub>2</sub> permeances that are as high as possible.<sup>7,8</sup> This analysis has further revealed that membranes having CO<sub>2</sub>/N<sub>2</sub> selectivities that are much greater than ca. 30 do not help, significantly; i.e., they would have little impact on the economics of the separation. When comparing the gas permeation properties of polymers, in general, permeance values ( $P/l$ ) are often used instead of permeabilities because they take a membrane thickness into account. More specifically, permeances are permeabilities that have been normalized with respect to a membrane's thickness. In practice, permeances are calculated by

dividing the observed flux ( $J$ ) by the pressure gradient ( $\Delta p$ ) that is applied across the membrane (eq. 1).<sup>11</sup> Here,  $P$  is a permeability coefficient that is characteristic of a given membrane/permeant combination and  $l$  is the thickness of the membrane. To date poly(ethylene glycol)-based polymers appear to be the most promising materials for a real-world capture of CO<sub>2</sub> from flue gas where CO<sub>2</sub>/N<sub>2</sub> selectivities of ca. 50 and CO<sub>2</sub> permeances of ca. 1000-2000 GPU have been reported.<sup>7,12,13</sup>

$$P/l = J/\Delta p \quad (1)$$

One unique polymer that has attracted broad attention in the gas separation area is poly[1-(trimethylsilyl)-1-propyne] (PTMSP). Owing to its high free volume and glassy state, PTMSP exhibits exceptionally high gas permeances.<sup>14</sup> However, these high permeances are usually accompanied by low permeation selectivities; i.e., there's a "trade-off" between permeance and selectivity.<sup>15</sup> In past studies, we and others have taken advantage of PTMSP's high permeability by using it as support material for extremely thin, permeation-selective Langmuir-Blodgett, Langmuir-Schafer and polyelectrolyte multilayers.<sup>16-18</sup> Other researchers have found that the grafting of poly(ethylene glycol)s to the surface of PTMSP results in a CO<sub>2</sub>/N<sub>2</sub> selectivity of ca. 80 with a CO<sub>2</sub> permeance as high as ca. 170 GPU.<sup>19</sup> A detailed comparison of a variety of other polymeric membranes with respect to CO<sub>2</sub> permeances and CO<sub>2</sub>/N<sub>2</sub> selectivities has previously been reported.<sup>12</sup>

Recently, we reported that the surface of PTMSP membranes can be modified using aqueous solutions of 3-mercaptopropanesulfonate under thiol-ene click reaction conditions to give membranes having a CO<sub>2</sub>/N<sub>2</sub> selectivity of ca. 20 with a CO<sub>2</sub> permeance of 530 GPU.<sup>20</sup> In an effort to reach the targeted CO<sub>2</sub>/N<sub>2</sub> selectivity of 30, we hypothesized that analogous *surface crosslinking* of PTMSP could result in a beneficial trade-off. Specifically, we reasoned that surface crosslinking would result in smaller pores at the surface of the membrane and an increase in

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Supporting information for this article (experimental procedures and characterization) can be found given under <http://>

the CO<sub>2</sub>/N<sub>2</sub> selectivity due to a greater reduction in diffusivity for the larger N<sub>2</sub> molecule relative to that of the smaller CO<sub>2</sub> molecule. We further reasoned that the CO<sub>2</sub> permeance should remain high as compared to most polymers that have been reported to date since surface crosslinking would involve only a thin outer layer of PTMSP. Although there have been other reports describing the crosslinking of PTMSP, to the best of our knowledge, none have been specifically designed to crosslink its surface with the goal of improved selectivity.<sup>21,22</sup>

To test our hypothesis, we chose dithiothreitol (DTT) as a surface crosslinking agent. Previous studies have shown that DTT is effective in crosslinking alkene-containing polymers *via* thiol-ene click reactions.<sup>23,24</sup> Because of its limited solubility in water (ca. 50 mg/mL), we envisioned that DTT would readily adsorb onto the hydrophobic surface of PTMSP and undergo thiol-ene crosslinking. We also reasoned that successful surface modification of PTMSP with DTT would be apparent by a significant increase in the membrane's hydrophilicity as its surface becomes hydroxylated.

With these ideas in mind, cast films of PTMSP (ca. 30 μm in thickness) were immersed in aqueous solutions containing varying concentrations of DTT plus 8 mg/mL of the free radical initiator, 2,2'-azobis(2-amidinopropane) dihydrochloride (AAPH, Fig. 1).<sup>20</sup> Surface modifications were then carried out by simply heating each solution for 5 h at 70°C under an argon atmosphere, followed by rinsing each film with deionized water. The advancing contact angle for water on untreated PTMSP was 90° ± 3°. The PTMSP surfaces that were treated with aqueous solutions that were 0.5, 5.0, 25 and 50 mg/mL in DTT showed contact angles of 85° ± 2°, 78° ± 3°, 58° ± 3° and 55° ± 2°, respectively. Examination of these films by ATR-FTIR analysis further revealed a steady increase in DTT content that accompanied this increase in hydrophilicity as evidenced by small but detectable increases in the O-H and C-S stretching regions (Figure 2).<sup>25,26</sup>

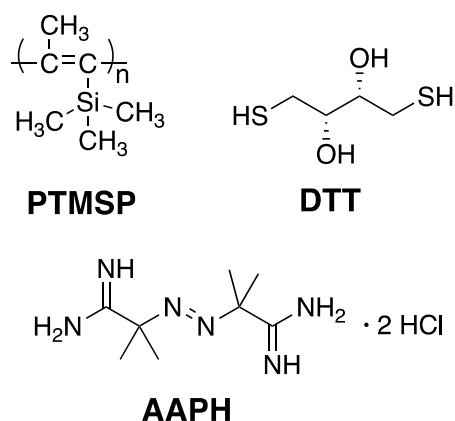


Fig. 1. Structures of PTMSP, DTT and AAPH used in this work.

Gas permeation measurements were made using a home-built constant volume-variable pressure apparatus.<sup>27</sup> The order that was used for gas measurements was H<sub>2</sub> followed by CO<sub>2</sub> and then N<sub>2</sub>. To ensure that no damage occurred during these analyses, H<sub>2</sub> measurements were repeated and found to be essentially unchanged. Table 1 lists the permeance values for each gas along with H<sub>2</sub>/N<sub>2</sub> and CO<sub>2</sub>/N<sub>2</sub> selectivities. Although elevated temperatures are known to produce thiyl radicals for thiol-ene

reactions even without radical initiators, maximum CO<sub>2</sub>/N<sub>2</sub> and H<sub>2</sub>/N<sub>2</sub> selectivities were obtained when the free radical initiator, AAPH, was included in the reaction (Table 1).<sup>28,29</sup> Thus, the production of thiyl radicals, and the resulting thiol-ene reaction, appear to be more efficient when a free radical initiator is employed.

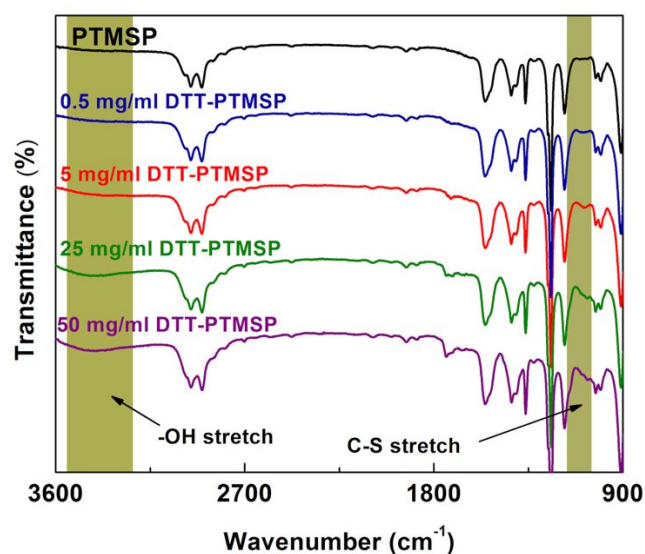


Fig. 2. ATR-FTIR spectra of PTMSP that has been treated with increasing concentrations of DTT under thiol-ene click reaction conditions..

Table 1. Permeances and Permeation Selectivities<sup>a</sup>

DTT (mg/mL)	H <sub>2</sub>	CO <sub>2</sub>	N <sub>2</sub>	H <sub>2</sub> /N <sub>2</sub>	CO <sub>2</sub> /N <sub>2</sub>
0.0 <sup>b</sup>	940	1900	420	2.2	4.5
0.5 <sup>b</sup>	350	820	120	2.9	6.8
	360	850	130	2.8	6.5
5.0 <sup>b</sup>	300	580	46	6.5	13
	310	600	47	6.6	13
25 <sup>c</sup>	280	480	25	11	19
	280	490	27	10	18
50 <sup>c</sup>	190	430	12	16	36
	190	430	13	15	33
50 <sup>c,d</sup>	180	410	11	16	37
	180	400	11	16	36
50 <sup>c,e</sup>	230	550	46	5.0	12
	250	560	47	5.3	12
50 <sup>c,f</sup>	180	320	8	23	40
	180	320	8	23	40
50 <sup>c,g</sup>	160	380	11	15	35
	160	370	11	15	34

<sup>a</sup>Permeance values are given in GPU units, where 1 GPU = 1 × 10<sup>-6</sup> (cm<sup>3</sup>/cm<sup>2</sup>·s·cm Hg). All permeances are ideal (single gas) values obtained at ambient temperatures using a pressure gradient of 2069 Torr. Unless noted otherwise, all surface modifications were carried out at 70°C for 5 h in the presence of 8 mg/mL AAPH. <sup>b</sup>Average values (± 5%) obtained from two independent measurements of the same sample. <sup>c</sup>Average values (± 5%) obtained from five independent measurements of the same sample. <sup>d</sup>24 h reaction time. <sup>e</sup>AAPH was absent in this surface treatment. <sup>f</sup>The membranes were aged for 30 days. <sup>g</sup>Water-saturated gases were employed

Examination of the surface of PTMSP by atomic force microscopy, before and after surface crosslinking under conditions that produced the maximum CO<sub>2</sub>/N<sub>2</sub> selectivity, revealed root mean-squared surface roughnesses (RMS) of 2.13 nm and 8.23 nm, respectively (Figure 3). Although we believe that crosslinking by DTT is the primary mode of surface modification of PTMSP, since H<sub>2</sub>S is known to be released from DTT under free radical conditions, it is possible that H<sub>2</sub>S may also be contributing to this surface crosslinking.<sup>30</sup>

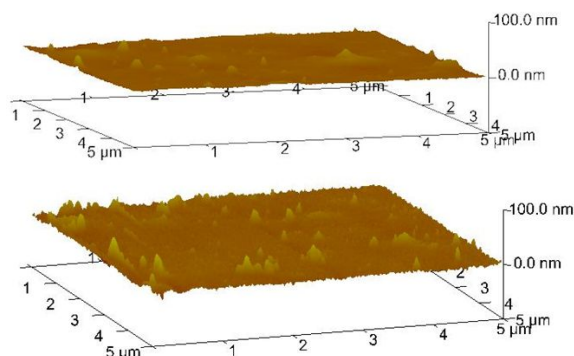


Fig. 3. AFM image of PTMSP before (top) and after DTT-modification using 50 mg/mL DTT, 8 mg/mL AAPH and 70°C for 5 h (bottom).

In Figure 4 are shown plots of CO<sub>2</sub> and N<sub>2</sub> permeances, and CO<sub>2</sub>/N<sub>2</sub> selectivities as a function of degree of surface modification as judged by contact angle measurements. Thus, significant increases in surface modification (i.e., increased hydrophilicity) were accompanied by lower permeances and higher CO<sub>2</sub>/N<sub>2</sub> selectivities.<sup>15</sup> The fact that the surface modification of PTMSP by 3-mercapto-1-propanesulfonate (an agent that is incapable of crosslinking) produces a surface that is even more hydrophilic than that produced with DTT but with significantly reduced CO<sub>2</sub>/N<sub>2</sub> and H<sub>2</sub>/N<sub>2</sub> selectivities, indicates that surface hydrophilicity plays a minor role in influencing the membrane's permeation and selectivity properties.<sup>20</sup> At the same time, this comparison provides inferential evidence for surface crosslinking by DTT. The fact that the H<sub>2</sub> permeances were always found to be greater than those of N<sub>2</sub> is a likely result of its smaller size and permeability coefficients that are dominated by diffusivity; i.e., the kinetic diameters for H<sub>2</sub> and N<sub>2</sub> are 0.289 and 0.364 nm, respectively.<sup>31</sup> This implies that changes in diffusivity are major contributors to the enhanced selectivity of this surface modified PTMSP. Since CO<sub>2</sub> has a kinetic diameter of 0.330 nm, which is significantly larger than that of H<sub>2</sub>, the fact that all of the CO<sub>2</sub> permeances were greater than those of H<sub>2</sub> is the likely result of a greater solubility contribution to the permeability coefficient for CO<sub>2</sub>; i.e.,  $P = S \times D$  where  $S$  and  $D$  are the solubility and diffusivity coefficients, respectively.<sup>15</sup> Whether specific interactions between CO<sub>2</sub> and surface-bound DTT contribute significantly to the increased CO<sub>2</sub>/N<sub>2</sub> selectivity of PTMSP remains to be established.

In preliminary studies, PTMSP membranes that were modified using 50 mg/mL of DTT and stored in a desiccator for 30 days at ambient temperature showed a ca. 25% decrease in their CO<sub>2</sub> permeances to 320 GPU with a ca. 10% increase in CO<sub>2</sub>/N<sub>2</sub> selectivity of 40.<sup>32</sup> In separate experiments, when freshly prepared membranes were exposed to water-saturated CO<sub>2</sub> and N<sub>2</sub> gases, the CO<sub>2</sub> permeances were reduced by ca. 10% to 380 GPU along with a negligible change in their CO<sub>2</sub>/N<sub>2</sub> selectivity (Table 1).

The significant improvement in the CO<sub>2</sub>/N<sub>2</sub> selectivity that we have observed for PTMSP through surface crosslinking may extend to other gaseous mixtures of interest. For example, surface crosslinked PTMSP might exhibit significant selectivity for H<sub>2</sub>O/C<sub>2</sub>H<sub>5</sub>OH separations where an ideal selectivity of 0.9 was observed for pure PTMSP.<sup>33,34</sup> In principle, an increased selectivity should be possible by taking advantage of the difference in the kinetic diameters for water (0.296 nm) and ethanol (0.430 nm).

Whether surface crosslinked PTMSP membranes in flat form, or in hollow fiber form (having surface areas 5 to 10 times greater per unit volume) can function, effectively, under real-world operating conditions for a given separation is a question that remains to be answered.<sup>35,36</sup> In a broader context, these results should encourage other investigators to consider related surface crosslinking strategies for improving the permeation properties of other high free volume polymers. Studies that are continuing in our laboratories are aimed at creating other surface crosslinked PTMSP membranes that may exhibit even higher CO<sub>2</sub> permeances with CO<sub>2</sub>/N<sub>2</sub> selectivities that are in excess of 30. Studies in progress are also being aimed at minimizing aging effects.<sup>37</sup>

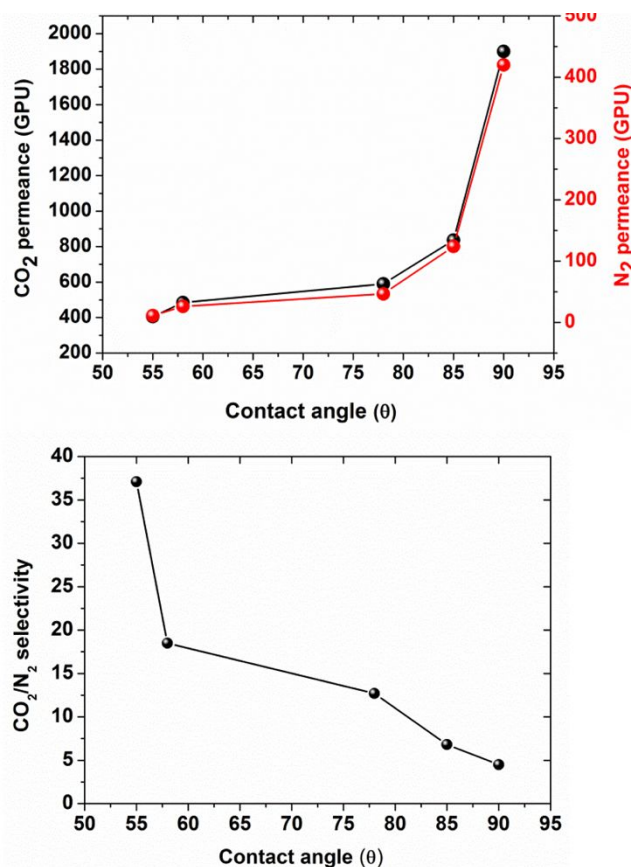


Fig. 4. Plot of (top) CO<sub>2</sub> and N<sub>2</sub> permeance and (bottom) CO<sub>2</sub>/N<sub>2</sub> selectivity as a function of the advancing contact angle for water.

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## Conflicts of interest

There are no conflicts of interest to declare.

## References

- IPCC, 2021: Climate Change 2021: The physical science basis. Contribution of working group I to the sixth assessment report of the intergovernmental panel on climate change [V. P. Masson-Delmotte, A. Zhai, S.L.Pirani, C. Connors, S. Péan, N. Berger, Y. Caud, L. Chen, M.I. Goldfarb, M. Gomis, K. Huang, E. Leitzell, J.B.R. Lonnoy, T.K. Matthews, T. Maycock, O. Waterfield, R. Yelekçi and R.Yu, B. Zhou (eds.)]. Cambridge University Press. In Press.
- F. Pattyn, C. Ritz, E. Hanna, X. Asay-Davis, R. DeConto, G.Durand, L.Favier, X. Fettweis, H.Goelzer, N.R.Golledge, P.K.Munneke, J.T.M.Lenaerts, S. Nowicki, A.J. Payne, A.Robinson, H.Seroussi, L. Trusel and M.van den Broeke, *Nat. Clim. Change*, 2018, **8**, 1053-1061.
- IEA (2021), Global Energy Review 2021, IEA, Paris <https://www.iea.org/reports/global-energy-review-2021>.
- X.Wang, C.Song, *Front. Energy Res.*, 2020, <https://doi.org/10.3389/fenrg.2020.560849>
- M. Afkhamipour and M.Nofarahi, *RSC Adv.*, 2017, **7**, 17857-17872.
- J-B.Lin, T.T.T.Nguyen, R.Vaidhyanathan, J.Burner, J.M.Taylor, H.Durekova, F.Akhtar, R.K. Mah, O. Ghaffari-Nik, S.Marx, N.Fylstra, S.S. Iremonger,, K.W. Dawson, P.Sarkar, P.Hovington, A.Rajendran, T.K.Woo and G.K.H.Shimizu, *Science*, 2021, **374**, 1464-1469.
- T.C.Merkel, H.Lin, X.Wei and R.Baker, *J. Membr. Sci.*, 2010, **359**, 126-139.
- M.G.Cowan, D.L.Gin, and R.D. Noble, *R. D. Acc. Chem. Res.*, 2016, **49**, 724-732.
- Y. Han and W.S.Winston Ho, *J. Membr. Sci.*, 2021, **628**, 119244-119267.
- S. Luo, K.A.Stevens, J.S.Park, J.D.Moon, Q.Liu and B.D. Freeman, *ACS Appl. Mater. Interfaces*, 2016, **8**, 2306-2317.
- M. Galizia, W.S.Chi, Z.P.Smith, T.C. Merkel, R.W.Baker and B.D. Freeman, *Macromolecules*, 2017, **50**, 7809-7843.
- R.S.Bhavsar, T. Mitra, D.J.Adams, A.I.Cooper and P.M.Budd, *J. Membr. Sci.*, 2018, **564**, 878-886.
- S.Luo, K.A.Stevens, J.S.Park, J.D. Moon, Q.Liu and B.D.Freeman, *ACS Appl. Mater. Interfaces*, 2016, **8**, 2306-2317.
- L.M.Robeson, *J. Membr. Sci.*, 2008, **320**, 390-400.
- B.D.Freeman, *Macromolecules*, 1999, **32**, 375-380.
- Y.Wang, V.Janout and S.L.Regen *Acc. Chem. Res.*, 2013, **46**, 2743-2754.
- C.Lin, Q.Chen, S.Yi, M.Wang, and S.L.Regen *Langmuir*, 2014, **30**, 687-691.
- J. Benito, J.Sanchez-Lainez, B.Zornoaz, S.Martin,M. Carta, R.Malpass-Evans, C.Tellez, N.B.McKeown, J.Coronas and I. Gascon, *ChemSusChem*, 2017, **10**, 4014-4017.
- C.H.Park, J.H. Lee, J.P. Jung, W. Lee, D.Y. Ryu and J.H. Kim, *Angew. Chem. Int. Ed.*, 2019, **58**, 1143-1147.
- N.B.Pramanik and S.L.Regen, *Langmuir*, 2020, **36**, 1768-1772.
- J.Jia and G.Baker, *J. Polym. Sci., B: Polym. Phys.* 1998, **36**, 959-968.
- S.D.Kelman, B.W. Rowe, C.W.Bielawski, S.Pas, A.J.Hill, D.R. Paul and B.D.Freeman, *J. Membr. Sci.*, 2008, **320**, 123-134.
- P.Sramkova, A. Zahoranova, Z.Kronekova, A.Siskova and J. Kronek, *J. Polym. Res.*, 2017, **24**:82.
- C.E.Hoyle and C.N.Bowman, *Angew. Chem. Int.*, 2010, **49**, 1540-1573.
- Y.Dong, H.Pang, H.B.Yang, C.Guo, J.Shao, Y.Chi, C.M.Li and T. Yu, *Angew. Chem. Int. Ed.*, 2013, **52**, 7800-7804.
- H. Ding, J-S.Wie and H-M.Xiong, *Nanoscale*, 2014, **6**, 13817-13823.
- C.Lin, E.R.Stedronsky and S.L.Regen, *ACS Appl. Mater. Interfaces*, 2017, **9**, 19525-19528.
- U.Biermann and J.O.Metzger, *Eur. J. Org. Chem.* 2018, 730-734.
- C.Wang, Y.Yu and S.L.Regen, *Angew. Chem. Int. Ed.*, 2017, **56**, 1639-1642.
- M.S.Akhlaq and C. von Sonntag,, *J. Am. Chem. Soc.*, 1986, **108**, 3542-3544.
- N.Mehio, S.Dai, and D. Jiang, *Phys. Chem. A.* 2014, **118**, 1150-1154.
- Z-X.Low, P.M. Budd, N.B. McKeown and D.A.Patterson, *Chem. Rev.* 2018, **118**, 5871-5911.
- M.G.Shalygin, A.A.Kozlova, A.I. Netrusov and V. Teplyakov, *Petroleum Chem.*, 2016, **56**, 977-986.
- V.P.Talluri, P.Patakova, T.Moucha and O.Vopicka, *Polymers*, 2019, **11**, 1943.
- E.Lasseguesste, J-C. Rouc and ,J-C. Remigy, *Ind. Eng. Chem. Res.*, 2013, **52**, 13146-13158.
- H.S.Lau and W.F. Yong, *J. Mater. Chem. A.*, 2021, **9**, 26454-26497. 9, DOI: 10.1039/d1ta07093b.
- C.H.Lau, P.T. Nguyen, M.R.Hill, A.W.Thornton, K. Konstas, C.M.Doherty, R.J.Mulder, L. Bourgeois, A.C.Y.Liu, D.J. Sprouster, J.P.Sullivan, T.J.Bastow, A.J. Hill, D.L. Gin and R.D. Noble, *Angew. Chem. Int. Ed.*, 2014, **53**, 5322-5326.