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## ARTICLE

## Structure, electrical and thermal properties of single-crystal BaCuGdTe<sub>3</sub>

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Single crystals of the quaternary chalcogenide BaCuGdTe<sub>3</sub> were obtained by direct reaction of elements allowing for a complete investigation of the intrinsic electrical and thermal properties of this previously uninvestigated material. The structure was investigated by high-resolution single-crystal synchrotron X-ray diffraction, revealing an orthorhombic crystal structure with the space group *Cmcm*. Although recently identified as a semiconductor suitable for thermoelectric applications from theoretical analyses, our electrical resistivity and Seebeck coefficient measurements show metallic conduction, the latter revealing strong phonon-drag. Temperature dependent hole mobility reveals dominant acoustic phonon scattering. Heat capacity data reveal a Debye temperature of 183 K and a very high density of states at the Fermi level, the latter confirming the metallic nature of this composition. Thermal conductivity is relatively high with Umklapp processes dominating thermal transport above the Debye temperature. The findings in this work lay the foundation for a more detailed understanding of the physical properties of this and similar multinary chalcogenide materials, and is part of the continuing effort in investigating quaternary chalcogenide materials and their suitability for use in technological applications.

### Introduction

Research into new materials and compositions, as well as an understanding of their physical properties, is critical in advancing technological fields of interest and enhancing device development for these applications. One class of materials that continues to be of great interest are transition metal chalcogenides, whether binary or multinary compositions<sup>1,2</sup>. In the case of binaries, this is partly due to the interest in 2D materials and their interesting properties with potential for applications including topological insulators<sup>3,4</sup>, magnetoresistance<sup>5</sup>, superconductivity<sup>6</sup> and ferromagnetism<sup>7,8</sup>. However, multinary chalcogenides are of interest for applications such as photovoltaics<sup>9–13</sup>, thermoelectrics<sup>14–18</sup>, and non-linear optics<sup>19–22</sup>. Several compositions can be synthesized with non-toxic and earth-abundant constituents, which is of particular interest for device development.

One class of quaternary chalcogenides that is of current interest are AMM'Q<sub>3</sub>, where A is an alkali or alkali earth metal, M is a transition metal, M' is a lanthanide or transition metal and Q is

a chalcogen. These materials have been known for three decades<sup>23</sup>, Ibers and co-workers having enumerated over 150 different compositions forming in three different structure types<sup>24</sup>. Most recently several hundred different compositions have been reported as stable by employing high-throughput density functional theory calculations<sup>25</sup>. Furthermore, there is a current strong interest in specific compositions for thermoelectric applications, as certain compositions have been predicted to possess low thermal conductivity<sup>25–28</sup>. Many of the initial theoretical investigations on AMM'Q<sub>3</sub> compounds predicted semiconducting behavior. K. Pal *et al.* reported that BaAgYTe<sub>3</sub> and BaCuYTe<sub>3</sub> may be suitable thermoelectric materials due to calculations indicating low lattice thermal conductivity<sup>26,27</sup>. In addition, theoretical investigations predicted BaCuGdTe<sub>3</sub> to be a semiconductor and a potential candidate for thermoelectric applications<sup>29</sup>. However, BaCuScTe<sub>3</sub> was revealed to be a poor metal even though semiconducting properties were predicted<sup>30</sup>. The crystal structures of all of the above-mentioned compositions belong to the orthorhombic *Cmcm* space group. In these structure types the lanthanide and transition metals have a direct impact on the electronic properties of these materials, therefore the specific stoichiometry also affects the assessment of these materials for thermoelectric applications. Nevertheless, it is noteworthy that there are scarce reports on the experimental transport properties for this large class of materials. Motivated by our continuing interest in quaternary chalcogenides as well as their physical properties for energy related applications, we investigated the structure, temperature-dependent transport properties and specific heat of BaCuGdTe<sub>3</sub>, a member of AMM'Q<sub>3</sub> family of compounds that forms in the orthorhombic

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*Cmcm* space group. Specifically, we grew single crystals and investigated the resistivity,  $\rho$ , Seebeck coefficient,  $S$ , Hall,  $R_H$ , thermal conductivity,  $\kappa$ , and isobaric heat capacity,  $C_p$ . To the best of our knowledge, there is no previous report on measured physical properties of this composition.

## Experimental

Single crystals of  $\text{BaCuGdTe}_3$  were grown by direct reaction of the constituent elements. In a glovebox with a  $\text{N}_2$  atmosphere, Ba pieces (99.2%, Alfa Aesar), Cu powder (99.9%, Alfa Aesar), Gd pieces (99.9999%, Ames Laboratories) and Te powder (99.8%, Acros Organics) were combined in 1:1:1:3 stoichiometric ratio in a silica ampoule and placed in a quartz tube. The quartz tube was sealed under vacuum before being placed in a resistive furnace. It was then heated to 1173 K within 48 hours and held at this temperature for 96 hours at which point it was slowly cooled to 673 K at 2.5 K/hr. The cooling rate was subsequently increased to 32 K/hr to bring the reaction mixture to room temperature. Rounded cuboid shaped dark single crystals were extracted from the resultant specimen under an optical microscope for structure characterization.

Single-crystal X-ray diffraction (XRD) measurements were conducted at NSF's ChemMatCARS, Sector 15 of the Advanced

Photon Source of Argonne National Laboratory. Data collection was accomplished using a Huber 3-axis diffractometer with a Pilatus3 X 2M CdTe detector<sup>31</sup>. Using Olex2<sup>32</sup>, the structure was solved with the XT<sup>33</sup> structure solution program using Intrinsic Phasing and refined with the XL<sup>34</sup> refinement package. An approximately  $0.7 \times 0.7 \times 1.0 \text{ mm}^3$  crystal was used for the measurement of electrical and thermal properties from 300 K to 20 K using a custom-built radiation-shielded vacuum probe<sup>35,36</sup> with a maximum experimental uncertainty of 20% associated with determination of the geometrical dimensions of the crystal. We have previously measured crystal of this, and smaller, size and employed similar procedures to that previously reported.<sup>37-40</sup> Low-temperature  $C_p$  data was measured using the heat capacity (HC) option on a commercial Quantum Design Physical Property Measurement System using the thermal N-grease accompanied by appropriate addenda. The maximum experimental uncertainty in the entire temperature range was estimated to be  $\pm 6\%$ . Temperature dependent 4-probe Hall measurements were conducted using the resistivity option at multiple magnetic fields with a maximum field of 2.4 T. A measurement sequence was set up to alternate the direction of the magnetic field at each field.

## Results and discussion

The single crystal refinement data are shown in Table 1. The atomic

**Table 1. Crystal data and structure refinement.**

<b>Empirical formula</b>	$\text{BaCuGdTe}_3$
<b>Formula weight/gmol<sup>-1</sup></b>	740.93
<b>Temperature/K, Radiation</b>	100.15, Synchrotron ( $\lambda = 0.49594 \text{ \AA}$ )
<b>Crystal system, Space group</b>	Orthorhombic, <i>Cmcm</i>
<b>a/<math>\text{\AA}</math></b>	4.4204(5)
<b>b/<math>\text{\AA}</math></b>	14.693(2)
<b>c/<math>\text{\AA}</math></b>	11.3601(12)
<b>Volume/<math>\text{\AA}^3</math></b>	737.84(14)
<b>Z</b>	4
<b><math>\rho_{\text{calc}} \text{ g/cm}^3</math></b>	6.670
<b><math>\mu/\text{mm}^{-1}</math></b>	10.725
<b>F(000)</b>	1220
<b>Crystal size/<math>\text{mm}^3</math></b>	$0.025 \times 0.025 \times 0.01$
<b>Diffractometer</b>	Huber 3-circles diffractometer
<b><math>2\theta</math> range for data collection/<math>^\circ</math></b>	2.50 to 38.58
<b>Index ranges</b>	$-5 \leq h \leq 5, -19 \leq k \leq 19, -15 \leq l \leq 15$
<b>Reflections collected</b>	9699
<b>Completeness to <math>\theta=17.311^\circ</math></b>	100.0%
<b>Independent reflections</b>	536 [ $R_{\text{int}} = 0.1045$ ]
<b>Data/restraints/parameters</b>	536/0/25
<b>Goodness-of-fit on <math>F^2</math></b>	1.026
<b>Final R indexes [<math>I \geq 2\sigma(I)</math>]</b>	$R_1 = 0.0436, wR_2 = 0.1086$

Largest diff. peak/hole / e Å<sup>-3</sup>

2.77/-2.55

coordinates and the isotropic ( $U_{eq}$ ) and anisotropic displacement parameters ( $U_{ij}$ ) are shown in Tables 2 and 3, respectively, with selected bond lengths given in Table 4. Our structure analyses indicated that BaCuGdTe<sub>3</sub> belongs to the Type II (AMM'Q<sub>3</sub>) category<sup>41</sup>. The structure is orthorhombic, space group *Cmcm*, with lattice parameters  $a = 4.4204(5)$  Å,  $b = 14.693(2)$  Å and  $c = 11.3601(12)$  Å. BaCuGdTe<sub>3</sub> can be viewed as having a two-dimensional structure with parallel [MM'Te<sub>3</sub>] slabs of alternating distorted CuTe<sub>4</sub> tetrahedra and GdTe<sub>6</sub> octahedra extending in the *ab* plane. These slabs are separated from each other by Ba atoms. Figure 1 illustrates the crystal structure and provides a perspective of the polyhedral view of the corner-shared GdTe<sub>6</sub> connected by CuTe<sub>4</sub> tetrahedra along the *ab* plane. Within the slab, the tetrahedral-tetrahedral, and octahedral-octahedral polyhedra are corner sharing, whereas the octahedral-tetrahedral polyhedra are edge-sharing. The Cu-Te bond lengths are similar to those of NaCuZrTe<sub>3</sub><sup>42</sup>, and the Gd-Te bond lengths are similar to those of CsZnGdTe<sub>3</sub><sup>43</sup>. The tetrahedral Te-Cu-Te and octahedral Te-Gd-Te angles deviate only slightly from the ideal tetrahedral angle of 109.47° and octahedral angles of 90° or 180°, respectively. Our structural data are in good agreement with that reported by Prakash *et al.*<sup>44</sup>.

To investigate the electrical properties, temperature dependent  $\rho$ ,  $S$  and  $R_H$  were measured. Measurements in two arbitrary directions

perpendicular to each other on different single crystals gave the same results. As shown in Figure 2,  $\rho$  increases with increasing temperature indicative of metallic behavior, with values comparable to that of other intermetallic compounds such as GdAl<sub>2</sub> and TbAl<sub>2</sub><sup>45</sup>. At the lowest measured temperatures  $d\rho/dT > 0$  indicating that the residual resistivity is not yet reached. Nevertheless, the relatively high residual resistance ratio (RRR  $\approx 5$ ) is comparable to that reported for single crystal metallic clathrate compositions as well as other ternary and quaternary chalcogenides<sup>46–52</sup>, and indicative of a good quality single crystal. We also considered the possibility that defects may affect the transport properties; however, our single crystal structure refinements show no fractional occupancy on any of the crystallographic positions. Hence, we conclude that the metallic behavior is intrinsic to BaCuGdTe<sub>3</sub>. We note that orthorhombic BaCuScTe<sub>3</sub> was also reported to be a metal<sup>30</sup>. As shown in Figure 3, all measured  $S$  values are positive for the entire measure temperature range indicating that holes are the majority carriers. The peak at approximately 60 K indicates strong phonon drag, an effect that is a significant contributor to  $S$  in pure metallic elements such as Cu and Ag, as well as for rare earth intermetallic binaries such as GdCu<sub>2</sub> and TbCu<sub>2</sub><sup>45,53</sup>. The solid line in the inset to Figure 3 is a linear fit of the form  $S = aT + bT^3$ , where the first and the second terms represent diffusion and phonon-drag contribution to  $S$ , respectively. From the

**Table 2. Atomic coordinates and equivalent\* isotropic atomic displacement parameters (Å<sup>2</sup>).**

Atom	$x/a$	$y/b$	$z/c$	$U_{eq}$ *	Atomic occupancy	# positions
Gd	0.5	0.5	0.5	0.0134(4)	1.0	4a
Ba	0	0.2505(1)	0.75	0.0143(4)	1.0	4c
Cu	0	0.5324(2)	0.75	0.0166(7)	1.0	4c
Te1	0.5	0.4290(1)	0.75	0.0127(4)	1.0	4c
Te2	0	0.6393(1)	0.9385(12)	0.0134(4)	1.0	8f

\*  $U_{iso}$  is defined as one third of the trace of the orthogonalized  $U_{ij}$  tensor.

**Table 3. Anisotropic Displacement Parameters (Å<sup>2</sup>).**

Atom	$U_{11}$	$U_{22}$	$U_{33}$	$U_{23}$	$U_{13}$	$U_{12}$
Gd	0.0088(6)	0.0135(6)	0.0180(7)	-0.0003(5)	0	0
Ba	0.0082(7)	0.0141(8)	0.0207(8)	0	0	0
Cu	0.0143(14)	0.0183(15)	0.0171(15)	0	0	0
Te1	0.0086(8)	0.0130(8)	0.0167(9)	0	0	0
Te2	0.0088(6)	0.0132(6)	0.0183(6)	0.0002(5)	0	0

\* The anisotropic atomic displacement factor exponent takes the form:  $-2\pi^2 [h^2a^*U_{11} + \dots + 2hka^*b^*U_{12}]$

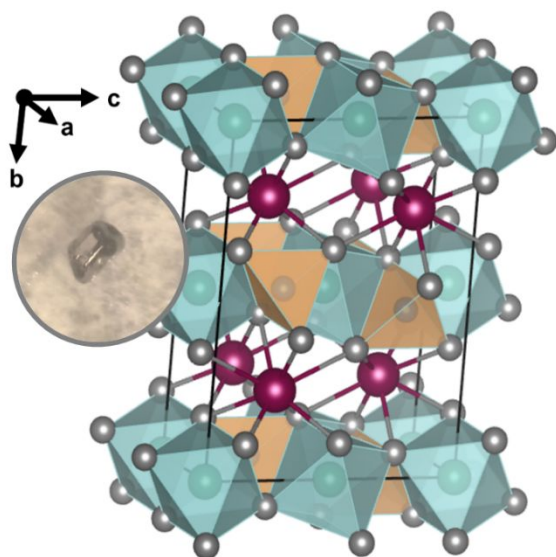
**Table 4. Selected bond lengths.**

Bond Length (Å)

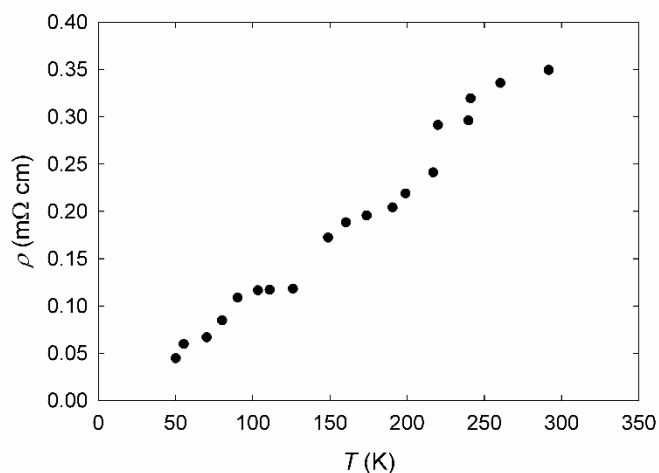
Gd-Te1	3.0256(7)
Gd-Te2	3.0920(9)
Cu-Te1	2.682(2)
Cu-Te2	2.655(2)
Ba-Te1	3.430 (2)
Ba-Te2	3.4844(12)

data fit the phonon-drag contribution,  $S_g = -0.0147^3 \mu\text{V/K}$ , is significantly lower than the values reported for Cu-Al alloys<sup>54</sup>. This effect is suppressed with increasing temperature as diffusion dominates at higher temperatures, leading to a steady increase in  $S$  with increasing temperature. The temperature dependent Hall,  $R_h$ , decreases with increasing temperature as shown in Figure 4. Near room temperature, the data becomes nearly temperature-independent with a hole concentration of  $5 \times 10^{20} \text{ cm}^{-3}$  at 300 K. The inset to Figure 4 shows temperature dependent hole mobility,  $\mu_h$ . The solid line represents a  $T^{-3/2}$  dependence indicative of acoustic phonon scattering. A hole effective mass,  $m^*$ , of  $0.3m_e$  obtained at 300 K is lower than that of  $\text{Cu}_2\text{MnSnSe}_4$ <sup>55</sup> and  $\text{Cu}_2\text{ZnSnSe}_4$ <sup>56</sup> and comparable with calculated  $m^*$  for  $\text{TlCuZrSe}_3$  and  $\text{BaCuYTe}_3$ <sup>27</sup>.

Low  $\kappa$  values have been calculated for certain compositions in the  $\text{AMM}'\text{Q}_3$  family of compounds, thus their interest as potential for

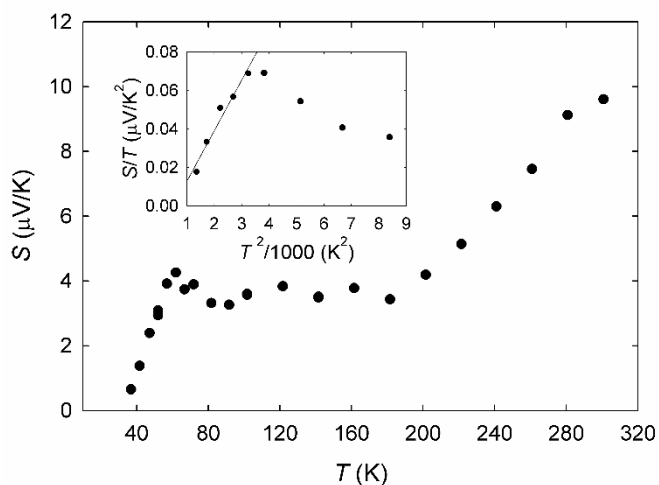


**Figure 1.** The crystal structure of  $\text{BaCuGdTe}_3$ . The layers of  $\text{GdTe}_6$  (light green) and  $\text{CuTe}_4$  (light orange) polyhedra with metal atoms in their center in the  $ab$  plane are connected by Ba (Red) atoms. Unit cell boundaries are marked by solid black lines. An image of the single crystal is also displayed.

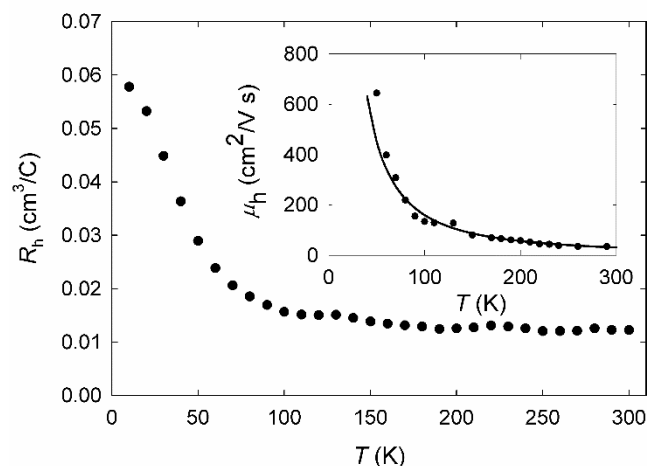


**Figure 2.** Temperature dependent  $\rho$  data.

thermoelectric applications. The alignment of  $\text{CuTe}_4$  tetrahedra and  $\text{GdTe}_6$  octahedra into polyhedral slabs in  $\text{BaCuGdTe}_3$  can presumably affect both the electronic and thermal properties of this material with the presence of lone pair electrons on tellurium<sup>30</sup>, as reported for other quaternary chalcogenide compositions<sup>57,58</sup>. To investigate the thermal properties of this material, we measured the low temperature  $C_p$ , as shown in Figure 5. The Dulong-Petit limit is achieved near 200 K indicating the majority of acoustic and optic mode frequencies are excited above this temperature.

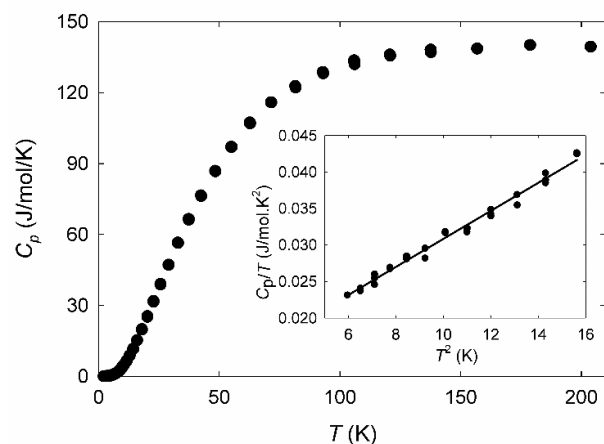


**Figure 3.** Temperature dependent  $S$  with the inset showing  $S/T$  vs  $T^2$  at low temperatures, where the solid line is a fit of the form  $S/T = a + bT^2$ .



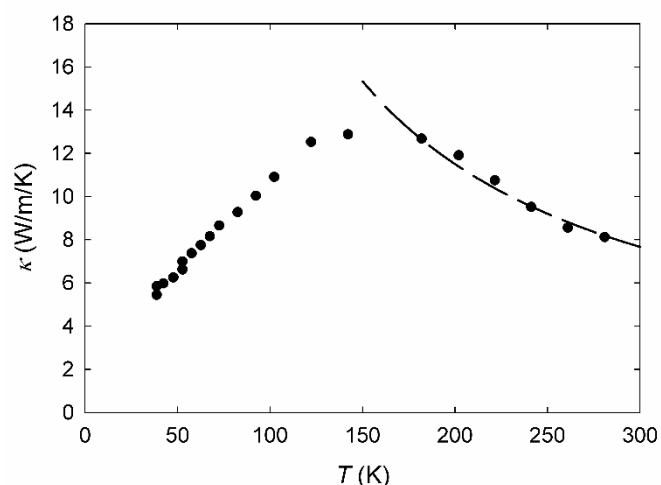
**Figure 4.** Temperature dependent  $R_h$  data. The inset shows temperature dependent  $\mu_h$  with the solid line representing  $T^{-3/2}$  dependence corresponding to acoustic phonon scattering.

The solid line in the inset of Figure 5 is a low temperature linear fit to the data of the form,  $C_p = \gamma T + \beta T^3$ , where the first and second terms represent the Sommerfeld coefficient and the lattice contribution, respectively<sup>59</sup>. Using the lattice contribution from the data fit,  $\beta = 1.8 \text{ mJ mol}^{-1} \text{K}^{-4}$ , the Debye temperature,  $\theta_D$ , can be determined from the low temperature  $C_p$  data using the relation  $\theta_D = (12\pi^4 R n_a / 5\beta)^{1/3}$ , where  $n_a$  is number of atoms per formula unit, and  $R$  is the universal gas constant. A value of 183 K thus obtained for  $\theta_D$  is in good agreement with the Dulong-Petit limit observed at high temperatures. The relatively high value of the Sommerfeld coefficient,  $\gamma = 11.8 \text{ mJ mol}^{-1} \text{K}^{-2}$ , obtained from our data fit corroborates the metallic nature of this material. Temperature dependent  $\kappa$  data for BaCuGdTe<sub>3</sub> are shown in Figure 6. The material displays relatively large  $\kappa$  over the entire measured temperature range, higher than that for BaCuScTe<sub>3</sub> at 300 K<sup>30</sup>. The electronic contribution to  $\kappa$ ,  $\kappa_E = LT/\rho$ , was estimated using the value of  $2.44 \times 10^{-8} \text{ V}^2/\text{K}^2$  for Lorentz number,  $L$ . The contribution from  $\kappa_E$  is relatively small (24% of  $\kappa$  at 300 K) for a metal and decreases with increasing temperature as expected. However,  $\kappa$  increases with



**Figure 5.** Temperature dependent  $C_p$  with the inset showing  $C_p/T$  vs  $T^2$

$T^2$  at low temperatures, where the solid line is a fit of the form  $C_p/T = \gamma + \beta T^2$ .



**Figure 6.** Temperature dependent  $\kappa$  data. The dashed line represents the  $1/T$  dependence above  $\theta_D$ .

increasing temperature up to approximately 180 K after which  $\kappa$  shows a  $1/T$  dependence indicative of Umklapp scattering, as shown in Figure 6. Although BaCuGdTe<sub>3</sub> displays metallic behavior at low temperatures,  $\kappa$  decreases with increasing temperature above  $\theta_D$ . This is in line with the relatively temperature independent behavior of  $R_h$  above  $\theta_D$ , and indicates no further contributions to thermal transport from the charge carriers. The distorted CuTe<sub>4</sub> tetrahedra and GdTe<sub>6</sub> octahedra extending in the  $ab$  plane in the crystal structure of BaCuGdTe<sub>3</sub>, as discussed above, presumably contributes to the temperature dependent behavior of  $\kappa$ , as in the case of BiAgOSe<sup>60</sup>.

Although BaCuGdSe<sub>3</sub> is a semiconductor with a relatively large bandgap<sup>61</sup>, the tellurium composition shows metallic behavior. A similar scenario has been reported for the p-type semiconductor BaCuScSe<sub>3</sub> and metallic BaCuScTe<sub>3</sub><sup>30</sup>. It should therefore be of interest to investigate other KCuZrS<sub>3</sub> type layered structure AMM'Q<sub>3</sub> compositions as this will aid in the fundamental understanding of these relatively unexplored but technologically important class of materials.

## Conclusions

Experimental electrical and thermal properties of BaCuGdTe<sub>3</sub> are reported for the first time. Temperature dependent  $\rho$  indicates metallic behavior with  $S$  data indicating significant phonon drag up to 60 K. Investigations into temperature dependent Hall revealed acoustic phonon scattering affecting electronic transport. Low temperature  $C_p$  measurements revealed a Debye temperature of 183 K, and a very high density of states at the Fermi level indicating metallic behavior corroborated by the relatively high  $\kappa$  observed for this material. With a majority of the predicted thermodynamically stable AMM'Q<sub>3</sub> compositions yet to be investigated, these results

contribute to the fundamental understanding of these and related materials of current interest for technologically important applications.

## Author Contributions

Wilarachchige D. C. B. Gunatilleke: investigation, formal analysis, visualization, writing – original draft. Winnie Wong-Ng: data curation, formal analysis. Teiyan Chang: data curation. Yu-Sheng Chen: data curation, formal analysis. George S. Nolas: conceptualization, investigation, supervision, writing – original draft, project administration.

## Conflicts of interest

There are no conflicts to declare.

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