Dalton Transactions

The Pressure-Stabilized Polymorph of Indium Triiodide

SCHOLARONE™ Manuscripts

COMMUNICATION

The Pressure-Stabilized Polymorph of Indium Triiodide

Danrui Ni, ^a Haozhe Wang, ^b Xianghan Xu, ^a Weiwei Xie ^b and Robert J. Cava *^a

Received 00th January 20xx, Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

A layered rhombohedral polymorph of indium (III) triiodide is synthesized at high pressure and temperature. The unit cell symmetry and approximate dimensions are determined by single crystal X-ray diffraction. Its R^3 crystal structure, with $a = 7.214$ Å, **and** *c* **= 20.47 Å, is refined by the Rietveld method on powder X-ray diffraction data. The crystal structure is based on InI6 octahedra sharing edges to form honeycomb lattice layers, though with considerable stacking defects. Different from ambient pressure InI3, which has a monoclinic molecular structure and a light-yellow color, high pressure InI3 is layered and has an orange color. The band gaps of both the monoclinic and rhombohedral variants of InI³ are estimated from diffuse reflectance measurements.**

High pressure synthesis and the influence of high-pressure treatment have attracted researchers' attention in recent years, as the application of pressure can lead to significant changes in reaction equilibria and affect both the structures and physical properties of compounds¹⁻³. High-pressure high-temperature synthetic techniques may introduce phase transitions and lead to different polymorphs⁴⁻⁶, or result in new formulas or structures⁷⁻⁹ that cannot be stabilized using traditional preparation methods at ambient pressure.

Metal trihalides often crystallize in crystal structures consisting of 1D-chains or 2D-honeycomb layers ¹⁰⁻¹². Polymorphism and phase transition behavior, as well as the structure-related properties of different phases, are aspects of their study ¹³⁻¹⁶, and pressure-induced polymorphism has been reported in several transition-metal-based systems¹⁷⁻¹⁹. In this context, some main-group-metal halide systems such as the indium iodide system²⁰ have also been reported to have relatively complex polymorphic behavior. Based on information in the

Inorganic Crystal Structure Database, indium triiodide (InI3) is suggested to have an aluminum triiodide type monoclinic structure with In_2I_6 dimers at ambient pressure²¹. Another InI_3 polymorph, mentioned in the literature as early as the 1940s²² as a low-temperature phase, was suggested to be light-sensitive and isostructural to the low-temperature phase of $CrCl₃²⁰$, but no modern structural refinement results were reported. In this report, we describe an R^3 polymorph of InI₃ with a honeycomb layered structure that is stabilized by high pressure synthesis. The structure was determined by powder X-ray diffraction (PXRD) via Rietveld refinement, with the space group symmetry and cell parameters qualitatively determined by single crystal Xday diffraction (SCXRD). Stacking faults occur in significant amounts in high pressure InI₃, which smear out the diffraction peaks (shown below), making single crystal structure refinements unreliable. Finally, some light absorption and physical properties are characterized.

Rhombohedral $Inl₃$ was prepared using high-pressure synthesis method (HP-InI₃). The ambient-pressure phase of monoclinic $Inl₃$ (AP-InI₃) was used as the starting material. Both materials are air-sensitive and hygroscopic. The purity of the AP-InI $_3$ starting material was confirmed by Le Bail fitting, as was its *P*21*/c* monoclinic symmetry (Figure 1A and 1B), consistent with the reported structure²¹. Different from the light-yellow colored AP -InI₃ powder, the sample after high pressure reaction (HP- $Inl₃$, shown in Figure 1C) appears to be orange after grinding (Figure 1B and 1D insets). Rietveld refinement of the PXRD data of HP-InI₃, collected at 300 K in an air-free environment (Figure 1D and Table 1), suggests that it has an *R*3 space group (#148) with lattice parameters *a* = 7.2144(4) Å and a *c* = 20.4704(9) Å. Due to a significant amount of disorder and stacking errors, the refinement gives 73.3% of the total indium content located at the honeycomb site (6*c* Wyckoff site), and 26.7% at a site in the honeycomb hole (the 3*a* site). The rhombohedral unit cell contains three layers of honeycomb lattice material (Figure 1C), formed by edge-sharing $Inl₆$ octahedra. The space group symmetry is confirmed by the single crystal diffraction data

a.Department of Chemistry, Princeton University, Princeton, NJ 08544, USA. b.Department of Chemistry, Michigan State University, East Lansing, MI 48824, USA.

Electronic Supplementary Information (ESI) available: CCDC deposition number 2311271; The experimental methods and details; Selected bond lengths and bond angles information. See DOI: 10.1039/x0xx00000x

COMMUNICATION Journal Name

Figure 1 (A) Crystal structure of monoclinic AP-InI₃ with In₂I₆ dimer; (B) Le Bail fitting of PXRD pattern of AP-InI₃, with a photo of powder sample shown as inset; (C) Crystal structure of R³ HP-InI₃, with a view through *c*-direction to reveal the in-plane honeycomb lattice, as well as the InI₆ coordination octahedron; (D) Rietveld refinement of HP-InI₃ PXRD pattern, with a photo of powder sample shown as the inset. The reflection positions are labeled with colored sticks in the powder diffraction patterns.

shown in Figure 2, especially in the (hk0) and (hk1) reciprocal lattice planes, although some ambiguous scattering may be observed due to the disorder in the system. (The streaking in the (h0l) and (0kl) reciprocal lattice planes is an indication of the disorder in the system.)

In HP-InI₃, the indium has a relatively symmetric octahedral coordination, with the In-I bond lengths ranging from 2.904 to 2.911 Å in the InI₆ octahedra (Table S1). When comparing these two polymorphs of InI₃, it is revealed that the In atoms in AP- $Inl₃$ which display a tetrahedral coordination, bond to iodine with distances at around 2.64 Å (the bonds to the iodines forming the unshared edges of the dimers) and 2.84 Å (the bonds forming the shared edges of the dimers)²¹. These In-I bond lengths are generally shorter than the ones in HP-InI $_3$, which suggests a stronger In-I bonding in the In_2I_6 dimer of AP-InI₃. However, based on the structures presented in Figure 1, AP-InI $_3$ is closer to being a molecular solid, in other words a much weaker interaction between different dimer units is observed. This may be due to the influence of iodine lone pairs in the AP polymorph, leading to the significantly lower density of AP-InI₃ (4.72 g/cm³) compared to HP-InI₃ (5.35 g/cm³), thus making the *R*3 polymorph more favored under pressure. This structural type, based on stacked honeycomb layers, is relatively common in metal trihalides¹⁴. Similar pressureinduced phase transition behavior can also be observed for $Mg(IO₃)₂$, which transits from a monoclinic $P2₁$ phase to a trigonal P3 between 7.5 and 9.7 GPa at ambient temperature²³. It is consistent with the commonly held principle that high pressure favors phases with higher density and larger coordination number.

The unit cell and color of HP -InI₃ match the description of the "low-temperature polymorph" of lnl_3 $20,24$. Although the latter paper is inclined to assign an *R*3 symmetry to that polymorph based on a Hamilton significance test, we have not found evidence for the absence of an inversion center when determining the structure of HP-InI₃ and thus attribute R_3 to the phase in this report. Previous reports also indicated that this phase is light-sensitive (while ours is not), and that it would

Table 1 Structural parameters and standardized crystallographic positions of HP-InI₃, from the Rietveld refinement of laboratory PXRD data collected at 300 K. wR = 12.29%, GOF = 2.479, reduced χ² = 6.15. Due to the multiplicity differences, the 6c site accounts for 73.3% of the In total while the 3*a* site accounts for 26.7%. (Standard deviations are indicated by the values in parentheses).

2 | *J. Name*., 2012, **00**, 1-3 This journal is © The Royal Society of Chemistry 20xx

Figure 2 The 0kl (top) hk0 (middle) and hk1 (bottom) reciprocal lattice planes of single crystal HP-InI3 at 300 K.

transform to monoclinic AP -InI₃ under light radiation, or by heating with a transition temperature ranging from 23 to 69 °C. Our observations confirm that HP -InI₃ is metastable, and that the samples may undergo a partial phase transition due to a delicate stimulus from the environment. However, it was also observed that with good protection, an HP-InI $_3$ sample was unchanged in either color or structure under ordinary room light at room temperature in a N_2 -protected glove box. This could be because a longer time, stronger light intensity, or higher energy radiation is needed to trigger HP-InI₃ to transform into the AP-phase than is needed for the low temperature form of InI₃. Another possible explanation is that an external factor (for example a tiny amount of moisture) might be needed to accelerate the phase transition. (This phenomenon has been observed in some halide systems, one instance being that $CsPbI₃$ was reported to catalytically transfer from a perovskite phase to a non-perovskite phase in the presence of moisture²⁵.) Diffuse reflectance measurements were conducted on both AP- $Inl₃$ and HP-InI₃. Due to the metastability of the HP-phase, however, the measured sample might have partially transformed to the ambient pressure phase during the measurement and thus two transitions can be observed in the HP -InI₃ sample reflectance curve (Figure 3). According to the comparison of the two reflectance curves, and the correlation with their sample colors, we attribute the absorption between 500 to 600 nm to the HP-InI $_3$ phase, and a second absorption at around 400 nm to the AP-phase. The band gaps were calculated

Figure3 Diffuse reflectance spectra of AP-InI₃ and HP-InI₃ samples. The inset presents the Tauc plots of indirect transitions to estimate the band gap values.

using Tauc plots (Figure 3 inset) based on the equation:²⁶ $(\alpha h \nu)^n = A(h \nu - E_g)$, where *A* is a constant and α is the absorption coefficient (cm-1). (*n* is 2 for direct transitions, and 0.5 for indirect transitions.) For indirect transitions, the values of the band gaps are calculated to be 2.71(5) eV for AP-InI $_3$, and 2.01(2) eV for HP-InI₃. DFT calculations obtained from the Open Quantum Materials Database (OQMD) provide calculated density of states for both R^3 structure InI₃ and $P2₁/c$ structure InI₃ ^{27,28}. Based on that information, $R^{\overline{3}}$ InI₃ has a calculated band gap of around 1.8 eV, while $P2₁/c$ InI₃ is calculated to have a band gap of 2.2 eV. Although it has been suggested that the band gaps estimated from diffuse reflectance using the Tauc plot may be usually underestimated²⁹, the larger band gap value of the monoclinic phase compared to the rhombohedral phase matches the experimental results.

The magnetic susceptibility (*M*/*H*) was measured versus increasing temperature on an HP -InI₃ powder sample under 1000 Oe. As is shown in the inset of Figure 4, temperatureindependent diamagnetic behavior is revealed, which is in line with the expectation of the electron configuration of In(III). Heat capacity measurements were also conducted from 2 to 200 K on dense HP-InI₃ sample pieces. As presented in Figure 4's main panel, no sharp transitions are observed, consistent with the magnetization behavior of the sample. Some small slope variation can be observed on the curve, which looks similar to

Figure 4 Magnetic susceptibility (inset) and heat capacity (main panel) data of HP-InI₃, plotted versus temperature.

COMMUNICATION Journal Name

what might happen for a second-order phase transition. A transition at around 93 K has been mentioned in previous reports²⁰. These phase transitions may be due to a minor structural transformation, suggesting that more complex polymorphic behavior may be found in the indium iodine system than is presently known. Thus it may be of future research interest to determine the low temperature crystal structure of HP-InI3.

Conclusions

A high-pressure stabilized InI³ polymorph was obtained by a high-pressure high-temperature synthetic method. Its rhombohedral structure and R^3 unit cell have been determined by SCXRD and Rietveld refinement of PXRD data. The orange color and honeycomb layered structure distinguish this HPphase from monoclinic AP-InI₃. Diffuse reflectance measurements are also conducted, from which the band gap values are estimated to be 2.71 eV for AP-InI₃ and 2.01 eV for HP-InI3, for indirect transitions. Magnetization and heat capacity have also been characterized, and suggest that further study of the polymorphic behavior of the indium iodine system may be of future interest. Compared with the description in previous reports, high pressure plays a crucial role in synthesis and forces the structure towards layered honeycomb InI₃, which further confirms the potential of high-pressure synthesis in exploring and developing new solid-state materials.

Acknowledgements

This work was funded in large part by the Gordon and Betty Moore foundation, EPiQS initiative, grant GBMF-9066. The SCXRD work performed in Weiwei Xie's group was supported by the U.S. DOE-BES under Contract DE-SC0022156.

Author Contributions

Danrui Ni: Conceptualization, Methodology, Investigation, Formal analysis, Visualization, Writing - Original Draft; **Haozhe Wang**: Validation, Investigation, Formal analysis, Visualization, Writing - Reviewing and Editing; **Xianghan Xu:** Validation, Formal analysis, Visualization, Software, Writing - Reviewing and Editing; **Weiwei Xie**: Data curation, Resource, Supervision, Writing - Reviewing and Editing; **Robert J. Cava***: Conceptualization, Resource, Supervision, Project Funding acquisition, administration, Writing- Reviewing and Editing.

Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

References

1 J. V. Badding, *Annu. Rev. Mater. Sci.*, 1998, **28**, 631–658.

2 D. A. Polvani, J. F. Meng, N. V. Chandra Shekar, J. Sharp and J. V. Badding, *Chem. Mater.*, 2001, **13**, 2068–2071.

3 B. Lorenz and C. W. Chu, in *Frontiers in Superconducting Materials*, ed. A. V. Narlikar, Springer, Berlin, Heidelberg, 2005, pp. 459–497.

4 Y. Shirako, X. Wang, Y. Tsujimoto, K. Tanaka, Y. Guo, Y. Matsushita, Y. Nemoto, Y. Katsuya, Y. Shi, D. Mori, H. Kojitani, K. Yamaura, Y. Inaguma and M. Akaogi, *Inorg. Chem.*, 2014, **53**, 11616–11625.

5 F. O. von Rohr, H. Ji, F. A. Cevallos, T. Gao, N. P. Ong and R. J. Cava, *J. Am. Chem. Soc.*, 2017, **139**, 2771–2777.

6 K. Nawa, Y. Imai, Y. Yamaji, H. Fujihara, W. Yamada, R. Takahashi, T. Hiraoka, M. Hagihala, S. Torii, T. Aoyama, T. Ohashi, Y. Shimizu, H. Gotou, M. Itoh, K. Ohgushi and T. J. Sato, *J. Phys. Soc. Jpn.*, 2021, **90**, 123703.

7 D. Ni, S. Guo, Z. S. Yang, K. M. Powderly and R. J. Cava, *Solid State Sciences*, 2019, **91**, 49–53.

8 D. Ni, S. Guo, K. M. Powderly, R. Zhong and R. J. Cava, *Journal of Solid State Chemistry*, 2019, **280**, 120982.

9 D. Ni, X. Gui, B. Han, H. Wang, W. Xie, N. P. Ong and R. J. Cava, *Dalton Trans.*, 2022, **51**, 8688–8694.

10 H. Hillebrecht, T. Ludwig and G. Thiele, *Zeitschrift für anorganische und allgemeine Chemie*, 2004, **630**, 2199–2204. 11 A. Banerjee, J. Yan, J. Knolle, C. A. Bridges, M. B. Stone, M. D.

Lumsden, D. G. Mandrus, D. A. Tennant, R. Moessner and S. E. Nagler, *Science*, 2017, **356**, 1055–1059.

12 T. Kong, K. Stolze, E. I. Timmons, J. Tao, D. Ni, S. Guo, Z. Yang, R. Prozorov and R. J. Cava, *Advanced Materials*, 2019, **31**, 1808074.

13 J. Angelkort, A. Schönleber and S. van Smaalen, *Journal of Solid State Chemistry*, 2009, **182**, 525–531. 14 M. A. McGuire, *Crystals*, 2017, **7**, 121.

15 J. R. Chamorro, T. M. McQueen and T. T. Tran, *Chem. Rev.*, 2021, **121**, 2898–2934.

16 D. Ni, K. P. Devlin, G. Cheng, X. Gui, W. Xie, N. Yao and R. J. Cava, *Journal of Solid State Chemistry*, 2022, **312**, 123240. 17 Y. Imai, K. Nawa, Y. Shimizu, W. Yamada, H. Fujihara, T.

Aoyama, R. Takahashi, D. Okuyama, T. Ohashi, M. Hagihala, S. Torii, D. Morikawa, M. Terauchi, T. Kawamata, M. Kato, H. Gotou, M. Itoh, T. J. Sato and K. Ohgushi, *Phys. Rev. B*, 2022, **105**, L041112.

18 D. Ni, X. Gui, K. M. Powderly and R. J. Cava, *Advanced Materials*, 2022, **34**, 2106831.

19 D. Ni, R. S. D. Mudiyanselage, X. Xu, J. Mun, Y. Zhu, W. Xie and R. J. Cava, *Phys. Rev. Mater.*, 2022, **6**, 124001.

20 P. P. Fedorov, A. I. Popov and R. L. Simoneaux, *Russ. Chem. Rev.*, 2017, **86**, 240.

21 J. D. Forrester, A. Zalkin and D. H. Templeton, *Inorg. Chem.*, 1964, **3**, 63–67.

22 F. Ensslin and H. Dreyer, *Zeitschrift für anorganische und allgemeine Chemie*, 1942, **249**, 119–132.

23 A. Liang, R. Turnbull, C. Popescu, F. J. Manjón, E. Bandiello, P. Rodriguez-Hernandez, A. Muñoz, I. Yousef, Z. Hebboul and D. Errandonea, *Phys. Rev. B*, 2022, **105**, 054105.

24 R. Kniep and P. Blees, *Angewandte Chemie International Edition in English*, 1984, **23**, 799–800.

25 D. B. Straus, S. Guo and R. J. Cava, *J. Am. Chem. Soc.*, 2019, **141**, 11435–11439.

26 J. Tauc, *Materials Research Bulletin*, 1968, **3**, 37–46.

27 J. E. Saal, S. Kirklin, M. Aykol, B. Meredig and C. Wolverton, *JOM*, 2013, **65**, 1501–1509.

28 S. Kirklin, J. E. Saal, B. Meredig, A. Thompson, J. W. Doak, M. Aykol, S. Rühl and C. Wolverton, *npj Comput Mater*, 2015, **1**, 1– 15.

29 A. B. Garg, D. Vie, P. Rodriguez-Hernandez, A. Muñoz, A. Segura and D. Errandonea, *J. Phys. Chem. Lett.*, 2023, **14**, 1762– 1768.