



# Modeling diurnal and annual ethylene generation from solar-driven electrochemical CO2 reduction devices

| Journal:                         | Energy & Environmental Science   |  |  |
|----------------------------------|--|--|--|
| Manuscript ID                    | EE-ART-02-2024-000545  |  |  |
| Article Type:                    | Paper  |  |  |
| Date Submitted by the<br>Author: | 113-FOD-71174  |  |  |
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#### Broader Context

Increasing greenhouse gas emissions requires the discovery and development of technologies capable of both removing  $CO_2$  and displacing current emissions-heavy technologies. Integrated solar fuels devices for  $CO_2$  reduction ( $CO_2R$ ) is a net-negative technology capable of converting waste  $CO_2$  streams into valuable chemical commodities. Streamlining the scaling process of these technologies will require tools and methodologies that can bridge the gap between bench-scale experimental results and large-scale operating systems. A key challenge for the operation of integrated solar fuels devices is maintaining optimal selectivity profiles under conditions of diurnal (e.g., throughout the day) and annual irradiance. Herein, we present a model which combines experimental  $CO_2R$  results, local meteorological data, and computational methods to develop a temperature and dependent diurnal and annual model of gaseous product output to streamline the co-design of large-scale, integrated solar-driven  $CO_2R$  systems.

# Modeling diurnal and annual ethylene generation from solar-driven electrochemical CO<sub>2</sub> reduction devices

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#### Abstract

Integrated solar fuels devices for  $CO_2$  reduction ( $CO_2R$ ) are a promising technology class towards achieving net-negative carbon emissions. Designing integrated CO<sub>2</sub>R solar fuels devices requires careful co-design of electrochemical and photovoltaic components as well as consideration of the diurnal and seasonal effects of solar irradiance, temperature, and other meteorological factors expected for 'on-sun' deployment. Using a photovoltaic-electrochemical (PV-EC) platform, we developed a temperature and potential-dependent diurnal and annual model using experimental CO<sub>2</sub>R performance of Cu-based electrocatalysts, local meteorological data from the National Solar Radiation Database (NSRD), and modeled performance of commercial c-Si PVs. We simulated diurnal product outputs with and without the effects of ambient temperature to determine gaseous product temperature sensitivity. From these outputs, we observed seasonal variation in gaseous product generation, with up to two-fold increases in ethylene productivity between the Winter and Summer, analyzed the consequences of dynamic cloud coverage, and identified periods where device cooling/heating mechanisms could be implemented to maximize ethylene generation. Finally, we modeled the annual ethylene generation for a scaled 1 MW solar farm at three different locations (Beijing, CN; Sydney, AUS; Barstow, CA) to determine the consequences of local meteorological climates on PV-EC CO<sub>2</sub>R product output, recording a maximum ethylene output of 18.5 tonne/yr at Barstow. Overall, this model presents a critical tool for streamlining the translation of experimental solar-driven electrochemical research to real-world implementation.

## Introduction

Tackling climate change requires commercializing net carbon negative technologies capable of converting waste CO<sub>2</sub> streams into valuable, multi-carbon fuels and chemicals using solar energy as the driving force. For emerging technologies, like integrated solar fuels devices for CO<sub>2</sub> reduction (CO<sub>2</sub>R), there is a need to develop tools and methodologies that can translate bench-scale experimental results to larger, pilot-scale on-sun operating systems. Successful scaling of integrated solar fuels CO<sub>2</sub>R technologies represents an opportunity to significantly reduce greenhouse gas emissions and to enable decentralization and off-grid operation of chemicals production<sup>1–3</sup>, allowing for flexibility in device location either close to the source of CO<sub>2</sub> generation or at the point of product consumption. Such flexibility could remove the need for long-haul chemicals transportation and the associated greenhouse gas emissions.<sup>4–7</sup>

A key challenge for integrated, on-sun solar-driven CO<sub>2</sub>R is promoting selective multi-carbon product generation throughout the diurnal operation cycle (i.e., throughout the day). These selectivity profiles are a complex function of operating parameters such as voltage, current density, and temperature. Copper is the most widely studied electrocatalyst for multi-carbon product generation from CO<sub>2</sub>R. However, it can produce up to 16 carbon products at a single potential, and small changes in the cathode potential can cause significant changes in reaction selectivity; for example, the Faradaic efficiency (FE) to C<sub>2</sub>H<sub>4</sub> has been shown to vary by more than 10% with cathode potential changes of ~50 mV.<sup>8-12</sup> Additionally, electrocatalyst properties, such as catalyst morphology, loading, and composition as well as microenvironment effects including pH, temperature, and mass transport rates can also significantly affect the selectivity of Cu-based CO<sub>2</sub>R.<sup>8,9,13-18</sup> Previous studies have shown that varying the size of copper nanoparticles between 20 – 70 nm can increase C<sub>2</sub>H<sub>4</sub> FE from <10% to ~40%<sup>19</sup>, while the presence of Cu<sup>+</sup> species and

sub-surface oxygen has enhanced C<sub>2</sub>H<sub>4</sub> FE to 60%.<sup>20</sup> Integrated solar fuels devices deployed outside must also contend with variability in incident solar irradiation and ambient temperature conditions, ranging from <0 - 50 °C, that will cause fluctuations in photovoltaic and electrochemical performance.<sup>21</sup> This further complicates product selectivity throughout the diurnal and annual cycles, with previous studies reporting that higher EC operating temperatures enhance overall total activity but increase H<sub>2</sub> selectivity at the cost of C<sub>2+</sub> products.<sup>13,22</sup> Successfully translating bench-scale solar fuels experiments to the development of pilot-scale plants will require careful, intentional co-design of the electrocatalyst and photovoltaic to the environmental conditions of a specific deployment location. Recent work has begun to address this challenge, focusing on solar-driven hydrogen evolution<sup>23–27</sup> and solar-driven CO<sub>2</sub>R under conditions of variable irradiation<sup>28–33</sup>. However, there remains a need to develop tools<sup>34</sup> that aid in understanding and predicting the transient, diurnal performance of integrated solar-fuels devices for complex, multi-product electrochemical reactions and the consequences of environmental conditions on the large-scale feasibility of these devices.

Herein, we develop a temperature and potential-dependent diurnal model that combines experimental results, consistent with previous electrochemical CO<sub>2</sub>R reports<sup>35–37</sup>, and real-world meteorological inputs (e.g., solar irradiance, ambient temperature) using computational regression, interpolation, and physics-based modeling techniques. This model is critical for bridging the gap between bench-scale experimental solar fuels devices and pilot-scale plants as it streamlines the location-specific co-design of the electrocatalytic and photovoltaic components of unassisted, integrated solar-driven CO<sub>2</sub>R systems. A photovoltaic-electrochemical platform (PV-EC) was utilized as the foundation for the model, where the photovoltaic and electrochemical components are co-designed for operation, but where no semiconductor/liquid junction is present.<sup>38</sup> The model aims to streamline the experimental prototyping phase, by identifying electrolyzer configurations (e.g., from a selection of catalysts) that maximize generation of a target product, and to enable the effective, intentional co-design of CO<sub>2</sub>R electrolyzers with large-scale PV modules. We analyzed the electrochemical CO<sub>2</sub>R product formation of nine Cu-based electrocatalysts under standardized experimental conditions. To probe the effects of ambient temperature, the bulk electrolyte temperature was varied as a proxy. The experimental data was fed into a non-linear regression and interpolation model and coupled to a physics-based model to simulate the PV, which utilized realworld meteorological data, including solar irradiance, ambient temperature, and wind speed.<sup>21</sup> The combined model simulates the diurnal product output with a temporal resolution of 5 min, identifying changes in product output due to seasonal variations and environmental effects. Expanding upon these results, we modeled the scaled, annual output of ethylene, for a 1 MW solar farm<sup>39,40</sup>, and the variations in annual product output which can be observed with changes in operating location. This work provides critical advancement in the implementation of solar-driven CO<sub>2</sub>R devices by providing a tool to quantitatively predict the location-specific diurnal and annual product output from a large-scale PV-EC CO<sub>2</sub>R system.

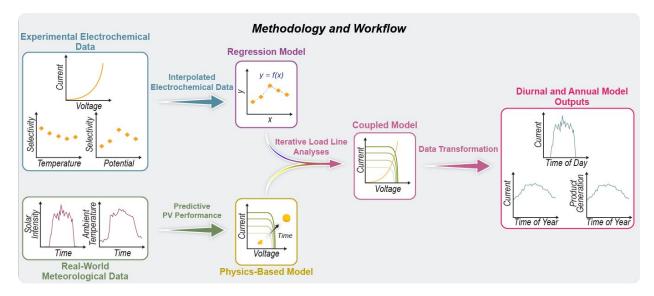
#### **Results and Discussion**

## Methodology for predicting the diurnal and seasonal product output of PV-EC CO<sub>2</sub>R devices

**Figure 1** illustrates the methodology for the model, combining bench-scale experimental electrocatalysis measurements with meteorological data sourced from the NSRD to generate a predictive model for MW-scale, real-world device performance outputs (e.g., product generation, selectivity) as a function of the time of the day and time of the year. The model focuses on the diurnal gas-phase product output and selectivity of unassisted PV-EC CO<sub>2</sub>R devices that use a Cubased catalyst and do not require external electrical bias; however, the model is readily generalizable to any solar-driven electrochemical reaction. Using the model, we investigated the consequences of two significant environmental factors on the performance of unassisted PV-EC CO<sub>2</sub>R devices: (1) varying incident solar irradiation and (2) varying temperature due to both device operation in ambient conditions and direct PV heating losses from irradiation.

A physics-based model, PVLIB<sup>41</sup>, is employed to predict PV performance, with changes in incident irradiation and PV module temperature considered for all data shown. Key components of this model include input meteorological data from the NSRD<sup>21</sup>, with a time resolution of 5 min, and the Faiman module temperature model<sup>42</sup> to capture the effects of ambient temperature, incident irradiation, convective heat transfer, and wind speed on the operating temperature of the PV module. To model the effect of incident solar irradiation on the operating point (**Figure S1**) and product distribution of the PV-EC CO<sub>2</sub>R device, the model assumes that the initial experimentally observed electrochemical activity and selectivity is maintained throughout the diurnal cycle. We used a combination of interpolation and regression models, and iterative load-line analyses, to integrate the discrete experimental data with the continuous, modeled PV data. Further analyses for PV-EC CO<sub>2</sub>R performance under conditions of assumed selectivity losses are also modeled (**see Note S1 for details**).

In this manuscript, we characterized performance using the following metrics: Faradaic efficiency (FE, %) to refer to the selectivity of the gaseous  $CO_2R$  products, activity (J, mA cm<sup>-2</sup>) to refer to the total current density observed on an electrocatalyst, productivity (J<sub>x</sub>, mA cm<sup>-2</sup>) to refer to the partial current density of specific products, and generation (tonne/yr) to refer to the annual generation of a specific PV-EC  $CO_2R$  product.



**Figure 1.** Flow diagram for the construction of the temperature and potential-dependent model. The model intakes experimentally derived electrocatalytic data describing the current-voltage characteristics, selectivity vs. voltage, and selectivity vs. temperature of the catalyst. This is combined with real-world meteorological data which the model transforms through the coupling of regression, interpolation, and physics-based modeling into outputs describing productivity vs. time of the day and product generation vs. time of the year.

## Evaluating CO<sub>2</sub> reduction performance of Cu-based electrocatalysts

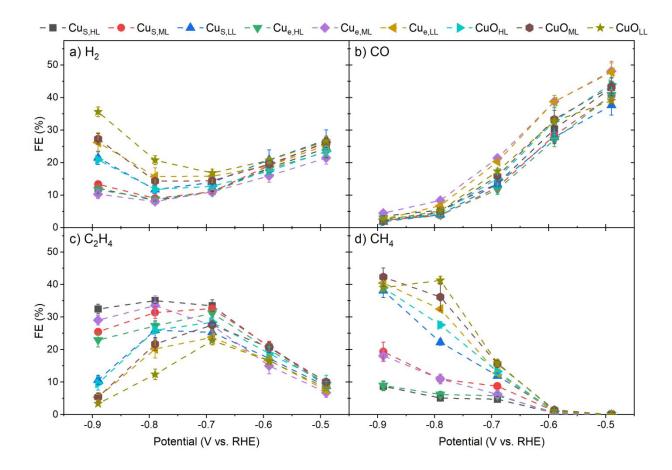
To collect the experimental electrochemical data required as the model input, we tested nine distinct Cu-based catalysts (**Table S1**) under standardized electrochemical conditions to establish a consistent baseline for comparison (see Methods). Multiple electrocatalysts were tested to develop one of the model's key features – the ability to identify the catalyst which maximizes the generation of a target product when coupled to a c-Si PV module with a maximum power output of 350 mW (Figure S2) under 1-sun intensity (1000 W m<sup>-2</sup>). In this work, we targeted the production of ethylene, which is projected to reach a global market size of 287 billion USD by  $2030^{43}$  and has a CO<sub>2</sub> emissions footprint of 1.56 tonne CO<sub>2</sub>/tonne ethylene<sup>44</sup>. Solar-driven CO<sub>2</sub>R offers a pathway to decarbonizing the ethylene industry and to long-term storage of solar energy in molecular form.<sup>45</sup>

For Cu-based electrochemical CO<sub>2</sub>R, numerous methods exist to steer selectivity and activity, including electrocatalytic methods such as catalyst morphology, loading, and chemical composition (e.g., oxides) and electrolyte-derived methods such as pH, flow rate, and electrolyte composition.<sup>15–17,46–51</sup> Each of these methods can be used to alter the electrochemical (EC) polarization curve, thereby changing the device operating point, which corresponds to the intersection between the EC polarization curve (current vs. voltage) and PV polarization curve (current vs. voltage) (**Figure S1**).<sup>52</sup> While it may be possible to dynamically vary electrolyte conditions to respond to diurnal variations, it is more difficult to alter the electrocatalyst, which is a fundamental driver of PV-EC CO<sub>2</sub>R activity and selectivity. <sup>8,9,21,53</sup> Furthermore, it is costly

and time-consuming to optimize an electrocatalyst to a photovoltaic system using experimental methods only.<sup>54–56</sup>

We utilized a gas diffusion electrode (GDE) electrochemical flow cell (**Figure S3**) which has similar structure to the membrane electrode assemblies (MEA) commonly used to achieve industrially relevant current densities.<sup>34,57,57,58</sup> Using a non-zero gap GDE flow cell allowed for simple incorporation of a reference electrode to the electrochemical device, enabling ready measurement of cathode potential ( $E_{WE}$ ), full cell voltage ( $V_{op}$ ), and selectivity (FE). We investigated the effects of catalyst morphology, loading, and chemical composition, by employing three different fabrication methods (Cu sputtering, Cu<sub>S</sub>; Cu e-beam deposition, Cu<sub>e</sub>; Cu reactive oxide sputtering, CuO) with three relative loadings for each catalyst (LL = lowest loading, ML = medium loading, HL = high loading) (**Table S1, Figures S4 – S7**).

**Figure 2** gives the Faradaic efficiency (FE) of the gaseous products (H<sub>2</sub>, CO, C<sub>2</sub>H<sub>4</sub>, and CH<sub>4</sub>) for each of the nine electrocatalysts vs.  $E_{WE}$  (**Figures S8 – 9**).<sup>59–61</sup> To connect the selectivity of the catalyst to the EC polarization curves, cyclic voltammetry was conducted prior to the potentiostatic experiments to determine the geometric current density (J<sub>geo</sub>) vs. cell voltage (V<sub>op</sub>) and J<sub>geo</sub> vs  $E_{WE}$  behavior of each electrocatalyst (**Figure S10**). All nine of the catalysts exhibit similar trends in their H<sub>2</sub> and CO selectivity, with H<sub>2</sub> reaching a minimum from -0.69 to -0.79 V vs. RHE and CO decreasing with applied potential. The C<sub>2</sub>H<sub>4</sub> maximum FE coincides with the H<sub>2</sub> minimum FE, reaching a value of ~35% for the Cu<sub>s,HL</sub> sample. At higher applied potentials, a deviation between catalysts occurs, with the CuO and lower loading Cu<sub>s</sub> and Cu<sub>e</sub> samples producing <10% FE for C<sub>2</sub>H<sub>4</sub>. As the C<sub>2</sub>H<sub>4</sub> FE decreases, CH<sub>4</sub> FE rapidly increases, reaching ~40% CH<sub>4</sub> FE at -0.89 V vs. RHE, which aligns with observations from previous studies into Cu-based CO<sub>2</sub>R.<sup>9,51,62-64</sup> While **Figure 2** gives a clear description of the electrocatalytic performance under the standardized conditions, it cannot alone predict the output when used in a PV-EC CO<sub>2</sub>R under diurnal conditions.



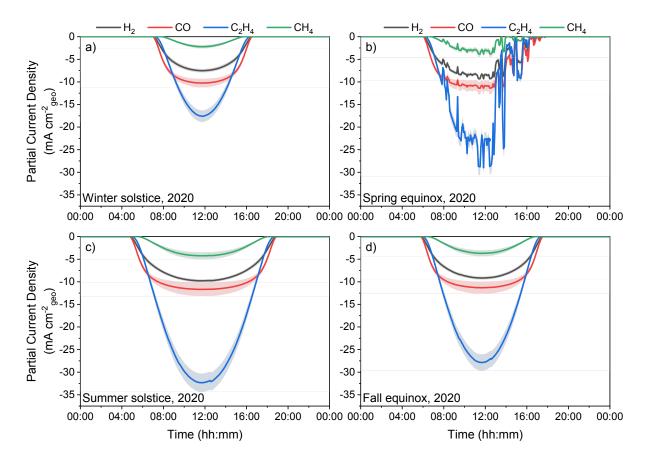
**Figure 2.** Faradaic efficiency of the gaseous products for (a)  $H_2$ , (b) CO, (c)  $C_2H_4$ , (d) CH<sub>4</sub> for CO<sub>2</sub>R experiments conducted at various applied working electrode potentials ( $E_{WE} = -0.49$ , -0.59, -0.69, -0.79, -0.89 V vs. RHE) for each of the nine electrocatalysts. The selectivity was determined via gas chromatography sampled during potentiostatic measurements. The first subscript on the catalyst name (S, e) refers to the catalyst fabricated via reactive oxide sputtering, e = electron-beam evaporation); CuO denotes catalysts fabricated via reactive oxide sputtering. The second subscript (LL, ML, HL) corresponds to the relative loading of the catalyst (LL = lowest loading, ML = medium loading, HL = highest loading). Error bars represent one standard deviation for at least three replicates.

#### Modeling the irradiance-driven diurnal output of PV-EC CO<sub>2</sub>R devices

Using the electrochemical data from Figure 2, recorded at 22 °C, we first modeled the output of the PV-EC CO<sub>2</sub>R devices under conditions of diurnal irradiance only, without the additional consideration of ambient temperature changes. **Figure 3** shows the results of the diurnal model for variations in incident irradiation, and compares the partial current density,  $J_x$ , of the four gas products over time, on the Cu<sub>S,HL</sub> electrocatalyst. This catalyst was selected as the representative due to the high ethylene outputs under conditions of diurnal irradiation variances only. The results of the model for each of the nine catalysts are available in **Table S2**.

Diurnal productivity for four representative days is given: the Winter solstice, Spring equinox, Summer solstice, and Fall equinox, using irradiation data for Barstow, CA in 2020.<sup>21</sup> Barstow was chosen for the model location as it has high average solar irradiation and currently houses largescale concentrated solar and PV power generation.<sup>65</sup> Figures S11 – 18 give the diurnal product output curves for the other eight electrocatalysts, while Figure S19 - 27 shows the corresponding FE selectivity during these four days. In the early hours of the morning, and later hours of the evening, when incident irradiation is low, H<sub>2</sub> and CO are the dominant products. These hours correspond to points of low V<sub>op</sub> and consequently, more positive E<sub>WE</sub>, favoring the formation of CO (Figure 2).<sup>8,9,66,67</sup> During the middle of the day, when solar irradiation is highest, C<sub>2</sub>H<sub>4</sub> productivity is maximized, with output greatest in the summer, when solar intensity is maximal. Despite the overall change in product selectivity throughout the day, the H<sub>2</sub>:CO ratio is similar across the four seasons, maintaining a range between 0.6 - 0.8. This range is consistent with the syngas ratio produced from coal gasification and presents an opportunity to utilize H<sub>2</sub> and CO generated from PV-EC CO<sub>2</sub>R as a replacement feedstock for the Fischer-Tropsch process (Figure S28).<sup>68,69</sup> Additionally, the consistency in H<sub>2</sub>:CO productivity ratio despite changes in total current density enables the future incorporation of unassisted integrated solar fuels CO<sub>2</sub>R devices with downstream processes to further react these products.<sup>70,71</sup>

The 'choppy' data observed during the Spring equinox (Figure 3b) is attributed to variations in solar irradiation due to dynamic cloud coverage. Comparing the C2H4 output in the Spring equinox to the Fall equinox, a similar maximum  $J_{C2H4}$  is observed, with a maximum of ~ -30 mA<sub>C2H4</sub> cm<sup>-2</sup> occurring around noon. However, the dynamic cloud coverage which is observed in the Spring equinox leads to significant fluctuations in the incident irradiation, and consequent operating point of the PV-EC device. This results in lower  $C_2H_4$  produced throughout the Spring equinox, with a daily production (integral of productivity over 24 h converted to a mole basis, as described in eqn. (8)) of  $0.42 \pm 0.03 \text{ mmol}_{C2H4} \text{ cm}^{-2}$  compared to  $0.59 \pm 0.04 \text{ mmol}_{C2H4} \text{ cm}^{-2}$  during the Fall (**Table S3**). A similar trend is observed for the other three gaseous products. Dynamic cloud coverage is an important environmental factor to consider when choosing locations for siting PV-EC CO<sub>2</sub>R devices as it can alter the device operating point over shorter timescales than those observed on sunny days. This sporadic change in operating point can reduce diurnal productivity and may have significant consequences for the reliability and durability of PV-EC CO<sub>2</sub>R device output and its ability to be integrated to downstream processes. While the cloud coverage observed in this work corresponds only to the four specific days in 2020, it provides insight into the effect of meteorological changes on the operation of unassisted PV-EC CO<sub>2</sub>R and should be investigated in future work to determine the long-term operation consequences of environmental phenomena on PV-EC CO<sub>2</sub>R diurnal product output and stability.



**Figure 3.** Modeling the diurnal product output of the  $Cu_{S,HL}$  system for an Si PV, with a 1-sun maximum power output of 350 mW located at Barstow, CA, for four representative days throughout 2020. (a) Winter solstice, (b) Spring equinox, (c) Summer solstice, (d) Fall equinox. Shaded regions represent uncertainty to one standard deviation.

## Modeling diurnal ambient temperature and irradiance effects in PV-EC CO<sub>2</sub>R devices

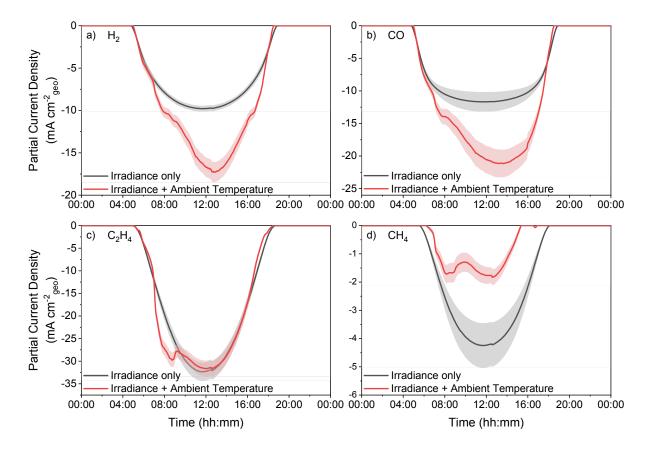
To consider additional environmental effects on PV-EC CO<sub>2</sub>R performance, we incorporated the effect of ambient temperature on electrochemical CO<sub>2</sub>R performance into our coupled model. Previous studies have reported that temperature can significantly influence selectivity changes of CO<sub>2</sub>R devices, with the extent of these effects varying with catalyst composition, electrolyte, and device configuration.<sup>13,22,72</sup> We utilized bulk electrolyte temperature as a proxy for ambient temperature and probed these effects using the Cu<sub>s,HL</sub> catalyst at temperatures of 5 °C, 15 °C, 22 °C, 35 °C, and 48.5 °C ± 0.5 °C (**see Note S1**). Electrochemical experiments were conducted in a similar process to the irradiance analysis, with cyclic voltammetry to produce EC polarization curves (J<sub>geo</sub> vs. V<sub>op</sub> and J<sub>geo</sub> vs. E<sub>WE</sub>) (**Figure S29**) followed by chronoamperometry to determine the selectivity (FE) for the gaseous products (**Figure S30**).

Increasing bulk electrolyte temperature, and consequently the temperature at the reaction interface, results in increased reaction rates and electrolyte conductivity.<sup>13,73,74</sup> In the EC polarization curves, these results manifest as increases in total current density and more positive  $E_{WE}$  values at higher temperatures for a given  $V_{op}$  (Figure S29). Each product exhibits a different temperature

dependence, with increasing temperatures resulting in greater selectivity towards  $H_2$ , decreases in the selectivity of  $C_2H_4$  and  $CH_4$ , and negligible change in CO selectivity. In addition to changes in reaction kinetics, these trends with temperature are also likely a function of factors including decreases in  $CO_2$  solubility with increasing temperature, changes in the buffer reactions of the  $CO_2$ -bicarbonate system, changes in transport properties, and products with higher activation energy (e.g.,  $C_2H_4$  and  $CH_4$ ) being more sensitive to increases in the operating temperature.<sup>75</sup>

The model integrates the influence of temperature by identifying the ambient temperature associated with the specific time of day and subsequently executing a load-line analysis to identify the operating point as a function of the PV polarization curve and relevant temperature-dependent EC polarization curve (Figure 1b).<sup>76</sup> This process is repeated to generate an array of operating points throughout the day as a function of both irradiance and ambient temperature. Figure 4 describes the output of this process, comparing the partial current density of C<sub>2</sub>H<sub>4</sub>, CH<sub>4</sub>, CO, and H<sub>2</sub> during the Summer solstice, on the Cu<sub>S HL</sub> catalyst, under 'variable irradiance only' and 'variable irradiance and ambient temperature' conditions. During the Summer solstice, the ambient temperatures reach a maximum of 38.5 °C (Figure S31) which enhances H<sub>2</sub> and CO productivity. C<sub>2</sub>H<sub>4</sub> productivity is negligibly affected while CH<sub>4</sub> productivity decreases by over two-fold. An analysis of the syngas ratio across all the seasons results in a wider range of H<sub>2</sub>:CO ratios observed, from 0.6 - 1.0 (Figures S32 - 34). The syngas ratio is most significantly affected by ambient temperature during the winter, where the colder temperatures favor C<sub>2</sub>H<sub>4</sub> selectivity at the cost of H<sub>2</sub> selectivity, and in the spring, where dynamic cloud coverage appears to increase the syngas ratio. While this larger ratio range will require more care with downstream reactors, the observed ratios are still within the boundaries of those obtained by conventional coal gasification (Figure S35).68,69

In the summer, spring, and fall, negligible changes in  $C_2H_4$  productivity are observed, indicating that the decreases in selectivity observed with warmer temperatures are balanced by the increase in total activity (**Table S4**). Comparatively, in the winter, a 25% increase in  $C_2H_4$  productivity occurs, suggesting that  $C_2H_4$  productivity could be optimized in warmer months through device cooling methods which increase  $C_2H_4$  selectivity while maintaining moderate total activity. A similar phenomenon occurs for  $CH_4$ , highlighting that device cooling mechanisms could be a powerful technique for steering diurnal product outputs. Ambient temperature also affects the shape of the diurnal productivity curves, skewing the H<sub>2</sub> and CO maximum productivity to the right, after the irradiance maximum. This is likely due to a delayed increase in ambient temperature which increases total activity. For  $C_2H_4$  and  $CH_4$ , two peaks are observed, occurring before and after the maximum irradiance and further emphasizing the balance between temperature and selectivity which must be maintained to maximize the productivity of temperature sensitive products.



**Figure 4.** Comparison of the diurnal output of the  $Cu_{S,HL}$  system during the Summer solstice, when only irradiance is considered vs. when irradiance and ambient temperature are considered. Data shown corresponds to the gaseous products, (A) H<sub>2</sub>, (B) CO, (C) C<sub>2</sub>H<sub>4</sub>, and (D) CH<sub>4</sub>. Error bars represent one standard deviation. Corresponding figures for the  $Cu_{S,HL}$  system for the winter solstice, spring equinox, and fall equinox are given in **Figures S32 – 34**.

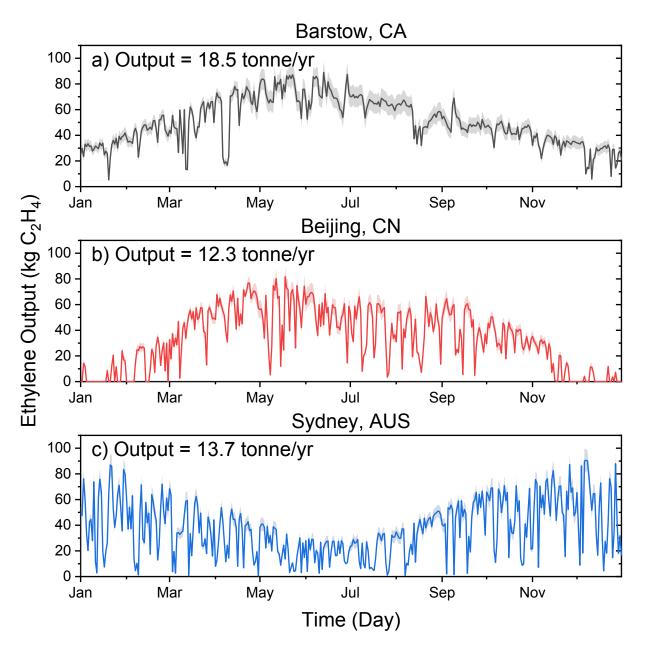
## The effects of scale and location on annual ethylene generation

Expanding upon the diurnal product output model, we calculated the average annual ethylene generation for a hypothetical 1 MW solar farm, by calculating a scaling factor based on the maximum power output of the modeled Si PV under 1-sun intensity, 350 mW (Note S2, Figure S36). The resulting output from Barstow, CA was compared to those of PV-EC CO<sub>2</sub>R devices located in Beijing, CN and Sydney, AUS. The calculation assumes that the scaled electrochemical device (~300 m<sup>2</sup>, divided into different electrochemical cells) maintains the selectivity and activity observed for the 1 cm<sup>2</sup> device. In practice, this performance is likely to have significant differences, and would involve utilizing an MEA over a GDE flow cell; however, this work aims to demonstrate the utility of the model and to highlight the impact of meteorological factors and location on product output. As data from scaled-up electrochemical cells (e.g., 1 m<sup>2</sup> MEA) becomes available, it can be readily incorporated into our existing model to determine scaled PV-EC CO<sub>2</sub>R performance.

Figure 5 provides a comparison of the scaled, annual results for the three locations. These results correspond to an ideal PV-EC CO<sub>2</sub>R device in which no catalyst performance degradation occurs. Under these conditions, Barstow produces the most ethylene via unassisted PV-EC CO<sub>2</sub>R, generating 18.5 tonne/yr compared to 12.3 tonne /yr for Beijing and 13.7 tonne/yr for Sydney (Table S5). The annual output curve for Barstow also varies less than the other two locations, emphasizing the effect of dynamic cloud coverage and weather events on the output of PV-EC CO<sub>2</sub>R devices. A particular factor experienced in Beijing is temperatures which fall outside of the experimental operating range (0 - 48.5 °C) during hours of irradiation. At temperatures outside of this range, it is no longer optimal to operate device, as this could risk damaging the electrochemical components; below 0 °C, issues of electrolyte freezing and mass transport arise, while above 50 °C, we observed deformation of the anion exchange membrane.<sup>77,78</sup> At these temperature extremes, the model assumes that the PV-EC device will not be operated to protect the device equipment (e.g., wind turbine cut-out speed).<sup>79</sup> However, future work could model the use of heating or cooling mechanisms to control the temperature of the device in the extreme temperature regions and begin to consider greater degrees of engineering complexity to assess the viability of operating solar-driven devices during nighttime or periods of excess electricity generation from the grid.

The development of stable CO<sub>2</sub>R electrocatalysts is ongoing, with state-of-the-art catalysts reaching operational lifetimes over 100 hours; however, most Cu-based catalysts exhibit stability on the scale of tens of hours.<sup>80–83</sup> Therefore, we conducted a preliminary analysis to begin considering the impact of catalyst degradation on product output and operating lifetimes (**Figures S37 – 38**). We modeled the annual generation curves for Barstow, CA, incorporating constant ethylene activity loss over 24 h periods (e.g., % decrease in C<sub>2</sub>H<sub>4</sub> partial current density from initial) and specifying minimum performance thresholds of ethylene activity (e.g., 60/80% of original partial current density) for the catalyst before it would be replaced. These results indicate that the minimum performance threshold has a stronger influence on annual generation than to the rate of activity loss. A 10% loss in annual C<sub>2</sub>H<sub>4</sub> generation (1.9 tonne C<sub>2</sub>H<sub>4</sub>/yr) compared to the ideal scenario is observed for a minimum threshold of 80% relative activity. Changing the degradation rate with a constant minimum threshold negligibly affected the annual output; however, the cost of changing the catalyst more frequently is expected to increase the cost of the output product.

The diurnal and annual model we have developed provides a tool to streamline the scaling of solardriven electrochemical technologies from bench-scale experimental results to larger installations. This model offers a bridge between bench-scale experiments and pilot scale operation by identifying, from various electrocatalyst and photovoltaic configurations, the configuration which optimizes the generation of a target product under varying meteorological conditions and solar irradiances at a specific location. Continued efforts to build upon this diurnal and annual model can accelerate the deployment of scaled solar-driven CO<sub>2</sub>R pilot plants, which will be required to construct fully validated models of large-scale systems that wholly capture the transient influences of location-specific irradiance, temperature, and other environmental factors. Previous research into semi-empirical regression modeling for scaled alkaline<sup>84</sup> and PEM water electrolyzers<sup>85,86</sup>, and electrochemical CO<sub>2</sub> reduction<sup>34,87</sup>, has highlighted the need for the synergistic pairing of modeling and experimental results to inform each progressive point of scaling. While this work presents a first step to enabling efficient, economic scaling of solar-driven electrochemical technologies, it has also identified areas of future work which should be considered. Key areas include: (1) understanding long-term durability of electrocatalysts, under steady and variable operating conditions; and (2) identifying unique challenges which arise when electrode and photovoltaic areas are scaled to the sizes required to operate a 1 MW+ scale solar fuels plant. As research and development efforts in this area progress, the model can be adapted to account for long-term degradation and challenges with device scaling. With progressions in scaling, it will be important to consider the integration of electrochemical solar fuels devices into the broader chemicals manufacturing infrastructure. In this vein, some important areas to research will include: (1) the integration of solar-driven electrochemical devices with upstream (e.g., potentially impure CO<sub>2</sub> streams) and downstream processes (e.g., syngas in existing thermochemical infrastructures) and the effects of variable outlet stream ratios on these processes; (2) detailed technoeconomic analyses (TEAs), life-cycle assessments (LCAs), studies of energy return on energy invested (EROEI), and environmental impact studies, among others, to obtain more accurate predictions of the environmental, climate, socio-economic, and sustainability benefits of solar fuels technologies; (3) the consequences of different electrochemical cell architectures and chemistries (e.g., alternative anode chemistries) on the diurnal and yearly output of solar fuels devices; and (4) develop a deeper understanding of the benefits of operating cell configurations under conditions of diurnal and stored electrical (or chemical) power to determine the role of solar-driven electrochemical technologies in the broader discussion of energy storage and grid balancing.



**Figure 5.** Annual output of  $Cu_{S,HL}$  PV-EC CO<sub>2</sub>R device for a PV-EC CO<sub>2</sub>R device scaled to a 1 MW solar farm located in: (a) Barstow, CA; (b) Beijing, CN; (c) Sydney, AUS. Shaded regions represent error to one standard deviation.

#### Conclusion

We have developed a temperature and potential-dependent model capable of converting benchscale electrochemical results into quantifiable product outputs under dynamic diurnal operating conditions (i.e., diurnal product output curves). Using electrochemical characterization (e.g., cyclic voltammograms, FE) in combination with real-world meteorological data, we investigated the consequences of meteorological and environmental effects on the performance of an unassisted PV-EC CO<sub>2</sub>R device. We investigated the CO<sub>2</sub>R performance of nine Cu-based catalysts across standardized experimental conditions and combined these results with a physics-based model of a Si PV module. In addition to capturing the effects of diurnal irradiance on productivity, with CO and H<sub>2</sub> productivity dominant in the early and late hours of the day, and C<sub>2</sub>H<sub>4</sub> productivity dominant during the middle of the day, the model captures environmental effects such as dynamic cloud coverage and provides insight into their resultant consequences on target product activity. Utilizing bulk electrolyte temperature as a proxy for ambient temperature, we observed that enhanced temperatures increase H<sub>2</sub> and CO activity, but that the H<sub>2</sub>:CO outlet ratio remains comparable to syngas derived from coal gasification and could be used for downstream reaction processes. Additionally, we found that  $C_2H_4$  and  $CH_4$  are particularly sensitive to warmer temperatures and propose that device cooling mechanisms could be key to maximizing C<sub>2</sub>H<sub>4</sub> output in warmer seasons (e.g., summer, fall) and in locations that experience extreme heat waves. Finally, we developed an annual model of product output, scaled to a 1 MW solar farm, to probe the consequences of siting location and catalyst degradation on annual ethylene generation. We observed a maximum generation of 18.5 tonne C<sub>2</sub>H<sub>4</sub>/yr at Barstow, CA and note that while integrated solar fuels devices have flexibility in their location, local climates could significantly affect their performance. The future deployment of solar-driven CO<sub>2</sub>R devices will require a synergistic pairing of modeling and experimental results, with these models and results being updated cyclically as progress toward scale-up informs next-generation designs. By integrating computation with experimental data, this work has developed a first step towards scaling of solardriven CO<sub>2</sub>R devices and can be used to optimize the location-specific output of a target electrochemical product for integration of solar fuels technology into broader chemicals manufacturing processes.

# Methods

## **Copper Electrocatalyst Deposition Methods**

Sigracet 39 BB (Fuel Cell Store) carbon paper was cut into  $6 \times 6 \text{ cm}^2$  pieces. The Cu electrocatalyst was deposited on the PTFE-treated side of the microporous layer. For the e-beam deposited samples, an AJA e-beam evaporator was used to deposit either 50 nm, 100 nm, or 200 nm of Cu at a rate of 1 Å/s. For the sputtered samples, Cu was sputtered at three different power settings: 200 W, 100 W, and 50 W for 600 s using a Lesker sputterer. For the CuO samples, reactive oxide sputtering was conducted on the Lesker sputterer by flowing O<sub>2</sub> at a flow rate of 4 sccm, with Ar flowing at 20 sccm. Following deposition of the catalyst layer, the electrode was cut into 1.5 x 1.5 cm<sup>2</sup>. Electrical connection was made to the catalyst via Cu electrical tape.

## **Electrocatalyst Characterization**

X-ray photoelectron spectroscopy (XPS) was used to analyze the surface of the electrodes pre- and post- electrolysis. Survey and high-resolution spectra were obtained using a PHI VersaProbe 3 XPS instrument with monochromatized Al K $\alpha$  (1486.4 eV) radiation in the Stanford Nano Shared Facilities. Survey spectra were conducted on a 200  $\mu$ m spot size to identify specific binding energies of interest using a pass energy of 224 eV with a 0.8 eV step size. High-resolution spectra were conducted on the same spot using a 55 eV pass energy with a 0.1 eV step size to identify surface chemical composition and oxidation states. An ion gun (2kV, 1.5 $\mu$ A) was used to sputter through the oxidized surfaces for 3 min after the initial survey and high-resolution spectra were collected. Automatic charge neutralization was used during sputtering to prevent charging effects and a total of ~15 - 20 nm was removed using SiO<sub>2</sub> on Si wafer as a reference. Survey and high-resolution spectra were calibrated to the adventitious C 1s peak at 284.8 eV. Peak fitting was performed using Shirley backgrounds and Gaussian-Lorentzian line shapes in CasaXPS software.

Scanning electron microscopy (SEM) was conducted using an FEI Magellan 400 XHR Scanning Electron Microscope to determine the morphology of the catalyst surface. An electron beam of 2 kV and 25 pA was used, and the samples were supported by conductive silver paint.

## **Electrochemical Cell Setup**

A gas diffusion electrochemical flow cell was constructed according to **Figure S3** with an active working electrode geometric area of 1 cm<sup>2</sup> and a counter electrode area of 4.0 cm<sup>2</sup>. The counter electrode was oversized to reduce the overpotential required to drive the oxygen evolution reaction. The porous Cu electrodes were used as the cathode, with Ni foil as the counter electrode, and an Ag/AgCl reference electrode in the cathode chamber. The catholyte and anolyte reservoirs each contained 13 mL of 1.0 M potassium hydroxide (KOH, Sigma-Aldrich, semiconductor grade, 99.99% trace metals basis) which was circulated through the electrochemical cell at 2 mL/min. A Sustainion anion exchange membrane (Dioxide Materials) was used to separate the working and counter electrode compartments to prevent product crossover. Electrochemistry was performed using a Biologic VMP3 potentiostat. Cyclic voltammetry (CV) measurements were performed to determine the full cell and working electrode polarization curves for a particular sample. For the

CV measurements, potentiostatic electrochemical impedance spectroscopy (PEIS) measurements were conducted after the experiment (to ensure that the full cell polarization curve reported completely uncompensated potentials between 1.5 and 4 V), and the measured resistance was compensated by 85% to prevent overcompensation over the large current range tested. Chronoamperometry was conducted at five potentials in 15 min intervals (-0.49, -0.59, -0.69, -0.79, and -0.89 V vs. RHE), for a total of 1 h 15 min, with PEIS and ZIR used to compensate for 85% of the measured resistance before the potential hold.

# CO<sub>2</sub> Reduction Product Analysis

For all potential holds, gases were sampled every 15 min, once for each potential, by a gas chromatograph (GC) (SRI Instruments, MG#5). The GC was equipped with a flame ionization detector (FID) to measure carbonaceous gas products and a thermal conductivity detector (TCD) to measure H<sub>2</sub> gas. The signal response of the FID and TCD was calibrated by comparison to a series of NIST-traceable standard gas mixtures (Airgas Inc.). The catholyte was analyzed after electrolysis via <sup>1</sup>H NMR using a Varian Inova 600 MHz NMR machine with phenol and DMSO internal standards. These results were quantified by comparison to an external standard calibration curve of the CO<sub>2</sub> reduction products.

## **Temperature Setup**

An Isotemp 3016S Recirculating Chiller was used to achieve bulk electrolyte temperatures of 5°C,  $15^{\circ}$ C,  $35^{\circ}$ C, and  $48.5^{\circ}$ C, with a ±0.5°C error, in the electrochemical cell. The cell was placed approximately 10 mm above the water bath. The electrolyte reservoirs and the tubing connecting them to the cell were submerged in the water bath. The volume of the reservoirs and the length of the tubing were also adjusted to achieve the target temperatures in the cell. The reservoir volumes for the 5°C,  $15^{\circ}$ C,  $35^{\circ}$ C, and  $48.5^{\circ}$ C temperatures were 20mL, 15mL, 20mL, and 30mL respectively.

To determine the chiller settings required to achieve the desired temperatures, an AMPROBE Multilogger thermometer was inserted into the cell to measure the temperature of the bulk electrolyte. For these experiments, MilliQ water was used as the electrolyte to prevent the corrosion of the thermocouple inserted into the cell. The temperature of the cell was monitored for at least 60 min, to determine how long it would take for the system's temperature to fully equilibrate (i.e., no temperature change within the cell for > 20 min). The chiller operating temperatures for the 5°C, 15°C, 35°C, and 48.5°C temperatures were  $3.5^{\circ}$ C, 14.2°C, 38.7°C, and 58.0°C respectively. For 5°C and 15°C, 15 min of waiting time was necessary for the temperature in the cell to equilibrate, while for 35°C and 48.5°C 30 min was necessary. For the electrochemical experiments at various operating temperatures, the original electrochemical cell setup was utilized, with 1 M KOH in the electrolyte reservoirs.

## Silicon Photovoltaic Model

To simulate the multi-cell silicon photovoltaic (PV) module, a combination of literature-defined and calculated PV parameters for a single c-Si diode ( $V_g = 1.12 \ eV$ ) solar cell at the Shockley-

Queisser radiative efficiency limit (S-Q limit) under AM 1.5G solar illumination were used.<sup>88,89</sup> The current-voltage (I-V) polarization curve for a single-diode equation was defined as:

$$i = i_{sc} - i_o \left[ \exp\left(\frac{qV}{nkT}\right) - 1 \right],\tag{1}$$

where  $i_{sc}$ ,  $i_o$ , and V are the short-circuit current, dark-saturation current, and cell potential, respectively. Moreover, q, n, k, and T are the fundamental charge of an electron, diode ideality factor, Boltzmann's constant, and temperature of the solar cell respectively. It is important to note that for silicon solar cells simulated at the S-Q limit, the short-circuit current and open circuit voltage  $V_{oc}$  play the most deterministic roles in the overall shape of the I-V curve.

Integrating over the solar spectral irradiance  $(I(\lambda))$ , with respect to wavelength, the short-circuit current density for a single diode c-Si  $(V_g = 1.12 \text{ eV})$  solar cell was defined<sup>90</sup>:

$$j_{sc} = \frac{i_{sc}}{A} = \frac{q}{hc} X \int_{0}^{\frac{hc}{qV_g}} I(\lambda) \lambda d\lambda, \qquad (2)$$

where h, c, X,  $V_g$ , A, and  $\lambda$  are Planck's constant, speed of light, solar concentration, the semiconductor bandgap, solar cell area, and photon wavelength, respectively. The solar concentration is calculated with the following,

$$X = \frac{I}{I_{AM1.5G}},\tag{3}$$

where *I* is the solar irradiance from NREL's TMY3 datasets and  $I_{AM1.5G}$  is the total solar irradiance at AM 1.5G conditions (100 mW cm<sup>-2</sup>).

Rearranging eqn. 1 for open-circuit conditions ( $V = V_{oc}$ , i = 0) and expressing currents as current densities (normalized by the same solar cell area A), the dark-saturation current density was calculated as follows:

$$j_o = \frac{i_o}{A} = \frac{j_{sc}}{\exp\left(\frac{qV_{oc}}{nkT}\right) - 1}.$$
(4)

Using  $j_{sc}$  calculated under full illumination (X = 1) AM 1.5G conditions specified by NREL (43.7 mA cm<sup>-2</sup>) and the  $V_{oc}$  for standard c-Si solar cells (~680 mV<sup>91</sup>),  $j_o$  was determined to be 1.49 × 10<sup>-10</sup> mA cm<sup>-2</sup>. To achieve a 100 mA short-circuit current, solar cells of  $A = 2.29 \ cm^2$  are needed and was subsequently multiplied by  $j_{sc}$  and  $j_o$  to recover the total short-circuit current and dark-saturation current. A Si PV module based on this single-diode solar cell would require approximately 6 PV cells in series ( $N_s = 6$ ) to achieve the required  $V_{oc}$  (4.1 V) to generate appreciable FE to ethylene.

The temperature of the solar cell module throughout the day as it is heated by the sun and convectively cooled by the wind was estimated using the Faiman module temperature model<sup>42</sup>:

$$T = T_a + E_{POA} \frac{\alpha(1 - \eta_m)}{U_0 + U_1 \times WS},\tag{5}$$

where  $T_a$ ,  $\alpha$ ,  $E_{POA}$ ,  $\eta_m$ ,  $U_0$ ,  $U_1$ , WS are the ambient air temperature, absorption coefficient of the module, irradiance incident on the plane of the module, efficiency of the PV module, convective heat transfer coefficient intercept, convective heat transfer coefficient slope, and wind speed respectively. The ambient air temperatures, windspeeds, and solar irradiances are obtained from 2020 NREL TMY3 dataset for Barstow, CA throughout the diurnal (datapoints every 5 min) and annual cycles (data points every 60 min).<sup>21</sup>

Assuming the multi-cell PV module has zero series resistance and negligibly large shunt resistance, these main parameters (solar irradiance *I*, temperature of the solar cell *T*,  $i_{sc}$ , and  $i_o$ ) were fed into the MATLAB coding package, PVLIB, an open-source toolbox developed by Sandia National Laboratories. The package uses PVsyst to recalculate  $i_{sc}$  and  $i_o$  as a function of solar irradiance and temperature of the solar cell, allowing for the construction of PV I-V curves (according to eqn. 1) with time series data specified throughout the day and year.<sup>41</sup> Additional parameters used to simulate the PV module are listed in **Table 1**.

| Parameter         | Value   | Units                   | Reference |
|-------------------|---------|-------------------------|-----------|
| n                 | 1       | -                       | 88        |
| $\mu_n$           | -0.0003 | K <sup>-1</sup>         | 41        |
| $\alpha_{sh,exp}$ | 5.5     | $m^2 s^{-1}$            | 41        |
| $\alpha_{i_{sc}}$ | -0.002  | A $K^{-1}$              | 41        |
| $\alpha_{abs}$    | 1       | -                       | 88        |
| η                 | 0.33    | -                       | 92        |
| Uo                | 25      | $W m^{-2} K^{-1}$       | 42        |
| U <sub>1</sub>    | 6.84    | W m $^{-3}$ s K $^{-1}$ | 42        |
|                   |         |                         |           |

Table 1: Parameters for modeling the PV module comprised of multiple c-Si solar cells.

# **Diurnal Model**

To model the diurnal output of the electrochemical cell, constrained by the I-V behavior of the PV module, a smoothing spline regression function (p = 0.99) was fitted to the experimental electrochemical polarization curve (total current  $i_{tot}$  vs. device operating potential  $V_{op}$ ). Because the active working electrode geometric area was 1 cm<sup>2</sup>, the use of current and current density in this analysis is synonymous. The cathode potential,  $E_{we}$ , was also related to  $V_{op}$  with a smoothing spline regression (p = 0.9999).

To perform a similar analysis on the partial current densities  $j_i$ , the discrete experimental  $j_i$  was calculated as the product between  $i_{tot}$  from the cyclic voltammetry experiment and the FEs from the chronoamperometry experiments, both matched to the same  $E_{we}$ ,

$$j_i = i_{tot,CV}(E_{we}) \times \frac{FE_{i,CA}(E_{we})}{100}.$$
 (6)

The discrete  $j_i$  was then fitted to a smoothing spline regression (p = 0.995) as a function of  $E_{we}$ . Therefore, if a  $V_{op}$  was identified, it was mapped to a corresponding  $i_{tot}$  and  $E_{we}$ , which was then further converted to  $j_i$ . FE was determined by dividing the calculated  $j_i$  by the same  $i_{tot}$  identified at  $V_{op}$  and  $E_{we}$ . Because FEs were only measured at a select few  $E_{we}$ , the error in experimental FE at all  $E_{we}$  was reasonably assumed from a modified Akima piecewise cubic Hermite interpolation function. The error in  $j_i$  was calculated with standard error propagation methods as follows:

$$\delta j_i = j_i \left[ \left( \frac{\delta i_{tot}}{i_{tot}} \right)^2 + \left( \frac{\delta F E_i}{F E_i} \right)^2 \right]^{\frac{1}{2}}.$$
(7)

Inputting the solar irradiance, ambient air temperature, and windspeed into PVLIB, the I-V curves from the PV module throughout the day are calculated.  $V_{op}$  for each PV I-V curve is calculated by finding the intersection between the PV I-V curves and the experimentally measured electrochemical polarization curves using MATLAB's fsolve() function. This point is then related to the product FE and  $j_i$  by calculating the expected  $E_{we}$  using  $V_{op}$ , with the assumption of an ideal electrochemical CO<sub>2</sub>R system such that at constant  $E_{we}$ , FE remains constant.  $j_i$  was then used to model the total product output throughout the diurnal cycle. The same overall approach is utilized to analyze the device's performance over an annual timescale.

To determine the specific molar production throughout the year, the specific partial current,  $i_i$ , is integrated with respect to time to obtain the expected charge passed to that product in the device. Faraday's law of electrolysis is used to convert electrical charge, Q, into moles of product,  $n_{mol}$ ,

$$n_{mol} = \frac{Q}{vF},\tag{8}$$

where v is the number of electrons involved in the specific half reaction.

#### **Temperature Model**

A similar approach is used to incorporate ambient temperature into the PV-EC simulations, except that an additional degree of freedom must now be considered in the 3D regression functions. Locally weighted smoothing linear regression (lowess) functions with spans = 0.4 and 0.2 were used to express  $i_{tot}$  as a function of  $V_{op}$  and temperature and  $V_{op}$  as a function of  $E_{we}$  and temperature respectively. Additionally, lowess regressions with spans = 0.2 were used to describe  $j_i$  as a function of  $E_{we}$  and temperature. Biharmonic spline interpolation (MATLAB® 4 griddata method) was used to model the error found between experimentally collected data points (interpolation between temperatures and  $E_{we}$ ). The error in  $j_i$  is propagated according to eqn. 7.

## Author Contributions

K.M.K.Y. and W.J.W. contributed equally to the ideation of the project, development of the model, and writing of the manuscript. M. R. P., L. W., S.L. contributed to the completion of experimental results and characterization. A.J.K. and J.C.B. contributed to the development of the model and writing of the manuscript. A.Z.W., A.T.B., A.C.N., and T.F.J. aided in the development of the project and writing of the manuscript.

## Conflicts of Interest

The authors declare no conflicts of interest.

## Acknowledgements

This work was funded by the Liquid Sunlight Alliance, which is supported by the U.S. Department of Energy, Office of Science, Office of Basic Energy Sciences, Fuels from Sunlight Hub under Award Number DE-SC0021266. Part of this work was performed in the nano@Stanford labs, which are supported by the National Science Foundation as part of the National Nanotechnology Coordinated Infrastructure under award ECCS-1542152. Some of this work was also performed at the Stanford Nano Shared Facilities (SNSF), supported by the National Science Foundation under award ECCS-2026822. K.M.K.Y. acknowledges the Chevron Energy Fellowship and the Stanford DARE fellowship for partial support. A.J.K. acknowledges funding from the National Science Foundation Graduate Research Fellowship under Grant No. DGE 2146752. J.C.B. acknowledges support from the National Defense Science and Engineering Graduate Fellowship (NDSEG) supported by the Army Research Office (ARO) under contract FA9550-21-F-0003.

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