

Photo-responsive Diels-Alder Based Azobenzenefunctionalized Main-chain Liquid Crystal Networks

| Journal: | Journal of Materials Chemistry C |
|----------------------------------|---|
| Manuscript ID | TC-ART-03-2024-001281.R1 |
| Article Type: | Paper |
| Date Submitted by the Author: | 18-Jun-2024 |
| Complete List of Authors: | Park, Minwook; University of California Santa Barbara, Chemistry and Biochemistry Guillen Campos, Jesus; University of California Santa Barbara, Chemistry and Biochemistry Stricker, Friedrich; University of California Santa Barbara, Chemistry and Biochemistry Read de Alaniz, Javier; University of California Santa Barbara, Chemistry and Biochemistry |



ARTICLE

Received 00th January 20xx, Accepted 00th January 20xx

DOI: 10.1039/x0xx00000x

Photo-responsive Diels–Alder Based Azobenzenefunctionalized Main-chain Liquid Crystal Networks

Minwook Park, Jesus Guillen Campos, Friedrich Stricker, and Javier Read de Alaniz*

Light-responsive liquid crystal elastomer networks (LCNs) have received significant interest due to their potential application in soft robotics and shape-morphing devices. Here, we present a systematic examination of light-responsive LCNs prepared using a catalyst-free Diels–Alder cycloaddition and a new azobenzene functionalized monomer for main-chain incorporation. The networks have robust mechanical stiffness that can be reversibly modulated by 1 GPa by turning the UV light on and off. This study highlights the contribution of photothermal softening to reversibly control rheological properties of the newly developed LCNs and demonstrates the ability to tune the modulus on demand. We believe this work will guide future developments of light-responsive LCNs based on the newly developed Diels–Alder cycloaddition.

Introduction, Results and discussion, Experimental

Liquid crystal networks (LCNs) represent a class of stimuliresponsive materials, offering a unique synergy between molecular order inherent in crystalline materials and elasticity of rubber.^{1,2} LCNs typically consist of two fundamental elements: a crosslinked matrix material that forms the primary elastomer network and responsive molecules that induce beneficial properties in polymers in response to heat,³ pH change,⁴ electrical,⁵ or optical stimuli.⁶ The interplay between the inherent anisotropy of liquid crystals, the network architecture, and the stimuli-responsive motifs introduce a remarkable level of tunability to obtain the desired thermo-mechanical and dynamic properties of the polymer.⁷

Incorporating photo-responsive molecules into LCNs enables spatial and temporal material responses, as well as modulation in the material's structure and behaviour corresponding to changes in light intensity, wavelength, or polarization. This adaptability can be controlled by a range of photo-responsive chromophores, including azobenzene,⁸ diarylethene,⁹ DASA,¹⁰ hydrazone,¹¹ as well as a number of organic dyes and photothermal agents.¹² Among these, azobenzene (Azo)-derived photoswitches have emerged as particularly versatile and the most widely used molecular motif. This is primarily due to their convenient chemical incorporation into LCNs, chemical tunability, large configurational change and reversible photoisomerization.¹³ Upon absorbing light of a specific wavelength (typically in the UV region), azobenzene transitions from the thermodynamically stable *trans*-state to the higher energy *cis*-state. This photoinduced isomerization is reversible, and the *cis*-isomer can convert back to the *trans*isomer through thermal relaxation (T-type) or exposure to a different wavelength of light (P-Type). These properties have contributed to the widespread use of azobenzene-derived photochromes in LCN-based actuators and responsive soft materials.

In 2022, we reported a new set of LC monomers that incorporates furan- and maleimide-functional groups, enabling the construction of LCNs using a catalyst-free Diels–Alder (DA) cycloaddition.¹⁴ Since network formation occurs via an uncatalyzed electrocyclization, this single-pot process can be performed open to ambient conditions, on a laboratory bench without the need for a photocatalyst or external reagents. By adding photochromic molecules, the material's properties such as stiffness and shape can be dynamically tuned by exposure to light. The work presented here examines the effects of light stimulation on the rheological properties, photothermal softening, and bending actuation of azo-containing uniaxially aligned glassy LCNs (**Figure 1a**). Such properties are important for future studies in soft robots,^{15,16} adhesion¹⁷ and for light-activated damping.¹⁸

Department of Chemistry and Biochemistry, University of California Santa Barbara, Santa Barbara, California 93106, United States.

Electronic Supplementary Information (ESI) available: [details of any supplementary information available should be included here]. See DOI: 10.1039/x0xx00000x

ARTICLE



(a) Schematic illustration of photo-responsive behavior in azo-containing LC network film

(b) Synthesis of Furan-functionalized Azo LC and two types of LC network film



Figure 1. (a) Photo-responsive behaviour in azo-containing liquid crystal network film (b) Synthetic route of furan-functionalized azobenzene liquid crystal (AzoFu) and film fabrication of two types of liquid crystal (LC) network (polydomain and monodomain): (i) K₂CO₃, KI, DMF, 90 °C, 12 hrs; (ii) annealed for 7 days at 50 °C, 78 °C, 95 °C respectively (in the case of 78 °C, planar surface alignment is used for fabricating monodomain LCN

Results

To incorporate azobenzene (Azo) core into the DA-based LCN system, furan-functional groups with a six-methylene chain spacer were prepared via carbodiimide EDC coupling with the terminal ends of azobenzene-diol (Azo diol) (Figure 1b).14 The synthetic procedure and characterization are described in supporting information (Figure S1 and Figure S2). To evaluate the properties of AzoFu building block, we first examined the thermodynamic behaviour and molecular packing structure using DSC and powder X-ray diffraction (Figure S3). In these experiments, we observed slightly different results in powder samples prepared by solvent-evaporation under vacuum and annealing at 80 °C (Figure S3a and S3b).¹⁹ In the thermodynamically stable state (annealed), the retention of a broad endothermic peak (84 °C) during repeated heating processes indicates a crystal melting point (T_m) and decent thermal stability. The nematic to isotropic point (T_{ni}) was not observed under these conditions. Given the length of single

AzoFu molecule (4.00 nm) and distinct higher order diffraction ((003), (004), (005)), the molecular packing structure of the liquid crystal is expected to possess a smectic A structure with the layer *d*-spacing of 3.87 nm.²⁰ These results suggest that the AzoFu molecules are uniformly dispersed within the LCN platform and well-aligned along the ordering direction of the matrix LC molecules. In contrast, the powder derived from the solvent-evaporated state formed a less stable state with closer packed π - π interaction with two distinct *d*-spacing values (0.31) and 0.30 nm), however no clear higher-order packing structure. To prepare photo-responsive AzoLCN, we blended monomers in the optimized molar ratios (MFu: 47%, MMal: 43.7%, AzoFu: 5.1%, TMal: 4.2%). The resulting mixture was then used to fabricate the AzoLCN based on our previous work (Figure 1; (ii)).14 Here, we considered two factors: surface enforced molecular orientation and processing temperature, which can influence the final physical properties and actuation behaviour of the AzoLCN.^{21,22} Motivated by these considerations, polydomain films (pAzoLCNs) were prepared at 50 °C, 78 °C, and 95 °C, respectively, while a monodomain film (mAzoLCN) was

Journal Name

prepared with planar surface alignment at 78 °C (nematic phase). The LC properties of these films were characterized by polarized optical microscopy (POM). As expected, there was no optical difference observed by POM for the polydomain AzoLCNs in any direction (**Figure 2a** and S4). For the aligned mAzoLCN, we observed a nearly uniform bright texture in the film set at 45° to the polarized optical filters, while no transmitted light was observed at 90°, indicating uniaxially well-aligned molecular orientation.

To investigate the ability to induce photo-responsive deformation in AzoLCN, free-standing cantilevers (13 mm x 1.3 mm x 10 μ m) were prepared and monitored under UV and visible light irradiation (Figure 2b and Figure S5). It is well known that the azobenzene's isomerization (trans-cis) can enhance nanoscale voids in the glassy LCN, thereby inducing volume distribution that leads to macroscopic deformation (expansion/contraction).²³ Upon UV light irradiation (20 mW cm⁻²) for 3 minutes, the glassy pAzoLCN quickly bent away from the UV light; however, it did not return to its original shape after UV light irradiation was stopped. Additionally, the pAzoLCN remained unchanged after subsequent visible light irradiation (3 minutes). During 5 cycles alternating between UV and visible light irradiation, the pAzoLCN continued to bend down, resulting in a -4.5 mm end tip deflection (inset of Figure 2b). The other pAzoLCN cantilevers prepared at different temperatures (at 50 °C for pAzoLCN50 and at 95 °C for pAzoLCN95) showed similar deformation behaviour with the same bending direction (Figure S5).



Figure 2. (a) POM images of polydomain AzoLCN (pAzoLCN) and monodomain AzoLCN (mAzoLCN) under the crossed-polarizers (P: polarizer direction). Tip deflection tracking for photo-responsive actuating behaviour of (b) pAzoLCN and (c) mAzoLCN|| under alternative UV or visible light irradiation each for 3 minutes. Inset are the images of the photo-responsive actuating behaviour at each point.

In the case of aligned mAzoLCNs (perpendicular alignment = mAzoLCN $^{\perp}$ and parallel alignment = mAzoLCN||), bending behaviour was influenced by the molecular alignment direction (= molecular long axis) (Figure S5).²¹ For example, the upper surface of mAzoLCN $^{\perp}$ contracted parallel to molecular long axis and curled slightly upwards. However, the whole cantilever still bent away from the UV source, similar to the behaviour observed in pAzoLCN. In contrast, the mAzoLCN|| cantilever bent toward the UV source, consistent with a buildup of the cis isomer at the surface nearest the light, disrupting the order of the LCN (Figure 2c). This leads to macroscopic deformation, observed as bending toward the light. There was no additional change beyond 3 minutes, presumably due to gravity. To test this hypothesis, the pAzoLCN cantilever was hung parallel to the direction of gravity. In this configuration, exposure to UV light from the right side of the cantilever caused bending away from the source (Figure S6). Overall, the observed bending angles for the pAzoLCN and mAzoLCN are consistent with a free-volume change and anisotropy-change, respectively (Figure S7).



Figure 3. (a) UV-vis spectra variation of AzoLCN under UV or visible light irradiation. (b) Temperature and absorption (365 nm) variation of AzoLCN before and after UV light irradiation. (c) A set of DSC heating data of AzoLCN at 5 $^{\circ}$ C min⁻¹.

To gain a better understanding of the photo-responsive behaviour of both the poly- and mono-domain cantilevers, we examined the kinetics of *trans-cis-trans* isomerization through time-resolved UV-vis spectra using LCN films (**Figure 3a**). Under constant UV light irradiation, the π - π * absorption peak at 360 nm decreased by approximately 45%, reaching steady state after 15 minutes. The metastable *cis*-isomer can be converted back to the *trans*-isomer within 4 seconds upon exposure to

ARTICLE

visible light. In the absence of light, the metastable *cis*-isomer gradually returns to the *trans*-form in the dark, reaching 66% recovery after 10 minutes (**Figure 3b**). Based on these results, we believe that the main-chain azobenzene photoswitch undergoes efficient switching between *trans*- and *cis*-isomers upon UV and Vis light irradiation. However, when investigate the mechanical properties (*vide infra*), we find that recovery of the modulus in the dark is on the order of seconds. The large mismatch between the relatively slow *cis*-to-*trans*-reversed isomerization of azobenzene molecules in the dark and the fast modulus change of the polymer when the light is turned off suggest the dynamics of the *trans*-to-*cis* and *cis*-to-*trans* isomerization process is decoupled from the driving forces contributing to the macroscopic change in the material properties.

Given these observations, we next investigated the photothermal effect using an IR camera (**Figure 3b**). The temperature of AzoLCN immediately increased from 20 °C to 35 °C ($\Delta T = 15$ °C) within 1 second and this temperature is maintained (± 2 °C) during the entire UV light irradiation (17 minutes). For comparison, the temperature change (ΔT) of an LCN sample with no azobenzene was only was only 5 °C (Figure S8). Prior reports have suggested that that azobenzene can generate remarkable local effective temperature heating (up to nearly 200 °C).²⁴ We speculate that the local temperature near to azobenzene molecules is also much higher than the recorded bulk surface temperature of 35 °C, which might explain this photothermal softening.

Based on these findings, we further investigated the photothermal effect using differential scanning calorimetry (DSC) and pAzoLCN films (Figure 3c). A sharp endothermic peak at 48 °C was observed during the first heating process, suggesting the existence of crystallization peak and a physically aged state.²⁵ This physically aged (PA) state is not observed during the second heating process, but is restored if the sample is held below glass transition temperature (T_g) for approximately 2 days (Figure S9). We postulate that the PA state is thermodynamically stable state, which might arise from the differences in the glassy molecular packing between the mesogens, alkyl chains and the Diels-Alder functionalized backbone. Overall, the Tg was observed around 48 °C and LCisotropic transition (T_{ni}) of AzoLCN was observed around 100 °C. It is noteworthy that this overlaps with retro-Diels-Alder reaction temperature (100-130 °C), which might decrease cross-linking density if temperatures are held above this temperature.²⁶ Indeed, when the sample is heated above T_{ni} , we observe a lower T_g (= 40 °C) peak, which is attributed to the gradual reorganization of molecular packing and a small population of the DA adducts undergoing bond breaking through the retro-DA cycloaddition. This would lower the crosslinking density and decrease ${\it T}_{\rm g}$. Consistent with the photothermal effects observe upon UV irradiation, the PA state was not observed upon exposure to UV light for 10 minutes (Figure 3c; blue graph). When AzoLCN was exposed to UV light for 10 minutes and subsequently visible light for 1 minute, it exhibited the same thermogram as the AzoLCN solely exposed to UV light. Additionally, the global molecular packing structure

in the Azo-LCN remained unchanged before and after UV light irradiation (Figure S10).



Figure 4. (a) Storage and loss modulus and (b) temperature tracking of AzoLCN under alternative UV light irradiation on and off. (c) Storage modulus tracking of AzoLCN under a series of conditions (i: UV light on and off, ii: alternative UV and visible light on and off, iii: visible light on and off, at intervals of 30 s respectively).

To investigate the impact of photothermal and photochemical effects on the mechanical properties of the LCN, we monitored the storage (G') and loss (G'') moduli using DMA under UV and visible light irradiation (Figure 4a and 4c). Upon UV light irradiation, the storage modules decreased almost immediately from ~2 GPa to ~1 GPa. Likewise, the loss modulus increased from 60 to 300 MPa (Figure 4a). The G' and G'' were maintained with similar values under irradiation (2 minutes). When the UV light was turned off, these values quickly returned to the initial value in approximately 6 seconds. Given the slow cis-trans back isomerization in the dark (Figure 2b), we attribute this change to photothermal softening. Similar results were observed using photothermal Disperse Blue 14 dye.²⁷ Additionally, the temperature of AzoLCN films was monitored during the irradiation cycles (UV light: 2 minutes, light-off: 30 seconds). As shown in Figure 4b, the temperature increase and decrease occur with a similar response time, approximately 6 seconds for a change in G' and <5 seconds for a change in temperature. These results are all consistent with photothermal effects mainly driving the mechanical response observed. The versatility of the modulus change is demonstrated in Figure 4c, where a series of reversible changes were achieved using both UV and visible light (Figure S11, S12 and 4c). For these experiments, the DMA sample was illuminated with broadband visible light (380-700 nm) on one side and with UV light (365 nm) on the opposite side (Figure S11). For condition (i) (zero to 6.5 minutes), UV light was cycled between on and off state (30 second each). As seen by the DMA, quick and reversible oscillation in G' was observed, reaching a maximum value of ~1.25 GPa and a minimum value of ~1.1 GPa. The increase and

Journal Name

with irradiation for 30 seconds, followed by 30 seconds interval without light. While

ARTICIF



there was a notable difference in the magnitude of changes between UV and visible light, both irradiation conditions resulted in a modulus decrease. For condition (iii), a modest change in *G*' was reversibly observed under repeated visible light irradiation (30 seconds each). While the response times are comparable to those in condition (i), the change in *G*' is less due to a reduced photothermal effect caused by visible light (Figure S12). Similar results were observed under constant irradiation with UV light and visible light pulses for 30 seconds or constant visible light irradiation and UV light pulses for 30 seconds (Figure S13). This reversibility of the mechanical property was affected by cross-linking density and oscillation speed (Figure S14). Additional explanation can be found in supporting information.

Figure 5 summarizes the photothermal effect and changes in mechanical properties of the DA-based LCN. Initially, the Azocontaining LC network is in its thermodynamically most stable state (1). This material is glassy and becomes physically aged over time, which is supported by the DSC data. In this state (1), the Azo isomer is the trans-isomer, densely packed, and exhibit the highest value in storage modulus (~2.2 GPa). Upon UV light irradiation, the temperature increases and photothermal softening occurs almost immediately, and the molecular azobenzene converts to the cis-isomer. This metastable state (2) exhibits a quasi-rubbery state, with a significant decrease in storage modulus (~1.3 GPa).²⁸ Under constant irradiation, a sustained rubber state is maintained. When the light is turned off, the material returns to the glassy state within seconds, and storage modulus is restored to ~2 GPa. From this state (3), conversion of the *cis*-isomer can occur thermally over a period of 5 hours or within seconds upon irradiation with visible light

to afford state (4). Finally, state (1) can be recovered upon physical aging over an extended period of time (~2 days).

Conclusions

In summary, we have developed a new photo-responsive liquidcrystal network by integrating readily available furanfunctionalized azobenzene LC building block. Polydomain and monodomain AzoLCNs can be prepared under mild ambient conditions without the need for additives or UV light photocuring using a modular Diels–Alder cycloaddition methodology. The networks exhibit robust mechanical stiffness that can be reversibly modulated by 1 GPa by turning the UV light on and off. Through a comprehensive investigation of polydomain LCNs, it was found that photothermal softening predominantly governs the actuation performance of the LCNs. By adjusting the sequence of UV and Vis light irradiation, we achieved the ability to tune the modulus on demand. We believe this work will guide future developments of lightresponsive LCNs.

Author Contributions

CRediT: **Minwook Park** data curation (lead), formal analysis (lead), writing-original draft (lead); **Jesus Guillen Campos** methodology (equal), writing-review & editing (equal); **Fredrick Stricker** methodology (equal), writing-review & editing (equal); **Javier Read de Alaniz** formal analysis (equal), writing-review & editing (lead), supervision (lead).

Journal Name

ARTICLE

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was primarily supported by the Office of Naval Research through a MURI on Photomechanical Material Systems (Grant No. ONR N00014-18-1-2624) with partial supported from the National Science Foundation (NSF) under Award No. EFMA-1935327.

Notes and references

- 1 D. C. Hoekstra, A. P. Schenning, M. G. Debije, *Soft Matter* **2020**, *16*, 5106–5119.
- 2 S. Pearson, J. Feng, A. del Campo, *Adv. Funct. Mater.* **2021**, *31*, 2105989.
- 3 J. Cui, D. M. Drotlef, I. Larraza, J. P. Fernández-Blázquez, L. F. Boesel, C. Ohm, M. Mezger, R. Zentel, A. del Campo, Adv. Mater. 2012, 24, 4601–4604.
- 4 S. Dai, P. Ravi, K. C. Tam, Soft Matter 2008, 4, 435–449.
- 5 E. Smela, O. Inganäs, I Lundström, *Science* **1995**, *268*, 1735–1738.
- 6 Y. Li, O. Rios, J. K. Keum, J. Chen, M. R. Kessler, ACS Appl. Mater. Interfaces 2016, 8, 15750–15757.
- 7 K. M. Herbert, H. E. Fowler, J. M. McCracken, K. R. Schlafmann, J. A. Koch, T. J. White, *Nat. Rev. Mater.* **2022**, *7*, 23–38.
- 8 J. Gao, M. Tian, Y. He, H. Yi, J. Guo, *Adv. Funct. Mater.* **2022**, 32, 2107145.
- 9 T. S. Hebner, M. Podgórski, S. Mavila, T. J. White, C. N. Bowman, *Angew. Chem. Int. Ed.* **2022**, *61*, 202116522.
- 10 J. Guillen Campos, F. Stricker, K. D. Clark, M. Park, S. J. Bailey, A. J. Kuenstler, R. C. Hayward, J. Read de Alaniz, *Angew. Chem. Int. Ed.* **2023**, *62*, 202214339.
- 11 A. Ryabchun, Q. Li, F. Lancia, I. Aprahamian, N. Katsonis, *J. Am. Chem. Soc.* **2019**, *141*, 1196–1200.
- 12 H. K. Bisoyi, A. M. Urbas, Q. Li, *Adv. Opt. Mater.* **2018**, *6*, 1800458.
- 13 X. Lu, S. Guo, X. Tong, H. Xia, Y. Zhao, *Adv. Mater.* **2017**, *29*, 1606467.
- 14 M. Park, F. Stricker, J. Guillen Campos, K. D. Clark, J. Lee, Y. Kwon, M. T. Valentine, J. Read de Alaniz, ACS Macro Lett. 2022, 12, 33–39.
- 15 M. Del Pozo, J. A. Sol, A. P. Schenning, M. G. Debije, *Adv. Mater.* **2022**, *34*, 2104390.
- 16 H. Shahsavan, L. Yu, A. Jákli, B. Zhao, *Soft Matter* **2017**, *13*, 8006–8022.
- 17 T. Ohzono, H. Minamikawa, E. Koyama, Y. Norikane, *Adv. Mater. Interfaces* **2021**, *8*, 2100672.
- 18 T. Takeshima, W. Y. Liao, Y. Nagashima, K. Beppu, M. Hara, S. Nagano, T. Seki, *Macromolecules* **2015**. *48*, 6378–6384.
- 19 M. Park, D. G. Kang, W. J. Yoon, Y. J. Choi, J. Koo, S. I. Lim, S. K. Ahn, K. U. Jeong, *Chem. Mater.* **2019**, *32*, 166–172.
- 20 D. J. Mulder, T. Liang, Y. Xu, J. ter Schiphorst, L. M. Scheres, B. M. Oosterlaken, Z. Borneman, K. Nijmeijer, A. P. Schenning, J. Mater. Chem. C 2018, 6, 5018–5024.
- 21 T. Ikeda, M. Nakano, Y. Yu, O. Tsutsumi, A. Kanazawa, *Adv. Mater.* **2003**, *15*, 201–205.
- 22 B. R.; Donovan, H. E. Fowler, V. M. Matavulj, T. J. White, *Angew. Chem. Int. Ed.* **2019**, *58*, 13744–13748.
- 23 M. Kondo, Y. Yu, T. Ikeda, Angew. Chem. Int. Ed. 2006, 45, 1378–1382.

- 24 J. Vapaavuori, A. Laventure, C. G. Bazuin, O. Lebel, C. Pellerin, J. Am. Chem. Soc. **2015**, 137, 13510–13517.
- 25 K. M. Lee, H. Koerner, D. H. Wang, L. S. Tan, T. J. White, *Macromolecules* **2012**, *45*, 7527–7534.
- 26 Z. C. Jiang, Y. Y. Xiao, L. Yin, L. Han, Y. Zhao, Angew. Chem. 2020, 132, 4955–4961.
- 27 J. E. Marshall, E. M. Terentjev, Soft Matter 2013, 9, 8547–8551.
- 28 X. Huang, L. Qin, J. Wang, X. Zhang, B. Peng, Y. Yu, *Adv. Funct. Mater.* **2022**, *32*, 2208312.

Data Availability Statement

Data for this article entitled "Photo-responsive Diels–Alder Based Azobenzene-functionalized Main-chain Liquid Crystal Networks" are available at Dryad at doi:10.5061/dryad.jsxksn0jv.

The temporary link can be shared during the peer-review process:

• Reviewer URL: https://datadryad.org/stash/share/AEfx_lwF6a_saOQGQEP-ahHf-7g1TpOKoT7BZwD8Mds.