



Cite this: *Chem. Commun.*, 2020, 56, 10100

One-dimensional electronic systems: metal-chain complexes and organic conductors†

Yukihiro Yoshida * and Hiroshi Kitagawa *

One-dimensional (1D) metal-chain complexes and organic conductors show many similarities as well as striking differences in structural and electronic properties, although constituent elements and orbitals that contribute to charge transfer in these systems are quite different. In this review, we highlighted the structural and electronic properties of neutral MMX-chain complexes ($M = Pt^{2+/3+}$, $X = I^-$) and tetramethyltetrafulvalene-based cation radical salts as typical examples of each group while comparing them with each other. This review primarily aims to construct a coherent body of knowledge of 1D electronic materials that might have been separately investigated. We have proposed future directions for the exploration of new and more advanced electronic materials not only having 1D character, but also residing in the dimensional crossover regime.

Received 13th June 2020,
Accepted 17th July 2020

DOI: 10.1039/d0cc04124f

rsc.li/chemcomm

1. Introduction

Intuitively, one-dimensional (1D) electronic systems seem to be extremely simple, but they have long been a central subject in solid-state sciences because of their totally different chemistry and

physics from those of their higher-dimensional counterparts.^{1–3} This is primarily because these systems offer a rich variety of exotic quantum phenomena such as charge/spin-soliton, polaron/bipolaron, bond-order wave, charge/spin-density wave, and Tomonaga–Luttinger liquid (non-Fermi liquid). In principal, 1D structures allow the formation of various kinds of ordered phases, which sometimes break the translational symmetry of lattice, charge, or spin degrees of freedom. Although quantum fluctuations, which play a prominent role in 1D systems, suppress long-range ordering at finite temperatures, transverse interactions,

Division of Chemistry, Graduate School of Science, Kyoto University, Kitashirakawa-Oiwakecho, Sakyo-ku, Kyoto 606-8502, Japan.

E-mail: yoshiday@ssc.kuchem.kyoto-u.ac.jp, kitagawa@kuchem.kyoto-u.ac.jp

† This feature article is dedicated to the memory of the late Prof. Peter Day, who passed away on May 19, 2020.



Yukihiro Yoshida

Yukihiro Yoshida earned his PhD degree in 2001 from Kyoto University under the direction of Prof. Gunzi Saito. After postdoctoral research at Kyoto University and Université de Rennes 1, he moved to Meijo University as an assistant professor. In 2017, he moved back to Kyoto University and is currently an associate professor at the same university. His research interests center on the design, synthesis, and

characterization of new materials, specifically in the field of organic (super)conductors, supramolecular rotors, ionic liquids, and proton-coupled electronic metal complexes including metal–organic frameworks. He has authored or co-authored more than 110 peer-reviewed publications.



Hiroshi Kitagawa

Hiroshi Kitagawa finished his PhD at Kyoto University in 1991 and, after working as Assistant Professor at IMS and JAIST, he was appointed as Associate Professor at the University of Tsukuba in 2000. He became Professor of Chemistry at Kyushu University in 2003 and was appointed to Kyoto University as Professor in 2009. He is also engaged at Japan Science & Technology Agency as Director of a network-type

research institution “Science and Creation of Innovative Catalysts”. He has published more than 400 original papers on research dealing with solid-state chemistry, coordination chemistry, nano-science, low-dimensional electron system, and molecule-based conductor.

which are usually present in actual systems, lead to an ordered state. Because anisotropic p and d orbitals of components as well as directional interactions, such as hydrogen and π - π interactions, between components are dominant factors in determining the anisotropic packing motif in metal-complex and molecular solids, it is apparent that these compounds are a promising platform for investigating low-dimensional electronic systems showing phase transitions as a function of temperature and pressure. In fact, the first spin-Peierls transition, which has been predicted to occur in certain compounds composed of spin-1/2 antiferromagnetic chains, was found in a molecular solid, TTF·Cu(tfd)₂ (TTF: tetrathiafulvalene, tfd: 1,2-bis(trifluoromethyl) ethene-1,2-dithiolate), in 1975,⁴ more than a decade prior to the first discovery of an inorganic spin-Peierls compound, CuGeO₃.⁵

In 1D metal complexes, an axial overlapping of the d_{2z} orbitals of group 10 metals, such as Ni, Pd, and Pt, is indispensable for securing conduction pathways. A representative example is prototype Krogmann's salt, K₂Pt(CN)₄Br_{0.3}·3H₂O (so-called KCP(Br)),⁶ which involves a 1D chain comprising square-planar PtCl₄ subunits arranged through Pt··Pt interactions. A marked overlap between the 5d_{2z} orbitals of Pt results in band formation, and 0.3 holes per Pt (*i.e.*, Pt^{2.3+}) serve as carriers in the solid with a partially filled band. In 1974, it was found that this complex behaves as metallic above approximately 250 K,⁷ below which the conductivity becomes thermally activated due to Peierls instability arising from the electron-phonon interactions in the 1D system. On the other hand, in molecular solids, such as that comprising electron-donating TTF and electron-accepting 7,7,8,8-tetracyano-*p*-quinodimethane (TCNQ) discovered independently by two groups in 1973,^{8,9} each component constructs a π -stacking column through π -orbital overlap, and the inter-column charge transfer (CT) interactions between TTF and TCNQ cause the generation of carriers within both columns. TTF·TCNQ with a partial CT (0.59) is currently regarded as the first molecular or synthetic metal and also shows complicated phase transitions associated with the Peierls instabilities at low temperatures.¹⁰ In both systems, itineracy depends largely on the balance between transfer energy ($t \approx W/4$ in 1D electronic systems; W : bandwidth), on- or inter-site Coulomb repulsion energy (U or V , respectively), and site-diagonal (or intra-site) and site-off-diagonal (or inter-site) electron-phonon coupling energy (S).¹¹ This review covers recent developments in each system after providing a general introduction to some materials comprehensively discussed in earlier review articles^{1-3,12-14} and compares the electronic behavior of 1D metal-chain complexes and organic conductors to provide a deeper understanding of how molecular and crystal structures affect the electronic properties. We mainly focused on neutral MMX-chain complexes, in which binuclear metal subunits (MM) are connected by a halogen (X), and cation radical salts, (TMTTF)₂X (TMTTF: tetramethyl-TTF, X: monoanion), in which cationic TMTTF^{0.5+} molecules form a weakly dimerized π -stacking column, as examples of 1D metal-complex and molecular solids, respectively (Fig. 1), because both systems are increasingly recognized to be full of examples of the relevance of electronic behavior to lattice, charge, and spin degrees of freedom.

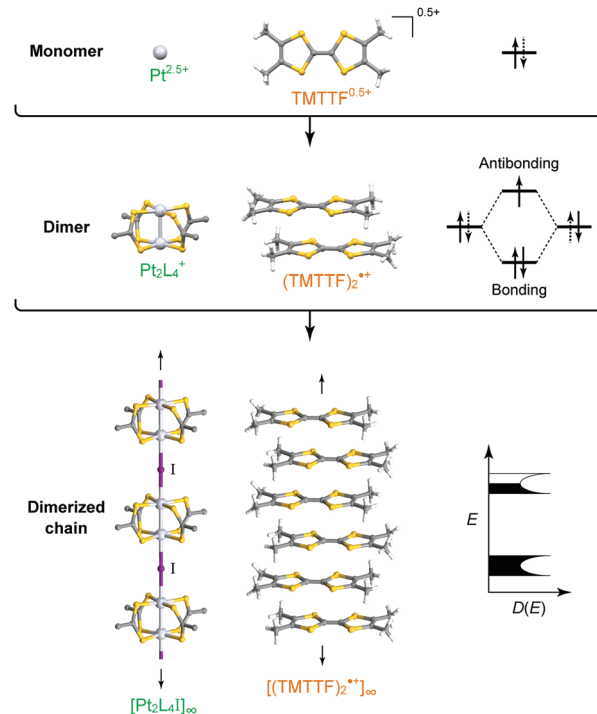


Fig. 1 Comparison of the monomeric, dimeric, and assembled structures of (left) Pt₂L₄ (L = RCS₂⁻) and (right) (TMTTF)₂X (X: monoanion) together with their electronic states ($D(E)$: density of states).

2. MMX-chain complex family

The first halogen-bridged MX-chain complex, in which metal (M) and halogen (X) species are arranged in an alternating manner through the antibonding combination of M(d_{2z}) and X(p_z) orbitals, was synthesized by Wolfram more than a century ago.¹⁵ In [Pt(EtNH₂)₄Cl]Cl₂·2H₂O, the so-called Wolfram's red salt, mixed valence Pt²⁺(d⁸) and Pt⁴⁺(d⁶) ions are arranged alternately primarily due to the significant site-diagonal electron-phonon interactions ($U < S$; negative- U effect;¹⁶ Fig. 2(a)).¹⁷ Such a mixed valence state is accompanied by opening of the energy gap (Δ) resulting in nonmagnetic insulating behavior. Replacing Pt with a lighter homologous element, *i.e.* Ni, provides an averaged valence state of +3, in which the unpaired electron is localized due to the on-site Coulomb interactions ($U > W$; Fig. 2(b)) while maintaining a spin degree of freedom. Regardless of their origin, all the MX-chain complexes reported to date show insulating behavior. Instead, the efforts in synthesizing chain complexes based on binuclear metal subunits (MMs) have been devoted to exploring diverse electronic states mainly because of the reduced electron-electron correlations. In MMX-chain complexes, a halogen (X) serves as a bridge between MMs, each of which is assembled by four bidentate ligands in a paddle-wheel fashion. In a PtPt subunit, the Pt²⁺(d⁸) ions adopt a planar four-coordinate geometry with no Pt²⁺-Pt²⁺ bond within the subunit, whereas Pt³⁺(d⁷) ions adopt an octahedral coordination geometry to form an intra-subunit Pt³⁺-Pt³⁺ bond.^{12,18} Therefore, each Pt³⁺ ion can form a covalent bond with a bridging iodine, *i.e.*, the iodine is displaced from the central position between two

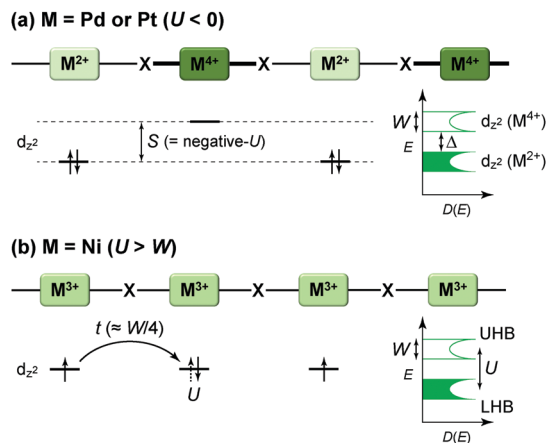


Fig. 2 Schematic electronic structures of (a) mixed-valence ($M = \text{Pd or Pt}$) and (b) Mott–Hubbard ($M = \text{Ni}$) insulating states in MX-chain complexes (UHB: upper Hubbard band, LHB: lower Hubbard band) related to physical parameters such as on-site Coulomb repulsion energy (U), bandwidth (W) or transfer energy ($t \approx W/4$ in 1D electronic systems), and electron–phonon coupling energy (S).^{18,25} S is proportional to a spatial distortion from the equilibrium structure with a coefficient β (*i.e.*, site-diagonal electron–phonon coupling constant), whereas t becomes smaller in proportional to the spatial distortion with a coefficient α (*i.e.*, site-off-diagonal electron–phonon coupling constant).^{27,28,30}

adjacent subunits depending on the valence state and approaches closer to Pt^{3+} . As shown in Fig. 3, four types of valence states appear in the MMX chain depending primarily on the valence states of Pt ions: (a) averaged-valence (AV) state: uniform chain composed of $\text{Pt}^{2.5+} - \text{Pt}^{2.5+}$ homovalent dimers, showing metallic ($U < W$) or Mott insulating ($U > W$) behavior; (b) charge-density wave (CDW) state: alternating chain comprising $\text{Pt}^{2+} - \text{Pt}^{2+}$ and $\text{Pt}^{3+} - \text{Pt}^{3+}$ homovalent dimers; (c) charge-polarization (CP) state: uniform chain composed of $\text{Pt}^{2+} - \text{Pt}^{3+}$ heterovalent dimers with a parallel orientation; (d) alternate charge-polarization (ACP) state: alternating chain comprising $\text{Pt}^{2+} - \text{Pt}^{3+}$ heterovalent dimers with an antiparallel orientation. It should be noted that the valence states Pt^{2+} and Pt^{3+} are more accurately represented by $\text{Pt}^{(2+\delta)+}$ and $\text{Pt}^{(3-\delta)+}$, respectively, where the δ value ($0 < \delta < 0.5$) significantly depends on the degree of orbital hybridization. The CP state retains the periodicity of the original AV state, whereas the CDW and ACP states described as Peierls and spin-Peierls states, respectively, deform their chain structures to produce a doubled periodicity as $-\text{Pt}-\text{Pt}-\text{I}-\text{Pt}-\text{Pt}-\text{I}-$. The spin degree of freedom, which is maintained in the AV and CP states, is essentially frozen in the CDW and ACP states in the extreme case ($\delta = 0$) due to the opening of the spin gap. The CP state with a permanent dipole moment has the potential to show ferroelectric behavior at a glance. In actual solids, these chains interact only weakly with each other through van der Waals forces, thereby identifying with a 1D electronic system.

The first MMX-chain complex, $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$, was synthesized by Bellitto *et al.* in 1983.¹⁹ In this solid, four bidentate dithiocarboxylate ligands (RCS_2^- ; R: alkyl group) are attached to two Pt ions in a square planar coordination for each Pt, and the paddle-wheel-shaped building units are assembled in a linear fashion by the bridging iodines. It is apparent that the ligands are responsible

for the spatial isolation of the chains from each other, producing a 1D electronic state. To date, several MMX-chain complexes have been synthesized using RCS_2^- ligands with different alkyl chain lengths. Another extensively studied line of research is $\text{A}_4[\text{Pt}_2(\text{pop})_4\text{I}] \cdot n\text{H}_2\text{O}$ (A: monocation, pop: $\text{P}_2\text{H}_2\text{O}_5^{2-}$), which is already the subject of some reviews^{12,20,21} and therefore will not be dealt with here. Typically, $\text{Pt}_2(\text{RCS}_2)_4\text{I}$ can be prepared by the thermal treatment of a Pt^{2+} dimer complex, $\text{Pt}_2(\text{RCS}_2)_4$, and a Pt^{3+} dimer complex, $\text{Pt}_2(\text{RCS}_2)_4\text{I}_2$, in toluene or toluene/hexane solution, where the acidic ligands RCS_2H are commonly synthesized by the reaction of appropriate alkyl Grignard reagents with carbon disulfide in tetrahydrofuran/diethyl ether solution at low temperatures.²² It is well-known that $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ with R = Me can also be prepared by the iodine oxidation of $\text{Pt}_2(\text{MeCS}_2)_4$ (Pt/I molar ratio of 2 : 1) in toluene solution.¹⁹

$\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ is not only the second example, after $\text{KCP}(\text{Br})$, of the transition-metal complex showing metallic behavior under ambient pressure but also the first example of metallic neutral chain complex. Its metallic behavior could be observed only when a high-purity single crystal was used for the transport measurements.^{23,24} At room temperature, the complex belongs to the monoclinic space group $C2/c$, and each infinite chain running along the b axis includes one $-\text{Pt}-\text{Pt}-\text{I}-$ unit in a unit cell (Fig. 4).^{18,19,25} Along the side-by-side direction ($\parallel c$), there are heteroatomic $\text{S} \cdots \text{S}$ contacts with a distance (*ca.* 3.8 Å) slightly longer than the sum of van der Waals radii (3.60 Å). Provided that the intra-subunit transfer energy (t_{MM}) is comparable to the inter-subunit transfer energy (t_{MXM}), the d_{2z} band is originally three-quarter-filled, $(5d_{2z})^2(5d_{2z})^1$. However, the possible condition $t_{\text{MM}} > t_{\text{MXM}}$ results in the splitting of d_{2z} into bonding d_σ and antibonding d_σ^* due to dimerization, latter of which is in the half-filled state (Fig. 3(a)). It is noteworthy that contrary to the d_σ orbital, the d_σ^* orbital possibly expands towards the outside of the subunit, which facilitates electron transfer between the subunits (*i.e.*, t_{MXM}). The conductivity of $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ along the chain direction is *ca.* 13 S cm^{-1} at room temperature and shows metallic behavior above approximately 300 K (Fig. 5(a)). A temperature-independent thermoelectric power above room temperature strongly suggests that the complex has a half-filled metallic state in this temperature range. The observation of a single C=S stretching band in the infrared (IR) spectra is firm evidence for the formation of the AV state. The insulating phase caused by a metal–insulator transition can be readily assigned to the CP state (Fig. 3(c)) because of its significant magnetic susceptibility (Fig. 5(b)) as well as the splitting of the C=S stretching mode (trapped-valence states Pt^{2+} and Pt^{3+}). The ^{129}I Mössbauer spectra showing two different iodine sites below 80 K indicate that the electronic state formed below 80 K is the ACP state (Fig. 3(d)), which is consistent with that predicted by theoretical studies.^{26–28} Because it is reasonable to expect that the Pt–I bonds are more compressible than the Pt–Pt bonds in MMX-chains, the t_{MXM} approaches the t_{MM} as the temperature is decreased. Thus, the degree of dimerization in the MMX-chain (*i.e.*, difference between t_{MM} and t_{MXM}) weakens upon cooling, which could have a favorable effect on the stability of a charge-ordered (CO) state as compared to that

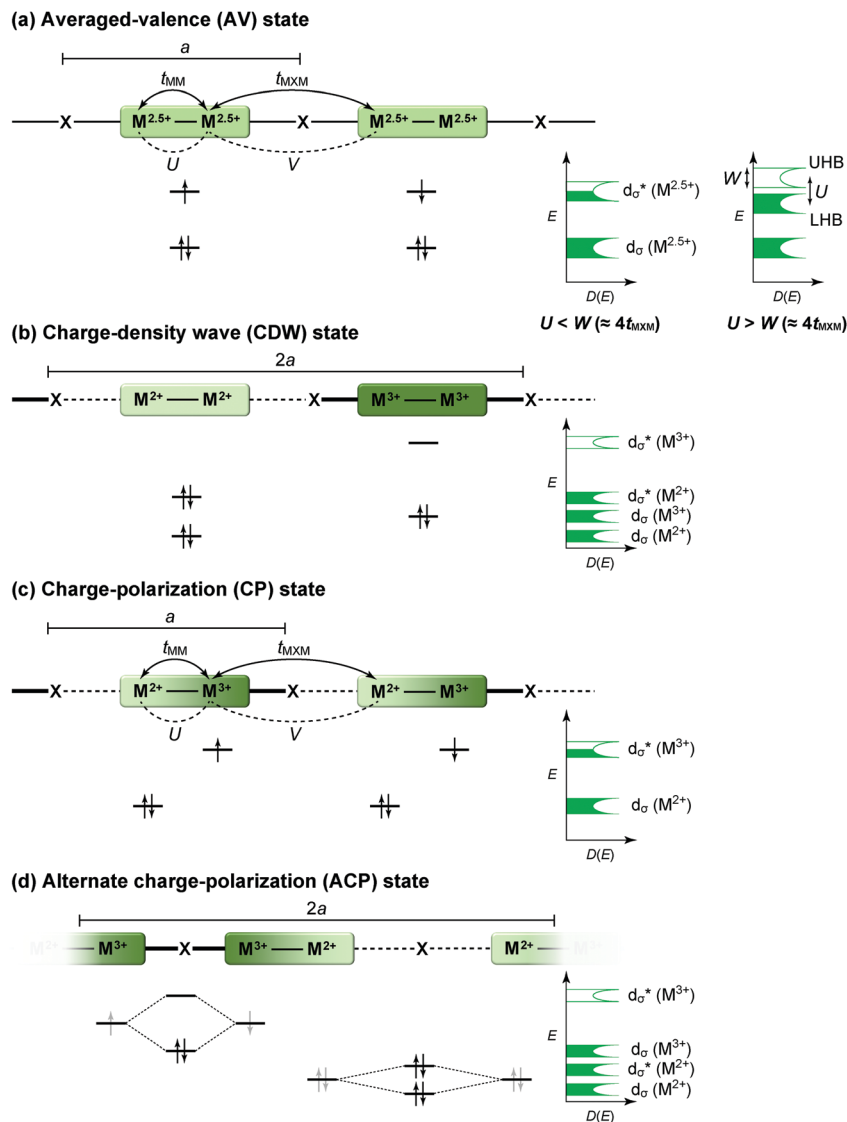


Fig. 3 Schematic electronic structures of the typical states in MMX-chain complexes: (a) averaged-valence (AV) state, (b) charge-density wave (CDW) state, (c) charge-polarization (CP) state, and (d) alternate charge-polarization (ACP) state (a : periodicity of an $-M-M-I-$ unit, t_{MM} : transfer energy within an MM subunit, t_{MXM} : transfer energy between the subunits, V : inter-site Coulomb repulsion energy).¹⁸

on a dimer-Mott state.²⁹ Although the increase in V owing to lattice compression must stabilize the CP state, the emergence of the ACP state at lower temperatures might be associated with the magnetic energy gain due to the opening of the spin gap. In this regard, site-off-diagonal electron-phonon interactions that modulate the transfer energies, especially t_{MXM} in this system, must be beneficial for realizing an ACP state with redistributed tetramers.^{27,30}

$Pt_2(EtCS_2)_4I$ (space group: $C2/c$), in which the methyl groups of ligands in $Pt_2(MeCS_2)_4I$ are replaced with longer ethyl groups, was synthesized by Mitsumi *et al.* via the reaction of $Pt_2(EtCS_2)_4$ with $Pt_2(EtCS_2)_4I_2$ in toluene.³¹ The complex shows a room-temperature conductivity of $5\text{--}30\text{ S cm}^{-1}$, comparable to that of $Pt_2(MeCS_2)_4I$, and a conductivity maximum at 205 K. Because the X-ray diffuse scattering lines corresponding to the 2-fold periodicity along the chain direction start to grow below

160 K, it is likely that the conductivity change at 205 K occurs due to a phase transition from the metallic AV to the Mott-insulating AV or charge-ordered CP state (Raman spectra suggest the coexistence of these states³²). X-ray structural refinement revealed that the low-temperature phase ($<160\text{ K}$) with a 2-fold $-Pt-Pt-I-$ unit repetition is identical to the ACP state, *i.e.*, $-Pt^{2+}-Pt^{3+}-I-Pt^{3+}-Pt^{2+}-I-$ (space group: $P\bar{1}$).³³ The emergence of the ACP state was confirmed by ^{129}I Mössbauer spectroscopy, which estimates the charge of iodine bridging the Pt^{3+} ions to be -0.4 and that of iodine bridging the Pt^{2+} ions to be -0.3 .³⁴ This structural feature is seemingly inconsistent with the magnetic susceptibility data, which indicate the presence of the spin degree of freedom even below 160 K (Fig. 6). This discrepancy can probably be explained by the difference between the degrees of lattice deformation within the chain. In fact, the difference between the $Pt^{2+}-I$ and $Pt^{3+}-I$ bond

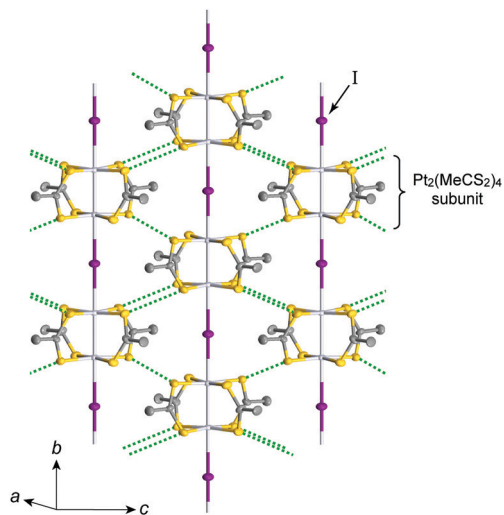


Fig. 4 Crystal structure of $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ at room temperature,^{18,19} where green dotted lines show heteroatomic S...S contacts between adjacent chains. Hydrogen atoms are omitted for clarity.

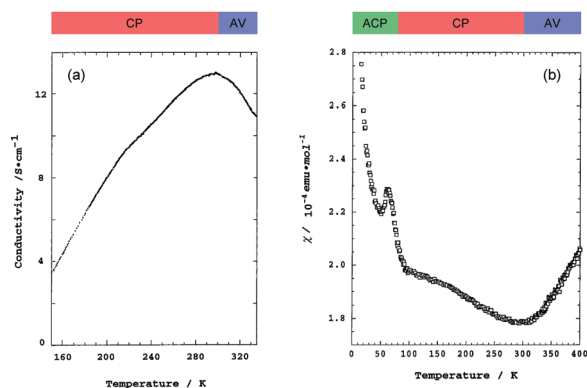


Fig. 5 Temperature dependence of (a) conductivity and (b) magnetic susceptibility (χ) of $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ (AV: averaged-valence state, CP: charge-polarization state, ACP: alternate charge-polarization state). Adapted with permission from ref. 18. Copyright 1999 American Chemical Society.

lengths (0.02–0.03 Å) is significantly smaller than that in the case of $\text{Pt}_2(n\text{-BuCS}_2)_4\text{I}$ (*ca.* 0.07 Å),³³ which shows a rapid drop in magnetic susceptibility associated with the first-order transition, as expected from the ideal ACP state (Fig. 6). As above-mentioned, the observation of ferroelectric behavior in the CP state is of fascinating interest. The distorted ESR spectra shown in Fig. 7 is indicative of the significantly reduced cavity quality factor, which is inversely proportional to dielectric loss, $\tan \delta$, possibly due to the emergence of ferroelectricity.³⁵ The temperature range of the distorted spectra (130–140 K) is below the transition temperature at which the superstructure associated with the ACP state starts to grow (160 K) because of the gradual growth of the ACP state.

Pressure has been applied to not only search the unexplored electronic states, but also control the electronic properties. As shown in Fig. 8, the room-temperature resistivity of $\text{Pt}_2(\text{EtCS}_2)_4\text{I}$ is steadily decreased upon applying hydrostatic pressure of up

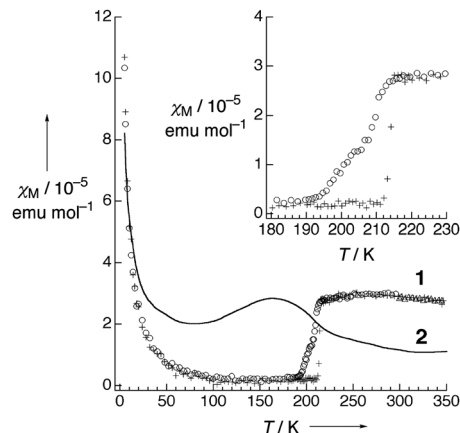


Fig. 6 Temperature dependence of magnetic susceptibility (χ_M) of $\text{Pt}_2(\text{EtCS}_2)_4\text{I}$ (**2**) and $\text{Pt}_2(n\text{-BuCS}_2)_4\text{I}$ (**1**; \circ : 1st cooling, $+$: 1st heating, Δ : 2nd cooling). Inset shows the phase transition behavior at around 210 K of **1**. Adapted with permission from ref. 33. Copyright 2002 Wiley-VCH.

to 3 GPa principally due to the increase in bandwidth caused by lattice compression.^{32,36} The conductivity maximum observed at 205 K under ambient pressure rapidly disappears when moderate pressure is applied and clearly revives under higher pressures (*ca.* 70 K at 2.2 GPa). Of particular importance is that the metallic behavior disappears again above this pressure and the conductivity shows thermally activated behavior. The insulating phase is readily assigned to the CP state based on the crystallographic and Raman studies as a function of pressure. Theoretical calculations predicted that the application of pressure leads to the stabilization of the reentrant CP state.²⁷ The reason must be the same as above-mentioned for low temperatures: the increased V relative to t_{MM} due to the shortened inter-subunit distance under pressure. Similar behavior of the stabilization of the CP state by applying pressure was observed for $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ (Fig. 9).³⁷ It is interesting to note that the proposed T - P phase diagrams for $\text{Pt}_2(\text{RCS}_2)_4\text{I}$ are quite different from that of $(\text{TMTTF})_2\text{X}$ (see below), in which the

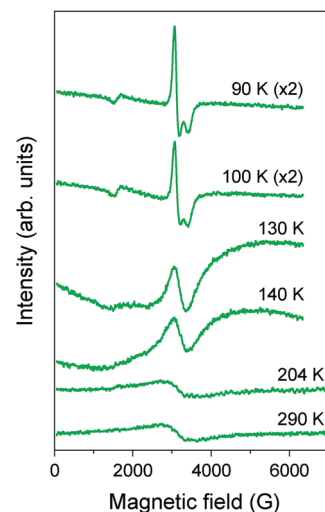


Fig. 7 Temperature dependence of ESR spectra of polycrystalline $\text{Pt}_2(\text{EtCS}_2)_4\text{I}$.³⁵

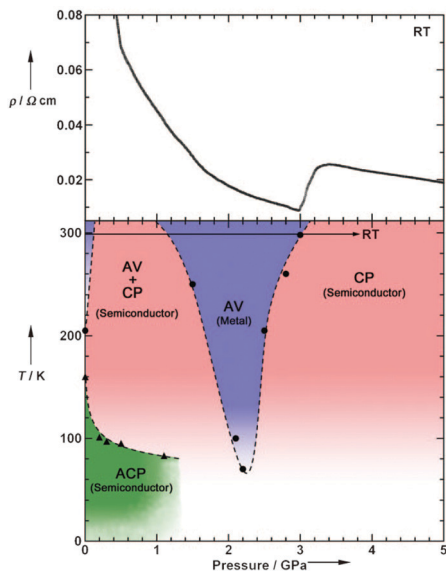


Fig. 8 (Upper) Pressure dependence of room-temperature resistivity (ρ) and (lower) electronic T - P phase diagram of $\text{Pt}_2(\text{EtCS}_2)_4\text{I}$, where a panel presented in the original paper³² was omitted. Adapted with permission from ref. 32. Copyright 2009 Wiley-VCH.

pressure transforms the salts from 1D (Tomonaga-Luttinger liquid) to 3D (Fermi liquid) metals to stabilize the AV state.^{14,38,39} Although the factors that determine the electronic state under pressure remain unclear, the persistent 1D character of $\text{Pt}_2(\text{RCS}_2)_4\text{I}$ due to the subtle interchain interactions may be a part of the reason.

A peculiar valence state was found for $\text{Pt}_2(n\text{-PenCS}_2)_4\text{I}$ with longer n -pentyl groups. It has a relatively low room-temperature conductivity (0.84 S cm^{-1} ⁴⁰) but retains metallic behavior down to 255–270 K⁴¹ or 235 K.⁴² On further cooling, the complex shows an abrupt drop in conductivity at *ca.* 200 K⁴¹ or 205 K,⁴² which is close to a first-order transition temperature (207.4 K) observed for the heat capacity profile.⁴⁰ At the same temperature, magnetic susceptibility rapidly vanishes with decreasing temperature possibly because of the transition from the paramagnetic (AV) to the nonmagnetic (CDW or ACP) state. To date, the valence state at low temperatures remains highly controversial

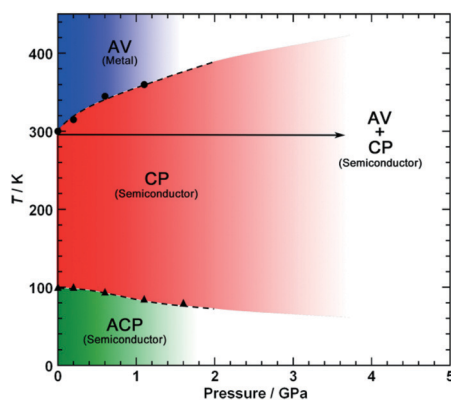


Fig. 9 Electronic T - P phase diagram of $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$. Adapted with permission from ref. 37. Copyright 2016 Wiley-VCH.

due to the complicated X-ray diffuse scattering patterns possibly arising from valence fluctuations.^{41,42} However, it is most likely that the low-temperature phase involves a 3-fold $-\text{Pt}-\text{Pt}-\text{I}-$ unit repetition as $-\text{Pt}^{3+}-\text{Pt}^{2+}-\text{I}-\text{Pt}^{2+}-\text{Pt}^{3+}-\text{I}-\text{Pt}^{2.5+}-\text{Pt}^{2.5+}-\text{I}-$, comprising alternating ACP and AV states. A similar sign of the 3-fold periodicity along the chain direction was found for $\text{Pt}_2(n\text{-PrCS}_2)_4\text{I}$ ⁴³ and $\text{Pt}_2(n\text{-BuCS}_2)_4\text{I}$.³³ Such an alternating chain composed of homovalent and heterovalent dimers has not been found in organic conductors, although there are 2D conducting layers comprising homovalent dimers with different charges (*e.g.*, $\kappa\text{-(BEDT-TTF)}_4(\text{Et}_4\text{N})[\text{M}(\text{CN})_6] \cdot n\text{H}_2\text{O}$; BEDT-TTF: bis(ethylenedithio)-TTF, $\text{M} = \text{Fe}^{3+}$ and Co^{3+} ; $n = 2$ ⁴⁴ and 3 ⁴⁵) and uniform and alternately charged columns (*e.g.*, $\alpha\text{-(BEDT-TTF)}_2\text{I}_3$ ⁴⁶). The phase diagram of $\text{Pt}_2(\text{RCS}_2)_4\text{I}$ with different alkyl chain lengths is illustrated in Fig. 10.

3. (TMTTF)₂X family

In the research of organic conductors, most of the interest in the chemistry field is focused on the synthesis of superconductors, which require increased electronic dimensionality to prevent the Peierls transition caused by the electron-phonon interactions in 1D systems.^{3,14} A good example is the BEDT-TTF cation radical salts mainly developed by Saito group^{52,53} and Williams and Schlueter group,^{54,55} which have provided a rich variety of 2D electronic materials with properties ranging from superconductors to quantum spin liquids. However, the collaborative efforts of chemists and physicists over the past

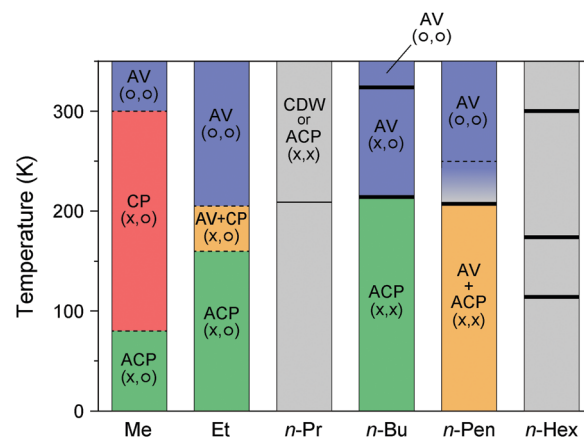


Fig. 10 Temperature dependence of the electronic states of $\text{Pt}_2(\text{RCS}_2)_4\text{I}$ with different alkyl chain (R) lengths, where the superlattice structure is not considered.^{12,18,31–33,40–43,47–51} Gray areas show the temperature range whose electronic state is unknown, and symbols in parentheses indicate the charge (left) and spin (right) degrees of freedom. Thick and thin vertical lines indicate the first- and higher-order transition temperatures, respectively, observed by heat capacity measurements.^{40,48,49,51} For $\text{R} = \text{Me}$, no heat-capacity anomaly was observed at around 300 and 80 K,⁴⁷ at which phase transitions occur between the AV and CP states and between the CP and ACP states, respectively.¹⁸ For $\text{R} = \text{Et}$, although numerical data of transition temperatures (*ca.* 180 and 230 K) observed by the heat capacity measurements are presented in ref. 12 and 50, the source and full extent of the anomalies are not shown. For $\text{R} = n\text{-Pen}$, no heat-capacity anomaly was observed at around 250 K,⁴⁰ at which a conductivity maximum was observed during the resistivity measurements (shown by a dotted line).^{41,42}

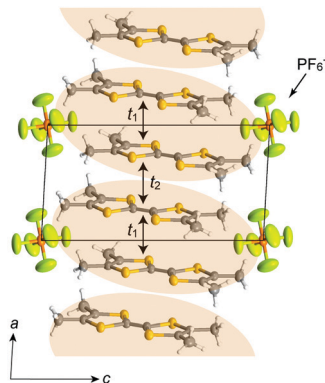


Fig. 14 Crystal structure of $(\text{TMTTF})_2\text{PF}_6$ at room temperature viewed along the b axis, where the TMTTF dimers are indicated by orange ellipses. t_1 and t_2 are intra- and inter-dimer transfer energies, respectively.

In $(\text{TMTTF})_2\text{X}$ (space group: $P\bar{1}$), the TMTTF molecules with an average charge of $+0.5$ weakly dimerize in a face-to-face fashion, and the TMTTF dimers with an $S = 1/2$ spin assemble in a stacking manner *via* π - π interactions along the a axis (Fig. 14). The zig-zag arrangement is attributed to the steric hindrance of counter anions. For $(\text{TMTTF})_2\text{PF}_6$, the intradimer (t_1) and interdimer (t_2) transfer energies at room temperature were estimated to be 230.6 and 170.2 meV, respectively, whereas the inter-column transfer energies are small but substantial ($|t| < 25.2$ meV). Theoretical calculation using the crystallographic data revealed that the bands are more dispersive along the stacking direction ($\parallel a$) than along the side-by-side direction ($\parallel b$) and the Fermi surface is composed of a wavy 1D hole-like surface along the a axis (Fig. 15). Accordingly, the room-temperature conductivity is the highest along the a axis, and anisotropy is approximately $\sigma_a : \sigma_b : \sigma_c = 2000 : 50 : 1$.⁶⁸ The formally three-quarter-filled HOMO band substantially splits into lower (bonding π) and upper (antibonding π^*) bands associated with the dimerized structure. Provided that t_2 is significantly higher than U within a dimer, *i.e.*, $U < 4t_2$, the upper band is regarded as a half-filled band with $W = 0.45$ eV. It is noteworthy that this value is significantly smaller than 1.70 eV of an isolated $\text{Pt}_2(\text{MeCS}_2)_4\text{I}$ chain calculated using the first principles method within the density functional theory (DFT) under the assumption of the AV state.⁶⁹ So far, five kinds

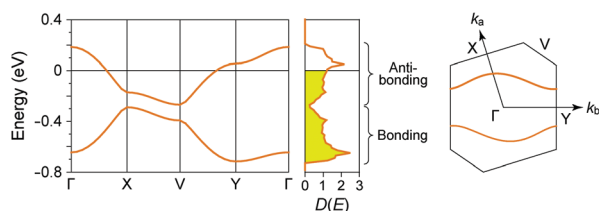


Fig. 15 Electronic band dispersions, densities of states ($D(E)$ in states per electronvolt per spin), and Fermi surfaces of $(\text{TMTTF})_2\text{PF}_6$, calculated on the basis of the extended Hückel method with the tight-binding approximation using crystallographic data at room temperature. The energy is provided relative to the Fermi energy. Note that the electron–electron correlation is not considered in this calculation.

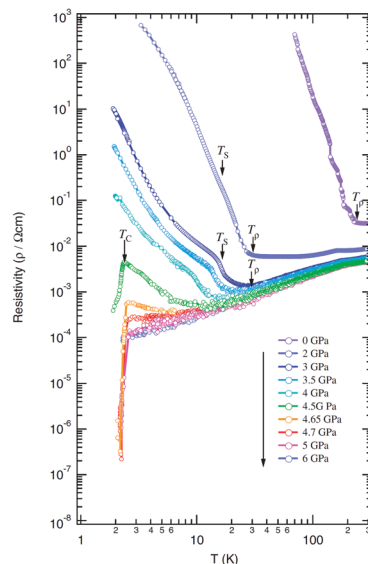


Fig. 16 Temperature dependence of resistivity of $(\text{TMTTF})_2\text{AsF}_6$ under various pressures, where T_p , T_s , and T_c are temperatures at minimum resistivity, spin-Peierls transition, and superconducting transition, respectively. Adapted with permission from ref. 72. Copyright 2007 Physical Society of Japan.

of superconductors have been found for $(\text{TMTTF})_2\text{X}$ ($\text{X} = \text{PF}_6^-$,^{70,71} AsF_6^- (Fig. 16),⁷² SbF_6^- ,⁷³ BF_4^- ,⁷⁴ and Br^- ⁷⁵) by applying pressure above a few gigapascals (Table 1).

Herein, we have mainly described the structural and electronic properties of $(\text{TMTTF})_2\text{PF}_6$ as an example of $(\text{TMTTF})_2\text{X}$ formed with centrosymmetric octahedral anions because other relevant salts have qualitatively similar properties to those of $(\text{TMTTF})_2\text{PF}_6$. $(\text{TMTTF})_2\text{PF}_6$ shows metallic behavior under ambient conditions as a result of the three-quarter-filled conduction band. The conductivity is *ca.* 40 S cm^{-1} at room temperature and exhibits a maximum at around 245 K possibly due to the electron–electron correlations (AV state; Fig. 17(a)).^{62,76} The fact that the magnetic susceptibility shows no anomaly at around the temperature is indicative of the occurrence of separation between the spin and charge degrees of freedom in 1D electronic systems.^{77,78}

On further cooling, the salt undergoes a phase transition to a $4k_F$ -CDW CO state of Wigner crystal type (= CP; Fig. 17(b)) at 67 K, as observed *via* ^{13}C NMR,^{79,80} dielectric,⁸¹ and optical spectroscopies.^{66,82} Band structure calculations of $(\text{TMTTF})_2\text{X}$ ($\text{X} = \text{PF}_6^-$, AsF_6^- , and SbF_6^-) demonstrated that the phase transition to the $4k_F$ -CDW CO state occurs below the temperature at which the degree of dimerization, $2(t_1 - t_2)/(t_1 + t_2)$, falls below 0.25.⁸³ The dimer-Mott state is stable at temperatures above which the degree of dimerization is higher than 0.25 due to significant dimerization, whereas weakened dimerization below this temperature bring it into the $4k_F$ -CDW CO state. The same could be speculated for $\text{Pt}_2(\text{RCS}_2)_4\text{I}$; however, the numerical values of t_{MM} ($= t_1$) and t_{MXM} ($= t_2$) have not been estimated to date.

Based on the IR spectroscopic study, the degree of charge disproportionation, 2δ , which is defined as the difference between

Table 1 Properties of typical (TMTTF)₂X^a

X	V _a (Å ³)	T _c (K)	T _{CO} (K)	T _{AO} (K)	GS	P _c (GPa)	T _c (K)	Ref. ^b
PF ₆ ⁻	109	245	67 [CP]		SP [15 K]	5.2	1.8	70 and 71
AsF ₆ ⁻	110	105	102 [CP]		SP [13 K]	4.5	2.4	72
SbF ₆ ⁻	121	154	157 [CP]		AF [8 K]	5.4	2.8	73
TaF ₆ ⁻	128	— ^c	177 [CP]		AF [9 K]	— ^d	— ^d	
BF ₄ ⁻	73	190	84 [ACP]	41	SS [41 K]	3.4	1.4	74
ClO ₄ ⁻	82	230	— ^d	73	AF [12 K]	— ^d	— ^d	
ReO ₄ ⁻	86	230	230 [ACP]	157	SS [157 K]	— ^d	— ^d	
SCN ⁻	71	240	160 [CP]	160	AF [7 K]	— ^d	— ^d	
Br ⁻	56	100	20 [CP]		AF [15 K]	2.6	0.8	75

^a V_a: anion volume (H. D. B. Jenkins, H. K. Roobottom, J. Passmor and L. Glasser, *Inorg. Chem.*, 1999, **38**, 3609); T_c: temperature at maximum conductivity; T_{CO}: charge-ordering temperature (CP: charge-polarization, ACP: alternate charge-polarization); T_{AO}: anion-ordering temperature; GS: ground state (SP: spin-Peierls, AF: antiferromagnetic ordering, SS: anion-order-induced spin-singlet. The numbers in brackets represent the transition temperature); P_c: critical pressure; T_c: critical temperature. ^b References for superconductivity. ^c Not measured. ^d Not observed.

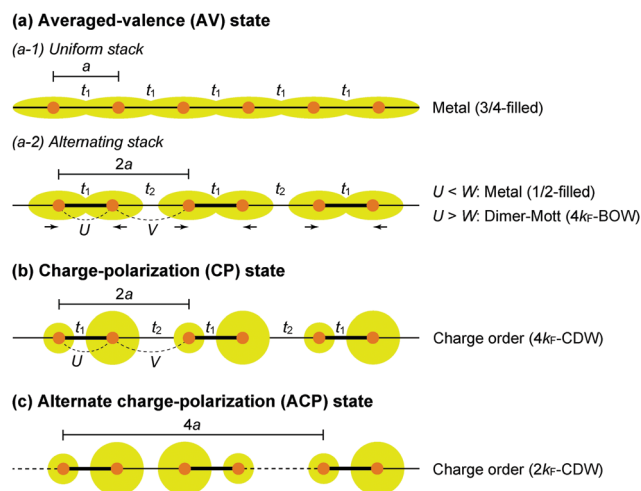


Fig. 17 Schematic electronic structures of typical states in (TMTTF)₂X, in which the average charge of TMTTF molecules is formally +0.5; (a) averaged-valence (AV) state in the cases of (a-1) uniform stack and (a-2) alternating stack, (b) charge-polarization (CP) state, and (c) alternate charge-polarization (ACP) state (*a*: periodicity for the uniformly arranged AV state). Orange spheres indicate a TMTTF molecule, whereas the size of the yellow sphere/ellipses corresponds to the amount of charge on each TMTTF molecule. Note that the state in (a-1) is not available for MMX-chain complexes due to the presence of bridging iodines. In (b) and (c), if the 4k_F and 2k_F modulations of electron density are accompanied by a lattice modulation (*i.e.*, bond-order wave (BOW)), they can be described as “4k_F-CDW + 4k_F-BOW” and “2k_F-CDW + 2k_F-BOW” states, respectively.

the charges of charge-rich and charge-poor TMTTF molecules, increases with the decrease in temperature below 67 K and eventually reaches 0.15 at 10 K (Fig. 18).^{66,84,85} The 2δ value is apparently smaller than those of (TMTTF)₂AsF₆ (0.21),⁶⁶ (TMTTF)₂SbF₆ (0.29),⁶⁶ and (TMTTF)₂TaF₆ (0.28)⁸⁴ with larger octahedral anions, presumably indicating that the chemical pressure induced by adopting smaller cations has a remarkable effect on the destabilization of the CO state. Additional evidence of the effect of lattice compression on 2δ was provided by the rapid decrease in 2δ upon the application of pressure.⁸⁶

Herein, we have considered that the orientation of the PF₆⁻ anions that are dynamically disordered on an inversion center becomes ordered at low temperatures. In this case, the space

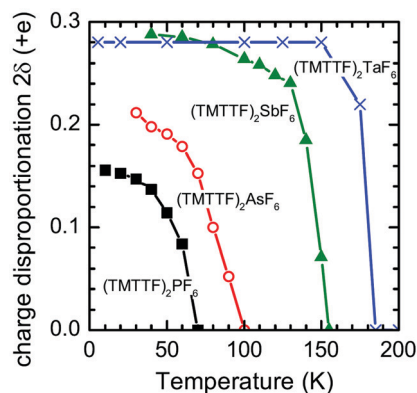


Fig. 18 Temperature dependence of the degree of charge disproportionation (2δ) of (TMTTF)₂X with X = PF₆⁻ (black closed squares), AsF₆⁻ (red open circles), SbF₆⁻ (green closed triangles), and TaF₆⁻ (blue crosses), estimated from the IR spectra.⁸⁵

group changes from $P\bar{1}$ to $P1$ due to the loss of the inversion center; however, the centrosymmetric nature of the anion disturbs the formation of any superstructure. It seems reasonable that the charges of two crystallographically independent TMTTF molecules in a unit cell deviate from the average charge of +0.5, *i.e.*, $+(0.5 + \delta)$ and $+(0.5 - \delta)$, causing a CP-type CO pattern. However, structural alternation in the stacking columns corresponding to the CO state could not be appreciably detected in crystallographic studies, although symmetry breaking was observed in the NMR spectra.⁷⁹ Therefore, the CO transition was temporarily considered as a “structureless” transition.⁸⁷ The first evidence of relevant structural modulation was recently found *via* synchrotron X-ray diffraction measurements,⁸⁸ which is not only the same as that previously considered based on experimental (ESR)⁸⁹ and theoretical^{90,91} studies, but also implies the formation of a 2D Wigner lattice. In such a 2D layer (within the *ab* plane), charge-rich TMTTF molecules with *ca.* +0.6 charge are preferably arranged so as not to be contiguous to each other as a consequence of long-range Coulomb repulsions (sum of the on-site (*U*) and inter-site (*V*) Coulomb repulsions).

Dielectric permittivity measurements show a peak divergence in the real part of the dielectric constant for (TMTTF)₂PF₆

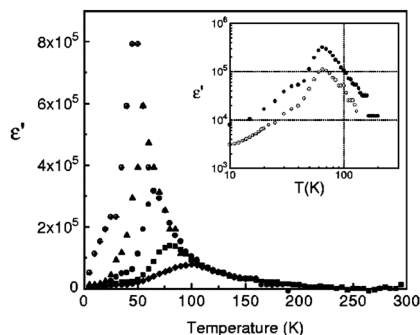


Fig. 19 Temperature dependence of the real part of the dielectric permittivity (ϵ') of $(\text{TMTTF})_2\text{PF}_6$ at different frequencies: 1 kHz (\oplus), 10 kHz (\blacktriangle), 100 kHz (\bullet), 1 MHz (\blacksquare), and 5 MHz (\blacklozenge). Inset shows the difference between the ϵ' of the salt obtained from different batches (100 kHz). Adapted with permission from ref. 92. Copyright 2000 American Physical Society.

(Fig. 19),⁹² $(\text{TMTTF})_2\text{AsF}_6$,⁹³ and $(\text{TMTTF})_2\text{SbF}_6$ ⁸¹ at each transition temperature. The unusual dielectric response that has a broad peak as a function of temperature and is strictly frequency-dependent bears the clear signatures of relaxor ferroelectrics, typical of the glass-like freezing of short-range ferroelectric ordering. It is noteworthy that the ground state varies depending on the anions; *i.e.*, spin-Peierls state in $(\text{TMTTF})_2\text{PF}_6$ and $(\text{TMTTF})_2\text{AsF}_6$ and antiferromagnetic ordered state in $(\text{TMTTF})_2\text{SbF}_6$.^{3,14,39} Theoretical study predicted that the CP-type CO state (including $4k_{\text{F}}$ -CDW) stabilized by the increase in V inhibits the spin-Peierls transition (including $2k_{\text{F}}$ -BOW; Fig. 20(a)), and therefore, $(\text{TMTTF})_2\text{SbF}_6$, which has a more stable CO state with a higher charge-ordering temperature, undergoes an antiferromagnetic transition (Fig. 20(b)) in place of a spin-Peierls transition at low temperatures.⁹⁴

Next, the structural and physical properties of $(\text{TMTTF})_2\text{ReO}_4$ as an example of $(\text{TMTTF})_2\text{X}$ formed with noncentrosymmetric tetrahedral anions have been described. $(\text{TMTTF})_2\text{ReO}_4$ has a room-temperature conductivity of *ca.* 20 S cm^{-1} and shows metallic behavior down to *ca.* 230 K,⁸⁷ at which it undergoes a $2k_{\text{F}}$ -CDW CO (= ACP; Fig. 17(c)) transition observed by various techniques such as X-ray diffraction,^{83,95} ESR,⁸⁹ and NMR⁹⁶

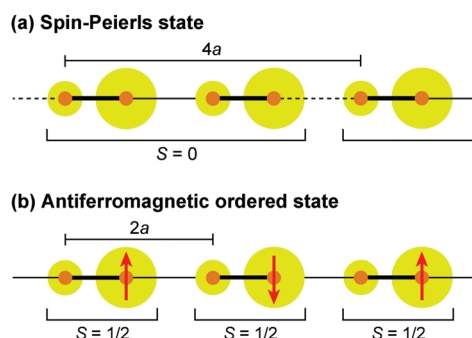


Fig. 20 Schematic electronic structures of (a) spin-Peierls state in $(\text{TMTTF})_2\text{PF}_6$ and $(\text{TMTTF})_2\text{AsF}_6$ and (b) antiferromagnetic ordered state in $(\text{TMTTF})_2\text{SbF}_6$.⁹⁴ Red arrows indicate a magnetically ordered spin, and other specifications are the same as those used in Fig. 17.

measurements. The noncentrosymmetric ReO_4^- anions are randomly orientated at room temperature, and the orientational disorder is frozen over two (or more) equivalent orientations upon cooling. An X-ray diffuse scattering study revealed that $(\text{TMTTF})_2\text{ReO}_4$ undergoes a first-order structural phase transition associated with anion ordering at 157 K, at which a superstructure with a critical wave vector $\mathbf{q}_{\text{AO}} = (1/2, 1/2, 1/2)$ to form a $2k_{\text{F}}$ modulation within the π -stacking columns becomes pronounced.⁹⁷ This transition is accompanied by the displacement of the anions, which results in the sizeable tetramerization of the TMTTF molecules within the column to stabilize the $2k_{\text{F}}$ -CDW CO state ($2\delta = 0.35$ at 200 K and 0.64 at 40 K).⁹⁸ At present, the precise reason why $(\text{TMTTF})_2\text{ClO}_4$ does not undergo a CO transition at low temperatures is not fully established; however, it is possible that the occurrence of anion ordering (73 K) prior to charge ordering causes spin-singlet formation due to doubled periodicity along the stacking direction. As shown in ref. 83, the degree of dimerization of $(\text{TMTTF})_2\text{ClO}_4$ is relatively independent of temperature and remains at a high level, which can affect the suppression of the CO state.²⁹ It must be true that the symmetry of counter anions is responsible for determining the CO pattern, whether CP or ACP, in the TMTTF π -stacking column (Table 1). To date, a successive phase transition from the AV to ACP state passing through the CP state on cooling, as observed for $\text{Pt}_2(\text{RCS}_2)_4\text{I}$, has not been reported for $(\text{TMTTF})_2\text{X}$.

$(\text{TMTTF})_2\text{X}$ formed with another type of noncentrosymmetric anion, linear SCN^- , exhibits some similarities with $(\text{TMTTF})_2\text{ReO}_4$ in addition to striking differences. With decreasing temperature, $(\text{TMTTF})_2\text{SCN}$ also undergoes a structural phase transition due to anion ordering at 160 K.⁹⁹ However, the superstructure is equal to $\mathbf{q}_{\text{AO}} = (0, 1/2, 1/2)$, indicating that the CO pattern is of $4k_{\text{F}}$ -CDW-type (= CP), as in the case of $(\text{TMTTF})_2\text{PF}_6$, because of no phase modulation within the π -stacking column. On further cooling, $(\text{TMTTF})_2\text{SCN}$ undergoes antiferromagnetic ordering below 7 K, whereas $(\text{TMTTF})_2\text{PF}_6$ and $(\text{TMTTF})_2\text{ReO}_4$ show a pronounced phase transition to nonmagnetic ground states arising from the spin-Peierls transition at 15 K and anion-order-induced tetramerization at 157 K, respectively.^{3,14,39}

As described above, the application of pressure modifies the CO behavior and also bring the salts into the 3D ordered electronic states such as antiferromagnetic ordered and superconducting states. Pressure dependence of the CO transition was investigated by ^{13}C NMR spectroscopy.^{100,101} For example, the CO transition temperature of $(\text{TMTTF})_2\text{AsF}_6$, which appears to be 102 K under ambient pressure, dramatically decreases with increasing pressure and eventually disappears under a pressure of 0.15 GPa.¹⁰⁰ Such a suppression of the CO state, as observed for $\text{Pt}_2(\text{EtCS}_2)_4\text{I}$ below 2.2 GPa (Fig. 8),³² is possibly a reflection of the increased W compared with V .

In the generic phase diagram of $(\text{TMTTF})_2\text{X}$ and $(\text{TMTSF})_2\text{X}$ systems shown in Fig. 13, which was originally proposed by Jérôme^{13,102} and subsequently made more sophisticated by Dressel,^{39,66} “loc” (as Mott insulating state), “1D”, and “2D” regimes are defined on the basis of the anisotropy of transport behavior as “ $d\rho_{\text{a}}/dT < 0$ and $d\rho_{\text{c}}/dT < 0$ ”, “ $d\rho_{\text{a}}/dT > 0$ and

$d\rho_c/dT < 0$ ”, and “ $d\rho_a/dT > 0$ and $d\rho_c/dT > 0$ ”, respectively,¹⁰³ where the a and c axes correspond to the most- and least-conducting directions, respectively. Therefore, the boundary between the “loc” and “1D” regimes is determined as a temperature at which the conductivity along the a axis exhibits a maximum (A charge gap (Δ_ρ) estimated from the slope of the Arrhenius plot is often substituted for the temperature due to thermal inaccessibility). Furthermore, the temperature at which the conductivity along the c axis exhibits a maximum provides a crossover between the “1D” and “2D” regimes and is therefore regarded as the Mott transition temperature (or Mott–Hubbard gap).¹⁰⁴ The crossover between the “2D” and “3D” regimes, which is only available for the (TMTSF)₂X system, is defined as the upper limit for a low-temperature 3D coherent domain.¹⁰⁵ The metal–insulator transition temperature is gradually decreased upon applying pressure possibly because of the decrease in the on-site Coulomb repulsion. This trend is exactly opposite to that found for Pt₂(RCS₂)₄I (R = Me and Et),^{32,37} although the electronic states just below the metal–insulator transition might be different in kind from each other. Increased electronic dimensionality in addition to the significant lattice compression may modify the electronic state in the favorable direction of a stabilized metallic phase. In fact, the calculated dimensionality parameter, defined as the ratio of inter-column (t_\perp) and intra-column (t_\parallel) transfer energies, increases with increasing applied pressure and decreasing temperature (Fig. 21).¹⁰⁶ Therefore, it is most likely that on cooling, Pt₂(RCS₂)₄I with a strong 1D character moves towards the lower left direction instead of towards the beneath direction under a certain pressure in the phase diagram shown in Fig. 13. Although the 1D character disturbs the antiferromagnetic ordering due to quantum fluctuations, the metal–insulator transition followed by the emergence of the CO state, as observed for most Pt₂(RCS₂)₄I, seems to be consistent with this possible movement.

It is fair to say that all (TMTTF)₂X salts reside in the Luttinger-liquid regime at high temperatures because of their 1D electronic character. Herein, we focused on the inter-column ($\parallel c$) transport behavior in the insulating regime, because the temperature dependence of the resistivity is described by the Luttinger liquid model: $\rho_c \sim T^{1-2\alpha}$, $\alpha = 1/4(K_\rho + K_\rho^{-1} - 2)$, where K_ρ is the Luttinger parameter (cf., $\rho_a \sim T^{16K_\rho-3}$ at $T > \Delta_\rho$ for quarter-filled

system).⁶⁷ An electronic system with stronger repulsive energies affords a lower K_ρ , and the critical value for Mott transition varies depending on the order of commensurability ($K_\rho^* = 1/n^2$; $n = 1$ for half-filled system and $n = 2$ for quarter-filled system). Clearly, the low K_ρ values observed for (TMTTF)₂PF₆ (0.18)¹⁰⁴ and (TMTTF)₂AsF₆ (0.20)¹⁰⁷ are associated with the quarter-filled HOMO band. Since the K_ρ values are close to 0.25 above which quarter-filled umklapp scattering is irrelevant, it is apparent that these salts are just on the insulating side of the boundary between the Tomonaga–Luttinger liquid and 1D Mott insulating state. Selenium substitution (TMTTF to TMTSF)¹⁰³ and application of pressure¹⁰⁸ rapidly bring these salts into the metallic regime with higher K_ρ values.

4. Outlook and conclusions

The 1D electronic subclass of metal-complex and molecular solids is very broad, and we have only covered the most informative examples in this review. As mentioned above, the half-filled band with antibonding character, which arises from the monocationic dimers as units, leads to the pronounced interdimer interactions owing to the orbital expanding towards the outside of dimers. The increased W as well as the decreased U owing to the dimer formation is considered to be the primary cause for the comparable U and W , which bring the solids into the vicinity of Mott boundary. Although the chemical modification of TMTTF molecules causes a dramatical change in the molecular arrangement in the cation radical salts, the basic structure of MMX-chain complexes remains unchanged because of the covalent bonding between the binuclear metal subunits through the bridging halogens. Considering the fact that the functional groups attached to the dithiocarboxylate ligands in MMX-chain complexes remain rather limited to n -alkyl (from methyl to n -hexyl) groups, attaching other functional groups to impart desired functionalities and longer alkyl groups that serve as van der Waals interaction sites may provide a significant step forward in exploring more versatile electronic materials, apart from the difficulty in synthesizing the ligands. Additionally, the subunits that are separated from each other by a bridging halogen are beneficial not only for the full substitution of Pt with Ni,^{109,110} but also for the partial substitution of Pt to afford heterobimetallic subunits PtM (M = Ni and Pd).¹¹¹ Such a replacement while maintaining the basic structure is not simply possible in (TMTTF)₂X, in which the columnar assembly of TMTTF molecules is achieved mainly by the weak π – π interactions.

The principal advantage of (TMTTF)₂X over the MMX-chain complexes might be the tunability of their electronic properties by chemical or physical pressure, which is mainly attributed to the softness of the lattice. The appearance of 3D ordered electronic states such as antiferromagnetic ordered and superconducting states that are not yet found in MMX-chain complexes can best be thought of as a manifestation of the intermolecular π – π and van der Waals interactions as dominant structure-directing forces. Additionally, by studying the relative

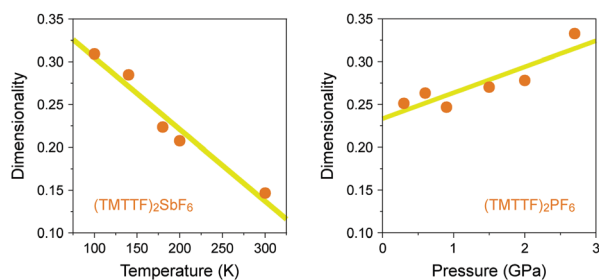


Fig. 21 (a) Temperature dependence of dimensionality (see text) of (TMTTF)₂SbF₆ and (b) pressure dependence of dimensionality of (TMTTF)₂PF₆ (data obtained from ref. 106). Yellow lines are guides to the eye.

properties of a variety of (TMTTF)₂X, a broad picture of the electronic behavior of this system has depicted not only the 1D regime but also the 1D–2D dimensional crossover regime. We hope that this review, which compares the structural and electronic properties of metal-chain complexes and organic conductors to provide insights into the relevant issues that are currently under debate, will stimulate further experimental and theoretical explorations of new and more advanced electronic materials that might surpass the existing electronic materials.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

The authors are grateful to Dr Toshikazu Nakamura (Institute for Molecular Science, Japan) and Dr Kenji Yonemitsu (Chuo University, Japan) for their helpful discussion. We also thank Dr Nakamura for providing crystallographic data.

Notes and references

- P. Day, *Mixed Valence Chemistry and Metal Chain Compounds*, in *Low-Dimensional Cooperative Phenomena*, ed. H. Keller, Plenum, New York, 1975, pp. 191–214.
- Extended Linear Chain Compounds*, ed. E. Conwell and J. S. Miller, Plenum Press, New York, 1982.
- T. Ishiguro, K. Yamaji and G. Saito, *Organic Superconductors*, Springer-Verlag, Berlin, 2nd edn, 1998.
- J. W. Bray, H. R. Hart, Jr., L. V. Interrante, I. S. Jacobs, J. S. Kasper, G. D. Watkins, S. H. Wee and J. C. Bonner, *Phys. Rev. Lett.*, 1975, **35**, 744.
- M. Hase, I. Terasaki and K. Uchinokura, *Phys. Rev. Lett.*, 1993, **70**, 3651.
- K. Krogmann and H. D. Hausen, *Z. Anorg. Allg. Chem.*, 1968, **358**, 67.
- H. R. Zeller and A. Beck, *J. Phys. Chem. Solids*, 1974, **35**, 77.
- J. P. Ferraris, D. O. Cowan, V. Walatka, Jr. and J. H. Perlstein, *J. Am. Chem. Soc.*, 1973, **95**, 948.
- L. B. Coleman, M. J. Cohen, D. J. Sandman, F. G. Yamagishi, A. F. Garito and A. J. Heeger, *Solid State Commun.*, 1973, **12**, 1125.
- Semiconductors and Semimetals*, ed. E. Conwell, Academic Press, San Diego, 1988, vol. 27.
- K. Nasu, *J. Phys. Soc. Jpn.*, 1983, **52**, 3865.
- Material Designs and New Physical Properties in MX- and MMX-Chain Compounds*, ed. M. Yamashita and H. Okamoto, Springer-Verlag, Wien, 2013.
- D. Jérôme, *Chem. Rev.*, 2004, **104**, 5565.
- J.-P. Pouget, P. Alemany and E. Canadell, *Mater. Horiz.*, 2018, **5**, 590.
- H. Wolffram, Dissertation, Königsberg, 1900.
- B. K. Chakraverty, *J. Phys. Lett.*, 1979, **40**, L99.
- M. B. Robin and P. Day, *Adv. Inorg. Chem. Radiochem.*, 1967, **10**, 247.
- H. Kitagawa, N. Onodera, T. Sonoyama, M. Yamamoto, T. Fukuwa, T. Mitani, M. Seto and Y. Maeda, *J. Am. Chem. Soc.*, 1999, **121**, 10068.
- C. Bellitto, A. Flamini, L. Gastaldi and L. Scaramuzza, *Inorg. Chem.*, 1983, **22**, 444.
- M. Yamashita, S. Takaishi, A. Kobayashi, H. Kitagawa, H. Matsuzaki and H. Okamoto, *Coord. Chem. Rev.*, 2006, **250**, 2335.
- H. Iguchi, S. Takaishi and M. Yamashita, *Chem. Lett.*, 2014, **43**, 69.
- J. Grote, F. Friedrich, K. Berthold, L. Hericks, B. Neumann, H.-G. Stammer and N. W. Mitzel, *Chem. – Eur. J.*, 2018, **24**, 2626.
- H. Kitagawa, N. Onodera, J.-S. Ahn, T. Mitani, M. Kim, Y. Ozawa, K. Toriumi, K. Yasui, T. Manabe and M. Yamashita, *Mol. Cryst. Liq. Cryst.*, 1996, **285**, 311.
- H. Kitagawa, N. Onodera, J.-S. Ahn, T. Mitani, K. Toriumi and M. Yamashita, *Synth. Met.*, 1997, **86**, 1931.
- H. Kitagawa and T. Mitani, *Coord. Chem. Rev.*, 1999, **190–192**, 1169.
- S. A. Borschch, K. Prassides, V. Robert and A. O. Solonenko, *J. Chem. Phys.*, 1998, **109**, 4562.
- S. Yamamoto, *J. Phys. Soc. Jpn.*, 2001, **70**, 1198.
- S. Yamamoto, *J. Phys. Chem. Solids*, 2002, **63**, 1489.
- M. Tsuchiizu, H. Yoshioka and Y. Suzumura, *J. Phys. Soc. Jpn.*, 2001, **70**, 1460.
- M. Kuwabara and K. Yonemitsu, *J. Mater. Chem.*, 2001, **11**, 2163.
- M. Mitsumi, T. Murase, H. Kishida, T. Yoshinari, Y. Ozawa, K. Toriumi, T. Sonoyama, H. Kitagawa and T. Mitani, *J. Am. Chem. Soc.*, 2001, **123**, 11179.
- K. Otsubo, A. Kobayashi, M. Hedo, Y. Uwatoko and H. Kitagawa, *Chem. – Asian J.*, 2009, **4**, 1673.
- M. Mitsumi, K. Kitamura, A. Morinaga, Y. Ozawa, M. Kobayashi, K. Toriumi, Y. Iso, H. Kitagawa and T. Mitani, *Angew. Chem. Int. Ed.*, 2002, **41**, 2767.
- A. Kobayashi, S. Kitao, M. Seto, R. Ikeda and H. Kitagawa, *Inorg. Chem.*, 2009, **48**, 8044.
- H. Tanaka and S. Kuroda, *et al.*, to be submitted.
- K. Otsubo, A. Kobayashi, H. Kitagawa, M. Hedo, Y. Uwatoko, H. Sagayama, Y. Wakabayashi and H. Sawa, *J. Am. Chem. Soc.*, 2006, **128**, 8140.
- K. Otsubo, T. Suto, A. Kobayashi, R. Ikeda, M. Hedo, Y. Uwatoko and H. Kitagawa, *Eur. J. Inorg. Chem.*, 2016, 4402.
- L. Degiorgi and D. Jérôme, *J. Phys. Soc. Jpn.*, 2006, **75**, 051004.
- M. Dressel, *Naturwissenschaften*, 2007, **94**, 527.
- K. Saito, S. Ikeuchi, Y. Nakazawa, A. Sato, M. Mitsumi, T. Yamashita, K. Toriumi and M. Sorai, *J. Phys. Chem. B*, 2005, **109**, 2956.
- A. Guijarro, O. Castillo, L. Welte, A. Calzolari, P. J. S. Miguel, C. J. Gómez-García, D. Olea, R. di Felice, J. Gómez-Herrero and F. Zamora, *Adv. Funct. Mater.*, 2010, **20**, 1451.
- M. Mitsumi, T. Yamashita, Y. Aiga, K. Toriumi, H. Kitagawa, T. Mitani and M. Kurmoo, *Inorg. Chem.*, 2011, **50**, 4368.
- M. Mitsumi, S. Umebayashi, Y. Ozawa, K. Toriumi, H. Kitagawa and T. Mitani, *Chem. Lett.*, 2002, 258.
- A. Ota, L. Ouahab, S. Golhen, Y. Yoshida, M. Maesato and G. Saito, and R. Świątlik, *Chem. Mater.*, 2007, **19**, 2455.
- P. Le Maguerès, L. Ouahab, P. Briard, J. Even, M. Bertault, L. Toupet, J. Ramos, C. J. Gomez-García and P. Delhaès, *Mol. Cryst. Liq. Cryst.*, 1997, **305**, 479.
- T. Kakiuchi, Y. Wakabayashi, H. Sawa, T. Takahashi and T. Nakamura, *J. Phys. Soc. Jpn.*, 2007, **76**, 113702.
- Y. Miyazaki, Q. Wang, A. Sato, K. Saito, M. Yamamoto, H. Kitagawa, T. Mitani and M. Sorai, *J. Phys. Chem. B*, 2002, **106**, 197.
- S. Ikeuchi, K. Saito, Y. Nakazawa, A. Sato, M. Mitsumi, K. Toriumi and M. Sorai, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **66**, 115110.
- S. Ikeuchi, K. Saito, Y. Nakazawa, M. Mitsumi, K. Toriumi and M. Sorai, *J. Phys. Chem. B*, 2004, **108**, 387.
- M. Sorai, M. Nakano and Y. Miyazaki, *Chem. Rev.*, 2006, **106**, 976.
- S. Ikeuchi, Y. Yamamura, M. Mitsumi, K. Toriumi, H. Saitoh, T. Atake and K. Saito, *Chem. Lett.*, 2009, **38**, 1190.
- G. Saito and Y. Yoshida, *Bull. Chem. Soc. Jpn.*, 2007, **80**, 1.
- G. Saito and Y. Yoshida, *Chem. Rec.*, 2011, **11**, 124.
- J. M. Williams, A. J. Schultz, U. Geiser, K. D. Carlson, A. M. Kini, H. H. Wang, W.-K. Kwok, M.-H. Whangbo and J. E. Schirber, *Science*, 1991, **252**, 1501.
- U. Geiser and J. A. Schlueter, *Chem. Rev.*, 2004, **104**, 5203.
- H. Kitagawa, S. Nakagami and T. Mitani, *Synth. Met.*, 2001, **116**, 401.
- K. Furukawa, T. Hara and T. Nakamura, *J. Phys. Soc. Jpn.*, 2009, **78**, 104713.
- E. Rose, C. Loose, J. Kortus, A. Pashkin, C. A. Kuntscher, S. G. Ebbinghaus, M. Hanfland, F. Lissner, Th. Schleid and M. Dressel, *J. Phys.: Condens. Matter*, 2013, **25**, 014006.
- J. P. Ferraris, T. O. Poehler, A. N. Bloch and D. O. Cowan, *Tetrahedron Lett.*, 1973, **14**, 2553.
- A. Mas, J.-M. Fabre, E. Torrelles, L. Giral and G. Brun, *Tetrahedron Lett.*, 1977, **18**, 2579.
- J.-M. Fabre, C. Galaine and L. Giral, *J. Phys., Colloq.*, 1983, **44**, C3–1153.
- P. Delhaes, C. Coulon, J. Amiel, S. Flandrois, E. Torrelles, J.-M. Fabre and L. Giral, *Mol. Cryst. Liq. Cryst.*, 1979, **50**, 43.

- 63 G. Brun, B. Liautard, S. Peytavin, M. Maurin, E. Toreilles, J.-M. Fabre, L. Giral and J. L. Galigné, *J. Phys., Colloq.*, 1977, **38**, C7-266.
- 64 D. Jérôme, A. Mazaud, M. Ribault and K. Bechgaard, *J. Phys., Lett.*, 1980, **41**, L95.
- 65 K. Bechgaard, K. Carneiro, F. B. Rasmussen and M. Olsen, *J. Am. Chem. Soc.*, 1981, **103**, 2440.
- 66 M. Dressel, M. Dumm, T. Knoblauch and M. Masino, *Crystals*, 2012, **2**, 528.
- 67 T. Giamarchi, *Chem. Rev.*, 2004, **104**, 5037.
- 68 B. Köhler, E. Rose, M. Dumm, G. Untereiner and M. Dressel, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2011, **84**, 035124.
- 69 A. Calzolari, S. S. Alexandre, F. Zamora and R. di Felice, *J. Am. Chem. Soc.*, 2008, **130**, 5552.
- 70 T. Adachi, E. Ojima, K. Kato, H. Kobayashi, T. Miyazaki, M. Tokumoto and A. Kobayashi, *J. Am. Chem. Soc.*, 2000, **122**, 3238.
- 71 D. Jaccard, H. Wilhelm, D. Jérôme, J. Moser, C. Carcel and J.-M. Fabre, *J. Phys.: Condens. Matter*, 2001, **13**, L89.
- 72 M. Itoi, M. Kano, N. Kurita, M. Hedou, Y. Uwatoko and T. Nakamura, *J. Phys. Soc. Jpn.*, 2007, **76**, 053703.
- 73 M. Itoi, C. Araki, M. Hedou, Y. Uwatoko and T. Nakamura, *J. Phys. Soc. Jpn.*, 2008, **77**, 023701.
- 74 P. Auban-Senzier, C. Pasquier, D. Jérôme, C. Carcel and J.-M. Fabre, *Synth. Met.*, 2003, **133-134**, 11.
- 75 L. Balicas, K. Behnia, W. Kang, P. Auban-Senzier, E. Canadell, D. Jérôme, M. Ribault and J.-M. Fabre, *Adv. Mater.*, 1994, **6**, 762.
- 76 M. Dressel, S. Kirchner, P. Hesse, G. Untereiner, M. Dumm, J. Hemberger, A. Loidl and L. Montgomery, *Synth. Met.*, 2001, **120**, 719.
- 77 C. Coulon, P. Delhaes and S. Flandrois, *J. Phys.*, 1982, **43**, 1059.
- 78 M. Dumm, A. Loidl, B. W. Fravel, K. P. Starkey, L. K. Montgomery and M. Dressel, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **61**, 511.
- 79 D. S. Chow, F. Zamborszky, B. Alavi, D. J. Tantilillo, A. Baur, C. A. Merlic and S. E. Brown, *Phys. Rev. Lett.*, 2000, **85**, 1698.
- 80 T. Nakamura, K. Furukawa and T. Hara, *J. Phys. Soc. Jpn.*, 2007, **76**, 064715.
- 81 P. Monceau, F. Y. Nad and S. Brazovskii, *Phys. Rev. Lett.*, 2001, **86**, 4080.
- 82 M. Dumm, M. Abaker and M. Dressel, *J. Phys. IV*, 2005, **131**, 55.
- 83 Y. Nogami, T. Ito, K. Yamamoto, N. Irie, S. Horita, T. Kambe, N. Nagao, K. Oshima, N. Ikeda and T. Nakamura, *J. Phys. IV*, 2005, **131**, 39.
- 84 Y. Oka, N. Matsunaga, K. Nomura, A. Kawamoto, K. Yamamoto and K. Yakushi, *J. Phys. Soc. Jpn.*, 2015, **84**, 114709.
- 85 T. Peterseim and M. Dressel, *Crystals*, 2017, **7**, 278.
- 86 I. Voloshenko, M. Herter, R. Beyer, A. Pustogow and M. Dressel, *J. Phys.: Condens. Matter*, 2017, **29**, 115601.
- 87 C. Coulon, S. S. P. Parkin and R. Laversanne, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1985, **31**, 3583.
- 88 S. Kitou, T. Fujii, T. Kawamoto, N. Katayama, S. Maki, E. Nishibori, K. Sugimoto, M. Takata, T. Nakamura and H. Sawa, *Phys. Rev. Lett.*, 2017, **119**, 065701.
- 89 T. Nakamura, *J. Phys. Soc. Jpn.*, 2003, **72**, 213.
- 90 H. Seo and H. Fukuyama, *J. Phys. Soc. Jpn.*, 1997, **66**, 1249.
- 91 S. Mazumdar, R. T. Clay and D. K. Campbell, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **62**, 13400.
- 92 F. Nad, P. Monceau, C. Carcel and J.-M. Fabre, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2000, **62**, 1753.
- 93 F. Nad, P. Monceau, C. Carcel and J.-M. Fabre, *J. Phys.: Condens. Matter*, 2000, **12**, L435.
- 94 K. Yoshimi, H. Seo, S. Ishibashi and S. E. Brown, *Phys. Rev. Lett.*, 2012, **108**, 096402.
- 95 Y. Nogami and T. Nakamura, *J. Phys. IV*, 2002, **12**, Pr9-Pr145.
- 96 T. Nakamura, K. Furukawa and T. Hara, *J. Phys. Soc. Jpn.*, 2006, **75**, 013707.
- 97 S. S. P. Parkin, J. J. Mayerle and E. M. Engler, *J. Phys., Colloq.*, 1983, **44**, C3-1105.
- 98 A. Pustogow, T. Peterseim, S. Kolatschek, L. Engel and M. Dressel, *Phys. Rev. B*, 2016, **94**, 195125.
- 99 C. Coulon, A. Maaroufi, J. Amiel, E. Dupart, S. Flandrois, P. Delhaes, R. Moret, J. P. Pouget and J. P. Morand, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1982, **26**, 6322.
- 100 F. Zamborszky, W. Yu, W. Raas, S. E. Brown, B. Alavi, C. A. Merlic and A. Baur, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2002, **66**, 081103.
- 101 W. Yu, F. Zhang, F. Zamborszky, B. Alavi, A. Baur, C. A. Merlic and S. E. Brown, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2004, **70**, 121101.
- 102 D. Jérôme, *Science*, 1991, **252**, 1509.
- 103 J. Moser, M. Gabay, P. Auban-Senzier, D. Jérôme, K. Bechgaard and J.-M. Fabre, *Eur. Phys. J. B*, 1998, **1**, 39.
- 104 P. Auban-Senzier, D. Jérôme, C. Carcel and J.-M. Fabre, *J. Phys. IV*, 2004, **114**, 41.
- 105 P. Auban-Senzier, D. Jérôme, N. Doiron-Leyraud, S. René de Cotret, A. Sedeki, C. Bourbonnais, L. Taillefer, P. Alemany, E. Canadell and K. Bechgaard, *J. Phys.: Condens. Matter*, 2011, **23**, 345702.
- 106 A. C. Jacko, H. Feldner, E. Rose, F. Lissner, M. Dressel, R. Valentí and H. O. Jeschke, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 155139.
- 107 B. Korin-Hamzić, E. Tafra, M. Basletić, A. Hamzić and M. Dressel, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2006, **73**, 115102.
- 108 A. Pashkin, M. Dressel and C. A. Kuntscher, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2006, **74**, 165118.
- 109 C. Bellitto, G. Dessy and V. Fares, *Inorg. Chem.*, 1985, **24**, 2815.
- 110 M. Mitsumi, Y. Youshida, A. Kohyama, Y. Kitagawa, Y. Ozawa, M. Kobayashi, K. Toriumi, M. Tadokoro, N. Ikeda, M. Okumura and M. Kurmoo, *Inorg. Chem.*, 2009, **48**, 6680.
- 111 G. Givaja, O. Castillo, E. Mateo, A. Gallego, C. J. Gómez-García, A. Calzolari, R. di Felice and F. Zamora, *Chem. – Eur. J.*, 2012, **18**, 15476.