


 Cite this: *RSC Adv.*, 2021, **11**, 28838

 Received 8th June 2021  
 Accepted 23rd August 2021

DOI: 10.1039/d1ra04434f

[rsc.li/rsc-advances](http://rsc.li/rsc-advances)

# A review of shape memory polymers based on the intrinsic structures of their responsive switches

 Lide Yang, <sup>a</sup> Jiankun Lou,<sup>a</sup> Jianmin Yuan\*<sup>a</sup> and Jianru Deng\*<sup>b</sup>

Shape memory polymers (SMPs), as stimuli-responsive materials, have attracted worldwide attention. Based on the history and development of SMPs, a variety of reports about SMPs in recent years are summarized in this paper. The responsive switches are analyzed and divided into two kinds according to their intrinsic structures: physical switch and chemical one. Then, detailed classification and comprehensive discussion of SMPs are further elaborated, based on the intrinsic structures of responsive switches and stimulation types. Finally, the development and prospect of SMPs are objectively predicted and forecasted.

## 1. Introduction

Animals, plants and other organisms in nature have the ability to adapt themselves to different environments by altering their own characteristics. Inspired by nature, scientists have designed and manufactured some intelligent materials that can perceive environmental stimuli, analyze and process information, make judgments and take corresponding measures. Shape memory materials, including shape memory alloys,<sup>1</sup> shape memory ceramics,<sup>2</sup> shape memory polymers (SMPs),<sup>3</sup> are a class of intelligent materials. SMPs are one kind of the most important shape memory materials due to their superior properties (*e.g.*, strong designability of molecular structure, excellent flexibility, good processability, and light weight) and broad application prospects, which have been greatly developed over the past few decades.<sup>4</sup>

In recent years, with the rapid development of artificial intelligence, there has been a worldwide upsurge of research interest in SMPs. SMPs, being of the functions of self-monitoring, self-repairing and self-adaptation, are the important basic materials for intelligent autonomous system. However, much higher and more comprehensive requirements for SMPs are put forward due to the demands of higher intelligence, more complex environments and wider applications. Thus, SMPs will not only face new opportunities but also be confronted with greater challenges. Now, great efforts have been done on shape memory materials and researchers have also published review articles about SMPs from different perspectives (*e.g.*, matrix resins, driving types, nanocomposites and applications).<sup>4,5</sup> The responsive switches are basic to the designs, manufactures and applications of SMPs. Based on summary and analysis of the development history and research progress for SMPs, the nature of the responsive switches is disclosed in this review, who is the reversible

interaction of physics or chemistry among polymer molecular chains or chain segments. And detailed classification and comprehensive discussion of SMPs are further elaborated according to the intrinsic structures of responsive switches and stimulation types (see Fig. 1), which is not only conducive for a better understanding of SMPs, but also helpful in designing or developing novel SMPs.<sup>6</sup>

## 2. History of SMPs

As far as we know, it has been nearly 100 years since the first discovery of shape memory materials, a kind of gold–cadmium alloys observed by Olander Sweden in 1932. Once heated to a threshold temperature, the deformed alloys could automatically

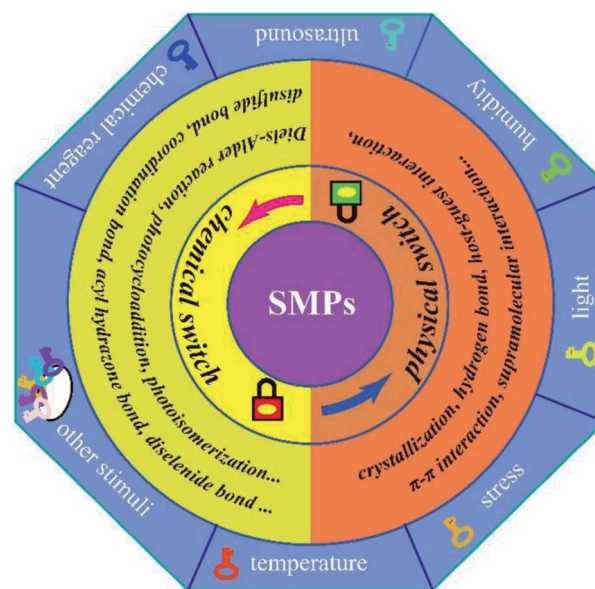


Fig. 1 Schematic diagram of the classification of SMPs based on the intrinsic structures of their responsive switches.

<sup>a</sup>College of Materials Science and Engineering, Hunan University, Changsha, 410082, P. R. China. E-mail: pangyuan2916@hnu.edu.cn

<sup>b</sup>College of Chemistry and Chemical Engineering, Hunan University, Changsha, 410082, P. R. China



revert to its original shape. Then, the concept of “elastic memory” was firstly proposed by Vernon in 1940s. In 1950s, Charles accidentally found the shape memory behaviours (shape fixation and shape recovery) of a chemical cross-linked polythene, opening the door for further study and applications of SMPs. It was an important milestone that the first commercial product of SMPs, thermal-shrinkable polyethylene pipe, came out in 1960. In 1970, National Aeronautics and Space Administration launched the application of SMPs in aerospace industry, and then restarted the research program for SMPs. In 1990, many kinds of SMPs were developed, such as polyolefin, copolymer, thermoplastic elastomer and polyester-alloy.<sup>7,8</sup>

In the last 30 years, SMPs have been paid great attention to by academic and industry scientists. There are many representative research findings, and a large number of SMPs have been developed for the requirements of applications.<sup>9,10</sup> The triggers of responsive switches in SMPs have been well studied, based on different external stimuli (*e.g.*, temperature, pH, chemical reagent, light, electricity, magnetism, pressure, humidity, ultrasonic or combinations thereof). Shape memory nanocomposites have been developed greatly in especial for their shape memory behaviours can be remotely controlled *via* light-thermal conversion, or magnetic-thermal conversion, or electro-thermal conversion, or acoustic-thermal conversion of nanomaterials. With the advent of artificial intelligence, more and more attention will be paid to SMPs.<sup>11,12</sup> There are, of course, plenty of scientific issues about SMPs that need to be answered.

### 3. The critical components of SMPs and the intrinsic structures of their responsive switches

Since the first SMP came into being, scientists have taken a lot of efforts to understand the mechanisms of shape memory

behaviours.<sup>13–17</sup> Although many mechanisms of SMPs based on responsive switches, or external stimuli, or phase transformation types are proposed, the changes in nature—the changes in physical or chemical structures of SMPs during shape memory processes—should be disclosed. It is generally believed that SMPs are composed of a stable phase, memorizing the original shape and preventing the polymer from flowing, and a reversible phase, hardening as the disappearance of stimuli (*e.g.*, temperature) and softening under stimuli. The stable phase, containing a stable network structure, is also a major contributor to the mechanical strength of a SMP. It has been shown that the entropic energy can be stored in the stable phase as a SMP is deformed from the original shape to a temporary one. When the stimulus on a temporarily deformed SMP is enough to trigger its responsive switch, the driving force generated from the stored energy prompts the SMP specimens to quickly revert to its original shape. The reversible phase is usually the linear chain structure with weak interchain interaction or the weakly cross-linked structure that is easily deformed (see Fig. 2). When the strength of a stimulus exceeds the switch threshold, the intermolecular interactions in the reversible phase are dropped to a very low level, and large deformation will occur under small external forces or even the driving force generated by the stored energy. The responsive switch is often included in reversible phase. Thus, as to complete a shape memory process, a SMP must have three critical components. The first is a stable phase (3D network structure), fixing the permanent shape of a SMP.<sup>4,18</sup> The second is a reversible phase (reversible structure) that can be easily deformed in response to external stimuli.<sup>4,19</sup> The third is a responsive switch that can open or close the door of shape deformations. Of course, an external stimulus is also the necessary factor.

At molecular level, the responsive switches just correspond to the physical or chemical weak interactions (compared with

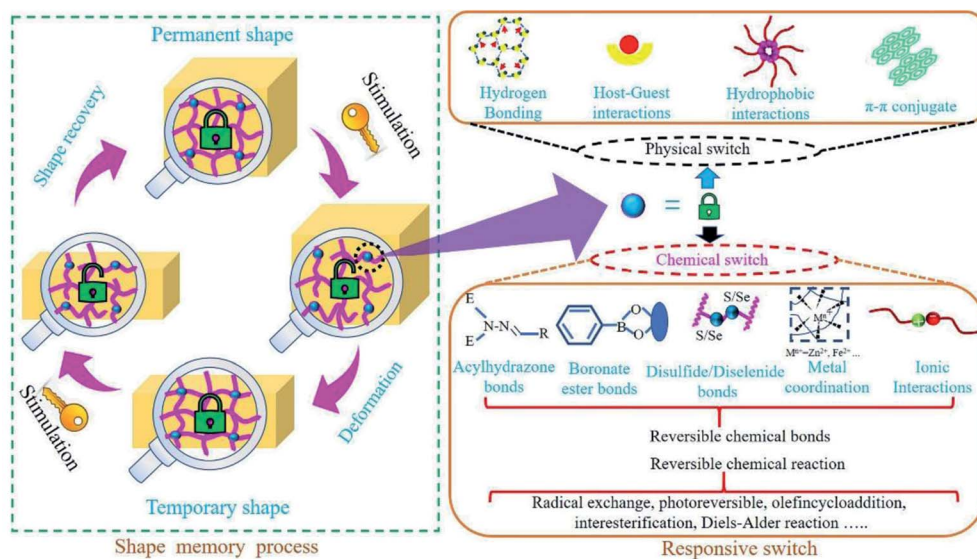


Fig. 2 Schematic diagram of shape memory process and responsive switches of SMPs.



the strong interactions of covalent bonds) among the molecular chains or chain segments of the reversible phase, which are the key factors for design and preparation of SMPs. One should note that the commonly called temperature switches of SMPs are generally referred to the temperature thresholds of shape memory recovery (see Fig. 2). Therefore, the intrinsic changes of SMPs during shape memory process involve the interconversion of dissociation and reconstruction for these weak interactions, the limited movements of the molecular chains or segments in reversible phase, and the variation of entropic energy stored in the network structure. Hu *et al.*<sup>4</sup> have explained the responsive switches from phase level and molecular level. At phase level, glass transition temperature ( $T_g$ ), melting temperature ( $T_m$ ) and isotropic temperature ( $T_i$ ), all can be used as switch thresholds. At molecular level, these intermolecular reversible factors such as non-chemical bond interactions, aggregation effects and chemical bonds can also be used as shape memory switches, that is, the shape memory switches can be physical, or chemical, or both (see Fig. 2).

Physical switches are usually resulted from some physical interactions<sup>20</sup> (*e.g.*, hydrogen bonding, supramolecular interaction,  $\pi$ - $\pi$  interaction) and chemical switches from some reversible chemical bonds<sup>21–26</sup> (*e.g.*, ionic bonds, metal coordination, disulfide bonds, acylhydrazone bonds, diselenide bonds, boronate ester bonds, oxime bonds, hydrozone bonds, imine bonds, thioester bonds) or reversible chemical reaction<sup>27–30</sup> (*e.g.*, radical exchange reaction, photoreversible olefin cycloaddition, interesterification, Diels–Alder (DA) reaction). Thus, a variety of reports about SMPs in recent years are classified and discussed in the following text, based on the intrinsic structures of responsive switches—the intermolecular interactions of physics or chemistry.

## 4. Classifications of SMPs

For better understanding the recent research of SMPs, the SMPs are divided into two kinds: one based on physical switches and

the other on chemical switches (see Table 1). Now that shape memory behaviours can be triggered by responsive switches, each type of the SMPs is further subdivided and discussed according to external stimuli (*e.g.*, temperature, stress, light, ultrasound, humidity, chemical reagent) that can turn on or turn off the switches.

### 4.1. SMPs based on physical switches

In general speaking, SMPs based on physical switches are those, whose responsive switches are associated with reversible physical interactions and may be triggered by external stimuli such as temperature, stress, light, ultrasound, humidity, chemical reagent, or combinations thereof.

**4.1.1. Temperature response.** Temperature is the most achievable and effective stimulus to trigger the responsive switches of SMPs.<sup>31</sup> Temperature rise of materials can be obtained through direct or indirect heating. Direct heating means heating materials by heat conduction or heat radiation. Indirect heating is realized by energy conversion (*e.g.*, light-thermal conversion, magneto-thermal conversion, electro-thermal conversion, and acoustic-thermal conversion) of the fillers integrated in polymer matrices. Nanomaterials are one kind of the most commonly used fillers for energy conversion. High temperature can induce phase transitions and dissociations of reversible physical interactions.

**4.1.1.1 Temperature induced phase transition.** It has been shown that phase transition is a commonly used responsive switch in SMPs. The phase transition temperature ( $T_{trans}$ ) as a switch threshold for reversible deformations might be glass transition temperature ( $T_g$ ), or melting temperature ( $T_m$ ), or isotropic temperature ( $T_i$ ).<sup>32,33</sup> Below the  $T_{trans}$ , the movements of polymer molecular chains or chain segments are restricted and the temporary shape is fixed. As the stimulation temperature rises above the  $T_{trans}$ , the movement of molecular chains or chain segments is intensified, and then the deformed SMP should restore to its initial shape.

**Table 1** Overview of selected examples of responsive switches and corresponding stimulation types

	Types of the switches	Stimulation types	Reference
Physical switches	Melting transition	Temperature, light, microwave, electricity	34, 37, 41, 59–63 and 67
	Glass transition	Temperature, magnetism, ultrasound, humidity	35, 36, 38–40, 42, 43, 62, 64, 68–70 and 75
	Hydrogen bonds	Temperature, stress, light, humidity, carbon dioxide, pH, non-aqueous solvents	44, 46–52, 54–58, 65, 66 and 71–76
	Host–guest interactions	pH	20
	Hydrophobic interactions	pH, dimethyl sulfoxide	80 and 84
Chemical switches	$\pi$ - $\pi$ interaction	Temperature	45 and 53
	Disulfide bonds	Temperature, hydrogen peroxide	86 and 121
	Diels–alder linkage	Temperature, ultrasound, light	87, 98 and 106
	Metal coordination	Light, ultrasound, complexing agent, electricity	88–91, 107, 108, 110–114 and 123
	Reversible photodimerization linkage	Light	99
	Photocycloaddition linkage	Light	100
	<i>cis</i> – <i>trans</i> isomerization	Light	101
	Borate ester bonds	pH, acetic acid solutions	115, 116, 118 and 119
	Acylhydrazone bonds	pH	117
	Donor–acceptor interactions	Hydrogen peroxide	120



For example, polycaprolactone (PCL) has been extensively discussed for its crystalline regions can play the role of network nodes and be used as the responsive switch. However, the  $T_m$  of PCL is above the human body temperature, limiting its application in biomaterials. An effective method was applied to decrease the  $T_m$  of a cross-linked PCL to around 33 °C *via* controlling the branching degree and molecular weight.<sup>34</sup> Beside the entropic energy stored in polymer network, the conformation energy stored in polymer molecular chains can also be used as the driving force of shape recovery. A SMP, whose main chain was polyacetylene in helical state like a spring and side chains were octadecyl groups like “comb teeth”, was synthesized (see Fig. 3a).<sup>35</sup> The energy stored in stretched “springs” served as driving energy of shape recovery and the phase transition of “comb teeth” was regarded as the responsive switch. To meet the need of application at high temperature, a SMP of SPEEK (sulfonated polyether ether ketone) ionomer was prepared.<sup>36</sup> In this strategy, the  $T_g$  of this SMP was improved by the Coulomb force between metal ions and sulfonate anions, endowing a switch temperature from 224 to 288 °C. The higher the valence state of the metal ions is, the better shape memory behaviours and higher modulus this SMP has.

In general, it is hard for pure SMPs to satisfy the comprehensive requirements of different industries. Composite technology is the most effective mean in the functional modification of SMPs.<sup>37</sup> A SMP composite excited by microwave (MV) was prepared by introducing zero-dimension silicon carbide (SiC) into polyvinyl alcohol (PVA) matrix.<sup>38</sup> The rapid heating of this SMP composite was achieved *via* the MV-thermal conversion of SiC nanofiller, and thus the rapid shape recovery was also realized. The nonuniform dispersion of nanofillers in polymer matrices is a serious problem for the preparation of SMP composites with superior properties. By using zinc dimethacrylate (ZDMA) as a compatibilizer as well as *in situ* reinforcer, a SMP composite of ferroferric oxide ( $Fe_3O_4$ )/carboxylic styrene butadiene rubber (XSBR) with uniform dispersed  $Fe_3O_4$  particles and magnetic-thermal conversion was achieved.<sup>39</sup> The mechanical properties and  $T_g$  of this SMP composite were enhanced for the additional ionic cross-linked points generated by the bridging role of ZDMA between  $Fe_3O_4$  particles and XSBR matrix. The doping technology of SMPs by using conductive polymers is a feasible method to realize electro-thermal response of shape memory behaviours. A shape memory poly-lactic acid (PLA) microfiber membrane coated with a layer of

conductive polypyrrole was synthesized *via* combining electrostatic spinning with chemical vapor polymerization technologies, whose shape memory behaviour was based on the rapid electro-thermal response.<sup>40</sup> In addition to the visible shape memory behaviours, those on microscale can be also applied in surface topography of polymer.<sup>41</sup> In recent research, some SMPs with tunable wettability have been developed *via* controlling their surface microstructure through shape memory functions.<sup>42,43</sup>

**4.1.1.2 Temperature induced dissociation of non-chemical bonds.** The responsive switches based on non-chemical bonds have been applied in SMPs.<sup>44,45</sup> These SMPs have at least two types of cross-linking networks, physically network and chemically one. The former acts as the reversible phase, while the latter serves as the stable phase. The tannic acid (TA) can form intramolecular or intermolecular hydrogen bonds in some hydrogels. This hydrogen bond networks can be dissociated or recombined. A SMP hydrogel was synthesized *via* blending of PVA and TA. The stronger hydrogen bonds between PVA and TA acted as the cross-linkage structure for memorizing the permanent shape, while the weaker hydrogen bonds among PVA chains served as the responsive switch (see Fig. 3b).<sup>46–49</sup> Analogously, a shape memory polybenzoxazine film consisting of arylamine mannich-type linkage was synthesized, whose chemical structure is different from that of the conventional polybenzoxazines.<sup>50</sup> Mannich polybenzoxazines possessed more weak hydrogen bonds of  $-OH \cdots \pi$  that only existed at relatively lower temperatures and served as the responsive switch. Moreover, a SMP was prepared *via* crosslinking of polydimethylsiloxane and poly(ethylene glycol) diglycidyl ether.<sup>51</sup> When the temperature rose above 37 °C, the weak hydrogen bonds among molecular chains or chain segments were disengaged, which led to the improvement for mobility of molecular chains or chain segments, as well as the shape recovery. Another SMP hydrogel was developed by the polymerization the acrylamide in an elastin-like polypeptide (ELP) solution.<sup>52</sup> This SMP hydrogel, holding dual networks of a physically cross-linking ELP and a chemically cross-linking polyacrylamide (PAM), exhibited excellent shape memory function, wherein the hydrogen bonds among the ELP molecular chains acted as the responsive switch. From the discussion above, the design of SMPs based on hydrogen bonds mostly relies on three groups: pyridine, ureidopyrimidine, and carboxylic acids.

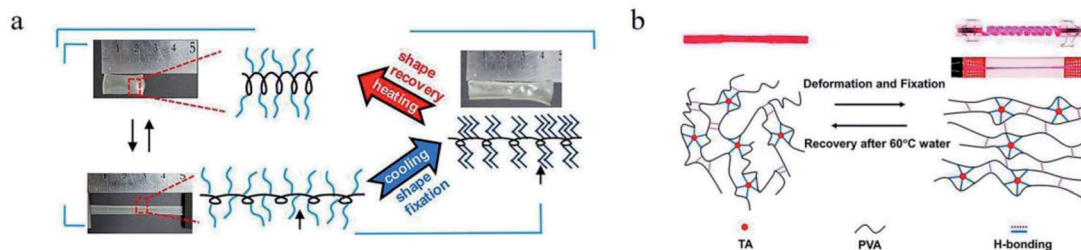


Fig. 3 (a) The shape memory process and mechanism of a SMP with spring-like main chains and comb teeth-like side chains.<sup>35</sup> Copyright 2019. Reproduced with permission from American Chemical Society. (b) Shape memory mechanism of the PVA–TA Hydrogel.<sup>46</sup> Copyright 2016. Reproduced with permission from American Chemical Society.



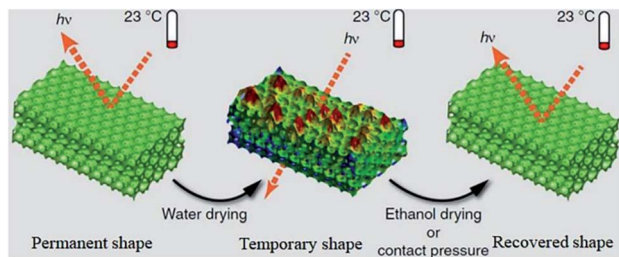


Fig. 4 A schematic illustration of the shape memory process of a SMP accompanied by reconfiguration of photonic crystals.<sup>57</sup> Copyright 2015. Reproduced with permission from Springer Nature.

Besides hydrogen bonds,  $\pi$ - $\pi$  interactions, weakly non-chemical bonds, can play important roles in shape memory properties by acting as the physical cross-linked points. A series of polyimides (PIs) with different chain structures were synthesized by a two-step method, showing a shape recovery ratio ( $R_r$ ) of above 93%.<sup>53</sup>

**4.1.2. Stress response.** The stress-responsive SMPs are those, whose responsive switches are triggered by stress. A SMP of poly-2-ureido-4[1H]-pyrimidone (UPy) was prepared,<sup>54-56</sup> whose temporary shape can be fixed through dimerization of unfolded UPy units, and shape recovery can be realized *via* dissociation of hydrogen bonds under stress. Fang *et al.*<sup>57</sup> synthesized a kind of stress-responsive photonic crystals basing on photocured copolymers of ethoxylated trimethylolpropane triacrylate (ETPTA) and polyethylene glycol diacrylate (PEGDA) oligomer (see Fig. 4). A disordered temporary shape can be achieved from a 3D ordered permanent shape at ambient conditions by evaporating water from the templated macropores. And the shape recovery accompanied by reconfiguration of photonic crystals can also be realized by a tiny contact pressure or even the surface tension gradient generated during solvent evaporating. They further studied a dynamic direct writing technology of ethoxylated trimethylolpropane triacrylate-*co*-polyethylene glycol diacrylate (ETPTA-*co*-PEGDA) copolymer by integrating the well-established templating nanofabrication and shape memory function.<sup>58</sup> The deformation and recovery process of this SMPs are attributed to the mutual transformation of the external energy provide by stress and the internal one stored in SMP system.

**4.1.3. Light response.** Light-responsive SMPs have attracted wide attention due to their unique advantages such as contactless and remote controlling, local application, and the ability free of surrounding intervention. They have an important application prospect in biomedical devices, soft drive machinery or wearable electronic devices. Most photoinduced shape memory behaviours are triggered by temperature thresholds, containing phase transition or dissociation of non-chemical bonds. Because the temperature variations of these SMPs are local and slight during shape memory progress, they are defined as light-responsive SMPs here.

**4.1.3.1 Light-induced phase transition.** The doping technology of SMPs by using light-thermal conversion fillers is a feasible method to realize light response of shape memory behaviours.<sup>59</sup> By using the reinforcing and light-thermal conversion functions of dopamine particles, Yang *et al.*<sup>60</sup> synthesized a SMP hydrogel with ultrafast near infrared ray (NIR) response. Plasmonic silver nanoparticles (AgNPs) as a class of light-thermal conversion nanofillers have also been applied in SMPs (see Fig. 5a).<sup>61</sup> In this system, the “nanorod-like” structure of cellulose nanocrystalline (CNC)-g-AgNPs is beneficial to the generation of plasmonic coupling between two individual NPs. Rare earth organic complex (REOC) with unique optical, electrical, magnetic and thermal functions, was blended with polymer to prepare a light-responsive REOC/SMP composite.<sup>62</sup> In this work, the existence of hydrogen bonds between nitrate groups of Sm (TTA) Phen (NO<sub>3</sub>)<sub>3</sub>(STPN) and carboxyl groups of cross-linked poly(methyl methacrylate-*co*-itaconic acid) (P(MMA-*co*-IA)) significantly improved the dispersion of STPN in polymer matrix. A SMP was developed by blending carbon dots and PVA.<sup>63</sup> The shape memory behaviours of this SMP could be quickly implemented by the melting process of PVA crystalline regions under 650 nm laser irradiation. In the above discussed SMPs, the light-thermal conversion fillers also offered additional physical cross-linked points, which improved the shape memory performance.

However, there are some drawbacks in doping technology of SMPs by using light-thermal conversion fillers due to the poor compatibility of light-thermal fillers in polymers. Recently, a kind of benzoxazine/epoxy (V-fa/ECO) copolymer was prepared from bio-based raw materials of vanillin, furfurylamine, paraformaldehyde, and epoxidized castor oil. Shape memory behaviours of this copolymer could be triggered by NIR

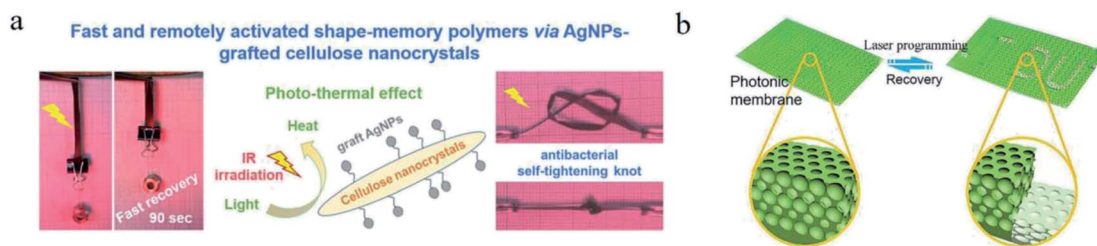


Fig. 5 (a) Digital images of SMP films actuated upon IR irradiation.<sup>61</sup> Copyright 2018. Reproduced with permission from American Chemical Society. (b) Schematic of shape memory process of photonic patterns.<sup>65</sup> Copyright 2019. Reproduced with permission from Royal Society of Chemistry.



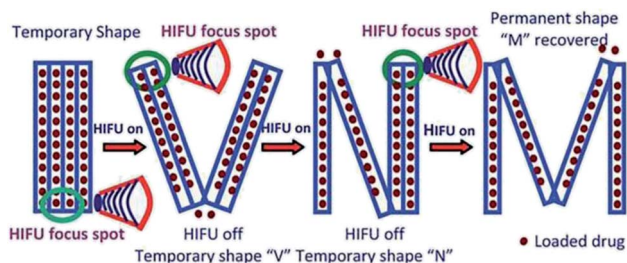


Fig. 6 Schematic illustration of the HIFU-enabled shape memory process and drug-controlled release.<sup>68</sup> Copyright 2012. Reproduced with permission from Royal Society of Chemistry.

without using any light-thermal fillers.<sup>64</sup> The mechanism was that light-thermal effect of  $\pi$ - $\pi$  interactions among polymer chains provided the energy for shape recovery and glass transition of the copolymer.

#### 4.1.3.2 Light-induced dissociation of the non-chemical bonds.

A photonic membrane with shape memory properties was prepared through integrating the copolymer of PEGDA-tetrafunctional aliphatic polyurethane acrylate (PUA) and  $\text{Cu}_2\text{S}$  nanoparticles.<sup>65,66</sup> The internal stress stored in this SMP could be released through light-thermal effect of  $\text{Cu}_2\text{S}$  nanoparticles, which improved the motion of the molecular segments in PEGDA-PUA copolymer and induced shape recovery (see Fig. 5b).

**4.1.4. Ultrasound response.** Ultrasound can penetrate much more deeply in the materials, triggering shape memory behaviours in a remote and controlled way. Basing on an ultrasound-responsive polyurethane (PU), Han *et al.*<sup>67</sup> studied a controlled drug release technology. Under high intensity focused ultrasound (HIFU), the shape recovery of the blend system of PU and drug brought about the enlargements of drug-to-body fluid contact and the drug diffusion coefficient. By using copolymer of methyl methacrylate and butyl acrylate, a similar research work has been done (see Fig. 6).<sup>68</sup> The shape memory mechanism was that ultrasonic effect rose the temperature of the SMPs above their  $T_g$ ,<sup>69</sup> leading to the trigger of their responsive switches.<sup>70</sup>

**4.1.5. Humidity response.** Thermal-responsive SMPs have always been the focus of research, but there are disadvantages for biomedical field. For example, if the responsive temperature

of SMPs is higher than the body temperature, it will damage cells and tissues. Humidity is regarded as a mild stimulus, which will be conducive to avoid the negative effects of direct heating.<sup>71,72</sup> A humidity-responsive SMP hydrogel based on polyacrylamide-chitosan (PAM-CS) was synthesized through free radical polymerization.<sup>73</sup> The temporary shape of this SMP hydrogel was fixed due to the formation of the physical microcrystalline crosslinking or chain entanglement as soaked in alkaline or saline solutions. These physical cross-linked networks could be dissociated by  $\text{H}_2\text{O}$  or  $\text{HCl}$  solution. Gu *et al.*<sup>74</sup> developed a humidity-responsive SMP hydrogel basing on PVA/silk hybrids (see Fig. 7). In this SMP hydrogel, the  $\beta$ -sheet crystals were regarded as network nodes to memory the permanent shape, while the hydrogen bonds within the amorphous regions of silks served as a switch for the reversible deformations. Some other humidity-responsive SMP composites were synthesized by introducing nanoparticles into the polymer matrices.<sup>75,76</sup> The nanoparticles were regarded as the additional physical cross-linked points to improve the shape memory properties of composite. The humidity-responsive shape memory effects (SMEs) are attributed to the plasticizing effect and the competitive hydrogen bonds. That is to say, the absorbed water in the composites can reduce the entanglement and the interaction of molecular chains, which will improve the movability of molecular chains and result in the shape recovery.

**4.1.6. Chemical reagent response.** Chemical reagent can also act as external stimuli to trigger the physically responsive switches of SMPs.<sup>77</sup> Chemical reagent responsive SMPs have been concerned due to their unique advantages such as excellent controllability and wide applied fields.

**4.1.6.1 Carbon dioxide ( $\text{CO}_2$ ) response.** Xu *et al.*<sup>78</sup> synthesized a  $\text{CO}_2$ -responsive hydrogel *via* copolymerization of 2-vinyl-4,6-diamino-1,3,5-triazine and *N,N*-dimethylacrylamide with polyethylene glycol diacrylates (see Fig. 8a). The physical cross-linkage generated from hydrogen bonds among diamino-triazine (DAT) groups served as a switch for the reversible deformation. The dissolved  $\text{CO}_2$  induced the protonation of DAT segments, leading to the dissociation of hydrogen bonds.

**4.1.6.2 pH response.** pH induced SMBs are fast and controllable. Li *et al.*<sup>79</sup> developed a pH-responsive SMP composite by blending of poly(ethylene glycol)-poly( $\epsilon$ -caprolactone)-polyurethane (PECU) and functionalized CNC with pyridine groups or carboxyl groups. The hydrogen bonds

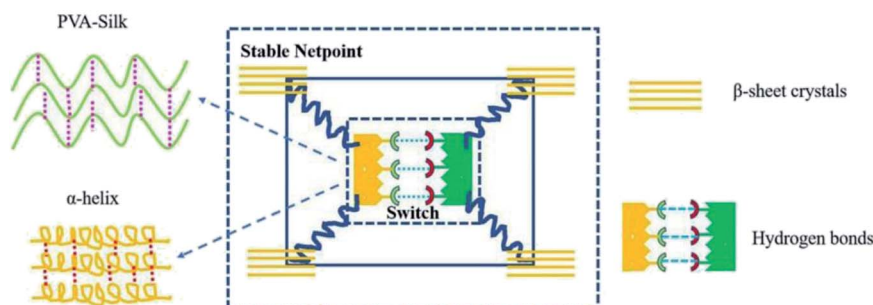


Fig. 7 Structure model of the shape memory PVA/silk hybrid.<sup>74</sup> Copyright 2019. Reproduced with permission from Elsevier.



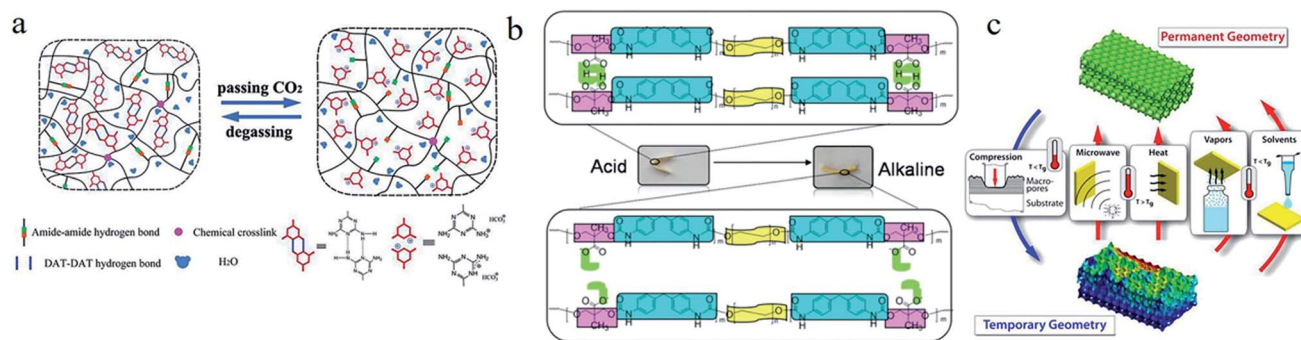


Fig. 8 (a) The CO<sub>2</sub>-triggered shape memory mechanism of a hydrogel with double hydrogen bonds.<sup>78</sup> Copyright 2015. Reproduced with permission from John Wiley & Sons Ltd. (b) Schematic route of the pH sensitive shape memory behavior of PEG-30%-MDI-DMPA.<sup>81</sup> Copyright 2016. Reproduced with permission from Royal Society of Chemistry. (c) A schematic diagram of the microstructural transitions between the 3D ordered permanent shape and disordered temporary shape of PU-based SMPs with solvent response.<sup>85</sup> Copyright 2017. Reproduced with permission from American Chemical Society.

between pyridine groups and hydroxyl groups were just the switch of this SMP composite, who were formed in alkali condition and weaken in acidic condition due to the protonation of pyridine group. A SMP hydrogel with both pH-response and temperature-response was developed by grafting dansyl groups (Dns) into the networks of polyacrylamide (PAAM).<sup>80</sup> In this work, cross-linked PAAM served as stable networks, while the hydrophobic association of Dns groups that could be modulated by pH or temperature was designed as the responsive switch. By using carboxylic dimerization as a switch, Song *et al.*<sup>81</sup> synthesized a pH-responsive SMP of polyethylene glycol-4,4'-diphenylmethane diisocyanate-dimethylol propionic acid (PEG-MDI-DMPA) (see Fig. 8b). The carboxylic dimers in polymer chains were dissociated under alkaline condition (pH 9) and rebuilt under acidic condition (pH 2). A SMP was prepared by cross-linking of  $\beta$ -cyclodextrin-modified alginate ( $\beta$ -CD-Alg) and diethylenetriamine (DETA)-modified alginate (DETA-Alg),<sup>82</sup> in which inclusion complexes (ICs) of  $\beta$ -CD and DETA served as a reversible phase, and the cross-linked alginate was regarded as a stable phase. It was shown that this SMP could be programmed into temporary shape at pH 11.5 and recover to its original shape at pH 7.

**4.1.6.3 Solvent response.** A hydrogel with solvent-responsive SMBs was synthesized by copolymerization of dopamine methacrylamide, *N,N'*-dimethylacrylamide, and *N,N'*-methylenebisacrylamide cross-linker.<sup>83</sup> In this hydrogel, hydrophobic association was used as responsive switch, who can be dissociated in dimethyl sulfoxide (DMSO) and rebuilt in H<sub>2</sub>O.

However, the catechol group was easy to be oxidized in aqueous solution, resulting in unstable mechanical properties of this hydrogel. By copolymerization of hydrophobic monomers, butyl methacrylate and methacrylic acid, in a mixed solvent of DMSO and H<sub>2</sub>O. Liao *et al.*<sup>84</sup> synthesized a series of solvent-responsive gels with stable mechanical properties. Fang and his cooperators<sup>85</sup> reported a solvent-responsive SMP with reconfigurable/rewritable macroporous photonic crystals (see Fig. 8c). Its shape recovery accompanied by reconfiguration of ordered photonic crystals could be triggered by multiple stimuli (*e.g.*, vapors, nonaqueous solvents, temperature, and MV

radiation).  $T_g$  of this SMP were decreased for the plasticizing effect of the solvent molecules permeating in polymer matrix.

## 4.2. SMPs based on chemical switches

SMPs based on chemical switches mainly refer the ones, whose responsive switches are based on reversible chemical reactions or reversible chemical bonds.

**4.2.1. Temperature response.** A polymer adhesive with dynamic disulfide bonds exhibited two levels of SMEs.<sup>86</sup> One was attributed to the reversible phase of the crystalline domains. The other was associated with the dynamic disulfide bonds. When heated to about 150 °C, the viscosity of the adhesive declined significantly for the temporary breaking of the dynamic disulfide bonds, which was benefit to the wetting of the adhesive on substrate. After cooled to room temperature, new disulfide bonds in the adhesive were built and a strong adhesion was achieved. Zhang *et al.*<sup>87</sup> synthesized a SMP that contained network nodes of Diels–Alder (DA) adducts. This SMP could undergo shape memory process in solid state at intermediate temperatures *via* breaking and recombining part of the DA adducts.

**4.2.2. Light response.** Coordination bonds provide a reversibility which can be used for the design of SMPs. Burnworth *et al.*<sup>88</sup> synthesized a metallosupramolecular SMPs that could be healed by light irradiation. Under UV irradiation, the dissociation of the multidentate ligand (Zn-Ns) led to a sharp drop in the viscosity of polymer, resulting in the selective healing of cracks.<sup>88,89</sup> Feng *et al.*<sup>90</sup> synthesized an organic cross-linked poly(acrylamide-*co*-*N*-vinylimidazole) hydrogel by using photoacid generator (PAG) as a trigger (see Fig. 9a). The temporary shape of this hydrogel could be fixed by the coordination bonds between the imidazole groups and metal ions. Under UV irradiation, PAG as proton donor could reduce the pH of this hydrogel below the acid dissociation constant ( $pK_a$ ) of imidazole groups, resulting in the disassociation of these coordination bonds, that is the reason of shape recovery. Bai *et al.*<sup>91</sup> synthesized a series of shape memory ionomers by blending PVA, polyacrylic acid (PAA) and bridge metal



ions (e.g.,  $\text{Fe}^{3+}$ ,  $\text{Cu}^{2+}$ ,  $\text{Co}^{2+}$ ), in which the responsive switches were the coordination bonds between metal ions and carboxyl groups. These coordination bonds could be temporarily decomposed by light-thermal conversion of metal ions themselves and rebuilt after the disappearance of NIR light.

In addition, photoinduced reversible chemical reactions can also be used in SMPs.<sup>92–97</sup> A dynamically cross-linked elastomer hybrid with self-healing and shape memory properties was synthesized by Shi *et al.*<sup>98</sup> In this hybrid, a reversible network was formed *via* DA reaction between macromolecular diene and carbon nanotubes (CNTs), and the reverse DA (rDA) reaction could be triggered by laser irradiation due to the light-thermal effect of CNTs. The network nodes between CNTs and furan-modified poly(styrene-*block*-butadiene-*block*-styrene) (SBS) could be rearranged for the liberation of SBS chains by rDA reaction, endowing the hybrids with self-healing ability. The combinations of DA reaction and rDA reaction could be used to reprogram the shape memory behaviours under laser irradiation or thermal treatment.

Banerjee *et al.*<sup>99</sup> synthesized a light responsive self-healing polymer by using low molecular weight tri-arm

polyisobutylene (PIB) with terminal groups of coumarin. The results showed that the reversible photodimerization of two adjacent coumarins could occur under UV irradiation of 365 nm, endowing this system with a stable macromolecular network. While under UV irradiation of 254 nm, the cross-linked PIB networks would be dissociated for the disaggregation of photodimers. Then the low viscosity PIB were flowing and filled the cut. Xie *et al.*<sup>100</sup> designed a SMP of poly(L-lactide)-poly(ethylene glycol)-(anthracene group) (PLA-PEG-A) copolymer (see Fig. 9b). It showed that the shape fixation of samples was realized by the photo-crosslinking, the photocycloaddition reaction of anthracene groups under UV light of 365 nm. Irradiated by UV light of 254 nm, part of the cyclized structures could be disassociated, leading to the decrease of the cross-linked density and the shape recovery. A special SMP has been synthesized by functionalizing the terminal groups of star-shaped PCL with 7-hydroxypropoxy-4-methylcoumarin.<sup>101</sup> This functionalized PCL could be cross-linked by the dimerization of coumarin groups under UV irradiation of 365 nm, and the cross-linked network could also be destroyed under UV irradiation of 256 nm, which was the basic reason for the light-responsive shape memory behaviour of this PCL. The *cis*-*trans* isomerization can be used as the switch of these SMPs. A swimming micro robot was designed by using the reversible process of light-responsive shape memory behaviours.<sup>102</sup> In this work, the *cis* isomers of azobenzene (azo) chromophores in the polymer films of paddles could be transformed into *trans* ones under UV irradiation and the opposite will be carried out under visible irradiation, which brought about the alternant bends of the paddles in two opposite directions and pushed forward the micro robot in liquid. Moreover, the moving direction of the micro robot could be guided by light source.  $T_g$  of polymer with azo groups could be altered by photoisomerization, which has also been applied in SMP (see Fig. 9c).<sup>103</sup> Under UV light irradiation (365 nm),  $T_g$  of this kind of SMPs is below room temperature for most of the azo groups are *cis* structure. While under visible light irradiation (530 nm),  $T_g$  is above room temperature for most of the azo groups are *trans* structure. Host-guest interactions between  $\alpha$ -cyclodextrin and azo groups could be reversible broken and recombination by *cis*-*trans* isomerization of azo. A shape memory film composed of azo-grafted poly(acrylate acid) (azo-PAA),  $\alpha$ -cyclodextrin modified alginate ( $\alpha$ -CD-Alg) and calcium chloride was synthesized.<sup>104</sup> In this SMP, the cross-linked network formed through network nodes—coordinate bonds of  $\text{Ca}^{2+}$ -carboxyl groups—served as the stable phase and host-guest interactions between  $\alpha$ -cyclodextrin and azo as the reversible switch. The SMP film could be programmed into temporary shapes as needed and recover to its permanent shape upon the stimuli of temperature.

**4.2.3. Ultrasound response.** Some chemical bonds, such as ester bond, disulfide bond, DA linkage, and coordinate bond,<sup>105</sup> can be temporally destroyed by ultrasound, which is an ideal strategy for SMPs. A kind of PU containing functional groups of DA adducts was synthesized, being of the functions of shape memory and self-healing under HIFU (see Fig. 10).<sup>106</sup> HIFU triggered rDA reactions and DA reshuffling reactions were the main reasons for the effective healing. Sijbesma *et al.*<sup>107,108</sup>

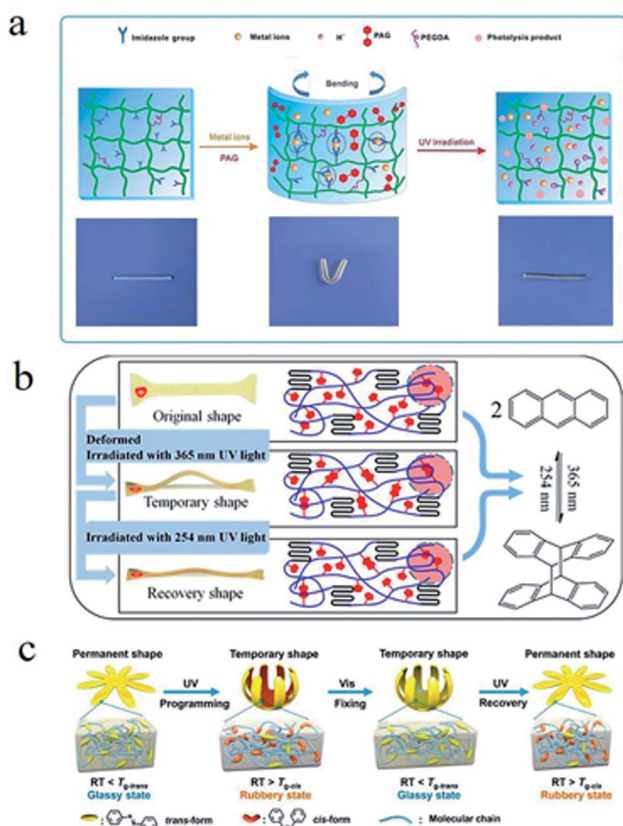


Fig. 9 (a) The UV light-triggered shape memory mechanism of an organic cross-linked poly(acrylamide-*co*-*N*-vinylimidazole) hydrogel.<sup>90</sup> Copyright 2015. Reproduced with permission from Royal Society of Chemistry. (b) The light-triggered shape memory process and mechanism of a PLA-PEG-A.<sup>100</sup> Copyright 2016. Reproduced with permission from American Chemical Society. (c) The shape memory process and mechanism of an SMP with azobenzene molecule.<sup>103</sup> Copyright 2018. Reproduced with permission from American Chemical Society.





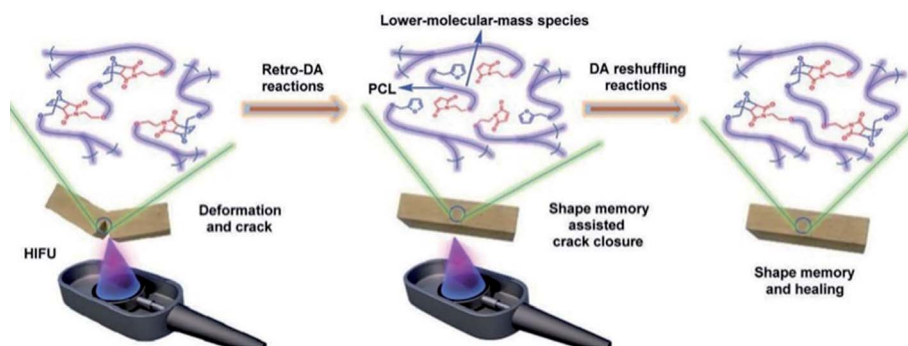


Fig. 10 A schematic diagram of the healing of a PCL-based PU with DA linkage assisted by HIFU-triggered shape memory.<sup>106</sup> Copyright 2014. Reproduced with permission from Royal Society of Chemistry.

synthesized a coordination polymer composed of low-molecular-weight diphenylphosphine telechelic poly(tetrahydrofuran) and palladium(II) dichloride. The palladium(II)-phosphine ligand could be destroyed under ultrasonic and reconstituted as the disappearance of ultrasonic, who can be used as a responsive switch of this SMP.

#### 4.2.4. Chemical reagent response

**4.2.4.1 Complexing agent response.** Coordination bonds of metal ions have been used as the switches in SMPs to generate additional cross-linkages for fixation of temporary shapes.<sup>109</sup> Some complexing agents can disrupt coordination bonds of polymer system by extracting metal ions, and these coordination bonds can be rebuilt when the metal ions are reincorporated.<sup>110</sup> For example, a copolymer hydrogel was prepared by

one-step photopolymerization of vinylimidazole (VI) and acrylonitrile (AN) in the presence of a polyethylene glycol-based cross-linker, named as PVI-AN hydrogel (see Fig. 11a).<sup>111</sup> The spiral shape of PVI-AN specimen could be fixed by the cross-linked network formed *via* network nodes of  $Zn^{2+}$ -imidazole coordination bonds. Once the hydrogel was transferred into EDTA solution, its cross-linked network would be disrupted for  $Zn^{2+}$  was captured by EDTA, inducing the realization of shape recovery. An interpenetrating network polymer of PAM and alginate was prepared by copolymerization of acrylamide (Am), host-guest cross-linker of azobenzene and  $\alpha$ -CD in an aqueous solution of alginate.<sup>112</sup> The temporary shape of this polymer could be fixed by  $Ca^{2+}$  and recover to its original shape *via* depriving  $Ca^{2+}$  by EDTA, due to the reversible coordination

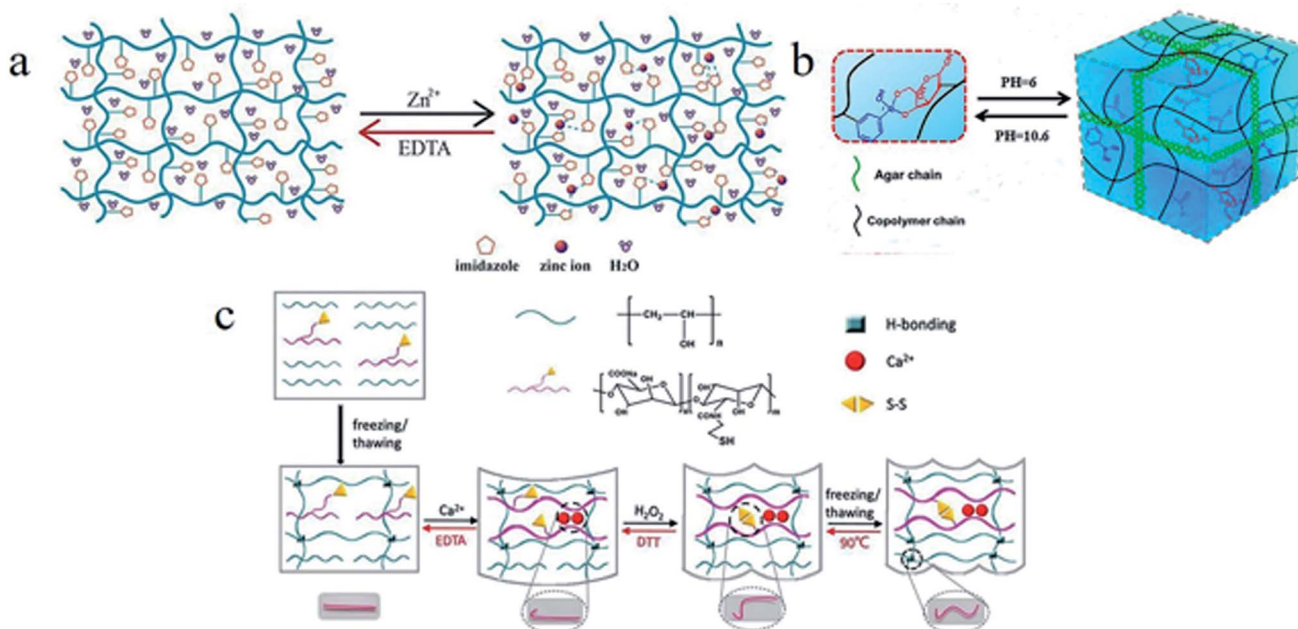


Fig. 11 (a) A schematic diagram of the EDTA-triggered shape memory mechanism of a PVI-AN hydrogel.<sup>111</sup> Copyright 2013. Reproduced with permission from Royal Society of Chemistry. (b) Mechanism of programmed shape memory process based on PBA-diol ester bonds.<sup>116</sup> Copyright 2017. Reproduced with permission from American Chemical Society. (c) A schematic diagram of the shape memory process and mechanism of a SMP hydrogel with three different types of reversible interactions, including  $Ca^{2+}$  ligand, a thiol-disulfide, and the hydrogen bonding.<sup>121</sup> Copyright 2018. Reproduced with permission from Royal Society of Chemistry.



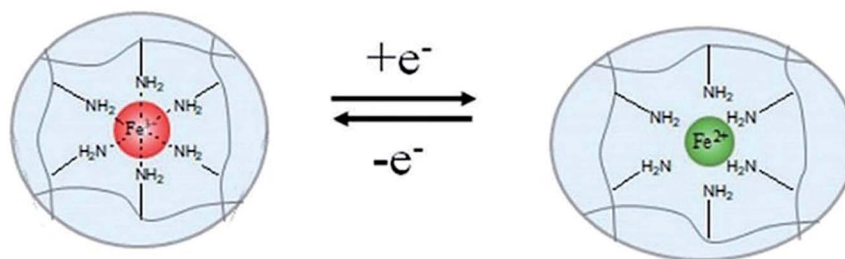


Fig. 12 Mechanism of shape memory effect and rapid reversible actuation of hydrogels with electrochemically controlled local metal ion coordination and crosslinking.<sup>123</sup> Copyright 2020. Reproduced with permission from Royal Society of Chemistry.

bonds of alginate–Ca<sup>2+</sup>. Zhang *et al.*<sup>113</sup> synthesized a similar SMP with rhodium–phosphine ligand. The rhodium–phosphine coordination bonds could be destroyed by triphenylphosphine solution and rebuilt in [RhCl(1,5-cyclooctadiene)]<sub>2</sub> solution. A SMP hydrogel with an interpenetrating network of physically cross-linked CMC and chemically cross-linked PAM could be fixed by chelation of CMC and Fe<sup>3+</sup>, and the original shape could be recovered by treatment of EDTA, oxalate or H<sub>2</sub>SO<sub>4</sub> solution for the chelation of CMC and Fe<sup>3+</sup> was destroyed.<sup>114</sup>

**4.2.4.2 pH response.** A SMP hydrogel with a responsive switch of reversible phenylboronic acid (PBA)–diol ester bonds was reported.<sup>115</sup> The temporary shape of this SMP hydrogel could be fixed in alkaline condition, and the shape recover was performed in acidic condition due to dissociation of the PBA–diol ester bonds. Moreover, Le *et al.*<sup>116</sup> fabricated a SMP hydrogel with triple responses that corresponded to stimuli of EDTA, pH, and temperature, respectively. In this work, shape fixation and shape restoration were associated with PBA–diol ester bonds, acrylic acid (AAc)–Fe<sup>3+</sup> coordination, and a physically cross-linked agar network (see Fig. 11b). A kind of hyaluronic acid-based injectable hydrogels with dual dynamic covalent bonds of acylhydrazone and disulfide linkages were synthesized.<sup>117</sup> These hydrogels displayed pH-responsive shape memory behaviours due to the reversible acylhydrazone bonds and the osmotic driving for the concentration gradient between the network and surroundings. A hydrogel of the copolymer of acrylamide, acrylic acid, agar, and PVA exhibited multiple shape memory behaviours based on the coordination bonds of carboxy–Fe<sup>3+</sup>, borate ester bonds, and hydrogen bonds. pH response of this hydrogel was associated with dynamic borate ester bonds.<sup>118</sup> A hydrogel was prepared by copolymerization of acrylamide, oxidized sodium alginate (OSA), and PVA, whose multiple-shape memory behaviours were based on imine, coordination, and borate bonds.<sup>119</sup> A temporary shape could be fixed by a treatment of borax solution for borate bonds were formed, and original shape could be recovered by soaking in acetic acid solutions for the borate ester bonds were destroyed.

**4.2.4.3 Redox agent response.** By integrating DNA/bipyridinium dithienylethene (DTE) and CMC modified with dopamine and self-complementary nucleic acid, Li *et al.*<sup>120</sup> prepared a multiple-responsive hydrogel. Shape memory behaviours of this hydrogel could be triggered by crown ether

(CE), or visible light, or redox agent. In this hydrogel, the donor–acceptor interactions of dopamine/DTE would be dissociated by oxidation of sodium persulfate, and recombined by reduction of vitamin C. Tang *et al.*<sup>121</sup> synthesized another multiple-responsive SMP hydrogel that contained three types of reversible interactions (Ca<sup>2+</sup> complexation, disulfide bond, hydrogen bond) by blending cysteamine-grafted alginate (SA-SH) and PVA (Fig. 11c). A temporary shape of hydrogel could be fixed by using a treatment of hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>), and the permanent shape could be recovered by introducing dithiothreitol (DTT). Wang *et al.*<sup>122</sup> reported a shape memory blend of poly(ethylene oxide) (PEO), poly(acrylic acid) (PAA), and TA. The temporary shape of this blend could be fixed *via* the ionic cross-linkage of Cu<sup>2+</sup>–carboxylate coordination bonds. And the shape recovery could be performed for the breaking of Cu<sup>2+</sup>–carboxylate coordination bonds induced by oxidation of sodium metabisulfite. Recently, some scholars have realized shape memory behaviours by means of electric redox. Rapid shape memory behaviours of a hydrogel device in air were realized by using electro-redox reaction of Fe<sup>3+</sup> and Fe<sup>2+</sup>, where the reversible coordination of Fe<sup>3+</sup> ions and amine groups served as a responsive switch (see Fig. 12).<sup>123</sup>

## 5. Conclusion and prospect

Based on deep analysis in the developments of SMPs, it is indicated that the nature of the responsive switch in SMPs is physical interaction or chemical one among polymer molecular chains or chain segments. For better understanding the mechanisms of their shape memory behaviours, the SMPs are divided into two kinds (one based on physical switches and the other on chemical switches) and in detail discussed in this paper, from of which it is shows that a lot of works have been conducted to improve the performance of SMPs and develop new types of SMPs. Although many shape memory mechanisms are deeply understood and a wide variety of SMPs have been developed, in order to satisfy the application requirements of intelligent materials, there is a long way to go. The SMPs with these properties such as high sensitivity, good responsive accuracy, multiple responsive switches, multifunctional responsive switches, and excellent mechanical characteristics, are needed in the future.

The design and development of new types of responsive switches may become an important part of research in SMPs in



future. The development of responsive switches with high-precision, quantifiable and controllable responses is getting attention for the requirement of intelligent control in intelligent materials. Moreover, the multifunctional responsive switches that mean a switch can respond to different stimuli will be greatly developed. The organisms in nature can sensitively and accurately respond to the external stimuli for their precisely and artfully designed structures. Therefore, precise control of the composition and structure of materials in micro scale or even molecular level is another important part of research in SMPs. With the development of material design (e.g., material computation and design combined with computer techniques) and preparation technology (e.g., 4D printing technology), this goal will be achieved gradually.

Although SMPs cannot rival the response properties of organisms in nature, many characters of the former such as service behaviors in high or low temperature, vacuum, high pressure, and chemical environments are better than that of the latter. In the future, SMPs as a kind of intelligent materials will be greatly developed and widely applied.

## Author contributions

Lide Yang: Investigation, Conceptualization, Methodology, Writing – Original draft preparation. Jiankun Lou: Investigation. Jianmin Yuan: Conceptualization, Methodology, Writing – Reviewing and Editing, Supervision, Validation. Jianru Deng: Methodology, Writing – Reviewing and Editing, Supervision, Validation.

## Conflicts of interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

## Acknowledgements

We gratefully acknowledge the valuable cooperation of professor Shengpei Su and the members of his laboratory (Hunan Normal University, China).

## Notes and references

- 1 A. Cohadesa, N. Hostettlera, M. Paucharda, C. J. G. Plummer and V. Michaud, *Compos. Sci. Technol.*, 2018, **161**, 22–31.
- 2 H. A. Rauch, Y. Chen, K. An and H. Z. Yu, *Acta Mater.*, 2019, **168**, 362–375.
- 3 W. Zhao, Z. Huang, L. Liu, W. Wang, J. Leng and Y. Liu, *Compos. Sci. Technol.*, 2020, **203**, 108563.
- 4 J. Hu, Y. Zhu, H. Huang and L. Jing, *Prog. Polym. Sci.*, 2012, **37**, 1720–1763.
- 5 Y. Xia, Y. He, F. Zhang, Y. Liu and J. Leng, *Adv. Mater.*, 2021, **33**, 2000713.
- 6 F. G. Bonifacich, O. A. Lambri, V. Recarte, V. Sánchez-Alarcos and J. I. Pérez-Landazábal, *Compos. Sci. Technol.*, 2021, **201**, 108538.
- 7 D. Ratna and J. Karger-Kocsis, *J. Mater. Sci.*, 2008, **43**, 254–269.
- 8 T. Xie, *Polymer*, 2011, **52**, 4985–5000.
- 9 S. Yang, Y. He, Y. Liu and J. Leng, *J. Mater. Chem. C*, 2020, **8**, 303–309.
- 10 Z. Cheng, D. Zhang, X. Luo, H. Lai, Y. Liu and L. Jiang, *Adv. Mater.*, 2020, **33**, 2001718.
- 11 X. Kuang, D. J. Roach, J. Wu, C. M. Hamel, Z. Ding, T. Wang, M. L. Dunn and H. J. Qi, *Adv. Funct. Mater.*, 2019, **29**, 1805290.1–1805290.23.
- 12 C. Lin, J. X. Lv, Y. S. Li and F. H. Zhang, *Adv. Funct. Mater.*, 2019, **29**, 201906569.
- 13 M. Zare, M. P. Prabhakaran, N. Parvin and S. Ramakrishna, *Chem. Eng. J.*, 2019, **374**, 706–720.
- 14 J. Fan and G. Li, *Nat. Commun.*, 2018, **9**, 1–8.
- 15 K. J. Wang, Y. G. Jia and X. X. Zhu, *Macromolecules*, 2017, **50**, 8570–8579.
- 16 J. Hu and S. Chen, *J. Mater. Chem.*, 2010, **20**, 3346–3355.
- 17 X. L. Wu, W. M. Huang, Y. Zhao and D. Zheng, *Polymers*, 2013, **5**, 1169–1202.
- 18 M. D. Hager, S. Bode, C. Weber and U. S. Schubert, *Prog. Polym. Sci.*, 2015, **49**, 3–33.
- 19 E. Wang, Y. Wu, M. Z. Islam, Y. Dong, Y. Zhu, F. Liu, Y. Fu, Z. Xu and N. Hu, *Mater. Lett.*, 2019, **238**, 54–57.
- 20 Z. C. Jiang, Y. Y. Xiao, Y. Kang, J. Bang and S. Zhang, *ACS Appl. Mater. Interfaces*, 2017, **9**, 20276–20293.
- 21 Y. Shi, M. Yoonessi and R. A. Weiss, *Macromolecules*, 2013, **46**, 4160–4167.
- 22 S. B. Zhou, L. X. Lu, T. Tian and S. S. Wu, *Polym. Chem.*, 2019, **10**, 1920–1929.
- 23 L. Ling, J. Li, G. Zhang, R. Sun and C. P. Wong, *Macromol. Res.*, 2018, **26**, 365–373.
- 24 S. Ji, W. Cao, Y. Yu and H. Xu, *Adv. Mater.*, 2015, **27**, 7740–7745.
- 25 W. N. Du, Y. Jin, S. Q. Lai, L. J. Shi, W. H. Fan and J. G. Pan, *Polymer*, 2018, **158**, 120–129.
- 26 M. M. Perera and N. Ayres, *Polym. Chem.*, 2020, **11**, 1410–1423.
- 27 H. Otsuka, K. Aotani, Y. Higaki and A. Takahara, *Chem. Commun.*, 2002, **23**, 2838–2839.
- 28 M. Abdallah, C. Yoshikawa, M. Hearn and G. P. Simon, *Macromolecules*, 2019, **52**, 2446–2455.
- 29 O. R. Cromwell, J. Chung and Z. Guan, *J. Am. Chem. Soc.*, 2015, **137**, 6492–6495.
- 30 M. J. Fan, J. L. Liu and X. Y. Li, *Ind. Eng. Chem. Res.*, 2014, **53**, 16156–16163.
- 31 J. R. Kumpfer and S. J. Rowan, *J. Am. Chem. Soc.*, 2011, **133**, 12866–12874.
- 32 M. Li, S. Fu, L. A. Lucia and Y. Wang, *Compos. Sci. Technol.*, 2020, **199**, 108371.
- 33 Z. J. Ding, L. Yuan, G. Z. Liang and A. J. Gu, *J. Mater. Chem. A*, 2019, **7**, 9736–9747.
- 34 M. Ebara, K. Uto, N. Idota, J. M. Hoffman and T. Aoyagi, *Adv. Mater.*, 2012, **24**, 273–278.



- 35 S. H. Kang, T. Aoki and G. Kwak, *Macromolecules*, 2019, **52**, 7984–7993.
- 36 Y. Shi and R. A. Weiss, *Macromolecules*, 2014, **47**, 1732–1740.
- 37 T. Y. Yang, M. Wang, F. Jia, X. Y. Ren and G. H. Gao, *J. Mater. Chem. C*, 2020, **8**, 2326–2335.
- 38 H. Du, Z. Song, J. Wang, Z. Liang, Y. Shen and F. You, *Sens. Actuators, A*, 2015, **228**, 1–8.
- 39 C. Liu, J. Huang, D. Yuan and Y. Chen, *Ind. Eng. Chem. Res.*, 2018, **57**, 14527–14534.
- 40 F. H. Zhang, Y. L. Xia, L. L. Wang, L. W. Liu, Y. J. Liu and J. S. Leng, *ACS Appl. Mater. Interfaces*, 2018, **10**, 35526–35532.
- 41 S. A. Turner, J. Zhou, S. S. Sheiko and V. S. Ashby, *ACS Appl. Mater. Interfaces*, 2014, **6**, 8017–8021.
- 42 C. Li, Y. Jiao, X. Lv, S. Wu and J. Chu, *ACS Appl. Mater. Interfaces*, 2020, **12**, 13464–13472.
- 43 H. Y. Zhang, H. Lai, Z. J. Cheng, D. J. Zhang, P. C. Liu, Y. F. Li and Y. Y. Liu, *Appl. Surf. Sci.*, 2020, **525**, 146525.
- 44 F. Song, Z. Li, P. Jia, M. Zhang, C. Bo, G. Feng, L. Hu and Y. Zhou, *J. Mater. Chem. A*, 2019, **7**, 13400–13410.
- 45 D. Liu, W. C. Nie, Z. B. Wen, C. J. Fan, W. X. Xiao, L. Bei, X. J. Lin, K. K. Yang and Y. Z. Wang, *ACS Macro Lett.*, 2018, **7**, 705–710.
- 46 Y. N. Chen, L. Peng, T. Q. Liu, Y. X. Wang, S. J. Shi and H. L. Wang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 27199–27206.
- 47 L. Yang, Z. Wang, G. Fei and H. Xia, *Macromol. Rapid Commun.*, 2017, **38**, 1700421.
- 48 W. Peng, G. Zhang, Q. Zhao and T. Xie, *Adv. Mater.*, 2021, 2102473.
- 49 S. Yang, Y. Zhang, T. Wang, W. Sun and Z. Tong, *ACS Appl. Mater. Interfaces*, 2020, **12**, 46701–46709.
- 50 S. Zhang, Q. Ran, Q. Fu and G. Yi, *Macromolecules*, 2018, **51**, 6561–6570.
- 51 H. Y. Lai, H. Q. Wang, J. C. Lai and C. H. Li, *Molecules*, 2019, **24**, 3224.
- 52 Y. Zhang, M. S. Desai, T. Wang and S. W. Lee, *Biomacromolecules*, 2020, **21**, 1149–1156.
- 53 Q. Wu, Y. Bai, Y. Chen, J. Ju, F. Zhang and T. Wang, *J. Mater. Chem. A*, 2015, **3**, 352–359.
- 54 R. J. Wojtecki, M. A. Meador and S. J. Rowan, *Nat. Mater.*, 2011, **10**, 14.
- 55 A. M. Kushner, J. D. Vossler, G. A. Williams and Z. Guan, *J. Am. Chem. Soc.*, 2009, **131**, 8766–8768.
- 56 Z. V. Guan, J. T. Roland, J. Z. Bai, S. X. Ma, T. M. Mcintire and M. Nguyen, *J. Am. Chem. Soc.*, 2004, **126**, 2058–2065.
- 57 Y. Fang, Y. Ni, S. Y. Leo, C. Taylor, V. Basile and P. Jiang, *Nat. Commun.*, 2015, **6**, 7416.
- 58 Y. Fang, Y. Ni, S. Y. Leo, B. Wang, V. Basile, C. Taylor and P. Jiang, *ACS Appl. Mater. Interfaces*, 2015, **7**, 23650–23659.
- 59 H. Koerner, G. Price, N. A. Pearce, M. Alexander and R. A. Vaia, *Nat. Mater.*, 2004, **3**, 115–120.
- 60 L. Yang, Z. Wang, G. Fei and H. Xia, *Macromol. Rapid Commun.*, 2017, **38**, 1700421.
- 61 A. Toncheva, F. Khelifa, Y. Paint, V. Michel, P. Lambert, P. Dubois and J. M. Raquez, *ACS Appl. Mater. Interfaces*, 2018, **10**, 29933–29942.
- 62 H. N. Wang, L. Fang, Z. Zhang, J. Epaarachchi, L. Li, X. Hu, C. Lu and Z. Xu, *Composites, Part A*, 2019, **125**, 105525.
- 63 S. Wu, W. Li, Y. Q. Sun, X. L. Pang, X. J. Zhang, J. L. Zhuang, H. R. Zhang, C. F. Hu, B. F. Lei and Y. L. Liu, *J. Mater. Chem. C*, 2020, **8**, 8935.
- 64 L. Amornkitbamrung, S. Srisaard, C. Jubsilp, C. W. Bielawski and S. Rimdusit, *Polymer*, 2020, **209**, 122986.
- 65 P. N. Chang, W. B. Niu, L. C. Qu and S. F. Zhang, *J. Mater. Chem. C*, 2019, **7**, 1896–1903.
- 66 M. Quan, B. Yang, J. Wang, H. Yu and X. Cao, *ACS Appl. Mater. Interfaces*, 2018, **10**(4), 4243–4249.
- 67 J. Han, G. Fei, G. Li and H. Xia, *Macromol. Chem. Phys.*, 2013, **214**, 1195–1203.
- 68 G. Li, G. Fei, H. Xia, J. Han and Y. Zhao, *J. Mater. Chem.*, 2012, **22**, 7692–7696.
- 69 B. Liu, H. Xia, G. Fei, L. Guo and W. Fan, *Macromol. Chem. Phys.*, 2013, **214**, 2519–2527.
- 70 M. Bao, Q. Zhou, W. Dong, X. Lou and Y. Zhang, *Biomacromolecules*, 2013, **14**, 1971–1979.
- 71 Q. Yin, S. Tu, M. Chen and L. Wu, *ACS Appl. Polym. Mater.*, 2020, **2**, 2858–2866.
- 72 G. Li, T. Gao, G. Fan, Z. T. Liu and Y. Zhao, *ACS Appl. Mater. Interfaces*, 2020, **12**, 6407–6418.
- 73 H. Xiao, C. X. Ma, X. X. Le and L. Wang, *Polymers*, 2017, **9**, 138.
- 74 L. Gu, Y. Z. Jiang and J. L. Hu, *Mater. Today: Proc.*, 2019, **16**, 1491–1496.
- 75 X. Qi, X. Yao, S. Deng, T. N. Zhou and Q. Fu, *J. Mater. Chem. A*, 2014, **2**, 2240–2249.
- 76 G. H. Yang, X. J. Wan, Y. J. Liu, L. Rui, Y. K. Su, X. R. Zeng and J. N. Tang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 34744–34754.
- 77 S. S. Song, R. H. Dong, D. Wang and A. X. Song, *Soft Matter*, 2013, **9**, 4209–4218.
- 78 B. Xu, Y. Y. Zhang and W. G. Liu, *Macromol. Rapid Commun.*, 2015, **36**, 1585–1591.
- 79 Y. Li, H. M. Chen, D. Liu, W. X. Wang, Ye. Liu and S. B. Zhou, *ACS Appl. Mater. Interfaces*, 2015, **7**, 12988–12999.
- 80 X. L. Gong, Y. Y. Xiao, M. Pan, K. Yang, B. J. Li and S. Zhang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 27432–27437.
- 81 Q. J. Song, H. M. Chen, S. B. Zhou, K. Q. Zhao, B. Q. Wang and P. Hu, *Polym. Chem.*, 2016, **7**, 1739–1746.
- 82 X. J. Han, Z. Q. Dong, M. M. Fan, Y. Liu, J. H. Li, Y. F. Wang, Q. J. Yuan, B. J. Li and S. Zhang, *Macromol. Rapid Commun.*, 2012, **33**, 1055–1060.
- 83 J. Huang, J. Liao, T. Wang, W. Sun and Z. Tong, *Soft Matter*, 2018, **14**, 2500–2507.
- 84 J. X. Liao, J. H. Huang, S. R. Yang, X. L. Wang, T. Wang, W. X. Sun and T. Zhen, *ACS Appl. Polym. Mater.*, 2019, **1**, 2703–2712.
- 85 Y. Fang, S. Y. Leo, Y. L. Ni, J. Y. Wang and B. C. Wang, *ACS Appl. Mater. Interfaces*, 2017, **9**, 5457–5467.
- 86 B. T. Michal, E. J. Spencer and S. J. Rowan, *ACS Appl. Mater. Interfaces*, 2016, **8**, 11041–11049.



- 87 G. Zhang, Q. Zhao, L. Yang, W. Zou, X. Xi and T. Xie, *ACS Macro Lett.*, 2016, **5**, 805–808.
- 88 M. Burnworth, L. Tang, J. R. Kumpfer, A. J. Duncan, F. L. Beyer, G. L. Fiore, S. J. Rowan and C. Weder, *Nature*, 2011, **472**, 334–337.
- 89 D. Habault, H. Zhang and Y. Zhao, *Chem. Soc. Rev.*, 2013, **42**, 7244–7256.
- 90 W. Feng, W. F. Zhou, S. D. Zhang, Y. J. Fan, Y. Akram and H. Y. Yang, *RSC Adv.*, 2015, **5**, 81784–81789.
- 91 Y. K. Bai, J. W. Zhang, D. D. Wen and B. Yuan, *J. Mater. Chem. A*, 2019, **7**, 20723–20732.
- 92 S. Kobatake, S. Takami, H. Muto, T. Ishikawa and M. Irie, *Nature*, 2007, **446**, 778–781.
- 93 Y. Zheng, M. Micic, S. V. Mello, M. Mabrouki, F. M. Andreopoulos, V. Konka, S. M. Pham and R. M. Leblanc, *Macromolecules*, 2002, **35**, 5228–5234.
- 94 L. Wang, X. F. Yang, H. M. Chen, T. Gong, W. B. Li, G. Yang and S. B. Zhou, *ACS Appl. Mater. Interfaces*, 2013, **5**, 10520–10528.
- 95 H. N. Van and P. F. E. Du, *Macromolecules*, 2018, **51**, 3405–3414.
- 96 H. Y. Jiang, S. Kelch and A. Lendlein, *Adv. Mater.*, 2006, **18**, 1471–1475.
- 97 G. Stoychev, A. Kirillova and L. Ionov, *Adv. Opt. Mater.*, 2019, **7**, 1900067.
- 98 J. Bai and Z. Shi, *ACS Appl. Mater. Interfaces*, 2017, **9**, 27213–27222.
- 99 S. Banerjee, R. Tripathy, D. Cozzens, T. Nagy, S. Keki, M. Zsuga and R. Faust, *ACS Appl. Mater. Interfaces*, 2015, **7**, 2064–2072.
- 100 H. Xie, M. J. He, X. Y. Deng, L. Du, C. J. Fan, K. K. Yang and Y. Z. Wang, *ACS Appl. Mater. Interfaces*, 2016, **8**, 9431–9439.
- 101 T. Defize, J. M. Thomassin, H. Ottevaere, C. Malherbe, G. Eppe, R. Jellali, M. Alexandre, C. Jérôme and R. Raphaël, *Macromolecules*, 2018, **52**, 444–456.
- 102 C. Huang, J. Lv, X. Tian, Y. Wang, Y. Yu and L. Jie, *Sci. Rep.*, 2015, **5**, 17414.
- 103 X. Zhang, C. Y. Zhu, B. Xu, Q. Lang, L. Qin, J. Wei and Y. L. Yu, *ACS Appl. Mater. Interfaces*, 2019, **11**, 46212–46218.
- 104 M. Pan, Q. Yuan, X. L. Gong, S. Zhang and B. J. Li, *Macromol. Rapid Commun.*, 2016, **37**, 433–438.
- 105 X. Q. Li, Z. H. Wang and H. S. Xia, *Front. Chem.*, 2019, **7**, 59.
- 106 X. Lu, G. Fei, H. Xia and Y. Zhao, *J. Mater. Chem. A*, 2014, **2**, 16051–16060.
- 107 J. M. J. Paulusse and R. P. Sijbesma, *Angew. Chem., Int. Ed.*, 2004, **43**, 4460–4462.
- 108 J. M. J. Paulusse, J. P. J. Huijbers and R. P. Sijbesma, *Chem.–Eur. J.*, 2006, **12**, 4928–4934.
- 109 H. Xiao, L. Wei, L. X. Xia, C. X. Ma, W. L. Zhao, J. Zheng, J. W. Zhang, Y. J. Huang and T. Chen, *Chem. Commun.*, 2016, **52**, 13292–13295.
- 110 J. Xu, Y. He, Y. T. Li and Q. S. Ye, *Mater. Lett.*, 2020, **260**, 126968.
- 111 W. Nan, W. Wang, H. Gao, W. Liu and W. Wang, *Soft Matter*, 2013, **9**, 132–137.
- 112 M. Xie, C. Wu, C. Chen, Y. Liu and C. Zhao, *Polym. Chem.*, 2019, **10**, 4852–4858.
- 113 P. F. Zhang, M. Behl, X. Z. Peng, M. Balk and A. Lendlein, *Chem. Mater.*, 2019, **31**, 5402–5407.
- 114 N. Li, G. Chen, W. Chen, J. Huang, J. Tian, X. Wan, M. He and H. Zhang, *Carbohydr. Polym.*, 2017, **178**, 159–165.
- 115 H. Meng, J. Zheng, X. Wen, Z. Cai, J. Zhang and T. Chen, *Macromol. Rapid Commun.*, 2015, **36**, 533–537.
- 116 X. Le, W. Lu, H. Xiao, L. Wang, C. Ma, J. Zhang, Y. Huang and T. Chen, *ACS Appl. Mater. Interfaces*, 2017, **9**, 9038–9044.
- 117 Y. Xu, G. G. Lu, M. Y. Chen, P. L. Wang, Z. L. Li, X. W. Han, J. Liang, Y. Sun, Y. J. Fan and X. D. Zhang, *Carbohydr. Polym.*, 2020, **250**, 116979.
- 118 M. Wang, J. P. Zhuge, C. Q. Li, L. B. Jiang and H. Yang, *Iran. Polym. J.*, 2020, **29**, 569–579.
- 119 M. Wang, B. Mo, B. Chen, L. Jiang and H. Yang, *Colloid Polym. Sci.*, 2020, **298**, 285–291.
- 120 Z. Li, G. Davidson-Rozenfeld, M. Vázquez-González, M. Fadeev, M. Zhang, H. Tian and I. Willner, *J. Am. Chem. Soc.*, 2018, **140**, 17691–17701.
- 121 L. Y. Tang, L. F. Wen, S. P. Xu, P. H. Pi and X. F. Wen, *Chem. Commun.*, 2018, **54**, 8084–8087.
- 122 S. Wang, G. Li, Z. W. Liu and Y. Zhao, *ACS Appl. Mater. Interfaces*, 2019, **11**, 30308–30316.
- 123 Y. Cong, S. Liu, F. Wu, H. Zhang and J. Fu, *J. Mater. Chem. B*, 2020, **8**, 9679–9685.

