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## Tuning the optical bandgap and piezoresistance in iridium-based molecular semiconductors through ligand modification†

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Square-planar  $d^8$  metal complexes are known to stack with short metal–metal distances in the solid-state, forming linear molecular chains with conductive pathways that can be enhanced under pressure. Although the influence of the metallic centre on the behaviour of these materials has been previously studied, the role and significance of ligand choice has received less focus. Here, we study the relationship between the structural, optical and conductive properties of a series of  $d^8$  iridium dicarbonyl complexes with different  $\beta$ -diketonate ligands using a combination of experimental and computational methods. Our results show that ligand choice contributes significantly to the optical transitions of the molecules in solution by lowering the LUMO energy for complexes with  $\pi$ -conjugation or electronegative atoms. We also show that ligand choice is a pathway for band-structure tuning in the molecular crystal through ligand size selection and associated structural packing, with complexes packing in linear metal–metal stacks exhibiting a smaller optical bandgap in the solid state. With pressure-dependent measurements, we confirm that that favourable metal–metal stacking in the solid obtained by appropriate ligand choice leads to higher conductivity at lower pressures. Our results provide insight for the design and application of  $d^8$  metal complexes in optoelectronic devices and the development of future molecular materials.

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## Introduction

Since the discovery of semiconducting behaviour in phthalocyanines and condensed aromatic hydrocarbons in the 1950s,<sup>1,2</sup> the field of molecular electronics has seen huge advances, from metallic conductivity in Krogman's salt ( $K_2Pt(CN)_4Br_{0.3}$ ) and charge transfer salt TTF-TCNQ,<sup>3,4</sup> to superconductivity.<sup>5</sup> Metal complexes in particular have shown a wide range of properties, with examples of Dirac electron systems,<sup>6</sup> Mott insulators,<sup>7</sup> and superconductivity.<sup>8</sup> Square-planar  $d^8$  metal complexes are of particular interest, due to their propensity to form linear stacks with short metal–metal

distances.<sup>9</sup> These strong intermolecular interactions can provide a conductive pathway which can be enhanced under the application of pressure; for example, square planar platinum complexes  $Pt(bqd)_2$  ( $bqd$  = benzoquinonedioximato) and  $Pt(dmg)_2$  ( $dmg$  = dimethylglyoximato) undergo a 4-fold and 15-fold order of magnitude increase in conductivity on application of pressure, becoming metallic at 0.6–1.4 and 5 GPa, respectively.<sup>10–13</sup> This behaviour originates from the increased interaction of  $Pt d_{z^2}$  and ligand  $\pi$ -orbitals along the stacking axis, which is highly compressible under pressure.<sup>14</sup> Piezoresistive behaviour has also been observed for other stacked  $d_8$  metal complexes, including Magnus' Green salt  $Pt(NH_3)_4PtCl_4$ , and  $Ir(CO)_2acac$  ( $acac$  = acetylacetonate) which undergo 1.5-fold and 5-fold order of magnitude changes in conductivity, respectively.<sup>15</sup>

While short metal–metal distances are key to this piezoresistive behaviour, there has been relatively little focus on the role played by the ligand. In this work, we investigate the impact of ligand choice on the molecular and solid-state optical and electronic properties of iridium dicarbonyl  $\beta$ -diketonate compounds (including  $Ir(CO)_2acac$ ). We present the synthesis of four  $Ir(CO)_2L$  ( $L$  = ligand) complexes and compare their molecular properties with support from (time-dependent)

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density functional theory (TD-)DFT calculations. We then present their crystal structures and ability to form thin films, and investigate their solid-state optical and electronic properties through diffuse reflectance measurements with support from solid-state band structure calculations. We then discuss the impact of ligand choice on the solid-state structure and therefore on the electronic and piezoresistive properties.

## Results and discussion

### Synthesis

The synthesis scheme for the  $\text{Ir}(\text{CO})_2(\beta\text{-diketonate})$  complexes is shown in Fig. 1a. The synthesis of  $\text{Ir}(\text{CO})_2\text{tfpb}$  (tfpb = benzoyltrifluoroacetone) is presented for the first time; complexes  $\text{Ir}(\text{CO})_2\text{dbm}$  (dbm = dibenzoylmethane) and  $\text{Ir}(\text{CO})_2\text{tfaa}$  (tfaa = trifluoroacetylacetone) have been reported previously,<sup>16</sup> however this was *via* a different synthetic route, and only IR and NMR data were reported. We study the properties of  $\text{Ir}(\text{CO})_2\text{acac}$  (acac = acetylacetone) in parallel to benchmark how our newly characterised complexes compare to the previously reported behaviour for this family of molecules. The complexes were each synthesised from the  $[\text{Ir}(\text{I})(\text{COD})\text{Cl}]_2$  dimer (COD = 1,5-cyclooctadiene), *via* reaction with carbon monoxide, followed by addition of base and the appropriate  $\beta$ -diketonate ligand. The complexes were obtained in moderate to good yields (38–77%) as crystals from hexane solutions.

### Molecular properties

The absorption spectra of the four complexes in solution are shown in Fig. 1b. All complexes exhibit optical transitions in the same region of the UV-vis (250–420 nm). The absorption of  $\text{Ir}(\text{CO})_2\text{tfaa}$  is similar in profile to that of  $\text{Ir}(\text{CO})_2\text{acac}$  due to the similarity in ligand structure and extent of conjugation, though with a slightly smaller optical gap. The absorbance of  $\text{Ir}(\text{CO})_2\text{dbm}$  and  $\text{Ir}(\text{CO})_2\text{tfpb}$  extend further towards the visible region than that of  $\text{Ir}(\text{CO})_2\text{tfaa}$ , a result of the extended conjugation in the ligand framework.

To gain further insight into the electronic structure of the complexes, hybrid (TD-)DFT calculations were performed to identify the frontier molecular orbitals and nature of the

electronic transitions. As we find the nature of the frontier orbitals to be very similar for all complexes, those of complex  $\text{Ir}(\text{CO})_2\text{tfpb}$  are presented in Fig. 2a, while the orbitals for the remaining complexes are available in the ESI† (Fig. S4–S7). The calculated HOMO and LUMO energies are shown in Table 1. In the case of all four complexes the HOMOs contain no contributions from the diketone ligand orbitals, such that the ligands are limited to exerting an inductive effect as seen in the slight lowering of the HOMO level for the two complexes containing electron-withdrawing fluorine substituents ( $\text{Ir}(\text{CO})_2\text{tfpb}$  and  $\text{Ir}(\text{CO})_2\text{tfaa}$ ). The large contribution of the diketone orbitals to the LUMOs of the complexes results in a lowering of the LUMO energy for complexes that have extended  $\pi$ -conjugation ( $\text{Ir}(\text{CO})_2\text{dbm}$  and  $\text{Ir}(\text{CO})_2\text{tfpb}$ ) and for complexes that have electronegative fluorine atoms ( $\text{Ir}(\text{CO})_2\text{tfpb}$  and  $\text{Ir}(\text{CO})_2\text{tfaa}$ ). This combination of effects results in the narrowest HOMO–LUMO gap for  $\text{Ir}(\text{CO})_2\text{tfpb}$ .

The absorption spectra of the complexes in solution were modelled by TD-DFT calculations, and showed good agreement between observed and predicted spectra with a similar ordering of transitions for all four complexes and a reasonably-good match of the number and position of intense peaks. These are presented in Fig. 2b for  $\text{Ir}(\text{CO})_2\text{tfpb}$ , and in Fig. S3 in the ESI† for the remaining complexes, with the nature of the low-energy optical transitions also available in the ESI† (Tables S6–S9). The calculated trend of HOMO–LUMO gap energies is consistent with the ordering of the onset of absorption for the complexes in solution, as presented in Table 1. Importantly, the frontier transitions of all the complexes are of metal-to-ligand charge transfer (MLCT) character, with electron density moving from the metal centre to either the central chelate or the vacant  $\pi^*$ -orbitals on either the CO ligands. Where the ligands possess extended conjugation  $\text{Ir}(\text{CO})_2\text{dbm}$  and  $\text{Ir}(\text{CO})_2\text{tfpb}$ , the  $\pi^*$ -orbitals on the phenyl rings also contribute to these frontier transitions.

### Solid-state properties

**Crystal packing.** The solid-state optical and electronic properties of square-planar organometallic complexes depend heavily on the interaction between metal atoms and  $\pi$ -orbitals

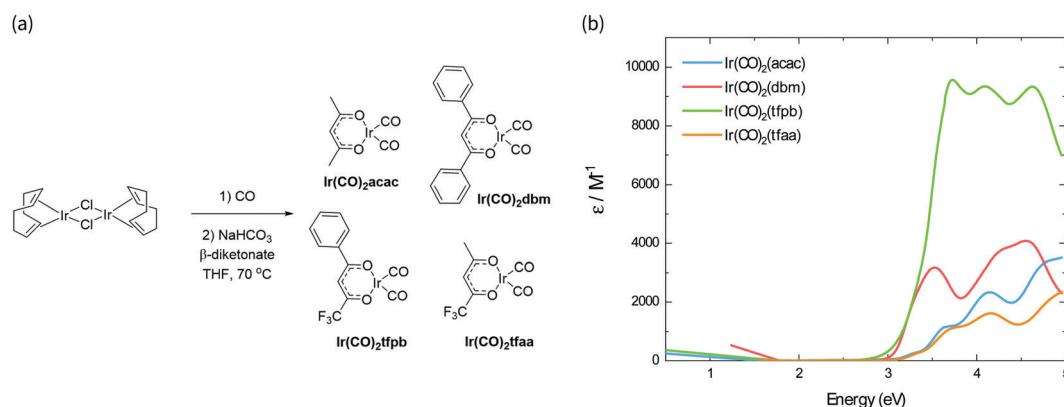


Fig. 1 (a) Synthesis scheme of  $\text{Ir}(\text{CO})_2(\beta\text{-diketonate})$  complexes. (b) Absorbance spectra of the complexes in solution, scaled by their respective molar absorption coefficients.



Fig. 2 (a) Frontier orbitals of the complex as modelled with DFT; (b) simulated (black curve) and experimental (green curve) absorbance spectra of the complex in hexane. The oscillator strength of the electronic transitions obtained *via* TD-DFT is shown (bars).

Table 1 Energies of the HOMO and LUMO of each complex

Complex	HOMO energy <sup>a</sup> (eV)	LUMO energy <sup>a</sup> (eV)	HOMO–LUMO gap (eV)	Absorption <sup>b</sup> onset (eV)
Ir(CO) <sub>2</sub> acac	−6.51	−1.75	4.76	3.1
Ir(CO) <sub>2</sub> dbm	−6.51	−2.43	4.08	2.9
Ir(CO) <sub>2</sub> tfpb	−6.76	−2.76	4.0	2.8
Ir(CO) <sub>2</sub> tfaa	−6.79	−2.46	4.33	3.1

<sup>a</sup> Calculated from DFT (B3LYP/LANL2DZ+6-31G\*). <sup>b</sup> Onset of absorption from spectra in Fig. 1b.

in adjacent molecules in the lattice.<sup>14,15,17</sup> Single crystals of the complexes were grown from hexane; the crystallographic data are given in Table 2 and the structures are presented in Fig. 3.

Complex Ir(CO)<sub>2</sub>tfaa shows a similar crystal packing to that of Ir(CO)<sub>2</sub>acac; the molecules form an infinite chain with an alternating distance of 3.2717(7) and 3.3140(7) Å between the iridium atoms, slightly longer than the metal–metal bonds of the Ir(CO)<sub>2</sub>acac complex. As with Ir(CO)<sub>2</sub>acac, the Ir atoms in the chain are slightly offset, forming a zig-zag with an angle of 170.755(19)°. Short F–O contacts (*e.g.* O2···F1 2.749(7) Å) exist between the carbon monoxide O and the CF<sub>3</sub> groups on adjacent molecules with the chain, and on adjacent chains. The similarity between the Ir(CO)<sub>2</sub>tfaa and Ir(CO)<sub>2</sub>acac structures can be explained by the comparable size and shape of the β-diketonate ligands.

By contrast, infinite chains of metal–metal interactions are not observed in complexes Ir(CO)<sub>2</sub>dbm and Ir(CO)<sub>2</sub>tfpb.

Compound Ir(CO)<sub>2</sub>dbm crystallises with a much larger unit cell, made up of chains of discrete dimers which stack along the *a*-axis. Each dimer has a short Ir–Ir distance of 3.1446(3) Å, with an angle of 1.82° between the planes of the two molecules, forming a ‘slipped’ metal–metal bond. The next closest dimer pair is related to the first *via* a translation along the *a* axis, such that the β-diketonate ligand sits over the Ir centre of molecule in an adjacent dimer. The phenyl rings are twisted (16–21°) with respect to the core of the ligand chelate, and do not show any sign of π-stacking. It is clear that the size of the β-diketonate ligand has disrupted the ability of the complex to form infinite chains of metal–metal interactions in the solid state.

In compound Ir(CO)<sub>2</sub>tfpb the β-diketonate ligand is smaller, with only one phenyl ring. As a result the molecules are still able to crystallise in metal–metal chains along the *a*-axis, however the packing is not as linear as that observed for Ir(CO)<sub>2</sub>acac or Ir(CO)<sub>2</sub>tfaa. The molecules crystallise as dimers, with each molecule related to its partner *via* a reflection and an in-plane 90° rotation, resulting in the phenyl rings stacking over each other whilst keeping the CF<sub>3</sub> groups of adjacent molecules far apart. The Ir–Ir distances are not as short as that of the other complexes, 3.5146(9) Å in the dimer and 3.9432(9) Å between dimers. The Ir atoms in the chain are also significantly offset, forming a zigzag with an angle of 159.04(3)°. In addition to this metal-centred stacking, the phenyl rings on the β-diketonate ligand form a chain of π-stacking interactions, 3.680(9) Å and 3.787(9) Å between ring centroids. It is clear that ligand choice has a significant impact on the structural packing of the

Table 2 Crystal structure data of Iridium dicarbonyl β-diketonate complexes

Complex	Ir(CO) <sub>2</sub> acac <sup>a</sup>	Ir(CO) <sub>2</sub> dbm	Ir(CO) <sub>2</sub> tfpb	Ir(CO) <sub>2</sub> tfaa
Space group	P $\bar{1}$	P $\bar{1}$	P2 <sub>1</sub> /c	P $\bar{1}$
Cell lengths (Å)	<i>a</i> = 6.48, <i>b</i> = 7.73, <i>c</i> = 9.16	<i>a</i> = 8.3064(4), <i>b</i> = 11.3030(5), <i>c</i> = 16.2709(9)	<i>a</i> = 7.333*(2), <i>b</i> = 17.1409(6), <i>c</i> = 19.7049(5)	<i>a</i> = 6.5643(4), <i>b</i> = 7.7556(4), <i>c</i> = 9.4383(3)
Cell angles (°)	$\alpha$ = 105.7, $\beta$ = 90.5, $\gamma$ = 100.7	$\alpha$ = 96.266(4), $\beta$ = 97.677(4), $\gamma$ = 100.017(4)	$\alpha$ = $\gamma$ = 90, $\beta$ = 92.083(2)	$\alpha$ = 101.495(3), $\beta$ = 90.337(4), $\gamma$ = 102.307(5)
Cell volume (Å <sup>3</sup> )	433.24	1477.08(13)	2475.46(13)	459.46(4)
<i>Z</i> , <i>Z'</i>	2, 0	4, 2	8, 2	2, 1

<sup>a</sup> Crystal parameters taken from ref. 18.

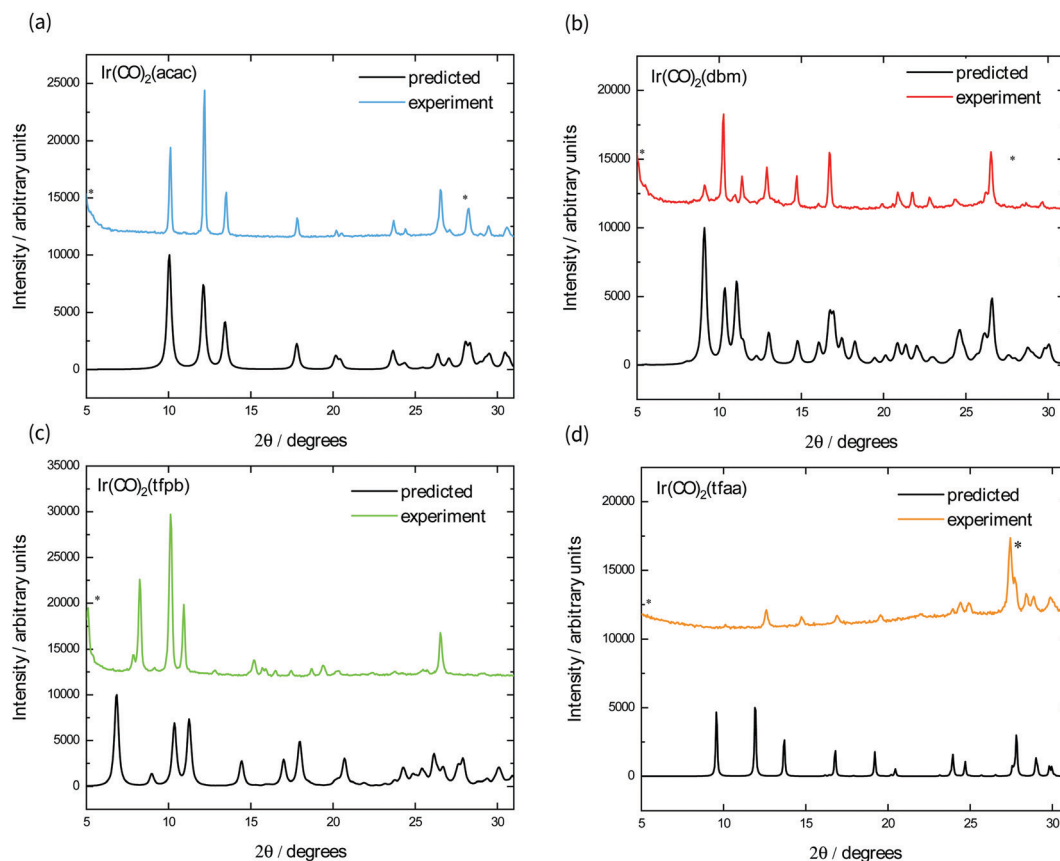


**Fig. 3** Schematic of the crystal structures of (a)  $\text{Ir}(\text{CO})_2\text{acac}$ , (b)  $\text{Ir}(\text{CO})_2\text{dbm}$ , (c)  $\text{Ir}(\text{CO})_2\text{tfpb}$ , (d)  $\text{Ir}(\text{CO})_2\text{tfaa}$ , obtained via single-crystal X-ray diffraction. Note that (b) and (c) have two molecules in each asymmetric unit. For all complexes, four molecules are shown to illustrate packing along the axis perpendicular to the plane of the molecules.

molecules in crystal form, with large ligands able to overcome the preference for forming metal-stacked chains.

We also fabricated films of each complex using vapour deposition, as square planar  $\text{M}(\text{CO})_2(\beta\text{-diketonate})$  complexes have been shown to be volatile at relatively low temperatures under reduced pressure.<sup>19,20</sup> Their powder X-ray diffraction patterns are shown in Fig. 4, alongside the predicted powder spectrum obtained from single crystal structures. The powder

patterns for  $\text{Ir}(\text{CO})_2\text{acac}$ ,  $\text{Ir}(\text{CO})_2\text{dbm}$ , and  $\text{Ir}(\text{CO})_2\text{tfaa}$  match well with the predicted single crystal patterns, with some evidence of preferred orientation indicated by the ratio of peak heights. The film of  $\text{Ir}(\text{CO})_2\text{tfpb}$ , however, exhibits peaks that do not correspond to the predicted spectrum, indicating either the existence of a different polymorph or decomposition of the material under the sublimation conditions.



**Fig. 4** Crystal structure of the vapour-deposited films of complexes (a)  $\text{Ir}(\text{CO})_2\text{acac}$ , (b)  $\text{Ir}(\text{CO})_2\text{dbm}$ , (c)  $\text{Ir}(\text{CO})_2\text{tfpb}$ , (d)  $\text{Ir}(\text{CO})_2\text{tfaa}$ , showing the predicted powder X-ray diffraction pattern (black) calculated from single-crystal structure data and experimental diffraction pattern for the films of each material. No experimental peaks were of sufficient intensity to be observed above (a)  $40^\circ$ , (b)  $30^\circ$ , (c)  $25^\circ$ . Peaks labelled with an asterisk (\*) correspond to the PXRD pattern of the fluorine-doped Tin Oxide (FTO) glass substrate.





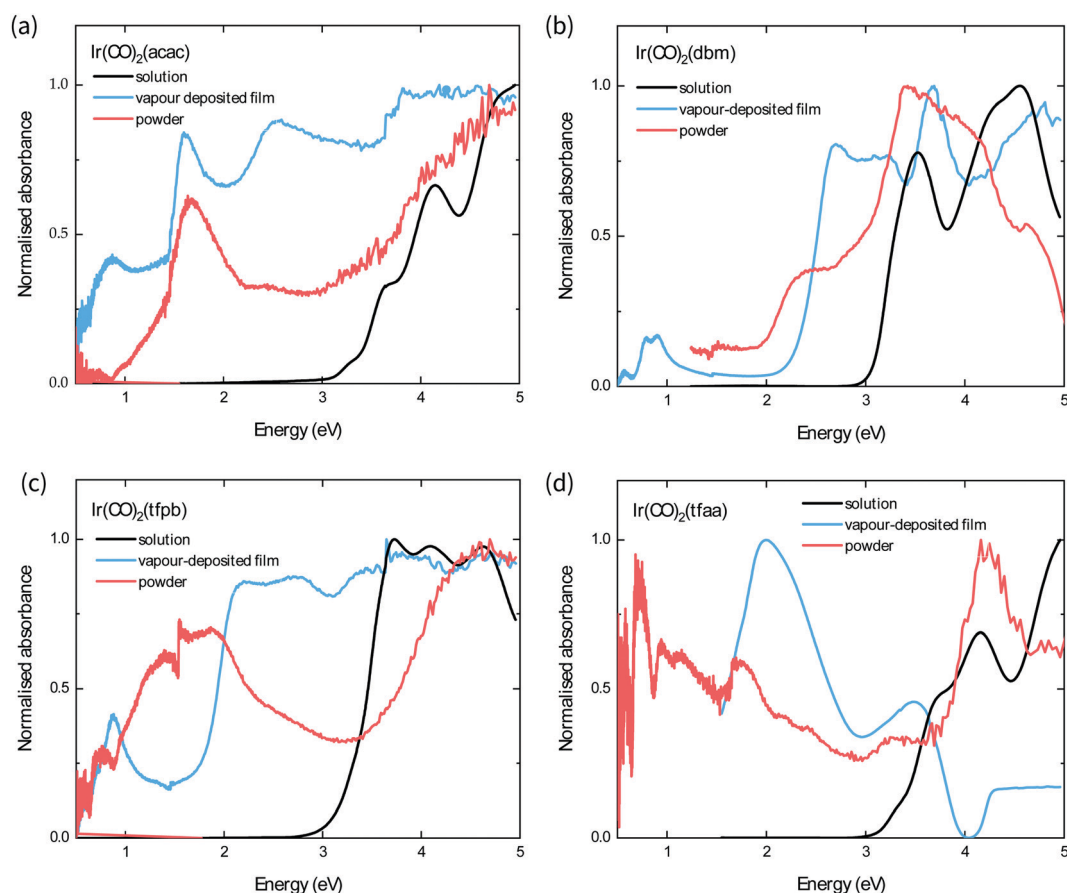
**Optical properties in the solid state.** We present solid-state absorption behaviour of the complexes in Fig. 5, measured with diffuse reflectance spectroscopy on bulk powders and thin-films obtained *via* sublimation for each material. Photographs of the films are shown in Fig. S15 (ESI†). The solution absorption spectra are included for reference alongside the Kubelka–Munk transforms of the solid-state complexes, which describe absorbance of each material. In all four complexes, the solid state and thin-film spectra show high-energy transitions as seen for single molecules in solution, as well as low energy absorption peaks only observed in the solid phases. This suggests new transitions emerging from strong intermolecular interactions present in the solid state.

The optical band gap of each complex is estimated from the absorption behaviour using the Tauc plot extrapolation method (see ESI†),<sup>21</sup> and the obtained values from the optical spectra of the powders follow the order  $\text{Ir}(\text{CO})_2\text{tfaa}$  (0.5 eV) <  $\text{Ir}(\text{CO})_2\text{tfpb}$  (1.0 eV) <  $\text{Ir}(\text{CO})_2\text{acac}$  (1.3 eV) <  $\text{Ir}(\text{CO})_2\text{dbm}$  (2.0 eV). This contrasts with the solution behaviour, where the complexes with the most conjugated ligands ( $\text{Ir}(\text{CO})_2\text{dbm}$  and  $\text{Ir}(\text{CO})_2\text{tfpb}$ ) had the smallest optical gap, and highlights the role of strong crystal packing interactions in determining solid-state optical behaviour. The powder and film spectra show similar features for all complexes with the exception of  $\text{Ir}(\text{CO})_2\text{tfpb}$ , where the

absorbance is reduced for the film between 1–2 eV compared to the powder. As the powder XRD measurements showed that the film has a different crystal structure to the measured single-crystal, it is possible that the inter-band transitions in the vapour-deposited film and the powder arise from differences in structure.

The band structures and projected density of states (PDOS) of the complexes in the solid-state were calculated with CASTEP<sup>22</sup> and are shown in Fig. 6. For all the complexes the valence band is comprised of Ir-based orbitals, while the frontier of the conduction band is made up of contributions from the  $\beta$ -diketonate ligands, with a small contribution from the CO ligands present for  $\text{Ir}(\text{CO})_2\text{acac}$  and  $\text{Ir}(\text{CO})_2\text{dbm}$ . Comparing these results with the molecular orbitals of the isolated molecules confirms that the solid-state valence band originates from the molecular HOMOs, particularly the  $d_{z^2}$  orbital, and that molecular LUMOs contribute to the conduction band.

In the complexes with linear metal–metal stacks in the crystalline solid ( $\text{Ir}(\text{CO})_2\text{acac}$ ,  $\text{Ir}(\text{CO})_2\text{tfaa}$  and,  $\text{Ir}(\text{CO})_2\text{tfpb}$ ), the valence band exhibits a high degree of dispersion along one axis (*ca.* 2 eV), corresponding to the metal–metal stacking axis (*c*-axis) in the crystal structure. This indicates that the intermolecular interactions are relatively strong along the chain. By contrast, the conduction bands are relatively flat,



**Fig. 5** UV-vis absorption spectra of the complexes in hexane solution (black curves) and Kubelka–Munk transforms obtained from diffuse reflectance measurements of vapour-deposited films (blue curves) and powders (red curves) of (a)  $\text{Ir}(\text{CO})_2\text{acac}$ , (b)  $\text{Ir}(\text{CO})_2\text{dbm}$ , (c)  $\text{Ir}(\text{CO})_2\text{tfpb}$ , (d)  $\text{Ir}(\text{CO})_2\text{tfaa}$ .





**Fig. 6** Predicted electronic transitions in the molecular solid: Band structure and corresponding projected density of states (PDOS) of each band, calculated using density functional theory with CASTEP. The valence bands (blue curves in band structure) are mainly of metal (blue curves in PDOS) character, whilst the conduction bands (red curves in band structure), are mainly of ligand character (red curves in PDOS). The X-axis labels on the electronic band structure diagrams represent high symmetry *k*-points, defined according to the standard convention for each corresponding Brillouin zone.

indicating there is relatively little overlap between the unoccupied molecular orbitals on transition to the solid state.

The band structure of  $\text{Ir}(\text{CO})_2\text{dbm}$  shows little to no dispersion for any of the bands, indicating that the orbitals are highly localised. This is a result of poor intermolecular orbital overlap in the solid state, a consequence of the dimeric packing in the crystal structure, and suggests that mobility of charge carriers will be limited.

The crystal structure of  $\text{Ir}(\text{CO})_2\text{tfpb}$  shows the presence of an infinite chain of metal–metal stacking interactions, but with longer bond lengths and a more staggered overlap compared to that of  $\text{Ir}(\text{CO})_2\text{acac}$  and  $\text{Ir}(\text{CO})_2\text{tfaa}$ . This translates to a lesser amount of dispersion in the valence band compared to the linear chain complexes, but a far greater dispersion than seen for the dimeric structure of  $\text{Ir}(\text{CO})_2\text{dbm}$ .

The simulated band gaps for the complexes follow the order  $\text{Ir}(\text{CO})_2\text{tfaa}$  (0.6 eV) <  $\text{Ir}(\text{CO})_2\text{acac}$  (1.0 eV) <  $\text{Ir}(\text{CO})_2\text{tfpb}$  (1.2 eV) <  $\text{Ir}(\text{CO})_2\text{dbm}$  (1.5 eV), which is largely in agreement with the trend from experimentally derived band gaps ( $\text{Ir}(\text{CO})_2\text{tfaa}$  <  $\text{Ir}(\text{CO})_2\text{tfpb}$  <  $\text{Ir}(\text{CO})_2\text{acac}$  <  $\text{Ir}(\text{CO})_2\text{dbm}$ ). This order reflects the varying amounts of dispersion in the band structures, which originate from the differing extent of intermolecular interactions in the solid state.

### Piezoresistive properties

The conductivity of  $\text{Ir}(\text{CO})_2\text{acac}$  has been shown to dramatically increase under the application of pressure; after applying

~14 GPa of external pressure the conductivity increases by approximately 5 orders of magnitude.<sup>15</sup> This increased conductivity is associated with a smooth decrease in the metal–metal distances along the chain. Herein, we have investigated the conductive behaviour of complexes  $\text{Ir}(\text{CO})_2\text{dbm}$ ,  $\text{Ir}(\text{CO})_2\text{tfpb}$  and  $\text{Ir}(\text{CO})_2\text{tfaa}$  under applied pressure in order to probe the influence of ligand choice on the solid state conductivity. The conductivity of the compressed powders was measured on a pellet using a diamond anvil cell (DAC) apparatus and is presented in Fig. 7 (see Methods for details).

In the case of complex  $\text{Ir}(\text{CO})_2\text{dbm}$ , the resistance of the compressed pellet was too high to be measured (>200 TΩ), even at elevated temperature (353 K). This high resistance is a due to the low dispersion of the frontier bands, which originated from the lack of metal–metal chains in the crystal structure.

As  $\text{Ir}(\text{CO})_2\text{tfaa}$  and  $\text{Ir}(\text{CO})_2\text{acac}$  have comparable crystal packing, it is plausible to expect any differences in pressure-dependent conductivity between the two originating from either (1) different initial band-gaps, (2) different degrees of compressibility of the lattice, or (3) presence of any phase transitions. The conductivity of complex  $\text{Ir}(\text{CO})_2\text{tfaa}$  increases steadily with greater pressure, increasing by approximately 2 orders of magnitude by 3.0 GPa. This increased conductivity is accompanied by a change in the optical properties with the pellet turning from green to brown (Fig. 7 insets), indicating a shift in the energies of the bands. This behaviour is comparable



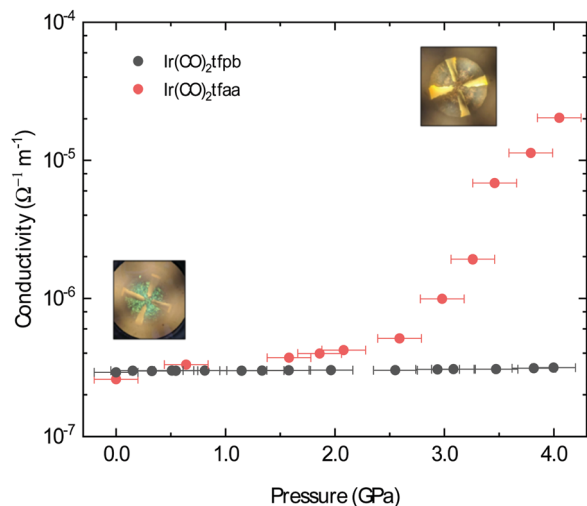


Fig. 7 Pressure-dependent conductivity of a compressed powders of  $\text{Ir}(\text{CO})_2\text{tfpb}$  and  $\text{Ir}(\text{CO})_2\text{tfaa}$ . Inset are pictures of the compressed  $\text{Ir}(\text{CO})_2\text{tfaa}$  inside the DAC at two different pressures, showing a colour change from green to brown indicating a shift in energy of the optical bands of the material. The error bars in the pressure axis arise from uncertainty in the collected pressure readings (see Methods for details). The error bars for conductivity values are too small to be visualised in the figure.

to that of  $\text{Ir}(\text{CO})_2\text{acac}$ , which experiences an order of magnitude increase in conductivity within a similar range.

By contrast, the conductivity of complex  $\text{Ir}(\text{CO})_2\text{tfpb}$  remains more or less constant at a lower value within the same pressure range, despite the fact that there is some degree of continuous metal–metal interactions in the structure as in the other complexes. This suggests that its structure does not compress in a way that enhances the metal–metal interactions, which are more staggered than those in  $\text{Ir}(\text{CO})_2\text{acac}$  or  $\text{Ir}(\text{CO})_2\text{tfaa}$ .

The difference in behaviour between the four complexes highlights the role the ligand plays in determining the optical and electronic behaviour of  $\text{Ir}(\text{CO})_2(\beta\text{-diketonate})$  complexes, by virtue of how it impacts the solid state crystal packing. Complexes with ligands that prevent the formation of infinite chains, or disrupt the linearity of the metal–metal stack, will either be highly insulating or only show a moderate response to the application of pressure.

## Conclusions

We synthesised four  $\text{Ir}(\text{CO})_2(\beta\text{-diketonate})$  complexes and investigated their structural and optical properties. We find that choice of  $\beta\text{-diketonate}$  ligand contributes significantly to the optical transitions of the molecules in solution by lowering the LUMO energy for complexes with  $\pi$ -conjugation or electronegative fluorine atoms. In the solid state, we observe that the choice of ligand plays a significant role in determining the structural packing in the molecular crystal. Importantly, with the support of molecular orbital and band structure calculations, we show that these differences in structural packing have a profound impact on the optical transitions in the solid state,

with complexes packing in metal–metal stacks showing a significant reduction in band gap. We also highlight the impact of this metal–metal stacking on the electronic properties of these materials, leading to a pressure-dependent increase in conductivity for  $\text{Ir}(\text{CO})_2\text{tfaa}$  and  $\text{Ir}(\text{CO})_2\text{acac}$ , while complexes without this favourable packing showed little piezoconductive response. These results demonstrate the importance of ligand modification within these materials set as an important tool for control over their optical and electronic properties. The separate influences of LUMO energy tuning using electron-withdrawing ligands, alongside band-structure tuning through ligand size and packing, provides insight and guidance for the optoelectronic application of semiconducting metal complexes, and the design and development of future molecular electronic materials.

## Materials and methods

### Organometallic synthesis

All commercially available chemicals were used without further purification. The iridium complexes were synthesised according to the following general method:

$\text{Ir}[(\text{COD})(\mu\text{-Cl})_2]$  (400 mg, 0.595 mmol) was dissolved in dry tetrahydrofuran (THF) (45 ml) under nitrogen atmosphere. The solution was degassed by three freeze–pump–thaw cycles, and allowed to thaw under vacuum. Carbon monoxide gas was then passed through the reaction flask for 5–10 minutes under vigorous stirring until the solution changed colour, from yellow-orange to blue-black. The CO was produced *in situ* from the reaction between oxalyl chloride and aqueous NaOH solution as described by Hansen *et al.*,<sup>23</sup> collected in a balloon and flowed directly into the reaction flask. Sodium hydrogen carbonate ( $\text{NaHCO}_3$ ) (4.00 g, 47 mmol) and the appropriate ligand (0.6 mmol) were then quickly added under a high flow of nitrogen, and the mixture heated at 70 °C for 3 hours with stirring. The solution changed back to a yellow-orange colour upon reaction completion. The reaction mixture was cooled, and the product collected through filtration after evaporating the solvent and washing with hexane.

Single crystals of all complexes were obtained by dissolving the reaction product in hexane, heating the solution to 100 °C and leaving to cool slowly. Due to their high solubility in hexane, the  $\text{Ir}(\text{CO})_2\text{tfpb}$  and  $\text{Ir}(\text{CO})_2\text{tfaa}$  solutions were placed in the fridge (4 °C) to cool for half an hour, then left in the freezer (−20 °C) overnight.

$\text{Ir}(\text{CO})_2\text{acac}$  (160 mg, 77%), obtained as bronze needles.  $^1\text{H}$  NMR (500 MHz, Chloroform-*d*)  $\delta$  = 5.8 (s, 1H), 2.13 (s, 6H) (in agreement with that reported in ref. 18).

**$\text{Ir}(\text{CO})_2\text{dbm}$  (250 mg, 44%) obtained as orange needles.**  $^1\text{H}$  NMR (500 MHz, Chloroform-*d*)  $\delta$  = 8.12–7.90 (m, 4H), 7.65–7.54 (m, 2H), 7.54–7.44 (m, 4H), 7.13 (s, 1H). HRMS: *m/z* (ESI) Found: 473.0355, 944.0539, 1414.0770;  $\text{C}_{17}\text{H}_{11}\text{IrO}_4$ ,  $[\text{C}_{17}\text{H}_{11}\text{IrO}_4]_2$  and  $[\text{C}_{17}\text{H}_{11}\text{IrO}_4]_3$  require 471.49, 942.98, 1414.47, respectively. Elemental analysis calculated for  $\text{C}_{17}\text{H}_{11}\text{IrO}_4$ : C, 43.31; H, 2.35. Found: C, 43.22; H, 2.205.



**Ir(CO)<sub>2</sub>tftp (151 mg, 63%), obtained as dark green needles.**

<sup>1</sup>H NMR (500 MHz, chloroform-*d*)  $\delta$  = 8.06–7.94 (m, 2H), 7.69 (ddt, *J* = 8.7, 7.2, 1.3 Hz, 1H), 7.58–7.50 (m, 2H), 6.91 (s, 1H). <sup>19</sup>F NMR (471 MHz, chloroform-*d*)  $\delta$  = –73.99. HRMS: *m/z* (ESI) Found: 922.0107, 1389.9503; [C<sub>12</sub>H<sub>6</sub>F<sub>3</sub>IrO<sub>4</sub>]<sub>2</sub> and [C<sub>12</sub>H<sub>6</sub>F<sub>3</sub>IrO<sub>4</sub>]<sub>3</sub> require 926.78 and 1390.17, respectively. Elemental analysis calculated for C<sub>12</sub>H<sub>6</sub>F<sub>3</sub>IrO<sub>4</sub>: C, 31.10; H, 1.31. Found: C, 31.19; H, 1.205.

**Ir(CO)<sub>2</sub>tfaa (117 mg, 38%), obtained as bright green needles.**

<sup>1</sup>H NMR (500 MHz, chloroform-*d*)  $\delta$  = 7.29 (s, 2H), 6.26 (d, *J* = 1.6 Hz, 1H), 2.32 (d, *J* = 1.6 Hz, 3H), 1.28 (s, 1H), 0.87 (s, 1H). <sup>19</sup>F NMR (471 MHz, chloroform-*d*)  $\delta$  = –74.19. HRMS: *m/z* (ESI) found: 409.96950; C<sub>7</sub>H<sub>4</sub>F<sub>3</sub>IrO<sub>4</sub> requires 401.96856. Elemental analysis calculated for C<sub>7</sub>H<sub>4</sub>F<sub>3</sub>IrO<sub>4</sub>: C, 20.95; H, 1.00. Found: C, 21.04; H, 1.08.

**Thin-film fabrication via vapour deposition**

Single crystals (<5 mg) of the desired material were sublimed onto fluorine-doped tin oxide (FTO) substrates under vacuum (300 mTorr) at approximately 120 °C. The substrate was placed at the end of a condenser finger cooled with dry ice/acetone.

**Crystallography methods**

Single crystal X-ray diffraction measurements were carried out on a Rigaku Oxford Diffraction SuperNova (compounds Ir(CO)<sub>2</sub>dmb and Ir(CO)<sub>2</sub>tftp) or a Rigaku Oxford Diffraction Gemini (Ir(CO)<sub>2</sub>tfaa) diffractometer using either Mo (Ir(CO)<sub>2</sub>dmb) or Cu (all others) K $\alpha$  radiation and a crystal temperature of 100 (Ir(CO)<sub>2</sub>tfaa) or 120 (all others) K. All structures were solved with ShelXT<sup>24</sup> and refined by full-matrix least-squares on *F*<sup>2</sup> using ShelXL,<sup>24</sup> interfaced through Olex2.<sup>25</sup>

**Optical characterization**

UV-visible absorption spectra of the complexes in solution and thin-films were carried out using a spectrophotometer (JASCO V-670), with a cell path-length of 1 cm. For solution measurements, a cuvette containing hexane was used as a reference. Molar absorption coefficients of each complex were obtained using the Beer–Lambert law. Diffuse reflectance (DR) spectra were measured with the same setup, equipped with an integrating sphere (SIN-723UV-Visible-NIR). For DR measurements, a few milligrams of each complex were ground into a fine powder with barium sulfate (BaSO<sub>4</sub>), to obtain a ‘dilute’ powder mixture, which was placed into the setup for measurement. The reflectance of a pure BaSO<sub>4</sub> ground sample was used as a reference.

**Computational methodology**

**Single-molecule calculations.** Hybrid density functional theory (DFT) was used to optimise the molecular geometries and calculate the molecular orbitals of each complex, and information on the excited states was calculated with time-dependent DFT (TD-DFT) using the Gaussian 09 package.<sup>26</sup> The Becke, 3-parameter, Lee–Yang–Parr hybrid functional (B3LYP) was used,<sup>27,28</sup> with the LANL2DZ basis set and pseudopotential used for the iridium atom, and the 6-31G\* basis set for C, H, O and F

atoms. The Polarization Continuum Model was used to model the behaviour in solvent (hexane). Molecular orbital images were obtained using Avogadro,<sup>29</sup> and predicted absorption spectra processed from TD-DFT oscillator strength data with Gausssum software,<sup>30</sup> applying a FWHM of 300 nm to each transition.

**Solid-state calculations**

The band structure and partial density of states for the complexes were calculated using density functional theory within the plane-wave pseudopotential approach as implemented in the CASTEP code.<sup>22</sup> The PBE functional was used with the default CASTEP ultrasoft pseudopotentials and a plane-wave basis set express at an energy cut-off of 750 eV. The primitive electronic Brillouin zone was sampled with a Monkhorst–Pack grid of 3  $\times$  5  $\times$  5 for Ir(CO)<sub>2</sub>acac, 5  $\times$  5  $\times$  5 for both Ir(CO)<sub>2</sub>dmb and Ir(CO)<sub>2</sub>tftp, and 4  $\times$  3  $\times$  3 for Ir(CO)<sub>2</sub>tfaa. The input geometries were those obtained from single-crystal X-ray diffraction experiments.

Ir(CO)<sub>2</sub>acac and Ir(CO)<sub>2</sub>dmb were optimised under full geometry optimisation, while Ir(CO)<sub>2</sub>tfaa was subjected to atom-only (ie lattice parameters held fixed, due to poor reproduction of the unit cell parameters under full optimisation – see ESI†). The following convergence criteria were applied: energy tolerance = 1  $\times$  10<sup>–5</sup> eV  $\text{\AA}^{-1}$ , atomic forces = 0.05 eV  $\text{\AA}^{-1}$ , atomic displacements = 5  $\times$  10<sup>–5</sup>  $\text{\AA}$ , and unit cell stress = 0.1 GPa using the Broyden–Fletcher–Goldfarb–Shanno (BFGS) method. Calculations on Ir(CO)<sub>2</sub>tftp were performed at single point energy, as due to the large system size the structure could not be optimised even under atom only conditions. The partial density of states plots were derived using the OptaDOS code.<sup>31,32</sup>

**Piezoresistivity measurements**

The high-pressure resistance measurements were performed in a Merrill–Bassett diamond anvil cell (DAC) with 0.8 mm culet anvils. The resistance was measured on the compressed powder pellet through the deposited on the anvil gold contacts (see Fig. 7 inset) using the Keithley 6517A electrometer. The resistance values were converted into conductivity using the Montgomery method for anisotropic materials.<sup>33</sup> The pressure was probed using the ruby fluorescence method,<sup>34</sup> with Daphne 7373 oil as the pressure transmitting medium.<sup>35</sup> The uncertainty in collected pressure readings (0.2 GPa) arises due to shifts of the ruby crystal from the centre of the gasket during the course of the measurement.

**Author contributions**

H. B. and N. R. designed and directed the work; S. E. B., and H. B. synthesised materials and performed optical experiments; S. E. B., H. B. and C. M. conducted simulations; S. A. performed pressure-dependent measurements, G. S. N. and S. M. conducted single-crystal X-ray diffraction measurements; K. K. and N. R. supervised the project; S. E. B. and H. B. wrote the manuscript with input from all authors.





## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 H. Kobayashi, A. Kobayashi and H. Tajima, *Chem. – Asian J.*, 2011, **6**, 1688–1704.
- 2 D. D. Eley, *Nature*, 1948, **162**, 819.
- 3 K. Krogmann, *Angew. Chem., Int. Ed. Engl.*, 1969, **8**, 35–42.
- 4 J. Ferraris, D. O. Cowan, V. Walatka and J. H. Perlstein, *J. Am. Chem. Soc.*, 1973, **95**, 948–949.
- 5 D. Jérôme, A. Mazaud, M. Ribault and K. Bechgaard, *J. Phys. Lett.*, 1980, **41**, 95–98.
- 6 R. Kato, H. B. Cui, T. Tsumuraya, T. Miyazaki and Y. Suzumura, *J. Am. Chem. Soc.*, 2017, **139**, 1770–1773.
- 7 J. B. Torrance, J. J. Mayerle, K. Bechgaard, B. D. Silverman and Y. Tomkiewicz, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1980, **22**, 4960–4965.
- 8 H. Tajima, M. Inokuchi, A. Kobayashi, T. Ohta, R. Kato, H. Kobayashi and H. Kuroda, *Chem. Lett.*, 1993, 1235–1238.
- 9 B. M. Anderson and S. K. Hurst, *Eur. J. Inorg. Chem.*, 2009, 3041–3054.
- 10 K. Takeda, I. Shirotni and K. Yakushi, *Chem. Mater.*, 2000, **12**, 912–916.
- 11 I. Shirotni, A. Kawamura, K. Suzuki, W. Utsumi and T. Yagi, *Bull. Chem. Soc. Jpn.*, 2006, **64**, 1607–1612.
- 12 K. Takeda, I. Shirotni, C. Sekine and K. Yakushi, *J. Phys.: Condens. Matter*, 2000, **12**, L483–L488.
- 13 K. Takeda, I. Shirotni and K. Yakushi, *Synth. Met.*, 2003, **133–134**, 415–416.
- 14 H. Benjamin, J. G. Richardson, S. A. Moggach, S. Afanasjevs, L. Warren, M. R. Warren, D. R. Allan, C. A. Morrison, K. V. Kamenev and N. Robertson, *Phys. Chem. Chem. Phys.*, 2020, **22**, 6677–6689.
- 15 L. V. Interrante and F. P. Bundy, *Inorg. Chem.*, 1971, **10**, 1169–1174.
- 16 F. Bonati and R. Ugo, *J. Organomet. Chem.*, 1968, **11**, 341–352.
- 17 L. F. Ballard and J. J. Wortman, *J. Appl. Phys.*, 1970, **41**, 4232–4236.
- 18 K. V. Zherikova, N. V. Kuratieva and N. B. Morozova, *J. Struct. Chem.*, 2009, **50**, 574–576.
- 19 E. L. Crane, Y. You, R. G. Nuzzo and G. S. Girolami, *J. Am. Chem. Soc.*, 2000, **122**, 3422–3435.
- 20 J. C. Hierro, R. Feurer and P. Kalck, *Coord. Chem. Rev.*, 1998, **178–180**, 1811–1834.
- 21 J. Tauc, R. Grigorovici and A. Vancu, *Phys. Status Solidi*, 1966, **15**, 627–637.
- 22 S. J. Clark, M. D. Segall, C. J. Pickard, P. J. Hasnip, M. I. J. Probert, K. Refson and M. C. Payne, *Zeitschrift für Krist.*, 2005, **220**, 567–570.
- 23 S. V. F. Hansen and T. Ulven, *Org. Lett.*, 2015, **17**, 2832–2835.
- 24 G. M. Sheldrick, *Acta Crystallogr., Sect. A: Found. Adv.*, 2015, **71**, 3–8.
- 25 O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, *J. Appl. Crystallogr.*, 2009, **42**, 339–341.
- 26 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, G. A. Petersson, H. Nakatsuji, X. Li, M. Caricato, A. Marenich, J. Bloino, B. G. Janesko, R. Gomperts, B. Mennucci, H. P. Hratchian, J. V. Ortiz, A. F. Izmaylov, J. L. Sonnenberg, D. Williams-Young, F. Ding, F. Lipparini, F. Egidi, J. Goings, B. Peng, A. Petrone, T. Henderson, D. Ranasinghe, V. G. Zakrzewski, J. Gao, N. Rega, G. Zheng, W. Liang, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, K. Throssell, J. A. J. Montgomery, J. E. Peralta, F. Ogliaro, M. Bearpark, J. J. Heyd, E. Brothers, K. N. Kudin, V. N. Staroverov, T. Keith, R. Kobayashi, J. Normand, K. Raghavachari, A. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, J. M. Millam, M. Klene, C. Adamo, R. Cammi, J. W. Ochterski, R. L. Martin, K. Morokuma, O. Farkas, J. B. Foresman and D. J. Fox, *Gaussian 09, Revision A.02*.
- 27 A. D. Becke, *J. Chem. Phys.*, 1993, **98**, 5648–5652.
- 28 C. Lee, W. Yang and R. G. Parr, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1988, **37**, 785–789.
- 29 M. D. Hanwell, D. E. Curtis, D. C. Lonie, T. Vandermeersch, E. Zurek and G. R. Hutchison, *J. Cheminform.*, 2012, **4**, 17.
- 30 N. M. O'Boyle, A. L. Tenderholt and K. M. Langner, *J. Comput. Chem.*, 2008, **29**, 839–845.
- 31 A. J. Morris, R. J. Nicholls, C. J. Pickard and J. R. Yates, *Comput. Phys. Commun.*, 2014, **185**(5), 1477–1485.
- 32 R. J. Nicholls, A. J. Morris, C. J. Pickard and J. R. Yates, *J. Phys.: Conf. Ser.*, 2012, **371**, 012062.
- 33 H. C. Montgomery, *J. Appl. Phys.*, 1971, **42**, 2971–2975.
- 34 G. J. Piermarini, S. Block, J. D. Barnett and R. A. Forman, *J. Appl. Phys.*, 1975, **46**, 2774–2780.
- 35 K. Yokogawa, K. Murata, H. Yoshino and S. Aoyama, *Jpn. J. Appl. Phys.*, 2007, **46**, 3636.

