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## REVIEW

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## 1 Introduction

Excessive carbon dioxide emissions from the use of fossil fuels have significantly affected the climate, resulting in global warming.<sup>1</sup> According to the "Global Carbon Budget 2023" report, it is estimated that the worldwide emissions of carbon dioxide will reach an unprecedented level of 36.8 billion tons in 2023, witnessing a 1.1% increase from the previous year. This

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Mimicking natural photosynthesis, artificial photosynthesis for the reduction of  $CO_2$  into valuable hydrocarbon fuels is a promising approach for solar energy utilization and carbon neutrality. However, great challenges are present in the development of efficient photocatalysts for  $CO_2$  reduction and controlling the selectivity for reduction products. This review summarizes the progress in photocatalyst design strategies to improve the efficiency and selectivity of photocatalytic  $CO_2$  reduction. Six popular modification methods are introduced, namely, creation of defect structures, cocatalyst loading, doping, heterojunction formation, single-atom engineering, and surface organometallic catalysis. The effects of these different strategies on the promotion of light absorption, charge separation and migration and catalyst surface reactions in the process of  $CO_2$  reduction are analyzed. In addition, the latest research results on selective reduction to  $C_1$ ,  $C_2$ , and  $C_{2+}$  products in  $CO_2$  and  $H_2O$  systems are summarized. Finally, the article delves into the future prospects and inherent hurdles in photocatalyst design, focusing on enhancing the selectivity of  $CO_2$  conversion towards specific products. This review provides insights into the efficiency and selectivity of photocatalytic  $CO_2$  reduction across various photocatalysts, thereby serving as valuable guidance for the advancement of high-performance photocatalysts.

not only underscores the inevitable depletion of humanity's predominant energy source—fossil fuels—but also highlights excessive  $CO_2$  emissions resulting from their combustion.<sup>2-6</sup> Therefore, it is imperative for experts to devise viable strategies to convert vast quantities of renewable energy sources such as solar energy, which can potentially supply the earth with approximately 120 000 TW of power annually, into usable chemical energy.<sup>7-11</sup> Possible chemical transformation pathways for  $CO_2$  reduction include enzymatic catalysis, electrocatalysis, chemical reforming, and photocatalysis (Scheme 1). In the case of enzymatic catalysis, it is easy to achieve C–C



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Scheme 1 Possible pathways for the chemical conversion of CO2.

coupling, which can convert  $CO_2$  into complex chemicals with high added value under mild conditions. However, a variety of problems such as the uncertain enzyme structure, poor stability, low activity, expensive cofactors, sensitivity to oxygen, and difficulty in purification are obstacles to large-scale greenhouse gas fixation.<sup>12</sup> Regarding electrocatalytic  $CO_2$  conversion, its reaction conditions are relatively mild, reaction rate is controllable, and selective generation of target products can be controlled by adjusting reaction potential as well as electrolyte



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and catalyst type reasonably. However, the activity and stability of electrocatalysis are still low at industrial current density. In addition, owing to the amplification effect caused by complex multi-field coupling such as electric field, flow field and thermal field, reaction performance, such as reaction life and energy consumption, is reduced, which further limits the large-scale application of CO<sub>2</sub> electrolysis technology.<sup>13</sup> In the chemical reforming reaction of CO<sub>2</sub>, the formed products are rich in variety and have heat resistance due to the carbonization reaction. However, if the chemical reforming reaction occurs in the absence of steam, traditional reforming catalysts will be rapidly deactivated.14 To date, there are no effective commercial catalysts that can operate without producing carbon; meanwhile, the chemical reforming process is an energy-intensive process that does not meet the requirements of green sustainability. Accordingly, the photocatalytic reduction of CO<sub>2</sub> to useful chemicals by simulating natural photosynthesis is the most ideal way. Compared with other methods, the photocatalysis process is carried out at room temperature and pressure, the raw materials are simple and easy to obtain, and the direct use of solar energy does not require auxiliary energy and can truly realize the recycling of carbon materials; thus, it is considered to be the most promising CO2 conversion method.<sup>15-19</sup> Not only does it tackle the issue of excessive carbon dioxide emissions, but it also addresses the escalating need for energy.

Photocatalytic CO<sub>2</sub> reduction has been extensively explored in terms of photocatalyst materials, product yield and selectivity, and mechanism.<sup>20,21</sup> Consequently, substantial progress has been made in enhancing the efficiency and product selectivity through diverse catalyst development strategies. Researchers have explored various strategies for the modification of catalysts aimed at improving their catalytic efficiency. Generally, there are two main concerns in the development of photocatalysts. Firstly, it is necessary to improve the separation and transfer of photogenerated charge in the photocatalyst *via* the typically heterojunction composition or cocatalyst loading. For instance, Cheng *et al.*<sup>22</sup> developed an ultra-thin Z-type heterojunction with a double-defect structure to enhance the efficiency of charge separation. Ran *et al.*<sup>23</sup> proposed that the presence of a co-catalyst can enhance the separation and



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Zizhong Zhang is Full Professor in the State Key Laboratory of Photocatalysis on Energy and Environment at Fuzhou University. He received his PhD in physical chemistry at Fuzhou University. His research interests mainly focus on heterogeneous photocatalysis in water splitting,  $CO_2$  conversion,  $CH_4$ transformation, and environmental purification. transfer of photoinduced electron-hole pairs and provided an overview of the co-catalysts developed to date. Secondly, it is necessary to enhance the CO<sub>2</sub> absorption and activation on photocatalysts via single atom or defect engineering. Zhang et al.24 proposed that anchoring Cu single atoms in COFs can provide more adsorption and activation sites for CO<sub>2</sub> molecules, showing excellent photocatalytic activity for reducing CO2. Huo et al.25 synthesized surface amine-functionalized SnO2 with oxygen vacancies, where the defects served as CO sites, enhancing the CO<sub>2</sub> adsorption capacity, and the presence of surface vacancies altered the local electron density distribution, facilitating electron injection into the  $\pi$  antibonding orbitals of CO<sub>2</sub>, thus effectively activating the CO<sub>2</sub> molecules. However, the total efficiency of photocatalytic CO2 reduction is far from satisfactory to date. Thus, the development of efficient photocatalysts is crucial to promote the CO2 reduction efficiency in photocatalytic systems.

The focus on photocatalyst performance in CO<sub>2</sub> reduction is accompanied with significant interest in the selectivity for the resulting products. Various products (such as CO, HCOOH, CH<sub>3</sub>OH, CH<sub>4</sub>, C<sub>2</sub>, and C<sub>2+</sub>) can result from this photocatalytic process.<sup>26,27</sup> Due to the complexity of the photocatalytic process, controlling the selectivity for the reduction products is very challenging.<sup>28-31</sup> Thus, it is necessary to understand the factors in the reaction to affect the product selectivity for efficient catalyst design. Both thermodynamic and kinetic aspects, including the energy levels of the conduction band and valence band of the photocatalyst,<sup>32,33</sup> reduction potential for CO<sub>2</sub> to the products, CO2 adsorption and activation, intermediate adsorption/desorption,34,35 lifetime and concentration of photogenerated electrons,<sup>36,37</sup> and competing reactions,<sup>38</sup> are the key influencing factors when considering the product selectivity.<sup>39,40</sup> Wang et al.<sup>41</sup> found that Ag could inhibit the H<sub>2</sub>O reduction reaction on a TiO<sub>2</sub> shell, demonstrating the effect of competitive reaction on the product selectivity. It was discovered by Wang et al.<sup>42</sup> that the energy band configuration of TiO<sub>2</sub> can be optimized through the incorporation of cobalt, ultimately influencing the selectivity for the CO<sub>2</sub> photolysis end-products. Xie et al.35 revealed that alkaline earth metal oxides possessed a higher aptitude for CO<sub>2</sub> absorption and activation, subsequently altering the reaction pathway and product selectivity in photocatalytic CO2 reduction. In addition, strong and weak adsorption between the intermediates and photocatalysts will also affect the selectivity of photocatalytic CO<sub>2</sub> reduction. When the interaction between the intermediates and photocatalyst is weak, their desorption and release from the surface of the photocatalyst will be easier, and these substances will become important by-products. Zhang et al.43 found through theoretical calculation that changing the adsorption pattern of the intermediate CO can improve its product selectivity. Wang et al.44 summarized the possible reaction mechanism of photocatalytic CO2 conversion to produce C1 and C2 products. Four principal mechanisms were outlined for the generation of these products on the surface of TiO<sub>2</sub>, including the formaldehyde route, carbene route, glyoxal pathway, and formyl route. Therefore, gaining a holistic comprehension of these distinct reaction mechanisms can facilitate the development of photocatalysts with improved activity and selectivity for CO<sub>2</sub> photoreduction.

This review comprehensively examines the advancements in photocatalyst optimization strategies aimed at enhancing the charge separation in photosynthesis and boosting the  $CO_2$ conversion process. The connection between the architectural design and the efficiency of photocatalysts and selectivity for  $CO_2$  photocatalytic reduction is analysed. Moreover, we outline the impending prospects and obstacles in photocatalyst design, particularly in augmenting the selectivity of  $CO_2$  reduction towards specific products. Serving as a synopsis of yield and selectivity across various photocatalysts for  $CO_2$  reduction, this review is intended to stimulate the advancement of highperformance photocatalysts for this purpose.

## 2 Principle of photocatalytic reduction of CO<sub>2</sub>

Photocatalytic CO<sub>2</sub> reduction with the H<sub>2</sub>O reaction process is complex and involves multiple reaction steps.45,46 The use of light as the excitation energy to trigger CO<sub>2</sub> reduction and H<sub>2</sub>O oxidation reactions on semiconductors can be separated into the following steps (Fig. 1):47-51 (1) adsorption activation of CO<sub>2</sub> molecules. The first key step in photocatalysis involves the absorption of CO<sub>2</sub> on the active sites on the catalyst surface. Adsorption interactions produce partially charged  $CO_2^{\delta-}$ , and different binding modes of  $CO_2^{\delta-}$  will produce different intermediates, which will influence the reaction trajectory to some extent, thus changing the product selectivity. (2) Upon the absorption of light, electron-hole pairs are generated. When the energy of the absorbed light exceeds the bandgap energy  $(E_g)$  of the catalyst, it results in an abundance of photoexcited electrons and holes. These charge carriers migrate towards the catalyst surface, where they engage in redox processes. To ensure that photocatalysts can reduce CO2 and oxidize H2O, semiconductors are required to have appropriate valence and conduction band positions.<sup>39,52-54</sup> The conductive band minimum should lie at a potential lower than that required for  $CO_2$  reduction, whereas the maximum points of the valence band should reside at a potential surpassing the level needed for water oxidation.

(3) Efficient charge carrier mobility and separation are crucial. Given that electron-hole pair recombination occurs significantly faster than charge transport and depletion, the lifespan of photoexcited electrons must be sufficiently



Fig. 1 Basic steps of photocatalytic conversion of CO $_{\rm 2}$  and H $_{\rm 2}O$  on semiconductors.

prolonged to facilitate photocatalytic redox reactions. The efficient separation of photogenerated electrons–holes improves the density of surface photogenerated electrons, which is beneficial for accelerating the redox reaction and promoting the generation of hydrocarbons.<sup>55–60</sup> (4) The photogenerated electrons on the surface active sites undergo a reduction reaction with CO<sub>2</sub> to produce hydrocarbons, while the photo-generated holes undergo an oxidation reaction with H<sub>2</sub>O to produce oxidation products such as O<sub>2</sub> and 'OH. This is a key step in photocatalytic CO<sub>2</sub> reduction. (5) The reaction products are desorbed from the surface of the photocatalyst, and the newly generated surface active sites participate in the next catalytic reaction.<sup>61–64</sup> To design an efficient photocatalyst for CO<sub>2</sub> conversion, the above-mentioned conditions must be satisfied simultaneously.

In terms of thermodynamics, the fundamental determinant of the efficiency in photocatalytic reactions is the semiconductor band structure. The relative position of the semiconductor conduction and valence bands determines the oxidation-reduction ability.65-67 In the context of photocatalytic conversion of CO<sub>2</sub>, it is essential that the conduction band (CB) of the photocatalysts possess a chemical potential that is notably more negative compared to the reduction potential of  $CO_2$ . Conversely, for the process of water oxidation, the valence band (VB) must have a positive potential exceeding the energy level required for water oxidation. Fig. 2 shows the bandgap, conduction band and valence band positions of some common semiconductor catalysts. The potential in Fig. 2 refers to a normal hydrogen electrode (NHE) reference with pH = 7 in an aqueous solution. According to different reaction mechanisms and pathways, CH<sub>4</sub>, CH<sub>3</sub>OH, CO, HCOOH, C<sub>2</sub>H<sub>6</sub>, C<sub>2</sub>H<sub>5</sub>OH, and even H2C2O4 are all possible products of CO2 reduction. Although CO has a high reduction potential, it is easier to produce than CH<sub>4</sub>, HCHO and CH<sub>3</sub>OH due to the fact that its formation only requires two electrons. Usually, when selecting a catalyst for  $CO_2$  reduction, the reduction potential of  $CO_2$  to the required products need to be more positive than the conduction band edge potential of the selected semiconductor.

The thermodynamic characteristics of photocatalytic CO<sub>2</sub> reduction significantly influence both the feasibility of the photocatalytic process and the ultimate reduction products. Alternatively, the kinetic aspects of this reduction govern the pace of the reaction in terms of rate and product formation. The detailed analysis of the photocatalytic kinetics is complex given that it involves many processes,<sup>68–71</sup> including the photogeneration and separation of holes and electrons and the adsorption and activation of the reactant molecules.

(1) Separation of photogenerated electron-hole pairs. In photocatalytic reactions, the recombination lifetime of the photogenerated electrons-holes is very fast, usually only a few picoseconds, exceeding their transport rate from the bulk phase to the surface layer (which takes hundreds of picoseconds). Simultaneously, the recombination rate of charge carriers on the catalyst surface is relatively rapid, occurring within tens of picoseconds, much faster than the speed at which they participate in catalytic reactions (ranging from a few nanoseconds to a few milliseconds).72-77 The swift combination of electrons and holes significantly diminishes the photocatalytic performance. The poor quantum efficiency observed in numerous photocatalysts is attributed to the premature recombination of these charge carriers before engaging with the adsorbed substances. Thus, enhancing the separation of charge carriers or inhibiting their recombination is pivotal to enhancing CO2 reduction reactions. Furthermore, the accelerated migration of photogenerated charge carriers towards the semiconductor catalyst surface contributes to elevating the rate of the multi-electron CO<sub>2</sub> reduction process, thereby enhancing the yield of the end products and the photocatalytic kinetics.54,78-81



Fig. 2 Schematic of the band gap, conduction band and valence band positions of common semiconductor catalysts.

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(2) Adsorption and activation of reactant molecules. The  $CO_2$ molecule is highly inert. In the CO<sub>2</sub> molecule, the carbon atom forms a chemical bond with the oxygen atoms through sp hybrid orbitals, influencing the reaction selectivity within the pair of carbon atom sp hybrid orbitals combined to create two  $\sigma$ bonds with the orbitals from two oxygen atoms, respectively. The other two p orbitals on the carbon atom, which are not involved in hybridization, overlap shoulder-to-shoulder with the p orbitals of the oxygen atom and form two delocalized bonds with three centres and four electrons.5,17,29,82 This structure shortens the distance between the C-O atoms and endows CO<sub>2</sub> molecules with certain triple bond characteristics. Hence, the C-O bond dissociation necessitates a significant energy input, making the thermodynamic activation of CO2 molecules arduous. The key kinetic aspects for the initiation of CO<sub>2</sub> reduction are the absorption and activation of CO<sub>2</sub> molecules on the catalyst surface. In photocatalytic reduction processes, CO<sub>2</sub> molecule activation demands its adsorption on the catalyst surface, followed by a structural transition from a linear to bent configuration. This is followed by the transfer of electrons from the photocatalyst surface to the distorted CO<sub>2</sub> molecules, thus facilitating CO2 activation.83-86

Consequently, the engagement of  $CO_2$  with the catalyst surface plays a crucial role in its absorption, effectively stimulating CO<sub>2</sub> molecules. During the photocatalytic conversion of CO<sub>2</sub>, the adsorption of CO<sub>2</sub> onto the photocatalyst surface primarily involves three types of adsorption modes including oxygen coordination, carbon coordination, and mixed coordination (Fig. 3). In this mechanism, the interplay between CO<sub>2</sub> and surface atoms generates partially charged  $CO_2^{\delta-}$  species. The production of the characteristic activation intermediates is decisive for the reaction route, which influences the selectivity for diverse products.<sup>29,39</sup> Furthermore, the dynamic characteristics of the multistage surface catalytic reaction in photolytic CO<sub>2</sub> conversion reveal that the catalytically active sites on the surface of a photocatalyst play a role in determining the product selectivity. Consequently, diverse active sites on the catalyst surface can give rise to distinct products.68,87-90 For example, the monodentate binding of C atoms with Lewis bases forms carboxyl radicals ('COOH) centred on the catalyst, while the



**Fig. 3** Three different CO<sub>2</sub> adsorption modes on the photocatalyst surface: (a) oxygen coordination; (b) carbon coordination; (c) mixed coordination. Reproduced from ref. 29 with permission from [the Royal Society of Chemistry], copyright [2016].

bidentate binding of two O atoms tends to favour H to link the carbon of  $CO_2^{\delta^-}$ , which causes formate anions to bind in a bidentate manner on the catalyst surface.

Activated  $CO_2$  molecules engage in a reaction with electrons that have been photo-induced and displaced towards the catalytic centre. To enhance the adsorption and activation of  $CO_2$ , photocatalysts with large surface areas or surface alkaline groups are usually used in the preparation of catalysts to provide more active sites.<sup>91-93</sup> The strategies employed include introducing defect vacancies,<sup>94-96</sup> introducing cocatalysts,<sup>97-101</sup> increasing the specific surface area and porosity of the catalyst,<sup>102,103</sup> and adjusting the acidity and alkalinity of the catalyst surface.<sup>104,105</sup>

In the photocatalytic process, the thermodynamics and kinetics are intertwined in the reaction, jointly determining the reaction efficiency and products. Thus, to improve the photocatalytic efficiency, it is necessary to comprehensively consider these two factors, optimizing the band structure, surface chemical properties, reaction conditions and photocatalytic system.<sup>106-108</sup>

Thus far, many reaction mechanisms for photocatalytic CO<sub>2</sub> reduction have been proposed.<sup>109,110</sup> The main three pathways of formaldehyde, carbene and glyoxal pathways are shown in Fig. 4. In the formaldehyde pathway, CO<sub>2</sub> initially undergoes electron excitation to form CO<sub>2</sub>. Subsequently, this species combines with a proton to yield a formic acid intermediate. This intermediate accepts additional protons to produce formic acid. Furthermore, through processes of electron coupling and proton transfer, formic acid serves as a foundation for the production of formaldehyde, methanol, and methane. This pathway can be used to explain the formation of HCOOH, HCHO and CH<sub>4</sub>, but the production of CO cannot be explained.111-114 In the carbene pathway, the production of CO can be explained. Activated CO2 accepts electrons to produce CO, and CO can be used as an intermediate product for the further formation of methane or methanol. C-C bond formation depends on the stabilization of the intermediary radical to prevent its recombination with the above-mentioned hydrogen atoms. In contrast, the glyoxal pathway predominantly yields C2 compounds, where CO<sub>2</sub> combines with electrons and protons to form formic acid, resulting in free formyl radicals (CHO') due to oxygen transfer and further electron acceptance. Eventually, the formyl groups are dimerized into glycol or evolve into other C2 and C<sub>3</sub> products. The photocatalytic conversion of CO<sub>2</sub> involves multiple steps, intermediates, and diverse end-products, making the process of product selectivity more complicated.54,114-116 Currently, CO2 conversion to C2+ products via the C-C coupling process mainly include the following reaction pathways: (a) \*CH<sub>3</sub> coupling, (b) \*CO hydrogenation to \*CHO and further coupling of \*CO and \*CHO, (c) \*CHO dimerization, and (d) \*CO linkage to \*COCO and a series of oxygen-binding intermediates,117 where the dimerization of the C1 intermediates on the catalyst surface to form C-C bonds is the key to the further formation of  $C_2$  products (Fig. 5). For example, the pathway to produce ethylene or ethane involves the coupling of carbene intermediates (CH<sub>2</sub> and CH<sub>3</sub>) in the carbene pathway, while the dimerization of formyl groups



Fig. 4 Three possible paths for carbon dioxide reduction: (a) formaldehyde pathway, (b) carbene pathway and (c) glyoxal pathway.

(CHO) in the glyoxal pathway can produce multiple  $C_1$  and  $C_2$ products. Given that the C<sub>2</sub> product involves multi-electroncoupled proton transfer and C-C coupling processes, the reactions are often more challenging. The dehydration step after  $CO_2$  receives 3 protons, forming the formyl radical (HC\*O). Subsequently, the formyl radical dimerizes to form glyoxal, which is reduced to the ethylene oxy (\*CH<sub>2</sub>-CHO) radical. This is further converted to acetaldehyde (CH<sub>3</sub>CHO) or C<sub>2</sub>H<sub>5</sub>OH, and the production of C<sub>2</sub>H<sub>4</sub> can be caused by CO dimerization, which has several steps similar to the glyoxal pathway, except that the intermediate (CH<sub>2</sub>CHO) changes from C-coordination to O-coordination.44,118-120 To attain precise control of these products, the key aspects of the photocatalytic CO<sub>2</sub> reaction pathway can be manipulated, such as manipulating the energy band architecture, enhancing the charge separation efficacy, and optimizing the absorption and activation of reactants.112-114

## 3 Strategies to design photocatalysts

Numerous photocatalysts have been explored for the photocatalytic reduction of  $CO_2$  into a value-added fuel, such as metal oxide (TiO<sub>2</sub>, ZnO, Cu<sub>2</sub>O, InVO<sub>4</sub>, Bi<sub>2</sub>WO<sub>6</sub>, *etc.*), metal sulphide (CdS, Ag<sub>2</sub>S, ZnIn<sub>2</sub>S<sub>4</sub>, *etc.*), carbon-based semiconductor (SiC, g- $C_3N_4$ , graphene, *etc.*), MOFs, COFs and other materials.<sup>17,23</sup> Nonetheless, these catalysts are associated with numerous limitations, including inadequate sunlight absorption efficacy, rapid recombination of photoexcited electrons and holes, and lack of active sites on their surface. These problems eventually lead to unsatisfactory reaction efficiency and product selectivity.<sup>55,56,121,122</sup> Therefore, many modification strategies have been commonly used for the preparation of photocatalysts, such as cocatalyst loading, doping, defect engineering, heterojunction formation, single-atom engineering, and surface



Fig. 5 Currently reported C–C coupling reaction pathways for photocatalyzed  $CO_2$  conversion to  $C_{2+}$  products. Reproduced from ref. 117 with permission from [ACS Publications], copyright [2023].



Scheme 2 Summary of commonly used modification strategies for photocatalysts.

organometallic catalysis, as shown in Scheme 2. We attempted get inspiration from the relevant cases to provide some new insights for the design of new efficient photocatalysts.

### 3.1 Defect construction

The balanced coordination of charge generation and separation in photocatalysts is crucial in the photocatalytic process. It has been reported that structural defects in semiconductors can improve the mobility of carrier charges and extend their lifetime to maintain the reaction kinetics,<sup>123</sup> including 3D volume defects (voids and disorders).<sup>124–126</sup> Point defects are generated through the removal of atoms in the lattice or doping. Line defects are formed by the dislocation of atoms in the lattice. Planar defects refer to the arrangement of atoms on both sides of the crystal surface, which is characterized by the arrangement of atoms deviating from the equilibrium position in the twodimensional direction.<sup>25,95,96,127–129</sup> Volume defects are caused by introducing a variety of substances into one or more crystal locations, resulting in lattice gaps or chaos. All types of defects affect the electronic structure and catalytic performance of metal oxides (MOS).<sup>130,131</sup>

Since Chen *et al.*<sup>132</sup> synthesized black  $TiO_2$  in 2011, research has focused on oxygen vacancies (OVs). It has been reported that defects can change the surface properties to expose new active centres in MOs.<sup>121,133</sup> The presence of vacancies is favourable for photoinduced electrons to preferentially enter the vacancies on the catalyst surface, suppressing the recombination loss of photogenerated electron–hole pairs, facilitating charge transportation, and enhancing surface chemical reactions. In addition, the binding mode and adsorption strength of  $CO_2$  on the surface of MOs are also related to the properties of the internal and external defects. OVs on the surface play key roles in adsorption and activation.<sup>134</sup> Liu<sup>135</sup> synthesized Cu(1)/ $TiO_{2-x}$  with surface OVs by annealing with an inert gas at 300 ° C. The results showed that the catalyst with defects had a higher



**Fig. 6** (a)  $CO_2$  adsorption mechanism diagram of TiO<sub>2</sub> with oxygen vacancies. (b) Low-temperature ammonium-assisted reduction of nitrogenfree doped  $WO_{3-x}$  nanorods. (c) Synthesis of  $CeO_2$  rich in anionic defects. (d and e) Photocatalytic  $CO_2$  reduction yield and stability of modified Bi1<sub>2</sub>O<sub>17</sub>Cl<sub>2</sub>. (a) Reproduced from ref. 135 with permission from [ACS Publications], copyright [2012]. (b) Reproduced from ref. 136 with permission from [Cell], copyright [2012]. (c) Reproduced from ref. 137 with permission from [Elsevier], copyright [2022]. (d and e) Reproduced from ref. 138 with permission from [Wiley Online Library], copyright [2018].

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CO yield than the catalyst without defects, which provided some active sites of Ti<sup>3+</sup> for the adsorption of O atoms from CO<sub>2</sub> (Fig. 6a). Thus far, many effective methods have been developed to control the formation of vacancies, such as calcination, hydrothermal/ chemical reduction, ion doping and solvothermal methods. High-temperature calcination is a process in which the sample is calcined at high temperature through an inert or reducing atmosphere to promote the diffusion and escape of surface lattice oxygen, and thus generate OVs. The vacancy concentration is affected by the calcination temperature and time. Generally, increasing the heating temperature and extending the calcination time can improve the vacancy concentration.<sup>95,127,129,139</sup> For example, Tan et al.140 found that annealing SrTiO3 at 375 °C had a higher OV concentration than annealing at 300 °C, and that extending the heating time by half an hour contributed to the formation of OVs. However, it tended to generate more OVs at a lower temperature under vacuum conditions.

Xing *et al.*<sup>141</sup> modified  $\text{TiO}_2$  with a low-temperature vacuum activation method to obtain stable  $\text{Ti}^{3+}$  and OVs. This method can also be used to enhance the visible light response of ZnO, WO<sub>3</sub> and other oxides. It is worth noting that a higher vacancy concentration is not better. A too high OV concentration may introduce a new impurity band with an unlocalized molecular orbital, which can trap electrons excited from the valence band and prevent them from reaching the conduction band, thus inhibiting charge migration.<sup>142</sup>

In contrast to high-temperature calcination, chemical reduction can produce vacancies at low temperature. When the reducing molecules adsorb on the MO surface, electrons transfer to the surface to capture the O atoms, finally forming OVs.<sup>136</sup> For example, Liu *et al.*<sup>136</sup> synthesized WO<sub>3-x</sub> nanorods *via* a low-temperature ammonium-assisted reduction method (Fig. 6b), showing an excellent photocatalytic performance in the reduction of CO<sub>2</sub> and H<sub>2</sub>O to CH<sub>4</sub>. Xu *et al.*<sup>143</sup> introduced OVs on the surface of SrTiO<sub>3</sub> by NaBH<sub>4</sub> treatment, and the vacancy led to the formation of intermediate gap states. Combined with the inherent energy level of SrTiO<sub>3</sub>, the SrTiO<sub>3-x</sub> photocatalyst exhibited enhanced activity in the reduction of CO<sub>2</sub> to CO under visible light irradiation.

Ion doping can also create vacancies in photocatalysts. The different chemical valence ion dopants replace the metal nodes in the oxide substrate, thus destroying the lattice oxygen order, and OVs are generated to maintain the charge balance. The use of cationic defects is a common method to introduce vacancies in MOs by replacing the original cation with a lower cost state or one with a lower vacancy formation energy.<sup>144</sup> For example, Zhou et al.<sup>145</sup> doped Ru in TiO<sub>2</sub> nanocrystals by the hydrothermal method and induced the formation of OVs. Metal ions and OVs cooperated to promote the adsorption of CO<sub>2</sub> and the separation of photogenic carriers, thus achieving the photocatalytic conversion of CO2 and H2O into CH4. Lai et al.137 synthesized Fe-containing CeO<sub>2</sub> with an OV porous structure via a one-step combustion method (Fig. 6c). The porous structure was conducive to the adsorption of CO<sub>2</sub>, and OVs promoted the activation of CO<sub>2</sub>. In addition to the doping of cations, anion defects can be constructed on MOs by introducing anions such as N, P, S, and halogens. Zeng *et al.*<sup>146</sup> calcined  $Co(NO_3)_2$  on an SBA-15 template and reintroduced fluorine to synthesize  $Co_{3^-}O_{4-x}F_x$  with defective, fluorine substitution and threedimensional mesoporous structure. The fluorine substitution and partial OVs altered the electronic structure to promote the ability to adsorb intermediates. This study proposed a new strategy of combining defects with fluorine chemistry, providing an idea for the design of defective fluoride oxides.

Hydrothermal/solvothermal reaction is a technique to synthesize crystalline materials by controlling the thermodynamic variables such as solvent type, additive composition, temperature and pressure, which can greatly control the morphology and structure, size and orientation of crystallinity. It is one of the best methods to prepare nanoscale materials.<sup>147</sup> Zhang *et al.*<sup>148</sup> synthesized Ti/TiO<sub>2</sub> interfacial photocatalysts with different concentrations of Ti<sup>3+</sup> sites *via* a hydrothermal one-step method. It was found that the interfacial coordination of unsaturated defect sites can not only narrow the band gap, but also promote the transfer and separation of interfacial photocatalytic activity.

In recent years, researchers have worked on developing bismuth oxide halide-based photocatalysts. OVs are easily introduced in BiOX nanotubes and thin nanosheets. Di et al.138 constructed Bi<sub>12</sub>O<sub>17</sub>Cl<sub>2</sub> featuring a twin-layered ultrathin tubular architecture enriched with surface oxygen vacancies. Under the illumination of a 300 W xenon lamp, the CO production rate for this engineered catalyst averaged 16.8 times higher (48.6  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>) than the pristine bulk Bi<sub>12</sub>O<sub>17</sub>Cl<sub>2</sub>. Concurrently, it accomplished the conversion of H<sub>2</sub>O to O<sub>2</sub>, with an average  $O_2$  yield of 23 µmol g<sup>-1</sup> h<sup>-1</sup>, sustaining its stability over 12 h of activity assessment (Fig. 6d and e). Wu et al.149 prepared BiOBr lamellar precursors via the hydrothermal method, and then separated them into atomic layers by ultrasound in an ice water bath. Finally, a large number of OVs was introduced on the surface of the atomic layers after 8 h ultraviolet irradiation because BiOBr with a high oxygen atomic density exposed (001) surface easily formed OVs under the irradiation of high-energy light (Scheme 3). Owing to the augmented light absorption in the visible spectrum of the modified BiOBr atomic layers and the facilitation of CO2 conversion into intermediary COOH\* by the surrounding surface oxygen vacancies, the reduction rate witnessed a remarkable increase of 24 times compared to pristine BiOBr (yielding 87.4  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>). These findings suggest that the introduction of defects on the catalyst surface can effectively optimize the CO<sub>2</sub> reduction mechanism, thereby paving the way



Scheme 3 Synthesis process of oxygen-deficient BiOBr photocatalysts. Reproduced from ref. 149 with permission from [Wiley Online Library], copyright [2018].

for innovative strategies in designing high-performance catalysts that operate under visible light for  $\rm CO_2$  reduction purposes.

In addition to oxygen vacancies, sulfur defects are very common in sulfide catalysts, and sulfur defects play an important role in photocatalytic CO<sub>2</sub> reduction by these catalysts. For instance, Yin et al.<sup>150</sup> created S vacancies in a thin layer of SnS<sub>2</sub> atoms through Ar plasma bombardment. It was found that the S vacancy could reduce the reactive energy barrier for water decomposition by constructing a specific reactive configuration. This led to a direct enhancement in water oxidation during the entire process of photocatalytic CO2 reduction by promoting the formation of O<sub>2</sub> molecules. Ultimately, a substantial improvement in the performance for the photoreduction of CO<sub>2</sub> was achieved. The photocatalytic CO<sub>2</sub> reduction process exhibited a CO yield of 25.71  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, which was 8.2 times higher than that of the initial SnS<sub>2</sub> catalyst. Gao et al.<sup>151</sup> proposed the introduction of sulfur defects in an atomic layer of AgInP<sub>2</sub>S<sub>6</sub> using an H<sub>2</sub>O<sub>2</sub> etching method. Through DFT calculation and in situ FTIR spectroscopy, it was demonstrated that introducing a sulfur vacancy in AgInP<sub>2</sub>S<sub>6</sub> can accumulate charge on the Ag atoms near the sulfur vacancy, allowing the exposed Ag site to effectively capture formed \*CO molecules. This enrichment in key reaction intermediates on the catalyst surface reduced the C-C binding coupling barriers and promoted ethylene production with a yield selectivity of up to 73%.

### 3.2 Cocatalyst loading

The integration of a metal co-catalyst in a photocatalyst significantly contributes to the augmentation of its photocatalytic performance. In certain instances, the addition of a co-catalyst not only boosts the efficiency of the photocatalytic reaction but also efficaciously influences the selectivity for the resulting products. The dispersion of the cocatalyst should be controlled reasonably during the catalytic process. Excess or slight amounts of active sites are not conducive to the reaction.97-101 Adrian Quindimil et al.<sup>152</sup> prepared Ni/Al<sub>2</sub>O<sub>3</sub> and Ru/Al<sub>2</sub>O<sub>3</sub> catalysts with different metal loadings. The loading of Ni and Ru provided new alkaline sites to promote the adsorption of CO<sub>2</sub>, and the noble metal Ru on Al<sub>2</sub>O<sub>3</sub> was more efficient in dissociating H<sub>2</sub> than the non-noble metal Ni. The conversion rate of CO2 was the highest when the loading of Ni was 12% and the loading of Ru was 4%. In addition, the presence of H<sub>2</sub> adsorption/dissociation points is important for the methanation of CO<sub>2</sub>, but pure MOs usually lack H<sub>2</sub> dissociation points. Thus, to solve this problem, Dreyer et al.<sup>153</sup> used flame spray pyrolysis to disperse 5% Ru on  $Al_2O_3$ , ZnO, MnO<sub>x</sub>, and CeO<sub>2</sub>, and found that the supported Ru nanoparticles could provide H<sub>2</sub> dissociation points, thus significantly improving the yield and selectivity for CH<sub>4</sub>.

The n-type semiconductor  $\text{TiO}_2$  is one of the most commonly used photocatalysts because of its advantages such as low price, non-toxic nature, and resistance to photocorrosion.<sup>154</sup> Dating back to 1978, Kraeutler and Bard first loaded Pt on TiO<sub>2</sub> for CO<sub>2</sub> reduction *via* photodeposition.<sup>155</sup> Subsequently, many teams were inspired to prove that the selectivity of CH<sub>4</sub> would be improved after loading the noble metal Pt on TiO<sub>2</sub>. They believed that the selective conversion of the CO product to CH<sub>4</sub> was because the Pt loaded on the photocatalyst surface formed a Schottky junction, which could extend the photoelectron lifetime, thus lowering the reaction barrier and promoting the formation of CH4.156-158 Thus, to explore the reason for the selective transformation of CO2 reduction after Pt loading, Ma et al.<sup>159</sup> loaded 1 wt% Pt on TiO<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub> and BiOBr, respectively, by photodeposition, and found that the selectivity of the three photocatalysts for CH<sub>4</sub> increased to nearly 100% (Fig. 7a). The gas adsorption energy and charge transfer amount for  $CO_2$ ,  $CO_2$ and CH<sub>4</sub> of Pt clusters with two configurations were calculated (Fig. 7b), and the results were attributed to the following reasons: (1) CO has strong adsorption capacity on Pt, and thus its desorption from the catalyst surface is difficult, preventing the detection of CO. (2) CO is used as an intermediate, and CO enriched on the Pt clusters is more conducive to obtaining electrons to form CH<sub>4</sub>. However, the adsorption capacity of CH<sub>4</sub> on Pt is very weak, which is manifested as physical adsorption and easy desorption. Therefore, the yield and selectivity for CH<sub>4</sub> were improved after Pt loading. This study provided further insight into how Pt promotes the highly selective production of CH<sub>4</sub> from photocatalyzed CO<sub>2</sub>, as well as a new strategy for exploring photocatalyzed CO2 reduction to CH4. Xiong158 deposited Pt and Cu2O nanoparticles together on the surface of anatase TiO<sub>2</sub> via NaBH<sub>4</sub> reduction. Compared with the experimental results for Pt or Cu<sub>2</sub>O loaded on TiO<sub>2</sub>, it was found that Pt promoted the formation of CH<sub>4</sub> and H<sub>2</sub>, whereas Cu<sub>2</sub>O augmented CO<sub>2</sub> absorption on the TiO<sub>2</sub> surface. It concurrently suppressed the chemical adsorption of H<sub>2</sub>O, subsequently hindering H<sub>2</sub> production. The combined effect of Pt and Cu<sub>2</sub>O facilitated the exceptionally selective conversion of CO2 into CH<sub>4</sub>, offering insights into the development of multifaceted photocatalysts (Fig. 7c). H<sub>2</sub> competition in the process of CO<sub>2</sub> and H<sub>2</sub>O photocatalysis is unfavourable for CO<sub>2</sub> reduction, and thus the cocatalyst should be selected to enhance CO2 adsorption and weaken H<sub>2</sub>O adsorption simultaneously.

The modification strategy of loading metals on TiO<sub>2</sub> alone has been widely reported. The electronic structure and geometric configuration of two metal alloy catalysts will be changed due to their combination, making the chemical interaction between the reaction intermediates and catalyst surface different. It can not only effectively reduce the kinetic overpotential but also control the product selectivity.<sup>161</sup> Neatu et al.<sup>160</sup> deposited Cu and Au nanoparticles on TiO<sub>2</sub> for two consecutive times and found that CO2 was reduced to CH4 in H<sub>2</sub>O under visible light irradiation. On the contrary, CH<sub>4</sub> was not detected under ultraviolet light irradiation, but  $H_2$ production increased significantly. The charges on TiO<sub>2</sub> were directly separated and electrons migrated to the alloy nanoparticles under ultraviolet irradiation. At this time, the Au-Cu alloy stored electrons and acted as a co-catalyst. Both the bare TiO<sub>2</sub> and Cu/TiO<sub>2</sub> were inactive under visible light irradiation, while Au nanoparticles acted as a light collector due to their plasma effect. During the reaction, electrons from Au were transferred to the oxidized Cu surface, and the Au atoms activated by plasma successfully were reduced.162 The main active

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**Fig. 7** (a) Product selectivity of three photocatalysts:  $Pt/TiO_2$ ,  $Pt/g-C_3N_4$  and Pt/BiOBr. (b) Gas adsorption energy and charge transfer of  $CO_2$ , CO and  $CH_4$  by two Pt clusters in the three photocatalysts. (c)  $CO_2$  reduction yield and selectivity of  $TiO_2$ ,  $Pt/TiO_2$ ,  $Cu_2O/TiO_2$  photocatalysis. (d) Schematic of the effect of different excitation wavelengths on the selectivity of  $TiO_2$  products. (a and b) Reproduced from ref. 159 with permission from [the Royal Society of Chemistry], copyright [2018]. (c) Reproduced from ref. 158 with permission from [Elsevier], copyright [2017]. (d) Reproduced from ref. 160 with permission from [ACS Publications], copyright [2014].

sites of  $CO_2$  existed on the surface of the reduced Cu atoms. The activation of  $CO_2$  by a single electron followed the so called "carbine pathway" to produce  $CH_4$ , and the excess holes were used to oxidize  $H_2O$  to produce  $O_2$  (Fig. 7d).

Enhancing the absorption and activation of CO<sub>2</sub> plays a pivotal role in boosting the efficiency of CO<sub>2</sub> reduction. Research indicates that the process of CO2 adsorption and activation can be significantly augmented by augmenting the specific surface area of the catalyst<sup>163,164</sup> and multiplying the available sites for CO2 adsorption.165-167 Therefore, for some photocatalytic semiconductors without CO2 adsorption and activation sites, it is necessary to fix the cocatalyst on the catalyst through doping or loading to construct active sites. Thus far, doping and loading synergistic strategies have also been widely used to improve the photocatalytic CO2 reduction performance. Wang et al.<sup>168</sup> modified Ga<sub>2</sub>O<sub>3</sub> by Zn species through impregnation and calcination to produce ZnGa<sub>2</sub>O<sub>4</sub>, and then supported a certain amount of Ag catalyst on ZnGa<sub>2</sub>O<sub>4</sub> surface by photodeposition, and the production of the products was improved by regulating the addition amount of Zn and the loading amount of Ag. It was found that the H<sub>2</sub> production decreased continuously when the additive amount of Zn was in the range of 0.1 mol% to 10 mol%, while the production of CO was almost unchanged, and thus the selectivity for the CO product reached nearly 100%. In addition, Huang et al.<sup>169</sup> synthesized a Pt/CoO<sub>x</sub>/N-TiO<sub>2</sub> composite photocatalyst, which was inspired by loading suitable double cocatalysts on the same catalyst. The synthesis process is shown in Scheme 4. Firstly, the layered titanite protonated precursor was prepared via the solvothermal method, and then the precursor was calcined for 2 h at 450 °C in an air atmosphere to obtain N-doped TiO<sub>2</sub> layered mesoporous spheres. The N-doping process was conducive to the introduction of OVs on the surface of TiO<sub>2</sub>. Then, Co(CH<sub>3</sub>- $(COO)_2 \cdot 4H_2O$  was used as a Co source to support  $CoO_x$  on Ndoped  $TiO_2$  via the in situ growth method.  $CoO_x$  was used as an oxidation cocatalyst for H<sub>2</sub>O oxidation. Finally, H<sub>2</sub>PtCl<sub>6</sub> was successfully reduced to Pt by photodeposition. The Pt loaded on the surface VOs of TiO<sub>2</sub> could capture and activate CO<sub>2</sub> as a reduction cocatalyst. Due to the synergistic action of VOs on the TiO<sub>2</sub> surface and the double cocatalyst, the yield of CO<sub>2</sub> conversion to CH<sub>4</sub> was further increased to 409.17  $\mu$ mol g<sup>-1</sup>. In this study, the efficiency of photocatalytic CO2 reduction and H<sub>2</sub>O oxidation was improved by introducing OVs to selectively deposit two cocatalysts on the surface of semiconductor materials to achieve dual active sites. However, it is worth noting that although the assumption that the entire REDOX reaction is separated into two half-reactions is conducive to the development of research, the distribution of cocatalysts on the semiconductor surface is random and uncontrollable, and there are still problems such as low yield, instability, and selectivity hurdles with co-catalysts such as sulphides. Hence, gaining profound insights into the photoexcited electron-hole transportation dynamics at the interface of co-catalyst/photocatalyst composites holds paramount importance for optimizing the cocatalyst design to enhance CO2 photocatalysis. Advanced characterization techniques can facilitate the visualization of



Scheme 4 Process for the synthesis of composite photocatalytic materials with Pt atoms and CoO<sub>x</sub> dual co-catalysts. Reproduced from ref. 169 with permission from [Elsevier], copyright [2022].

electron-hole transfer, reaction intermediates, and endproducts, thereby elucidating the intricacies of the reaction mechanism.

### 3.3 Doping

Doping different elements in a catalyst is one of the common strategies to optimize its catalytic performance, which can be divided into non-metallic doping, metal doping, and co-doping according to the type of doping elements. Graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>), a typical representative of carbon-based materials, has emerged as one of the highly desirable photocatalysts, owing to its responsiveness to visible light, stable physical and chemical properties, convenient preparation, and other advantages. Furthermore, the bandgap of g-C<sub>3</sub>N<sub>4</sub> is approximately 2.7 eV, with its conduction band situated at around -1.1 eV. This negatively positioned conduction band is notably lower than the reduction potential required for converting CO<sub>2</sub> into CO, CH<sub>4</sub>, CH<sub>3</sub>OH, and various other fuel molecules.<sup>170-176</sup>

Regrettably, the light absorption capacity and surface area achieved through the conventional calcination of  $g-C_3N_4$ remain inadequate, and the swift recombination of photoexcited electrons and holes significantly hampers the catalytic efficiency of bulk  $g-C_3N_4$ . Thus, researchers suggested the incorporation of distinct components into  $g-C_3N_4$  to modify its crystal and electronic configuration, thereby manipulating its bandgap size, enhancing its absorption in the visible light spectrum, augmenting its specific surface area, and boosting its charge separation effectiveness.

g- $C_3N_4$  is a type of non-metallic semiconductor. If nonmetals are added to it, then it can continue to maintain its non-metallic properties, such as high ionization energy and electronegativity.<sup>177</sup> The common non-metallic doping elements include B, P, S, O, and halogens. During the doping process, these elements easily acquire electrons to form covalent bonds with other compounds. The doping of g- $C_3N_4$  is divided into interstitial doping and substitution doping based on the distance between and within the plane of the principal lattice planes and the atomic radius of the dopant.<sup>178</sup> Substitution doping usually forms a more stable structure than interstitial doping.<sup>179-181</sup> Its dopant atoms displace the atoms in the main lattice and form chemical bonds with the surrounding atoms. For example, S atoms and N atoms have a similar atomic radius, and thus S atoms can easily replace N atoms, and eventually form an S–C bond in g- $C_3N_4$ .<sup>180</sup> Arumugam *et al.*<sup>182</sup> reported the comparative effect of the incorporation of non-metals B, O, P and S into g- $C_3N_4$  for the photocatalytic reduction of CO<sub>2</sub> to CH<sub>4</sub> with H<sub>2</sub>O. The synthesis process of the catalyst is demonstrated in Scheme 5. Sodium dihydrogen phosphate was used as the P source, boric acid as the B source, thioureone as the S source, and hydrogen peroxide as the O source. Upon thorough



Scheme 5 Schematic of the fabrication of porous  $g-C_3N_4$ , B, O, P, and S-doped  $g-C_3N_4$  samples. Reproduced from ref. 182 with permission from [Elsevier], copyright [2022].

### Review

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blending, the catalyst was subjected to heat treatment at a temperature of 550 °C, at a gradual rate of 5 °C per minute, under an inert gas environment, and maintained for a duration of 5 h. The observations indicated an enhancement in CH<sub>4</sub> production when four non-metal elements were introduced in the mixture, surpassing the initial  $g-C_3N_4$  yield. Notably, the sample doped with S demonstrated the highest methane formation rate at 7.8 mmol  $g^{-1} h^{-1}$ . Although S doping did not increase the specific surface area or boost the light absorption abilities of the catalyst, it notably facilitated the efficient separation and mobility of charges. Another type of interstitial doping refers to the mixing of atoms in the gap of the primary lattice. If elements with a large radius different from C and N atoms are added to g-C<sub>3</sub>N<sub>4</sub> (such as Na, K, Br, and I), interstitial doping tends to occur.183,184 In the study by Zhu and colleagues,185 the impact of incorporating F, Cl, Br, and I on the catalytic properties of single-layered g-C<sub>3</sub>N<sub>4</sub> was investigated through first-principles calculations. They proposed that the dual-coordinated N atoms served as the key active sites for photocatalytic reactions, but the dual-coordination N atoms were not always the most stable doping sites.<sup>186</sup> In the case of halogen atoms, they were preferentially doped into the interstitial space surrounded by three triazine units. The F element contributed to the VB, while other halogen atoms participated in the conduction band, reducing the band gap from 1.18 eV to 0.64-1.14 eV. It also exhibited strong light absorption in the wavelength range of 200-1000 nm.184

Without introducing foreign elements,  $g-C_3N_4$  can be selfdoped under certain conditions, that is, C-doped and Ndoped. It has been found that the uniform substitution of N atoms by C atoms during C self-doping causes the formation of delocalized  $\pi$  bonds between the substituting C and the sixmembered ring, which affects the band and electron structure of  $g-C_3N_4$ . This results in a decrease in its band gap, together with an elevated rate of charge-carrier migration and light absorption.<sup>187</sup> In the process of N self-doping, N atoms are

prone to lose extra electrons and become n-type semiconductors, resulting in an intermediate energy gap state to enhance the visible light absorption, which improves the separation efficiency of electrons and holes, reduces the recombination of photogenerated carriers and prolongs the lifetime of photogenerated charges.<sup>188</sup> Fang et al.<sup>188</sup> calcined the precursor and nitrogen-rich additives together to prepare  $C_3N_{4+x}$ . When there were more dopants, it showed metal conductivity, which absorbed more visible light than the original g-C<sub>3</sub>N<sub>4</sub> and accelerated the charge transfer. Liu et al.<sup>189</sup> calcined urea and carboxyl-functionalized polystyrene nanospheres in a furnace at 550 °C for 2 h and polymerized them into N-doped C-grafted carbon nitride (Fig. 8a). The modification led to a significant increase in the CO<sub>2</sub> adsorption capacity, jumping from the initial 0.04 mmol  $g^{-1}$  to 0.22 mmol  $g^{-1}$  at a temperature of 298 K. The improved performance was attributed to the heightened presence of N atoms following the N self-doping of  $C_3N_4$ , which boosted the concentration of basic sites. Operating under visible light, the N-doped Cmodified carbon nitride catalyst exhibited a remarkable CO generation rate of 15.4  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, representing a significant sevenfold enhancement over the undoped g-C<sub>3</sub>N<sub>4</sub>. Introducing alkali metals into g-C<sub>3</sub>N<sub>4</sub> is another facile strategy to enhance its photocatalytic efficiency. The interplay between the doped alkali metal and triazine ring in g-C<sub>3</sub>N<sub>4</sub> leads to an electron shift from the nitrogen atom lone pair to the metallic atom, causing electron redistribution. This process suppresses the recombination of photogenerated carriers and strengthens the absorption of visible light. Zhang et al.<sup>190</sup> studied the effect of doping g- $C_3N_4$  with Li, Na, K and Rb on its photocatalytic  $CO_2$  reduction performance. Among them, Rb-doped g-C<sub>3</sub>N<sub>4</sub> showed the strongest light absorption. Rb-doped g-C3N4 had a thinner lamellar structure with mesopores than pure  $g-C_3N_4$  (Fig. 8b-e). The porous structure was favourable for light absorption and CO<sub>2</sub> adsorption. The change in the electronic structure of all the g-C<sub>3</sub>N<sub>4</sub> samples doped with different alkali metals resulted in



**Fig. 8** (a) Diagram of N-doped C-grafted carbon nitride. (b–e) TEM images of undoped  $g-C_3N_4$  and the  $g-C_3N_4$  doped with Rb. (f) Change in band gap before and after doping Rb and change in optical response after doping different alkali metals. (a) Reproduced from ref. 189 with permission from [Elsevier], copyright [2021]. (b–f) Reproduced from ref. 190 with permission from [Elsevier], copyright [2020].

a redshift in its optical response. After doping Rb, the band gap of the catalyst decreased from 2.65 to 2.0 eV (Fig. 8f).

In addition to alkali metal doping, g-C<sub>3</sub>N<sub>4</sub> doped by Co, Ni, Cu and other metals has also been widely studied. Wang et al.<sup>191</sup> synthesized an Mo-doped g-C<sub>3</sub>N<sub>4</sub> photocatalyst with a vermicular mesoscopic structure and higher surface area by pyrolysis. After 8 h UV irradiation, g-C<sub>3</sub>N<sub>4</sub> doped with 4% Mo showed the highest activity, reducing CO<sub>2</sub> to CO and CH<sub>4</sub> with yields of 887  $\mu$ mol g<sup>-1</sup> and 123  $\mu$ mol g<sup>-1</sup> (Fig. 9a and b), respectively. The introduction of Mo led to an expansion in the specific surface area, boosting the light absorption efficiency, prolonging the lifespan of photogenerated carriers, and consequently elevating the photocatalytic performance. Similarly, Dong et al.194 used MgCl<sub>2</sub> as a magnesium source to add different amounts of Mg to  $g-C_3N_4$  via a one-pot method. The integration of 4 wt% Mg to g-C<sub>3</sub>N<sub>4</sub> led to a reduction in its bandgap from 2.70 to 2.65 eV, concurrently creating an intermediate gap. This newly formed mid-gap state functioned as an electron trap, enhancing the efficiency of electron-hole pair separation. Consequently, the 4 wt% Mg-doped g-C<sub>3</sub>N<sub>4</sub> catalyst demonstrated the optimal photocatalytic performance under both ultraviolet and visible light conditions.

To overcome the different inherent shortcomings of singleelement doping, researchers proposed the strategy of doping two or more elements, namely co-doping. This method takes advantage of each dopant for collaborative catalysis, and the catalytic efficiency is usually higher than that of single element doping; thus, it has been widely investigated in many fields.

Babu et al.<sup>192</sup> synthesized g-C<sub>3</sub>N<sub>4</sub> co-doped with B and S by placing boric acid, thiourea and melamine in a muffle furnace through one pot in situ calcination (Fig. 9c). The B and S elements were introduced in the lattice to enhance the light absorption, accelerate the charge separation and migration, and increase the specific surface area of g-C<sub>3</sub>N<sub>4</sub>. The defects generated by the dopants captured photogenerated electrons, thus inhibiting the recombination of charge carriers. Wang et al.<sup>193</sup> mixed dicyandiamide with KBH<sub>4</sub> and calcined it for 2 h at 550 °C. The H<sub>2</sub> generated by the thermal decomposition of KBH<sub>4</sub> created N defects. The band gap gradually narrowed with an increase in the doping concentration. The band gap of pure g-C<sub>3</sub>N<sub>4</sub> is 2.74 eV, which decreased to 2.61 eV when 3% K and B elements were added. Moreover, the N defects caused by H<sub>2</sub> generated by the thermal decomposition of KBH<sub>4</sub> reduced the band gap to 2.35 eV (Fig. 9d). In this study, when N atoms and K dopants were close to each other, the electron density became enriched, and the presence of K changed the electron spatial distribution, which made K act as an electron donor and promoted the electron transfer between adjacent layers. The adjacent atoms experienced an electron-abundant state due to the incorporation of both nitrogen vacancies and boron. However, the electron enrichment resulting from nitrogen vacancies exceeded that from boron doping, leading to a higher concentration of basic sites. Consequently, the nitrogen defects exhibited a more robust CO<sub>2</sub> adsorption capacity than B and K doping. B maintained a high reduction potential and compensated for the adverse effect of K. The K, B and N vacancies



**Fig. 9** (a and b) Photocatalytic  $CO_2$  reduction yield and stability of  $g-C_3N_4$  with different doping amounts of Mo. (c) Synthesis of B, S-co-doped  $g-C_3N_4$  by *in situ* calcination method. (d) Comparison of the band gap of  $g-C_3N_4$  doped with K and B. (a and b) Reproduced from ref. 191 with permission from [Elsevier], copyright [2016]. (c) Reproduced from ref. 192 with permission from [ACS Publications], copyright [2018]. (d) Reproduced from ref. 193 with permission from [Elsevier], copyright [2018].

cooperated to realize the photocatalytic reduction of  $CO_2$  to  $CH_4$  and CO without a sacrificial agent.

### 3.4 Heterojunction formation

In the conventional heterojunction framework, there are three classifications including bridging gap type (type-I), interlaced gap type (type-II), and fracture gap type (type-III).<sup>195</sup> As illustrated in Fig. 10, a type-I semiconductor setup exhibits a scenario where the conduction band (CB) of semiconductor A resides higher than that of semiconductor B; conversely, the valence band (VB) of semiconductor A is situated beneath that of semiconductor B. Electrons and holes shift collaboratively towards semiconductor B. This movement leads to a decline in the effectiveness of charge separation, subsequently reducing the redox potential of the reaction at lower levels. In the context of type-II semiconductors, A possesses a higher CB and VB. Here, electrons transit from the CB to B, while holes move from the VB to A, enabling the efficient separation of electron-hole pairs. However, the redox ability is still weak. In the case of type-III, the band gaps do not overlap and electron-hole pairs cannot transfer. Therefore, researchers need to develop other types of heterojunctions to overcome the limitation of traditional heterojunctions.196,197

Exploring the charge migration dynamics in Z-scheme heterojunctions, researchers have categorized this phenomenon into three distinct classifications (Fig. 11) including the conventional Z-scheme heterojunction, all-solid-state Z-scheme heterojunction, and direct Z-scheme heterojunction.<sup>198</sup> The traditional Z-scheme heterojunction was first proposed by Bard in simulating the direction of charge transfer in natural photosynthesis.<sup>199</sup> The traditional Z-scheme photocatalytic system consists of two different semiconductors (S1 and S2) and a receptor/donor (A/D) pair. During the photocatalytic reaction, the electrons generated by S2 will react with A to give electrons to D, and the holes in S1 will react with D to give electrons to A. Thus, this structure can effectively realize the separation of electron–hole pairs and has strong redox ability. However, this type of heterojunction is limited to the solution phase, resulting



**Fig. 10** Three types of traditional heterojunction: (a) bridging gap type (type-I), (b) interlaced gap type (type-II) and (c) fracture gap type (type-III). Reproduced from ref. 195 with permission from [Wiley Online Library], copyright [2017].

in difficult charge transfer and narrow application range. In 2006, Tada et al.<sup>200</sup> introduced the all-solid-state Z-scheme heterojunction design to facilitate targeted charge migration and broaden its applicability. The photocatalytic system was comprised of a duo of dissimilar semiconductors with noble metal nanoparticles functioning as the electronic mediator. This system has seen substantial endorsement. However, the reliance on scarce and costly noble metal nanoparticles as the electron medium in this solid-state Z-scheme system hinders its wide application. Consequently, researchers have increasingly directed their focus towards investigating direct Zheterojunctions. In 2013, Yu et al.201 created a direct Z-type heterojunction comprised of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>. In 2016,<sup>202</sup> they conducted an investigation into the electron transportation dynamics within the g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> heterostructure. Our research group used a self-assembly method to vertically fix one edge of a single layer of MoS<sub>2</sub> nanosheets on the surface of SiC nanoparticles to form a marigold-like heterojunction, SiC@MoS2.203 This Z-scheme structure could expose the surface of the photocatalyst to the maximum extent to fully contact with the reactants. The high selectivity for the CH<sub>4</sub> product was attributed to the rapid reduction of CO<sub>2</sub> on the surface of SiC by high electron mobility, and the production of O2 was produced by the rapid oxidation of H<sub>2</sub>O on the surface of MoS<sub>2</sub> by high hole mobility. Upon optimization, the catalyst with a 60% MoS<sub>2</sub> composition was determined to be the most favourable blend. It exhibited the CH<sub>4</sub> generation rate of 323  $\mu$ L g<sup>-1</sup> h<sup>-1</sup> and an O<sub>2</sub> production rate of 620  $\mu$ L g<sup>-1</sup> h<sup>-1</sup>. These findings indicated that the molar proportion of CH<sub>4</sub> to O<sub>2</sub> approached a 1:2 ratio, aligning with the chemical equilibrium of the  $CO_2(g) + 2H_2O(g)$  $= CH_4(g) + 2O_2(g)$  reaction.

To a certain extent, the low efficiency of photocatalytic reactions can be attributed in part to the significant disparity in the electron and hole migration rates within semiconductor catalysts, resulting in the accumulation of internal photogenerated holes, which subsequently hinder the production of photogenerated electrons, thereby shortening the lifetime of photogenerated carriers. Based on this, we constructed a direct and indirect Z-type heterostructure coupled photocatalytic system.<sup>204</sup> As shown in Scheme 6, firstly, Pt nanoparticles were loaded on SiC by photodeposition, and then dispersed in an aqueous solution containing Cu<sup>2+</sup> and IrCl<sub>6</sub><sup>3-</sup>. Under ultraviolet irradiation, Cu<sup>2+</sup> was reduced to Cu<sub>2</sub>O, and IrCl<sub>6</sub><sup>3-</sup> was oxidized to  $IrO_x$ . Finally, the Cu<sub>2</sub>O-Pt/SiC/IrO<sub>x</sub> composite catalyst was obtained. This catalyst exhibited improved photocatalytic efficiency by constructing a multi-photocatalyst integrated system to enhance the lifetime of the photogenerated carriers and redox ability. Regarding the photo-catalytic conversion of CO<sub>2</sub> and H<sub>2</sub>O into HCOOH and O<sub>2</sub>, the respective production rates were 896.7  $\mu mol~g^{-1}~h^{-1}$  for HCOOH and 440.7  $\mu mol~g^{-1}~h^{-1}$  for  $O_2$ .

In 2019 and 2020, Yu proposed the S-scheme heterojunction to improve the electron transfer problem of the type-II and Zscheme heterojunctions.<sup>205,206</sup> The S-scheme heterojunction is a composition involving a reduction photocatalyst (RP) and an oxidation photocatalyst (OP). As depicted in Fig. 12a, the conduction band (CB), valence band (VB), and Fermi energy



Fig. 11 Three types of Z-scheme heterojunctions: (a) traditional Z-scheme, (b) all-solid-state Z-scheme, and (c) direct Z-scheme.



Scheme 6 Process for the formation of  $Cu_2O-Pt/SiC/IrO_x$  by photodeposition. Reproduced from ref. 204 with permission from [Nature Publishing Group], copyright [2020].

levels of the RP are positioned above that of the OP to attain equilibrium in the Fermi energy levels at the interface, where the electrons from the RP will naturally transition to the OP. During the electron transfer process, the RP side retains a positive charge and the OP side retains a negative charge, resulting in the formation of an internal electric field at the interface from RP to OP (Fig. 12b). Electrons in both RP and OP are transferred from VB to CB under photoexcitation (Fig. 12c), and the system has the maximum reduction and oxidation capacity.55 Accordingly, many researchers began to study this emerging topic to reduce CO<sub>2</sub> to hydrocarbon fuel. However, the selection of OP and RP is crucial, which requires that their band structure matches and they have characteristics conducive to CO<sub>2</sub> adsorption and activation. Yang et al.<sup>207</sup> synthesized the Sscheme heterojunction g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>/C with a thickness of about 5 nm using electrostatic self-assembly and calcination methods.

As shown in Scheme 7, firstly, a certain amount of urea was calcined at 550 °C to condense into g-C<sub>3</sub>N<sub>4</sub> with layers of stacked flower structure. The pH of the suspension was adjusted to 4 with 0.5 M HCl and deionized water to bring H<sup>+</sup> to the surface. This protonation process was conducive to the stripping of  $g-C_3N_4$  into layers. Then, the prepared  $Ti_3C_2T_r$ monolayer with an area greater than 5  $\mu$ m  $\times$  5  $\mu$ m was added to the suspension as the base plane to obtain g-C<sub>3</sub>N<sub>4</sub> with a stretched fold layer. The monolayer was calcined at 400 °C, where the Ti layers were converted into TiO<sub>2</sub> nanoparticles and fixed on the graphite layer to form a rough surface. The g-C<sub>3</sub>N<sub>4</sub> with a stretched fold layer was tightly attached to the TiO<sub>2</sub> layer to prevent g-C<sub>3</sub>N<sub>4</sub> from agglomerating. Under Xe lamp irradiation for 3 h, the catalyst reduced CO<sub>2</sub> to CH<sub>4</sub> and CO with yields of 3.7  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> and 25.96  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, respectively. The improved performance of the catalyst was attributed to the formation of an S-scheme heterojunction between the two semiconductors, and its layer-to-layer structure could expose the maximum optical absorption area, which provided an effective channel for photogenerated carrier migration. These design schemes provided a new idea for the preparation of multi-junction photocatalysts, which are expected to be extended to other semiconductor catalysts.

Although type-II, Z-scheme and S-scheme heterojunctions all have staggered energy band structures, type-II and Z-scheme



Fig. 12 Carrier transfer mechanism of S-type heterojunctions: (a) band structure of RP and OP, (b) formation of IEF after RP and OP coupling, (c) photogenic carrier transfer and separation under light irradiation.



Scheme 7 Process for the preparation of  $g-C_3N_4/TiO_2/C$ . Reproduced from ref. 207 with permission from [Wiley Online Library], copyright [2021].

heterojunctions present challenges in terms of thermodynamics, photogenerated carrier transfer kinetics and energy utilization. Alternatively, the S-scheme heterojunction, as a new strategy to increase the separation and migration rate of photogenerated carriers under the influence of an internal electric field, has largely eliminated many of the inherent defects of the traditional Z-type heterojunction and other heterojunctions, and therefore has become a research hotspot in the field of photocatalytic energy and environmental applications. Deng et al.<sup>208</sup> synthesized an In<sub>2</sub>O<sub>3</sub>/Nb<sub>2</sub>O<sub>5</sub> S-scheme heterojunction by mixing Nb<sub>2</sub>O<sub>5</sub> and In<sub>2</sub>O<sub>3</sub> in the same electrospinning solution and roasting at high temperature. The heterojunction synthesized by this method exhibited close interface contact and promoted ultra-fast (<10 ps) electron transfer at the interface. The optimized In<sub>2</sub>O<sub>3</sub>/Nb<sub>2</sub>O<sub>5</sub> nanofibers extended the lifetime of the catalyst and improved the CO<sub>2</sub> reduction performance due to the rapid interfacial charge transfer and effective CO<sub>2</sub> activation on the catalyst. The CO yield was as high as 0.21 mmol  $g^{-1}$  h<sup>-1</sup>. Combined with FSTAS analysis, it was shown that the photoelectron transfer on the CB of In<sub>2</sub>O<sub>3</sub> to VB of Nb<sub>2</sub>O<sub>5</sub> inhibits the self-carrier recombination, effectively separates the electrons in the CB of Nb<sub>2</sub>O<sub>5</sub> and the holes in the VB of In<sub>2</sub>O<sub>3</sub>, prolongs the life of the nano-hybrid materials, and proves the potential of the S-type heterojunction interface in ultrafast charge transfer. In addition, Sun et al. 209 synthesized a novel Stype heterojunction Ga<sub>2</sub>S<sub>3</sub>/CuS catalyst using a simple two-step hydrothermal method. A series of characterization analyses of PL and time-resolved PL showed that the S-type heterostructure is advantageous for the improvement of the photogenerated electron-hole separation efficiency. Density functional theory also proved that the Ga<sub>2</sub>S<sub>3</sub>/CuS interfacial electric field can promote the adsorption of CO<sub>2</sub> and intermediate products. In situ KPFM indicated the transfer of photogenerated electrons from Ga<sub>2</sub>S<sub>3</sub> to CuS under light conditions. This study elucidated the action mechanism of the intrinsic electric field at the S-type heterojunction interface and provides a new way to improve the selectivity of CO<sub>2</sub> reduction to CH<sub>4</sub> in photocatalysis.

### 3.5 Single-atom engineering

In 2011, Zhang et al.<sup>210</sup> constructed a Pt-O-Fe structure and successfully fixed a Pt atom to  $FeO_x$  and proposed the term single-atom catalysts (SACs), which advanced the process of elucidating the catalytic active site at the atomic level. In 2014, in the work by Yang et al.,<sup>211</sup> a groundbreaking one-step approach was employed to fix individual metal atoms (including Pt, Pd, Rh, and Ru) on the surface of TiO<sub>2</sub>, revealing that these single-atom catalysts exhibited a superior photocatalytic H<sub>2</sub> generation performance compared to TiO<sub>2</sub>-supported metal clusters. After that, the field of photocatalysis opened the door to SACs. It was also realized that as the size of metal nanoparticles was gradually reduced to a single atom, the individual atoms embedded in the carrier could regulate the energy band and electronic structure and change the surface structure of the catalyst through the overlap of electron orbitals and interaction with the carrier ligand.<sup>212</sup> The support of SACs usually acts as a light collection unit. As the first step of photocatalysis, light collection is also a necessary step to achieve visible light absorption, which can provide sufficient electronhole pairs for subsequent catalytic reactions. The catalytic activity and selectivity are fundamentally shaped by the metallic single atoms functioning as active centres. Excited electron transfer from these metal atoms to the support leads to a reshuffle of charge distribution between the individual metal atoms and the support. Consequently, this alteration in local electronic properties influences the photocatalytic performance.<sup>210,213,214</sup> If these three parts are properly optimized, the final atomic utilization rate will reach nearly 100%. Based on this, researchers began to design various single-atom construction methods, and this modification strategy has become one of the most promising research directions in the field of photocatalysis.215-227

Wet chemistry is a common and simple traditional method for preparing supported monatomic catalysts because it can be done in ordinary laboratory operations. It includes the codeposition method,<sup>210,213</sup> sol-gel method,<sup>228,229</sup> impregnation method,230,231 depositional precipitation method232,233 and strong electrostatic adsorption method.234 Among them, the metal atoms prepared by the co-deposition method and sol-gel method may be buried in the support, leading to a decrease in atomic utilization. The metal atoms prepared by the impregnation method can be directly anchored on the surface of the support to avoid the above-mentioned problems, but there is a problem of uneven dispersion. The final crystal size of the sedimentation-precipitation method depends on the nucleation and growth rate during the precipitation process, which is difficult to control. When the metal load is high, the particle size and dispersion are uneven, and thus the dispersion can only be maintained evenly under a low load. In the process for the creation of strong electrostatic adsorption, the conditions are not controllable and the surface active sites are not uniform. It can be seen that wet chemical synthesis methods are simple and direct, but they lack precise control of the size and dispersion of sites, structure and composition of the active units of the catalyst.235 Single metal atoms have a higher surface energy than nanoclusters and nanoparticles, resulting in their easier migration and aggregation during preparation.<sup>236</sup> It has been reported that when the support is doped with heteroatoms,24,94-97 vacancies,237-240 surface groups102,103 and single-atom step edges,<sup>25,79,104</sup> the defects formed by the introduction of surface groups can be used as anchoring sites of metal single atoms to prevent the agglomeration of the atoms. When designing catalysts, the stability of single metal atoms is a crucial factor. One approach involves attaching single metal atoms directly to light-absorbing substances, necessitating an abundance of anchoring sites and reaction sites across the support surface. Some traditional semiconductors, such as  $TiO_{2}$ ,<sup>75,100,105,106</sup> CeO<sub>2</sub>,<sup>215,229,232,233,241,242</sup> and FeO<sub>x</sub>,<sup>210,213,243</sup> can be replaced by a single atom because of their surface metal cations and can interact with oxygen anions to achieve the anchoring of single metal atoms.

Although many MOs are regarded as effective carriers of SACs, the single-atom load is still unsatisfactory due to the limited number of constructed sites, and their structure and properties are difficult to adjust at the molecular level, and thus researchers are urgently searching for ideal materials that can increase the load by orders of magnitude.<sup>96,244-250</sup> In this case, organic polymers can satisfy these requirements.

 $g-C_3N_4$  is also a popular photocatalyst for anchoring single metal atoms. The unique cavity in its structure has six N atoms, which are suitable as anchoring sites and their lone pair electrons can combine with single metal atoms with empty or partially empty orbitals, breaking the limitations of anchoring traditional materials and insufficient reaction sites, and significantly improving the load content of single metal atoms.<sup>251–253</sup> Theoretically, the ultimate load content depends on the maximum number of matches between the metal atoms



**Fig. 13** (a) Schematic of the introduction of defects in  $g-C_3N_4$  by Pt cationic groups and oxygen-containing functional groups. (b) Diagram of Ni confinement in  $g-C_3N_4$  vacancy ligands with fewer porous layers. (c) Anchoring site synthesis diagram of  $Cu-N_4$  in P-doped  $g-C_3N_4$ . (a) Reproduced from ref. 254 with permission from [Wiley Online Library], copyright [2020]. (b) Reproduced from ref. 255 with permission from [Wiley Online Library], copyright [2022]. (c) Reproduced from ref. 256 with permission from [Wiley Online Library], copyright [2023].

separated by the target and the anchoring sites. Cheng et al.<sup>254</sup> increased the density of monatomic active sites by using a selflimiting method to confine Ni to the g-C<sub>3</sub>N<sub>4</sub> vacancy ligand with fewer porous layers by using an unsaturated edge (Fig. 13a). Shi et al.<sup>255</sup> introduced Pt cationic groups and oxygen-containing functional groups into the defect edge of defective g-C<sub>3</sub>N<sub>4</sub> to form intermediates, and low-temperature calcination could make the Pt atoms closely connected with it without affecting the structure of the support (Fig. 13b). Xie et al.256 designed Cu-N<sub>4</sub> anchoring sites in P-doped g-C<sub>3</sub>N<sub>4</sub> to promote the formation of C-C coupled intermediates and adjusted the intermediate energy level of the reaction path of the target product to increase the selectivity for the high value-added product C<sub>2</sub>H<sub>4</sub>, reaching 53.2% (Fig. 13c). To increase the single atom load, Ma et al.257 prepared Co/g-C<sub>3</sub>N<sub>4</sub> via a two-step calcination process, where the load of Co restricts the potential for efficient photocatalytic  $CO_2$  reduction, thereby hindering the conversion of  $CO_2$  into valuable products. This research introduces an innovative approach for the development of SACs with a high loading and superior dispersion.

At present, the commonly used method for the synthesis of SACs with an M-N-C structure is high-temperature pyrolysis, but the coordination environment of  $M-N_x$  in the catalyst is uncertain.258 Compared with the SACs obtained by pyrolysis, the structure of the molecular catalyst is controllable, which is helpful for the study of the mechanism of single active sites.<sup>259,260</sup> Determining the consistency of the surface active sites is a future research focus in the field of SACs, which is conducive to improving the selectivity for CO<sub>2</sub> reduction products. The loading of Co atoms on the surface was up to 24.6 wt%, which achieved the reduction of CO2 to CH3OH (941.9  $\mu$ mol g<sup>-1</sup>) under 300 W Xe lamp irradiation for 4 h without adding sacrificial and photosensitizing agents. The key points of this experiment are shown in Scheme 8. (1) The massive g-C<sub>3</sub>N<sub>4</sub> formed porous sheets with a high specific surface area after being treated with concentrated H<sub>2</sub>SO<sub>4</sub>, which could provide enough coordination sites for anchoring Co atoms to inhibit atomic aggregation. (2) The synthesis adopted two calcination steps. In the first stage, it was maintained at 130 °C for 9 h to ensure that the H<sub>2</sub>O adsorbed in the mixture of g-C<sub>3</sub>N<sub>4</sub> and  $Co(NO_3)_2 \cdot 6H_2O$  was removed cleanly. In the second stage, the temperature was increased to 550 °C to vaporize the Co atoms not fixed on the surface. When the g-C<sub>3</sub>N<sub>4</sub> bulk and cobalt nitrate were subjected to direct calcination in one step, the tendency for aggregation on the surface led to the formation

of  $CoO_x$ . This is the reason for the lack of active sites in low-loading single-atom catalysts (SACs).

### 3.6 Surface organometallic catalysis

Surface organometallic catalysis (SOMC)<sup>261-263</sup> is the rational bonding of molecular precursors (organometallic or coordinate compounds) to the surface of a carrier (such as oxides, metal nanoparticles, and carbon materials) according to the rules of coordination chemistry. The carrier surface controls the reactivity of the molecular precursors, and the grafted molecular complexes can directly act as active centres or be modified after heat treatment under vacuum, inert gas or reactive gas before participating in catalytic reactions.<sup>264</sup> The life of some organometallic compounds sensitive to oxygen, water and temperature can be extended after grafting to a support, which is new molecular technology for the synthesis of single active site, homogeneous functional heterogeneous catalysts. Different from SACs, organometallic compounds retain their partial distribution after grafting, which is a special case of SACs. Abundant –OH groups are present on the oxide surface at 700  $^\circ$ C,<sup>265</sup> and thus organometallic compounds are usually connected to the isolated hydroxyl groups on the surface of the oxide to form M-O bonds. Given that oxides have been the most widely used graft support, we focus on describing the design strategies of SOMC supported by solid oxides. Molecular-level synthesis provides great help in improving the structural uniformity of catalysts.

The synthesis of heterogeneous catalysts using molecular strategies requires understanding the controlled structure and catalytic behaviour of surface species to establish structureactivity relationships. Specific organometallic species have active sites similar to that of supported MOs, and they usually possess a larger fraction of active sites, making structureactivity relationships possible. There are many -OH present on the surface of MOs and they are directly related to the density and coordination surroundings of the metal sites; thus, they are often considered the grafting sites.<sup>265</sup> Accordingly, controlling the number of -OH groups on the surface of the oxide support becomes the key to the successful synthesis of SOMC. The type and density of the surface groups (MOs-OH, MOs-O-Ms or  $MOs-OH_2$ ) of the oxide carriers can be changed by certain treatment. As shown in Scheme 9, heat treatment under vacuum can remove the adsorbed H2O and condense the nearby MOs-OH sites, leaving more MOs-O-MOs between the metal complex and the oxide, and this process reduces the density of



Scheme 8 Schematic representation of  $Co/g-C_3N_4$  by a two-step calcination method (blue: N atoms; grey: C atoms; and red: Co atoms). Reproduced from ref. 260 with permission from [Wiley Online Library], copyright [2022].



Scheme 9 Process of dehydroxylation and grafting of MOs. Reproduced from ref. 265 with permission from [Elsevier], copyright [1998].

functional MOs–OH on the oxide surface. When the active B<sup>-</sup> ligand is attacked by unstable protons, the site-isolated MOs–OH reacts with the molecular precursor  $A_xMB_y$  to form well-defined surface species as single-site MOs–O–MA<sub>x</sub>B<sub>y–1</sub>, which can also interact with neighbouring MOs–O–MOs.<sup>264</sup>

Evaluating the structure of the grafted complex at the molecular and atomic levels is a key step in understanding its stability and reactivity, and thus it is necessary to determine the structure of the carrier prior to the preparation of SOMCs. In general, carriers with a high surface area and a certain number of hydroxyl groups on their surface (such as TiO<sub>2</sub>, SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, and ZrO<sub>2</sub>) are we preferred, but we need to consider that some oxides tend to undergo phase transitions or sintering when heated in the presence of H<sub>2</sub>O. One way to ameliorate this dilemma is to moisten MO powder with H<sub>2</sub>O, and then slowly evaporate it in a drying oven. The fine powder is condensed into larger, more easily filtered particles, and then the carrier is calcined at a temperature lower than dehydroxylation to remove organic impurities.

In the case of the metal precursors, researchers usually choose hydrocarbons, amides, pyridines, porphyrins, and siloxane, sometimes grafted in organic solvents to avoid surface chemical reactions that dissolve or reprecipitate the oxide support in  $H_2O$ , making the preparation process more complex.<sup>266–271</sup>

The graft of organometallic compounds can be anchored by organometallic fixation272,273 or ALD,274,275 which takes advantage of the homogeneity of the molecular precursors. Both methods enable a level of control and flexibility that contributes to the clarification of structure and mechanism, compared with traditional catalyst synthesis techniques, advanced molecularlevel preparation provides a way for rational catalyst design. The specific grafting process is as follows: firstly, the carrier is placed in a vacuum infrared tube for dehydroxylation.<sup>276</sup> The reactor in the infrared tube can slide up and down and its head contains infrared transparent windows (CaF2, KBr, etc.) to obtain a spectrum and facilitate the monitoring of the consumption of hydroxyl groups on the surface through infrared spectroscopy during the experiment. Subsequently, the volatile metal precursor can be grafted in the gas phase by sublimating the compound to a carrier using a vacuum tube attached to the reactor. It is important to note that the grafting of the molecular precursors to the oxide carrier requires strict anhydrous and anaerobic conditions because most organics decompose easily in H<sub>2</sub>O and O<sub>2</sub>. SOMCs form isolated sites more selectively than impregnation, which facilitates the binding of metal sites into the carrier.

Compared with single-metal catalysts, bimetal catalysts have double active sites, and the synergistic effect of the two metals leads to an improvement in their stability, activity and reaction selectivity.<sup>277-279</sup> However, it is more difficult to control the uniform reaction sites on the carrier surface, which is affected by many factors, such as temperature, metal type, dispersion, particle size, and precursor metal feed ratio. Thus far, this has been a serious challenge. In the work by Lian,<sup>280</sup> the concept of a "dual surface organometallic complex" was introduced through the hydrogenation of supported bimetallic cluster catalysts, drawing from the principles of surface organometallic chemistry. This innovative idea served as a foundational approach for the development of multifaceted metal catalysts, featuring numerous active sites, which could be anchored to a variety of supports.

# 4 Reaction efficiency and product selectivity of photocatalysis $CO_2$ with $H_2O$

At present, the utilization of diverse semiconductor materials in photocatalytic CO<sub>2</sub> conversion encounters challenges of low efficacy and unsatisfactory product selectivity. Various approaches to boost the efficiency of CO<sub>2</sub> reduction have been compiled. Focusing on improving the selectivity of the process towards economically viable and sustainable fuel production, we review the progress in manipulating the product selectivity in photocatalytic CO<sub>2</sub> reduction reactions. Additionally, we classify the prospective photocatalytic products together with recent photocatalysts, aiming to offer ideas for the development of efficient photocatalysts. In 2012, Kuhl et al.<sup>281</sup> found that CO<sub>2</sub> could be reduced to 16 different products. If more than two electrons and protons are transferred, CH<sub>4</sub>, CH<sub>3</sub>OH, C<sub>2</sub>H<sub>5</sub>OH,  $C_2H_4$ , and other products may be formed. The product selectivity is primarily influenced by both the dynamics and thermodynamics (Fig. 14). The dynamic factors include the separation efficiency of photogenerated electrons-holes, distinct active sites, and gaseous and liquid products. For example, when two electrons and protons are transferred, CO



Fig. 14 Diverse elements influencing the selectivity of photocatalytic  $CO_2$  conversion and prospective enhancement approaches.

#### Review

and HCOOH may be formed. Also, the adsorption/desorption characteristics of reactants/intermediates, *etc.* Regarding thermodynamics, by optimizing the band structure to elevate the CB position, certain semiconductor materials can facilitate the production of compounds that are thermodynamically unfavorable, thereby influencing the product selectivity. Hence, we can explore these two aspects and employ suitable strategies for catalyst modification to enhance the product selectivity.

When the adsorbed H<sub>2</sub>O molecules undergo an oxidation reaction by the photogenerated hole, oxygen and H<sup>+</sup> ions are produced on the semiconductor surface. Electrons with reducing power also reduce H<sup>+</sup> to hydrogen. Moreover, when the pH of the reaction solution is low, the rate of HER increases to produce hydrogen gas. Thus, hydrogen evolution is the competitive reaction to affect the efficiency of photocatalytic activity and the product selectivity of CO2 reduction. In this case, suppressing H<sub>2</sub> generation and improving the CO<sub>2</sub> reduction efficiency are important considerations in CO<sub>2</sub> reduction with H<sub>2</sub>O. Surface engineering of catalysts and controlling the reaction system are often considered for suppressing H<sub>2</sub> generation. Selecting catalysts with a higher affinity for CO<sub>2</sub> than for protons is favourable to suppress HER. For example, metal catalysts such as copper, silver, and gold have shown good selectivity for CO2 reduction.282-284 In addition, the construction of alkaline sites on the surface of the photocatalyst can enhance the CO<sub>2</sub> adsorption capacity. Meanwhile, increasing the hydrophobicity of the catalyst surface can inhibit proton reduction.<sup>91-93</sup> The reaction system operating in nearneutral or weakly alkaline electrolyte can reduce the availability of protons, thereby inhibiting HER. This can be achieved by using buffers or adjusting the electrolyte composition. Moreover, some electrolyte ions also affect the adsorption of CO<sub>2</sub> and protons on the catalyst surface. For example, bicarbonate ions can promote CO2 reduction. Therefore, we can also inhibit HER by adjusting the ionic strength and composition of the reaction system. Implementing a flow cell reactor improves the mass transport and increases CO<sub>2</sub> concentration at the electrode surface, thereby reducing the competition from HER.285,286 By integrating these strategies, the selectivity and efficiency of the p-CO<sub>2</sub>R process can be improved, while minimizing the competing HER.

In the photocatalytic CO<sub>2</sub> reduction system, using electrolytes of primarily salts also has an important influence in the activity and selectivity of the reduced products. Research has indicated that the adsorption of sulfites, K<sub>2</sub>SO<sub>4</sub>, NaCl, NaHCO<sub>3</sub>, etc., and alkalis such as NaOH on the surface of the photocatalyst induces alterations in its surface properties, which can hinder CO<sub>2</sub> reduction and inter-particle contact between the photocatalytic particles and reduce the electron transport efficiency between the photocatalyst particles.287 Furthermore, adding alkaline salt to the reaction solution can facilitate the production of CO and enhance its selectivity. This heightened CO<sub>2</sub> reduction activity can be attributed to favorable pH conditions and effective provision of hydrated CO2 molecules by adding alkaline salt to supply reactants. Simultaneously, an appropriate concentration of alkaline additives will also aid in stabilizing certain photo-corrosive metal sulfide materials

during  $CO_2$  reduction.<sup>285</sup> Additionally, Das *et al.*<sup>288</sup> examined how varying the NaOH concentration affects the catalytic activity by adjusting the solution pH. Their research revealed that methane production increases with an increase in the pH level due to the increased solubility of carbon dioxide in the reaction medium at higher pH values, leading to a greater concentration of  $CO_2$  near the catalyst surface, which is conducive for carbon dioxide reduction. However, when reaching a certain threshold level of pH value, insufficient proton supply on the adsorbed  $CO_2$  surfaces becomes detrimental for photocatalytic  $CO_2$  reduction. Therefore, we believe that adjusting the type of reaction solution can enhance and stabilize the photocatalytic performance for  $CO_2$ .

### 4.1 C1 products

Although various  $C_1$ ,  $C_2$  and  $C_{2+}$  products are produced from  $CO_2$  reduction, the majority of current research on photocatalytic  $CO_2$  reduction is focused on the generation of  $C_1$ products, such as CO, CH<sub>4</sub>, CH<sub>3</sub>OH, and HCOOH. These  $C_1$ products can be used directly as an energy source or serve as a cost-effective carbon source. The conversion of  $CO_2$  into lowcost  $C_1$  products using solar energy through environmentally friendly methods holds significant importance and plays a pivotal role in meeting the present energy and chemical requirements.<sup>78,121,122</sup>

**4.1.1 CO** and  $CH_4$ . CO and  $CH_4$  are the most common products in the photocatalytic reduction  $CO_2$  with  $H_2O$ . There have been many reports on the development of novel photocatalysts for CO and  $CH_4$  production. The adsorption and desorption characteristics of CO on the surface of the photocatalyst greatly influence the product selectivity during photocatalyzed  $CO_2$  reduction.<sup>13,55,120,289</sup> Due to its suitable reduction potential and two-electron reduction process, CO is readily obtained as the primary product. Tables 1 and 2 list the recent advances in the photocatalytic reduction of  $CO_2$  with  $H_2O$  to produce CO and  $CH_4$ .

According to Table 1, it can be seen that great progress has been made in the work on the reduction of CO<sub>2</sub> emissions via CO<sub>2</sub> conversion to CO. Without the need for a sacrificial agent, the highest activity recorded was 608  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>,<sup>290</sup> and the selectivity reached almost 100%.291 To achieve the optimal activity and selectivity, compared with traditional oxide photocatalysts, researchers are more inclined to modify Bi-based photocatalysts due to their unique layered structure. The layered structure of Bi-based photocatalysts facilitates electronhole pair separation when the  $Br_{4p}$  electrons transition to the Bi<sub>6p</sub> orbitals, resulting in an internal electric field between the [Bi<sub>2</sub>O<sub>2</sub>]<sup>2+</sup> layers and X<sup>-</sup> layers. This prolongs the photogenerated charge carrier lifetimes, thereby improving the photocatalytic performance.310-312 However, compared with CO, obtaining high-added value products such as CH<sub>4</sub> is imperative but very difficult. Currently, the photocatalytic CO<sub>2</sub> conversion to CH<sub>4</sub> reported under the condition of no sacrificial agent has the highest activity of 1640 µmol g<sup>-1</sup> h<sup>-1</sup>,<sup>298</sup> and selectivity of 100%,159 but it cannot have the two advantages of high activity and selectivity simultaneously. This because it requires the

Table 1 Summary of the photocatalytic systems for CO<sub>2</sub> reduction with H<sub>2</sub>O to product CO

Catalyst	Light source	Production/µmol $g^{-1}$ $h^{-1}$	Selectivity	Reference
Ag <sub>1</sub> @PCN	300 W Xe lamp	160	94%	70
AgBr/BiOBr(OVs)	300 W Xe lamp	212.6	90.3%	133
Cu SAs/TiO <sub>2</sub>	300 W Xe lamp	65.8	100%	249
Ag-Cu <sub>2</sub> O/TiO <sub>2</sub>	300 W Xe lamp	13.19	73.7%	282
BiOBr/Bi <sub>2</sub> S <sub>3</sub> (OVs)	300 W Xe lamp	100.8	74.8%	283
Au nanoparticle-loaded TiO <sub>2</sub>	300 W Xe lamp	608	99%	290
Mn <sub>1</sub> Co <sub>1</sub> -CN	300 W Xe lamp	47	${\sim}100\%$	291
Ag-loaded Ga <sub>2</sub> O <sub>3</sub> /ZnGa <sub>2</sub> O <sub>4</sub>	400 W Hg lamp	_	${\sim}100\%$	289
NiAl-LDH/Ti <sub>3</sub> C <sub>2</sub>	300 W Xe lamp	11.82	92%	292
Defective CeO <sub>2</sub>	300 W Xe lamp	7	${\sim}100\%$	293
Ultrathin Bi <sub>4</sub> O <sub>5</sub> Br <sub>2</sub>	300 W Xe lamp	31.57	99.5%	294
g-C <sub>3</sub> N <sub>4</sub> /FeWO <sub>4</sub>	Visible light ( $\lambda > 420 \text{ nm}$ )	6	99%	295
$TiO_2/BiVO_4$ -4	300 W Xe lamp	17.33	100%	296
Bi-Bi <sub>2</sub> Sn <sub>2</sub> O <sub>7</sub>	300 W Xe lamp	114.1	100%	297

Table 2 Summary of the photocatalytic systems for  $CO_2$  reduction with  $H_2O$  to produce  $CH_4$ 

Catalyst	Light source	Production/µmol $g^{-1}$ $h^{-1}$	Selectivity	Reference
V <sub>O,N</sub> -NBCN	300 W Xe lamp	5.90	_	79
Fe/SiC	300 W Xe lamp	30.0	94.3%	101
Pt@Ag-TiO <sub>2</sub>	350 W Xe lamp	160.3	87.9%	131
Pt/TiO <sub>2</sub>	300 W Xe lamp	150.04	${\sim}100\%$	159
$WO_{3-x}$	300 W Xe lamp	1640	95%	298
Pt@h-BN	300 W Xe lamp	184.7	99.1%	299
Pt/SiC nanosheets	300 W Xe lamp	13.6	96.79%	300
Bi <sub>2</sub> WO <sub>6</sub> -Cl nanosheets	300 W Xe lamp	1.66	94.98%	301
CNBr	300 W Xe lamp	16.68	70.27%	302
Pt@Def-CN	300 W Xe lamp	6.3	97%	303
Au@Pd-TiO <sub>2</sub>	300 W Xe lamp	26.32	99.7%	304
HAP/TiO <sub>2</sub>	300 W Xe lamp	4.64	99.1%	305
N-doped C dot/CoAl-LDH/g-C <sub>3</sub> N <sub>4</sub>	300 W Xe lamp	5.69	99%	306
Bi <sub>2</sub> MoO <sub>6</sub> /Co <sub>3</sub> O <sub>4</sub>	300 W Xe lamp	28.5	100%	307
V <sub>O</sub> -Nb <sub>2</sub> O <sub>5</sub>	300 W Xe lamp	19.14	94.1%	308
(Pt/TiO <sub>2</sub> )@rGO	300 W Xe lamp	41.3	99.1%	309

active site on the catalyst surface to enrich more electrons during the reaction to complete the eight-electron conversion process. Among the many strategies to improve the performance of photocatalysts, designing heterojunctions is one of the most convenient ways in terms of current technology and research. In this way, the combination of two catalysts with disparate band gaps can effectively enhance the separation efficiency of photoelectrons and holes, thereby improving the overall performance. In the work by Wang et al.,133 they integrated BiOBr with other semiconductor materials to create a heterojunction system. As depicted in Fig. 15a, within the BiOBr/AgBr heterojunction, the electrons gathered in the conduction band (CB) of AgBr were prompted to recombine with holes present in the valence band (VB) of BiOBr due to the influence of Coulomb interaction and an internal electric field (IEF). This unique S-type band alignment promoted the highly effective spatial separation of photogenerated electron-hole pairs. As a direct consequence, this led to a remarkable improvement in both the CO generation rate and selectivity compared to the untreated BiOBr. The yield of CO and CH<sub>4</sub> amounted to 212.6  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> and 5.7  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, respectively, demonstrating an approximate increase of 9.2-fold and 5.2-fold over pristine BiOBr. Remarkably, the CO selectivity attained an outstanding percentage of 90.31%. Wang et al.282 constructed Ag, CuO, and TiO<sub>2</sub> into a ternary Z-scheme heterojunction catalyst via a one-step reduction method. The Zscheme heterojunction formed by P25 and CuO improved the separation efficiency of photogenerated electrons and holes. Alternatively, Ag nanoparticles further improved the electron transport efficiency and promoted the absorption of visible light, preventing the self-reduction of CuO. Compared with P25 and the binary heterojunction formed by Ag or CuO and TiO<sub>2</sub> alone, the constructed ternary Z-scheme heterojunction catalyst greatly increased the yield of CO, and the selectivity reached 73.74%.

However, traditional heterojunctions still exhibit deficiencies such as poor interface contact and low active site density. Through morphology regulation, 2D/2D



**Fig. 15** (a) Electron transfer mechanism of BiOBr and AgBr before and after illumination. (b) Schematic of 3D layered nanocomposite LDH/Ti<sub>3</sub>C<sub>2</sub> with high specific surface area by *in situ* hydrothermal method. (c and d)  $Ga_2O_3/ZnGa_2O_4$  photocatalytic performance test: yield, selectivity and stability. (a) Reproduced from ref. 113 with permission from [Elsevier], copyright [2022]. (b) Reproduced from ref. 292 with permission from [Elsevier], copyright [2021]. (c and d) Reproduced from ref. 289 with permission from [Nature Publishing Group], copyright [2020].

heterojunctions are constructed to achieve good contact between two-dimensional materials, thereby effectively enhancing the migration rate of photogenerated charge carriers and accelerating surface reactions. This approach improves the adsorption and transport of CO<sub>2</sub>, ultimately enhancing both the selectivity and activity, such as LDH/Ti<sub>3</sub>C<sub>2</sub>,<sup>292</sup> and Ga<sub>2</sub>O<sub>3</sub>/ ZnGa<sub>2</sub>O<sub>4</sub>.<sup>289</sup> Shi proposed the construction of a 3D layered nanocomposite, LDH/Ti<sub>3</sub>C<sub>2</sub>, with a high specific surface area via an in situ hydrothermal approach, as depicted in Fig. 15b. The resulting 2D/2D LDH/Ti<sub>3</sub>C<sub>2</sub> established a close contact interface, effectively promoting the separation of photoelectron-hole pairs in NiAl-LDH. Additionally, it offered abundant REDOX active sites with enhanced exposure, leading to improved CO<sub>2</sub> absorption and activation. Photocatalysis reduced CO<sub>2</sub> to CO with a selectivity of up to 92% and excellent stability. Also, the selective enhancement of CO was observed in the Ga2O3/ ZnGa<sub>2</sub>O<sub>4</sub> heterojunction catalyst. H<sub>2</sub> generation was significantly inhibited when the concentration of ZnGa<sub>2</sub>O<sub>4</sub> increased from 0.1 mol% to 10.0 mol%. Finally, the Ga<sub>2</sub>O<sub>3</sub>/ZnGa<sub>2</sub>O<sub>4</sub> heterostructure achieved nearly 100% selectivity towards CO generation, as illustrated in Fig. 15c and d. The investigation revealed that the heterojunction effectively facilitated the segregation of photogenerated electron-hole pairs; however, it lacked adequate electron enrichment capacity. Thus, to compensate for this, dual cocatalysts with contrasting attributes are strategically engineered on the photocatalyst surface to govern the migration of electrons and holes. Metal atoms can be used as CO2 adsorption activation and reduction sites.249,289,313 Fig. 16 depicts the process in which Long et al.<sup>290</sup> implemented a phase doping approach to integrate conventional face-centred cubic (fcc) phase Au nanoparticles onto TiO2. They accomplished photoreduction by altering the chemical composition of

the fcc phase Au NP surface. The resulting gold nanoparticles (T–Au NPs) successfully solved the thermodynamic bottleneck of CO<sub>2</sub> conversion to CO, and the 5-T–Au/TiO<sub>2</sub> composite exhibited the maximum CO production rate at 609 µmol g<sup>-1</sup> h<sup>-1</sup>. The T–Au nanocomposites played a dual role, functioning both as an electron storage medium to enhance the electron withdrawal from TiO<sub>2</sub> and as an active centre for CO<sub>2</sub> conversion reactions. The exceptional CO selectivity exhibited by T–Au/TiO<sub>2</sub> may stem from the lowered energy threshold for CO formation due to the presence of dual Au NPs, combined with easier desorption on the catalyst surface. Nonetheless, the



Fig. 16 (a) Schematic of  $CO_2$  photoreduction on gold nanoparticles loaded on TiO<sub>2</sub> photocatalyst. (b) Schematic representation of twin stacking fault in Au crystal viewed along the [110] zone axis. Reproduced from ref. 290 with permission from [Wiley Online Library], copyright [2022].

restricted affordability and scarcity of noble metals limit their broad application as ancillary catalysts in photocatalytic processes. Consequently, numerous researchers have been actively seeking viable alternatives to noble metal co-catalysts. However, in comparison with noble metal co-catalysts, the progress in developing non-noble metal counterparts has been relatively sluggish.

In recent years, single-atom catalysts (SACs) have attracted significant interest from researchers due to their active site specificity, and have become one of the frontiers of contemporary catalysis research. Li et al.<sup>291</sup> loaded bimetallic single atoms onto CN to form double active sites. When Mn was combined with Fe, Co, Ni, and Cu, the CO formation rate of the Mn<sub>1</sub>Co<sub>1</sub>-CN catalyst was the highest, reaching 47  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> (Fig. 17a), and the CO selectivity was almost 100%. In the photocatalytic CO<sub>2</sub> conversion reaction, the Mn single atom as the active centre was conducive to the accumulation of holes and accelerating the oxidation of H<sub>2</sub>O (Fig. 17b). Simultaneously, the Co single atom was conducive to the adsorption of CO<sub>2</sub>, thereby accelerating the reduction of  $CO_2$  to CO by electrons (Fig. 17c). To enhance our understanding of SAC mechanisms, density functional theory (DFT) simulations can be advantageous. Nevertheless, if the active centre of the catalyst does not align with the actual active centre, theoretical applications become limited.78,80,299 Therefore, it is crucial to further investigate in situ characterization methods for SACs and achieve their controllable preparation, particularly double or multiple singleatom photocatalysts, to advance sustainable development.

The formation of the high-value-added product CH<sub>4</sub> typically involves a multi-electron reaction. To facilitate the production of the desired product through this multi-electron reaction, it is

essential to accumulate a sufficient number of photogenerated electrons at the reactive site. Noble metals, non-metallic impurities, and vacancy defects in photocatalysts can serve as electron sinks to provide an ample supply of electrons for selective product generation.79,81,314 For example, Wang et al.300 loaded ultrafine Pt nanoparticles on SiC nanosheets, and found that the ultrafine Pt nanoparticles ( $\sim$ 1.8 nm) captured (Fig. 18ad) and accumulated photogenerated electrons in SiC nanoparticles to promote the 8-electron reaction to produce CH<sub>4</sub>. Simultaneously, the CO produced by the 2-electron reaction was chemically absorbed by Pt NPs, and the high-energy electron SiC NSs provided by Pt NPs triggered further CH<sub>4</sub> generation from Pt-CO (Fig. 18e). The CH<sub>4</sub> selectivity of the designed 2 wt% Pt/SiC NSs reached 96.79%. In addition, this study also found that the proton provided by the H<sub>2</sub>O oxidation half-reaction played a role in the formation of CH<sub>4</sub>, and the surface modification increased the CO<sub>2</sub> adsorption and activation sites, enhanced the light absorption, and accelerated the charge separation. Li et al.<sup>301</sup> designed a series of chlorine-modified  $Bi_2WO_6$  nanosheets to study the effect of protons produced by H<sub>2</sub>O oxidation on CH<sub>4</sub> generation. The results showed that the selectivity for CH<sub>4</sub> was regulated by the chloride loading and reached up to 94.98% at the optimum loading. Further intermediate product studies and density functional theory calculations also confirmed that the Cl<sup>-</sup> on Bi<sub>2</sub>WO<sub>6</sub> nanosheets not only promoted the oxidation of H<sub>2</sub>O but also reduced the energy barrier generated by the intermediate \*CHO, thus promoting the formation of CH<sub>4</sub> (Fig. 18f and g). Xu et al.<sup>302</sup> studied carbon nitrogen bromide (CNBr) (Fig. 19a). They found that it could stably and efficiently photocatalyze the reduction of CO2 to CH4 with a selectivity of 70.27%. In their synthesized catalysts, the Br



Fig. 17 (a) Photocatalytic CO yield diagram of bimetallic single-atom Mn and CO on CN. (b) Effects of single atoms of Mn and CO on CN in the photocatalytic oxidation of  $H_2O$ . (c) Effects of single atoms of Mn and CO on CN in photocatalytic CO<sub>2</sub> reduction. Reproduced from ref. 291 with permission from [Wiley Online Library], copyright [2022].



**Fig. 18** (a–d) TEM images of ultrafine SiC nanoparticles. (e) Diagram of loading Pt nanoparticles on SiC nanosheets. (f and g) DFT calculation, free energy diagrams for  $Bi_2WO_6$  and  $Cl-Bi_2WO_6$ , respectively. (a–e) Reproduced from ref. 300 with permission from [Elsevier], copyright [2020]. (f and g) Reproduced from ref. 301 with permission from [ACS Publications], copyright [2020].

atom supplanted the N atom within the tri-s-triazine units, thereby enhancing the local charge separation. The presence of Br acted as an active site, drawing  $CO_2$  to the surface of the catalyst. This facilitated the activation of adsorbed CO<sub>2</sub> by the locally generated photoelectrons, ultimately leading to CH<sub>4</sub> production through hydrogenation. Employing density functional theory (DFT) simulations, it was uncovered that the introduction of Br doping effectively diminished the energy barrier associated with the rate-determining step, thus accelerating the reaction dynamics. This led to an increase in the generation of \*CHO species, subsequently boosting the selectivity for CH<sub>4</sub> production. Xiong et al.<sup>303</sup> designed defects by the hydrothermal treatment of CN, and then introduced Pt single atoms in the defect centre to improve the catalytic efficiency. The CH<sub>4</sub> formation rate of the Pt@Def-CN catalyst was 6.3 µmol  $g^{-1}$  h<sup>-1</sup>, and the selectivity reached 97%. In contrast, the CH<sub>4</sub> formation rate of pure CN was only 0.3  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> (Fig. 19b).

Feng *et al.*<sup>304</sup> used a surfactant-free sedimentary reduction method to deposit both Au and Pd metals on OV-rich  $TiO_2$  (Fig. 19c), and they uncovered the significance of metal sites and semiconductor charge carriers in the integrated process of

 $CO_2$  reduction and  $H_2O$  oxidation. In particular,  $TiO_2$  with an abundance of OVs enhanced water oxidation by supplying more protons. Subsequently,  $CO_2$  and protons were activated at the Au@Pd interface, leading to the formation of surface-attached Au@Pd- $CO_2$ <sup>•-</sup> and Au@Pd-H intermediates. By regulating the proportion of Au@Pd- $CO_2$ <sup>•-</sup>/Au@Pd-H through the modification of the Au@Pd nanoparticle ratio and OV concentration, the efficiency of the catalyst was optimized for oxygen and methane generation, achieving a remarkable  $CH_4$  selectivity of 96%.

Our previous examples consistently demonstrate that the photocatalytic performance can be effectively enhanced through the precise regulation of the heterostructure. However, despite observing a significant increase in activity in all cases, no corresponding improvement in selectivity was observed. Unlike cocatalysts and dopants, it remains unclear whether manipulating the heterostructure can enhance the methane selectivity. This is because the study of photocatalytic materials containing heterojunctions is limited by the difficulty of accurately dealing with the complexity of the interface. Simultaneously, considering the complexity of the synthesis process, how to improve the stability of the reaction process is still



Fig. 19 (a) Reaction pathways of  $CO_2$  photocatalyzed by CNBr. (b) Performance test of Pt@Def-CN catalyst. (c) Schematic diagram of Au and Pd deposition on OV-rich TiO<sub>2</sub> by deposition reduction method. (a) Reproduced from ref. 302 with permission from [Elsevier], copyright [2023]. (b) Reproduced from ref. 303 with permission from [Wiley Online Library], copyright [2022]. (c) Reproduced from ref. 304 with permission from [Elsevier], copyright [2019].

a difficult problem to solve. Therefore, other modification strategies can also be chosen, such as the construction of metal active sites, which is more advantageous than non-metal active sites, while dual or multiple active sites are more conducive to the reaction. In addition, the construction of surface defects may also have the corresponding acceleration effect on the adsorption of CO<sub>2</sub>.<sup>121</sup>

4.1.2 CH<sub>3</sub>OH. CH<sub>3</sub>OH is one of the most promising liquid products for CO<sub>2</sub> reduction, which facilitates fuel cell storage and transportation. Compared with CH<sub>4</sub>, CH<sub>3</sub>OH has higher added value and wider application and is a more ideal product of CO<sub>2</sub> reduction.<sup>79,82,315</sup> For example, CH<sub>3</sub>OH is a precursor to many essential chemicals (aromatics, methane, methyl methacrylate, and fatty acid methyl esters).316,317 Current studies have found that many catalysts can produce CH<sub>3</sub>OH. However, the generation of CH<sub>3</sub>OH involves a 6-electron process, thus necessitating an exceptionally prolonged carrier lifetime to facilitate electron accumulation for achieving selectivity. Table 3 lists the recent advances in the photocatalytic reduction of CO<sub>2</sub> with H<sub>2</sub>O to produce CH<sub>3</sub>OH. According to the reports in the literature, the highest activity for photocatalytic CO2 reduction to produce CH<sub>3</sub>OH in pure water systems reached 511.1  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>,<sup>318</sup> and the selectivity was close to 100%.<sup>319</sup> However, the inherent defects of a single photocatalytic material limit its full potential in photocatalytic activity. As shown in Table 3, researchers have explored various strategies such as ion doping, micro-morphology design, and defect regulation to optimize photocatalysts. Among the materials, carbon-based non-metallic materials such as graphene show great promise in the field of photocatalytic CO<sub>2</sub> reduction for methanol production. With unique surface properties and a textured structure, carbon-based non-metallic materials not only demonstrate excellent photocatalytic activity but can also be used as a cocatalyst for combination with other semiconductors to improve their overall performance. Additionally, carbonbased non-metallic materials offer advantages over semiconductors due to their cost-effectiveness and ease of preparation.

Most work still begins with the design of heterojunctions for the improvement of methanol production. For example, in the work by Ding et al.,<sup>320</sup> a composite catalyst was synthesized by integrating the conventional  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> with g-C<sub>3</sub>N<sub>4</sub>, aiming to establish a Z-scheme heterojunction. This integration facilitated the separation of photogenerated electrons and holes, thereby enhancing the efficiency of the catalyst. At a specific ratio of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> to g-C<sub>3</sub>N<sub>4</sub> of 4:6, the designed catalyst demonstrated a peak CH<sub>3</sub>OH generation rate of 5.63  $\mu$ mol g<sup>-1</sup>  $h^{-1}$ , with a near-perfect selectivity of nearly 100% (Fig. 20a). Wang et al.<sup>321</sup> combined two different Bi-based catalysts, BiVO<sub>4</sub> and Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>, to form a heterojunction, enhancing the capability of electron-hole pair separation, significantly boosting the photocatalytic efficiency for CO2 reduction with H2O to produce CH<sub>3</sub>OH and CO. Specifically, the catalyst performance was optimal when 10% Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub> was incorporated, yielding CH<sub>3</sub>OH and CO at 16.6  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> and 13.29  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, and these figures represented a 12.39-fold and 5.68-fold increase compared to pristine BiVO4, and a 9.88-fold and 2.8-fold improvement over pure Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub>, respectively (Fig. 20b).

Zhou et al.322 proposed a new concept of providing sufficient protons for activated H<sub>2</sub>O without producing strong oxidizing radicals to achieve the highly selective photocatalytic reduction of CO2 and H2O to CH3OH. They loaded cobalt sulphide on carbon nitride and found that cobalt sulphide significantly weakened the overpotential of H<sub>2</sub>O oxidation and promoted the formation of protons without forming strong oxidation radicals such as 'OH and ' $O_2^-$ . Thus, the directed activation of H<sub>2</sub>O promoted the sequential coupling of multiple protons/electrons with CO<sub>2</sub> to form the vital intermediates CHO\* and CH<sub>3</sub>O\*, and finally increased the rate of  $CH_3OH$  formation to 97.3 µmol g<sup>-1</sup>  $h^{-1}$  and the selectivity reached 87.2%.

Although the construction of heterojunctions can achieve the reduction of CO<sub>2</sub> to CH<sub>3</sub>OH, their selectivity is unsatisfactory, possibly because electrons are not transferred quickly enough to the key intermediates CHO\* and CH<sub>3</sub>O\* to participate in the reaction. Based on this, many researchers began to investigate metal doping. Hou et al.323 developed a new flower-

Table 3 Summary of the photocatalytic systems for $CO_2$ reduction with H <sub>2</sub> O to produce CH <sub>3</sub> OH					
Catalyst	Light source	Production/µmol $g^{-1}$ $h^{-1}$	Selectivity	Reference	
Co/g-C <sub>3</sub> N <sub>4</sub> SAC	300 W Xe lamp	235.5	96.2%	257	
2Re-In <sub>2</sub> O <sub>3</sub>	300 W Xe lamp	265.5	100%	317	
Pr <sub>1</sub> -N <sub>4</sub> O <sub>2</sub> /CN	300 W Xe lamp	511.1	92.4%	318	
Cu <sub>x</sub> In <sub>5</sub> S <sub>8</sub> -Cu <sub>v</sub> Se	300 W Xe lamp	5.25	${\sim}100\%$	319	
$\alpha$ -Fe <sub>2</sub> O <sub>3</sub> /g-C <sub>3</sub> N <sub>4</sub>	300 W Xe lamp	5.63	${\sim}100\%$	320	
BiVO <sub>4</sub> /Bi <sub>4</sub> Ti <sub>3</sub> O <sub>12</sub>	300 W Xe lamp	16.6	78.9%	321	
C <sub>3</sub> N <sub>4</sub> -CoS	300 W Xe lamp	97.3	87.2%	322	
BiOBr/La	300 W Xe lamp	63.12	92.4%	323	
Ti-WO <sub>3</sub>	300 W Xe lamp	16.8	88.9%	324	
o-BiVO <sub>4</sub>	300 W Xe lamp	398.3	${\sim}100\%$	325	
<sup>m</sup> CD/CN	300 W Xe lamp	13.9	99.6%	326	
CdS/Al <sub>2</sub> O <sub>3</sub>	10 W LED lamp	144.5	_	327	
Bi <sub>2</sub> S <sub>3</sub> /CeO <sub>2</sub>	300 W Xe lamp	168.35	_	328	
Fe <sub>3</sub> O <sub>4</sub> @ZrO <sub>2</sub> @TiO <sub>2</sub>	400 W Xe lamp	301.29	_	329	
Fe@TiO <sub>2</sub> /SrTiO <sub>3</sub>	300 W Xe lamp	154.20	98.9%	330	
In <sub>2</sub> O <sub>3</sub> -WO <sub>3</sub>	355 nm laser beam	496	_	331	



Fig. 20 (a) Photocatalytic CO<sub>2</sub> reduction activity of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> heterojunction. (b) Measurement of CO<sub>2</sub> reduction yield by photocatalysis with different ratios of BiVO<sub>4</sub>/Bi<sub>4</sub>Ti<sub>3</sub>O<sub>12</sub> heterojunction. (c) AFM of Ti-doped WO<sub>3</sub> ultra-thin nanosheets. (a) Reproduced from ref. 320 with permission from [Elsevier], copyright [2019]. (b) Reproduced from ref. 321 with permission from [Elsevier], copyright [2020]. (c) Reproduced from ref. 324 with permission from [the Royal Society of Chemistry], copyright [2022].

like catalyst, La/BiOBr, by doping La into BiOBr and adjusting its morphology. This modification expanded the photoresponse range of the original BiOBr and diminished the recombination of photogenerated electron-hole pairs. The introduction of La<sup>3+</sup> ions facilitated the migration of electrons residing in the conduction band of BiOBr to La<sup>3+</sup>, enhancing their lifetime and reducing the charge recombination rate, ultimately boosting the production of CH<sub>3</sub>OH. The reported yield of CH<sub>3</sub>OH in this investigation was 63.12 µmol g<sup>-1</sup> h<sup>-1</sup>, accompanied by a selectivity of 88.9%. In the work by Xie *et al.*,<sup>324</sup> an innovative approach was employed to synthesize ultrathin nanosheets with the integration of Ti atoms in the WO<sub>3</sub> structure (Fig. 20c). The reported methanol formation rate for this composite was 16.8 µmol g<sup>-1</sup> h<sup>-1</sup>, a significant enhancement of 3.3 times compared to the undoped WO<sub>3</sub> nanosheets. Impressively, the selectivity towards methanol achieved an outstanding 88.9%. Besides, Xie *et al.*<sup>325</sup> used hexadecyl trimethyl ammonium bromide to construct vanadium vacancies on BiVO<sub>4</sub> and synthesized Vv-rich and Vv-poor with the one-unit-cell thickness *o*-BiVO<sub>4</sub> atomic layers (Fig. 21a). The improvement in light absorption capacity was due to the construction of new defects and the increase in vacancy concentration. The one-unit-cell *o*-BiVO<sub>4</sub> layer with abundant vanadium vacancies had a high surface photovoltage intensity,



**Fig. 21** (a) Schematic of the synthesis of a Vv-rich and Vv-poor o-BiVO<sub>4</sub> atomic layer with a single cell thickness. (b) Comparison of electron lifetime of CN <sup>m</sup>CD/CN and sCD/CN. (c) Mass spectra of the products after <sup>13</sup>CO<sub>2</sub> photoreduction. (a) Reproduced from ref. 325 with permission from [ACS Publications], copyright [2017]. (b and c) Reproduced from ref. 326 with permission from [Nature Publishing Group], copyright [2020].

which ensured its high carrier separation efficiency, and thus increased the CH<sub>3</sub>OH generation rate to 398.3  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>. To further enhance the selectivity for methanol, Tang et al.326 employed a microwave method for synthesizing carbon nitride (CN) modified with carbon dots (<sup>m</sup>CD). This <sup>m</sup>CD possessed unique hole-accepting properties, which could prolong the electron lifetime of CN by six times, thereby facilitating the generation of the 6e<sup>-</sup> product CH<sub>3</sub>OH in the CO<sub>2</sub> reduction reaction (Fig. 21b). The CH<sub>3</sub>OH selectivity approached 100%, and the internal quantum efficiency in the visible light region reached 2.1%, as evidenced by the transient absorption spectrum and isotope labelling, as shown in Fig. 21c. These <sup>m</sup>CD could rapidly extract holes from CN, preventing the surface adsorption of methanol and promoting water oxidation on methanol, thus enhancing the selectivity of CO<sub>2</sub> reduction to alcohol. This study presented a novel strategy for efficiently and selectively reducing carbon dioxide into high value-added chemicals using water.

At present, SACs have become a popular modification strategy because of their high atomic utilization. Thus, Zhou et al.332 prepared Co/g-C3N4 SACs via pyrolysis, and the load of Co atoms on their surface was as high as 24.6 wt%. The CH<sub>3</sub>OH generation rate was increased to 235.5  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> with a selectivity of 96.2%. Their discovery revealed that an extraordinarily high concentration of individual cobalt atoms was firmly situated on the surface of g-C<sub>3</sub>N<sub>4</sub>, forming a Co-N<sub>2</sub>C configuration, where each Co atom was bonded with two nitrogen atoms and one carbon atom. These evenly dispersed Co-N<sub>2</sub>C sites had dual functions, serving both as electron accumulation hubs and as locations for CO<sub>2</sub> absorption and activation. Compared with other catalyst modification measures, SACs still have a great advantage in the photocatalytic CO<sub>2</sub> reduction to CH<sub>3</sub>OH. It can be seen that the design on the molecular or atomic scale of catalyst modification is more favourable for improving the photocatalytic performance.

**4.1.3** HCOOH. In the process of CO<sub>2</sub> reduction, HCOOH is one of the important intermediates, and most of the products from CO<sub>2</sub> reduction need to go through the HCOOH intermediate. HCOOH is also an important chemical raw material, which can be used as a hydrogen storage material. On an industrial scale, HCOOH is mainly produced from methyl formate, which is synthesized from CH<sub>3</sub>OH and CO in the presence of strong salts and bases. Therefore, it is highly desirable to develop environmentally friendly and sustainable processes for HCOOH production through CO2 conversion using solar, wind, and/or hydroelectric power sources.333

Although photocatalytic CO2 reduction to HCOOH is a dynamic favourable process with double electron transfer, the efficiency and selectivity of the photocatalyzed CO<sub>2</sub> to HCOOH are not high at present. Most catalysts need to reduce CO<sub>2</sub> to HCOOH under certain conditions, and pure CO<sub>2</sub> is not suitable as a good carbon source. Typically, bicarbonate is used as a carbon source and a sacrificial agent is added to convert it into HCOOH. Therefore, how to efficiently convert CO2 into HCOOH is a major challenge at present. Table 4 lists the recent advances in the photocatalytic reduction of CO2 with H2O to HCOOH product.

As can be seen in Table 4, although many modification strategies have been employed to expand the light absorption, accelerate the charge carrier separation, and promote CO<sub>2</sub> adsorption and activation, without the addition of a sacrificial agent or photosensitizer, only low-value gas products have been produced. At present, the highest activity reported for the photocatalytic CO2-induced HCOOH production is 896.7 µmol  $g^{-1}$  h<sup>-1</sup>.<sup>334</sup> In recent years, metal oxides, graphene, sulphides, etc. have been widely explored to promote charge separation by forming Schottky or Z-scheme heterojunctions to improve the CO2 reduction reaction performance. However, the photocatalytic activity for HCOOH production is not ideal. An increasing number of researchers have begun to look at MOF photocatalysts to improve the photocatalytic performance, because MOFs can be endowed with clear and isolated sites to anchor the catalytic substance compared to other solid carriers with non-uniformly dispersed sites. Zhang et al.335 developed a CuO/Ag/UiO-66 Z-type heterojunction composite catalyst and investigated the impact of Ag additive on enhancing the selectivity for CO2 reduction. Comparative studies were conducted on CuO, CuO/UiO-66, and CuO/Ag/UiO-66. Pure CuO primarily produced CH<sub>3</sub>OH. Upon the formation of the CuO/UiO-66 heterostructure, HCOOH was observed as the main product. However, with the addition of Ag, the selectivity significantly increased from 63% to 95.9% (Fig. 22a). This is probably the highest selectivity achieved without the use of sacrificial agents in liquid-phase photocatalytic CO2 reduction. This enhancement was attributed to the highly dispersed Ag acting as an electron transport medium in the CuO/Ag/UiO-66 Z-type

Table 4         Summary of the photocatalytic systems for CO2 reduction to produce HCOOH					
Catalyst	Light source	Production/ $\mu$ mol g <sup>-1</sup> h <sup>-1</sup>	Selectivity	Reference	
CuZn/SrTiO <sub>3</sub>	Hg Xe lamp (240–300 nm)	8	75%	333	
Cu <sub>2</sub> O-Pt/SiC/IrO <sub>x</sub>	300 W Xe lamp	896.7	—	334	
CuO/Ag/UiO-66	300 W Xe lamp	63	95.9%	335	
NH2-C@Cu2O	Al Ka (1486.6 eV, 250 W)	138.65	92%	336	
Pt/In <sub>2</sub> O <sub>3</sub> /g-C <sub>3</sub> N <sub>4</sub>	3 W LED light (420 nm)	63.1	90.1%	337	
g-C <sub>3</sub> N <sub>4</sub> /Cu-TiO <sub>2</sub>	Visible light	2534.5	—	338	
Cu@Cu2O/C-350	300 W Xe lamp	81.58	—	339	
PCN-222/CsPbBr <sub>3</sub> QDs	300 W Xe lamp (420 nm)	189.9	${\sim}100\%$	340	
Ir SACs/NH <sub>2</sub> -UiO-66	300 W Xe lamp	3.38	96.3	341	



**Fig. 22** (a) Adsorption model of CO<sub>2</sub> on CuO/Ag/UiO-66 catalyst. (b) Activities of pure CuO, UiO-66, CuO/UiO-66 and heterojunction CuO/Ag/UiO-66. (c) Schematic diagram of the preparation of  $NH_2$ –C@Cu<sub>2</sub>O photocatalyst by low temperature carbonization method. (d) Yield test diagram of  $NH_2$ –C@Cu<sub>2</sub>O photocatalytic CO<sub>2</sub> reduction. (a and b) Reproduced from ref. 335 with permission from [Elsevier], copyright [2022]. Reproduced from ref. 336 with permission from [the Royal Society of Chemistry], copyright [2023].

heterojunction to facilitate the efficient transport and separation of photogenerated electrons. Simultaneously, the *in situ* infrared characterization and theoretical calculations also demonstrated that Ag exhibits strong adsorption capacity for \*H protons, favouring their combination with the  $CO_2$  primary activation intermediate \*OCO through oxygen coordination at nearby sites (Fig. 22b). This led to the generation of the formic acid intermediate \*OCOH, and ultimately resulted in highly selective formic acid products. These findings highlight how Ag effectively modulates the  $CO_2$  reduction pathway by regulating the reactive intermediate species.

Li et al.336 synthesized a metal-organic skeleton (MOF) precursor material (-NH2-Cu-MOF) to enhance the CO2 adsorption and improve the long-term stability of the catalyst (Fig. 22c). The method adopted a low-temperature carbonization process (300 °C) to synthesize the -NH<sub>2</sub> carbon skeletonmodified Cu<sub>2</sub>O photocatalyst. In this study, acid-base neutralization was utilized to enhance the adsorption of -NH<sub>2</sub> on the acidic CO<sub>2</sub>, thereby activating -NH<sub>2</sub> as the active centre. Moreover, the introduction of amino functional groups served as an electron-withdrawing agent, thereby significantly inhibiting carrier recombination. Simultaneously, the unordered in situ carbon network originating from the organic ligands played a protective role in preventing Cu<sub>2</sub>O from photo-corrosion. Consequently, this led to improved selectivity and productivity in the conversion of CO<sub>2</sub> to HCOOH. The achieved formic acid yield increased to 138.65  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> (Fig. 22d), which was

three-times higher than that achieved with pure Cu<sub>2</sub>O. These findings demonstrated that modifying Cu<sub>2</sub>O with an  $-NH_2$ modified carbon skeleton offers a novel approach for designing efficient and stable photocatalysts. Li *et al.*<sup>337</sup> developed a Pt/ In<sub>2</sub>O<sub>3</sub>/g-C<sub>3</sub>N<sub>4</sub> multifunctional catalyst *via* the controlled assembly method. This catalyst produced H atoms through water splitting, adsorbed and activated CO<sub>2</sub> molecules, and inhibited the recombination of photogenerated electrons and holes by accelerating the electron transfer and forming a heterojunction. Thus, it showed high activity and selectivity for HCOOH in the photocatalytic CO<sub>2</sub> reduction process driven by visible light.

Plasmonic metals display outstanding photocatalytic activity in the visible region due to their surface plasmon resonance effect for CO<sub>2</sub> reduction, which is attributed to their generation of hot electrons. Among the ionized metals such as Au, Ag, and Cu nanocrystals, Cu-based catalysts have attracted significant attention in the field of plasma catalysis due to their abundant reserves and cost-effectiveness. Saidina Amin *et al.*<sup>338</sup> loaded Cu onto g-C<sub>3</sub>N<sub>4</sub> and TiO<sub>2</sub> (Fig. 23). In the process of catalyst treatment, Cu was easily oxidized to form CuO, and thus formed ternary heterojunctions. The electrons on the loaded Cu were transferred to CuO, and thus the CO<sub>2</sub> adsorbed on the active sites obtained electrons to produce CH<sub>3</sub>OH and HCOOH. In the designed catalyst, enhancements in both photocatalytic activity and product specificity were achieved through strategic manipulation of the g-C<sub>3</sub>N<sub>4</sub> to Cu–TiO<sub>2</sub> proportion. At a balance



Fig. 23 Schematic reaction mechanism of  $g-C_3N_4/Cu/TiO_2$  composites for the reduction of  $CO_2$  to  $CH_3OH$  and HCOOH under UV light (b) and visible light. (a). Reproduced from ref. 338 with permission from [Elsevier], copyright [2017].

of 30% g-C<sub>3</sub>N<sub>4</sub> and 70% Cu-TiO<sub>2</sub>, the system demonstrated a superior performance with the CH<sub>3</sub>OH and HCOOH production rates reaching 1287  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> and 2534.5  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, respectively. Li et al.339 successfully employed a Cu-MOF precursor to synthesize homogeneous core-shell nanoparticles consisting of a Cu<sub>2</sub>O shell coated with a Cu core. The surface Cu<sub>2</sub>O shell served as the active site for the adsorption and activation of CO2 molecules, while the heterojunction between the plasma Cu and Cu<sub>2</sub>O facilitated efficient photoelectron transport to the adsorbed CO2 molecules. In the absence of any additives, the composite catalyst of Cu@Cu2O/C-350 demonstrated photocatalytic reduction of CO<sub>2</sub> to HCOOH with a yield of 67.35  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>. This study paved the way for developing cost-effective copper-based photocatalysts. According to previous studies, it is clear that the photocatalytic reduction of CO<sub>2</sub> to HCOOH remains a significant challenge at present. Among the various strategies, constructing heterojunctions has emerged as a common approach. Additionally, band gap engineering can modulate the band structure of photocatalysts to match the desired reduction potential for CO to HCOOH (-0.61 V).

For instance, in the work by Murugesan et al.,<sup>16</sup> the successful construction of a direct Z-scheme heterostructure involving AgCl/g-C<sub>3</sub>N<sub>4</sub> was accomplished, resulting in a remarkable selectivity of 88.9% for CO2 to HCOOH. Nonetheless, this approach struggles to hinder other thermodynamically more favourable reactions, such as the conversion of  $CO_2$  to  $CH_3OH$  (-0.38 V). Moreover, the integration of organic sacrificial agents, including methanol, ethanol, triethanolamine, trimethylamine, and ethylenediaminetetraacetic acid, is instrumental in exhausting the photogenerated holes, enhancing the electron-hole separation, and supplying protons for \*H reduction in CO. For instance, Hao et al.18 utilized isopropyl alcohol as a sacrificial agent within MOF membranes (Ir SACs/NH<sub>2</sub>-UiO-66). Nevertheless, it should be noted that the addition of these reagents inevitably complicates the product separation and increases the expense of photocatalytic CO<sub>2</sub> emission reduction, which contradicts the advantages of green photocatalytic systems and hinders clean energy applications. Therefore, there is an imperative need to explore universal approaches for achieving the highly selective production of HCOOH through photocatalytic CO2.337

### 4.2 C<sub>2</sub> and C<sub>2+</sub> products

In the series of products of CO<sub>2</sub> reduction, C<sub>2</sub> and C<sub>2+</sub> products have higher added value compared with other products. However, only a few catalyst systems can produce C<sub>2</sub> products. Moreover, most of the C<sub>2</sub> products produced are additional products rather than the main product, and the yield is very low. The photocatalysis of CO<sub>2</sub> with H<sub>2</sub>O to produce high-yield and selective C<sub>2</sub> products is a major challenge using the current catalytic systems.54,120 Present investigations emphasize that obtaining C2 and C2+ compounds relies on minimizing the recombination of photogenerated electrons and holes, while augmenting their separation efficiency within the catalyst. Additionally, an effective catalyst must possess robust capabilities for C-C bonding.342,343 As shown in Table 5, we summarized the recent progress in photocatalytic CO<sub>2</sub> reduction with H<sub>2</sub>O, leading to the formation of C<sub>2</sub> and C<sub>2+</sub> derivatives. As can be seen in Table 5, there are few studies on  $C_2$  products at present. Regarding the reported photocatalysts, the yield of C2+ products formed by photocatalytic CO<sub>2</sub> reduction with H<sub>2</sub>O as the reducing agent is generally low, and most of them have low selectivity. Among them, the peak efficiency in the photocatalytic conversion of CO2 into C2H6 stands at an impressive 38.2  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>.<sup>344</sup> Predominantly, researchers opt for metal oxides as photocatalysts due to their excellent stability. However, unmodified photocatalysts often encounter constraints stemming from swift carrier recombination and an incompatible energy band structure, necessitating their enhancement through modification. Chen et al.345 fabricated Ni nanocluster-embedded black TiO2 (Ni/TiO2(OVs)) photocatalysts with integrated dual active sites through a strategy that merged metal support and surface defect architectures. The combination of Ni nanoclusters and oxygen vacancies (OVs) not only established a fast electron transfer pathway, but also the band gap was reduced to enhance the response to visible light. The Ni/TiO<sub>2</sub>(OV) photocatalyst possessed stable CO<sub>2</sub> binding sites and the lowest activation energy (0.08 eV), which improved the product selectivity and enhanced the photocatalytic activity. The principal output was CH<sub>3</sub>CHO with a production rate of 1.67  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, an output 18 times greater than that achieved by the commercial TiO2. Tatsumi Ishihara and collaborators aimed to transform CO<sub>2</sub> into C<sub>2</sub> compounds by

Table 5	Summary of	the photocatalytic	c systems for	CO <sub>2</sub> reduction	with $H_2O$ to	produce $C_2$
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Catalyst	Light source	Production/µmol $g^{-1}$ $h^{-1}$	Selectivity	Reference
Au <sub>1</sub> /RP	300 W Xe lamp	C <sub>2</sub> H <sub>6</sub> 1.32	96%	342
Hydrogenated SrTiO <sub>3</sub>	Ultraviolet lamp (8 W)	$C_2H_2$ 6.5	76%	343
$Cd_{0.8}Zn_{0.2}S/In_2O_3$	300 W Xe lamp	$C_2H_6$ 38.2	82.6%	344
Ni/TiO <sub>2</sub> (OVs)	300 W halogen lamp	CH <sub>3</sub> CHO 1.67	_	345
$WO_3 \cdot 0.33H_2O$ nanotube	Solar light	CH <sub>3</sub> COOH 9.4	85%	346
$Cu^{\delta^+}/CeO_2 - TiO_2$	300 W Xe lamp (320-850 nm)	$C_2H_4$ 4.51	73.9%	347
CuGaS <sub>2</sub> @CuO	450 W Xe lamp (with UV cut-off filter)	$C_2H_4$ 20.6	75.1%	348
P/Cu SAs@CN	300 W Xe lamp	C <sub>2</sub> H <sub>6</sub> 616.6	_	349
ZSM-5@NiV <sub>2</sub> Se <sub>4</sub>	300 W Xe lamp	$C_2H_6$ 1.85	90.1%	350
Bi <sub>19</sub> S <sub>27</sub> Cl <sub>3</sub>	420 W Xe lamp	CH <sub>3</sub> CH <sub>2</sub> OH 5.19	85%	351
AuIr@GaN	300 W Xe lamp	$C_2H_6$ : 58 800	5.6%	352
CuO <sub>x</sub> @p-ZnO	300 W Xe lamp	$C_2H_4$ 2.7	32.9%	353
Pd-Co <sub>3</sub> O <sub>4</sub>	300 W Xe lamp	CH <sub>3</sub> COOH 13.8	100%	354

surmounting the energy barrier associated with multipleelectron transfer. In their approach,<sup>346</sup> OVs played a crucial role in capturing a substantial quantity of electrons to overcome this energy barrier. However, due to the unstable OVs, low concentration and easy oxidation on the surface, they designed ultrathin  $WO_3 \cdot 0.33H_2O$  nanosheets with a large number of stable OVs (Fig. 24a), which could stably and efficiently convert CO<sub>2</sub> into CH<sub>3</sub>COOH by using water as the oxidant under visible light. The yield reached 9.4 µmol g<sup>-1</sup> h<sup>-1</sup>, and the selectivity reached 85%. In the work by Huang *et al.*,<sup>347</sup> an interface-modification approach was employed to construct a Cu<sup> $\delta^+$ </sup>/CeO<sub>2</sub>-TiO<sub>2</sub> photocatalyst, featuring Cu<sup> $\delta^+$ </sup> sites uniformly distributed across CeO<sub>2</sub>-TiO<sub>2</sub> heterostructures (Fig. 24b). This novel photocatalyst exhibited a remarkable enhancement in



Fig. 24 (a) Schematic illustration of the hydrothermal procedure for the synthesis of oxygen-deficient hydrated  $WO_3 \cdot 0.33H_2O$  nanotube. (b) Schematic diagram of the construction of  $Cu^{\delta+}/CeO_2$ -TiO<sub>2</sub> photocatalyst. (c) Simulation of CuO *in situ* on the upper surface of wurtzite phase of CuGaS<sub>2</sub>. (a) Reproduced from ref. 346 with permission from [ACS Publications], copyright [2018]. (b) Reproduced from ref. 347 with permission from [ACS Publications], copyright [2022]. (c) Reproduced from ref. 348 with permission from [Wiley Online Library], copyright [2023].

both efficiency and selectivity for  $C_2H_4$  production, reaching 4.51 µmol g<sup>-1</sup> h<sup>-1</sup> and 47.5%, respectively. These figures represented a 2.36-fold and 1.32-fold improvement in that by the  $Cu^{\delta+}/TiO_2$  catalyst, respectively. Peter *et al.*<sup>348</sup> proposed that surface restructuring facilitated C–C bonding in the CO<sub>2</sub>-to- $C_2H_4$  conversion process. The *in situ* formation of wurtzite phase CuGaS<sub>2</sub> occurred on the catalyst surface, accompanied by the development of a thin CuO layer atop the initial catalyst (Fig. 24c). Consequently, CO<sub>2</sub> was effectively transformed into  $C_2H_4$  with a high selectivity of 75.1% and a yield of 20.6 µmol g<sup>-1</sup> h<sup>-1</sup>.

Some noble metals can enrich electrons, and single atoms can be used as catalytic active sites. In the work by Li et al.,<sup>342</sup> they dispersed individual Au atoms onto red phosphorus (Au<sub>1</sub>/ RP). The electronically rich phosphorus atoms adjacent to these Au single atoms served as active sites for CO<sub>2</sub> activation, as illustrated in Fig. 25. This configuration notably diminished the energy barrier for C-C bonding, thereby enhancing the kinetic process of C<sub>2</sub>H<sub>6</sub> formation. Consequently, Au<sub>1</sub>/RP exhibited a C<sub>2</sub>H<sub>6</sub> selectivity of 96% and activity of 1.32  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup>, which was achieved without the use of a sacrificial agent. Mao et al.349 developed a photocatalyst consisting of dual P and Cu sites integrated on g-C<sub>3</sub>N<sub>4</sub> (P/Cu SACs@CN). This innovative design resulted in an impressive C<sub>2</sub>H<sub>6</sub> generation rate of 616.6  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> during the conversion of CO<sub>2</sub> into hydrocarbons. This research highlighted that the presence of a dual-site structure on the surface of g-C<sub>3</sub>N<sub>4</sub> played a pivotal role in the effective conversion of CO<sub>2</sub> into multi-carbon compounds. The kinetic study of charge carriers also showed that the multielectron transfer process for CO<sub>2</sub> reduction, attributed to the electron and hole trapping sites of the Cu and P double sites in the CN matrix, promoted the formation of charge-rich Cu sites, endowing the P/Cu SAC@CN photocatalysts with excellent charge separation capability.

According to the available studies, evidently all elements contribute to electron and proton accumulation, including elevated photon intensity and energy, presence of co-catalysts and reduced charge recombination, which positively impact the generation of  $C_2$  and higher carbon compounds. From doping nonmetals to metals and building single atoms to building double active sites, catalysts tend to be designed for active sites.

Therefore, more catalyst designs are not satisfied with a single method but use multiple strategies for simultaneous modification such as doping metal or non-metal and surface defect construction. Recent studies have also shown that the presence of both can indeed improve the selectivity for  $C_2$ products; however, they also increase the difficulty of explaining the formation mechanism.

In addition, long-term stability is essential for the practical application of photocatalytic CO2 reduction. However, most photocatalytic materials are prone to deactivation during reduction reactions. Furthermore, the development of the stable photocatalyst materials for CO2 reduction still faces great challenges. This, many studies have been devoted to addressing this issue. Some approaches have been studied to improve the stability of photocatalytic CO<sub>2</sub> reduction reaction systems. Photocatalytic CO<sub>2</sub> reduction can usually be performed in highly polar (e.g., dimethylformamide (DMF), dimethyl sulfone (DMSO), and acetonitrile (ACN)), medium polar (e.g., ethyl acetate (EA) and tetrahydrofuran (THF)), and low polar (e.g., ether and toluene) reaction media. However, semiconductors such as halide perovskites are severely unstable when exposed to water, highly polar hydrophobic solvents, oxygen, light, and heat. This is because they have strong ionic properties, making them vulnerable to damage upon contact with polar reaction media. Consequently, they immediately degrade into other structures, and thus choosing the appropriate reaction medium can improve their stability.355 Also, a protective layer is designed to encapsulate unstable semiconductors to prevent their degradation due to various factors such as photocorrosion and strong acids and bases. Li et al.356 reported the synthesis of a double-doped CsPbCl3 nanocrystal (NC) with core-shell structure as a CO<sub>2</sub> reduction photocatalyst. Due to the size difference between the dopant and bulk ion, surface segregation occurs during the process of elastic stress relaxation. Based on the surface segregation effect, a solid BI-rich shell could be constructed to give the CsPbCl<sub>3</sub> NCs a stable crystal structure and morphology. The core-shell double doped CsPbCl<sub>3</sub> nanocrystal structure maintained the initial photocatalytic efficiency of 97% without changes in its structure, morphology and size under a harsh environment such as strong irradiation and high humidity. Ziarati et al.357 synthesized a three-dimensional coreshell TiO2-x/LDH photocatalyst through three steps including



Fig. 25 (a) CO<sub>2</sub> reduction yield of Au<sub>1</sub>/RP photocatalysis. (b) Diagram of implanting single-atom Au reaction intermediates on red phosphorus. (a and b) Reproduced from ref. 342 with permission from [ACS Publications], copyright [2022].

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solvothermal, hydrogen treatment and hydrothermal reaction. This structure reveals high efficiency for the photoreduction of carbon dioxide into a solar fuel in the absence of a precious metal co-catalyst. The CH<sub>4</sub> yield increased gradually with the extension of the reaction time, and still reached 63  $\mu$ mol g<sup>-1</sup> h<sup>-1</sup> after 12 h illumination. It was concluded that the core-shell photocatalyst avoids the spontaneous movement of surface dispersed particles through space interaction and isolation, and effectively improves the thermal stability and service life of the composite catalyst. However, in practical applications, is necessary to consider the fact that the encapsulation layer should not affect the photo-adsorption efficiency and mass transfer capacity of the semiconductor itself while playing the role of a protective layer.

## 5 Conclusion and perspective

The use of semiconductor materials and solar photocatalytic reduction of CO<sub>2</sub> to synthesize hydrocarbon fuels is currently one of the ideal models for purifying the environment and the renewable utilization of carbon resources. Herein, we reviewed the common strategies for the design of photocatalysts for CO<sub>2</sub> reduction in recent years, including defect construction, cocatalyst loading, doping, heterojunction formation, singleatom engineering, and surface organometallic catalysis. These modification methods boost the efficiency of electron and hole separation and migration, manipulate the electronic and energy band configuration, and strengthen the capability to adsorb CO2 to improve the CO2 reduction performance and regulate the product selectivity. Currently, photocatalysis technology has made some progress and breakthroughs in cleaning the environment and converting solar energy into renewable energy sources through the efforts of researchers around the world. However, given that photocatalysis is a complex reaction process, there are still some bottlenecks in CO<sub>2</sub> photoreduction technology, as follows: (1) the product activity is low. (2) The unclear reaction path and intermediates lead to inaccurate regulation of the final target products. (3) At present, the main reduction products mainly consist of CO and CH<sub>4</sub>, where it is difficult to generate multi-carbon products. (4) The understanding of the catalytic reaction mechanism is still not thorough, such as the multi-scale migration and separation mechanisms of photogenerated charges, and the kinetics and mechanism of catalyst surface interface reaction, which limit the understanding of the catalytic reaction properties.

To break through these bottlenecks, it is necessary to develop new types of photocatalytic materials and study the basic issues in photocatalytic reactions, which is also the next development direction of photocatalysis. For the design of photocatalysts, researchers can continue to study the properties of the photo-absorption of the catalyst, the mechanism of photogenerated carrier separation and the tendency of migration to the surface, which are conducive to improving the photocatalytic efficiency. Based on the research results, we believe that some attempts and explorations can be made in the following three aspects. Firstly, dual cocatalysts can be designed to improve the photocatalytic reduction activity of  $CO_2$ , as well

as study the loading form and preparation method. Secondly, because a composite of semiconductor materials can improve the charge separation rate and expand the spectral response range, research can be conducted on composite semiconductor photocatalytic materials. Thirdly, microstructure control is also an important factor in improving the photocatalytic activity of semiconductors. Due to the fact that photocatalytic reactions occur on the surface of materials, special surface microstructures in materials can significantly affect their photocatalytic activity.

The entire process of photocatalytic CO<sub>2</sub> reduction involves multiple steps, intermediates, and different products, making the process of product selectivity more complex. To accurately regulate the product, we can first clarify the reaction mechanism of photocatalytic CO<sub>2</sub> reduction and optimize its key steps to adjust the product. The photocatalytic CO<sub>2</sub> reduction process involves carrier kinetics and surface catalysis. These processes are influenced by numerous factors, including band structure, charge separation efficiency, and the absorption and activation of the reactants. The dynamics of CO<sub>2</sub> adsorption and activation significantly impact the performance and selectivity in photocatalytic CO<sub>2</sub> conversion; however, revealing the kinetics of these steps in photocatalytic systems remains an arduous task. Therefore, the systematic investigation of the factors influencing the mechanism of photocatalytic CO2 reduction and product selectivity is extremely important. The carrier kinetics and the energy barrier associated with each step in the photocatalytic CO<sub>2</sub> reduction processes can be elucidated through theoretical simulations, and in situ or space/time resolution advanced characterization technology. This will facilitate an indepth exploration of the factors governing the product generation and uncover the underlying mechanism of photocatalytic  $CO_2$  reduction. It is expected that joint investigations on the structure-effect relationship of photocatalysts using solar energy to convert CO2 and H2O into various carbon fuels will break new ground. This provides valuable theoretical guidance for developing highly selective photocatalytic CO<sub>2</sub> reduction catalysts, thus achieving the goal of carbon neutrality.

The surface of a catalyst always changes in the catalytic process. At present, most of the studies on surface reconstruction are carried out in electrocatalytic and thermocatalytic reactions, but there are few studies on the photocatalytic process. Therefore, studies on the surface reconstruction of the catalyst during the reaction may be the future research direction of photocatalytic CO2 reduction. Surface reconstruction typically exerts the following impacts on catalytic reactions. Firstly, it alters the distribution of active sites by changing the atomic arrangement and crystal structure of the catalyst surface. This modification can either enhance or diminish the number and utilization of active sites, thereby influencing the catalytic performance. Secondly, surface reconfiguration can induce alterations in the interaction between the reactants and catalyst, thereby affecting the adsorption behavior of the reactants. Simultaneously, altering the diffusion of reactants on the catalyst surface significantly influences the reaction rate and selectivity. Furthermore, structural damage or deactivation of the active sites may occur during certain catalytic processes. By

restructuring the surface structure through the rearrangement or repair of surface atoms, the stability and service life of the catalyst can be enhanced. Therefore, comprehending and regulating surface reconfiguration are pivotal in designing efficient new catalysts as well as optimizing their catalytic performance. Also, studying the underlying mechanisms of surface reconstruction together with kinetic processes can 16 D

## surface reconstruction together with kinetic processes can clarify the structure–activity relationships, providing a theoretical foundation for the design and preparation of catalysts.

## Data availability

No primary research results, software or code has been included and no new data were generated or analysed as part of this review.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- 1 S. Chu and A. Majumdar, Nature, 2012, 488, 294-303.
- F. P. Garcia de Arquer, C. T. Dinh, A. Ozden, J. Wicks, C. McCallum, A. R. Kirmani, D. H. Nam, C. Gabardo, A. Seifitokaldani, X. Wang, Y. C. Li, F. Li, J. Edwards, L. J. Richter, S. J. Thorpe, D. Sinton and E. H. Sargent, *Science*, 2020, 367, 661–666.
- 3 B. D. Patterson, F. Mo, A. Borgschulte, M. Hillestad, F. Joos, T. Kristiansen, S. Sunde and J. A. van Bokhoven, *Proc. Natl. Acad. Sci. U. S. A.*, 2019, **116**, 12212–12219.
- 4 X. Zhang, F. Xie, X. Li, H. Chen, Y. She, C. Wang, Z. Mo, W. Yang, P. Hou, C. Wu, H. Xu and H. Li, *Appl. Surf. Sci.*, 2021, **542**, 22075–22082.
- 5 M. Mikkelsen, M. Jørgensen and F. C. Krebs, *Energy Environ. Sci.*, 2010, **3**, 43–81.
- 6 S. N. Habisreutinger, L. Schmidt-Mende and J. K. Stolarczyk, *Angew. Chem., Int. Ed.*, 2013, **52**, 7372–7408.
- 7 J. Bonin, A. Maurin and M. Robert, *Coord. Chem. Rev.*, 2017, **334**, 184–198.
- 8 Q. Wang, J. Cai, G. V. Biesold-McGee, J. Huang, Y. H. Ng, H. Sun, J. Wang, Y. Lai and Z. Lin, *Nano Energy*, 2020, 78, 105313.
- 9 J. Chen, R. Abazari, K. A. Adegoke, N. W. Maxakato, O. S. Bello, M. Tahir, S. Tasleem, S. Sanati, A. M. Kirillov and Y. Zhou, *Coord. Chem. Rev.*, 2022, **469**, 214664.
- 10 Q. Xu, Z. Xia, J. Zhang, Z. Wei, Q. Guo, H. Jin, H. Tang, S. Li, X. Pan, Z. Su and S. Wang, *Carbon Energy*, 2022, 5, e205.
- 11 B. M. Pirzada, A. H. Dar, M. N. Shaikh and A. Qurashi, *ACS Omega*, 2021, **6**, 29291–29324.
- 12 A. Li, X. Cao, R.-Z. Fu and S.-Q. Guo, *Biotechnol. Adv.*, 2024, 72, 108347.

- 13 Y. Wang, P. Han, X. Lv, L. Zhang and G. Zheng, *Joule*, 2018, 2, 2551–2582.
- 14 S. Wang and G.-Q. Lu, Energy Fuels, 1996, 10, 896-904.
- 15 J. Schneider, M. Matsuoka, M. Takeuchi, J. Zhang, Y. Horiuchi, M. Anpo and D. W. Bahnemann, *Chem. Rev.*, 2014, **114**, 9919–9986.
- P. Murugesan, S. Narayanan, M. Manickam,
  P. K. Murugesan and R. Subbiah, *Appl. Surf. Sci.*, 2018, 450, 516–526.
- 17 C.-H. Huang and C.-S. Tan, *Aerosol Air Qual. Res.*, 2014, **14**, 480–499.
- 18 Y. C. Hao, L. W. Chen, J. Li, Y. Guo, X. Su, M. Shu and Q. Zhang, *Nat. Commun.*, 2021, **12**, 2682.
- 19 W. C. Chueh, C. Falter, M. Abbott, D. Scipio, P. Furler, S. M. Haile and A. Steinfeld, *Science*, 2010, 330, 1797–1801.
- 20 D. D. Zhu, J. L. Liu and S. Z. Qiao, Adv. Mater., 2016, 28, 3423-3452.
- 21 A. Corma and H. Garcia, J. Catal., 2013, 308, 168-175.
- 22 S. Cheng, Z. Sun, K. H. Lim, A. A. Wibowo, T. Zhang, T. Du, L. Liu, H. T. Nguyen, G. K. Li, Z. Yin and S. Kawi, *ACS Catal.*, 2023, 13, 7221–7229.
- 23 J. Ran, M. Jaroniec and S. Z. Qiao, *Adv. Mater.*, 2018, **30**, 1704649.
- 24 Y. Zhang, L. Cao, G. Bai and X. Lan, Small, 2023, 19, 230035.
- 25 Y. Huo, P. Zhang, J. Chi, F. Fang, Y. Song and D. Sun, *Adv. Energy Mater.*, 2024, **14**, 2304282.
- 26 Z. Wu, S. Guo, L.-H. Kong, A.-F. Geng, Y.-J. Wang, P. Wang, S. Yao, K.-K. Chen and Z.-M. Zhang, *Chin. J. Catal.*, 2021, 42, 1790–1797.
- 27 D.-C. Liu, D.-C. Zhong and T.-B. Lu, *EnergyChem*, 2020, 2, 10034.
- 28 W. Chen, B. Han, C. Tian, X. Liu, S. Liang, H. Deng and Z. Lin, *Appl. Catal.*, B, 2019, 244, 996–1003.
- 29 X. Chang, T. Wang and J. Gong, *Energy Environ. Sci.*, 2016, 9, 2177–2196.
- 30 G. Yin, Q. Bi, W. Zhao, J. Xu, T. Lin and F. Huang, *ChemCatChem*, 2017, **9**, 4389-4396.
- 31 W. Yu, D. Xu and T. Peng, *J. Mater. Chem. A*, 2015, **3**, 19936–19947.
- 32 X. Li, J. Wen, J. Low, Y. Fang and J. Yu, *Sci. China Mater.*, 2014, **57**, 70–100.
- 33 S. Sorcar, Y. Hwang, C. A. Grimes and S.-I. In, *Mater. Today*, 2017, **20**, 507–515.
- 34 A. Meng, S. Wu, B. Cheng, J. Yu and J. Xu, *J. Mater. Chem. A*, 2018, **6**, 4729–4736.
- 35 S. Xie, Y. Wang, Q. Zhang, W. Deng and Y. Wang, ACS *Catal.*, 2014, **4**, 3644–3653.
- 36 J. Low, L. Zhang, T. Tong, B. Shen and J. Yu, J. Catal., 2018, 361, 255–266.
- 37 F. Xu, J. Zhang, B. Zhu, J. Yu and J. Xu, *Appl. Catal.*, B, 2018, 230, 194–202.
- 38 J. L. White, M. F. Baruch, J. E. Pander, Y. Hu, I. C. Fortmeyer, J. E. Park, T. Zhang, K. Liao, J. Gu, Y. Yan, T. W. Shaw, E. Abelev and A. B. Bocarsly, *Chem. Rev.*, 2015, **115**, 12888–12935.
- 39 J. Fu, K. Jiang, X. Qiu, J. Yu and M. Liu, *Mater. Today*, 2020, 32, 222–243.

- Review
- 40 A. Vasileff, C. Xu, Y. Jiao, Y. Zheng and S.-Z. Qiao, *Chem*, 2018, 4, 1809–1831.
- 41 Y. Wang, Q. Guan and J. Xiang, Int. J. Surg., 2018, 55, 98– 102.
- 42 T. Wang, X. Meng, G. Liu, K. Chang, P. Li, Q. Kang, L. Liu, M. Li, S. Ouyang and J. Ye, *J. Mater. Chem. A*, 2015, 3, 9491– 9501.
- 43 H. Zhang, T. Wang, J. Wang, H. Liu, T. D. Dao, M. Li, G. Liu, X. Meng, K. Chang, L. Shi, T. Nagao and J. Ye, *Adv. Mater.*, 2016, 28, 3703–3710.
- 44 Y. Wang, E. Chen and J. Tang, *ACS Catal.*, 2022, **12**, 7300–7316.
- 45 K. Zhao, W. Pang, S. Jiang, C. Hu, P. Liu, D. Cui, X. An, B. Tian, C. Gao, P. Zhang, M. Tian, D. Fu and H. Zhao, *Nano Res.*, 2023, **16**, 4812–4820.
- 46 B. Lei, W. Cui, P. Chen, L. Chen, J. Li and F. Dong, ACS Catal., 2022, 12, 9670–9678.
- 47 J. Di, G. Hao, G. Liu, J. Zhou, W. Jiang and Z. Liu, *Coord. Chem. Rev.*, 2023, 482, 215057.
- 48 S. Yin, Y. Zhou, Z. Liu, H. Wang, X. Zhao, Z. Zhu, Y. Yan and P. Huo, *Nat. Commun.*, 2024, **15**, 437.
- 49 V. P. Indrakanti, J. D. Kubicki and H. H. Schobert, *Energy Environ. Sci.*, 2009, **2**, 745–758.
- 50 Z. Xie, H. Luo, S. Xu, L. Li and W. Shi, *Adv. Funct. Mater.*, 2024, **34**, 2313886.
- 51 H.-N. Wang, Y.-H. Zou, H.-X. Sun, Y. Chen, S.-L. Li and Y.-Q. Lan, *Coord. Chem. Rev.*, 2021, **438**, 213906.
- 52 H. J. Son, C. Pac and S. O. Kang, Acc. Chem. Res., 2021, 54, 4530-4544.
- 53 S. Singh, A. Modak, K. K. Pant, A. Sinhamahapatra and P. Biswas, *ACS Appl. Nano Mater.*, 2021, 4, 8644–8667.
- 54 J. Albero, Y. Peng and H. García, *ACS Catal.*, 2020, **10**, 5734– 5749.
- 55 L. Wang, B. Zhu, J. Zhang, J. B. Ghasemi, M. Mousavi and J. Yu, *Matter*, 2022, 5, 4187–4211.
- 56 Y. Yin, W. Jing, F. Wang, Y. Liu and L. Guo, *Carbon*, 2023, **214**, 118317.
- 57 Q. Liu, J. Lin, H. Cheng, L. Wei and F. Wang, *Angew. Chem.*, *Int. Ed.*, 2023, **62**, e202312151.
- 58 X. Li and Q.-L. Zhu, EnergyChem, 2020, 2, 10033.
- 59 J. Liang, H. Yu, J. Shi, B. Li, L. Wu and M. Wang, *Adv. Mater.*, 2023, 35, 2209814.
- 60 W. Jiang, H. Loh, B. Q. L. Low, H. Zhu, J. Low, J. Z. X. Heng, K. Y. Tang, Z. Li, X. J. Loh, E. Ye and Y. Xiong, *Appl. Catal.*, *B*, 2023, **321**, 122079.
- 61 S. Wang, X. Han, Y. Zhang, N. Tian, T. Ma and H. Huang, *Small Struct.*, 2020, **2**, 2000061.
- 62 L. j. Guo, Y. j. Wang and T. He, *Chem. Rec.*, 2016, **16**, 1918– 1933.
- 63 F. Qi, Y. Pu, D. Wu, X. Tang and Q. Huang, *Chem. Rec.*, 2023, 23, e202300078.
- 64 W. A. Thompson, E. Sanchez Fernandez and M. M. Maroto-Valer, *ACS Sustain. Chem. Eng.*, 2020, **8**, 4677–4692.
- 65 J. Wang, H. Li, P. Gao, Y. Peng, S. Cao and M. Antonietti, *Chem. Eng. J.*, 2022, **443**, 136447.

- 66 E. Gong, S. Ali, C. B. Hiragond, H. S. Kim, N. S. Powar, D. Kim, H. Kim and S.-I. In, *Energy Environ. Sci.*, 2022, 15, 880–937.
- 67 S. Sun, M. Watanabe, J. Wu, Q. An and T. Ishihara, *J. Am. Chem. Soc.*, 2018, **140**, 6474–6482.
- 68 G. T. S. T. da Silva, A. E. Nogueira, J. A. Oliveira, J. A. Torres, O. F. Lopes and C. Ribeiro, *Appl. Catal.*, *B*, 2019, 242, 349– 357.
- 69 Y. Xiang, W. Dong, P. Wang, S. Wang, X. Ding, F. Ichihara, Z. Wang, Y. Wada, S. Jin, Y. Weng, H. Chen and J. Ye, *Appl. Catal.*, *B*, 2020, 274, 119096.
- 70 S. Hu, P. Qiao, X. Yi, Y. Lei, H. Hu, J. Ye and D. Wang, *Angew. Chem., Int. Ed.*, 2023, **62**, e202304585.
- 71 Y. Zheng, X. Shen, M. Lin, M. Zhu, B. Yang, J. Yan, Z. Zhuang and Y. Yu, *Small*, 2024, **20**, 2306836.
- 72 X. Shi, X. a. Dong, Y. Sun, S. Zhang and F. Dong, *Research*, 2022, **2022**, 9818792.
- 73 S. Cao, Y. Li, B. Zhu, M. Jaroniec and J. Yu, J. Catal., 2017, 349, 208–217.
- 74 M. Li, M. Wang, L. Zhu, Y. Li, Z. Yan, Z. Shen and X. Cao, *Appl. Catal.*, *B*, 2018, **231**, 269–276.
- 75 P. Wang, S. Guo, Q. P. Zhao, S. Y. Xu, H. Lv, T. B. Lu and Z. M. Zhang, Angew. Chem., Int. Ed., 2024, 63, e202312450.
- 76 S. Hu, P. Qiao, X. Liang, G. Ba, X. Zu, H. Hu, J. Ye and D. Wang, *Appl. Catal.*, B, 2024, 346, 123737.
- 77 L. Liu, H. Huang, Z. Chen, H. Yu, K. Wang, J. Huang, H. Yu and Y. Zhang, *Angew. Chem., Int. Ed.*, 2021, **60**, 18303– 18308.
- 78 S. Hussain, Y. Wang, L. Guo and T. He, J. Photochem. Photobiol., C, 2022, 52, 100538.
- 79 J. Jiang, X. Wang, Q. Xu, Z. Mei, L. Duan and H. Guo, *Appl. Catal.*, B, 2022, 316, 121679.
- 80 Q. V. Le, V.-H. Nguyen, T. D. Nguyen, A. Sharma, G. Rahman and D. L. T. Nguyen, *Chem. Eng. Sci.*, 2021, 237, 116547.
- 81 R. Long, Y. Li, Y. Liu, S. Chen, X. Zheng, C. Gao, C. He, N. Chen, Z. Qi, L. Song, J. Jiang, J. Zhu and Y. Xiong, *J. Am. Chem. Soc.*, 2017, **139**, 4486–4492.
- 82 S. A. Fors and C. A. Malapit, ACS Catal., 2023, 13, 4231-4249.
- 83 Q. Chen, Y. Zhang, E. You, Q. Jiang, X. Chen, Y. Wang, Z. Song, K. Chang, Z. Xie and Q. Kuang, *Small*, 2022, 18, 2204924.
- 84 L. Zou, Z. A. Chen, D. H. Si, S. L. Yang, W. Q. Gao, K. Wang,
  Y. B. Huang and R. Cao, *Angew. Chem., Int. Ed.*, 2023, 62, e202309820.
- 85 M. Xing, Y. Zhou, C. Dong, L. Cai, L. Zeng, B. Shen, L. Pan, C. Dong, Y. Chai, J. Zhang and Y. Yin, *Nano Lett.*, 2018, 18, 3384–3390.
- 86 T. Guo, X. Xu, Z. Xu, F. You, X. Fan, J. Liu and Z. Wang, *Adv. Mater.*, 2024, 202402071.
- 87 Y.-H. Chen, M.-Y. Qi, Y.-H. Li, Z.-R. Tang, T. Wang, J. Gong and Y.-J. Xu, *Cell Rep. Phys. Sci.*, 2021, 2, 10031.
- 88 K. Li, T. Peng, Z. Ying, S. Song and J. Zhang, *Appl. Catal.*, *B*, 2016, **180**, 130–138.
- 89 F. Q. Yan, X. Y. Dong, Y. M. Wang, Q. Y. Wang, S. Wang and S. Q. Zang, *Adv. Sci.*, 2024, 2401508.

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- 90 Z. Wang, J. Zhu, X. Zu, Y. Wu, S. Shang, P. Ling, P. Qiao, C. Liu, J. Hu, Y. Pan, J. Zhu, Y. Sun and Y. Xie, *Angew. Chem., Int. Ed.*, 2022, **61**, e202203249.
- 91 Z. Sun, T. Liao, Y. Dou, S. M. Hwang, M.-S. Park, L. Jiang, J. H. Kim and S. X. Dou, *Nat. Commun.*, 2014, 5, 3813.
- 92 S. Yue, Z. Zhao, T. Zhang, P. Wang and S. Zhan, *Mater. Today Energy*, 2024, 40, 101482.
- 93 Y. Wang, Z. Hu, W. Wang, Y. Li, H. He, L. Deng, Y. Zhang, J. Huang, N. Zhao, G. Yu and Y.-N. Liu, *Appl. Catal.*, *B*, 2023, 327, 122419.
- 94 J. Hou, M. Yang, Q. Dou, Q. Chen, X. Wang, C. Hu and R. Paul, *Chem. Eng. J.*, 2022, 450, 122319.
- 95 X. Jia, C. Hu, H. Sun, J. Cao, H. Lin, X. Li and S. Chen, *Appl. Catal.*, *B*, 2023, **324**, 122232.
- 96 X. Xiong, C. Mao, Z. Yang, Q. Zhang, G. I. N. Waterhouse, L. Gu and T. Zhang, *Adv. Energy Mater.*, 2020, **10**, 2002928.
- 97 Q. Chen, W. Mo, G. Yang, S. Zhong, H. Lin, J. Chen and S. Bai, Small, 2021, 17, 2101105.
- 98 D. Hong, T. Kawanishi, Y. Tsukakoshi, H. Kotani, T. Ishizuka and T. Kojima, *J. Am. Chem. Soc.*, 2019, 141, 20309–20317.
- 99 D. Kim, S. Bhattacharjee, E. Lam, C. Casadevall, S. Rodríguez-Jiménez and E. Reisner, *Small*, 2024, 202400057.
- 100 K. Li, S. Zhang, Y. Li, J. Fan and K. Lv, *Chin. J. Catal.*, 2021, 42, 3–14.
- 101 Q. Lin, J. Zhao, P. Zhang, S. Wang, Y. Wang, Z. Zhang, N. Wen, Z. Ding, R. Yuan, X. Wang and J. Long, *Carbon Energy*, 2024, 6, e435.
- 102 L. Gao, Y. Li, M. Xiao, S. Wang, G. Fu and L. Wang, 2D Mater., 2017, 4, 025031.
- 103 Y. Zhang, Z. Xu, Q. Wang, W. Hao, X. Zhai, X. Fei, X. Huang and Y. Bi, *Appl. Catal.*, *B*, 2021, **299**, 120679.
- 104 C. Chen, Y. Li and P. Yang, Joule, 2021, 5, 737-742.
- 105 H. Wang, H. Feng, Y. Bao, J. Wu, X. Qu, X. Zhang, J. Liu and
   H. Wang, *Catal. Sci. Technol.*, 2023, 13, 4996–5004.
- 106 P. Hu, G. Liang, B. Zhu, W. Macyk, J. Yu and F. Xu, ACS Catal., 2023, 13, 12623–12633.
- 107 F. Xu, Z. Li, R. Zhu, Y. Chu, Z. Pan, S. Xia, J. Fu, Z. Xiao, X. Ji,
   M. Liu and B. Weng, *Appl. Catal.*, *B*, 2022, **316**, 121615.
- 108 K. Yang, Z. Yang, C. Zhang, Y. Gu, J. Wei, Z. Li, C. Ma, X. Yang, K. Song, Y. Li, Q. Fang and J. Zhou, *Chem. Eng. J.*, 2021, **418**, 129344.
- 109 L. Yuan, M. Y. Qi, Z. R. Tang and Y. J. Xu, *Angew. Chem., Int. Ed.*, 2021, **60**, 21150–21172.
- 110 L. Liu and Y. Li, Aerosol Air Qual. Res., 2014, 14, 453-469.
- 111 Y. Wang, E. Chen and J. Tang, *ACS Catal.*, 2022, **12**, 7300–7316.
- 112 K. Li, B. Peng and T. Peng, ACS Catal., 2016, 6, 7485-7527.
- 113 Y. Zhou, Z. Wang, L. Huang, S. Zaman, K. Lei, T. Yue, Z. a. Li, B. You and B. Y. Xia, *Adv. Energy Mater.*, 2021, 11, 2003159.
- 114 J. Ran, M. Jaroniec and S. Z. Qiao, *Adv. Mater.*, 2018, **30**, 1704649.
- 115 S. Nahar, M. F. M. Zain, A. A. H. Kadhum, H. A. Hasan and M. R. Hasan, *Materials*, 2017, **10**, 629.

- 116 B. T. Barrocas, N. Ambrozova and K. Koci, *Materials*, 2022, **15**, 967.
- 117 L. Liao, G. Xie, X. Xie and N. Zhang, J. Phys. Chem. C, 2023, 127, 2766–2781.
- 118 L. Wang, W. Chen, D. Zhang, Y. Du, R. Amal, S. Qiao, J. Wu and Z. Yin, *Chem. Soc. Rev.*, 2019, **48**, 5310–5349.
- S. Wang, J. Wang, Y. Wang, X. Sui, S. Wu, W. Dai, Z. Zhang,
   Z. Ding and J. Long, ACS Catal., 2024, 14, 10760–10788.
- 120 Y. Wu, Q. Hu, Q. Chen, X. Jiao and Y. Xie, *Acc. Chem. Res.*, 2023, **56**, 2500–2513.
- 121 S. Cheng, Z. Sun, K. H. Lim, T. Z. H. Gani, T. Zhang, Y. Wang, H. Yin, K. Liu, H. Guo, T. Du, L. Liu, G. K. Li, Z. Yin and S. Kawi, *Adv. Energy Mater.*, 2022, **12**, 2200389.
- 122 H. Wang, L. Zhang, Z. Chen, J. Hu, S. Li, Z. Wang, J. Liu and X. Wang, *Chem. Soc. Rev.*, 2014, **43**, 5234–5244.
- 123 K. Zhong, P. Sun and H. Xu, Small, 2024, 202310677.
- 124 L. Ran, J. Hou, S. Cao, Z. Li, Y. Zhang, Y. Wu, B. Zhang, P. Zhai and L. Sun, *Sol. RRL*, 2020, 4, 1900487.
- 125 P. Raizada, V. Soni, A. Kumar, P. Singh, A. A. Parwaz Khan, A. M. Asiri, V. K. Thakur and V.-H. Nguyen, *Journal of Materiomics*, 2021, 7, 388–418.
- 126 E. Pastor, M. Sachs, S. Selim, J. R. Durrant, A. A. Bakulin and A. Walsh, *Nat. Rev. Mater.*, 2022, 7, 503–521.
- 127 Y. Ji and Y. Luo, J. Am. Chem. Soc., 2016, 138, 15896-15902.
- 128 M. Zhang, D. Zhang, X. Jing, B. Xu and C. Duan, Angew. Chem., Int. Ed., 2024, 63, e202402755.
- 129 X. Zu, Y. Zhao, X. Li, R. Chen, W. Shao, Z. Wang, J. Hu, J. Zhu, Y. Pan, Y. Sun and Y. Xie, *Angew. Chem., Int. Ed.*, 2021, **60**, 13840–13846.
- 130 S. Vinoth, W.-J. Ong and A. Pandikumar, *Coord. Chem. Rev.*, 2022, **464**, 214541.
- 131 Y. Wang, Q. Lai, Y. He and M. Fan, *Catal. Commun.*, 2018, 108, 98–102.
- 132 Z. Geng, X. Kong, W. Chen, H. Su, Y. Liu, F. Cai, G. Wang and J. Zeng, *Angew. Chem., Int. Ed.*, 2018, **57**, 6054–6059.
- 133 Z. Miao, Q. Wang, Y. Zhang, L. Meng and X. Wang, *Appl. Catal.*, B, 2022, 301, 120802.
- 134 W. Tu, Y. Zhou and Z. Zou, *Adv. Mater.*, 2014, **26**, 4607–4626.
- 135 L. Liu, C. Zhao and Y. Li, *J. Phys. Chem. C*, 2012, **116**, 7904–7912.
- 136 D. Liu, C. Wang, Y. Yu, B.-H. Zhao, W. Wang, Y. Du and B. Zhang, *Chem*, 2019, 5, 376–389.
- 137 H. Lai, X. Zeng, T. Song, S. Yin, B. Long, A. Ali and G.-J. Deng, *J. Colloid Interface Sci.*, 2022, **608**, 1792–1801.
- 138 J. Di, C. Zhu, M. Ji, M. Duan, R. Long, C. Yan, K. Gu, J. Xiong, Y. She, J. Xia, H. Li and Z. Liu, *Angew. Chem.*, *Int. Ed.*, 2018, 57, 14847–14851.
- 139 F. Chen, Z. Ma, L. Ye, T. Ma, T. Zhang, Y. Zhang and H. Huang, *Adv. Mater.*, 2020, **32**, 1908350.
- 140 H. Tan, Z. Zhao, W.-b. Zhu, E. N. Coker, B. Li, M. Zheng,
  W. Yu, H. Fan and Z. Sun, *ACS Appl. Mater. Interfaces*,
  2014, 6, 19184–19190.
- 141 M. Xing, J. Zhang, F. Chen and B. Tian, *Chem. Commun.*, 2011, 47, 4947–4949.

- 142 L. Li, X. Feng, Y. Nie, S. Chen, F. Shi, K. Xiong, W. Ding,
  X. Qi, J. Hu, Z. Wei, L.-J. Wan and M. Xia, ACS Catal.,
  2015, 5, 4825–4832.
- 143 H. Xu, C. Yan, R. Li, L. Song and S. Ouyang, *Nanotechnology*, 2022, **33**, 085401.
- 144 Z. Shi, D. Shi, L. Zhang and Y. Cao, *J. Power Sources*, 2022, 535, 231421.
- 145 Y. Zhou, Q. Zhang, X. Shi, Q. Song, C. Zhou and D. Jiang, *J. Colloid Interface Sci.*, 2022, **608**, 2809–2819.
- 146 H. Zeng, M. h. Oubla, X. Zhong, N. Alonso-Vante, F. Du,Y. Xie, Y. Huang and J. Ma, *Appl. Catal.*, *B*, 2021, 281, 119535.
- 147 Z. Li, K. Zhu, J. Li and X. Wang, *CrystEngComm*, 2014, 16, 10112–10122.
- 148 Y. Zhang, Y. Li, H. Yu, K. Yu and H. Yu, *J. Mater. Sci. Technol.*, 2022, **106**, 139–146.
- 149 J. Wu, X. Li, W. Shi, P. Ling, Y. Sun, X. Jiao, S. Gao, L. Liang, J. Xu, W. Yan, C. Wang and Y. Xie, *Angew. Chem., Int. Ed.*, 2018, 57, 8719–8723.
- 150 S. Yin, X. Zhao, E. Jiang, Y. Yan, P. Zhou and P. Huo, *Energy Environ. Sci.*, 2022, **15**, 1556–1562.
- 151 W. Gao, S. Li, H. He, X. Li, Z. Cheng, Y. Yang, J. Wang, Q. Shen, X. Wang, Y. Xiong, Y. Zhou and Z. Zou, *Nat. Commun.*, 2021, **12**, 4717.
- 152 A. Quindimil, U. De-La-Torre, B. Pereda-Ayo, A. Davó-Quiñonero, E. Bailón-García, D. Lozano-Castelló, J. A. González-Marcos, A. Bueno-López and J. R. González-Velasco, *Catal. Today*, 2020, **356**, 419–432.
- 153 J. A. H. Dreyer, P. Li, L. Zhang, G. K. Beh, R. Zhang,
  P. H. L. Sit and W. Y. Teoh, *Appl. Catal.*, *B*, 2017, 219, 715–726.
- 154 X. Chen and S. S. Mao, Chem. Rev., 2007, 107, 2891-2959.
- 155 B. Kraeutler and A. J. Bard, *J. Am. Chem. Soc.*, 1978, **100**, 4317–4318.
- 156 W.-N. Wang, W.-J. An, B. Ramalingam, S. Mukherjee, D. M. Niedzwiedzki, S. Gangopadhyay and P. Biswas, *Nano Res.*, 2013, 58, 305.
- 157 S. Xie, Y. Wang, Q. Zhang, W. Fan, W. Deng and Y. Wang, *Chem. Commun.*, 2013, **49**, 2451–2453.
- 158 Z. Xiong, Z. Lei, C.-C. Kuang, X. Chen, B. Gong, Y. Zhao, J. Zhang, C. Zheng and J. C. S. Wu, *Appl. Catal.*, *B*, 2017, 202, 695–703.
- 159 Z. Ma, P. Li, L. Ye, L. Wang, H. Xie and Y. Zhou, *Catal. Sci. Technol.*, 2018, **8**, 5129–5132.
- 160 Ş. Neaţu, J. A. Maciá-Agulló, P. Concepción and H. Garcia, J. Am. Chem. Soc., 2014, 136, 15969–15976.
- 161 C. Kim, F. Dionigi, V. Beermann, X. Wang, T. Moller and P. Strasser, *Adv. Mater.*, 2019, **31**, e1805617.
- 162 Y. Sugano, Y. Shiraishi, D. Tsukamoto, S. Ichikawa, S. Tanaka and T. Hirai, *Angew. Chem., Int. Ed.*, 2013, 52, 5295–5299.
- 163 S. Wang and X. Wang, Appl. Catal., B, 2015, 162, 494-500.
- 164 D. Li, M. Kassymova, X. Cai, S.-Q. Zang and H.-L. Jiang, *Coord. Chem. Rev.*, 2020, 412, 213262.
- 165 X. Meng, S. Ouyang, T. Kako, P. Li, Q. Yu, T. Wang and J. Ye, *Chem. Commun.*, 2014, **50**, 11517–11519.

- 166 X. Feng, F. Pan, H. Zhao, W. Deng, P. Zhang, H.-C. Zhou and Y. Li, *Appl. Catal.*, *B*, 2018, 238, 274–283.
- 167 T. Ghanbari, F. Abnisa and W. M. A. Wan Daud, *Sci. Total Environ.*, 2020, **707**, 135090.
- 168 Z. Wang, K. Teramura, Z. Huang, S. Hosokawa, Y. Sakata and T. Tanaka, *Catal. Sci. Technol.*, 2016, **6**, 1025–1032.
- 169 Y. Huang, K. Li, J. Zhou, J. Guan, F. Zhu, K. Wang, M. Liu, W. Chen and N. Li, *Chem. Eng. J.*, 2022, **439**, 135744.
- 170 G. Liu, K. Du, S. Haussener and K. Wang, *ChemSusChem*, 2016, **9**, 2878–2904.
- 171 Z. Sun, H. Wang, Z. Wu and L. Wang, *Catal. Today*, 2018, **300**, 160–172.
- 172 X. Wang, S. Blechert and M. Antonietti, *ACS Catal.*, 2012, 2, 1596–1606.
- 173 W. J. Ong, L. L. Tan, Y. H. Ng, S. T. Yong and S. P. Chai, *Chem. Rev.*, 2016, **116**, 7159–7329.
- 174 J. Fu, J. Yu, C. Jiang and B. Cheng, *Adv. Energy Mater.*, 2018, 8, 1701503.
- 175 X. Dong and F. Cheng, J. Mater. Chem. A, 2015, 3, 23642– 23652.
- 176 X. Huang, W. Gu, Y. Ma, D. Liu, N. Ding, L. Zhou, J. Lei,
  L. Wang and J. Zhang, *Res. Chem. Intermed.*, 2020, 46, 5133–5164.
- 177 Q. Lin, L. Li, S. Liang, M. Liu, J. Bi and L. Wu, *Appl. Catal., B*, 2015, **163**, 135–142.
- 178 X. Ma, Y. Lv, J. Xu, Y. Liu, R. Zhang and Y. Zhu, *J. Phys. Chem. C*, 2012, **116**, 23485–23493.
- 179 F. Almomani, R. Bhosale, M. Khraisheh, A. Kumar and M. Tawalbeh, *Appl. Surf. Sci.*, 2019, **483**, 363–372.
- 180 H. Guo, Z. Shu, D. Chen, Y. Tan, J. Zhou, F. Meng and T. Li, *Chem. Phys.*, 2020, **533**, 110714.
- 181 S. Patnaik, D. P. Sahoo and K. Parida, *Carbon*, 2021, **172**, 682–711.
- 182 M. Arumugam, M. Tahir and P. Praserthdam, *Chemosphere*, 2022, **286**, 131765.
- 183 T. Xiong, W. Cen, Y. Zhang and F. Dong, *ACS Catal.*, 2016, 6, 2462–2472.
- 184 H.-Z. Wu, L.-M. Liu and S.-J. Zhao, *Appl. Surf. Sci.*, 2015, **358**, 363–369.
- 185 B. Zhu, J. Zhang, C. Jiang, B. Cheng and J. Yu, *Appl. Catal.*, B, 2017, 207, 27–34.
- 186 X. Ma, Y. Lv, J. Xu, Y. Liu, R. Zhang and Y. Zhu, *J. Phys. Chem. C*, 2012, **116**, 23485–23493.
- 187 G. Dong, K. Zhao and L. Zhang, Chem. Commun., 2012, 48, 6178–6180.
- 188 J. Fang, H. Fan, M. Li and C. Long, J. Mater. Chem. A, 2015, 3, 13819–13826.
- 189 B. Liu, M. Liu, L. Tian, F. Guo, Y. Xia, T. Wang, W. Hu and R. Guan, *Appl. Surf. Sci.*, 2021, **540**, 148411.
- 190 H. Zhang, Y. Tang, Z. Liu, Z. Zhu, X. Tang and Y. Wang, *Chem. Phys. Lett.*, 2020, **751**, 137467.
- 191 Y. Wang, Y. Xu, Y. Wang, H. Qin, X. Li, Y. Zuo, S. Kang and L. Cui, *Catal. Commun.*, 2016, 74, 75–79.
- 192 P. Babu, S. Mohanty, B. Naik and K. Parida, *ACS Appl. Energy Mater.*, 2018, **1**, 5936–5947.
- 193 K. Wang, J. Fu and Y. Zheng, *Appl. Catal.*, *B*, 2019, **254**, 270–282.

- 194 X. Dong, S. Zhang, H. Wu, Z. Kang and L. Wang, *RSC Adv.*, 2019, **9**, 28894–28901.
- 195 J. Low, J. Yu, M. Jaroniec, S. Wageh and A. A. Al-Ghamdi, *Adv. Mater.*, 2017, **29**, 1601694.
- 196 H. J. Yun, H. Lee, N. D. Kim, D. M. Lee, S. Yu and J. Yi, *ACS Nano*, 2011, 5, 4084–4090.
- 197 A. Iwase, Y. H. Ng, Y. Ishiguro, A. Kudo and R. Amal, *J. Am. Chem. Soc.*, 2011, **133**, 11054–11057.
- 198 A. Alhebshi, E. Sharaf Aldeen, R. S. Mim, B. Tahir and M. Tahir, *Int. J. Energy Res.*, 2021, **46**, 5523–5584.
- 199 A. J. Bard, J. Photochem., 1979, 10, 59-75.
- 200 H. Tada, T. Mitsui, T. Kiyonaga, T. Akita and K. Tanaka, *Nat. Mater.*, 2006, 5, 782–786.
- 201 J. Yu, S. Wang, J. Low and W. Xiao, *Phys. Chem. Chem. Phys.*, 2013, **15**, 16883–16890.
- 202 J. Liu, B. Cheng and J. Yu, Phys. Chem. Chem. Phys., 2016, 18, 31175–31183.
- 203 Y. Wang, Z. Zhang, L. Zhang, Z. Luo, J. Shen, H. Lin, J. Long, J. C. S. Wu, X. Fu, X. Wang and C. Li, *J. Am. Chem. Soc.*, 2018, 140, 14595–14598.
- 204 Y. Wang, X. Shang, J. Shen, Z. Zhang, D. Wang, J. Lin, J. C. S. Wu, X. Fu, X. Wang and C. Li, *Nat. Commun.*, 2020, **11**, 3043.
- 205 J. Fu, Q. Xu, J. Low, C. Jiang and J. Yu, *Appl. Catal., B*, 2019, **243**, 556–565.
- 206 Q. Xu, L. Zhang, B. Cheng, J. Fan and J. Yu, *Chem*, 2020, **6**, 1543–1559.
- 207 Y. Yang, D. Zhang, J. Fan, Y. Liao and Q. Xiang, *Sol. RRL*, 2021, 5, 2000351.
- 208 X. Deng, J. Zhang, K. Qi, G. Liang, F. Xu and J. Yu, *Nat. Commun.*, 2024, **15**, 4087.
- 209 Y. Sun, K. Lai, N. Li, Y. Gao and L. Ge, *Appl. Catal., B*, 2024, 357, 124302.
- 210 B. Qiao, A. Wang, X. Yang, L. F. Allard, Z. Jiang, Y. Cui, J. Liu, J. Li and T. Zhang, *Nat. Chem.*, 2011, 3, 634–641.
- 211 J. Xing, J. F. Chen, Y. H. Li, W. T. Yuan, Y. Zhou, L. R. Zheng,
  H. F. Wang, P. Hu, Y. Wang, H. J. Zhao, Y. Wang and
  H. G. Yang, *Chem.-Eur. J.*, 2014, 20, 2138–2144.
- 212 Z. Wang, X. Hao, S. Gerhold, P. Mares, M. Wagner, R. Bliem, K. Schulte, M. Schmid, C. Franchini and U. Diebold, *J. Phys. Chem. C*, 2014, **118**, 19904–19909.
- 213 J. Lin, A. Wang, B. Qiao, X. Liu, X. Yang, X. Wang, J. Liang, J. Li, J. Liu and T. Zhang, *J. Am. Chem. Soc.*, 2013, 135, 15314–15317.
- 214 J. C. Matsubu, V. N. Yang and P. Christopher, *J. Am. Chem. Soc.*, 2015, **137**, 3076–3084.
- 215 J. Jones, H. Xiong, A. T. DeLaRiva, E. J. Peterson, H. Pham, S. R. Challa, G. Qi, S. Oh, M. H. Wiebenga, X. I. Pereira Hernández, Y. Wang and A. K. Datye, *Science*, 2016, 353, 150–154.
- 216 M. Moliner, J. E. Gabay, C. E. Kliewer, R. T. Carr, J. Guzman, G. L. Casty, P. Serna and A. Corma, *J. Am. Chem. Soc.*, 2016, 138, 15743–15750.
- 217 S. Wei, A. Li, J.-C. Liu, Z. Li, W. Chen, Y. Gong, Q. Zhang, W.-C. Cheong, Y. Wang, L. Zheng, H. Xiao, C. Chen, D. Wang, Q. Peng, L. Gu, X. Han, J. Li and Y. Li, *Nat. Nanotechnol.*, 2018, 13, 856–861.

- 218 W. Huang, S. Zhang, Y. Tang, Y. Li, L. Nguyen, Y. Li, J. Shan, D. Xiao, R. Gagne, A. I. Frenkel and F. Tao, *Angew. Chem.*, *Int. Ed.*, 2016, **55**, 13441–13445.
- 219 C. Wang, X.-K. Gu, H. Yan, Y. Lin, J. Li, D. Liu, W.-X. Li and J. Lu, ACS Catal., 2017, 7, 887–891.
- 220 N. Cheng, S. Stambula, D. Wang, M. N. Banis, J. Liu,
  A. Riese, B. Xiao, R. Li, T.-K. Sham, L.-M. Liu,
  G. A. Botton and X. Sun, *Nat. Commun.*, 2016, 7, 13638.
- 221 B. J. O'Neill, D. H. K. Jackson, J. Lee, C. Canlas, P. C. Stair, C. L. Marshall, J. W. Elam, T. F. Kuech, J. A. Dumesic and G. W. Huber, *ACS Catal.*, 2015, 5, 1804–1825.
- 222 J. S. King, A. Wittstock, J. Biener, S. O. Kucheyev, Y. M. Wang, T. F. Baumann, S. K. Giri, A. V. Hamza, M. Baeumer and S. F. Bent, *Nano Lett.*, 2008, 8, 2405–2409.
- 223 C. Liu, C.-C. Wang, C.-C. Kei, Y.-C. Hsueh and T.-P. Perng, *Small*, 2009, **5**, 1535–1538.
- 224 S. Sun, G. Zhang, N. Gauquelin, N. Chen, J. Zhou, S. Yang, W. Chen, X. Meng, D. Geng, M. N. Banis, R. Li, S. Ye, S. Knights, G. A. Botton, T.-K. Sham and X. Sun, *Sci. Rep.*, 2013, 3, 1775.
- 225 Y. Cao, S. Chen, Q. Luo, H. Yan, Y. Lin, W. Liu, L. Cao, J. Lu, J. Yang, T. Yao and S. Wei, *Angew. Chem., Int. Ed.*, 2017, 56, 12191–12196.
- 226 G. E. Johnson, C. Wang, T. Priest and J. Laskin, *Anal. Chem.*, 2011, **83**, 8069–8072.
- 227 G. E. Johnson, T. Priest and J. Laskin, *ACS Nano*, 2012, 6, 573–582.
- 228 Z. Zhang, Y. Zhu, H. Asakura, B. Zhang, J. Zhang, M. Zhou, Y. Han, T. Tanaka, A. Wang, T. Zhang and N. Yan, *Nat. Commun.*, 2017, 8, 16100.
- M. Kottwitz, Y. Li, R. M. Palomino, Z. Liu, G. Wang, Q. Wu,
  J. Huang, J. Timoshenko, S. D. Senanayake,
  M. Balasubramanian, D. Lu, R. G. Nuzzo and
  A. I. Frenkel, ACS Catal., 2019, 9, 8738–8748.
- 230 K. Mou, Z. Chen, X. Zhang, M. Jiao, X. Zhang, X. Ge, W. Zhang and L. Liu, *Small*, 2019, **15**, 1903668.
- 231 M. J. Islam, M. Granollers Mesa, A. Osatiashtiani, M. J. Taylor, J. C. Manayil, C. M. A. Parlett, M. A. Isaacs and G. Kyriakou, *Appl. Catal.*, *B*, 2020, **273**, 119062.
- 232 H. Wang, J. Shen, J. Huang, T. Xu, J. Zhu, Y. Zhu and C. Li, *Nanoscale*, 2017, **9**, 16817–16825.
- 233 X. Wang, X.-P. Fu, W.-Z. Yu, C. Ma, C.-J. Jia and R. Si, *Inorg. Chem. Front.*, 2017, **4**, 2059–2067.
- 234 Z. Bo, L. R. McCullough, S. Dull, M. A. Ardagh, J. Wang and J. Notestein, *J. Chem. Phys.*, 2019, **151**, 214703.
- 235 S. L. Wegener, T. J. Marks and P. C. Stair, *Acc. Chem. Res.*, 2012, **45**, 206–214.
- 236 T. W. Hansen, A. T. DeLaRiva, S. R. Challa and A. K. Datye, *Acc. Chem. Res.*, 2013, **46**, 1720–1730.
- 237 S. Zhang, L. Nguyen, J.-X. Liang, J. Shan, J. Liu, A. I. Frenkel,
  A. Patlolla, W. Huang, J. Li and F. Tao, *Nat. Commun.*, 2015,
  6, 7938.
- 238 X.-L. Ma, J.-C. Liu, H. Xiao and J. Li, *J. Am. Chem. Soc.*, 2018, **140**, 46–49.
- 239 J. Wan, W. Chen, C. Jia, L. Zheng, J. Dong, X. Zheng,
  Y. Wang, W. Yan, C. Chen, Q. Peng, D. Wang and Y. Li, *Adv. Mater.*, 2018, 30, 1705369.

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- 240 G. Liu, A. W. Robertson, M. M.-J. Li, W. C. H. Kuo, M. T. Darby, M. H. Muhieddine, Y.-C. Lin, K. Suenaga, M. Stamatakis, J. H. Warner and S. C. E. Tsang, *Nat. Chem.*, 2017, 9, 810–816.
- L. Nie, D. Mei, H. Xiong, B. Peng, Z. Ren, X. I. P. Hernandez,
  A. DeLaRiva, M. Wang, M. H. Engelhard, L. Kovarik,
  A. K. Datye and Y. Wang, *Science*, 2017, 358, 1419–1423.
- 242 Q. Fu, H. Saltsburg and M. Flytzani-Stephanopoulos, *Science*, 2003, **301**, 935–938.
- 243 R. Lang, W. Xi, J.-C. Liu, Y.-T. Cui, T. Li, A. F. Lee, F. Chen,
  Y. Chen, L. Li, L. Li, J. Lin, S. Miao, X. Liu, A.-Q. Wang,
  X. Wang, J. Luo, B. Qiao, J. Li and T. Zhang, *Nat. Commun.*, 2019, 10, 234.
- 244 S. Ji, Y. Qu, T. Wang, Y. Chen, G. Wang, X. Li, J. Dong, Q. Chen, W. Zhang, Z. Zhang, S. Liang, R. Yu, Y. Wang, D. Wang and Y. Li, *Angew. Chem., Int. Ed.*, 2020, 59, 10651–10657.
- 245 H. Li, Q. Song, S. Wan, C. W. Tung, C. Liu, Y. Pan, G. Luo,
  H. M. Chen, S. Cao, J. Yu and L. Zhang, *Small*, 2023, 19, 2301711.
- 246 Y. Li, B. Li, D. Zhang, L. Cheng and Q. Xiang, *ACS Nano*, 2020, 14, 10552–10561.
- 247 M. Ma, Z. Huang, D. E. Doronkin, W. Fa, Z. Rao, Y. Zou, R. Wang, Y. Zhong, Y. Cao, R. Zhang and Y. Zhou, *Appl. Catal.*, *B*, 2022, **300**, 120695.
- 248 G. Wang, C.-T. He, R. Huang, J. Mao, D. Wang and Y. Li, *J. Am. Chem. Soc.*, 2020, **142**, 19339–19345.
- 249 H. Yin, F. Dong, D. Wang and J. Li, *ACS Catal.*, 2022, **12**, 14096–14105.
- 250 D. Zeng, H. Wang, X. Zhu, H. Cao, Y. Zhou, W. Wang, L. Zhang and W. Wang, *Chem. Eng. J.*, 2023, **451**, 138801.
- 251 X. Li, W. Bi, L. Zhang, S. Tao, W. Chu, Q. Zhang, Y. Luo, C. Wu and Y. Xie, *Adv. Mater.*, 2016, 28, 2427–2431.
- 252 G. Vilé, D. Albani, M. Nachtegaal, Z. Chen, D. Dontsova, M. Antonietti, N. López and J. Pérez-Ramírez, Angew. Chem., Int. Ed., 2015, 54, 10987.
- 253 Z. Chen, Q. Zhang, W. Chen, J. Dong, H. Yao, X. Zhang,
  X. Tong, D. Wang, Q. Peng, C. Chen, W. He and Y. Li, *Adv. Mater.*, 2018, 30, 1704720.
- 254 L. Cheng, H. Yin, C. Cai, J. Fan and Q. Xiang, *Small*, 2020, **16**, 2002411.
- 255 X. Shi, Y. Huang, Y. Bo, D. Duan, Z. Wang, J. Cao, G. Zhu,
  W. Ho, L. Wang, T. Huang and Y. Xiong, *Angew. Chem.*, *Int. Ed.*, 2022, 61, e202203063.
- 256 W. Xie, K. Li, X.-H. Liu, X. Zhang and H. Huang, *Adv. Mater.*, 2023, **35**, 2208132.
- 257 M. Ma, Z. Huang, D. E. Doronkin, W. Fa, Z. Rao, Y. Zou, R. Wang, Y. Zhong, Y. Cao, R. Zhang and Y. Zhou, *Appl. Catal.*, *B*, 2022, **300**, 120695.
- 258 X. Li and Z. Xiang, Nat. Commun., 2022, 13, 57.
- 259 J. Gu, C.-S. Hsu, L. Bai, H. M. Chen and X. Hu, *Science*, 2019, 364, 1091–1094.
- 260 D. Grammatico, A. J. Bagnall, L. Riccardi, M. Fontecave, B.-L. Su and L. Billon, *Angew. Chem., Int. Ed.*, 2022, 61, e202206399.
- 261 Y. Liang and R. Anwander, *Dalton Trans.*, 2013, **42**, 12521–12545.

- 262 P. Serna and B. C. Gates, Acc. Chem. Res., 2014, 47, 2612– 2620.
- 263 C. Copéret, M. Chabanas, R. Petroff Saint-Arroman and J.-M. Basset, *Angew. Chem., Int. Ed.*, 2003, 42, 156–181.
- 264 M. Rimoldi and A. Mezzetti, *Catal. Sci. Technol.*, 2014, 4, 2724–2740.
- 265 J.-M. Basset, F. Lefebvre and C. Santini, *Coord. Chem. Rev.*, 1998, **178–180**, 1703–1723.
- 266 S. R. Docherty and C. Copéret, *J. Am. Chem. Soc.*, 2021, **143**, 6767–6780.
- 267 H. Ma, Y. Wang, Y. Xu and W. Shi, *Appl. Catal., B*, 2024, **349**, 123857.
- 268 D. Li, M. Kassymova, X. Cai, S.-Q. Zang and H.-L. Jiang, *Coord. Chem. Rev.*, 2020, **412**, 213262.
- 269 J. Wang, X.-F. Cui, B. Liu, S. Ling, R. Long and Y.-J. Xiong, J. Am. Chem. Soc., 2018, 140, 16514–16520.
- 270 P. Du, Y.-N. Bo, H. Li, R. Liu, C.-M. Wang, H.-J. Liu, T. K. D. Liu, Z. Lu, C. Gao and Y.-J. Xiong, *Appl. Catal.*, *B*, 2023, **330**, 122667.
- 271 H. Wang, Z. Zhan, Q. Huang, S.-Q. Li, X.-X. Yi, Q. Tang and J.-Y. Wang, *Small*, 2022, **18**, 2105083.
- 272 Y. Xiao and T.-S. Chung, *Energy Environ. Sci.*, 2011, 4, 201–208.
- 273 Z. Zhang, H. Gao, H. Wu, Y. Qian, L. Chen and J. Chen, *ACS Appl. Nano Mater.*, 2018, **1**, 6463–6476.
- 274 Q. Peng, B. Gong and G. N. Parsons, *Nanotechnology*, 2011, 22, 155601.
- 275 V. Rozyyev, J. G. Murphy, E. Barry, A. U. Mane, S. J. Sibener and J. W. Elam, *Appl. Surf. Sci.*, 2021, **562**, 149996.
- 276 C. Coperet, A. Comas-Vives, M. P. Conley, D. P. Estes,A. Fedorov, V. Mougel, H. Nagae, F. Nunez-Zarur andP. A. Zhizhko, *Chem. Rev.*, 2016, **116**, 323–421.
- 277 W. Yi, W. Yuan, Y. Meng, S. Zou, Y. Zhou, W. Hong, J. Che,
  M. Hao, B. Ye, L. Xiao, Y. Wang, H. Kobayashi and J. Fan,
  ACS Appl. Mater. Interfaces, 2017, 9, 31853–31860.
- 278 T. W. Goh, C.-K. Tsung and W. Huang, ACS Appl. Mater. Interfaces, 2019, 11, 23254–23260.
- 279 C. Dong, Y. Li, D. Cheng, M. Zhang, J. Liu, Y.-G. Wang, D. Xiao and D. Ma, ACS Catal., 2020, 10, 11011–11045.
- 280 K. Lian, Y. Yue, J.-M. Basset, X. Liu, L. Chen, C. Ozsoy-Keskinbora, X. Bao and H. Zhu, *J. Phys. Chem. C*, 2022, 126, 16663–16671.
- 281 K. P. Kuhl, E. R. Cave, D. N. Abram and T. F. Jaramillo, *Energy Environ. Sci.*, 2012, 5, 7050–7059.
- 282 X. Wang, Z. Jiang, H. Chen, K. Wang and X. Wang, *J. Alloys Compd.*, 2022, **896**, 163030.
- 283 Z. Miao, Y. Zhang, N. Wang, P. Xu and X. Wang, J. Colloid Interface Sci., 2022, 620, 407–418.
- 284 H. Huang, J. Zhao, B. Weng, F. Lai, M. Zhang, J. Hofkens, M. B. J. Roeffaers, J. A. Steele and J. Long, *Angew. Chem.*, *Int. Ed.*, 2022, **61**, e202204563.
- 285 Q. Ge, Y. Liu, K. Li, L. Xie, X. Ruan, W. Wang, L. Wang, T. Wang, W. You and L. Zhang, *Angew. Chem., Int. Ed.*, 2023, **62**, e202304189.
- 286 R. Das, S. Chakraborty and S. C. Peter, *ACS Energy Lett.*, 2021, **6**, 3270–3274.

- 287 S. Yoshino, A. Iwase, Y. Yamaguchi, T. M. Suzuki, T. Morikawa and A. Kudo, *J. Am. Chem. Soc.*, 2022, 144, 2323–2332.
- 288 R. Das, S. Sarkar, R. Kumar, S. D. Ramarao, A. Cherevotan, M. Jasil, C. P. Vinod, A. K. Singh and S. C. Peter, *ACS Catal.*, 2021, **12**, 687–697.
- 289 R. Pang, K. Teramura, M. Morishita, H. Asakura, S. Hosokawa and T. Tanaka, *Commun. Chem.*, 2020, **3**, 137.
- 290 H. Huang, J. Zhao, B. Weng, F. Lai, M. Zhang, J. Hofkens, M. B. J. Roeffaers, J. A. Steele and J. Long, *Angew. Chem.*, *Int. Ed.*, 2022, **61**, e202204563.
- 291 H. Ou, S. Ning, P. Zhu, S. Chen, A. Han, Q. Kang, Z. Hu, J. Ye, D. Wang and Y. Li, *Angew. Chem., Int. Ed.*, 2022, 61, e202206579.
- 292 Q. Shi, X. Zhang, Y. Yang, J. Huang, X. Fu, T. Wang, X. Liu, A. Sun, J. Ge, J. Shen, Y. Zhou and Z. Liu, *J. Energy Chem.*, 2021, **59**, 9–18.
- 293 D. Jiang, W. Wang, E. Gao, S. Sun and L. Zhang, *Chem. Commun.*, 2014, **50**, 2005–2007.
- 294 Y. Bai, P. Yang, L. Wang, B. Yang, H. Xie, Y. Zhou and L. Ye, *Chem. Eng. J.*, 2019, **360**, 473–482.
- 295 R. Bhosale, S. Jain, C. P. Vinod, S. Kumar and S. Ogale, *ACS Appl. Mater. Interfaces*, 2019, **11**, 6174–6183.
- 296 M. Wu, C. Chen, Y.-Y. Xu, C.-Y. Ma, M.-Y. Song and G.-B. Jiang, *J. Am. Chem. Soc.*, 2024, **146**, 9163–9171.
- 297 W. Yan, Y. Zhang and Y. Bi, *Angew. Chem., Int. Ed.*, 2023, **63**, e202316459.
- 298 S. Xue, C.-G. Wei, M.-S. Xiao, C. Liang, J.-L. Wang, C. Yang, W.-D. Xing, S.-B. Wang, W. Lin, Z.-Y. Yu and X. Wang, *Proc. Natl. Acad. Sci. U. S. A.*, 2024, **121**, e2319751121.
- 299 W. Bi, Y. Hu, H. Jiang, L. Zhang and C. Li, *Adv. Funct. Mater.*, 2021, **31**, 2010780.
- 300 C. Han, B. Wang, N. Wu, S. Shen and Y. Wang, *Appl. Surf. Sci.*, 2020, **515**, 145952.
- 301 Y.-Y. Li, J.-S. Fan, R.-Q. Tan, H.-C. Yao, Y. Peng, Q.-C. Liu and Z.-J. Li, *ACS Appl. Mater. Interfaces*, 2020, **12**, 54507– 54516.
- 302 P. Yan, F. Ji, W. Zhang, Z. Mo, J. Qian, L. Zhu and L. Xu, J. Colloid Interface Sci., 2023, 634, 1005–1013.
- 303 X. Shi, Y. Huang, Y. Bo, D. Duan, Z. Wang, J. Cao, G. Zhu,
  W. Ho, L. Wang, T. Huang and Y. Xiong, *Angew. Chem.*, *Int. Ed.*, 2022, 61, e202203063.
- 304 J. Fan, L. Cheng, Y. Liu, Y. He, Y. Wang, D. Li and J. Feng, *J. Catal.*, 2019, **378**, 164–175.
- 305 R. Chong, Y. Fan, Y. Du, L. Liu, Z. Chang and D. Li, *Int. J. Hydrogen Energy*, 2018, **43**, 22329–22339.
- 306 W.-K. Jo, S. Kumar and S. Tonda, *Composites, Part B*, 2019, **176**, 107212.
- 307 J. Fan, L. Shi, H. Ge, J. Liu, X. Deng, Z. Li and Q. Liang, *Adv. Funct. Mater.*, 2024, 202412078.
- 308 J. Wu, J. Zhu, W. Fan, D. He, Q. Hu, S. Zhu, W. Yan, J. Hu, J. Zhu, Q. Chen, X. Jiao and Y. Xie, *Nano Lett.*, 2024, 24, 696– 702.
- 309 Y. Zhao, Y. Wei, X. Wu, H. Zheng, Z. Zhao, J. Liu and J. Li, *Appl. Catal., B*, 2018, **226**, 360–372.
- 310 X. Wu, H. Ling Tan, C. Zhang, Z. Teng, Z. Liu, Y. Hau Ng, Q. Zhang and C. Su, *Prog. Mater. Sci.*, 2023, 133, 101047.

- 311 M. Xu, J. Yang, C. Sun, L. Liu, Y. Cui and B. Liang, *Chem. Eng. J.*, 2020, **389**, 124402.
- 312 S. Song, Z. Xing, H. Zhao, Z. Li and Z. Wei, *Green Energy Environ.*, 2023, **8**, 1232–1264.
- 313 Y. Zhao, Y. Wei, X. Wu, H. Zheng, Z. Zhao, J. Liu and J. Li, *Appl. Catal., B*, 2018, **226**, 360–372.
- 314 S. Yao, J. He, F. Gao, H. Wang, J. Lin, Y. Bai, J. Fang, F. Zhu, F. Huang and M. Wang, *J. Mater. Chem. A*, 2023, **11**, 12539– 12558.
- 315 C. Hu, Z. Jiang, Q. Wu, S. Cao, Q. Li, C. Chen, L. Yuan, Y. Wang, W. Yang, J. Yang, J. Peng, W. Shi, M. Zhai, M. Mostafavi and J. Ma, *Nat. Commun.*, 2023, 14, 4767.
- 316 Q. Liu, H. Cheng, T. Chen, T. W. B. Lo, Z. Xiang and F. Wang, *Energy Environ. Sci.*, 2022, **15**, 225–233.
- 317 C. Shen, X. Y. Meng, R. Zou, K. Sun, Q. Wu, Y. X. Pan and C. J. Liu, *Angew. Chem., Int. Ed.*, 2024, **63**, 202402369.
- 318 M. Ma, Z. Huang, L. Li, W. Zhang, R. Guo, R. Zhang, W. Fa, C. Han, Y. Cao, S. Yu and Y. Zhou, *Appl. Catal.*, *B*, 2023, **330**, 122626.
- 319 H. Zhao, J. Duan, Z. Zhang and W. Wang, *ChemCatChem*, 2021, **14**, e202101733.
- 320 H. Guo, M. Chen, Q. Zhong, Y. Wang, W. Ma and J. Ding, *J. CO2 Util.*, 2019, **33**, 233–241.
- 321 X. Wang, Y. Wang, M. Gao, J. Shen, X. Pu, Z. Zhang, H. Lin and X. Wang, *Appl. Catal.*, *B*, 2020, **270**, 118876.
- 322 M. Ma, Z. Huang, R. Wang, R. Zhang, T. Yang, Z. Rao,
  W. Fa, F. Zhang, Y. Cao, S. Yu and Y. Zhou, *Green Chem.*, 2022, 24, 8791–8799.
- 323 W. Jiao, Y. Xie, F. He, K. Wang, Y. Ling, Y. Hu, J. Wang, H. Ye, J. Wu and Y. Hou, *Chem. Eng. J.*, 2021, **418**, 129286.
- 324 P. Ling, J. Zhu, Z. Wang, J. Hu, J. Zhu, W. Yan, Y. Sun and Y. Xie, *Nanoscale*, 2022, **14**, 14023–14028.
- 325 S. Gao, B. Gu, X. Jiao, Y. Sun, X. Zu, F. Yang, W. Zhu, C. Wang, Z. Feng, B. Ye and Y. Xie, *J. Am. Chem. Soc.*, 2017, **139**, 3438–3445.
- 326 Y. Wang, X. Liu, X. Han, R. Godin, J. Chen, W. Zhou, C. Jiang, J. F. Thompson, K. B. Mustafa, S. A. Shevlin, J. R. Durrant, Z. Guo and J. Tang, *Nat. Commun.*, 2020, 11, 2531.
- 327 M. M. Kandy and V. G. Gaikar, *Mater. Res. Bull.*, 2018, **102**, 440–449.
- 328 S. Ijaz, M. F. Ehsan, M. N. Ashiq and T. He, *Catal. Sci. Technol.*, 2015, 5, 5208–5215.
- 329 M. Habibi Zare and A. Mehrabani-Zeinabad, ACS Appl. Energy Mater., 2023, 6, 3173-3199.
- 330 W.-X. Shi, J.-R. Huang, S.-Y. Xu, H. Luo and J.-W. Zhang, *Adv. Mater.*, 2024, **36**, 2306906.
- 331 M. A. Gondal, M. A. Dastageer, L. E. Oloore and U. Baig, J. Photochem. Photobiol., A, 2017, 343, 40–50.
- 332 M. Ma, Z. Huang, D. E. Doronkin and Y. Zhou, *Appl. Catal.*, B, 2022, **300**, 120695.
- 333 G. Yin, H. Abe, R. Kodiyath, S. Ueda, N. Srinivasan,A. Yamaguchi and M. Miyauchi, *J. Mater. Chem. A*, 2017, 5, 12113–12119.
- 334 Y. Wang, X. Shang, J. Shen, Z. Zhang, D. Wang, J. Lin, J. C. S. Wu, X. Fu, X. Wang and C. Li, *Nat. Commun.*, 2020, **11**, 3043.

- 335 M. Zhang, X. Wang, X. Qi, H. Guo, L. Liu, Q. Zhao and W. Cui, *J. Catal.*, 2022, **413**, 31–47.
- 336 Q. Zhu, Y. Cao, Y. Tao, T. Li, Y. Zhang, H. Shang, J. Song and G. Li, J. CO2 Util., 2021, 54, 101781.
- 337 J. He, P. Lv, J. Zhu and H. Li, *RSC Adv.*, 2020, **10**, 22460–22467.
- 338 D. O. Adekoya, M. Tahir and N. A. S. Amin, *J. CO2 Util.*, 2017, **18**, 261–274.
- 339 H. Wang, S. Cheng, X. Cai, L. Cheng, R. Zhou, T. Hou and Y. Li, *Catal. Commun.*, 2022, **162**, 106372.
- 340 P. Wang, X. Ba, X. Zhang, H. Gao, M. Han, Z. Zhao, X. Chen, L. Wang, X. Diao and G. Wang, *Chem. Eng. J.*, 2023, 457, 141248.
- 341 Y.-C. Hao, L.-W. Chen, J. Li, Y. Guo, X. Su, M. Shu, Q. Zhang, W.-Y. Gao, S. Li, Z.-L. Yu, L. Gu, X. Feng, A.-X. Yin, R. Si, Y.-W. Zhang, B. Wang and C.-H. Yan, *Nat. Commun.*, 2021, **12**, 2682.
- 342 H. Ou, G. Li, W. Ren, B. Pan, G. Luo, Z. Hu, D. Wang and Y. Li, *J. Am. Chem. Soc.*, 2022, **144**, 22075–22082.
- 343 H. Xu, Z. Wang, H. Liao, D. Li, J. Shen, J. Long, W. Dai, X. Wang and Z. Zhang, *Appl. Catal.*, *B*, 2023, 336, 122935.
- 344 H. Li, M. Xia, X. Wang, B. Chong, H. Ou, B. Lin and G. Yang, *Appl. Catal., B*, 2024, **342**, 123423.
- 345 T. Billo, F. Y. Fu, P. Raghunath, I. Shown, W. F. Chen, H. T. Lien, T. H. Shen, J. F. Lee, T. S. Chan, K. Y. Huang, C. I. Wu, M. C. Lin, J. S. Hwang, C. H. Lee, L. C. Chen and K. H. Chen, *Small*, 2018, 14, 1702928.
- 346 S. Sun, M. Watanabe, J. Wu, Q. An and T. Ishihara, *J. Am. Chem. Soc.*, 2018, **140**, 6474–6482.

- 347 T. Wang, L. Chen, C. Chen, M. Huang, Y. Huang, S. Liu and
   B. Li, ACS Nano, 2022, 16, 2306–2318.
- 348 S. Chakraborty, R. Das, M. Riyaz, K. Das, A. K. Singh, D. Bagchi, C. P. Vinod and S. C. Peter, *Angew. Chem., Int. Ed.*, 2023, **135**, e202216613.
- 349 Z. Chen, G. Wang, T. Wang, D.-S. Wang and J.-J. Mao, *Angew. Chem., Int. Ed.*, 2022, **134**, e202210789.
- 350 Y. Tian, R. Wang, S. Deng, Y. Tao, W. Dai, Q. Zheng,
  C. Huang, C. Xie, Q. Zeng, J. Lin and H. Chen, *Nano Lett.*, 2023, 23, 10914–10921.
- 351 K. Das, R. Das, M. Riyaz, A. Parui, D. Bagchi, A. K. Singh, A. K. Singh, C. P. Vinod and S. C. Peter, *Adv. Mater.*, 2022, 35, 2205994.
- 352 B. Zhou, Y. Ma, P. Ou, Z. Ye, X.-Y. Li, S. Vanka, T. Ma, H. Sun, P. Wang, P. Zhou, J. K. Cooper, Y. Xiao, I. A. Navid, J. Pan, J. Song and Z. Mi, *Nat. Catal.*, 2023, 6, 987–995.
- 353 W. Wang, C. Deng, S. Xie, Y. Li, W. Zhang, H. Sheng, C. Chen and J. Zhao, *J. Am. Chem. Soc.*, 2021, 143, 2984– 2993.
- 354 P. Du, J.-Y. Ding, J.-C. Zhu, Q. Hu, D.-P. He, Y. Wu, W.-X. Liu, S. Zhu, J. Hu, J.-F. Zhu, Q.-X. Chen, X.-C. Jiao and Y. Xie, *Angew. Chem., Int. Ed.*, 2024, 63, e202400828.
- 355 R. t. Guo, J. Wang, Z. x. Bi, X. Chen, X. Hu and W. g. Pan, *Small*, 2022, **19**, 2206314.
- 356 J. Zhu, L. Zhou, Y. Zhu, J. Huang, L. Hou, J. Shen, S. Dai and C. Li, *Small*, 2022, **18**, 2104399.
- 357 A. Ziarati, A. Badiei, R. Grillo and T. Burgi, *ACS Appl. Mater. Interfaces*, 2019, **11**, 5903–5910.