


 Cite this: *RSC Adv.*, 2023, **13**, 26683

# Raman spectroscopy investigation of magnesium oxide nanoparticles

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We investigate Raman spectra (100  $\text{cm}^{-1}$  to 3900  $\text{cm}^{-1}$ ) of magnesium oxide nanoparticles with nominal sizes of 10 nm, 20 nm, 40 nm, 50 nm, and 300 nm. The crystal structure of MgO prohibits first-order modes and yet, there are numerous reports of relatively intense peaks throughout the literature. Raman signals at approximately 278  $\text{cm}^{-1}$  and 445  $\text{cm}^{-1}$  that were attributed to MgO nanoparticles by previous authors are shown to belong to layers of  $\text{Mg}(\text{OH})_2$  formed on the surface of MgO nanoparticles. Through an annealing process at 400 °C in an  $\text{O}_2$  atmosphere, we observe that modes in the 3700  $\text{cm}^{-1}$  spectral region, which are a signature of OH groups, disappear together with modes at 278  $\text{cm}^{-1}$  and 445  $\text{cm}^{-1}$ , thus establishing a necessary criterion to associate all of these peaks to the presence of OH groups on the surface.

 Received 5th July 2023  
 Accepted 26th August 2023

DOI: 10.1039/d3ra04492k

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## Introduction

Magnesium oxide forms highly stable crystals with face-centered cubic structure and space group *Fm3m*. In its nanoparticle form, thanks to a substantial surface-to-volume ratio, the surface adsorptive properties per weight become large leading to many applications.<sup>1</sup> For instance, MgO nanoparticles are applied in purification of fresh water and the treatment of wastewater as they have excellent adsorptive properties with respect to *e.g.* Cd(II) and Pb(II).<sup>1,2</sup> When MgO nanoparticles are poured into aqueous environments containing these metals, the concentration of unbound contaminants decreases drastically. Nano-powdered magnesium oxide also has antibacterial properties in such a way that it is employed to deactivate bacteria, *i.e.* *Escherichia coli* and *Staphylococcus*.<sup>1,3</sup> The specific surface chemistry has therefore attracted attention since the earliest observations of these effects.

Optical Raman properties of MgO single crystal are well-known: bulk MgO has a very large optical bandgap of approximately 7.8 eV (ref. 4) and is inactive in first-order Raman scattering because of its rocksalt structure.<sup>5</sup> Nevertheless, many authors report Raman modes in MgO micro- and nanoparticles that they attribute to first-order scattering (one photon – one phonon).<sup>6–9</sup> Table 1 exemplifies the contribution of authors who worked on Raman properties of MgO nanoparticles. For instance, Ishikawa *et al.* who investigated optical Raman properties of MgO nanoparticles report first-order Raman modes at 280  $\text{cm}^{-1}$ , 446  $\text{cm}^{-1}$ , 1088  $\text{cm}^{-1}$ , and 1120  $\text{cm}^{-1}$  observable in

MgO nanoparticles sized from 30 nm to 3000 nm.<sup>6</sup> They suggested that the forbidden bulk Raman modes of MgO single crystal become active because the size of the nanoparticles allowed the selection rules to relax. Their consideration was based on the numerically determined phonon density of states that peaks around 280  $\text{cm}^{-1}$  and 460  $\text{cm}^{-1}$  (Fig. 1b) but that requires momentum to become accessible in the zone centre. So far, there is no experimental evidence for that size-dependent surface relaxation to occur in MgO.

Before Ishikawa, Sangster *et al.* had determined the phonon dispersion relation of MgO single crystal by inelastic neutron scattering measurements.<sup>10</sup> Ishikawa *et al.* compared their own observed Raman modes to the phonon density of states MgO single crystal (Fig. 1) measured by Sangster *et al.*, and they attribute the 280  $\text{cm}^{-1}$  mode to TA phonon at the zone boundary X, which occurs at 301  $\text{cm}^{-1}$  whereas the 446  $\text{cm}^{-1}$  mode is ascribed to the TO zone center mode at 393  $\text{cm}^{-1}$  in the phonon density of states (Fig. 1). Phonons other than those in the zone center do not respect momentum conservation of first order processes. Moreover, as regards the second attribution, there is a significant deviation (13.2%) between the value 446  $\text{cm}^{-1}$  observed on the Raman spectrum and the 393  $\text{cm}^{-1}$  mode on the dispersion relation. Not to mention that the peak at 1088  $\text{cm}^{-1}$  is attributed to a surface mode since there is no trace of it in the phonon dispersion relation, even though surface modes are known to appear in the TO–LO phonon gap, which is not the case here. As for the peak at 1120  $\text{cm}^{-1}$ , no explanation is provided.

Morozov *et al.* observe even more peaks: 275  $\text{cm}^{-1}$ , 370  $\text{cm}^{-1}$ , 448  $\text{cm}^{-1}$ , 560  $\text{cm}^{-1}$ , 960  $\text{cm}^{-1}$ , 1080  $\text{cm}^{-1}$ , and 1120  $\text{cm}^{-1}$  regarding experimental Raman measurements on MgO nanoparticles.<sup>9</sup> They assign the peaks at 275  $\text{cm}^{-1}$ , 448  $\text{cm}^{-1}$ , 1080  $\text{cm}^{-1}$ , and 1120  $\text{cm}^{-1}$  the same way as Ishikawa. Morozov

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Table 1 Summary of the Raman peak assignments for MgO nanoparticles by different authors in the literature (all units are in  $\text{cm}^{-1}$ )

Author	Ishikawa <i>et al.</i> <sup>6</sup>	Böckelmann <i>et al.</i> <sup>11</sup>	Morozov <i>et al.</i> <sup>9</sup>	Kim <i>et al.</i> <sup>7</sup>	Sangster <i>et al.</i> <sup>10</sup>
Experiment	Raman spectroscopy on MgO nanoparticles	Raman spectroscopy on MgO nanoparticles	Raman spectroscopy on MgO nanoparticles	Raman spectroscopy on MgO nanoparticles	Phonon dispersion relation measured by inelastic neutron scattering
TA mode at X	280	—	275	290	~300
TO mode at $\Gamma$	446	—	448	448	~393
LO mode at $\Gamma$	—	—	960	—	~722
LA mode at X	—	—	370	—	~410
Surface modes	1080	—	1080, 1120	1080, 1121	—
Other detected peaks	1120	595, 719, 978, and 1096	560	—	—

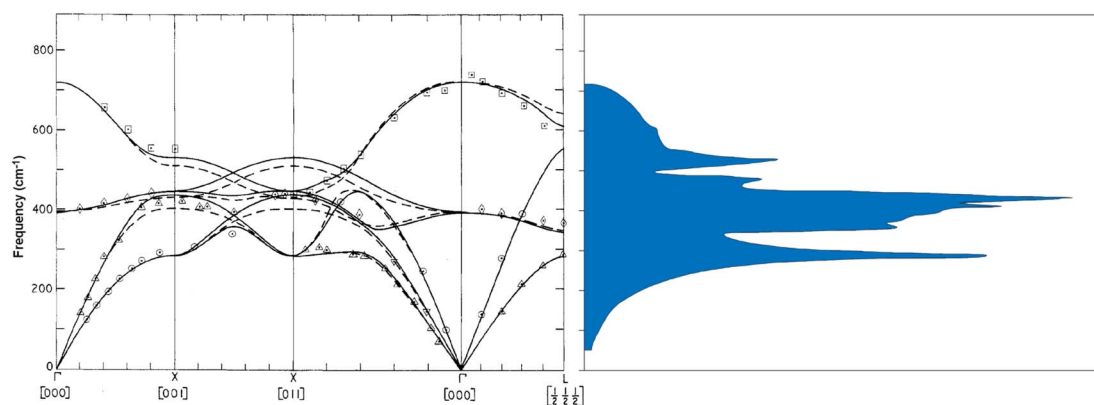


Fig. 1 (a) Phonon dispersion diagram calculated by inelastic neutron scattering for MgO bulk. The solid and the dashed lines correspond to two different models the authors used to model the experimental data. (b) Phonon density of states associated with the dispersion diagram. (from ref. 10 DOI: 10.1088/0022-3719/3/5/017 © IOP Publishing. Reproduced with permission. All rights reserved.)

*et al.* claim the peak at  $370 \text{ cm}^{-1}$  to be the LA phonon mode on the dispersion curve (Fig. 1a), but one can see that the LA mode at  $370 \text{ cm}^{-1}$  is rather at the zone boundary X, so that if this process would occur, momentum would again not be conserved. The authors ascribe the modes at  $1080 \text{ cm}^{-1}$  and  $1120 \text{ cm}^{-1}$  to surface modes although they are not in the TO-LO gap. As for the mode at  $960 \text{ cm}^{-1}$ , Morozov attributes it to the LO mode ( $722 \text{ cm}^{-1}$ ) on the dispersion diagram although there is a substantial deviation with that mode.

Böckelmann and his team synthesized MgO nanoparticles of two size distributions (30 nm et 60 nm) by three different methods.<sup>11</sup> Raman spectroscopy of MgO nanoparticles that have an average size of 30 nm have shown modes at  $595 \text{ cm}^{-1}$ ,  $719 \text{ cm}^{-1}$  and  $1096 \text{ cm}^{-1}$ , regardless of the synthesis method, whereas those that have an average size of 60 nm gave Raman modes at  $592 \text{ cm}^{-1}$  and  $1088 \text{ cm}^{-1}$ . Until this day, none of the modes  $595 \text{ cm}^{-1}$  and  $592 \text{ cm}^{-1}$  seems to be reported by another author as far as MgO nanoparticles are concerned. The cause of Raman mode at  $978 \text{ cm}^{-1}$ , which can be found on the spectrum of some of their microcrystals of MgO, is not clear.

Kim *et al.* obtained the same Raman peaks as Ishikawa for MgO nanoparticles.<sup>7</sup> They attribute the peaks the same way as Ishikawa, except for the peak at  $1121 \text{ cm}^{-1}$ , which they consider to be a surface phonon mode.

Apart from MgO nanoparticles, MgO in other forms and shapes (nanotubes, nanocubes of a variety of sizes) as well as thin films of MgO have other phonon modes that have been identified.<sup>12–14</sup> Some of these modes were dependent on the size and shape<sup>12,13,15,16</sup> while other modes were rather constant and identified as surface phonon modes.<sup>6–9</sup> It is understandable that until this day the assignment of Raman peaks of MgO nanoparticles is still under debate. The temptation to invoke a correlation with the phonon density of states (PDOS) as illustrated in Fig. 1b is understandable as it predicts a strong probability to activate modes *via* the relaxation of momentum preservation. Because Raman properties of MgO nanoparticles are subject to many discussions and inconsistencies, not to mention that authors have divergent views on the matter, the scope of this work is to shed light on Raman properties of MgO nanoparticles by clarifying some of the peaks obtained, and by explaining the effects that cause them. The key will be to carefully distinguish between pure magnesium oxide and “nominally pure” magnesium oxide and to take into consideration its strong hygroscopic nature.

## Experimental methods

High purity MgO nanoparticles (99+%) of different nominal sizes (10 nm, 20 nm, 40 nm, 50 nm, and 300 nm) were



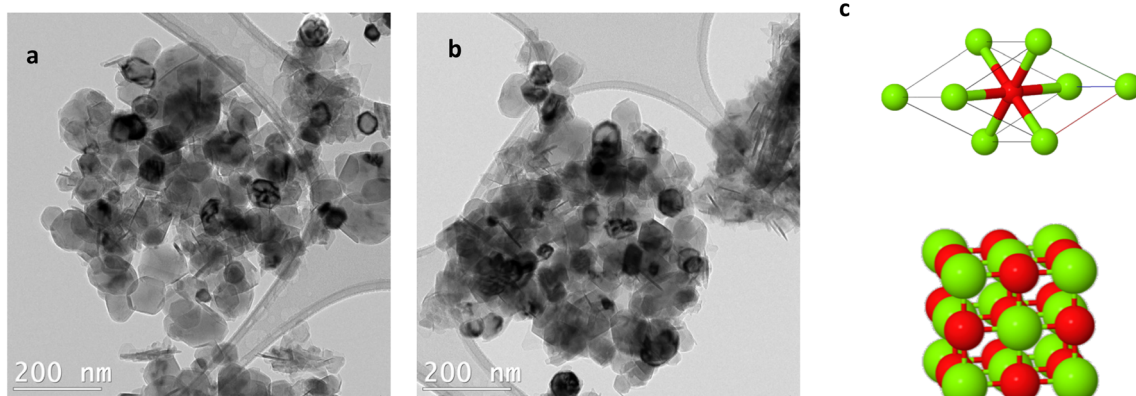


Fig. 2 (a) and (b) are TEM images of MgO nanoparticles of nominal size 50 nm. (c) Primitive cell (top) and conventional cell (bottom) of MgO nanoparticles. Green atoms are  $\text{Mg}^{2+}$ ; red atoms are  $\text{O}^{2-}$ .

purchased from US Research Nanomaterials Inc. The nanoparticles have been synthesized by combustion method from magnesium metal ingot. The nanoparticles' sizes are the nominal sizes that is only a label to identify them, as a thorough analysis on their size distributions has not been made in this paper. Transmission electronic microscopy (TEM) images were acquired using a JEOL JEM-2100F transmission electron microscope with a Schottky field-emission gun operated at 200 kV. Sample preparation for TEM was made by dispersing a small amount of MgO nanoparticles in 1 mL of methanol by ultrasonication. TEM images of nominal size 50 nm are represented on Fig. 2. MgO has a face-centered cubic structure with primitive and conventional cells as depicted on Fig. 2c.

Raman spectra were obtained on a Horiba spectrometer (iHR320) system with a linearly polarized continuous wave DPSS laser (Cobolt 04-01, of wavelength 473 nm, TEM 00, 25 mW), objective of  $50\times$  (Olympus, LMPLFLN) with numerical aperture of 0.50. All spectra were acquired with a CCD detector  $1024 \times 256$  pixel (Horiba Synapse BIDD) with  $2400 \text{ l mm}^{-1}$  grating for some of the spectra (approximately  $1 \text{ cm}^{-1}$  hardware resolution), and  $1200 \text{ l mm}^{-1}$  for others (approximately  $2.4 \text{ cm}^{-1}$  hardware resolution). The power of the laser before our sample was about 14 mW for all spectra. All spectra were acquired in backscattering geometry without analyzer. The exposition time varied from 120 to 360 seconds, and the number of spectra acquired for averaging (accumulation) varied from 3 to 5 depending on each sample in order to get the best signal-to-noise ratio. The calibration was made with a xenon lamp (Ocean Optics, XE-1 Xenon Calibration light source, 916–1984 nm) and a numerical baseline subtraction.

X-ray powder diffraction (XRD) measurements were made using a diffractometer Bruker D8 Advance (from Bruker AXS, Inc.) with an X-ray beam of 40 kV generated by a Cu  $K\alpha_1$  (15 406 Å) and a  $K\alpha_2$  (15 444 Å) sources with a ratio 2 : 1. The step size of the spectra is  $0.04^\circ$  for all spectra.

All annealing processes were conducted with a high temperature quartz tube furnace (model GSL-1000X, MTI Corporation, USA) at  $400^\circ\text{C}$  during 1 hour with an oxygen flow of about  $80 \text{ mL min}^{-1}$ . The heating rate was  $20^\circ\text{C min}^{-1}$  during

a temperature ramp of 20 minutes. The oxygen atmosphere for the annealing was chosen in order to prevent reduction of the MgO nanoparticles' surfaces by oxygen desorption. After the annealing, MgO nanoparticles were cooled down with the intrinsic time constant of the furnace, *i.e.* without any assisted cooling process, in the same  $\text{O}_2$  flow, reducing the contact with moisture at still elevated temperatures. The cooling time varied between 3 to 5 hours. When the nanoparticles reached a temperature around  $60^\circ\text{C}$ , they were removed from the furnace tube to be characterized by Raman spectroscopy or XRD.

## Results and discussion

### A Raman spectroscopy characterization

Fig. 3 depicts Raman spectra of as-received MgO nanoparticles for Raman shifts ranging from  $100 \text{ cm}^{-1}$  to  $3900 \text{ cm}^{-1}$ . On Fig. 3a, we can observe that regardless of their sizes, all nanoparticles have peaks at  $278 \text{ cm}^{-1}$  and  $445 \text{ cm}^{-1}$ . We also acquired high wavenumber spectra (Fig. 3b and c) where we can see Raman peaks between  $3600 \text{ cm}^{-1}$  and  $3800 \text{ cm}^{-1}$ . These high frequency modes are traces of OH groups that are formed on the surface of the nanoparticles. MgO is known to be highly hygroscopic, therefore it reacts with physisorbed water forming  $\text{Mg}(\text{OH})_2$ .<sup>17,18</sup> On Fig. 3c, we can distinguish three peaks labeled A, B, and C. The highest peak in intensity labeled A (approximately at  $3646 \text{ cm}^{-1}$ ) is close to the  $3652 \text{ cm}^{-1}$  OH symmetric stretching mode  $A_{1g}$  in bulk crystalline  $\text{Mg}(\text{OH})_2$ ,<sup>19–22</sup> we therefore assign peak A to the bulk  $A_{1g}$  mode in  $\text{Mg}(\text{OH})_2$ . Peaks B ( $3669 \text{ cm}^{-1}$ ), and C ( $3734 \text{ cm}^{-1}$ ), which are lower in intensity have also been observed by other authors<sup>22,23</sup> and are equally associated with  $\text{Mg}(\text{OH})_2$ .

Several theoretical and experimental investigations report that on  $\{100\}$  surfaces, water is only physisorbed while chemisorption occurs in the presence of defects on the surface, *i.e.* corners and edges.<sup>24–26</sup> Atoms on  $\{100\}$  surfaces have a coordination number of five; atoms on the edge of a terrace have a coordination number of four; and atoms at the corner have coordination number of three.<sup>19,27</sup> It has been reported by other



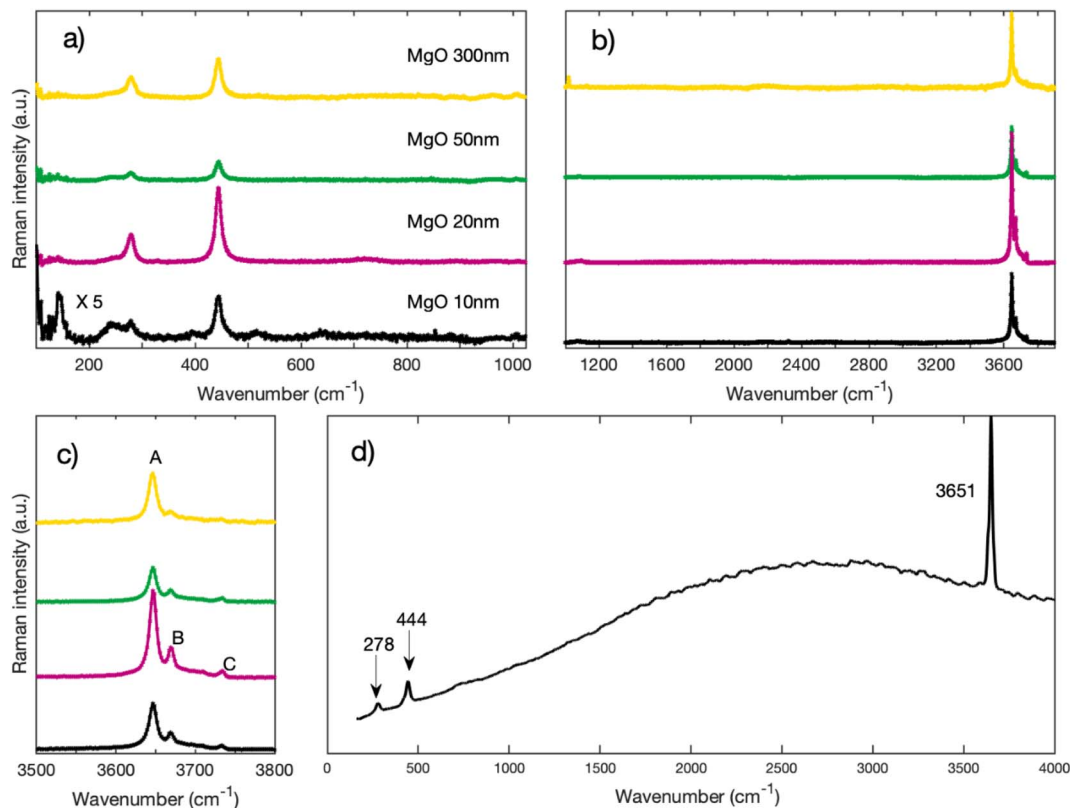


Fig. 3 (a) Raman spectra of MgO nanoparticles for low wavenumbers acquired with a grating of  $2400 \text{ l mm}^{-1}$  (b) Raman spectra of MgO nanoparticles for higher wavenumbers acquired with a grating of  $1200 \text{ l mm}^{-1}$  (c) zoom on the region around  $3700 \text{ cm}^{-1}$  of spectra of Fig. 1b. (d) Raman spectrum of  $\text{Mg}(\text{OH})_2$  from RRUFF database.<sup>31</sup>

authors that atoms in the MgO surface that have a coordination number of five - terrace atoms - may not form chemical bonds with water. Chemisorption appears to be associated to surface imperfections on  $\{100\}$  surfaces and therefore to ions with lower coordination.<sup>28,29</sup> An alternative to imperfections on presumably  $\{100\}$  surfaces is the presence of other surfaces in the habitus of MgO nanoparticles (see TEM images from Fig. 2a and b), e.g. face  $\{110\}$ . Apart from  $\{110\}$  surfaces in octahedrally shaped nanoparticles, surfaces with higher Miller indices may also be present. We would like to remind that  $\{111\}$  surfaces are terminated with ionic species of a single kind (either  $\text{O}^{2-}$  or  $\text{Mg}^{2+}$ ) and therefore unstable; they tend to recrystallize into surfaces with different Miller indices<sup>30</sup> or to reduce their free energy through chemisorption of OH-groups forming  $\text{Mg}(\text{OH})_2$ .

In order to verify if the peaks  $278 \text{ cm}^{-1}$ ,  $445 \text{ cm}^{-1}$ , and those in the high frequencies belong intrinsically to the MgO nanoparticles, we annealed each MgO sample during an hour in a quartz furnace under an  $\text{O}_2$  flow. Most of the cooling process was made inside the furnace under oxygen atmosphere with the intrinsic time constant (approximately 1 hour) of the thermal heat capacity of the furnace.

Fig. 4 depicts the spectra of each sample before and after annealing. One can clearly see that, for all sized nanoparticles, the peaks in the  $3700 \text{ cm}^{-1}$  region are no longer visible (Fig. 4b) after annealing. The annealed spectra are almost featureless in the  $3700 \text{ cm}^{-1}$  region for MgO nanoparticles with 300 nm and

50 nm nominal size. As for MgO nanoparticles with a nominal size of 20 nm, the annealed spectrum has a small remaining feature around  $3600 \text{ cm}^{-1}$ , while the spectrum of MgO nanoparticles with a nominal size of 10 nm has a broad band between  $3000 \text{ cm}^{-1}$  and  $3800 \text{ cm}^{-1}$ . Smaller particles have a larger surface-to-volume ratio than larger ones and therefore facilitate the detection of adsorbed species. We therefore attribute the features observed in the smallest nanoparticles of 10 and 20 nm size respectively to physisorbed molecular water.<sup>32</sup> Even with the presence of some physisorbed water on the surface, our post-annealing spectra suggest that the OH peaks, and hence the  $\text{Mg}(\text{OH})_2$ , are no longer present on the annealed nanopowder. It has been reported by other authors that  $\text{Mg}(\text{OH})_2$  annealed at temperatures of  $400 \text{ }^\circ\text{C}$  and higher transforms into MgO.<sup>17,33</sup> A small peak at  $1088 \text{ cm}^{-1}$  that is found on the annealed spectra of each nanoparticle appears after the annealing (Fig. 4b). It has been reported by many authors that MgO reacts with  $\text{CO}_2$  to form  $\text{MgCO}_3$ .<sup>34-36</sup> We assign that peak at  $1088 \text{ cm}^{-1}$  to the carbonate group  $\text{CO}_3^{2-}$  based on the Raman spectrum of  $\text{MgCO}_3$  (see Fig. 5), the lower frequency peaks that appear on the  $\text{MgCO}_3$  spectrum (about  $210 \text{ cm}^{-1}$  and  $330 \text{ cm}^{-1}$ ) being too weak to appear on our spectra. According to literature, an annealing of  $700 \text{ }^\circ\text{C}$  is required to desorb these ions from the surface of  $\text{MgO}$ .<sup>34-36</sup>

Most importantly, both peaks at  $278 \text{ cm}^{-1}$  and  $445 \text{ cm}^{-1}$  also disappear and do not spontaneously form again upon cooling





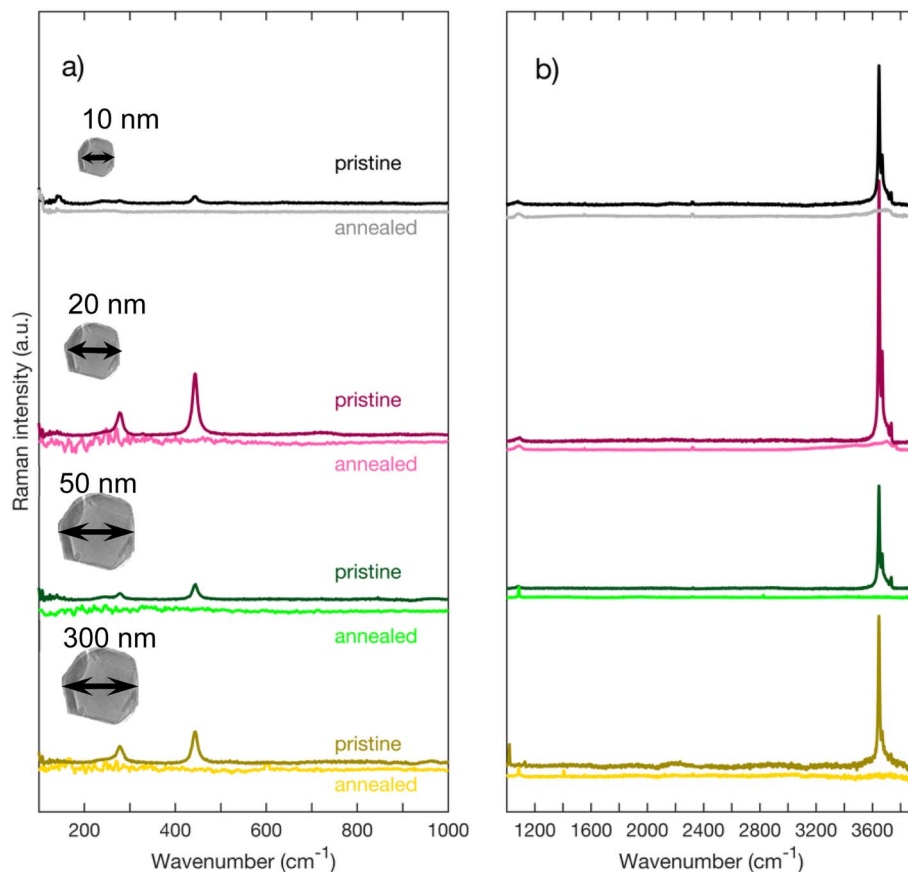


Fig. 4 (a) and (b) Raman spectra of MgO nanoparticles before (pristine) and after annealing at 400 °C of an hour with oxygen flow. (a) The spectral range below 1000  $\text{cm}^{-1}$  (approx. 1  $\text{cm}^{-1}$  per pixel), has been acquired with a grating of 2400  $\text{l mm}^{-1}$ . (b) The spectral range above 1000  $\text{cm}^{-1}$  (approx. 2  $\text{cm}^{-1}$  per pixel) has been acquired with a grating of 1200  $\text{l mm}^{-1}$ . The spectra obtained after annealing have been integrated approximately 5 times longer to highlight the absence of any remaining features except for weak signatures of free water in the samples with the highest surface to volume ratio and the strongest capillarity.

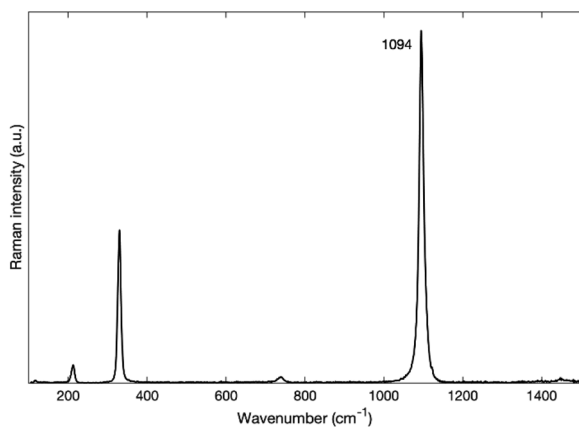


Fig. 5 Raman spectrum of  $\text{MgCO}_3$  from RRUFF database.<sup>31</sup>

when physisorption of water occurs again. This is a necessary requirement to assign these peaks to the presence of  $\text{Mg}(\text{OH})_2$ . For comparison purposes, the spectrum of  $\text{Mg}(\text{OH})_2$  is plotted on Fig. 3d from RRUFF database<sup>31</sup> with a clear presence of peaks at 278  $\text{cm}^{-1}$ , 444  $\text{cm}^{-1}$  and 3651  $\text{cm}^{-1}$ .

So far, for MgO nanoparticles, and as expected for the bulk structure, there seem to be no peaks that belong intrinsically to MgO nanoparticles or peaks that have appeared because of the reduction of symmetry due to the nanoscale size of the nanoparticles. The peaks at 278  $\text{cm}^{-1}$  and 445  $\text{cm}^{-1}$  are present in the bulk phase of  $\text{Mg}(\text{OH})_2$  and therefore are a signature of a surface contamination on nanoscale MgO, just as the disappearing peaks around 3700  $\text{cm}^{-1}$ .

#### XRD characterization

We carried out XRD measurements of powder diffraction to verify the nature and the crystal structure of the nanoparticles after the annealing to acknowledge that they still consist of MgO. In Fig. 6, the XRD spectra of all four nanopowders are depicted. Comparison of these peaks with those from the reported values of MgO (JCPDS number 00-045-0946) clearly confirms that for all sizes our nanopowders are MgO. Furthermore, no phase transition of MgO, that could possibly interfere with the interpretation of XRD data, is reported in the respective temperature range. XRD, even on nanopowders is however not sufficiently sensitive to confirm the presence of  $\text{MgCO}_3$



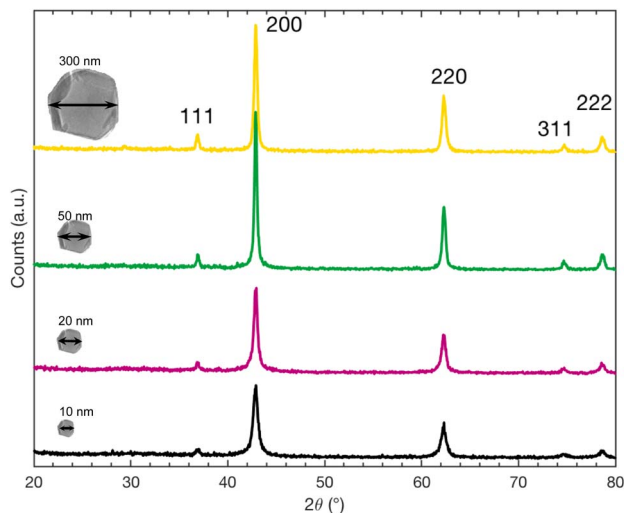


Fig. 6 XRD measurements of MgO nanoparticles annealed at 400 °C under oxygen flow.

## Conclusions

In this work, we have revisited the origin of several peaks obtained in Raman spectra of nominally pure MgO nanoparticles that have preoccupied researchers as their observation collides with the symmetry requirement of bulk MgO being Raman silent. All observations throughout recent literature may be conclusively explained through almost inevitable surface contaminations when MgO is handled under ambient conditions. Particularly, we have shown that the Raman peaks at  $278\text{ cm}^{-1}$  and  $445\text{ cm}^{-1}$  that were occasionally assigned to MgO do not actually belong to pure MgO nanoparticles. As a matter of fact, after an annealing process of 400 °C under  $\text{O}_2$  atmosphere during an hour, MgO nanoparticles do not exhibit these Raman peaks at  $278\text{ cm}^{-1}$  and  $445\text{ cm}^{-1}$  anymore. These peaks are rather assigned to  $\text{Mg}(\text{OH})_2$  that forms on the MgO nanoparticles due to a chemical reaction with humidity contained in ambient air.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

MD, APR, and AM appreciate the help of Ms Katharina Kohlmann for the XRD measurements. A. Ruediger gratefully acknowledges financial support through an NSERC discovery grant.

## References

- 1 A. A. Pilarska, Ł. Klapiszewski and T. Jesionowski, Recent development in the synthesis, modification and application of  $\text{Mg}(\text{OH})_2$  and MgO: a review, *Powder Technol.*, 2017, **319**, 373–407.

- 2 M. Hua, S. Zhang, B. Pan, W. Zhang, L. Lv and Q. Zhang, Heavy metal removal from water/wastewater by nanosized metal oxides: a review, *J. Hazard. Mater.*, 2012, **211–212**, 317–331.
- 3 S. Makhluף, R. Dror, Y. Nitzan, Y. Abramovich, R. Jelinek and A. Gedanken, Microwave-Assisted Synthesis of Nanocrystalline MgO and Its Use as a Bactericide, *Adv. Funct. Mater.*, 2005, **15**, 1708–1715.
- 4 D. M. Roessler and W. C. Walker, Electronic Spectrum and Ultraviolet Optical Properties of Crystalline MgO, *Phys. Rev.*, 1967, **159**, 733–738.
- 5 N. B. Manson, W. von der Ohe and S. L. Chodos, Second-Order Raman Spectrum of MgO, *Phys. Rev. B: Solid State*, 1971, **3**, 1968–1972.
- 6 K. Ishikawa, N. Fujima and H. Komura, First-order Raman scattering in MgO microcrystals, *J. Appl. Phys.*, 1985, **57**, 973–975.
- 7 H. Kim and H. Kim, Fabrication and Raman Studies of MgO/SnO<sub>2</sub> Core-Shell Heteronanowires, *Acta Phys. Pol., A*, 2009, **116**, 58–61.
- 8 K. Krishnamoorthy, J. Y. Moon, H. B. Hyun, S. K. Cho and S.-J. Kim, Mechanistic investigation on the toxicity of MgO nanoparticles toward cancer cells, *J. Mater. Chem.*, 2012, **22**, 24610.
- 9 Iu. G. Morozov, S. Sathasivam, O. V. Belousova, I. P. Parkin and M. V. Kuznetsov, Effect of synthesis conditions on room-temperature ferromagnetic properties of MgO nanoparticles, *J. Alloys Compd.*, 2018, **765**, 343–354.
- 10 M. J. L. Sangster, G. Peckham and D. H. Saunderson, Lattice dynamics of magnesium oxide, *J. Phys. C: Solid State Phys.*, 1970, **3**, 1026–1036.
- 11 H. K. Böckelmann and R. G. Schlecht, Raman scattering from microcrystals of MgO, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1974, **10**, 5225–5231.
- 12 M. J. Lagos, A. Trügler, U. Hohenester and P. E. Batson, Mapping vibrational surface and bulk modes in a single nanocube, *Nature*, 2017, **543**, 529–532.
- 13 L. Savio, E. Celasco, L. Vattuone, M. Rocca and P. Senet, MgO/Ag(100): confined vibrational modes in the limit of ultrathin films, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **67**, 075420.
- 14 L. Chen, C. Xu, X.-F. Zhang and T. Zhou, Raman and infrared-active modes in MgO nanotubes, *Phys. E*, 2009, **41**, 852–855.
- 15 Y. Hwang, R. Souda, T. Aizawa, W. Hayami, S. Otani and Y. Ishizawa, *Jpn. J. Appl. Phys.*, 1997, **36**, 5707.
- 16 V. Shpakov, A. Gotte, M. Baudin, T. Woo and K. Hermansson, MgO(001) surface phonons from *ab initio* calculations, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2005, **72**, 195427.
- 17 L. Kumari, W. Z. Li, C. H. Vannoy, R. M. Leblanc and D. Z. Wang, Synthesis, characterization and optical properties of  $\text{Mg}(\text{OH})_2$  micro-/nanostructure and its conversion to MgO, *Ceram. Int.*, 2009, **35**, 3355–3364.
- 18 M. A. Alavi and A. Morsali, Syntheses and characterization of  $\text{Mg}(\text{OH})_2$  and MgO nanostructures by ultrasonic method, *Ultrason. Sonochem.*, 2010, **17**, 441–446.



- 19 A. Kondo, R. Kurosawa, J. Ryu, M. Matsuoka and M. Takeuchi, Investigation on the Mechanisms of  $\text{Mg}(\text{OH})_2$  Dehydration and  $\text{MgO}$  Hydration by Near-Infrared Spectroscopy, *J. Phys. Chem. C*, 2021, **125**, 10937–10947.
- 20 A. Suslu, K. Wu, H. Sahin, B. Chen, S. Yang, H. Cai, T. Aoki, S. Horzum, J. Kang, F. M. Peeters and S. Tongay, Unusual dimensionality effects and surface charge density in 2D  $\text{Mg}(\text{OH})_2$ , *Sci. Rep.*, 2016, **6**, 20525.
- 21 P. Dawson, C. D. Hadfield and G. R. Wilkinson, *The polarized infra-red and Raman spectra OF  $\text{Mg}(\text{OH})_2$  and  $\text{Ca}(\text{OH})_2$* , Pergamon Press, 1973, vol. 34.
- 22 A. Maltseva, V. Shkirskiy, G. Lefèvre and P. Volovitch, Effect of pH on  $\text{Mg}(\text{OH})_2$  film evolution on corroding Mg by *in situ* kinetic Raman mapping (KRM), *Corros. Sci.*, 2019, **153**, 272–282.
- 23 F. Khairallah and A. Glisenti, Synthesis, characterization and reactivity study of nanoscale magnesium oxide, *J. Mol. Catal. A: Chem.*, 2007, **274**, 137–147.
- 24 N. H. de Leeuw, G. W. Watson and S. C. Parker, Atomistic Simulation of the Effect of Dissociative Adsorption of Water on the Surface Structure and Stability of Calcium and Magnesium Oxide, *J. Phys. Chem.*, 1995, **99**, 17219–17225.
- 25 H. Onishi, C. Egawa, T. Aruga and Y. Iwasawa, Adsorption of Na atoms and oxygen-containing molecules on  $\text{MgO}(100)$  and  $(111)$  surfaces, *Surf. Sci.*, 1987, **191**, 479–491.
- 26 H. DUNSKI, W. JOWIAK and H. SUGIER, Dehydroxylation of the surface of magnesium oxide by temperature programmed desorption, *J. Catal.*, 1994, **146**, 166–172.
- 27 C. Chizallet, G. Costentin, M. Che, F. Delbecq and P. Sautet, Revisiting Acido-basicity of the  $\text{MgO}$  Surface by Periodic Density Functional Theory Calculations: Role of Surface Topology and Ion Coordination on Water Dissociation, *J. Phys. Chem. B*, 2006, **110**, 15878–15886.
- 28 S. Knop, T. L. C. Jansen, J. Lindner and P. Vöhringer, On the nature of OH-stretching vibrations in hydrogen-bonded chains: Pump frequency dependent vibrational lifetime, *Phys. Chem. Chem. Phys.*, 2011, **13**, 4641.
- 29 G. Lipinski, K. Jeong, K. Moritz, M. Petermann, E. F. May, P. L. Stanwix and M. Richter, Application of Raman Spectroscopy for Sorption Analysis of Functionalized Porous Materials, *Adv. Sci.*, 2022, **9**, 2105477.
- 30 R. Plass, J. Feller and M. Gajdardziska-Josifovska, *Morphology of  $\text{MgO}(111)$  surfaces: artifacts associated with the faceting of polar oxide surfaces into neutral surfaces*, 1998, vol. 414.
- 31 B. Lafuente, R. T. Downs, H. Yang and N. Stone, in *Highlights in Mineralogical Crystallography*, De Gruyter, 2015, pp. 1–30.
- 32 A. de Lima Ribeiro, C. Artlett and H. Pask, A LIDAR-Compatible, Multichannel Raman Spectrometer for Remote Sensing of Water Temperature, *Sensors*, 2019, **19**, 2933.
- 33 M. A. Alavi and A. Morsali, Syntheses and characterization of  $\text{Mg}(\text{OH})_2$  and  $\text{MgO}$  nanostructures by ultrasonic methods, *Ultrason. Sonochem.*, 2010, **17**, 441–446.
- 34 D. K. Aswal, K. P. Muthe, S. Tawde, S. Chodhury, N. Bagkar, A. Singh, S. K. Gupta and J. V. Yakhmi, XPS and AFM investigations of annealing induced surface modifications of  $\text{MgO}$  single crystals, *J. Cryst. Growth*, 2002, **236**, 661–666.
- 35 J. Geler-Kremer, A. B. Posadas and A. A. Demkov, Preparation of clean  $\text{MgO}$  surface by oxygen plasma: Comparison with standard substrate cleaning procedures, *J. Vac. Sci. Technol., B: Nanotechnol. Microelectron.: Mater., Process., Meas., Phenom.*, 2020, **38**, 062201.
- 36 T. Awaji, K. Sakuta, Y. S. Yoshiyuki Sakaguchi and T. K. Takeshi Kobayashi, Improved Surface Crystallinity of  $\text{MgO}$  Crystal Substrate through Annealing in Oxygen Atmosphere, *Jpn. J. Appl. Phys.*, 1992, **31**, L642.

