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**Balancing Functional Properties and Environmental Impact
of Graphitic Carbon Nitride: A Case Study on Boron Doping
Syntheses**

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Environmental Significance

Graphitic carbon nitride is a 2D nanomaterial that has emerged as a promising material to address challenges pertaining to water contamination, carbon emissions, and energy production. Its properties as a visible light photocatalyst hold immense value, yet controlling its properties to achieve adaptability and sustainability remains unrealized. This study uncovers ways to tune carbon nitride's optical and chemical characteristics through boron doping and diverse synthesis methods. The work also introduces the MAterial Properties and Sustainability, or MAPS, paradigm, an approach to harmonize physicochemical characteristics with sustainability benchmarks at the material synthesis stage. By considering both dimensions in tandem, design of graphitic carbon nitride is pursued for high-performing properties while minimizing its environmental footprint.

Abstract

Graphitic Carbon nitride (g-CN) possesses properties that make it suitable for various applications, including photocatalysis, carbon dioxide reduction, sensing, water-splitting, and nitrogen fixation. To overcome performance limitations arising from limited visible-light absorption and rapid recombination of photo-induced charge carriers, we employ heteroatom doping in a minimally impactful manner. Boron as a dopant is chosen due to its electron deficiency compared to carbon and nitrogen, enabling the replacement of either element in the g-CN backbone and influencing the optical properties of boron-doped carbon nitride (BCN). Our investigation reveals that the replacement of carbon atoms by boron within the g-CN framework influences the BCN's optical bandgap, 1.67 eV to 2.52 eV, and effectively modulates electron-hole recombination. Further, using different synthesis approaches and boron precursors results in vastly different material morphology. The boron-doping-induced structural defects lead to bandgap energy reduction. However, we find that this reduction does not correlate with the suppression of electron-hole recombination. In addition to studying physicochemical properties that underline BCN functional performance across a wide range of energy and environmental applications, we compare the environmental impacts across the multiple BCN syntheses. This assessment encompasses a comparison across nine TRACI midpoint impact categories, such as global warming potential, human health impacts from carcinogenic and noncarcinogenic substances, and ecotoxicity. Our life cycle impact assessment results demonstrate that electricity is the major contributor to the overall impacts of BCN synthesis, regardless of the synthesis technique used. Further, we propose a MAPS, Material Properties and Sustainability, evaluation approach that simultaneously considers physicochemical properties and sustainability metrics to gain valuable insights into designing minimally impactful, high-performing carbon nitride materials.

Keywords: green synthesis, nanotechnology, life cycle assessment, doped carbon nitride, sustainability tradeoffs

1.0 Introduction

Nanotechnology underpins technological advances of tomorrow with wide-ranging impacts in the fields of electronics and computing, healthcare, energy and environment, material science, and manufacturing. However, the versatile functionality of nanotechnology demands complementary research on the environmental impact, health and safety risks, and scalable transitioning from lab-scale to large-scale production of nanomaterials. With the global nanotechnology market projected to reach an impressive USD 23.6 billion by 2026¹, it is imperative to investigate the emerging physicochemical properties of nanomaterials alongside the corresponding environmental impact of the processes involved in their synthesis and manufacturing.

Graphitic carbon nitride (g-CN) is a promising nanomaterial with expanding applications due to its favorable physicochemical properties², such as metal-free composition³, tunable structure^{4,5}, biocompatibility⁶, chemical inertness⁷, and visible-light response⁸⁻¹⁰. The current and future applications of carbon nitride span photocatalysis^{3,7,9}, hydrogen generation¹¹⁻¹³, carbon dioxide reduction⁷, and sensing⁴. In its pristine form, carbon nitride suffers from a few drawbacks, such as limited bandgap for visible-light absorption and rapid recombination of photo-induced electron-hole pairs⁸. Collectively, these drawbacks negatively influence the charge transfer and generation of radical oxidative species, which are crucial for carbon nitride's efficacy in the above-listed applications.

There are several approaches to improve g-CN properties, including doping with metals/non-metals⁵, synthesizing nanocomposites with graphene¹⁴⁻¹⁶, and surface functionalization^{17,18}. Among the suggested approaches, elemental doping with heteroatoms is considered an effective strategy to modulate the optical and electronic properties of carbon nitride^{5,19}. Heteroatom doping is a promising approach since it is metal-free, allows adjustment of visible-light activation, and enhances catalytic activity due to modification of the g-CN electronic structure and surface properties^{5, 20}. In this study, we investigate boron atom doping because boron is a p-type semiconductor dopant that can introduce positive charge carriers into the g-CN framework to facilitate the flow of electric current^{21,22}. Moreover, boron has also been found effective in regulating the band structure²³⁻²⁵ that can influence the light absorption and emission characteristics of g-CN. This motivates us to evaluate the optimal amount of B that could be doped

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3 into the g-CN framework, its influence on the optical and electronic properties, and its correlation
4 with various boron-doped carbon nitride (BCN) synthesis approaches.
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8 There are various strategies to synthesize graphitic carbon nitride and BCN, with supramolecular
9 and non-supramolecular approaches being widely popular. The supramolecular approach of
10 synthesis involves molecular self-assembly of melamine and cyanuric acid and has demonstrated
11 improved local order, charge separation, and electrical conductivity for g-CN^{26,27}. The non-
12 supramolecular (NS) approach involves direct calcination of g-CN precursors such as melamine,
13 urea, and dicyandiamide^{25,28,29}. While both approaches to molecular assembly have been pursued,
14 there remains an opportunity to realize how the synthesis approach influences BCN
15 physicochemical properties. Herein, we report a methodology we developed that adopts the
16 supramolecular approach for BCN synthesis using boric acid. In addition, we synthesize BCN
17 using the NS approach with two different boron precursors, boric acid and ammonia borane. Given
18 the diverse strategies for synthesizing BCN and the range in resultant physicochemical properties,
19 we take the opportunity to demonstrate how we couple material design with environmental impact
20 assessment, using life cycle assessment (LCA), providing guidance on sustainable BCN
21 development. This investigation into BCN physicochemical properties that emerge from different
22 synthesis approaches, in conjunction with a sustainability assessment of the associated BCN
23 production, demonstrates a unique methodology that we propose to importantly inform the future
24 development of a promising visible light photocatalyst in its wide-ranging applications.
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39 **2.0 Experimental**

40 *2.1. Chemicals*

41 Boric acid (BA, 99%), cyanuric acid (CA, 98%), and dicyandiamide (DCDA, 99%) were
42 purchased from Sigma-Aldrich (St. Louis, MO). Melamine (M, 99%) and borane-ammonia
43 complex (AB, >85%), Ethanol (190 proof), were purchased from Acros Organics B.V.B.A. (Geel,
44 Belgium), TCI chemicals (Tokyo, Japan), and Decon Labs Inc. (King of Prussia, PA), respectively.
45 All reagents employed in the synthesis were used without further purification.
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53 *2.2. Synthesis of the supramolecular boron-doped carbon nitride (S-BCN) sample set*

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3 X mmol ($X = 0, 1, 2, 3$) of BA was stirred at 600 rpm with 1 g of melamine and 1.02 g of cyanuric
4 acid (M:CA molar ratio = 1:1) in a parafilm-covered beaker with 40 ml ethanol for 3 h at room
5 temperature. Next, the mixture was sonicated inside the same beaker using a bath sonicator
6 (Aquasonic Model 150HT) for 1 h. Subsequently, the sonicated mixture was dried overnight at 80
7 °C. After drying, the resultant powder was ground using mortar and pestle and transferred to a
8 quartz crucible for calcination inside the muffle furnace (Thermo Scientific Lindberg/Blue M
9 TF55035A-1). The muffle furnace was heated to 500 °C with a ramp rate of 2.5 °C/min. Once the
10 maximum temperature was reached, the powder sample was calcined for 3 h under the N₂
11 (prepurified gas 99.9985%, Matheson UN1066) gas flow (rate 100 mL/min), after which the
12 furnace was turned off and cooled naturally. After reaching room temperature, the crucible was
13 removed, samples were weighed, and stored in glass vials in the dark. The yield obtained for this
14 synthesis scheme was 24.03%.

25 26 *2.3. Synthesis of the non-supramolecular boron-doped carbon nitride from boric acid (NS-BCN- 27 BA) sample set*

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29 BCN was synthesized based on the approach demonstrated by Peng et al.²⁵ Briefly, X mmol ($X =$
30 0, 1, 2, 3, 6) of BA was stirred at 600 rpm with 1 g of melamine in a beaker with 10 ml deionized
31 (DI) water for 3 h at room temperature. The mixture was dried overnight at 100 °C. The resultant
32 powder was ground using mortar and pestle and transferred to an alumina crucible for calcination
33 inside the muffle furnace (*vide supra*). The muffle furnace was heated to 550 °C at a ramp rate of
34 5 °C/min. The powder was calcined for 4 h at 550 °C under the Air (zero gas, Matheson UN1002)
35 gas flow (rate 100 ml/min), after which the furnace was turned off and was cooled naturally. The
36 powder sample was collected from the crucible, washed with water while on filter paper, dried,
37 ground, and stored in glass vials in the dark. The yield achieved for this synthesis method was
38 61.30%.

39 40 41 42 43 44 45 46 47 48 *2.4. Synthesis of the non-supramolecular boron-doped carbon nitride from ammonia borane (NS- 49 BCN-AB) sample set*

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51 BCN was synthesized from ammonia borane following the method by Wang et al.³⁰ Briefly, X g
52 ($X = 0, 0.02, 0.1, 0.2$, equivalent to 0, 0.64, 3.20, and 6.40 mmol, respectively) of AB was added
53 to 10 ml deionized water in a beaker and stirred for 5 min. (Note: mass, g, was used instead of
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3 mmol to keep the synthesis consistent with Wang et al. approach.) After stirring, 2 g of DCDA
4 was added to the aqueous solution and stirred for 1 h at 600 rpm and room temperature. The
5 mixture was heated at 100 °C overnight to evaporate the solvent. The resultant powder was then
6 ground using mortar and pestle and transferred to a quartz crucible for calcination inside the muffle
7 furnace. The calcination was carried out at 600 °C (ramp rate of 2.39 °C/min) for 4 h under N₂ gas
8 flow (rate 100 ml/min), after which the furnace was turned off and cooled to room temperature
9 naturally. The calcined powder was ground and stored in glass vials in the dark. The yield observed
10 for this synthesis technique was 21.78%.
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19 *2.5. Material Characterization*

20 Powder X-ray diffraction (XRD) was used to confirm the crystal structure of graphitic carbon
21 nitride. Samples were analyzed using a Bruker D8 Discover instrument (Cu Ka, 40 kV, 40 mA).
22 The chemical compositions of the synthesized BCN samples were analyzed by X-ray
23 photoelectron spectroscopy (XPS) using an ESCALAB 250Xi (Thermo Scientific) instrument with
24 a monochromatic Al Ka X-ray source (1486.7 eV, spot size of 650 mm). Optical band gap data
25 were collected with a LAMBDA-750 UV-vis-NIR spectrophotometer (PerkinElmer L750)
26 equipped with a 60 mm integrating sphere using barium sulfate as the standard reference. Steady-
27 state photoluminescence (ss-PL) was used to determine the charge separation efficiency of BCN
28 samples. Spectra were measured on a FluoroMax-3 spectrometer (Jobin Yvon Horiba) with
29 excitation at carbon nitride's characteristic excitation wavelength of 330 nm.³¹ Transmission
30 electron microscopy (TEM) imaging with electron energy loss spectroscopy (EELS) for elemental
31 mapping was performed on a JEOL JEM-2100F TEM instrument operated at 200 kV. Sample
32 preparation for all the characterization techniques has been discussed in the Supplementary
33 Information (**Section S1**).
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46 *2.6. Sustainability Assessment of Syntheses*

47 *2.6.1. Scope, System Boundary, and Functional Unit*

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50 A cradle-to-gate life cycle impact assessment was conducted for the three BCN syntheses. This
51 includes all inputs, emissions, and transportation associated with raw material acquisition through
52 the production of the BCN (e.g., upstream resource extraction, energy required for producing
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3 reagents, and direct emissions from energy generation). We adopted a mass-based functional unit
4 of one gram of BCN synthesized, instead of a performance-based functional unit, for two reasons:
5 (i) the focus of this work and intended use of the life cycle assessment results is to inform material
6 synthesis-property tradeoffs, and (ii) BCN is proposed for use in many energy and environmental
7 applications (e.g., water disinfection, water splitting, biosensing, carbon dioxide reduction), each
8 having distinct use cases and thus, different associated functional units. Our intent is to remain as
9 broadly applicable as possible, and the mass-based functional unit is most transferrable to future
10 studies and other researchers.
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19 *2.6.2. Impact Assessment*

20 A life cycle impact assessment was conducted on the three BCN synthesis routes described above
21 (refer to **Section S2** and **Tables S1-S5** for the inventory details of each synthesis route). The impact
22 assessment was performed using SimaPro 9.1 (PRé Consultants). The United States Life Cycle
23 Inventory (USLCI)³² was chosen when available; otherwise, Ecoinvent 3 was selected³²⁻³⁴.
24 Environmental impacts were modeled using the United States Environmental Protection Agency's
25 Tool for the Reduction and Assessment of Chemical and Other Environmental Impacts (TRACI)
26 2.1 assessment method³⁵. Environmental impact categories include global warming potential
27 (GWP, in kg CO₂ eq), photochemical smog formation (PS, in kg O eq), acidification (AC, kg SO
28 eq), eutrophication (EU, kg N eq), human health impacts from toxic carcinogenic (HHC) and
29 noncarcinogenic (HHNC) substances (in comparative toxic units for human toxicity impacts, or
30 CTUh), respiratory effects (RE, in kg PM_{2.5} eq), ecotoxicity (EC, in comparative toxic units for
31 aquatic ecotoxicity impacts, or CTUe), and fossil fuel depletion (FF, in MJ surplus).
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43 *2.6.3. Uncertainty Analysis*

44 Uncertainty analysis was performed using Monte Carlo simulation (1,000 runs, SimaPro 9.1) to
45 assess the uncertainty of the cumulative unit process LCI data associated with each synthesis route.
46 For unit processes available in the selected databases (i.e., USLCI, Ecoinvent), the default
47 lognormal distributions were used. For unit processes created by the authors, the uncertainty
48 factors are calculated for each input and output data considering a lognormal distribution and
49 utilizing the Pedigree matrix approach (refer to **Section S3** and **Tables S1-S5** for uncertainty plots
50 and for the standard deviations generated using pedigree matrix approach, respectively).
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3.0 Results and Discussion

3.1. Modulating boron doping of carbon nitride

3.1.1. Confirming the characteristic tri-s-triazine structure formation

Density functional theory studies demonstrate that the tri-s-triazine structure (**Figure 1**) is the most stable among all the allotropes of g-CN, and thus, it is recognized as its fundamental building block.^{36,37} The tri-s-triazine structure consists of three nonequivalent nitrogen atoms (N1, N2, N3) and two nonequivalent carbon atoms (C1, C2). We employed X-ray photoelectron spectroscopy (XPS) to verify the presence of the tri-s-triazine bonding moieties. The C1s spectra of all synthesized BCN samples (shown in **Tables S6-S8** under the C1s column) exhibit peaks at 284.4 eV (C-C), representing adventitious carbon impurities introduced during synthesis and XPS sample preparation.^{38,39} Additionally, peaks were observed at 288 eV (C1-N2=C2)^{27,30,40}, along with some peaks above 290 eV identified as satellite peaks²⁷. The N1s BCN spectra (shown in **Tables S6-S8** under the N1s column) show peaks at 398.5 eV (C1-N2=C2)^{30,31,38}, 399.5 eV (N-H, pyrrolic nitrogen)^{36,37}, and 400.1 eV (N3, graphitic)^{36,37}, along with peaks above 401 eV, which were identified as satellite peaks.^{41,42} The presence of satellite peaks (in the N1s spectra) indicates the phenomenon of pi-pi electron excitation, characteristic of the pi-conjugated structure in the tri-s-triazine ring of g-CN⁴³. Thus, regardless of the synthesis approach adopted, XPS indicates the attainment of the tri-s-triazine backbone.

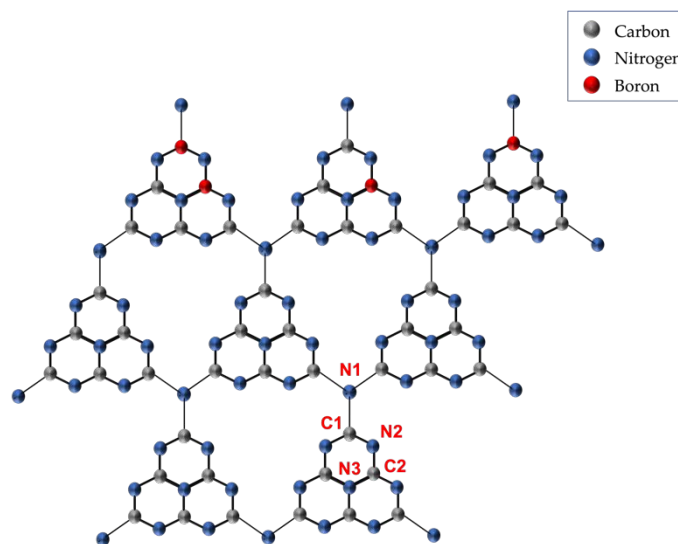


Figure 1. The tri-s-triazine structure for graphitic carbon nitride with possible boron doping sites indicated (red atoms).

3.1.2. The extent of boron doping varies with the synthesis approach

The boron doping atomic percentages were determined using XPS as the average of three different locations on the BCN sample (refer to **Section S1** for sample preparation and characterization details). The synthesis-specific boron atomic percentages are listed in **Table 1**. Interestingly, when the same amount of dopant (molar ratio) was used during synthesis, significantly lower B atomic percentages were observed for the SBCN samples compared to the NSBCN-BA samples. Shalom et al. show that during the SBCN synthesis, the formation of the melamine-cyanuric acid complex as an intermediate is stabilized by hydrogen bonds²⁶. This stabilization can have two effects: (i) allowing fewer boron atoms to enter the tri-s-triazine framework, and/or (ii) exposing fewer melamine molecules to react with boron atoms. These factors could lead to the lower observed boron doping in the SBCN samples (**Table 1**).

Table 1. Boron doping atomic percentages and bandgap energies (in eV) for the three BCN sample sets.

SBCN			NSBCN-BA			NSBCN-AB		
Sample	B% (avg ± stdev)	BG (eV)	Sample	B% (avg ± stdev)	BG (eV)	Sample	B% (avg ± stdev)	BG (eV)
0 mmol	0	2.52	0 mmol	0	2.33	0 mmol	0	1.94
1 mmol	0.75 ± 0.01	2.57	1 mmol	4.88 ± 0.98	1.86	0.64 mmol	4.84 ± 0.42	1.94
2 mmol	1.33 ± 0.23	2.34	2 mmol	5.50 ± 0.25	1.82	3.20 mmol	8.13 ± 0.45	1.74
3 mmol	1.61 ± 0.29	2.53	3 mmol	7.90 ± 0.45	2.01	6.40 mmol	17.41 ± 0.11	1.67
			6 mmol	22.85 ± 0.17	2.80			

3.1.3. Boron replaces carbon atoms in the tri-s-triazine structure

The B1s peak deconvolution reveals a single peak in the range of 191.7-192.2 eV in all samples (**Figure 2**) corresponding to B-N bonds^{25,44}. This observation indicates that the B atom replaces the C atom (**Figure 1**) (note: it remains unclear if the replaced C atom(s) were C1, C2, or both because XPS only confirms the presence of the B-N bond in the B1s spectra and cannot resolve the location-specific differences for similarly hybridized atoms) in the g-CN tri-s-triazine structure. The broader and less intense peak observed in the SBCN sample (**Figure 2a**) could be a result of lower boron doping (see **Table 1**). Additional peaks were observed at 188 eV and 197

eV, particularly for the NSBCN-AB (**Figure 2c**) samples. A previous study⁴⁵ suggests that the B1s peak at 188 eV represents B-C bonds. Since this B-C peak is observed in the highest doped sample of the NSBCN-AB sample set (B atomic % = 17.41, in the case of 6.4 mmol NSBCN-AB), it is possible that boron is replacing some nitrogen atoms in the g-CN framework. A 6 mmol NSBCN-BA sample was synthesized to investigate whether a similar trend was observed with an increase in doping and for the presence of a B-C peak. The same trend was not observed, nor was the B-C peak (**Table S7**, B1s column). We believe this difference emerged from the use of different precursors (melamine and boric acid versus dicyandiamide and ammonia borane) involved in the NSBCN-BA synthesis. The peak at 197 eV is possibly a satellite peak⁴⁶, resulting from shake-ups, energy loss, plasmons, and electron excitation.

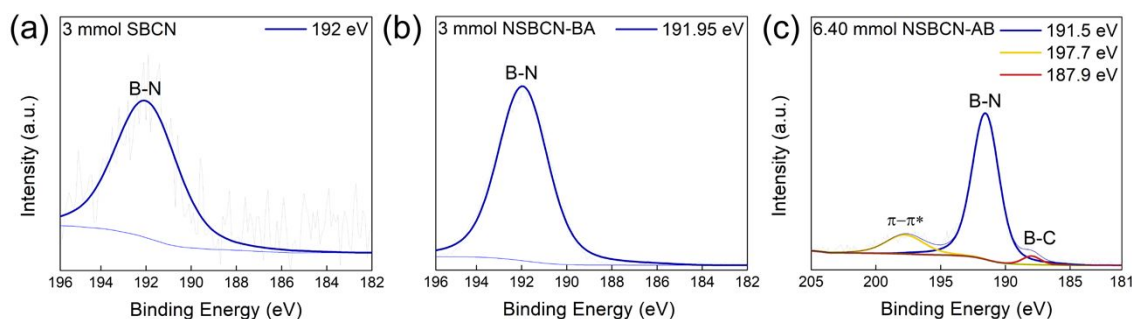


Figure 2. B1s peak deconvolution in (a) 3 mmol SBCN, (b) 3 mmol NSBCN-BA, and (c) 6.4 mmol NSBCN-AB. These samples were chosen as they are the highest-doped samples (considered for further characterization) from each synthesis route and showcase the maximum possible extent of boron doping in this study. All other B1s spectra are included in **Tables S6-S8**, under the B1s columns.

3.2. Boron atoms distort the interlayer stacking

The crystal structure of the synthesized samples was investigated using XRD. The obtained diffractograms for the three sample sets (**Figure 3**) indicate the presence of the characteristic peaks for carbon nitride at $2\theta = 13^\circ$ (representing the tri-s-triazine unit^{27,30,38,44}) and 27° (representing interlayer stacking^{27,30,38,44}). The changes in these characteristic peaks, including peak broadening and diminished intensity, are indicative of interlayer distortion (e.g., exfoliation, reduced layer thickness). For SBCN samples (**Figure 3a**), the 13° peak disappears at B precursor amounts greater than 1 mmol, and the intensity of the 27° peak decreases with increasing doping levels. Similar trends are seen for the non-supramolecular sample sets NSBCN-BA (**Figure 3b**)

and NSBCN-AB (**Figure 3c**). The reduction in peak intensities is attributed to boron atoms interfering with the two-dimensional, layered carbon-nitride structure, which deforms the interlayer stacking order. The XRD results of the supramolecular sample set show relatively invariable peak intensity for the 27° peak across the sample set. This suggests a higher preserved local order of the pristine g-CN structure. The higher-order is attributed to the tendency of SBCN's melamine-cyanuric acid complex (formed *in situ* during calcination)²⁶ to comprise three melamine molecules in the same plane in the formation of the tri-s-triazine unit. There is a consistent peak distortion of the 27° peak in the NSBCN-BA sample set with an increase in doping. There is less distortion in the 27° peak until the 3.20 mmol of the NSBCN-AB sample set. These varying observations across the two non-supramolecular approaches imply the possible role of different precursors (i.e., melamine and boric acid in NSBCN-BA versus dicyandiamide and ammonia borane in NSBCN-AB) in the way boron doping interferes with the interlayer stacking of carbon nitride.

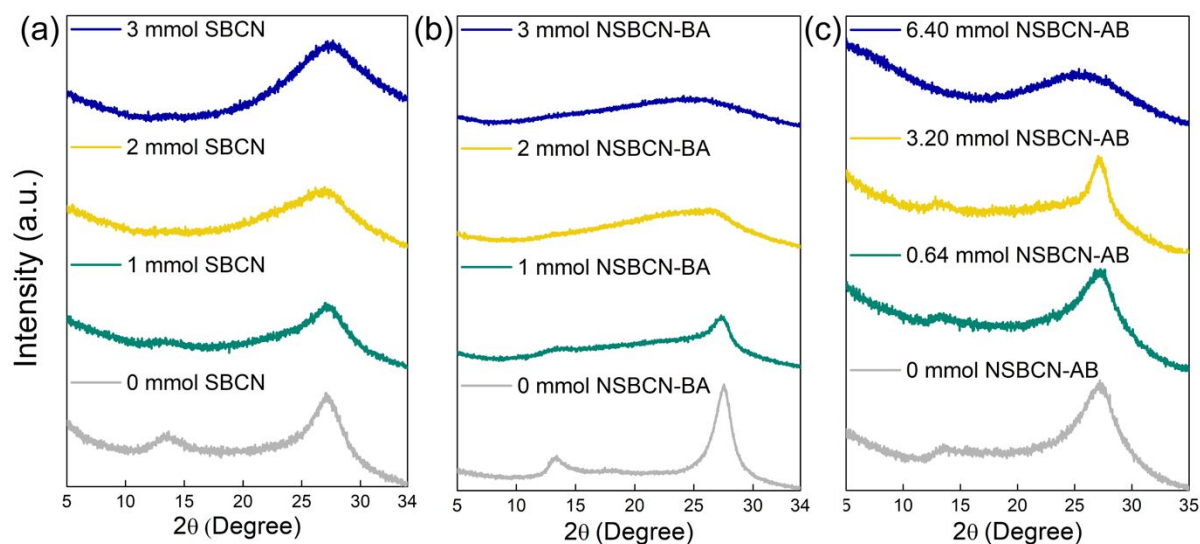
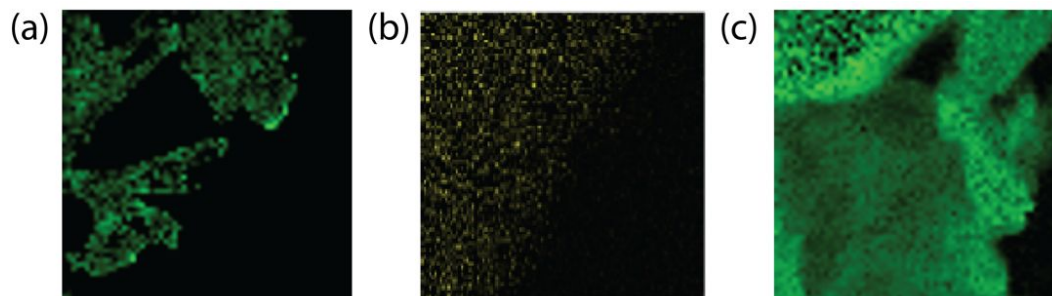


Figure 3. XRD diffractograms for (a) S-BCN set, (b) NS-BCN-BA set, (c) NS-BCN-AB set

3.3. Synthesis method influences BCN morphology

Transmission electron spectroscopy (TEM) illuminates the morphological characteristics of the BCN samples synthesized under different conditions, and electron energy loss spectroscopy (EELS) reveals the elemental distribution across the synthesized samples (**Figure 4**). The supramolecular approach (SBCN, using boric acid as B precursor) resulted in exfoliated samples with an increase in porosity as the molar ratio of the B dopant increases (**Figure 5 a, b**). Porosity

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3 in g-CN emerges from the release of gases, such as CO₂, NH₃, and CO, during the high-temperature
4 calcination step of the synthesis⁴⁷. Increased porosity for BCN has been observed by Mahveladi-
5 Shamsabadi, et al.⁴⁸, and is believed to emerge from the decomposition of the boric acid molecule
6 into H₂O gas. Increased porosity can provide more active sites on the BCN surface for chemical
7 reactions and facilitate efficient transfer of charges between the reaction molecules, particularly in
8 applications such as photocatalysis^{47,49}. The non-supramolecular synthesis with ammonia borane
9 (NSBCN-AB, **Figure 5e, f**) resulted in a similar porous morphology, having a curled sheet
10 structure with round edges. For the non-supramolecular synthesis with boric acid (NSBCN-BA,
11 **Figure 5c, d**), the TEM analysis reveals a different morphology resembling a sheet structure with
12 distinct edges. The stacked nanosheet thickness appears to increase (less transparent) with
13 increased doping, which is corroborated by the XRD results. The observed peak distortion for the
14 $2\theta = 27^\circ$ peak in the NSBCN-BA and NSBCN-AB samples with increased B doping % indicates
15 deformed interlayer stacking, which can influence the observed nanosheet thickness.
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36 **Figure 4.** EELS of (a) 3 mmol SBCN, (b) 6 mmol NSBCN-BA, (c) 6.40 mmol NSBCN-AB.

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38 Green indicates B atom location within the BCN sheet.
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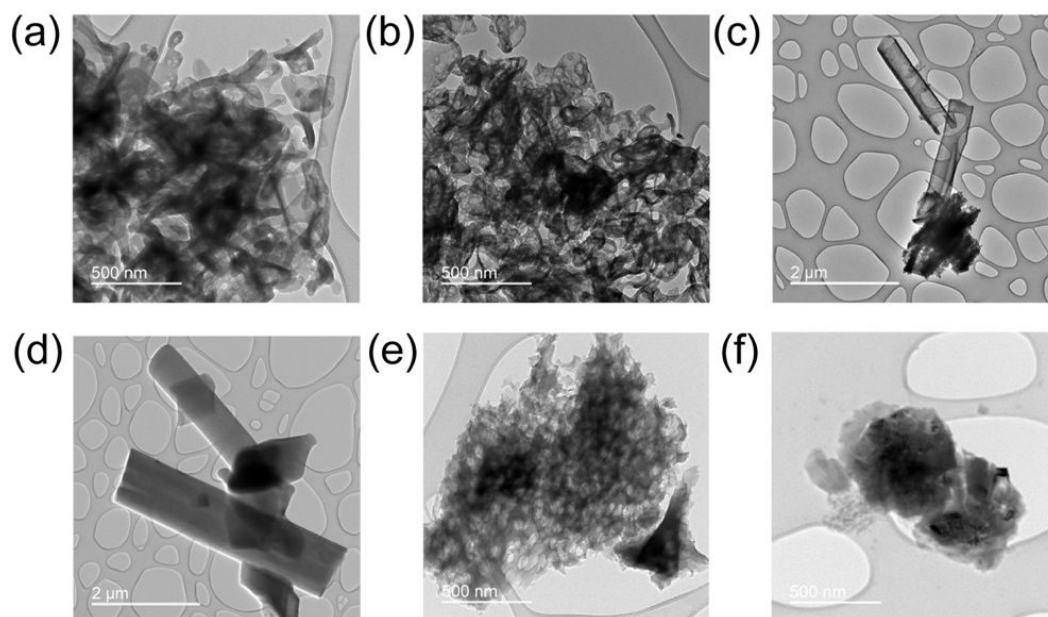


Figure 5. TEM images of: (a) 0 mmol S-BCN, (b) 3 mmol S-BCN, (c) 0 mmol NSBCN-BA, (d) 6 mmol NSBCN-BA, (e) 0 mmol NSBCN-AB, (f) 6.40 mmol NSBCN-AB. Note: the scale bar for (e) and (f) is 500 nm

3.4. Boron doping influences the optical properties of carbon nitride

3.4.1. Modulating boron atomic percentages influence the optical bandgap

The optical bandgap (or band gap energy, E_{gap}) of a semiconducting material is the energy required to transition an electron from its valence band to the conduction band. Upon reaching the conduction band, the electron can move freely in the crystal lattice and serve as a charge carrier. The E_{gap} (in eV, **Table 1**) of the synthesized samples was determined from measured UV-vis-DRS data via Tauc plots (**Figure S2**). There is a negligible change in E_{gap} observed in S-BCN samples (**Table 1**), despite there being a minor difference of 0.86 atomic % (between 0 mmol SBCN and 3 mmol SBCN) doping. There is a gradual decrease in the band gap of the NSBCN-BA sample set with increased doping until 3 mmol, when the E_{gap} increases. This increasing trend is maintained for the 6 mmol sample. The bandgap increment for higher B doped samples (i.e., 3 mmol and 6 mmol NSBCN-BA) is on account of the quantum confinement effect introduced by the presence of higher B atoms in the g-CN framework.³⁹

Conversely, there is a consistent decreasing trend of optical bandgap observed for the NS-BCN-AB samples. The band gap of inorganic semiconductor materials has been found to narrow with

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3 increasing structural defects⁵⁰. Additionally, our PL spectra (*vide infra*, **Figure 6**) of the NSBCN-
4 AB sample set reveal a difference in photoluminescence peak intensities (i.e., intensity increases
5 for NSBCN-AB samples) with increasing B doping, which is influenced by the structural defects
6 generated⁵⁰. This suggests that the lowering of bandgap in the NSBCN-AB sample set is due to
7 the doping-induced structural defects. The optical bandgap range observed in this work is
8 compared to the BCN optical bandgap ranges in some previous research studies (**Table 2**). An
9 important advancement in our work, synthesizing BCN using various approaches, is the
10 observation of the lower bound of the bandgap range shifting further into the region conducive to
11 visible light.
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19 **Table 2:** Summary of bandgaps found in the B-doped carbon nitride literature

Study	Boron Precursors	Optical bandgap range (in eV)
BCN for photocatalytic UO ₂ ²⁺ reduction ⁵¹	Boric acid	2.49 – 2.57
BCN for photocatalytic degradation of organic dyes ⁵²	Diboron trioxide	2.61 – 2.75
BCN for photocatalytic water splitting ⁵³	Ammoniotrihydroborate	2.55 – 2.68
BCN for photocatalytic NO _x removal ⁴⁴	Boric acid	2.54 – 2.70
BCN synthesized in this work	Boric acid and Ammonia borane	1.67 – 2.52

3.4.2. The approach to boron doping differentially influences electron-hole recombination

36 Charge separation reflects the ability of electrons and holes to remain separated. This is essential
37 for photocatalytic applications of g-CN, where electrons and holes play a critical role in generating
38 reactive oxidative species. Photoluminescence (PL) spectroscopy elucidates the charge separation
39 efficiency. The intensity of PL peaks signifies the intensity of light emission exhibited by the
40 photon released when an electron and hole pair undergo recombination. The greater the peak
41 intensity, the greater the electron-hole recombination.²⁷
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50 The PL peak intensities for SBCN (**Figure 6a**) and NSBCN-AB (**Figure 6c**) do not trend with
51 atomic %B. Notably, the 1 mmol SBCN sample and the 6.40 mmol NSBCN-AB sample exhibit
52 reduced recombination compared with the other samples in their respective sets, as indicated by
53 the lower-than-expected intensity. For SBCN, the multiple PL runs (i.e., three independent
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attempts to disperse each sample in a solvent and measuring PL intensities every time) returned different trends in relative intensity, suggesting heterogeneity in doping across the BCN samples prepared via the supramolecular method. For the 6.40 mmol NSBCN-AB sample, we believe the departure from the trend could emerge from the saturation of doping sites or the amount of ammonia borane is so high that it uniquely interferes with the tri-s-triazine bond formation.

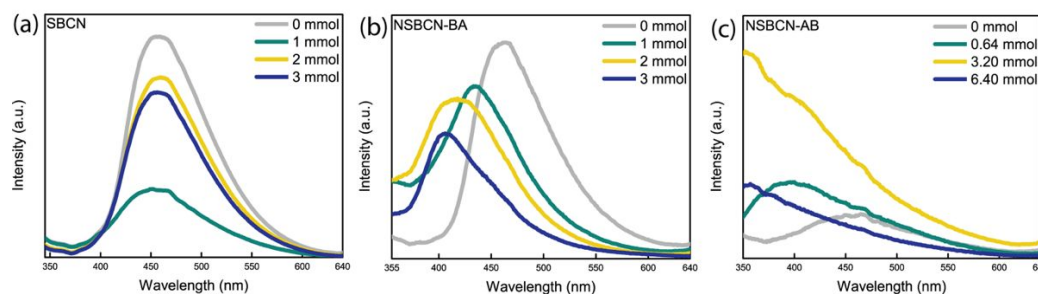


Figure 6. Photoluminescence spectra for: (a) S-BCN, (b) NSBCN-BA, (c) NSBCN-AB

There are notable blue-shifts (i.e., peak shifts towards lower wavelengths) in the non-supramolecular sample sets: NSBCN-BA (**Figure 6b**) and NS-BCN-AB (**Figure 6c**). The blue shift is ascribed to the interlayer strain and the quantum confinement effect induced by boron doping.^{39,54} The PL intensity of NSBCN-BA is observed to decrease with an increase in atomic %B, suggesting relatively low recombination with increased doping. This differing trend in recombination could emerge from the differences in morphology with the distinct edges providing charge-trapping or recombination sites (**Figure 5 c-d**). Combined, these observations illuminate the importance of material structure variations generated during synthesis, and influenced by different doping regimens, on photocatalytic properties of g-CN.

3.5. Life cycle assessment to capture environmental impacts of different synthesis approaches

With growing research interest in g-CN, there is a critical opportunity to inform and direct its development in a way that is minimally impactful. Life cycle impact assessment (LCIA) is an established methodology that can be used to quantify environmental impacts associated with any material. Previous LCIA of engineered nanomaterials identify energy demand from raw material extraction and synthesis processes as the predominant contributor to environmental impacts⁵⁵⁻⁵⁷. Given that we achieve a range of physicochemical properties through B doping and use different synthesis methods, including varying processes and precursors, we employ LCIA to (i) identify high-impact components of BCN production, (ii) determine environmental impact tradeoffs of our synthesis routes, and (iii) elucidate possible avenues to reduce the environmental footprint. Taking

these cradle-to-gate (i.e., raw material acquisition through synthesis) impacts and the physical-chemical properties of each BCN material together, we can quantitatively compare the performance and environmental tradeoffs.

LCIA was conducted for the production of 1 g of BCN via the two synthesis approaches and two boron precursors (Figure 8). The synthesis processes were divided into four categories: electricity to power the reactor, chemical reagents, precursors, and emissions. Each was analyzed according to the nine TRACI impact categories: (i) Global Warming Potential (GWP); (ii) Photochemical Smog (PS); (iii) Acidification (AD); (iv) Eutrophication (EU); (v) Carcinogenics (CA); (vi) Non carcinogenics (NC); (vii) Respiratory Effects (RE); (viii) Ecotoxicity (EC); (ix) Fossil Fuel Depletion (FF). Since each category has a unique unit, the impacts are presented for each category as the relative contribution of the four categories (**Figure 7a-c**). The combined results indicate that synthesis-related energy demand is the greatest contributor to the impacts regardless of the approach. Further, the approaches are plotted together on a relative basis to compare the total impacts of the three synthesis methods.

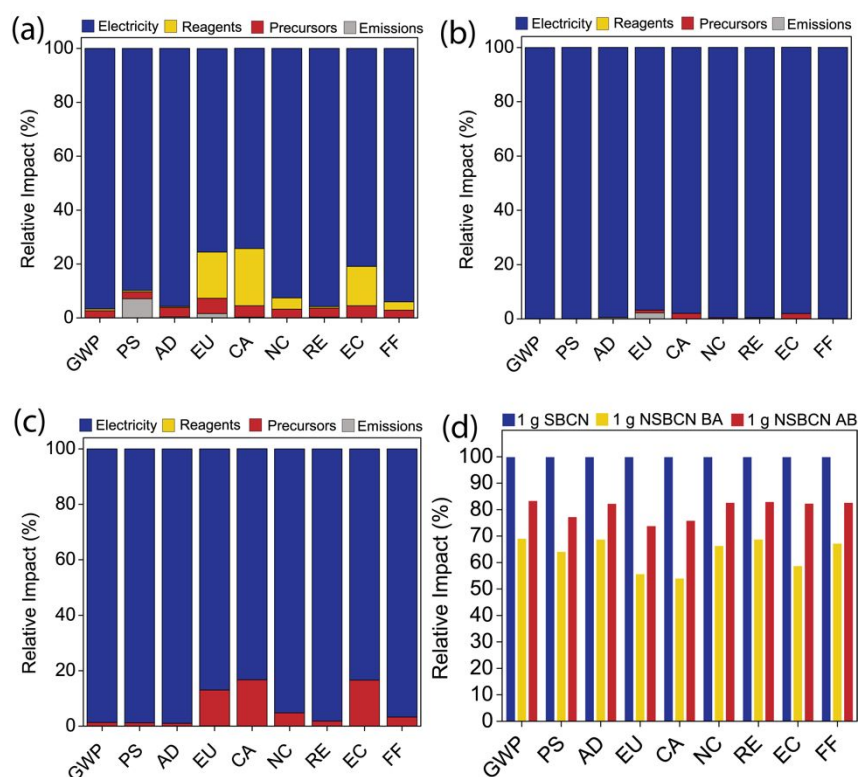


Figure 7. Process contributions associated with the nine TRACI impact categories to produce 1g of (a) 1 mmol SBCN, (b) 1 mmol NSBCN-BA, and (c) 0.64 mmol NSBCN-AB; and (d) the relative impact assessment for synthesizing 1 g of BCN using the three synthesis methods.

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5 Across the three synthesis methods, electricity contributes the greatest percent contribution
6 (>87%) to the overall impact for every TRACI impact category. The electricity source assumed in
7 this analysis was ‘Electricity country mix – Low Voltage – Electricity, at grid, US/US’. This
8 electricity source consists of electricity generation from multiple fuel sources: biomass, coal,
9 petroleum, geothermal, natural gas, nuclear, solar, hydroelectric, and wind energy sources. An in-
10 depth analysis of electricity impacts in SimaPro revealed that sources such as bituminous coal,
11 natural gas, and lignite coal in the mix were the primary impact components. While the energy
12 demand per mass of BCN produced may decrease as production scales up⁵⁸⁻⁶⁰, energy demand is
13 likely to remain the predominant component of material synthesis impacts. Thus, a potential
14 strategy to reduce the impact of energy demand during BCN syntheses is to adopt an electricity
15 source with minimal fossil fuel.
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26 The reagents are another contributor to the overall impacts, particularly in SBCN synthesis (**Figure**
27 **7a**). The solvents used in the syntheses are ethanol (SBCN, **Figure 7a**) and deionized water
28 (NSBCN-BA, **Figure 7b**, and NSBCN-AB, **Figure 7c**). The range in percent contribution
29 attributed to ethanol (the only reagent used in this synthesis) in SBCN synthesis varies between
30 3%-17% in the EU, HHC, HHNC, EC, and FF categories. Although ethanol is considered a green
31 solvent⁶¹⁻⁶³, these results – comparing **Figure 7a** to **7b-c** - importantly highlight the benefit of
32 using water, whenever possible, as the solvent used in chemical processes.
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40 Among the three synthesis approaches, SBCN production has the highest relative impact across
41 all the TRACI impact categories considered in this investigation (**Figure 7d**). The high impact of
42 SBCN production emerges from the utility of an extra precursor (i.e., cyanuric acid, which was
43 employed to form the melamine-cyanuric supramolecular complex) and an additional processing
44 step (i.e., bath sonication). Moreover, for impact categories such as smog formation and
45 eutrophication potential in the SBCN production (**Figure 7a**), a considerable relative impact
46 contribution of synthesis emissions was observed. These synthesis emissions are likely on account
47 of isocyanic acid, which is an emission associated with cyanuric acid precursor employed in the
48 SBCN synthesis.
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3.6 Design of graphitic carbon nitride that considers physicochemical property and environmental impact metrics

With near infinite possibilities for synthesizing new materials, the knowledge informing how to narrow the design space to achieve desired properties for specific applications holds immense value. Equally critical is determining whether it is possible to achieve equivalent functional performance with a lower environmental impact through choices made at the material inception (i.e., at synthesis). We propose a Material Properties and Sustainability (or MAPS) approach (**Figure 8**), which includes simultaneous consideration of relevant environmental impact metrics (here, using the global warming potential, kg CO₂) and physicochemical properties of g-CN, such as bandgap energy (in eV) and photoluminescence light intensity (normalized arbitrary units, A.U.; details in **Section S6** for calculating the normalized arbitrary units for PL intensities), that underline photocatalytic performance. In this way, we compare g-CN options for achieving performance goals and elucidate environmental impact tradeoffs (or lack thereof) within the potential design space. To illustrate the versatility of these MAPS, we include g-CN materials synthesized and characterized by researchers for these same performance properties; one example of oxygen-doped carbon nitride (OCN)⁶⁴ (ref.) and carbon-doped carbon nitride (CCN)⁶⁵. The adaptability across functional properties is straightforward, given the universal metrics adopted for primary g-CN properties. We are unable to include these samples in the MAPS plots that include our environmental impact unit (GWP), which elucidates an opportunity and need within the advanced material research community (*vide infra*).

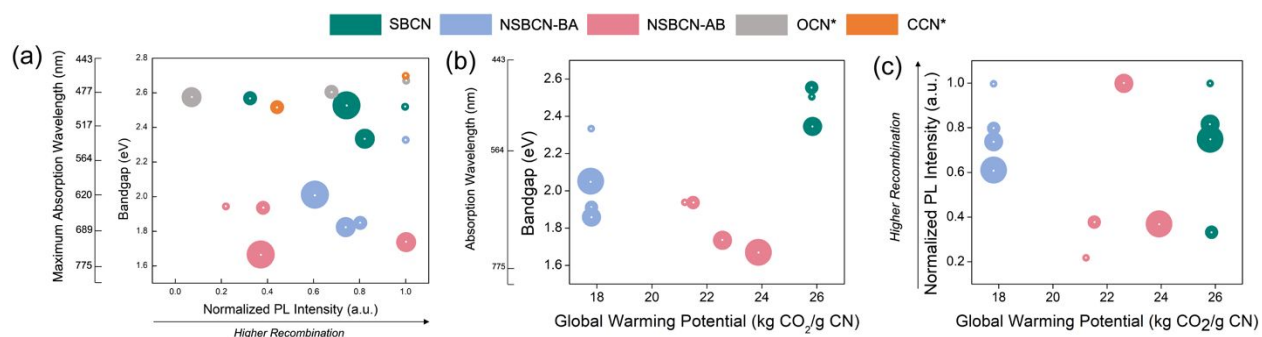


Figure 8. Example MAPS plots showing, (a) optical bandgap and photoluminescence intensity tradeoffs for the synthesized BCN samples sets along with OCN⁶⁴ and CCN⁶⁵ sample sets, (b) tradeoffs between optical bandgap with the global warming potential associated with different g-CN synthesis methods (this plot shares the maximum absorption wavelength y-axis scale with plot (a)), and (c) tradeoffs between the photoluminescence intensity and the synthesis method's global

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3 warming potential. The size of the colored circles is proportional to the heteroatom atomic %
4 dopant (B, O, or C). The center white dots represent the data point associated with the x-y
5 coordinates.
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10 The optical band gap, the associated maximum wavelength of absorption (nm), and the
11 photoluminescence peak are material properties that represent critical optical performance
12 properties for wide-ranging applications of g-CN. Lower photoluminescence intensity indicates
13 less electron-hole recombination, enhancing the potential for radical oxidative species generation,
14 which is critical for downstream reactions underlining photocatalysis and water splitting. Optical
15 band gap is directly related to the energy of light absorbed, thus reducing the bandgap allows for
16 absorbance of longer, lower energy, visible light. We have chosen GWP as a representative
17 environmental impact metric because it is commonly used across the LCA literature and, while
18 the magnitude of impacts change between categories, the relative impacts will be similar (plots
19 including additional environmental impact categories are included in **Figure S4**). Minimizing
20 GWP of synthesis is desired, as it indicates a lower environmental impact. Keeping the above
21 information in mind, the MAPS plots (**Figure 8**) provide insights into the associated tradeoffs and
22 thus, can assist in selecting g-CN materials that meet the desired criteria for specific applications
23 while informing the potential to simultaneously reduce the material's embodied resource footprint.
24 Specific constraints and desired threshold values for band gap and PL intensity will vary by
25 specific application. For many environmental applications, the lower band gap is desired to attain
26 efficiency at harvesting lower energy (higher wavelength) visible light (as opposed to high energy
27 UV, for example). While PL intensity on its own does not comprehensively represent the
28 intricacies of light interacting with g-CN to produce reactive oxygen species, in general, lower
29 recombination is desired. This would direct us to options in the lower left side of the MAPS,
30 **Figure 8a**. In this example, we would be choosing between three samples synthesized using the
31 non-supramolecular approach and ammonia borane precursor. The specific sample chosen would
32 depend on how the specific tradeoffs influence performance considering the entire system of a
33 given application.
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53 Considering these properties alongside GWP (**Figures 8b-c**) aids in further honing the optimal
54 alternative(s). Considering band gap and GWP (**Figure 8b**) there is a clear tradeoff if the lowest
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3 band gap energy is desired. The ammonia borane samples sacrifice approximately 4 kg CO₂/g CN
4 compared with the boric acid samples (NSBCN-BA). (Note: ammonia borane is not available in
5 SimaPro's databases, thus the process was established by the authors based on a patent⁶⁶, which is
6 why the GWP varies slightly compared with BCN samples prepared using reagents available in
7 the databases.) Considering electron-hole recombination, the 1 mmol and 2 mmol NSBCN-AB
8 outperform the alternatives. While the conclusions from these MAPS may ultimately lead to a
9 performance-environmental impact tradeoff, the material or product designer enters the
10 development process equipped with concrete guidance enabling an informed decision.
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18 Expanding these plots to include additional alternatives (e.g., with other heteroatom dopants, other
19 modifications to manipulate properties linked to material performance) will facilitate sustainable
20 development of g-CN applications, which pertain to clean energy harvesting, storage, water
21 decontamination, among others. Solving these grand challenges must be pursued with solutions
22 that are holistically sustainable to avoid shifting burdens between life cycle stages and unintended
23 consequences. One critical limitation precluding inclusion of additional g-CN samples (e.g., OCN
24 and CCN samples in **Figure 8a**) pertains to variability in the underlying assumptions across LCA
25 studies and lack of standardization in reporting across synthesis reports. For example, Aquino de
26 Carvalho, et al.⁶⁷ assumed 100% yield in their LCIA of various g-CN syntheses due to the lack of
27 yield data reported. Here, we include our actual yield for our BCN samples prepared, thus
28 preventing a fair GWP comparison across samples.
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39 **4.0 Conclusion**

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43 The demand for design and production of new materials is concomitant with technological and
44 societal advances. Further, materials will underline transformative solutions to our global grand
45 challenges and thus, must avoid introducing embodied and direct adverse implications on the
46 environment and human health. Advances in materials science and engineering thus, must co-
47 occur with considerations of their impact early in the design phase. In addition to reporting on
48 novel synthetic approaches to dope graphitic carbon nitride with varying degrees of B, the study
49 herein presents a framework for how these multiple objectives can be successfully pursued. Given
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3 the demonstrated ability to modulate properties critical for applications via B doping, this work
4 evokes new directions in g-CN design and synthesis.
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8 Synthesizing BCN using multiple approaches and dopant precursors illuminated that increasing
9 boron doping percentage introduces structural defects that modulate the optical properties of BCN.
10 These defects lead to bandgap energy reduction; however, we find that this reduction does not
11 correlate with the suppression of electron-hole recombination. SBCN sample set shows
12 inconsistency in variation of both bandgap and electron-hole recombination. NSBCN-BA shows
13 a decrease in band gap with doping along with reduced electron-hole recombination. NSBCN-AB
14 experiences a band gap reduction with doping, but an increase in electron-hole pair recombination.
15 These findings suggest that the structural defects introduced by doping can act as charge-trapping
16 sites in NSBCN-BA and recombination centers in NSBCN-AB. Exploring BCN morphological
17 variations shed light on possible explanations for the synthesis-induced variations in g-CN optical
18 properties. However, further investigations on the dynamic behavior of g-CN-associated
19 photocatalysis are encouraged by employing techniques such as time-resolved transient absorption
20 spectra analysis and transient PL spectrum.
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32 Our LCIA results demonstrate that electricity is the major contributor to the overall impacts of
33 BCN synthesis, regardless of the synthesis technique used. The results further emphasize the
34 importance of minimizing precursors and processing steps during g-CN synthesis to reduce the
35 associated environmental footprint. The introduced MAPS plot approach combines critical
36 material metrics for performance and environmental impact, allowing visualization and
37 quantitative tradeoff evaluation between material alternatives. This approach aims to guide the
38 community in making informed decisions about material design and synthesis.
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