



Cite this: *Nanoscale*, 2018, **10**, 7575

# MoS<sub>2</sub> nano flakes with self-adaptive contacts for efficient thermoelectric energy harvesting†

Qingqing Wu, Hatef Sadeghi \* and Colin J. Lambert \*

We examine the potential of the low-dimensional material MoS<sub>2</sub> for the efficient conversion of waste heat to electricity via the Seebeck effect. Recently monolayer MoS<sub>2</sub> nano flakes with self-adaptive Mo<sub>6</sub>S<sub>6</sub> contacts were formed, which take advantage of mechanical stability and chemical covalent bonding to the MoS<sub>2</sub>. Here, we study the thermoelectric properties of these junctions by calculating their conductance, thermopower and thermal conductance due to both electrons and phonons. We show that thermoelectric figures of merit *ZT* as high as ~2.8 are accessible in these junctions, independent of the flake size and shape, provided the Fermi energy is close to a band edge. We show that Nb dopants as substituents for Mo atoms can be used to tune the Fermi energy, and despite the associated inhomogeneous broadening, room temperature values as high as *ZT* ~ 0.6 are accessible, increasing to 0.8 at 500 K.

Received 26th February 2018,  
Accepted 29th March 2018

DOI: 10.1039/c8nr01635f

[rsc.li/nanoscale](http://rsc.li/nanoscale)

## Introduction

The quest for high-performance thermoelectric devices and materials, which convert waste heat to reusable electrical energy with high efficiency, is a long-sought-after goal of modern materials science.<sup>1–4</sup> A thermoelectric device or material is characterised by its thermoelectric figure of merit  $ZT = S^2GT/\kappa$ , where *S*, *G*, *T*, *κ* represent the Seebeck coefficient (thermopower), electrical conductance, temperature and thermal conductance. The latter is given by  $\kappa = \kappa_e + \kappa_p$ , where  $\kappa_e$  and  $\kappa_p$  are the electron and phonon contributions to thermal conductance respectively.<sup>5–7</sup> Consequently an efficient thermoelectric device requires a large Seebeck coefficient and electrical conductance and simultaneously low thermal conductance. The Mott formula<sup>8</sup>  $S \propto -\frac{\partial \ln T(E)}{\partial E}|_{E=E_F}$  indicates that a large Seebeck coefficient can be obtained if the Fermi energy *E<sub>F</sub>* happens to coincide with a steep slope of electron transmission coefficient *T(E)*, which describes the passage of electrons of energy *E* from the source to the drain of a device. This is a good approximation provided *T(E)* remains linear in the scale of *k<sub>B</sub>T* around Fermi energy where *k<sub>B</sub>* is Boltzmann's constant.<sup>8</sup>

In the past couple of decades, although the thermoelectric performance of the bulk, thin films, and superlattices of Bi, Te, or Sb alloys materials have improved,<sup>9</sup> they are not yet sufficiently efficient for future energy demands and further-

more, some of them are toxic with limited global supply.<sup>10</sup> Recently the potential of the nanoscale devices composed of organic molecules or 2D materials sandwiched between metallic electrodes was recognized. Molecular-scale devices are particularly interesting, because their transport properties may be tailored by chemical modification of the active part of the device,<sup>11–14</sup> varying the contacting configuration,<sup>15–17</sup> device architecture<sup>18–20</sup> and the Fermi level alignment by gating or doping.<sup>21–26</sup> Despite the fact that the nanoelectronic systems are of great interest, it remains a challenge to identify suitable molecules and contacting strategies<sup>27,28</sup> which overcome inhomogeneous broadening and junction variability.

Two dimensional materials provide an alternative approach to thermoelectricity in low-dimensional systems. Graphene is the most widely explored 2D material, but is not suitable since the pristine graphene does not have a bandgap and the material has a high in-plane thermal conductance. As an alternative to graphene, transition metal chalcogenides such as monolayer molybdenum disulfide (MoS<sub>2</sub>) may be attractive alternatives,<sup>29,30</sup> since their band gaps could be used to optimise thermoelectricity. Here we show that this is indeed the case for monolayer MoS<sub>2</sub>, which has a direct bandgap of around 1.6 eV (see Fig. 1 in the ESI†).

Although some contacting strategies have been reported to characterize the electronic and thermoelectric properties of MoS<sub>2</sub>,<sup>20,31</sup> the contacting to metallic electrodes remains a challenge. Recently, it was shown that using a focused electron beam, structures consisting of MoS<sub>2</sub> connected directly to Mo<sub>6</sub>S<sub>6</sub> could be formed.<sup>32</sup> Mo<sub>6</sub>S<sub>6</sub> nanowires (see Fig. S2 in ESI†) can be used as electrodes, because they possess metallic properties and a robust structural conformation.<sup>33,34</sup> Large scale production of MoS<sub>2</sub> nano flakes with self-adaptive Mo<sub>6</sub>S<sub>6</sub>

Quantum Technology Centre, Physics Department, Lancaster University, LA1 4YB Lancaster, UK. E-mail: [h.sadeghi@lancaster.ac.uk](mailto:h.sadeghi@lancaster.ac.uk), [c.lambert@lancaster.ac.uk](mailto:c.lambert@lancaster.ac.uk)

†Electronic supplementary information (ESI) available. See DOI: 10.1039/c8nr01635f



contacts is possible, since their electronic structures are insensitive to the shape of MoS<sub>2</sub> monolayers.<sup>32</sup> Furthermore, it has been reported that parallel-wire bundles of Mo<sub>6</sub>S<sub>6</sub> nanowires are metallic and mechanically stable.<sup>35</sup> Mo<sub>6</sub>S<sub>6</sub> nanowires might therefore be used as a flexible nano wires in electronic devices, because bending does not significantly change their electronic properties.<sup>36</sup>

Although these junctions have been realized, their potential as thermoelectric devices is unexplored. These devices are advantageous, because they provide new direct and covalent contacting possibilities to MoS<sub>2</sub> edges. Furthermore, MoS<sub>2</sub> has shown high thermopower and thermoelectric properties using other contacting strategies.<sup>20,31</sup> In what follows, we show that flexible routes to tailoring their thermoelectric properties such as doping or gating of both MoS<sub>2</sub> and Mo<sub>6</sub>S<sub>6</sub> make such devices very attractive for future thermoelectricity.

## Results and discussion

In this letter, we calculate electronic and vibrational properties of MoS<sub>2</sub> monolayers connected to Mo<sub>6</sub>S<sub>6</sub> nanowire electrodes (Fig. 1) and show that high Seebeck coefficients and low phononic thermal conductances lead to thermoelectric figures of merit as high as  $ZT \sim 2.8$ . We then demonstrate that thermoelectric properties of these junctions could be tuned by introducing Nb dopants into either MoS<sub>2</sub> nano flakes or Mo<sub>6</sub>S<sub>6</sub> electrodes.

Fig. 1 shows a thermoelectric device consisting of a central MoS<sub>2</sub> monolayer connected to Mo<sub>6</sub>S<sub>6</sub> nanowires as current-carrying leads. The device geometry is similar to that obtained experimentally<sup>32</sup> by steering a focused electron beam onto a single MoS<sub>2</sub> monolayer. After making holes in the monolayer, the remaining slab reconstructs to form self-adaptive Mo<sub>6</sub>S<sub>6</sub> leads contacted to MoS<sub>2</sub> nano flakes.<sup>32</sup>

Fig. 2 shows the electron transmission coefficient obtained from the density-functional-theory (DFT) mean field Hamiltonian of the converged ground state geometry of the device (see methods for more details) shown in the Fig. 1b. There exists a transmission gap of approximate 1.65 eV for electron energies  $E$  between  $-0.55$  eV and  $1.1$  eV (relative to

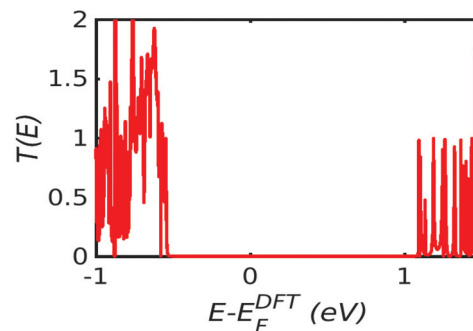


Fig. 2 Electron transmission of the junction in Fig. 1.

the DFT-predicted Fermi energy  $E_F^{\text{DFT}}$ ). This is due to the gap of the pristine two-dimensional MoS<sub>2</sub> (see Fig. S1 in the ESI†) and is in good agreement with previous studies.<sup>30,37</sup> It is known that a step-like transmission coefficient can lead to a high Seebeck coefficient, provided the Fermi energy  $E_F$  lies close to such steps.<sup>8,38</sup> In our case, the step like transmission functions around  $E_F - E_F^{\text{DFT}} = -0.55$  eV and  $1.1$  eV are due to the valance and conduction bands of MoS<sub>2</sub>.

From the transmission function in Fig. 2, the conductance, the Seebeck coefficient, the thermal conductance due to the electrons and electronic thermoelectric figure of merit  $ZT_e = \frac{S^2 G T}{\kappa_e}$  can be obtained (see methods). Fig. 3a shows the electrical conductance for different Fermi energies at room temperature. Due to the thermal averaging of  $T(E)$  around  $-0.55$  eV and  $1.1$  eV, two peaks are obtained around these energies. They also have high slope and therefore, a high Seebeck coefficient is also obtained around these energies

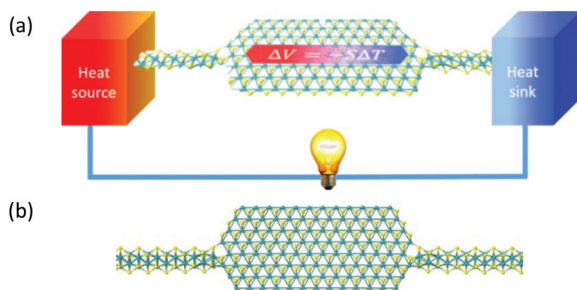


Fig. 1 (a) Schematic of Mo<sub>6</sub>S<sub>6</sub> nanowire–MoS<sub>2</sub>–Mo<sub>6</sub>S<sub>6</sub> nanowire junction. (b) The Mo<sub>6</sub>S<sub>6</sub>–MoS<sub>2</sub>–Mo<sub>6</sub>S<sub>6</sub> junction used in the simulation: yellow atoms represent sulphur and green represents molybdenum.

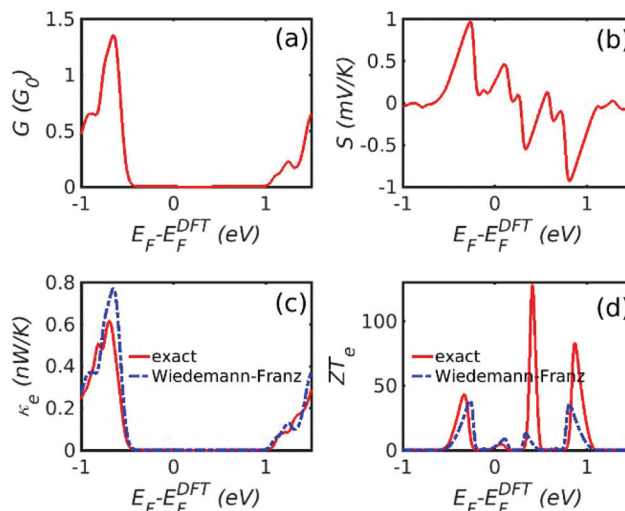


Fig. 3 Electronic contribution to thermoelectric properties as a function of the pristine DFT-predicted Fermi energy at 300 K. (a) Electrical conductance  $G$ , (b) Seebeck coefficient  $S$ , (c) electronic contribution to thermal conductance  $\kappa_e$  and (d) electronic thermoelectric figure of merit  $ZT_e$ .



(Fig. 3b). Fig. 3c and d show the electronic thermal conductance and the electronic contribution to the figure of merit  $ZT_e$ . Fig. 3d depicts the variation of  $ZT_e$  with the Fermi energy. There appear three large resonances up to 50, 100 and 80 around  $-0.5$ ,  $0.5$  and  $1$  eV. In Fig. 3, the red solid curves are exact values obtained using eqn (2)–(5). The blue dashed curves show the prediction of the Wiedemann-Franz law, which is valid provided  $T(E)$  varies approximately linearly with  $E$  near  $E = E_F$  on the scale of  $k_B T$ . Although the two are in close agreement, the Wiedemann-Franz law tends to overestimate the thermal conductance due to electrons and therefore underestimate  $ZT_e$ . However,  $ZT_e$  neglects the contribution from phonons in the denominator and therefore to obtain the full  $ZT$ , we now compute the thermal conductance due to phonons.

The results of our phonon transport calculation are shown in Fig. 4, where Fig. 4a shows the phonon transmission spectrum and Fig. 4b shows the corresponding phonon thermal conductance. Compared to the electronic thermal conductance within the gap, the phonon ( $0.044$  nW K $^{-1}$ ) contribution to thermal conductance is much higher. Consequently, the total  $ZT$  is lower than  $ZT_e$ . As shown in Fig. 5, a value as high as  $ZT \sim 2.8$  at 300 K is obtained around  $E_F = -0.5$  eV, which is higher than unity and higher than currently-reported values for other materials at room temperature.<sup>39–41</sup> Two peaks in the thermoelectric figure of merit curve at  $-0.5$  eV and  $1$  eV are due to two step-like transmission features at  $-0.5$  eV and  $1$  eV.

To demonstrate that the high  $ZT$  at  $E_F - E_F^{\text{DFT}} = -0.5$  eV and  $1$  eV is due to the valence and conduction bands of MoS $_2$  and not an edge effect, we consider other junctions shown in Fig. S3 and S5 in the ESI,<sup>†</sup> obtained by increasing the size of the sample, changing the edge shape and applying periodic boundary conditions in transverse direction. Fig. S3–S5<sup>†</sup> show that their corresponding transmission coefficients possess two main features: step-like transmission features associated with the valence and conduction bands of the MoS $_2$  monolayer around  $E = -0.5$  eV and  $1$  eV and sharp resonances with small width inside the gap. Local density of states calculations around the DFT Fermi energy (Fig. S4 in the ESI<sup>†</sup>) reveal that these resonances are due to localized states at the edges of MoS $_2$  and are sensitive to the shape of the edges and size of the flake. However, the features due to the valence and conduction bands of MoS $_2$  are resilient and are less dependent on the details of the junction. Furthermore, the transmission

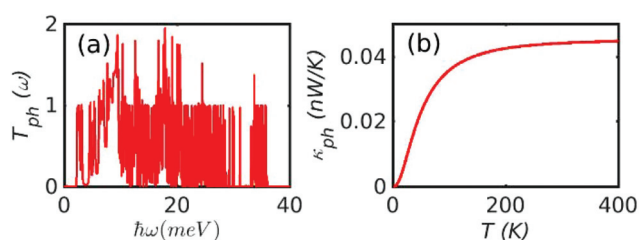


Fig. 4 Phonon transmission spectra (a) and the corresponding thermal conductance (b).

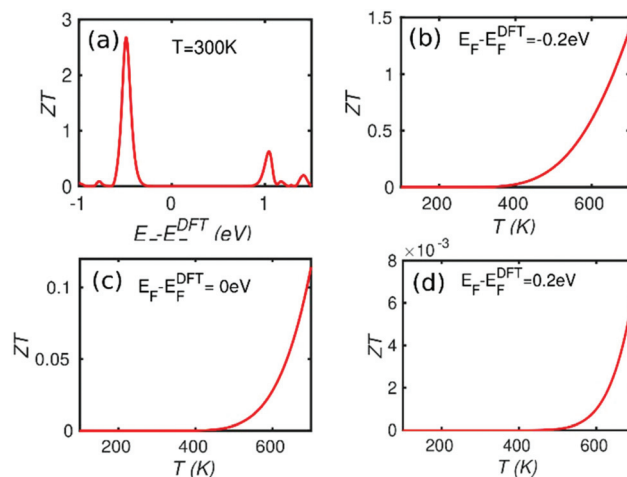
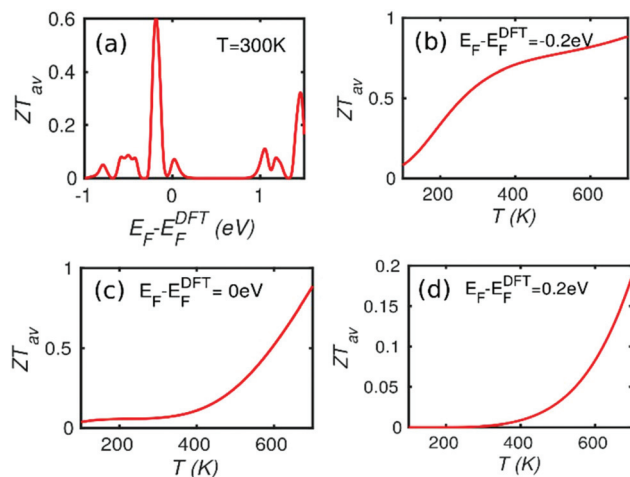


Fig. 5 (a) The thermoelectric figure of merit at 300 K for the structure shown in Fig. 1(b). (b)–(d) The thermoelectric figure of merit as a function of temperature for three particular Fermi energies ( $E_F - E_F^{\text{DFT}} = -0.2$  eV,  $E_F - E_F^{\text{DFT}} = 0$  eV,  $E_F - E_F^{\text{DFT}} = 0.2$  eV) for the structure shown in Fig. 1(b). These calculations are valid provided the forces entering the phonon dynamical matrix are harmonic. Anharmonic effects are negligible, provided the temperature does not approach the melting point of the material. The melting point of MoS $_2$  is greater than 1400 K and therefore we have restricted our analysis to approximately half that value.

amplitude and slope is higher close to the conduction and valence band edges, which is promising for an efficient thermoelectric device. High  $ZT$  is therefore obtained regardless of the shape of the MoS $_2$  flake as shown in Fig. S8 in the ESI<sup>†</sup> at  $E_F - E_F^{\text{DFT}} = -0.5$  eV and  $1$  eV. These results indicate that the higher  $ZT$  peak is mainly due to the valence band edge of the MoS $_2$  monolayer and the edge states do not play a significant role.

In reality, the Fermi energy may be determined by extrinsic factors such as doping. Indeed Nb atoms were used in the past to tune the Fermi energy of MoS $_2$  to form a p-type semiconductor.<sup>21,23</sup> To demonstrate that in the presence of dopants, a high  $ZT$  is accessible in these devices, we now show that in the both Mo $_6$ S $_6$  electrodes and MoS $_2$  flakes, Nb doping can be used to shift the position of the valence band edge towards the Fermi energy. Fig. S9 of the ESI<sup>†</sup> shows results for six different dopant configurations. By replacing Mo atoms by Nb substituents in the monolayer MoS $_2$  flake, the Fermi level shifts towards the valence band and simultaneously new transmission peaks due to defect states near the valence band are formed. The precise value of  $ZT$  is sensitive to the positions and concentration of the dopants, because multiple scattering of electrons between dopants placed in the different locations, causes the transmission coefficient to change (Fig. S9 in the ESI<sup>†</sup>). In a real device consisting of many structures such Fig. 1b and Fig S9,<sup>†</sup> placed in parallel, the ensemble average of transport coefficients are relevant, as defined by the equations in section 10 of the ESI.<sup>†</sup> Fig. 6a shows the resulting ensemble averaged figure of merit  $ZT_{\text{av}}$  versus Fermi energy at one par-





**Fig. 6** (a) The ensemble average of the total thermoelectric figure of merit at 300 K for the configurations with p-type dopants Nb replacing Mo atoms in ESI Fig. S9.† (b)–(d) The ensemble average of the total thermoelectric figure of merit as the function of temperature when  $E_F - E_F^{DFT} = -0.2\text{ eV}$ ,  $E_F - E_F^{DFT} = 0\text{ eV}$ ,  $E_F - E_F^{DFT} = 0.2\text{ eV}$  for the configurations with p-type dopants Nb replacing Mo atoms in ESI Fig. S9.†

ticular temperature and Fig. 6(b–d) show  $ZT_{av}$  as a function of temperature for three different Fermi energies. Fig. 6b shows that even after allowing for inhomogeneous broadening due to random locations of dopants, room temperature values of  $ZT_{av} = 0.6$  are possible, increasing to 0.8 at 500 K.

## Conclusion

In summary, we have studied electron and phonon transport through  $\text{MoS}_2$  nano flakes with self-adaptive  $\text{Mo}_6\text{S}_6$  contacts and demonstrated that large Seebeck coefficients and electrical conductance, combined with low thermal conductance due to electrons and phonons lead to large room temperature thermoelectric figures of merit up to 2.8. Even after allowing for inhomogeneous broadening due to random locations of dopants, room temperature values of  $ZT_{av} = 0.6$  are possible, increasing to 0.8 at 500 K. This demonstrates that  $\text{MoS}_2$  nano flakes are an attractive material for the design of high efficiency, nanoscale, thermoelectric energy harvesters and conversely for nanoscale Peltier cooling.

## Computational methods

The structures of  $\text{Mo}_2\text{S}-\text{Mo}_6\text{S}_6$  junction obtained in the experiments<sup>32</sup> were optimized using the first-principles DFT code SIESTA<sup>42</sup> with a double- $\zeta$  (DZ) basis set. Subsequently, the nanoscale  $\text{Mo}_6\text{S}_6-\text{Mo}_2\text{S}-\text{Mo}_6\text{S}_6$  sandwich device was constructed. From the relaxed xyz coordinates of the system, sets of xyz coordinates were generated by displacing each atom in positive and negative x, y, and z directions by  $\delta q' = 0.01\text{ \AA}$ . The forces in three directions  $q_i = (x_i, y_i, z_i)$  on each atom were then

calculated by DFT without geometry relaxation. These values of forces were combined with the method described in ref. 7 to calculate the dynamical matrix and thermal conductance due to phonons. Furthermore, the mean-field Hamiltonian and overlap matrices were extracted from the first-principles calculation and used to obtain the thermoelectric properties of the devices using Gollum.<sup>43</sup> The electron transmission coefficient  $T(E)$  as a function of energy is calculated through the formula:

$$T(E) = \text{Tr}[\Gamma_R(E)G^R(E)\Gamma_L(E)G^{R\dagger}(E)] \quad (1)$$

where  $\Gamma_{L,R}(E) = i(\Sigma_{L,R}(E) - \Sigma_{L,R}^\dagger(E))/2$  is the anti-Hermitian part of self-energy which describes the broadening of electron transmission resonance;  $\Sigma_{L,R}(E)$  is the self-energy due to the left or right electrodes;  $G^R$ ,  $G^{R\dagger}$  are the retarded and advanced Green's function respectively. The thermoelectric properties can be obtained from:

$$G = G_0 L_0 \quad (2)$$

$$S = -\frac{L_1}{|e|TL_0} \quad (3)$$

$$\kappa_e = -2\frac{L_0L_2 - L_1^2}{hTL_0} \quad (4)$$

$$ZT_e = \frac{L_1^2}{L_0L_2 - L_1^2} \quad (5)$$

$$ZT = \frac{S^2GT}{\kappa_e + \kappa_p} \quad (6)$$

where

$$L_n(T) = \int_{-\infty}^{+\infty} dE (E - E_F)^n T(E) \left( -\frac{\partial f(E)}{\partial E} \right) \quad (7)$$

The electrical conductance  $G$ , Seebeck coefficient  $S$  and thermal conductance due to electrons  $\kappa_e$  can be combined to obtain electronic thermoelectric figure of merit  $ZT_e$ . By including the thermal conductance due to phonons  $\kappa_p$ , the total thermoelectric figure of merit  $ZT$  is calculated.  $T$  is the mean temperature  $(T_1 + T_2)/2$ ,  $G_0 = 2e^2/h$  is the conductance quantum;  $h$  is the Planck's constant;  $e$  is the charge of electron;  $f(E) = (1 + \exp(E - E_F/k_B T))^{-1}$  is the Fermi-Dirac probability distribution function and  $E_F$  is the Fermi energy. The relationship between  $ZT$  and the power efficiency is discussed in section 11 of the ESI.†

## Author information

HS and CJL conceived and conducted the project. Q. W. and H. S. carried out the calculations. All authors took part in the discussions and writing the manuscript.

## Conflicts of interest

There are no conflicts to declare.



## Acknowledgements

H. S. acknowledges the Leverhulme Trust for Leverhulme Early Career Fellowship no. ECF-2017-186. Further support from the UK EPSRC is acknowledged, through grant no. EP/M014452/1, EP/P027156/1, EP/N017188/1 and EP/N03337X/1. Support from the European Commission is provided by the FET Open project 767187 – QuIET.

## Notes and references

- 1 A. L. Moore and L. Shi, Emerging challenges and materials for thermal management of electronics, *Mater. Today*, 2014, **17**, 163–174.
- 2 Y. Kim, W. Jeong, K. Kim, W. Lee and P. Reddy, Electrostatic control of thermoelectricity in molecular junctions, *Nat. Nanotechnol.*, 2014, **9**, 881–885.
- 3 J. R. Widawsky, P. Darancet, J. B. Neaton and L. Venkataraman, Simultaneous determination of conductance and thermopower of single molecule junctions, *Nano Lett.*, 2012, **12**, 354–358.
- 4 H. Sadeghi, S. Sangtarash and C. J. Lambert, Electron and heat transport in porphyrin-based single molecule transistors with electro-burnt graphene electrodes, *Beilstein J. Nanotechnol.*, 2015, **6**, 1413–1420.
- 5 P. Reddy, S. Jang, R. A. Segalman and A. Majumdar, Thermoelectricity in molecular junctions, *Science*, 2007, **315**, 1568–1571.
- 6 C. A. Perroni, D. Ninno and V. Cataudella, Electron-vibration effects on the thermoelectric efficiency of molecular junctions, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 275303.
- 7 H. Sadeghi, S. Sangtarash and C. J. Lambert, Oligoynes Molecular Junctions for Efficient Room Temperature Thermoelectric Power Generation, *Nano Lett.*, 2015, **15**, 7467–7472.
- 8 C. J. Lambert, H. Sadeghi and Q. H. Al-galiby, Quantum-interference-enhanced thermoelectricity in single molecules and molecular films, *C. R. Phys.*, 2016, **17**, 1084–1095.
- 9 X. Zhang and L. Zhao, Thermoelectric materials: Energy conversion between heat and electricity, *J. Mater.*, 2015, **1**, 92–105.
- 10 L. E. Bell, Cooling, heating, generating power, and recovering waste heat with thermoelectric systems, *Science*, 2008, **321**, 1457–1461.
- 11 Q. H. Al-Galiby, H. Sadeghi, L. A. Algharagholi, I. Grace and C. Lambert, Tuning the thermoelectric properties of metallo-porphyrins, *Nanoscale*, 2016, **8**, 2428–2433.
- 12 M. Noori, H. Sadeghi and C. J. Lambert, High-performance thermoelectricity in edge-over-edge zinc-porphyrin molecular wires †, *Nanoscale*, 2017, **9**, 5299–5304.
- 13 S. V. Aradhya and L. Venkataraman, Single-molecule junctions beyond electronic transport, *Nat. Nanotechnol.*, 2013, **8**, 399.
- 14 M. L. Perrin, R. Frisenda, M. Koole, J. S. Seldenthuis, J. A. C. Gil, H. Valkenier, J. C. Hummelen, N. Renaud, F. C. Grozema, J. M. Thijssen, D. Dulić and S. J. H. van der Zant, Large negative differential conductance in single-molecule break junctions, *Nat. Nanotechnol.*, 2014, **9**, 830–834.
- 15 F. Schwarz and E. Lörtscher, Break-junctions for investigating transport at the molecular scale, *J. Phys.: Condens. Matter*, 2014, **26**, 474201.
- 16 P. Moreno-García, M. Gulcur, D. Z. Manrique, T. Pope, W. Hong, V. Kaliginedi, C. Huang, A. S. Batsanov, M. R. Bryce, C. Lambert and T. Wandlowski, Single-molecule conductance of functionalized oligoynes: Length dependence and junction evolution, *J. Am. Chem. Soc.*, 2013, **135**, 12228–12240.
- 17 H. Sadeghi, S. Sangtarash and C. J. Lambert, Robust molecular anchoring to graphene electrodes, *Nano Lett.*, 2017, **17**, 4611–4618.
- 18 M. Noori, H. Sadeghi, Q. Al-Galiby, S. W. D. Bailey and C. J. Lambert, High cross-plane thermoelectric performance of metallo-porphyrin molecular junctions, *Phys. Chem. Chem. Phys.*, 2017, **19**, 17356–17359.
- 19 Q. Wu, H. Sadeghi, V. M. García-Suárez, J. Ferrer and C. J. Lambert, Thermoelectricity in vertical graphene-C60-graphene architectures, *Sci. Rep.*, 2017, **7**, 11680.
- 20 H. Sadeghi, S. Sangtarash and C. J. Lambert, Cross-plane enhanced thermoelectricity and phonon suppression in graphene/MoS<sub>2</sub> van der Waals heterostructures, *2D Mater.*, 2017, **4**, 15012.
- 21 K. Dolui, I. Rungger, C. Das Pemmaraju and S. Sanvito, Possible doping strategies for MoS<sub>2</sub> monolayers: An ab initio study, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **88**, 075420.
- 22 Q. H. Al-Galiby, H. Sadeghi, D. Z. Manrique and C. J. Lambert, Tuning the Seebeck coefficient of naphthalenediimide by electrochemical gating and doping, *Nanoscale*, 2017, **9**, 4819–4825.
- 23 S. Gemming, G. Seifert and I. Vilfan, Li doped MoS<sub>6</sub> nanowires: Elastic and electronic properties, *Phys. Status Solidi B*, 2006, **243**, 3320–3324.
- 24 M. Ghorbani-Asl, A. N. Enyashin, A. Kuc, G. Seifert and T. Heine, Defect-induced conductivity anisotropy in MoS<sub>2</sub> monolayers, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **88**, 245440.
- 25 H. Sadeghi, S. Sangtarash and C. J. Lambert, Enhanced thermoelectric efficiency of porous silicene nanoribbons, *Sci. Rep.*, 2015, **5**, 9514.
- 26 R. Gillen, J. Robertson and J. Maultzsch, Indirect doping effects from impurities in MoS<sub>2</sub>/h-BN heterostructures, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2014, **90**, 75437.
- 27 W. Hong, D. Z. Manrique, P. Moreno-García, M. Gulcur, A. Mishchenko, C. J. Lambert, M. R. Bryce and T. Wandlowski, Single molecular conductance of tolanes: Experimental and theoretical study on the junction evolution dependent on the anchoring group, *J. Am. Chem. Soc.*, 2012, **134**, 2292–2304.



- 28 Q. Wu, X. Zheng, X. Shi, J. Lan, H. Hao and Z. Zeng, Electron transport enhanced by electrode surface reconstruction: a case study of C 60 -based molecular junctions, *RSC Adv.*, 2014, **4**, 44718–44725.
- 29 C. K. Gan, Theoretical study of thermoelectric properties of few-layer MoS2 and WSe2. *Phys. Chem. Chem. Phys.*, 2014, **16**, 10866–10874.
- 30 E. S. Kadantsev and P. Hawrylak, Electronic structure of a single MoS 2 monolayer, *Solid State Commun.*, 2012, **152**, 909–913.
- 31 J. Wu, H. Schmidt, K. K. Amara, X. Xu, G. Eda and B. Özyilmaz, Large thermoelectricity via variable range hopping in chemical vapor deposition grown single-layer MoS2, *Nano Lett.*, 2014, **14**, 2730–2734.
- 32 J. Lin, O. Cretu, W. Zhou, K. Suenaga, D. Prasai, K. I. Bolotin, N. T. Cuong, M. Otani, S. Okada, A. R. Lupini, J.-C. Idrobo, D. Caudel, A. Burger, N. J. Ghimire, J. Yan, D. G. Mandrus, S. J. Pennycook and S. T. Pantelides, Flexible metallic nanowires with self-adaptive contacts to semiconducting transition-metal dichalcogenide monolayers, *Nat. Nanotechnol.*, 2014, **9**, 436–442.
- 33 A. Leen, S. Nano and S. Facilities, Torsional Deformations in Sub-Nanometer MoS Interconnecting Wires, *Nano Lett.*, 2015, **16**, 1210–1217.
- 34 B. Radisavljevic, A. Radenovic, J. Brivio, V. Giacometti and A. Kis, Single-layer MoS2 transistors, *Nat. Nanotechnol.*, 2011, **6**, 147–150.
- 35 J. Kibsgaard, A. Tuxen, M. Levisen, E. Levisen, S. Gemming, G. Seifert, J. V. Lauritsen and F. Besenbacher, Atomic-scale structure of Mo6S6 nanowires, *Nano Lett.*, 2008, **8**, 3928–3931.
- 36 I. Popov, S. Gemming, S. Okano, N. Ranjan and G. Seifert, Electromechanical switch based on Mo6S6 nanowires, *Nano Lett.*, 2008, **8**, 4093–4097.
- 37 D. Costanzo, S. Jo, H. Berger and A. F. Morpurgo, Gate-induced superconductivity in atomically thin MoS2 crystals, *Nat. Nanotechnol.*, 2016, **11**, 339–344.
- 38 H. Sadeghi, S. Sangtarash and C. J. Lambert, Enhancing the thermoelectric figure of merit in engineered graphene nanoribbons, *Beilstein J. Nanotechnol.*, 2015, **6**, 1176–1182.
- 39 H. Liu, X. Shi, F. Xu, L. Zhang, W. Zhang, L. Chen, Q. Li, C. Uher, T. Day and G. J. Snyder, Copper ion liquid-like thermoelectrics, *Nat. Mater.*, 2012, **11**, 422–425.
- 40 L. Han, D. V. Christensen, A. Bhowmik, S. B. Simonsen, L. T. Hung, E. Abdellahi, Y. Z. Chen, N. V. Nong, S. Linderöth and N. Pryds, Scandium-doped zinc cadmium oxide as a new stable n-type oxide thermoelectric material, *J. Mater. Chem. A*, 2016, **4**, 12221–12231.
- 41 R. Chetty, A. Bali and R. C. Mallik, Tetrahedrites as thermoelectric materials: an overview, *J. Mater. Chem. C*, 2015, **3**, 12364–12378.
- 42 J. M. Soler, E. Artacho, J. D. Gale, A. Garcia, J. Junquera, P. Ordejón and D. Sánchez-Portal, SIESTA Method For Ab Initio Order-N Materials Simulation, *J. Phys.: Condens. Matter*, 2002, **14**, 2745–2779.
- 43 J. Ferrer, C. J. Lambert, V. M. García-Suárez, D. Z. Manrique, D. Visontai, L. Oroszlany, R. Rodríguez-Ferradás, I. Grace, S. D. Bailey, K. Gillemot, H. Sadeghi and L. A. Algharagholy, GOLLUM: A next-generation simulation tool for electron, thermal and spin transport, *New J. Phys.*, 2014, **16**, 093029.

