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A high-performance wearable pressure sensor based on an MXene/PVP composite nanofiber membrane for health monitoring†

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Flexible and wearable pressure sensors have attracted extensive attention in domains, such as electronic skin, medical monitoring and human–machine interaction. However, developing a pressure sensor with high sensitivity, mechanical stability and a wide detection range remains a huge challenge. In this work, a flexible capacitive pressure sensor, based on a $\text{Ti}_3\text{C}_2\text{T}_x$ (MXene)/polyvinyl pyrrolidone (PVP) composite nanofiber membrane (CNM), prepared *via* an efficient electrospinning process, is presented. The experimental results show that even a small mass fraction of MXene can effectively decrease the compression modulus of the PVP nanofiber membrane, thus enhancing the sensing performance. Specifically, the sensor based on (0.1 wt% MXene)/PVP CNM has a high sensitivity (0.5 kPa^{-1} at 0–1.5 kPa), a fast response/recovery time (45/45 ms), a wide pressure detection range (0–200 kPa), a low detection limit ($\sim 9 \text{ Pa}$) and an excellent mechanical stability (8000 cycles). Due to its superior performance, the sensor can monitor subtle changes in human physiology and other signals, such as pulse, respiration, human joint motions and airflow. In addition, a 4×4 sensor array is fabricated that can accurately map the shape and position of objects with good resolution. The high-performance flexible pressure sensor, as developed in this work, shows good application prospects in advanced human–computer interface systems.

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1. Introduction

In recent years, flexible pressure sensors exhibit promising prospects in advanced applications, especially in the fields of electronic skin, flexible display screens, medical monitoring, and human–machine interaction.^{1–12} In general, the practical applications of flexible pressure sensors mainly involve the tactile pressure interaction between the users and the devices, ranging from slight stimuli to heavy touches.¹³ Therefore, it is crucial for flexible pressure sensors to be highly sensitive and operate within a wide detection pressure range, thereby exhibiting good sensing capability, similar to the human skin. According to the signal conversion mechanism, flexible pressure sensors can be mainly divided into four types: piezoresistive, piezoelectric, capacitive, and triboelectric pressure sensors.^{14–22} Among them, capacitive

pressure sensors have the advantages of relatively high sensitivity, fast response time and low power consumption, thereby attracting high research interest.

Many research studies reveal that the introduction of microstructures into dielectric layers of capacitive pressure sensors can further improve their sensing performance. For example, Cui *et al.* presented a pressure sensor with a 1D pyramidal micrographic structure on a polydimethylsiloxane (PDMS) substrate and a dielectric layer of polystyrene (PS) microspheres, with a sensitivity of 0.741 kPa^{-1} in the 0–1 kPa pressure range, a pressure range of 0–50 kPa and a response time of 150 ms.²³ Ji *et al.* proposed a novel dielectric layer based on the gradient microsphere structure (GDA) to achieve a high sensitivity (0.065 kPa^{-1}) of capacitive sensors with an ultra-wide linear range (0–1700 kPa) and a fast response time (100 ms).²⁴ Although microstructures can effectively enhance sensitivity, the manufacturing processes involved, such as photolithography, are usually expensive and complicated, failing to meet the requirements of most practical applications. In comparison, the electrospinning process is simple, efficient and inexpensive, while the fabricated nanofiber membranes have high porosity, large specific surface area and a low compression modulus, making them suitable for practical applications on a large scale.^{25,26}

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The selection of the dielectric layer material is also particularly vital. Polymer materials, which are commonly used as dielectric layers, generally have a high compression modulus, inhibiting the optimization of the sensor sensitivity. In order to solve this problem, conductive fillers can be added to polymer materials to form nanocomposites, thus decreasing the compression modulus while maintaining the flexibility. For example, sensors based on carbon nanotubes (CNTs)/polyvinylidene fluoride (PVDF), silver nanowires (AgNWs)/thermoplastic polyurethane (TPU) and zinc oxide nanowires (ZnONWs)/polydimethylsiloxane (PDMS) elastomer composite membranes, have been studied.^{27–29} However, these one-dimensional (1D) materials such as CNTs, AgNWs and ZnONWs, have low surface areas and tend to aggregate into bundles, which will partly affect the sensing performance. In order to solve the above problems, two-dimensional (2D) materials can be used.

MXene is a kind of metal carbide and nitride with a 2D layered structure. The surface functional groups, such as O, F and OH, lead to the formation of semiconductor functionalized MXene ($M_{n+1}X_nT_x$), where M is an early transition metal (Ti, Mo, V, *etc.*), X is carbon or nitrogen, x is the termination number and T is a functional group (O, F or OH).^{30,31} Thanks to its abundant functional groups, MXene can be dispersed in water or other organic materials. Compared to other 2D conductive nanofillers, MXene has the advantages of high electrical conductivity, large specific surface area and high aspect ratio.^{32,33} $Ti_3C_2T_x$ is the most widely studied MXene and is used in many applications, including energy storage, electromagnetic interference shielding, catalysis, optoelectronics and sensors.^{30,34,35} Therefore, it is an ideal additive for the nanocomposite dielectric layer.

In this work, a flexible capacitive pressure sensor, based on an MXene ($Ti_3C_2T_x$)/polyvinyl pyrrolidone (PVP) composite nanofiber membrane (CNM), was fabricated using a simple and efficient electrospinning process. The PVP micro-structure network has excellent physiological inertness and biocompatibility and is widely studied for its antihemorrhagic and antibacterial wound protection effects.^{36,37} Furthermore, a small mass fraction of MXene is added into the PVP nanofiber membrane, effectively reducing its compressive modulus and improving the sensing performance. The results indicate that the produced capacitive pressure sensor, based on $f(\text{MXene})/\text{PVP}$ ($f = 0.1$ wt%) CNM, has a high sensitivity of 0.5 kPa^{-1} at 0–1.5 kPa, a fast response/recovery time of 45/45 ms, and a low detection limit (~ 9 Pa) and it shows an excellent mechanical stability during 8000 operation cycles. In addition, the sensor can be used to detect human health signals, such as pulse, respiration and human joint bending, among others. Finally, a 4×4 sensor microarray is constructed for mapping the pressure distribution of objects placed on a flat surface.

2. Experimental section

2.1 Preparation of the MXene/PVP solution

MXene/PVP composites were chosen as dielectric materials, while *N,N*-dimethylformamide (DMF) (99.5%) was used as the

organic solvent. 5 g 9.48 mg mL^{-1} DMF solution was added in a 20 mL vial, followed by adding 1 g PVP particles ($M_r = 1\,300\,000$) into the DMF solution, at a ratio of 1 : 5. 5 mg mL^{-1} MXene solution was added into the PVP/DMF suspension solution, while the mass fractions of MXene and PVP were set at 0.05 : 99.95, 0.1 : 99.9 and 0.15 : 99.85, respectively. Finally, the solutions were kept under magnetic stirring for 24 hours, until they became homogeneous.

2.2 Preparation of the MXene/PVP CNM

MXene/PVP CNM was prepared using the electrospinning process. First, a high voltage electric field of 15 kV was applied at the end of the syringe needle. The syringe's propulsion speed was 10 mL h^{-1} . The distance between the aluminum foil of the receiving device and the needle was set at 15 cm. The ambient humidity was less than 20%. Afterwards, $f(\text{MXene})/\text{PVP}$ ($f = 0, 0.05, 0.1$ and 0.15 wt%) mixed solution portions were drawn with a 1 mL syringe and used for electrospinning. The prepared films were obtained by drying for 10 min on a heating table, at 150°C . Finally, the MXene/PVP CNMs were prepared by UV irradiation (UV-2550, Shimadzu) for 40 minutes.

2.3 Preparation of the sensor

The prepared MXene/PVP CNM was used as the dielectric layer, while the polyimide (PI) film sputtered with Cu was used as the electrode. The MXene/PVP CNM was sandwiched between two layers of Cu-sputtered PI electrodes, forming a flexible capacitive pressure sensor. The physical appearance of the fabricated device is shown in Fig. S1.† Silver paste was spread on the edge of the electrode, while the copper wire was put through subsequent performance testing. Finally, the device was packaged with transparent tape to isolate noise and prevent signal distortion.

2.4 Microstructure characterization of the sensor

The microstructure and morphology of the PVP, MXene particles and MXene/PVP CNMs were characterized using a field emission scanning electron microscope (FESEM, Hitachi S4800). A transmission electron microscope (TEM, FEI Talos F200X) was employed to further characterize the microstructure of MXene. Elemental dispersion analysis of the MXene/PVP CNM was performed using an Energy Dispersive Spectrometer (EDS).

2.5 Electromechanical characterization of the sensor

Regarding electromechanical characterization, a computer-controlled stepper motor (ZQ-9908) was used to apply static and dynamic loading to the sensor. The capacitive response was measured using an LCR (inductance, capacitance, resistance) meter (CYCLETEST ECA200) at a frequency of 1 MHz and a voltage of 1 V. To measure human physiological signals, the sensors were fixed on the wrist, mask, muscle, finger joint and wrist joint, respectively. The copper wires were used to connect the sensors to the LCR meter. The real-time change of the capacitance value was recorded by a computer. Similarly, to test



the perception of the external physical pressure, the sensors were fixed on a table and beaker, respectively. Pressure was generated by squeezing a rubber balloon or holding a beaker, and the capacitance signal was recorded using an LCR meter and a computer.

2.6 Preparation and measurement of the 4×4 sensor array

The fabrication process of a 4×4 sensor array was as follows. The patterned mask was placed on the surface of a PI film for Cu target sputtering to prepare the upper and lower electrodes. The width and spacing of each electrode were fixed at 0.5 cm. Next, the MXene/PVP CNM, prepared by electrospinning, was cut into 16 equally sized pieces of $0.5 \times 0.5 \text{ cm}^2$, to prevent crosstalk between adjacent capacitors. The middle MXene/PVP CNM dielectric layer was assembled face to face with the upper and lower electrodes, to form a 4×4 sensor array. As shown in Fig. S2,[†] the capacitance of the sensor array was measured orderly using a dual comparator (LM393), and wirelessly transmitted to a computer through the WIFI module (ESP8266). Finally, the capacitance cloud map can be directly displayed on the screen of C++ programmed software.

3. Results and discussion

The fabrication procedure of the proposed flexible capacitive pressure sensor, based on MXene/PVP CNM, is schematically illustrated in Fig. 1. The process starts with thorough mixing of the MXene solution and the PVP particles. Next, the MXene/PVP CNM was prepared by electrospinning, followed by UV treatment.^{38,39} Finally, the MXene/PVP CNM was sandwiched between Cu-sputtered PI electrodes to obtain a complete pressure sensor. The detailed experimental steps can be viewed in the “Experimental section”. Fig. 2(a) and (b) show the FESEM images of the PVP and MXene particles. It can be obtained from the size distribution that the average sizes of the PVP and MXene particles are $108 \mu\text{m}$ and $18 \mu\text{m}$, respectively. Most importantly, the organ-like MXene particles are assembled by

multilayer MXene nanosheets, as shown in Fig. 2(b). As further analysis of the MXene nanosheet, the high-resolution TEM image and selected-area electron diffraction (SAED) pattern (Fig. 2(c)) indicate the hexagonal lattice characteristics of the MXene nanosheet, which is consistent with other reported studies.^{40,41} Fig. 2(d) shows the FESEM image of the $f(\text{MXene})/\text{PVP}$ ($f = 0.1 \text{ wt}\%$) CNM, where the fibrous scaffolds appear intertwined, forming a fiber network. Observing the selected area through an enlarged magnification (Fig. 2(e)), some light-colored nanosheets appear densely attached to each fiber. Further analysis of the selected area using energy dispersive spectroscopy (EDS) indicates that the area contains chemical elements of C, O, N, F and Ti, verifying that the light-colored nanosheets are actually MXene. It can be seen from Fig. 2(f) that the assembled sensor is a typical sandwich structure in a cross-sectional view. The Cu-sputtered PI electrode has a thickness of $50 \mu\text{m}$, while the MXene/PVP CNM dielectric layer has a thickness of $23 \mu\text{m}$.

The corresponding capacitance C of the pressure sensor, based on the MXene/PVP CNM, equal to that of a traditional parallel-plate capacitor, is given by:

$$C = \frac{\epsilon_0 \epsilon_r A}{d} \quad (1)$$

where A is the overlapping area of the two electrodes, d is the distance between the two electrodes, and ϵ_r and ϵ_0 are the dielectric constants of the CNM and free space, respectively.⁴² When a vertical pressure P is applied on the sensor, the ratio of the variation of the capacitance ΔC to the initial capacitance C_0 can be expressed as:

$$\frac{\Delta C}{C_0} = \frac{\frac{\epsilon_0 \epsilon_r A}{d} - \frac{\epsilon_0 \epsilon_{r0} A}{d_0}}{\frac{\epsilon_0 \epsilon_{r0} A}{d_0}} = \left(\frac{\epsilon_r d_0}{d \epsilon_{r0}} \right) - 1 \quad (2)$$

where ϵ_r/ϵ_{r0} represents the dielectric constant of the CNM with/without pressure, while d/d_0 is the distance between the two

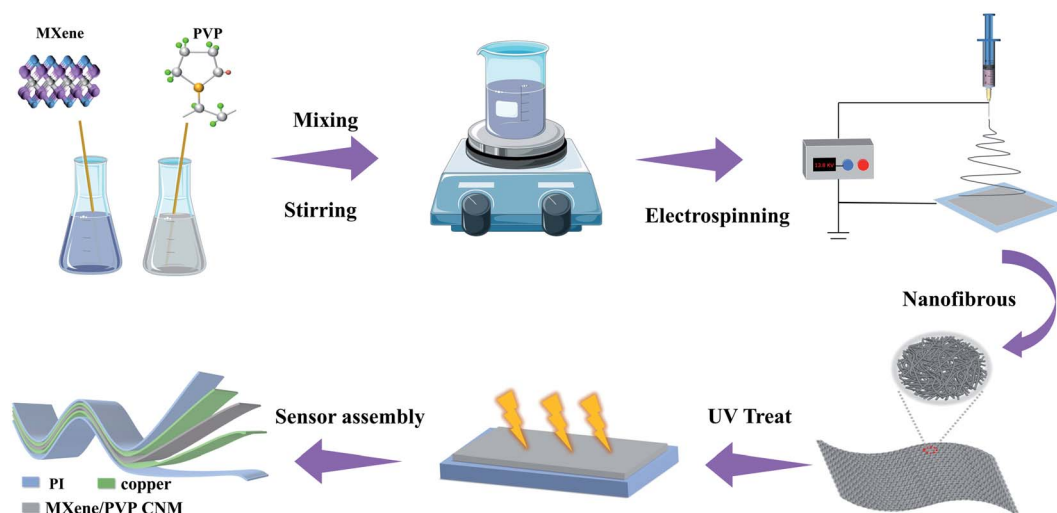


Fig. 1 Schematic diagram of the fabrication process of the MXene/PVP CNM-based pressure sensor.



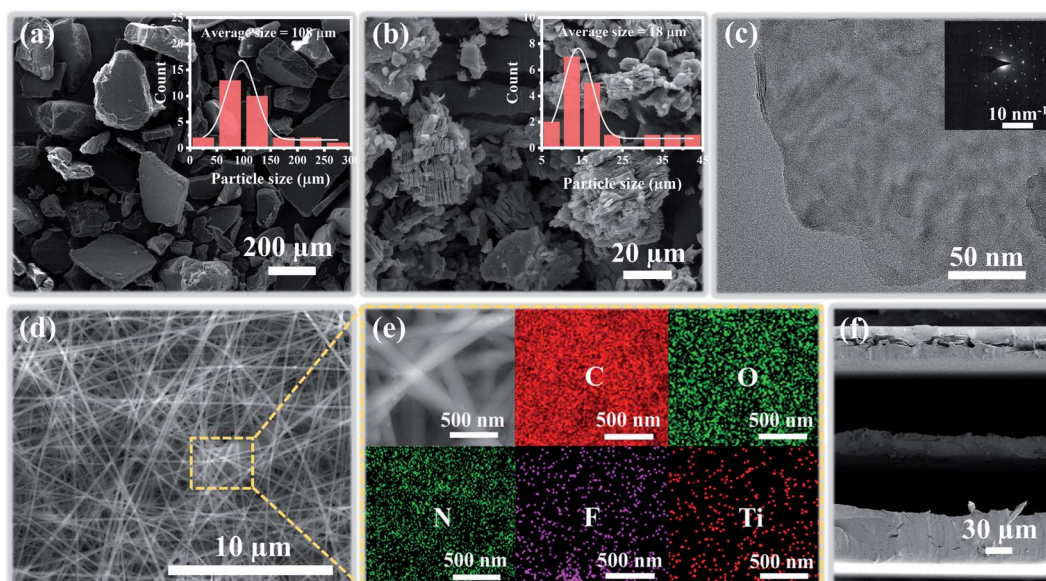


Fig. 2 FESEM images of the (a) PVP and (b) MXene particles. (c) TEM image of the MXene nanosheet and its SAED pattern. (d) FESEM image of the MXene/PVP CNM. (e) Partial enlarged view of the MXene/PVP CNM and corresponding EDS elemental mapping of C, O, N, F and Ti. (f) Cross-sectional SEM image of the assembled sensor.

electrodes with/without pressure. In addition, the sensitivity of the sensor is defined as:

$$S = \delta(\Delta C/C_0)/\delta P \quad (3)$$

where S can be obtained by calculating the slope of the pressure–capacitance curve. On the one hand, the MXene/PVP CNM, prepared by electrospinning, contains a large number of air gaps, which are progressively filled by MXene/PVP fibers as the pressure increases, thus increasing the dielectric constant ϵ_r . On the other hand, the introduction of a small fraction of MXene into PVP can not only decrease its compression modulus, but also have a small impact on its dielectric constant ϵ_{r0} . Based on the synergy of the above two aspects, the sensing performance can be effectively improved.

The dielectric layer plays a great role in improving the overall performance of the capacitive pressure sensor. First, MXene/PVP CNMs electrospun at different times (20, 40 and 60 min) were prepared, and the influence of the dielectric layer thickness (electrospinning time) on the sensing performance is discussed. As shown in Fig. 3(a), at a constant pressure of 200 kPa, the pressure sensor based on the MXene/PVP CNM electrospun at 20 min has a $\Delta C/C_0$ of 1.419, which is larger than that of the sensors electrospun at 40 min (0.977) and 60 min (0.750). Since the thin dielectric layer (electrospun at 20 min) exhibits more pores, ϵ_r and C will increase rapidly under an applied pressure, leading to excellent sensing performance. However, the ultra-thin dielectric layer is easily damaged under high pressure, resulting in a short circuit. Therefore, the optimal electrospinning time for the MXene/PVP CNM dielectric layer is 20 min.

In order to evaluate the effect of the addition of MXene on the sensing performance, S of the sensors, based on $f(\text{MXene})/$

PVP ($f = 0, 0.05, 0.1$ and 0.15 wt%) CNMs, is illustrated in Fig. 3(b). S first increases, with the increase of f ($0 \leq f \leq 0.1$ wt%), followed by a decrease, as f ($0.1 < f \leq 0.15$ wt%) further increases, exhibiting a maximum value at $f = 0.1$ wt%. S of the sensor with $f = 0.1$ wt%, in the low-pressure region (0 – 1.5 kPa), is 0.5 kPa^{-1} , whereas in the high-pressure region (60 – 200 kPa), it is 2.57×10^{-3} kPa^{-1} . This occurrence can be explained by the strain–stress analysis plot in Fig. 3(c) and the force–displacement analysis plot in Fig. S3.† As shown in Fig. 3(c), the compression modulus of the MXene/PVP CNM, which can be obtained by the slope of the curve, decreases from 7.22 to 5.93 , as f increases from 0 to 0.1 wt%. However, as f further increases from 0.1 to 0.15 wt%, the compression modulus rises from 5.93 to 6.72 . A similar phenomenon can also be seen in Fig. S3.† The compressive stiffness, obtained by the slope of the curve, decreases from 86.11 to 61.64 as f increases from 0 to 0.1 wt%, and then rises to 72.70 as f further increases to 0.15 wt%. In other words, the compressive flexibility first decreases ($0 \leq f < 0.1$ wt%), and then increases ($0.1 < f \leq 0.15$ wt%). This can be attributed to the fact that a small fraction of MXene can weaken the fluidity of the PVP chain and soften the PVP nanofiber membrane, resulting in a lower compression modulus/stiffness.⁴³ Nevertheless, an excess of MXene will easily agglomerate inside the PVP nanofiber and tend to create a 3D physical network, thus increasing the compression modulus/stiffness.^{44,45} Consequently, the sensor with $f = 0.1$ wt% possesses the maximum S value.

As shown in Fig. 3(d), the initial capacitance C_0 of the sensors based on $f(\text{MXene})/\text{PVP}$ ($f = 0, 0.05, 0.1$ and 0.15 wt%) CNMs slightly increases along f , as a result of the elevation of the effective dielectric constant caused by the addition of MXene, which is consistent with the principle of formula (1). However, the relative capacitance ΔC first increases and then



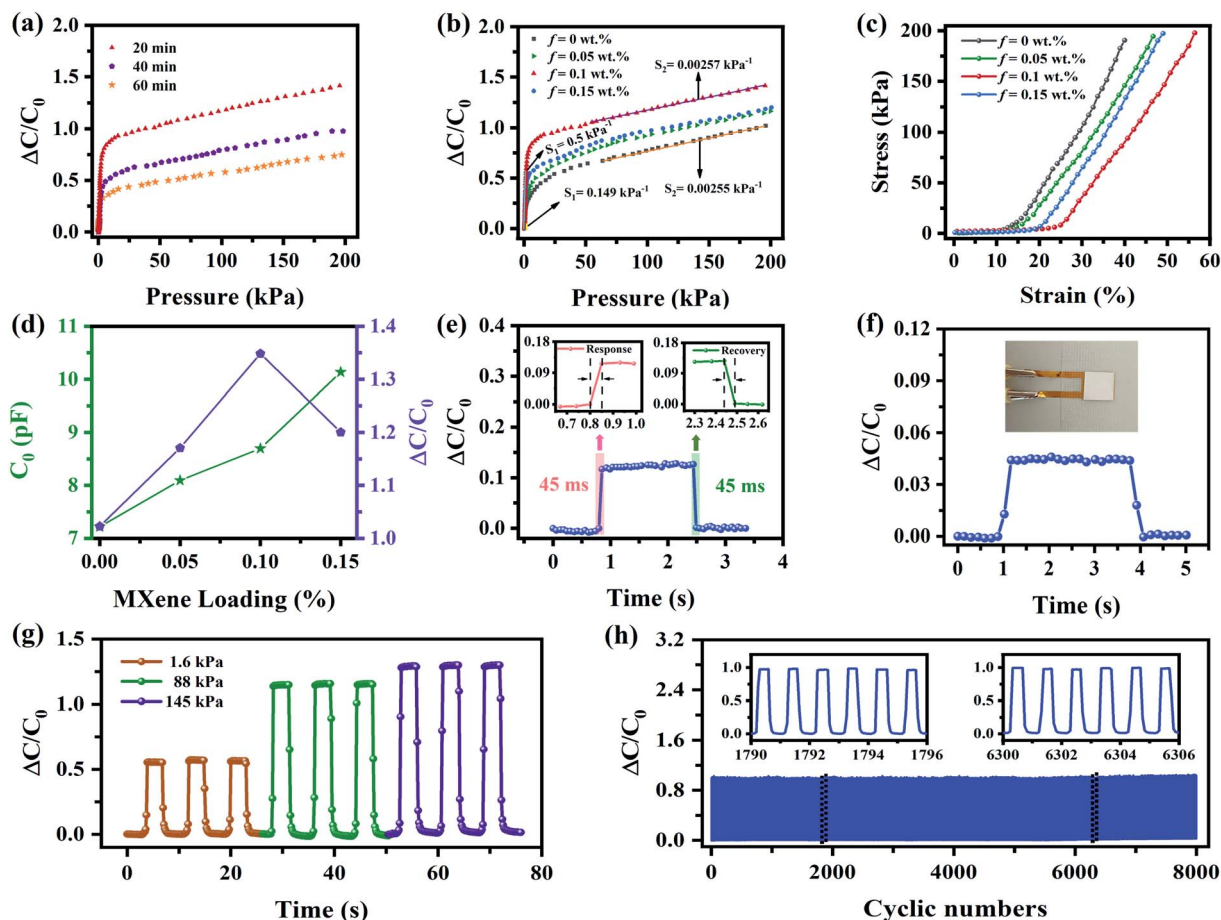


Fig. 3 Experimental sensing performance of the MXene/PVP CNM-based pressure sensor. (a) Relative capacitance change ($\Delta C/C_0$) of the pressure sensors based on the MXene/PVP CNMs electrospun at different times (20, 40 and 60 min). (b) Relative capacitance change ($\Delta C/C_0$) of the pressure sensors based on f (MXene)/PVP ($f = 0, 0.05, 0.1$ and 0.15 wt%) CNM dielectric layers. (c) Compressive stress–strain performance of the sensors. (d) Relation of the initial capacitance C_0 and the relative change in capacitance $\Delta C/C_0$ of the sensors with the MXene fraction amount. (e) Response and recovery time at a pressure of 240 Pa. (f) Demonstration of the lower limit of detection sensitivity, under sequential loading/unloading of a piece of tissue paper (~ 9 Pa). (g) Relative change in capacitance response, under repeatedly applied pressures of 1.6 kPa, 88 kPa, and 145 kPa. (h) Cycling stability test for 8000 loading and unloading cycles, at 34 kPa. The insets present the selected cycles at the beginning and end of the stability test.

decreases with f , matching the variation of the compressive modulus, as illustrated in Fig. 3(c). Similar to most of the dielectric composites, the results shown in Fig. S4† demonstrate that C_0 of the sensors based on f (MXene)/PVP ($f = 0, 0.05, 0.1$ and 0.15 wt%) CNMs decreases as the frequency increases,

which is caused by the Maxwell–Wagner–Sillars interface polarization inside the material.⁴⁶ Furthermore, the conductivity of the sensors, as a function of frequency, is shown in Fig. S5.† The sensors show conductive behavior when the frequency is greater than 10^5 Hz, while the conductivity is proportional to f . This

Table 1 Sensing performance, including the sensitivity, response time, pressure range and stability of the MXene/PVP CNM-based capacitive pressure sensor and other reported state-of-the-art current pressure sensors

Nanofiber material	Sensitivity	Response time	Pressure range	Stability [cycle]	Ref.
ACG-M hydrogel	—	500 ms	—	—	47
RGO coated PU foam	0.67/kPa	10 ms	0–30 kPa	100	48
CNFs/PDMS@Ni foam	0.6/kPa	30 ms	0–20 kPa	10 000	49
PLGA–PCL nanofiber	0.863/kPa	251 ms	0–5 kPa	—	50
Poly (ionic liquid) nanofiber	0.49/kPa	30 ms	0–11 kPa	300	51
PDMS ion gel/PVDF–HFP nanofiber	0.43/kPa	—	0–1.6 kPa	—	52
RCU nanofiber	0.174/kPa	—	0–10 kPa	1000	53
PVP/MXene nanofiber	0.5/kPa	45ms	0–200 kPa	8000	This work



indicates that the MXene can form a stable conductive seepage bridge in the PVP matrix at high frequencies.

Regarding other performance aspects of the sensor with $f = 0.1$ wt%, Fig. 3(e) shows its response and recovery time. When a force of 240 Pa is applied, the relative capacitance $\Delta C/C_0$ increases rapidly and the response time is 45 ms. After unloading, the capacitor returns to its initial level rapidly, and the recovery time is also 45 ms. The device shows not only a fast response/recovery time but also an excellent advantage in sensing tiny objects. In Fig. 3(f), a piece of paper (~ 9 Pa) is placed lightly on the sensor. The relative capacitance increases rapidly from 0 to 0.043, returning to 0 after the paper is picked up. Based on the demonstrated excellent performance, the sensor appears to have a similar perception ability to human skin. The pressure stability under repeated iterations is also studied. Fig. 3(g) displays the stability tests of the sensor at different levels of pressure. Forces of 1.6 kPa, 88 kPa and 145 kPa are repeatedly applied three times, while observations show

that the pressure response is stable, without obvious fluctuations. In addition, the sensor is subjected to fatigue tests up to 8000 times, as shown in Fig. 3(h), where the inset represents an enlarged view of selected cycles. According to the illustrations, the device shows an excellent mechanical stability throughout the test interval, which proves the long endurance of the proposed device. Overall, the performance of the sensor in this work is comparable to those of other reported state-of-the-art pressure sensors, as shown in Table 1.^{47–53}

4. Applications

Wearable miniature electronic devices have gained interest in recent years. Specifically, heart rate detection has become an attractive feature of such devices.^{54,55} Fig. 4(a) shows the ability of the proposed sensor to detect the heart rate. The designed sensor is fixed on the volunteer's wrist, along with medical tape to detect his pulse signal. The real-time pulse of the volunteer is

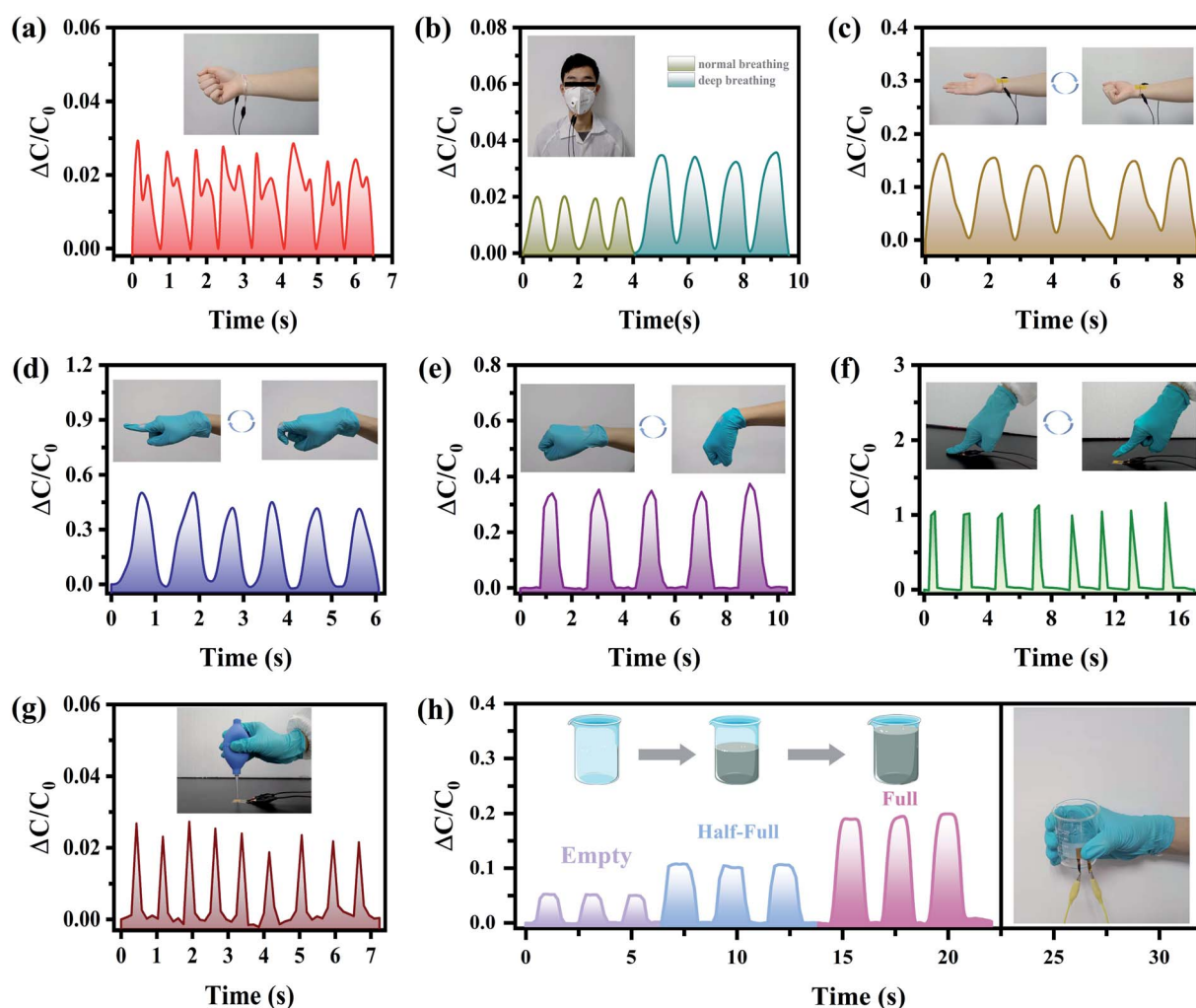


Fig. 4 Applications of the MXene/PVP CNM-based sensor. (a) Real-time monitoring of human pulse wave. (b) Breath detection of normal and deep breathing. (c) The capacitance response to the muscle contraction and relaxation by opening the palm and clenching the fist. (d) The capacitance response to a finger bending at 90 degrees. (e) The capacitance response to the wrist bending at 60 degrees. (f) The capacitance response to the repeated pressing with the finger. (g) The capacitive response to a light airflow. (h) The capacitive response to holding an empty, half-full and full beaker.



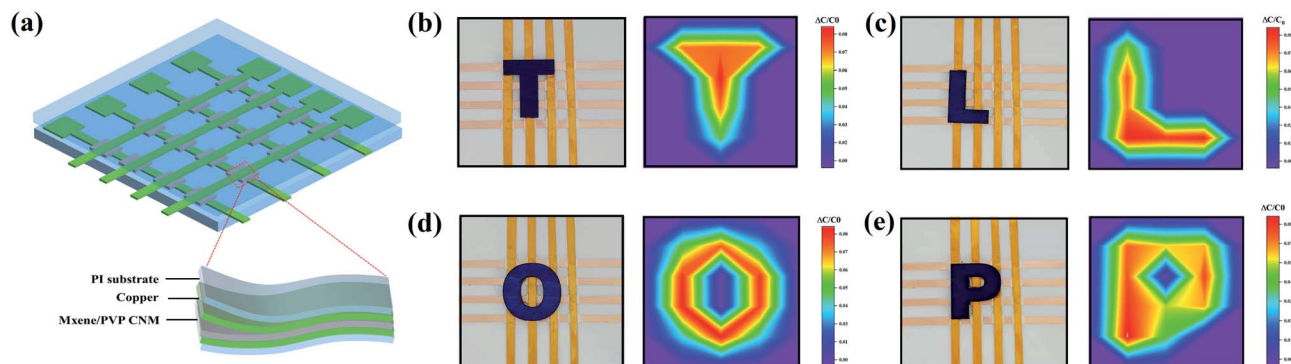


Fig. 5 Spatial pressure distribution based on the sensor array. (a) Schematic diagram of the 4×4 sensor array. (b–e) Placement of 3D letters T, L, O and P on the sensor matrix. On the left side there is the actual shape, while on the right there is the capacitance mapping.

comparable to the pulse rate of 90 beats per min, usually detected in healthy adults. In addition, respiratory monitoring is also used, as an indispensable part of medical diagnosis. As shown in Fig. 4(b), the designed sensor is installed on the mask to monitor the normal and deep breathing of the volunteer, so as to obtain important biological signals regarding human health based on the detected breathing frequency. As the volunteer exhales, the air pressure pushes the sensor to compress, causing the relative capacitance to increase. As the volunteer inhales, the air pressure disappears and the relative capacitance gradually decreases, returning to its original value. During a deep breath, the change of the relative capacitance is larger due to a higher air pressure.

In a further evaluation of the practical and intelligent application of the designed sensor, it is attached to body joints to detect the physiological signals of movements, providing the user with motion related feedback and guidance.^{56,57} Specifically, the contraction and relaxation of the wrist muscles are illustrated in Fig. 4(c). As the palm makes a fist, the muscle protrudes, exerting a pressure on the sensor, thereby increasing the relative capacitance. Monitoring muscle movements and receiving clear feedback enhance the development of healthy exercise habits in a short period of time, while improving efficiency and reducing the risk of injury. Following this, in order to monitor the joint motion of human hands, the sensor is connected to the finger (Fig. 4(d)) and wrist (Fig. 4(e)). The relative capacitance increases, as the finger/wrist bends, while it returns to the original value, when the finger/wrist relaxes. These signals show a fast and steady response to various joint motions, reflecting the sensor's ability to monitor movements in both small joints (such as fingers) and large joints (such as wrists). Next, the sensor's response to other physical stimuli is also tested. As shown in Fig. 4(f), when a finger repeatedly applies force on the sensor, the variation of the relative capacitance can quite accurately reflect the size and duration of that force. Furthermore, the sensor successfully demonstrates its ability to sense tiny airflow, as shown in Fig. 4(g). When a rubber balloon is squeezed, the generated airflow is distributed on the surface of the sensor, resulting in an obvious wave peak, within a rapid response time. This behavior will be of great help to prove its usefulness in the monitoring of

atmospheric airflow and wind direction in the future. Moreover, the sensor can distinguish objects of different weights. During the respective experiment, one holds a beaker in his hand, while the sensor is fixed between the thumb and the beaker. As an empty/half-full/full beaker of water is picked up and put down repeatedly, the capacitance signal increases with the weight of the object, indicating that more force of the hand is required (Fig. 4(h)).

In order to verify the ability of the pressure sensor to obtain load information in space, a 4×4 sensor array is designed, as shown in Fig. 5(a). The upper and lower parts of the sensing matrix are Cu-sputtered PI electrodes, while the middle part is the MXene/PVP CNM dielectric layer. When 3D printed letters ("T", "L", "O", "P") are placed on different pixels of the sensor matrix, the device can identify the overall pressure distribution. The magnitude of the pressure applied by different weights can also be obtained based on the value of the relative capacitance change, as illustrated in Fig. 5(b)–(e). The detailed data can be found in the ESI.† These results indicate the application potential of the sensing matrix in the human–machine interface field.

5. Conclusion

In conclusion, a high-performance flexible capacitive pressure sensor, based on MXene/PVP CNM, was developed. Different mass fractions of MXene ($f = 0, 0.05, 0.1, 0.15$ wt%) were added to the PVP membrane to fabricate composite dielectric layers, *via* electrospinning. Due to the high porosity and low compression modulus, the pressure sensor with $f = 0.1$ wt% demonstrated the best mechanical and electrical properties, including a high sensitivity (0.5 kPa^{-1} at $0\text{--}1.5 \text{ kPa}$), a fast response/recovery time (45/45 ms), a wide operating pressure range ($0\text{--}200 \text{ kPa}$), an ultra-low detection limit ($\sim 9 \text{ Pa}$) and an excellent stability (8000 cycles). In addition, due to its outstanding electromechanical properties, the designed sensor can be used in various pressure monitoring applications, such as pulse, respiration, human joint bending and hand-holding a beaker. Finally, a pressure mapping test of the 4×4 sensing array was performed, showing its great potential in future wearable medical devices and health monitoring. The



developed high-performance sensor has broad application prospects in generic advanced wearable electronic devices.

Conflicts of interest

There are no conflicts to declare.

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References

- Q. Ma, L. Sun and B. Wang, *Matter*, 2022, **5**, 782–784.
- X. Wei, H. Li, W. Yue, S. Gao, Z. Chen, Y. Li and G. Shen, *Matter*, 2022, **5**, 1481–1501.
- H. B. Choi, J. Oh, Y. Kim, M. Pyatykh, J. C. Yang, S. Ryu and S. Park, *ACS Appl. Mater. Interfaces*, 2020, **12**, 16691–16699.
- D. J. Lipomi, M. Vosgueritchian, B. C.-K. Tee, S. L. Hellstrom, J. A. Lee, C. H. Fox and Z. Bao, *Nat. Nanotechnol.*, 2011, **6**, 788–792.
- J. Chen, L. Li, Z. Zhu, Z. Luo, W. Tang, L. Wang and H. Li, *Mater. Today Chem.*, 2022, **23**, 100718.
- Q. Wu, Y. Qiao, R. Guo, S. Naveed, T. Hirtz, X. Li, Y. Fu, Y. Wei, G. Deng, Y. Yang, X. Wu and T.-L. Ren, *ACS Nano*, 2020, **14**, 10104–10114.
- K. Meng, X. Xiao, W. Wei, G. Chen, A. Nashalian, S. Shen, X. Xiao and J. Chen, *Adv. Mater.*, 2022, **34**, 2109357.
- W. Wang, S. Gao, Y. Li, W. Yue, H. Kan, C. Zhang, Z. Lou, L. Wang and G. Shen, *Adv. Funct. Mater.*, 2021, **31**, 2101201.
- S. Sundaram, P. Kellnhofer, Y. Li, J.-Y. Zhu, A. Torralba and W. Matusik, *Nature*, 2019, **569**, 698–702.
- Y. Zheng, K. S. Beloshenko, S. Shulga, R. Wojnarowska and S. Prokhorenko, *Ukr. J. Phys. Opt.*, 2017, **18**, 102–108.
- Y. Zheng, K. S. Beloshenko, M. Makarovskiy, Y. Guliyova and S. Prokhorenko, *Ukr. J. Phys. Opt.*, 2017, **18**, 225–231.
- X. Liao, W. Wang, L. Zhong, X. Lai and Y. Zheng, *Nano Energy*, 2019, **62**, 410–418.
- H. Niu, S. Gao, W. Yue, Y. Li, W. Zhou and H. Liu, *Small*, 2020, **16**, 1904774.
- L. Bi, Z. Yang, L. Chen, Z. Wu and C. Ye, *J. Mater. Chem. A*, 2020, **8**, 20036.
- B. Chen, L. Zhang, H. Li, X. Lai and X. Zeng, *J. Colloid Interface Sci.*, 2022, **617**, 478–488.
- E. S. Hosseini, L. Manjakkal, D. Shakthivel and R. Dahiya, *ACS Appl. Mater. Interfaces*, 2020, **12**, 9008–9016.
- Z. Chen, Z. Wang, X. Li, Y. Lin, N. Luo, M. Long, N. Zhao and J. B. Xu, *ACS Nano*, 2017, **11**, 4507–4513.
- Y. Guo, S. Gao, W. Yue, C. Zhang and Y. Li, *ACS Appl. Mater. Interfaces*, 2019, **11**, 48594–48603.
- M. Yue, Y. Wang, H. Guo, C. Zhang and T. Liu, *Compos. Sci. Technol.*, 2022, **220**, 109263.
- J. Zhang, L. Wan, Y. Gao, X. Fang, T. Lu, L. Pan and F. Xuan, *Adv. Electron. Mater.*, 2019, **5**, 1900285.
- D. Yang, H. Guo, X. Chen, L. Wang, P. Jiang, W. Zhang, L. Zhang and Z. L. Wang, *J. Mater. Chem. A*, 2020, **8**, 23827–23833.
- H. J. Lee, K.-Y. Chun, J. H. Oh and C.-S. Han, *ACS Sens.*, 2021, **6**, 2411–2418.
- J. Cui, B. Zhang, J. Duan, H. Guo and J. Tang, *Materials*, 2017, **10**, 1439.
- B. Ji, Q. Zhou, M. Lei, S. Ding, Q. Song, Y. Gao, S. Li, Y. Xu, Y. Zhou and B. Zhou, *Small*, 2021, **17**, 2103312.
- X. Li, Y. Zhao, Y. Bai, X. Zhao, R. Wang, Y. Huang, Q. Liang and Z. Huang, *Electrochim. Acta*, 2017, **230**, 445–453.
- H. Matsumoto and A. Tanioka, *Membranes*, 2011, **1**, 249–264.
- X. Yang, Y. Wang and X. Qing, *Sens. Actuators, A*, 2019, **299**, 111579.
- S. Zhao, W. Ran, D. Wang, R. Yin, Y. Yan, K. Jiang, Z. Lou and G. Shen, *ACS Appl. Mater. Interfaces*, 2020, **12**, 32023–32030.
- G.-W. Hsieh, L.-C. Shih and P.-Y. Chen, *Nanomaterials*, 2022, **12**, 256.
- L. Li, X. Fu, S. Chen, S. Uzun, A. Levitt, C. E. Shuck, W. Han and Y. Gogotsi, *ACS Appl. Mater. Interfaces*, 2020, **12**, 15362–15369.
- M. Mao, K.-X. Yu, C.-F. Cao, L.-X. Gong, G.-D. Zhang, L. Zhao, P. Song, J.-F. Gao and L.-C. Tang, *Chem. Eng. J.*, 2022, **427**, 131615.
- J. Xu, T. Peng, X. Qin, Q. Zhang, T. Liu, W. Dai, B. Chen, H. Yu and S. Shi, *J. Mater. Chem. A*, 2021, **9**, 14147–14171.
- S. Zhao, W. Ran, D. Wang, R. Yin, Y. Yan, K. Jiang, Z. Lou and G. Shen, *ACS Sens.*, 2021, **6**, 2630–2641.
- S. Sharma, A. Chhetry, M. Sharifuzzaman, H. Yoon and J. Y. Park, *ACS Appl. Mater. Interfaces*, 2020, **12**, 22212–22224.
- C. Jiang, C. Wu, X. Li, Y. Yao, L. Lan, F. Zhao, Z. Ye, Y. Ying and J. Ping, *Nano Energy*, 2019, **59**, 268–276.
- X. Liu, M. Hou, X. Luo, M. Zheng, X. Wang, H. Zhang and J. Guo, *Biomacromolecules*, 2021, **22**, 319–329.
- S. K. Das, T. Parandhaman and M. D. Dey, *Green Chem.*, 2021, **23**, 629.
- F. Hassouna, S. Therias, G. Mailhot and J.-L. Gardette, *Polym. Degrad. Stab.*, 2009, **94**, 2257–2266.
- E. Yilgor, O. Kaymakci, M. Isik, S. Bilgin and I. Yilgor, *Appl. Surf. Sci.*, 2012, **258**, 4246–4253.
- Q. Yu, B. Zhao, X. Ren, C. Zhu, Q. Wang, Y. Lin, W. Zeng, Z. Chen and S. Wang, *Langmuir*, 2022, **38**, 5494–5501.
- L. Ding, Y. Wei, L. Li, T. Zhang, H. Wang, J. Xue, L.-X. Ding, S. Wang, J. Caro and Y. Gogotsi, *Nat. Commun.*, 2018, **9**(1–7), 155.
- X. Zeng, Z. Wang, H. Zhang, W. Yang, L. Xiang, Z. Zhao, L.-M. Peng and Y. Hu, *ACS Appl. Mater. Interfaces*, 2019, **11**, 21218–21226.
- Y. Zhang, Y. Zhao, S. Tan and Z. Zhang, *Polym. Chem.*, 2017, **8**, 1840–1849.
- H. Zhang, Z. Zhang, K. Friedrich and C. Eger, *Acta Mater.*, 2006, **54**, 1833–1842.
- W. Yang, N.-W. Li, S. Zhao, Z. Yuan, J. Wang, X. Du, B. Wang, R. Cao, X. Li, W. Xu, Z. L. Wang and C. Li, *Adv. Mater. Technol.*, 2018, **3**, 1700241.
- Prateek, V. K. Thakur and R. K. Gupta, *Chem. Rev.*, 2016, **116**, 4260–4317.



- 47 S.-N. Li, Z.-R. Yu, B.-F. Guo, K.-Y. Guo, Y. Li, L.-X. Gong, L. Zhao, J. Bae and L.-C. Tang, *Nano Energy*, 2021, **90**, 106502.
- 48 L. Zhao, F. Qiang, S.-W. Dai, S.-C. Shen, Y.-Z. Huang, N.-J. Huang, G.-D. Zhang, L.-Z. Guan, J.-F. Gao, Y.-H. Song and L.-C. Tang, *Nanoscale*, 2019, **11**, 10229.
- 49 S.-W. Dai, Y.-L. Gu, L. Zhao, W. Zhang, C.-H. Gao, Y.-X. Wu, S.-C. Shen, C. Zhang, T.-T. Kong, Y.-T. Li, L.-X. Gong, G.-D. Zhang and L.-C. Tang, *Composites, Part B*, 2021, **225**, 109243.
- 50 H. B. K. Muhammad, M. Ali, A. M. Soomro, S. W. Kima, H. B. Kima, B. G. Lee and K. H. Choi, *Sens. Actuators, A*, 2019, **294**, 140–147.
- 51 Z. H. Wang, Y. Si, C. Y. Zhao, D. Yu, W. Wang and G. Sun, *ACS Appl. Mater. Interfaces*, 2019, **11**, 27200–27209.
- 52 M. F. Lin, J. Q. Xiong, J. X. Wang, K. S. Parida and P. S. Lee, *Nano Energy*, 2018, **44**, 248–255.
- 53 Y. B. Shin, C. J. Han, Y. M. Kim and J. W. Kim, *Adv. Eng. Mater.*, 2021, **24**, 2100903.
- 54 X. Qian, M. Su, F. Li and Y. Song, *Acta Chim. Sin.*, 2016, **74**, 565–575.
- 55 Y. Khan, A. E. Ostfeld, C. M. Lochner, A. Pierre and A. C. Arias, *Adv. Mater.*, 2016, **28**, 4373–4395.
- 56 S. Park, J. Y. Cho, B. J. Lee, J.-M. Hwang, M. Lee, S. Y. Hwang, K. M. Kim, K. H. Lee and D. Park, *Sci. Rep.*, 2020, **10**, 11772.
- 57 M. Yang, Y. Cheng, Y. Yue, Y. Chen, H. Gao, L. Li, B. Cai, W. Liu, Z. Wang, H. Guo, N. Liu and Y. Gao, *Adv. Sci.*, 2022, **9**, 2200507.

