

## REVIEW

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## ZnO nanostructured materials and their potential applications: progress, challenges and perspectives

Sauvik Raha and Md. Ahmaruzzaman \*

Extensive research in nanotechnology has been conducted to investigate new behaviours and properties of materials with nanoscale dimensions. ZnO NPs owing to their distinct physical and chemical properties have gained considerable importance and are hence investigated to a detailed degree for exploitation of these properties. This communication, at the outset, elaborates the various chemical methods of preparation of ZnO NPs, viz., the mechanochemical process, controlled precipitation, sol–gel method, vapour transport method, solvothermal and hydrothermal methods, and methods using emulsion and micro-emulsion environments. The paper further describes the green methods employing the use of plant extracts, in particular, for the synthesis of ZnO NPs. The modifications of ZnO with organic (carboxylic acid, silanes) and inorganic (metal oxides) compounds and polymer matrices have then been described. The multitudinous applications of ZnO NPs across a variety of fields such as the rubber industry, pharmaceutical industry, cosmetics, textile industry, opto-electronics and agriculture have been presented. Elaborative narratives on the photocatalytic and a variety of biomedical applications of ZnO have also been included. The ecotoxic impacts of ZnO NPs have additionally been briefly highlighted. Finally, efforts have been made to examine the current challenges and future scope of the synthetic modes and applications of ZnO NPs.

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## 1. Introduction

Found in the Earth's crust as the mineral zincite, ZnO appears as a white powder and is nearly insoluble in water. Zinc oxide is the second most abundant metal oxide after iron. It is inexpensive and safe, and can be prepared easily. The ZnO powder has been widely used as an additive in a plethora of materials and products including ceramics, glass, cement, rubber, lubricants, paints, ointments, adhesives, plastics, sealants, pigments, foods, batteries, ferrites and fire retardants. ZnO can assume three possible polymorphs, rock salt, wurtzite and zinc blende. The rock salt polymorph exists under conditions of high pressure and is therefore very rare. At ambient pressure and temperature, ZnO has a hexagonal wurtzite crystal structure with two lattice parameters, *a* and *c*, with values of 0.3296 nm and 0.52065 nm, respectively. In hexagonal wurtzite, O<sup>2−</sup> ions occupy the hexagonal close packing (hcp) array of lattice sites while Zn<sup>2+</sup> ions place themselves in alternate tetrahedral holes in contrast with the zinc blende structure wherein O<sup>2−</sup> ions form a cubic close packing (ccp) array with Zn<sup>2+</sup> ions occupying tetrahedral voids. ZnO has a wide band gap of ~3.37 eV and a large exciton binding energy of 60 meV. The physical and chemical behaviors of zinc oxide nanoparticles are simply tuned

by changing the morphology using various synthesis routes and different materials.

ZnO has a slew of unique chemical and physical properties, viz., high chemical stability, high electrochemical coupling coefficient, broad range of radiation absorption and high photostability, which make it among all metal oxides a key technological material and confer upon it its wide applications in varied fields. ZnO is categorized as a group II–VI semiconductor in materials science because zinc belongs to the 2<sup>nd</sup> group while oxygen belongs to the 6<sup>th</sup> group of the periodic table. Its covalence is on the borderline demarcating ionic and covalent semiconductors. Besides, it has good transparency, high electron mobility, an outsized exciton binding energy (60 meV), wide band gap (3.37 eV),<sup>1</sup> strong room temperature luminescence, high thermal and mechanical stability at room temperature, broad range of radiation absorption and high photostability that make ZnO the most favorite multitasking material.<sup>2,3,5,6</sup> As a result of its distinctive optical and electrical properties<sup>4</sup> it is considered to be a possible material in electronic applications, optoelectronic applications and laser technology. ZnO among nano-sized metal oxides has also been further extensively exploited to derive possible benefits from its antimicrobial and antitumor activities.<sup>7</sup> Because of its blocking and absorbing capabilities ZnO finds inclusion in some cosmetic lotions.<sup>8</sup> ZnO can also be used in human medicine as an astringent (for wound healing), and to treat hemorrhoids, eczema and excoriation.<sup>9</sup> ZnO nanoparticles have recently

Department of Chemistry, National Institute of Technology Silchar, 788010, Assam, India. E-mail: mda2002@gmail.com



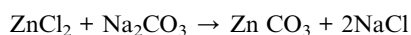
attracted attention owing to their unique features. There are numerous promising applications of ZnO nanoparticles in veterinary science due to their wound healing, antibacterial, antineoplastic and antigenic properties. Recently, many research studies and experimental analyses have improved the efficiency of zinc oxide (ZnO) materials by producing nanostructures where each nano-dimension is reduced to generate nanowires, thin films and other structures for plenty of applications including defense against intracellular pathogens and brain tumors.<sup>10</sup> One-dimensional structures include nanorods,<sup>11–13</sup> nanoneedles,<sup>14</sup> nanohelices, nanosprings, nanorings,<sup>1</sup> nanoribbons,<sup>15</sup> nanotubes,<sup>16–18</sup> nanobelts,<sup>19</sup> nanowires<sup>20–22</sup> and nanocombs.<sup>23</sup> Nanoplates/nanosheets and nanopellets<sup>24,25</sup> are their two-dimensional forms while flowers, dandelions, snowflakes, coniferous urchin-like structures, *etc.*<sup>26–29</sup> count as the three-dimensional morphologies of ZnO nanoparticles. Nevertheless, the challenges in terms of the potential toxic effects of ZnO nanoparticles do require special attention.

## 2. Chemical methods for synthesis of zinc oxide nanoparticles

A range of precursors and a variety of conditions such as temperature, time, concentration of reactants and so forth can be employed for the chemical synthesis of ZnO nanoparticles. These chemical techniques by virtue of adjustment of the various aforesaid parameters allow better control of morphology in terms of size and geometry. ZnO NPs can be chemically synthesized using the following techniques: the mechanochemical process, controlled precipitation, sol-gel method, vapour transport method, solvothermal and hydrothermal methods, and methods using emulsion and microemulsion environments (Fig. 1). The details giving the gist of the various aforementioned strategies are presented in Table 1. The detailed synthesis conditions, experimental variables, main mechanisms, properties and applications are highlighted (Table 1). The different chemical methods used for the synthesis of ZnO NPs are enumerated below.

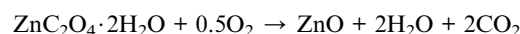
### 2.1 Mechanochemical process

In the mechanochemical process the starting materials are subjected to a high-energy dry milling which sets off a reaction by way of ball-powder impacts in a ball mill, at a low temperature. A diluent, often the by-product of the reaction (salt *viz.* NaCl), accompanies the starting materials during the procedure.<sup>30,31</sup> The diluent functions as a reaction medium and effects separation of the nanoparticles thereby restricting their subsequent growth. This in turn prevents serious agglomeration. The starting materials conventionally used in this method are mainly anhydrous ZnCl<sub>2</sub>, Na<sub>2</sub>CO<sub>3</sub> and NaCl. The following reactions take place during the procedure:



Ao *et al.*<sup>32</sup> carried out a mechanochemical process of synthesizing ZnO NPs by exploiting the reaction between ZnCl<sub>2</sub> and Na<sub>2</sub>CO<sub>3</sub> and using NaCl as a diluent.<sup>32</sup> The pure nanocrystalline ZnO was obtained by removing the by-product NaCl and finally drying in a vacuum. TEM images showed moderately aggregated ZnO nanoparticles of size less than 100 nm which were prepared by a 6 h milling followed by a thermal treatment at 600 °C for 2 h. The effect of milling time and annealing was carefully investigated in the study. A decrease in nanocrystallite size from 25 nm to 21.5 nm was observed as the milling time increased from 2 to 6 h after which it attained steadiness. This phenomenon was chalked up to a critical effect prevailing in the course of milling. The crystal size, however, was found to increase with temperature with the rise being steep after 600 °C. The activation energies for nanocrystallite growth in different temperature ranges were calculated using the Scott equation. The activation energy was found to be 3.99 for growth in between 400 and 600 °C while it reached 20.75 kJ mol<sup>−1</sup> beyond 600 °C. The higher growth rate at higher temperatures was thus attributed to extensive interfacial reactions driven by greater activation energy.

A reduction in particle size on prolonging the milling time was reported by Stanković *et al.*<sup>33</sup> in their work of preparing ZnO NPs using ZnCl<sub>2</sub> and oxalic acid as starting materials. Their scheme involved milling the powders of starting materials in a Retsch PM4 planetary ball mill using a ball-to-powder weight ratio of 10 : 1. Different intervals of time were chosen for milling to investigate the influence of milling time on particle size. Pastes of zinc oxalate thus obtained were then air dried at 100 °C for 1 h and then subjected to thermal decomposition at 450 °C to obtain ZnO NPs.



While the XRD analysis substantiated a perfect long-range order and a pure wurtzite structure of the synthesized ZnO powders regardless of the milling time, Raman spectroscopy revealed that lattice defects and impurities were introduced into ZnO powders at the middle-range scale depending on milling duration. Extended milling was found to reduce crystal defects but introduce impurities. The SEM images suggested that the milling duration of the reactant mixture positively regulated the morphology of the particles irrespective of the additional thermal treatment.

ZnO NPs were also prepared through a mechanochemical method by using ZnCl<sub>2</sub>, NaCl and Na<sub>2</sub>CO<sub>3</sub> as starting materials.<sup>34</sup> A solid phase reaction triggered by milling the starting powders led to the isolation of ZnCO<sub>3</sub> in the NaCl matrix. The ZnCO<sub>3</sub> was finally subjected to a thermal treatment at 400 °C which induced its decomposition to ZnO. The anatomization of TEM results indicated a mean particle size of 26.2 nm. The mean nanocrystallite size evaluated from the XRD peak width at 2θ = 36° using the Scherrer equation was found to be 28.7 nm. Meanwhile, the surface area of the ZnO nanopowder evaluated





Fig. 1 Various strategies for the fabrication of ZnO NPs.

from BET analysis was  $47.3 \text{ m}^2 \text{ g}^{-1}$  corresponding to a spherical particle size of 27 nm.

Another study on the optical properties of ZnO NPs synthesized through mechanochemical means and using  $\text{ZnCl}_2$ , NaCl and  $\text{Na}_2\text{CO}_3$  as raw materials was conducted by Moballegh *et al.*<sup>35</sup> The XRD and TEM results revealed that particle size increased with calcination temperature. The work proposed improved optical properties as a result of the decrease in particle size owing to the enhanced ratio of surface to volume in ZnO NPs. In another study<sup>36</sup> a mixture of starting powders (anhydrous  $\text{ZnCl}_2$ ,  $\text{Na}_2\text{CO}_3$  and NaCl) was milled at 250 rpm and then calcined at  $450^\circ\text{C}$  for 0.5 h to yield ZnO NPs with a crystallite size of 28.5 nm as estimated from subsequent XRD analysis. The particle size that emerged from TEM and SEM analysis ranged in between 20 and 30 nm. The incongruent particle size estimated from BET analysis was ascribed to an agglomeration of nanoparticles in the course of drying.

The foremost shortcoming of the procedure exists in its fundamental difficulty encountered in the homogeneous grinding of the powder and controlled minimization of the particles to the required size. Note that the particle size reduces with increasing time and intensity of milling. However, if the powder is subjected to milling for longer periods of time, the chances of contamination increase. A highly shrunk size of nanoparticles is the prime advantage that can be extracted from the method apart from the benefit of a significantly low cost of

generation coupled with diminished agglomeration of particles and pronouncedly homogeneous crystallite morphology and architecture. The mechanochemical process is particularly desirable for large-scale production of ZnO NPs.

## 2.2 Controlled precipitation

Controlled precipitation is often resorted to when the idea is to set off a large-scale generation of nanoparticles in the powder form with reproducible properties for their use as industrial products. The method involves two stages: (a) precipitation of a precursor of ZnO from the solution, and (b) thermal treatment of this precursor with subsequent milling to induce physical disintegration of the calcined agglomerates and to get rid of impurities. Various methods may be employed for precipitation of ZnO by exercising control over a range of limiting factors that include reaction temperature, pH conditions, duration of precipitation and the medium used for washing. Using  $\text{ZnSO}_4$  and  $\text{NH}_4\text{OH}$ , Sadraei<sup>37</sup> prepared nano-ZnO flakes. Nano-sized ZnO flakes as confirmed by SEM analysis were eventually achieved. XRD analysis of the flakes revealed the hexagonal structure of ZnO flakes with high crystallinity and a diameter of 30 nm.

Kumar *et al.*<sup>38</sup> used zinc acetate ( $\text{Zn}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$ ) and NaOH as reagents, and the settled white powder was separated followed by washing with deionized water thrice and dried overnight under dust-free conditions at room temperature. XRD



Table 1 Chemical methods of synthesis of ZnO

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications	Advantages	Ref.
Mechanochemical process	ZnCl <sub>2</sub> , Na <sub>2</sub> CO <sub>3</sub> and NaCl	Calcination, 2 h, 600 °C, milling for 2–6 h	Milling time and heat-treatment temperature on ZnO nanocrystallite sizes	ZnO nanocrystallite growth is homogeneous, crystal nuclei were formed with decomposition of ZnCO <sub>3</sub> and grew by emergence of the secondary formed ZnO. The driving force of the interfacial reaction came from the activation energy. Higher activation energy above 600 °C leads to a higher growth rate for the ZnO nanocrystallite	Hexagonal structure; particle diameter: 21–25 nm	Simplicity, relatively low-cost equipment, large-scale production, and applicability for a variety of materials. Operates at room temperature, which increases safety and reduces energy utilization. Induces not only morphological and structural changes of the particles but also modifies their optical and electrical properties and prevents the agglomeration of the synthesized particles	32
Precipitation process	ZnCl <sub>2</sub> and oxalic acid	Calcination, 1 h, 400 °C, milling for 0.5–4 h	Oxalic acid and wet-milling conditions on the ZnO average particle size and morphology	—	Hexagonal structure; particle diameter: 1 µm to 50–90 nm	Hexagonal structure; flower production typifies the shape (length of each petal did not exceed 800 nm); application: antimicrobial activity with high production yield	33
	ZnCl <sub>2</sub> , NaCl and Na <sub>2</sub> CO <sub>3</sub>	Calcination, 0.5 h, 300–450 °C, milling for 9 h	Calcination temperature on particle size and structural properties of ZnO nanoparticles	—	Hexagonal structure; particle diameter: 27.7–56.3 nm		35
Precipitation process	ZnSO <sub>4</sub> and NH <sub>4</sub> OH	Reaction: 50–60 °C; drying: 60 °C, 8 h	—	—	Hexagonal structure; flakes, particle diameter: 30 nm	Hexagonal structure; flower production typifies the shape (length of each petal did not exceed 800 nm); application: antimicrobial activity with high production yield	37
	Zn(OAc) <sub>2</sub> ·2H <sub>2</sub> O and NaOH	Reaction: 30 min, 75 °C; drying: Room temperature, overnight	—	On heating the solution of zincate ions, the molecules start to rearrange into hexagonal ZnO nanorods after growing along the (0001) direction. When the molecules got saturated, the ZnO nuclei grew to give rod shaped ZnO. Over time, these freshly formed nanorods deposited on the surface of formerly formed crystalline nanorods resulting in a leaf-like structure first	Hexagonal structure; flakes, particle diameter: 30 nm		38





Table 1 (Contd.)

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications	Advantages	Ref.
				and a number of such leaves came together in an ordered array which appeared as flower shaped ZnNSs	Hexagonal structure; ZnO (A): particle diameter is 40 nm. ZnO (B): particle diameter is 40 nm. Photocatalytic degradation of methyl orange dye		39
	Zn(OAc) <sub>2</sub> ·2H <sub>2</sub> O, (NH <sub>4</sub> ) <sub>2</sub> CO <sub>3</sub> , and polyethylene glycol	Drying: 100 °C, 12 h. ZnO (A): the dried precipitate was ball-milled for 1 h followed by calcination at 450 °C for 3 h to produce ZnO powder which was further ball-milled for 3 h. ZnO (B): the precipitate was ball-milled for an hour and then a 1 : 1.0 powder and 1-butanol mixture was prepared which was dried by heterogeneous azeotropic distillation; calcination at 450 °C for 3 h. Aging: 96 h, ambient temperature, calcination: 2 h, 500 °C	Reaction temperature and time, concentration of oleic acid				
	ZnCl <sub>2</sub> , NH <sub>4</sub> OH, and CTAB			Particle formation is a very complex process and involves nucleation, growth, coagulation and flocculation. Addition of surfactant CTAB affects the nucleation during the crystallization process. After nucleation, the surfactant can influence particle growth, coagulation and flocculation	Zincite structure; particle diameter: 54–60 nm, BET = ~17 m <sup>2</sup> g <sup>-1</sup>		41
	Zn(NO <sub>3</sub> ) <sub>2</sub> , NaOH, SDS, and TEA (triethanolamine)	Precipitation: 50–55 min, 101 °C	Addition of sodium dodecyl sulfate (SDS) and triethanolamine (TEA)	Dissolution–reprecipitation mechanism	Wurtzite structure, rod-like shape (L: 3.6 µm, D: 400–500 nm), nut-like and rice-like shapes, size: 1.2–1.5 µm		42



Table 1 (Contd.)

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications	Advantages	Ref.
Sol-gel	Zn(OAc) <sub>2</sub> ·2H <sub>2</sub> O, polyvinyl pyrrolidone (PVP) and NaOH	Reaction: 60 °C; vigorous stirring for 1 h. Calcination: 600 °C, 1 h			Wurtzite structure; platelet-like ZnO with a grain size of 150 nm transformed into rod-shaped ZnO with a diameter of 100 nm at 3 × 10 <sup>-4</sup> M PVP	Sol-gel shows many advantages over other techniques such as its simplicity and low equipment cost	43
	Zn(OAc) <sub>2</sub> ·2H <sub>2</sub> O and oxalic acid	Reaction: refluxed at 50 °C, 1 h; drying at 80 °C for 20 h; calcination: 650 °C, 4 h			Wurtzite structure; uniform, spherically shaped ZnO nanoparticles with a crystallite size of 20 nm; BET surface area of 10 m <sup>2</sup> g <sup>-1</sup> ; 69.75% degradation of phenol and 67.98% degradation of benzoic acid in 120 min under UV light		44
	Zinc 2-ethylhexanoate, 2-propanol, and tetramethylammonium (TMAH)	Reaction: room temperature; aging: 30 min; drying: 60 °C	Weight ratio of 2-propanol and tetramethylammonium (TMAH)		Cylinder-shaped crystallites, diameter: 25–30 nm; height: 35–45 nm		46
Vapour transport method	Zn and water vapour or oxygen	Heating: 1 h, 800 °C, pressure: 0.03–0.05 MPa, cooling rate: 7 °C per minute	Influence of the atmosphere		With H <sub>2</sub> O: nanoflowers constructed by tens of ZnO nanosheets with random orientations. With O <sub>2</sub> : hexagonal nanorod arrays, non-uniform sized nanorods	The vapour transport method has been emphasised because of the easy control of thicknesses, morphologies and crystal structures of ZnO films and nanostructures by varying the precursor gas, substrate temperature and substrate materials	48
	ZnO powder	Heating in a horizontal tube furnace: 1350 °C, 30 min; deposition: 400–500 °C under an Ar pressure of 250 Torr		For crystal growth, after initial nucleation, the subsequent growth stage strongly governs the final morphology of the crystal. In O <sub>2</sub> gas, the growth of ZnO is simply along the (001) direction due to the fastest growth kinetics in this direction and absence of side or reverse reactions. Due to the small thickness of the nanobelts, spontaneous polarization normal to the nanobelt leads to the growth of helical nanostructures. The mechanism for the helical growth is attributed to the consequence of minimizing the total	Wurtzite; nanobelts with widths of 10–60 nm, thickness of 5–20 nm and lengths up to several hundreds of micrometers		49





Table 1 (Contd.)

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications	Advantages	Ref.
	Zn powder and O <sub>2</sub>	Heating in a furnace at 450 °C, 550 °C, and 650 °C at a rate of 10 °C min <sup>-1</sup> , feeding O <sub>2</sub> into the reaction zone at a rate of 5 mL min <sup>-1</sup> for 30, 45, and 60 min after reaching a furnace temperature of 450 °C.	Growth temperature and growth time	energy contributed by spontaneous polarization and elasticity The growth mechanism of 1D ZnO nanostructures can be divided into three stages, as follows: first, the Zn vapor and catalytic Cu form liquid alloy droplets during the heating process at a certain temperature, representing the initial stage of the nucleation process. Second, crystal nucleation occurs upon gaseous species adsorption until supersaturation is reached, and the formed sites serve as nucleation sites on the substrate. Finally, the axial growth of the nanorods begins from these sites	450 °C: ZnO nanorods with a diameter and length of 19–27 nm and 2.8 µm, respectively. 550 °C: ZnO nanorods with a diameter and length of 85 nm and 3.8 µm, respectively. 650 °C: ZnO nanorods with a diameter and length of 190–350 nm and 3.9 µm, respectively, covered with short nanorods with a diameter of 95 nm and length of 900 nm at the tips. 30 min growth time: ZnO nanorods with a diameter of 19–27 nm and a length of 2.8 µm. 45 min growth time: ZnO nanorods with a diameter of 65–190 nm and a length of 3.2 µm. 60 min growth time: ZnO nanorods with a diameter of 80–250 nm and a length of 3.8 µm		50
	Zn and O <sub>2</sub>	Heating in a furnace at 750 °C, feeding O <sub>2</sub> into the reaction zone at a rate of 50 mL min <sup>-1</sup> for 15 min	Gas flow rate, growth temperature, position from the zinc source, and reaction time can affect the size, morphology, and density of the zinc oxide nanostructures	The growth mechanism of zinc oxide tetrapods is believed to occur by growth of four wurtzitic arms from an octahedral zinc-blende embryo, each at a 109.5° angle from the adjacent one. The tapered ends of some of the tetrapod arms indicate continued growth of zinc oxide when the oxygen flow had been turned off but residual	ZnO nanotetrapods: arm lengths, 0.5–3.5 µm and diameters of 120–350 nm		51



Table 1 (Contd.)

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications	Advantages	Ref.
	Zn and O <sub>2</sub>	Heating in a furnace at 700 °C, 800 °C and 900 °C, 50 sccm of oxygen flow for 2 h	Different evaporation temperatures	oxygen remains in the growth chamber	Wurtzite; ZnO tetrapods with an arm diameter of 22 nm and length of 90 nm. ZnO tetrapods have excellent supercapacitive performance. The maximum capacitance is 160.4 F g <sup>-1</sup> at a current density of 1.0 A g <sup>-1</sup> . Excellent capacitance retention of 94.3% over 1000 cycles		52
Hydrothermal method	ZnCl <sub>2</sub> and NaOH	pH 5–8 100 °C 160 °C 180 °C 200 °C 220 °C	Reaction temperature and template agents (organic compounds) 10 h 6 h 6 h 5 h		As temperature was increased, the ZnO particle morphologies changed Bullet-like; 100–200 nm Rod-like; 100–200 nm Sheets; 50–200 nm Polyhedra; 200–400 nm Crushed stone-like; 50–200 nm	The hydrothermal technique is a promising alternative synthetic method because of the low process temperature and great ease of controlling the particle size. The hydrothermal process has several advantages over other growth processes such as use of simple equipment, employing a growth time of 6 h the grain size was found to increase from 7 nm to 16 nm with temperature rise from 100 °C to 200 °C. The average grain size of ZnO synthesized at 200 °C for 12 h revealed an increase from 12 nm to 24 nm with elevation in concentration of NaOH from 0.2 M to 0.5 M	53
	Zn(OAc) <sub>2</sub> ·2H <sub>2</sub> O, NaOH and methanol	100–200 °C; 6–12 h; 0.2–0.5 M NaOH	Concentration of precursors (NaOH), reaction temperature and growth time		With 0.3 M NaOH and employing a growth time of 6 h the grain size was found to increase from 7 nm to 16 nm with temperature rise from 100 °C to 200 °C. The average grain size of ZnO synthesized at 200 °C for 12 h revealed an increase from 12 nm to 24 nm with elevation in concentration of NaOH from 0.2 M to 0.5 M	as use of simple equipment, low cost, large area uniform production, environment friendliness and less hazardous nature. The low reaction temperatures make this method an attractive one for microelectronics and plastic electronics. This method has also been successfully employed to prepare nanoscale ZnO and other luminescent materials. The particle properties such as morphology and size can be controlled <i>via</i> the hydrothermal process by adjusting the reaction temperature, time and concentration of precursors	54





Table 1 (Contd.)

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications Advantages	Ref.
Solvothermal method	ZnSO <sub>4</sub> , NaOH, Na <sub>2</sub> CO <sub>3</sub> and stearic acid; using the resulting ZnO nanoparticles as precursors	Reaction temperature: 60 °C; water-ethanol medium in an autoclave at 180 °C for 72–186 h	Precursors and time	The process appears to occur <i>via</i> an agglomeration/melting mechanism and leads to nanoneedles of relatively large dimensions The formation mechanism of one-dimensional nanostructures does appear to be related more to a rolling-up/surfactant-segregation process than with the characteristic ZnO crystallite growth	The ZnO wurtzite phase was formed. Average grain diameter of 27 nm using the Scherrer formula Nanoneedles with a diameter of 450–900 nm, length of 8–20 nm and aspect ratio of 0.05  Nanorods with a diameter of 40–160 nm, length of 5–8 nm and aspect ratio of 0.014	56
		180 °C: 72 h				
		180 °C: 168 h				
Emulsion or microemulsion method	Zn powder, trimethylamine <i>N</i> -oxide and 4-picoline <i>N</i> -oxide in organic solvents Zinc oleate in decane and NaOH in water or ethanol	Reaction: 24–100 h, 180 °C	Oxidants and solvents, trace amount of water in solvent	Nanowires with a diameter of 30–50 nm, length of 0.8–1 nm and aspect ratio of 0.04. Photocatalytic activities with respect to the degradation of methylene blue ZnO rod-like and particle-like nanostructures with diameters ranging in between 24 and 185 nm Morphologies obtained: spherical agglomerates, needle shapes, near-hexagonal shapes, near-spherical shapes and irregular agglomerates. Diameters obtained: 2–10 μm, 90–600 nm, 100–230 nm and ~150 nm	57	58
		Stirring: 2 h, room temperature or 90 °C, and maintaining the decane/water interface during stirring				

Table 1 (Contd.)

Chemical methods of synthesis	Precursors	Synthesis conditions	Experimental variables	Main mechanisms	Properties and applications	Advantages	Ref.
	Zinc acetate and KOH or NaOH. Cyclohexane as an organic phase, and nonylphenyl polyoxyethylene glycol ethers as a mixture of emulsifiers in emulsion formation	Stirring: 9000 rpm; destabilization: 80 °C; drying at 120 °C	Concentration of $\text{Zn}(\text{CH}_3\text{COO})_2$ solution. Precipitating agent. Amount of zinc acetate/cyclohexane ( $\text{cm}^3$ ). Dosing rate of KOH (or NaOH) to $\text{Zn}(\text{CH}_3\text{COO})_2$ ( $\text{cm}^3 \text{ min}^{-1}$ )		Morphologies such as solids (Z1), ellipsoids (Z2), rods (Z3) and flakes (Z4) with modal diameters of $\sim 396 \text{ nm}$ , $\sim 396 \text{ nm}$ , $\sim 1110 \text{ nm}$ and $\sim 615 \text{ nm}$ . Values of $8 \text{ m}^2 \text{ g}^{-1}$ , $10.6 \text{ m}^2 \text{ g}^{-1}$ , $12 \text{ m}^2 \text{ g}^{-1}$ and $23 \text{ m}^2 \text{ g}^{-1}$ could be respectively assigned to samples Z1, Z2, Z3, and Z4		59

revealed the formation of hexagonal ZnO nanostructures. SEM and TEM analyses revealed the formation of crystalline ZnO flowers in which a bunch of ZnO nanorods assembled together to form a leaf-like structure followed by flower-shaped ZnO nanostructures. The ZnO nanoflowers were each formed by the combination of 8–10 leaf-like petals as shown. The length of each petal did not exceed 800 nm. The as-synthesized ZnO nanostructures showed good antimicrobial activity towards Gram-positive bacteria *Staphylococcus aureus* as well as Gram-negative bacteria *Escherichia coli* with a MIC/MBC of  $25 \text{ mg L}^{-1}$ .  $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$  and  $(\text{NH}_4)_2\text{CO}_3$  were employed as reagents by Hong *et al.*<sup>39</sup> in their method of synthesizing ZnO NPs. XRD and TEM tests revealed particle sizes of 40 and 30 nm. Heterogeneous azeotropic distillation thoroughly prevents agglomeration and reduces the size of ZnO NPs.

Raoufi *et al.*<sup>40</sup> used the aqueous solutions of  $\text{Zn}(\text{NO}_3)_2$  and  $(\text{NH}_4)_2\text{CO}_3$  as reagents in their experiments. The precursors were synthesized by adding  $\text{Zn}(\text{NO}_3)_2$  solution in a drop-wise manner into  $(\text{NH}_4)_2\text{CO}_3$  solution under stirring. The molar ratio of  $\text{Zn}(\text{NO}_3)_2$  to  $(\text{NH}_4)_2\text{CO}_3$  was maintained at 1 : 1.5. The precipitate obtained was filtered. It was further repeatedly washed with deionized water and ethanol. Subsequently, it was subjected to drying at 100 °C for 6 h. The precursor was annealed at temperatures ranging in between 250 and 550 °C for 4 h to obtain ZnO NPs. XRD analysis demonstrated an enhancement in crystallinity with a rise in annealing temperature. Using the Scherrer equation, the crystallite size was however found to increase from 8.34 to 27.58 nm as the annealing temperature was increased from 250 to 550 °C. The mean crystallite size obtained from BET study also confirmed an increase from 8.60 to 42.51 nm upon an increase in annealing temperature from 250 to 550 °C. The conspicuous incongruence of crystallite size caused agglomeration with a rise in temperature due to the increased growth rate.

In the precipitation method of synthesizing nanopowders, it is more or less a ritual these days to use surfactants that would enable control over the growth of particles with the simultaneous prevention of coagulation and flocculation of particles thereby preventing an appreciable reduction in the final yield. The surfactants act as chelates encapsulating the metal ions in an aqueous medium. Wang *et al.*<sup>41</sup> used  $\text{ZnCl}_2$  and  $\text{NH}_4\text{OH}$  and a cationic surfactant, CTAB (cetyltrimethyl-ammonium bromide), for the generation of ZnO NPs. The formation of sharply crystalline ZnO NPs with a wurtzite structure and crystallite size of 40.4 nm was confirmed by XRD data, while TEM examination of the powder bore out the formation of spherical nanoparticles of size 50 nm.

Nano-sized crystals of ZnO possessing a wide variety of shapes were generated using  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  and NaOH by Li *et al.*<sup>42</sup> with sodium dodecyl sulfate (SDS) and triethanolamine (TEA) as cationic surfactants. These surfactants were found to exert great influences on the shape and size of the ZnO NPs. Heating a  $\text{Zn}(\text{OH})_2$  precursor in  $\text{H}_2\text{O}$  solvent at 101 °C for 4 h gave rod-like ZnO crystals, with diameters ranging from 400 to 500 nm and an average length of about 3.6  $\mu\text{m}$  (aspect ratio 8 : 1). When 0.01 M SDS was added into the reaction solution (keeping the other conditions identical), nut-like ZnO crystals

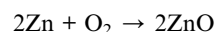
### 2.3 Sol-gel method

Suwanboon *et al.*<sup>43</sup> using  $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ , polyvinyl pyrrolidone (PVP) and NaOH prepared nano-structured ZnO crystallites *via* the sol-gel method. The XRD characterization revealed a wurtzite structure having an average crystallite size of about 45 nm. The role of PVP at its different concentrations on the morphology was checked. There occurred a shift from a platelet-like to a rod shape with an increase in PVP concentration. TEM images bore out the grain size of platelet-like ZnO to be 150 nm while the diameter of the rod-shaped ZnO was likewise determined to be 100 nm. In another sol-gel method-based synthesis by Benhebal *et al.*<sup>44</sup> zinc acetate dihydrate and oxalic acid were used to generate ZnO nanopowder with ethanol as a solvent which showed a hexagonal wurtzite structure. The crystallite size obtained from the Scherrer equation was found to be 20 nm. The SEM micrograph confirmed the formation of uniform, spherically shaped ZnO nanoparticles. BET analysis revealed a surface area of  $10 \text{ m}^2 \text{ g}^{-1}$ . This was characteristic of a material with low porosity, or a crystallized material.

In a study conducted by Ristic *et al.*<sup>46</sup> nano-structured ZnO crystallites were obtained using the sol-gel route. From XRD examination and using the Scherrer formula, the average value of the basal diameter of the cylinder-shaped crystallites was found to be 25–30 nm, while the height of the crystallites was 35–45 nm. The sol-gel method presents a host of advantages in comparison with the previously mentioned methods. Prime amongst its merits are the low cost of the apparatus and raw materials, reproducibility and flexibility of generating nanoparticles.<sup>47</sup>

In this process, zinc and oxygen or oxygen mixture vapours are transported and are allowed to react with each other resulting in the formation of ZnO nanoparticles. The vapour transport method has been emphasized because of the easy control of thicknesses, morphologies and crystal structures of ZnO films and nanostructures by varying the precursor gas, substrate temperature and substrate materials.<sup>48</sup> Kong and Wang<sup>49</sup> reported free-standing ZnO nanobelts that grew along the *a*-axis, and their large top and bottom surfaces were the polar (0001) facets. The structurally controlled, high-purity and high-yield ZnO nanobelts were synthesized by a solid-vapor process. SEM and TEM analyses show that the product has a belt shape with widths of 10–60 nm, thickness of 5–20 nm and lengths up to several hundreds of micrometers. The dominant component of the as-synthesized sample is nanobelts with a uniform size distribution, but a significant number of nanobelts have a ring shape, which has not been found for single crystal nanobelts or nanowires of any other materials.

ZnO nanoparticles were also prepared on Au-coated Si substrates by Wu *et al.*<sup>48</sup> using water vapour and oxygen gas. The advantage of using a Au catalyst is the fast growth rate of the oxide material because of the catalytic effect of Au, but the main disadvantage is the remains of the Au catalyst, which may influence the purity of the ZnO nanostructures. Au covered Si substrates were prepared by depositing a 200 nm thick layer of Au film on cleaned Si(100) wafers using radio frequency sputtering (JC500-3/D) from an Au target. The synthesis procedure included the following steps: (a) flowing Ar gas through the quartz tube for 0.5 h to remove the air in it; (b) increasing the furnace temperature to 800 °C with the Ar gas flow; (c) maintaining the temperature at 800 °C for 1 h and replacing the Ar flow by a mixture of either Ar/H<sub>2</sub>O vapour or Ar/O<sub>2</sub> gas. The pressure of the quartz tube is maintained at 0.03–0.05 MPa; (d) switching off the furnace and cooling down the sample to RT at a rate of 7 °C per minute. The reactions of metallic Zn with H<sub>2</sub>O vapour and O<sub>2</sub> gas can be written, respectively, as follows:



Novel one-dimensional single-crystalline ZnO nanorod and nanoneedle arrays on a Cu catalyst layer-coated glass substrate

were investigated by Alsultany *et al.*<sup>50</sup> via a simple physical vapour deposition method by thermal evaporation of Zn powder in the presence of O<sub>2</sub> gas. The ZnO nanorods and nanoneedles were synthesized along the *c*-axis growth direction of the hexagonal crystal structure. The diameter and growth rate of the high-quality and well oriented one-dimensional ZnO nanostructures were achieved as a function of varying growth temperature and growth time. At 450 °C, ZnO nanorods were uniformly distributed at a high density on the entire substrate surface and quasi-aligned, and small average diameters were obtained. The diameters and lengths of the obtained nanorods were in the range of 19–27 nm and 2.8 μm, respectively. When the temperature was increased to 550 °C, ZnO nanorods grew perpendicular to the substrate, uniformly throughout their length, and with more consistent shape and dimensions, with approximately 85 nm width and 3.8 μm length. The morphological change and distribution occurred at a growth temperature of 650 °C, and ZnO nanorods with a hexagonal shape at the tips of rods of hexagonal hierarchical structures were formed. These rods possessed a typical hierarchical structure with lengths and diameters of approximately 190–350 nm and 3.9 μm, respectively, whereas short nanorods with a diameter of 95 nm and length of 900 nm were observed on the tip of each rod of hexagonal hierarchical structures. As Cu metal catalysts were used in the study, the growth mechanism of 1D ZnO nanostructures presented therein followed the VLS method. This method could be divided into three stages, as follows: first, the Zn vapor and catalytic Cu formed liquid alloy droplets during the heating process at a certain temperature, representing the initial stage of the nucleation process. Second, crystal nucleation occurred upon gaseous species adsorption until supersaturation was reached, and the formed sites served as nucleation sites on the substrate. Finally, the axial growth of the nanorods began from these sites. Based on this study of the mechanism in the presence of Cu metal catalysts at different growth temperatures and according to the nucleation theory of the VLS growth mechanism, the Cu catalyst nanoclusters formed because of capillarity, which caused beading of the Cu layer at high growth temperature. Consequently, the Cu–Zn alloy process reached a certain solubility depending on the temperature; then, the Zn vapor began to precipitate out at the interface between the surface and droplet. That in turn determined the diameter and size of the nanostructures depending on the size of the liquid alloy droplets. Notably, large-scale ZnO nanorods with a lower diameter were formed at a low growth temperature of 450 °C. The Zn metal powder (melting point of 419 °C) vapor pressure at 450 °C was sufficiently high to investigate the growth of ZnO nanorods on the glass substrate via the VLS method, and the decrease in Zn vapor as a result of the decrease in the growth temperature led to a low lateral growth rate compared with the axial growth rate of the 1D nanostructure. In contrast, the higher growth temperature could also lead to the formation of hierarchical nanostructures. In addition, at high growth temperature along with the consumption of the Zn vapor during growth, the diameter of the nanorods markedly decreased. This condition consequently caused the production of rods with a typical hierarchical

structure. At a growth time of 30 min, ZnO nanorods were obtained with a diameter of 19–27 nm and a length of 2.8 μm. When the growth time increased to 45 min, nanoneedles were obtained. The needles exhibited mean diameters of 65–190 nm and length of 3.2 μm. On the other hand, nanoneedles grown at 60 min were approximately 80–250 nm in diameter and 3.8 μm in length.

Diep and Armani<sup>51</sup> designed a flexible light-emitting nanocomposite based on ZnO nanotetrapods (NTPs) which they prepared using a vapour transport technique. The CVT synthesis of the ZnO NTPs was self-catalyzed. In the TEM images, the lattice fringes were clearly visible, indicating the single-crystalline nature of the nanostructures. The lattice spacing was found to be 2.6 Å, indicating growth in the [0001] direction. X-ray diffraction (XRD) and energy-dispersive X-ray spectroscopy (EDX) analysis were also performed to confirm the crystal structure and elemental composition of the NTPs. Based on an analysis of the TEM and SEM images, the ZnO NTP arm lengths ranged from 0.5 μm to 3.5 μm and the diameters varied from 120 nm to 350 nm.

Luo *et al.*<sup>52</sup> also constructed ZnO tetrapods as potential electrode materials for low-cost and effective electrochemical supercapacitors using an oxidative-metal-vapor-transport method. The SEM images of the ZnO tetrapods collected at different temperatures showed that the products obtained were pure and uniform, and the tetrapods consist of four arms branching from one center, and the angles between the arms were nearly the same, analogous to the spatial structure of the methane molecule. As for the size variation with collected temperatures, it transpired that smaller size tetrapods were obtained with lower evaporation temperature. This demonstrated the power of the technique for controlling the size of the tetrapods. ZnO tetrapods with arms as thin as about 170 nm and shorter than 4000 nm were revealed by SEM analysis. The XRD pattern of the ZnO tetrapods showed that all the diffraction peaks could be indexed to a wurtzite 5 structure with lattice constants of  $a = 0.324$  nm and  $c = 0.519$  nm. The TEM and high resolution TEM (HRTEM) images of the ZnO tetrapods revealed that the arm diameter and length of the tetrapods are, on average, about 22 nm and 90 nm, respectively. The HRTEM image of a single arm revealed clear fringes perpendicular to the arm axis and these fringes were spaced by about 0.25 nm consistent with the interplanar spacing of (0002) suggesting that the nanowire growth direction was along [0001].

## 2.5 Hydrothermal method

The hydrothermal method is a technique often used for the easy generation of nanoparticles. This method of chemical synthesis bears an inherently favourable attribute of being harmless to the environment. Conventionally, the method involves the use of an autoclave. The substrates in the form of a mixture are placed within the autoclave and then subjected to a gradual heating up to a temperature of 100–300 °C. The heating continues over a specific period of time and cooling then follows. This causes the formation of crystal nuclei followed by nuclei growth.





A 1 : 2 mixture of  $\text{ZnCl}_2$  and NaOH in an aqueous medium was employed by Chen *et al.*<sup>53</sup> A white  $\text{Zn}(\text{OH})_2$  precipitate was formed and hydrothermal treatment yielded ZnO NPs. XRD and TEM tests were conducted to ascertain the average size and morphology of the ZnO NPs so formed. Two factors, namely temperature and reaction time, were found to exert great effects on the size and architecture of ZnO NPs. On increasing the temperature, the particle morphology was found to change from rod-like to polyhedra. Further in the study, it was also revealed that different templates cast varying influences on the size and morphology of the nanostructured particles. Furthermore, a noticeable rise in crystallinity and lengthening of particle size with pH were recorded as a shortcoming of the procedure.

Aneesh *et al.*<sup>54</sup> carried out an experiment in which they used  $\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$ , NaOH and methanol as reagents. The ZnO NPs thus formed had a hexagonal wurtzite structure. XRD analysis demonstrated an enhancement in average grain size with rising temperature and concentration of the substrates. The average grain size of ZnO NPs prepared from 0.3 M NaOH employing a growth time of 6 h was found to increase from 7 to 16 nm with temperature rise from 100 to 200 °C. The average grain size of ZnO synthesized at 200 °C for 12 h revealed an increase from 12 to 24 nm with elevation in concentration of NaOH from 0.2 M to 0.5 M.

This process has many advantages over other methods. Organic solvents do not find use in this process. This coupled with the omission of supplementary processes like grinding and calcination within the ambit of the method endows it with the much sought after eco-friendly character. Low operating temperatures, the diversified morphologies and sizes of the resulting nano-crystals depending on the composition of the starting mixture and the process temperature and pressure, the greatly pronounced crystallinity of the nanoparticles and their high purity are factors that surely make the process more advantageous than others.<sup>54,55</sup>

## 2.6 Solvothermal method

The prime objective of using the solvothermal technique of generating a wide range of nano-structured metal oxides is to achieve enormous surface area, marked crystallinity and great thermal stability. Thermal decomposition of an organometallic compound in an organic medium forms the basis of this procedure. A solvothermal method was adopted by Segovia *et al.*<sup>56</sup> to fabricate ZnO NPs. The work resulted in the production of ZnO with a range of morphologies – nanoneedles, nanorods and nanowires – obtained under different solvothermal conditions.  $\text{ZnSO}_4$ , NaOH,  $\text{Na}_2\text{CO}_3$  and stearic acid were employed as chemical reagents while ethanol–water mixtures were used as solvents. Constant agitation of the reaction mixture consisting of  $\text{ZnSO}_4$ ,  $\text{Na}_2\text{CO}_3$  and NaOH at 60 °C yielded the ZnO precursor. XRD data revealed the formation of the ZnO wurtzite phase. Estimation of particle size using the Scherrer formula approach indicated an average grain diameter of 27 nm. With ZnO nanoparticles as precursors and 180 °C thermal treatment carried out for 72 h, nanoneedles having a diameter of 450–900 nm, length of 8–20 nm and aspect ratio of 0.05 were formed. With ZnO

nanoparticles and stearic acid as precursors and 180 °C thermal treatment carried out for 168 h, nanorods having a diameter of 40–160 nm, length of 5–8 nm and aspect ratio of 0.014 were formed. With the  $\text{ZnO}(\text{stearic acid})_{0.38}$  nanocomposite as a precursor and 180 °C thermal treatment carried out for 168 h, nanowires having a diameter of 30–50 nm, length of 0.8–1 nm and aspect ratio of 0.04 were formed.

Chen *et al.*<sup>57</sup> also used a solvothermal route to generate ZnO NPs. They eventually prepared nano-structured ZnO crystals that were devoid of hydroxyl groups. They carried out a reaction of zinc powder with trimethylamine *N*-oxide ( $\text{Me}_3\text{N} \rightarrow \text{O}$ ) and 4-picoline *N*-oxide (4-pic  $\rightarrow$  NO). The medium for the reaction was a mixture of organic solvents (toluene, ethylenediamine (EDA) and *N,N,N',N'*-tetramethylethylenediamine (TMEDA)) contained in an autoclave which was kept at 180 °C. It was observed that the size and morphology of the ZnO nanoparticles/nanowires were greatly influenced by the oxidants used and the ligating capacities of the solvents. The ramifications of the presence of water in the system were additionally investigated. It emerged that the presence of traces of water catalyzed the zinc/4-picN  $\rightarrow$  O reaction and exerted an effect on the size of the nano-structured ZnO crystallites thus obtained. Depending on the reaction conditions, the ZnO nanostructures had diameters ranging in between 24 and 185 nm. The solvothermal synthesis method has many advantages. Foremost among them is the fact that reactions can be carried out under determined conditions. As a result, nano-structured ZnO with a range of architectures can be generated by exercising due control over the reaction conditions.

## 2.7 Method using an emulsion or microemulsion environment

Emulsion techniques have recently gained extensive use in preparing ZnO nanoparticles. Vorobyova *et al.*<sup>58</sup> employed an emulsion to precipitate ZnO by allowing zinc oleate (mixed in decane) to react with sodium hydroxide (mixed in ethanol or water). The following reaction illustrates the entire process:



SEM and XRD analysis showed that the particle size and phase location were both dependent upon the conditions (ratio of two-phase components, substrates and temperature) employed for the accomplishment of the process. Depending on the process conditions, ZnO NPs with different particle morphologies were obtained. The morphologies that formed during the process included spherical agglomerates, needle shapes, near-hexagonal shapes, near-spherical shapes and irregular agglomerates. These NPs further had a wide range of diameters. Some had diameters ranging in between 2 and 10  $\mu\text{m}$ , while the diameters of others ranged from 90 to 600 nm, some others had diameters in between 100 and 230 nm and yet others were characterized by diameters hovering around 150 nm.

Kołodziejczak-Radzimska *et al.*<sup>59</sup> used zinc acetate and KOH or NaOH in an emulsion system. For the generation of an



emulsion, cyclohexane was utilized. Cyclohexane was held to have furnished a ready organic phase, and also essayed the role of a surfactant that wasn't ionic. In this method for emulsion formation cyclohexane was used as an organic phase, and nonylphenyl polyoxyethylene glycol ethers NP3 and NP6 were used as a mixture of emulsifiers. By tailoring the ZnO precipitation process by way of altering the precipitating agent, substrate ingredients and the tempo of substrate dosing, an amazing variety of ZnO nanostructures were designed. Four samples were obtained, labelled Z1, Z2, Z3, and Z4, composed of particles of different shapes. Morphologies such as solids (Z1), ellipsoids (Z2), rods (Z3) and flakes (Z4) with modal diameters of  $\sim 396$  nm,  $\sim 396$  nm,  $\sim 1110$  nm and  $\sim 615$  nm were obtained. They were further characterized by their considerable surface areas. Values of  $8\text{ m}^2\text{ g}^{-1}$ ,  $10.6\text{ m}^2\text{ g}^{-1}$ ,  $12\text{ m}^2\text{ g}^{-1}$  and  $23\text{ m}^2\text{ g}^{-1}$  could be respectively assigned to samples Z1, Z2, Z3, and Z4.

If a surfactant possessing balanced hydrophilic and lipophilic properties is used in the right proportion, a different oil and water system will be produced. The system remains an emulsion, but exhibits some characteristics that are different from emulsions. These new systems are "microemulsions". The drop size in a microemulsion is significantly smaller than in an emulsion, and lies in the range  $0.0015\text{--}0.15\text{ }\mu\text{m}$ .<sup>60,61</sup> In contrast to emulsions, microemulsions form spontaneously under appropriate conditions. This synthesis method does not require any complex preparation procedure, sophisticated equipment or rigorous experimental conditions, but still provides possibilities in controlling the size and morphology of the ZnO powders in a size scale approaching nanometers. Even though the product yield is low, the narrow size distribution due to well-dispersed cage-like small reactors ( $5\text{--}100$  nm) formed under uniform nucleation conditions is the superior aspect of the ZnO nanoparticles obtained by microemulsion routes. Such low-dimensional uniform ZnO nanostructures offering size and morphology dependent tunable electrical and optical properties are of particular technological interest for applications such as quantum dots, UV-emission optoelectronic and lasing devices, and transparent conducting thin films.

Li *et al.*<sup>62</sup> prepared ZnO NPs by a facile process based on microemulsion. The synthesized ZnO NPs were marked by a wide range of morphologies and further exhibited significant optical properties. The microemulsion (ME) was generated by using a solution of heptane and hexanol in the mole ratio 3 : 1 as the oil phase and polyoxyethylene *tert*-octylphenyl ether (Triton X-100) as nonionic surfactant. A definite amount of Triton X-100 was put in the oil phase, and the mixture was subjected to stirring for a while up until the mixture became transparent. Thus, a microemulsion (ME) consisting of 0.20 M Triton X-100 in the oil phase was prepared. The next step entailed the preparation of two microemulsion solutions, namely ME-1 and ME-2, containing different reactants. 3 mL of a 0.25 M  $\text{Zn}(\text{NO}_3)_2$  aqueous solution having various concentrations of PEG400 additive was added to 30 mL ME. This microemulsion solution was named ME-1. Likewise, ME-2 was obtained by adding 3 mL of a 0.5 M NaOH aqueous solution to 30 mL ME. Next, ME-1 was first added to ME-2 in a slow manner with constant stirring. This in turn resulted in a mixture that

was subjected to hydrothermal treatment at  $140\text{ }^\circ\text{C}$  for 15 h in a 100 mL Teflon-lined autoclave. The autoclave was then left to cool down to room temperature by itself. Collection of precipitates was done by way of their centrifugation at 5000 rpm. This was followed by washing and then drying in air at  $60\text{ }^\circ\text{C}$ . XRD peaks confirmed the hexagonal wurtzite structure of the prepared ZnO NPs. The marked crystallinity of the nanoparticles was figured out from the sharpness of the diffraction peaks. The crystallite size was found to decrease from 40 to 27 nm with a rise in the concentration of PEG400 additive from zero to 50.0% in  $\text{Zn}(\text{NO}_3)_2$  aqueous solution exhibiting the regulatory influence of microemulsions as well as the PEG400 additive on the nucleation and growth of ZnO NPs and their end architecture. TEM analysis further revealed that the morphology of the ZnO NPs underwent an alteration from needle-like to columnar and finally to spherical as the PEG400 concentration rose from zero to 50.0%.

Yildirim and Durucan<sup>63</sup> also synthesized ZnO NPs through the use of microemulsions. They made an endeavour to reshape the microemulsion *modus operandi* with an eye to generate monodisperse ZnO nanostructures. They subjected the zinc complex precipitate obtained in the course of the microemulsion method to thermal decomposition. Subsequent calcination was adopted. The use of glycerol as the internal phase of a reverse microemulsion imparted the intended modification. The synthesized ZnO NPs had spherical shapes. They were monodisperse and their diameter measured in between 15 and 24 nm.

All the procedures involving chemical synthesis of ZnO NPs generate a few toxic chemicals and their adsorption on the surface increases the likelihood of harmful effects being wielded in medical applications. Further, these approaches include reactions requiring high temperature and intense pressure for their commencement while some reactions require operations in an inert atmosphere or under inert conditions. Toxic materials such as metallic precursors, toxic templates and capping agents and even  $\text{H}_2\text{S}$  find application in quite a few chemical routes.<sup>64</sup> Very often toxic substances are employed for the generation of nano-structured particles and for their stabilization as well. This in turn produces secondary products and residues that are detrimental to the ecosystem.<sup>65,66</sup>

### 3. Green methods for the synthesis of ZnO nanoparticles

The usual procedures adopting the chemical routes are costly. They often involve the use of chemical substances as reducing agents and these substances more often than not turn out to be toxic.<sup>67</sup> Green routes of synthesis of nanoparticles cut down the risk of pollution at the originary level. These procedures lay greater focus on prevention of generation of wastes than their treatment or removal after they are formed. The principle emphasizes upon the choice of reagents which are eco-friendly. Thus, eco-friendly chemicals and safe solvents such as water are used alongside gentle natural extracts in such procedures. Although chemical methods are quick and easier for the





generation of nanoparticles the biogenic techniques are superior and eco-friendly.<sup>68–70</sup> Biogenic routes involving the use of extracts generated from microorganisms and plants have thus been preferred over chemical procedures of fabricating ZnO NPs. Besides, ZnO NPs obtained by way of green methods demonstrate pronounced biocidal activity against a host of pathogens in comparison with chemically generated ZnO NPs.<sup>71</sup> Bio-inspired procedures of nanoparticle generation have of late been observed to employ systems including bacteria, yeast and other types of fungi.<sup>72</sup> Nevertheless, a greatly detailed procedure of maintaining cell cultures and intracellular synthesis causes inevitable complications in the use of microorganisms in the generation of nanoparticles while numerous purification measures additionally increase the difficulty. Table 2 is a brief presentation of the various methods adopted by researchers to fabricate ZnO nanostructures using green routes. The detailed synthesis conditions, experimental variables, surface morphology, mechanisms and applications are discussed (Table 2).

An extract prepared from Ajwain (*Carom-Trachyspermum ammi*) seeds has also been used to synthesize ZnO NPs.<sup>70</sup> The work boasts of its operation under ambient temperature conditions. The ZnO NPs were found to have a wurtzite structure. The synthesized ZnO nanostructures were morphologically characterized by FE-SEM images. The ZnO nanostructure showed uniform hexagonal plates, as well as irregular and highly aggregated nanoparticles with a rough surface. The average diameter of the nano-sized ZnO clusters has been observed to be ~41 nm. XRD results showed an increase in interplanar spacing with an increase in the extract volume from 0.2474 nm to 0.2765 nm with a simultaneous decrease in crystallite size from 39.51 nm to 28.112 nm. The band gap also fell from 3.592 eV to 3.383 eV as the amount of extract increased. Phytoconstituents in the extract thus evidently played a key role of reductants and furthermore acted as capping agents in the generation and stabilization of ZnO NPs.

Jamdaini *et al.*<sup>72</sup> used an aqueous flower extract of *Nyctanthes arbortristis* for making ZnO NPs. The starting materials consisted of zinc acetate dihydrate and sodium hydroxide. XRD results showed an average crystallite size of 16.58 nm while TEM analysis revealed that the individual particle size ranged within 12–32 nm and the nanoparticles were obtained in the form of aggregates. In a very recent study,<sup>73</sup> *Ulva lactuca* seaweed extract was used to prepare ZnO nanoparticles. XRD analysis revealed strong characteristic peaks of ZnO suggesting high crystallinity of the synthesized material. Further, the average crystallite size thus calculated was found to range in between 5 and 15 nm. TEM micrographs revealed an agglomeration of asymmetrically shaped NPs bearing an average crystallite size of 15 nm.

*Muraya koenigii* seed extract was also recently reported to have been used as a stabilizer as well as a reductant in the preparation of ZnO NPs.<sup>74</sup> Sharp diffraction peaks in XRD results indicated remarkable crystallinity of the NPs whose average crystallite size was calculated to be 70–100 nm. Both SEM and TEM micrographs revealed nanoparticles with an average size of about 100 nm and bearing a wide range of

morphologies – spherical, triangular, radial, hexagonal, rod-like and rectangle-shaped.

One recent experiment used *Calotropis procera* leaf extract and  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  to synthesize ZnO NPs.<sup>75</sup> An XRD test confirmed a hexagonal wurtzite structure of the nanoparticles with marked crystallinity. The average crystallite size was calculated using the Scherrer equation and found to be 24 nm. Diffuse Reflectance Spectroscopy (DRS) revealed a band gap of 3.1 eV for the synthesized nanoparticles. In the FT-IR analysis of the synthesized ZnO NPs, a peak attributed to the metal–oxygen bond of ZnO appeared in between 500 and 700  $\text{cm}^{-1}$ . Further, a conspicuous shift and broadening of peaks corresponding to functional groups like hydroxyl, aldehyde, amine, ketone, and carboxylic acid suggests their participation in the stabilization of ZnO by the extract. Surface attachment of groups like aldehyde, amine, phenol and terpenoid enhances stabilization additionally allowing the extract to function as a bio-template thereby preventing aggregation of ZnO NPs. TEM images revealed an average particle size of 15–25 nm, while SAED and HR-TEM further confirmed the high crystallinity of the material prepared.

The effects of *Artocarpus heterophyllus* leaf extract and varying temperatures on the morphology and properties of the ZnO NPs thus prepared were studied by Vidya C. *et al.*<sup>76</sup> XRD results show an increase in crystallinity and average crystallite size with temperature, the diffraction peaks being increasingly sharper and narrower with temperature. The particles were all spherical and a grain size of 50 nm was obtained from SEM images. SEM analysis also shows similar trends of size and morphology upon temperature variation. TEM analysis revealed a particle size of ~10–15 nm at 400 °C, ~15–25 nm at 600 °C and ~25–30 nm at 800 °C. This further corroborated the results of XRD and SEM tests. Diffuse Reflectance Spectroscopy (DRS) showed a decrease in the calculated band gaps with increasing calcination temperatures.

Archana *et al.*<sup>77</sup> used *Moringa oleifera* natural extract and  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  for the preparation of ZnO NPs. They took different volumes of the extract, viz. 2, 6, 10 and 14 mL, to prepare ZnO NPs which were accordingly labeled ZnO-2, ZnO-6, ZnO-10 and ZnO-14. The PXRD results of all the samples showed great crystallinity. They had a hexagonal wurtzite structure. And the average crystallite size was found to be 21.6 nm. Field Emission Scanning Electron Microscopy (FE-SEM) analysis showed highly crystalline ZnO-10 and ZnO-14 having a spherical shape and average crystallite size of 20–150 nm. HR-TEM micrographs revealed *d*-spacing of 0.28 and 0.19 nm for the (001) and (101) planes of wurtzite ZnO. The band gaps calculated using the results from Diffuse Reflectance Spectroscopy (DRS) had values of 2.92 eV for ZnO-2, 3.05 eV for ZnO-6, 3.12 eV for ZnO-10 and 3.10 eV for ZnO-14. The increase in band gap with the amount of fuel was attributed to quantum size effects.

In their research work, Rajeswari Rathnasamy *et al.*<sup>78</sup> used papaya leaf extract for the synthesis of ZnO NPs. Both FESEM and TEM data revealed an average size of ~50 nm for the individual nanoparticles. The extract of *Nephelium lappaceum* L. (rambutan) peels (a natural ligation agent) was put into use for



Table 2 Green synthesis of ZnO nanoparticles

Source	Synthesis conditions	Experimental variables	Shape/morphology	Mechanism and applications	Size	Ref.
<i>Carom-Trachyspermum ammi</i> seed extract	2 mL of the extract was slowly added dropwise to a 25 mL solution of 0.05 M Zn(NO <sub>3</sub> ) <sub>2</sub> . Magnetic stirring for 2 h at 50 °C. Centrifugation and drying at room temperature at 35 °C	—	Uniform hexagonal plates, irregular and highly aggregated nanoparticles with a rough surface	Anti-bacterial activities on both Gram-positive ( <i>Staphylococcus aureus</i> ) and Gram-negative ( <i>Pseudomonas aeruginosa</i> ) bacteria	~41 nm	70
<i>Nyctanthes arbor-tristis</i> flower extract	0.01 M solution of zinc acetate and flower extract were added at a pH of 12 and the acetate, pH, and solution was stirred for 2 h. A white precipitate was obtained and dried at 60 °C overnight	Concentration of zinc acetate, pH, and temperature	Aggregate of nanoparticles	Nanoparticles were tested for their antifungal potential and were found to be active against all five tested phytopathogens with the lowest MIC value recorded being 16 µg mL <sup>-1</sup>	~12–32 nm	72
<i>Ulva lactuca</i> seaweed extract	<i>U. lactuca</i> extract was added into 1 mM zinc acetate and kept under magnetic stirring at 70 °C for 3–4 h. The mixture was centrifuged at 4000 rpm for 10 min and the solid product was collected and heated at 450 °C for 4 h	—	Agglomeration of asymmetrical shaped nanoparticles	Excellent photocatalytic activity on methylene blue. High antibiofilm activity on 4 species of Gram-positive and -negative bacteria	~15 nm	73
<i>Muraya koenigii</i> seed extract	20 mL of <i>Muraya koenigii</i> seed extract was mixed with 80 mL of zinc nitrate (Zn(NO <sub>3</sub> ) <sub>2</sub> ) and 2.0 M NaOH solution was added with vigorous stirring for 3–5 h, and incubated overnight at room temperature. Zinc oxide nanoparticles (white precipitate) were washed with distilled water and ethanol and dried at room temperature	—	Spherical, triangle, radial, hexagonal, rod and rectangle shaped	ZnO nanoparticles used for antimicrobial activities using human pathogenic bacterial and fungal species	~100 nm	74
<i>Calotropis procera</i> leaf extract	Leaf extract was added to 50 mL distilled water and heated up to 70 °C and 6 g of Zn(NO <sub>3</sub> ) <sub>2</sub> ·6H <sub>2</sub> O was added and evaporated. Calcined at 400 °C for 3 h	—	Spherical	When Zn(NO <sub>3</sub> ) <sub>2</sub> ·6H <sub>2</sub> O is mixed with <i>C. procera</i> leaf extract, the Zn <sup>2+</sup> ions disperse consistently and form a complex with active sites of hydroxyl groups. Polyphenolic molecules that interact with divalent Zn <sup>2+</sup> cations forming a bridge between two hydroxyl groups from two different chains come from the polyphenolic groups in close contact with Zn <sup>2+</sup> . The divalent cations keep the molecules together and form various structures of zinc complex. Photocatalytic degradation of methyl orange with an efficiency of 81% within 100 min under UV light	~15–25 nm	75





Table 2 (Contd.)

Source	Synthesis conditions	Experimental variables	Shape/morphology	Mechanism and applications	Size	Ref.
<i>Artocarpus heterophyllus</i> leaf extract	5 g of zinc nitrate hexahydrate was added to 150 mL leaf extract and heated at 80 °C and calcined in a muffle furnace at 400, 600 and 800 °C for 1 h	Calcination temperature	Spherical	Photo-degradation of Congo red dye	~10–15 nm at 400 °C, ~15–25 nm at 600 °C and ~25–30 nm at 800 °C	76
<i>Moringa oleifera</i>	2.97 g of zinc nitrate hexahydrate was dissolved in <i>Moringa oleifera</i> natural extract and heated on a hot plate with a stirrer to form a gel kind of product and kept in a muffle furnace maintained at 400 °C	Concentration of <i>Moringa oleifera</i> natural extract and time	Clusters of spherical nanoparticles	ZnO nanoparticles with smaller size show better H <sub>2</sub> evolution rates up to 360 μmol h <sup>-1</sup> g <sup>-1</sup> . It is noteworthy that ZnO nanoparticles prepared via novel green synthesis exhibit oxygen vacancies and register enhanced photocatalytic activity as well as good photostability	100–200 nm	77
<i>Carica papaya</i> leaf extract	To zinc acetate dihydrate (5 mmol) papaya leaf extract was added and the mixture heated at a temperature of 60 °C for 2 h under stirring at a pH of 8. Finally, it was washed with a water and ethanol mixture and dried at 80 °C for 12 h	—	Spherical	Photocatalyst for methylene blue dye degradation (complete degradation within 180 min in the presence of UV) and photo-anode with an energy conversion efficiency of 1.6% with a current density of 8.1 mA cm <sup>-2</sup> in dye sensitized solar cells	~50 nm	78
<i>Nephetium lappaceum</i> L. fruit peel extract	A volume of 50 mL was prepared and then 10 mL of rambutan peel extract was added to 0.1 M Zn(NO <sub>3</sub> ) <sub>2</sub> ·6H <sub>2</sub> O with heating at a temperature of 80 °C for 2 h and then incubated at room temperature for 1 day to form zinc-ellagate and dried in an oven at 40 °C for 8 h. ZnO nanoparticles were obtained on direct decomposition of the zinc-ellagate complexes in a muffle furnace at 450 °C	—	Multidimensional chain-like structures in which spherical nanoparticles were intertwined with each other	—	~20–50 nm	79
<i>Moringa oleifera</i> leaf extract	50 mL of <i>Moringa oleifera</i> extract was added to Zn(NO <sub>3</sub> ) <sub>2</sub> ·6H <sub>2</sub> O at room temperature with a pH of 5 and subjected to heat treatment in air at 500 °C for 1 h	Concentration of zinc salt and calcination temperature	Drying at 100 °C; agglomerates of spherical particles; annealing at 500 °C; nanorods in addition to the clusters of spherical nanostructures	Three chemical reactions of the solvated Zn <sup>2+</sup> ions are considered with the phytochemicals of <i>Moringa oleifera</i> , i.e. with a phenolic acid, a flavonoid and vitamin based compounds. An altered chemical behavior of L-ascorbic acid and zinc nitrate, probable oxidation of biological compound i.e. L-ascorbic acid to L-dehydroascorbic acid via free radicals, followed by	~12.27–30.51 nm	80



Table 2 (Contd.)

Source	Synthesis conditions	Experimental variables	Shape/morphology	Mechanism and applications	Size	Ref.
<i>Catharanthus roseus</i> leaf extract	An aqueous leaf extract of <i>C. roseus</i> was added to 0.025 M aqueous zinc acetate and pH adjusted to 12 and the solution was dried in a vacuum	—	Spherical	electrostatic attraction between the free radical and cation of the precursors. Electrochemical investigations by cyclic and square wave voltammetry Antibacterial activity was evaluated. Among the four bacterial species tested, <i>Pseudomonas aeruginosa</i> is more susceptible when compared with the other three species and may be used for the preparation of antibacterial formulations against <i>Pseudomonas aeruginosa</i>	~23–57 nm	81
<i>Camellia sinensis</i> leaf extract	ZnO NPs using the aqueous extract of green tea leaves. In the prepared extract zinc acetate was dissolved by way of magnetic stirring. Intense stirring was eventually applied on this solution for 5–6 h; a temperature of about 150 °C was maintained during this time. The solid mass thus obtained subsequently underwent a 4500 rpm centrifugation for 15 min; this act was repeated again. Finally washing and drying at 80 °C for 7 to 8 h yielded agglomerates of irregularly shaped ZnO NPs. UV spectroscopy analysis showed maximum absorption at about 330 nm. The size of the particles was determined using a particles size analyzer. The average diameter of the particles was found to be 853 nm	—	Agglomerates of irregularly shaped nanoparticles	These nano-sized ZnO demonstrated remarkable antimicrobial properties against Gram-positive and Gram-negative bacteria as well as against a fungal strain	~853 nm	82
<i>Citrus aurantifolia</i> fruit extract	50 mL of aqueous <i>Citrus aurantifolia</i> extract was boiled to 60–80 °C. <sup>83</sup> It was followed by the addition of a specific amount (5 g) of Zn(NO <sub>3</sub> ) <sub>2</sub> to the solution as its temperature rose to 60 °C. The reaction mixture so prepared was then boiled until a deep yellow coloured paste was left. This paste was then collected and heated in a furnace in the presence of air at 400 °C for 2 h to eventually yield a powder. This powder bearing a faint white colour was further ground in	—	Near spherical shaped nanoparticles	—	~9–10 nm	83



Table 2 (Contd.)

Source	Synthesis conditions	Experimental variables	Shape/morphology	Mechanism and applications	Size	Ref.
<i>Oryza sativa</i> rice extract	a mortar-pestle. The synthesized nanoparticles were characterized by moderate stability. They had near spherical shapes with the most probable particle-size in the range of 9–10 nm. ZnO NPs were prepared by the hydrothermal method. The method involved the use of zinc acetate, sodium hydroxide, and uncooked rice flour at several ratios at 120 °C for 18 h. The rice bio-template was found to exert considerable influences upon the size and morphology of ZnO NPs.	—	Flake-, flower-, star-, toothed-edge flake-like, rose- and rod-like structures for 0.25 g, 0.50 g, 1.0 g, 2.0 g, 4.0 g and 8.0 g uncooked rice, respectively	—	~200–800 nm, ~800–2000 nm, ~200–1000 nm, ~250–700 nm, ~200–700 nm, ~150–700 nm and ~40–100 nm for 0.25 g, 0.50 g, 1.0 g, 2.0 g, 4.0 g and 8.0 g uncooked rice, respectively	84
<i>Passiflora caerulea</i> . L. leaf extract	The leaf extract was prepared by maintaining a temperature of 70 °C for 8 min. 50 mL of aqueous 1 mM zinc acetate $[\text{Zn}(\text{O}_2\text{CCH}_3)_2 \cdot (\text{H}_2\text{O})_2]$ was prepared and subjected to stirring for 1 h. Subsequently, to this solution, a 20 mL of NaOH solution was slowly added. This was followed by a slow addition of 25 mL of plant extract. As a consequence, the color of the reaction mixture was found to change after incubation for an hour. This solution was again subjected to stirring for 3 h. The subsequent appearance of a yellow color confirmed the generation of ZnO NPs. The precipitate so obtained was centrifuged at 8000 rpm at 60 °C for 15 min. Thereafter, the pellets that resulted were dried in a hot air oven at a temperature of 80 °C for 2 h.	—	Spherical	—	~30–50 nm	85
Sucrose (as a capping agent)	Zinc acetate $[\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}]$ and sucrose ( $\text{C}_{12}\text{H}_{22}\text{O}_{11}$ ) served as the precursor and capping agent, respectively. The precursor was prepared by dissolving 4.3900 g of zinc acetate in 50 mL of double distilled water and stirring for 30 min at 60 °C. During the process, 3.4229 g of sucrose solution was slowly added. The resultant solution was stirred for 2 h at the same temperature. The solution was then bone dried at 80 °C and was calcined in an atmosphere of air at 400 °C for an hour. The end product was finely ground using an agate mortar to obtain the required	—	Granular	Carbon coated ZnO nanoparticles are used for symmetric supercapacitor device fabrication. The symmetric device yields a specific cell capacitance of 92 $\text{F g}^{-1}$ at a specific current of 2.5 $\text{A g}^{-1}$	~10–100 nm	86



Table 2 (Contd.)

Source	Synthesis conditions	Experimental variables	Shape/morphology	Mechanism and applications	Size	Ref.
	ZnO/C nanocomposite. Similarly, without sucrose we synthesized pure ZnO nanoparticles					
Whey (as a chelating agent)	Firstly, zinc citrate was obtained by mixing $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ with citric acid (CA), previously dissolved in distilled water ( $0.1 \text{ g mL}^{-1}$ ), at a molar ratio of 1 : 3. Whey (W) was subsequently added to these solutions at a mass ratio of CA : W = 60 : 40 to promote citrate polymerization. The resultant mixture was homogenized and maintained under magnetic stirring for 1 h at $80^\circ\text{C}$ to complete gelatinization. The obtained brown colored and transparent stable resin (xerogel) aspect was heated to $100^\circ\text{C}$ in order to eliminate excess water. Then, the xerogel was calcined at temperatures ranging from 200 to $1000^\circ\text{C}$ for one hour to obtain the resultant ZnO nanoparticles	Calcination temperature	Spherical	—	With an increase in calcination temperature from $400$ to $1000^\circ\text{C}$ , the size of nanoparticles increased from $18.3$ to $88.6 \text{ nm}$	87
<i>Citrus sinensis</i> fruit peel extract	An aqueous extract of orange peel was used as the biological reducing agent for the synthesis of ZnO NPs from zinc acetate dihydrate. The ZnO NPs were synthesized by mixing $2 \text{ g}$ of zinc nitrate with $42.5 \text{ mL}$ of the extracts. These mixtures were then stirred for 60 minutes and then placed in a water bath at $60^\circ\text{C}$ for 60 minutes. Subsequently, the mixtures were dried at $150^\circ\text{C}$ and then heat-treated at $400^\circ\text{C}$ for 1 hour	Annealing temperature and synthesis pH	Spherical	Ligation takes place between the functional components of the orange peel and the zinc precursor. The organic substances (flavonoids, limonoids, and carotenoids) in orange peel extract act as ligand agents. These hydroxyl aromatic ring groups, one of the extract components, form complex ligands with zinc ions. Through the process of nucleation and shaping, nanoparticles are stabilized and formed. The mixture of the organic solution is then decomposed directly upon calcination at $400^\circ\text{C}$ resulting in the release of ZnO nanoparticles. Antibacterial activities toward <i>E. coli</i> and <i>S. aureus</i> : without UV light, the bactericidal rate towards <i>E. coli</i> was over 99.9%, while the bactericidal rate towards <i>S. aureus</i> varied in the relatively wide range of 89–98%	$400^\circ\text{C}$ , $700^\circ\text{C}$ and $900^\circ\text{C}$ : $35\text{--}60 \text{ nm}$ , $70\text{--}100 \text{ nm}$ and $200\text{--}230 \text{ nm}$ , respectively. pH values of 6.0 and 8.0: $10\text{--}20 \text{ nm}$ and $400 \text{ nm}$ . pH value of 10.0: Agglomerates of blocks with lengths of $\sim 370 \text{ nm}$ and widths of $\sim 160 \text{ nm}$	88



the preparation of ZnO NPs in another investigation.<sup>79</sup> The bio-mediated ZnO NPs were found to be spherical in shape. They were characterized by diameters between 20 and 50 nm. Some of the particles were found in agglomerated form. After a day, multi-dimensional chain-like structures formed. In these chains spherical nanoparticles were found intertwined to each other.

An investigation conducted by Matinise *et al.*<sup>80</sup> used *Moringa oleifera* extract as a remarkably operative chelating agent to prepare ZnO nanoparticles. The ZnO NPs eventually obtained were characterized by a particle size in between 12.27 and 30.51 nm. The sample obtained just after drying at 100 °C consisted of agglomerates of spherical particles while that obtained after annealing at 500 °C also had nanorods in addition to the clusters of spherical nanostructures.

The biocomponents of leaves of *Catharanthus roseus* have also been utilized to prepare ZnO NPs with zinc acetate and sodium hydroxide as reagents.<sup>81</sup> SEM micrographs revealed that in addition to the individual ZnO-NPs, aggregates were also formed and they were spherical with diameter ranging from 23 to 57 nm. Sharp and clear XRD peaks confirmed high purity and excellent crystallinity. Shah *et al.*<sup>82</sup> generated ZnO NPs using the aqueous extract of green tea (*Camellia sinensis*) leaves. The size of the particles was determined using a particle size analyzer. The average diameter of the particles was found to be 853 nm. These nano-sized ZnO particles demonstrated remarkable antimicrobial properties against Gram-positive and Gram-negative bacteria as well as against a fungal strain.

In another experiment, 50 mL of aqueous *Citrus aurantifolia* extract was boiled to 60–80 °C.<sup>83</sup> It was followed by the addition of a specific amount (5 g) of Zn(NO<sub>3</sub>)<sub>2</sub> to the solution as its temperature rose to 60 °C. The synthesized nanoparticles were characterized by moderate stability. They had near-spherical shapes with the most probable particle size in the range of 9–10 nm. The extract of *Oryza sativa* rice<sup>84</sup> was also used to generate ZnO NPs. The extract has been considered a renewable bio-resource. Its abundance adds to its list of merits. The extract has also been cited as a source of bio-template that typically assists the generation of a variety of multifunctional nanostructured materials. ZnO NPs were prepared using the hydrothermal method. The method involved the use of zinc acetate, sodium hydroxide, and uncooked rice flour at several ratios at 120 °C for 18 h. The rice bio-template was found to exert considerable influences upon the size and morphology of ZnO NPs. Fig. 2 shows field emission scanning electron microscopy (FESEM) images of the samples synthesized at different concentrations of uncooked rice (UR). To investigate the effects of raw rice on the resulting ZnO morphology, FESEM was conducted on ZnO synthesized without UR (Fig. 2a and b). As seen in Fig. 2c and d, the ZnO structures were mostly flake-like structures assembling together. They were much more ordered in contrast to the one synthesized without UR (as a control) (Fig. 2a and b). The diameter of ZnO flakes dramatically decreased after adding 0.25 g UR. This was proposed to have occurred due to the inhibition of lateral growth of ZnO crystals. It was further proposed that the accessibility of the zinc ions to the ZnO crystal seeds was controlled by a bio-template.

However, the size of particles seemed to increase when the synthesis was done using 0.25 g UR. Different morphologies of the as-synthesized ZnO were observed with increasing the amount of uncooked rice to 0.5 g. Particles with a very small flower-like shape could be observed (Fig. 2e and f). A lower magnification FESEM image indicated that the mentioned structure showed denticulated petals aggregated and form larger flowers of particles. Notably the size of the ZnO particles had been obviously decreased for the sample prepared using 0.5 g UR. In addition, the tooth-like flakes were more dominant for the ZnO sample prepared using 0.5 g UR compared to the one synthesized using 0.25 g UR. Fig. 1g and h indicate the FESEM images of the ZnO sample synthesized using 1 g UR. A very unique star-like structure could be clearly observed at low to high magnification. The star-like structure contained small flakes with denticulated edges which attach to other similar flakes in the center. A closer look showed that the lateral flake acted as a substrate for other flakes to grow on the surface and form a star-like structure. It was therefore argued that the branched pattern for soft templates of starch revealed that the semicrystalline granules of starch were made from concentric rings in which the amylose and amylopectin basic components were aligned perpendicularly to the growth rings and to the granule surface. Fig. 2g and h show that the size of the star-like ZnO particles decreased in comparison with the previous lower amount of uncooked rice. In the case of ZnO crystals synthesized at 2 g UR, increasing the amount of bio-template resulted in different morphologies of ZnO particles being produced. It formed lots of agglomerated toothed-edge flakes which became a secondary unit for larger particles. The star-like shape of the particles could be perceived in some areas but aggregation seemed to be dominant and prevented clearer observation of the particles as they really are. Fig. 2k and l show the FESEM images of the as-synthesized ZnO particles synthesized using 4 g UR. The ZnO morphology changed to flower-like structures, mostly rose-like shapes. A detailed view of the flower-like particles revealed that their flakes had the largest diameter compared to other samples. In the case of ZnO synthesized using 8 g UR, a new morphology, different from other and control samples, was observed. The ZnO crystals appear mostly as rods with around 100 nm size. Moreover, agglomerated without any specific shape, particles coexisted with nanorods in the structure of ZnO synthesized using 8 g UR. Fig. 3 shows the particle size distribution of the ZnO samples synthesized using 0.25, 0.5, 1, 2, 4, and 8 g UR. The particle size distribution of ZnO synthesized without rice is also given for comparison. As shown in Fig. 3, the range of particle size for ZnO synthesized without UR lies between 200 and 800 nm. When 0.25 g UR was used in the synthesis, the size of particles increased dramatically to 800–2000 nm. Notably the size of ZnO synthesized using 0.5 g UR considerably decreased to the 200–1000 nm range. The decreasing trend continued for the sample synthesized using 1 g UR and with a size range of 250–700 nm. Although this distribution was quite similar to that of ZnO synthesized without a bio-template, it was slightly narrower. On the basis of the particle size distribution for the samples synthesized using 2 and 4 g UR, it could be clearly observed that the size of



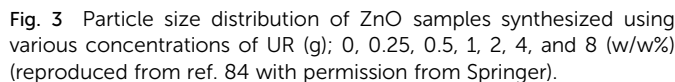


Fig. 2 FESEM images of ZnO prepared using different concentrations of uncooked rice (g): 0 (a and b), 0.25 (c and d), 0.5 (e and f), 1 (g and h), 2 (i and j), 4 (k and l), and 8 (m and n) (reproduced from ref. 84 with permission from Springer).

particles decreased to 200–700 nm and 150–700 nm, respectively. In the case of the ZnO sample synthesized using 8 g UR, the size of particles was within the nano regime, between 40 and

100 nm. As mentioned in the growth mechanism suggested by the study, adding a bio-template, which presumably acts as a flocculant, forces aggregation. Therefore, the surface-active





Sucrose was used in a study as the capping agent to synthesize a ZnO/C nanocomposite adapting the sol-gel method.<sup>86</sup> The presence of carbon in the prepared ZnO/C was confirmed through EDAX. SEM images of the ZnO/C samples indicate a wide distribution of particles ranging from 10 to 100 nm and exhibit only an irregular granular feature. This kind of surface morphology was argued to be more suitable for supercapacitor electrode materials. Electrochemical investigations of the ZnO/C electrode were carried out using cyclic voltammetry, galvanostatic charge-discharge, and electrochemical impedance spectroscopy. The ZnO/C electrode exhibits a maximum specific capacitance of 820 F g<sup>-1</sup> at a constant specific current of 1 A g<sup>-1</sup>. The symmetric aqueous supercapacitor device exhibits a specific cell capacitance of 92 F g<sup>-1</sup> at a specific current of 2.5 A g<sup>-1</sup>. The aqueous symmetric supercapacitor device achieved an energy density of 32.61 W h kg<sup>-1</sup> and a power density of approximately 1 kW kg<sup>-1</sup> at a discharge current of 1.0 A g<sup>-1</sup>. It has been found that the cells have an excellent electrochemical reversibility (92% after 400

Another report presented an efficient, environmentally friendly, and simple approach for the green synthesis of ZnO nanoparticles (ZnO NPs) using orange fruit peel extract.<sup>89</sup> The approach aimed to both minimize the use of toxic chemicals in nanoparticle fabrication and enhance the antibacterial activity and biomedical applications of ZnO nanoparticles. The sample obtained without annealing exhibited relatively small spherical particles (10–20 nm) which were coagulated in large clusters on a matrix of residual organic material from the reducing agents. In the samples annealed at 400 °C and 700 °C, the particle sizes were randomly distributed and ranged from 35 to 60 nm and 70 to 100 nm, respectively. For an annealing temperature of 900 °C, the particle size increased intensively in the range of 200–230 nm. It was thus found that the morphology and size of the ZnO NPs depended on the annealing temperature. Specifically, with increasing annealing temperature, the particle size tended to increase and shape larger particles due to crystal growth. For pH values of 4.0 and 6.0, the particles were sphere-like in shape, and were distorted with distinct grain boundaries and low coagulation. At pH = 6, the particle size was in the 10–20 nm range and exhibited relative separation. Meanwhile, for a pH of 8.0, the particles had a variable shape and were coagulated in large clusters around 400 nm in size with indistinct grain boundaries. For a pH of 10.0, the particles were coagulated into large blocks with lengths of ~370 nm and widths of ~160 nm. The ZnO NPs exhibited strong antibacterial activity toward *Escherichia coli* (*E. coli*) and *Staphylococcus aureus* (*S. aureus*) without UV illumination at an NP concentration of 0.025 mg mL<sup>-1</sup> after 8 h of incubation. In particular, the bactericidal activity towards *S. aureus* varied extensively with the synthesis parameters. This study presents an efficient green synthesis route for ZnO NPs with a wide range of potential applications, especially in the biomedical field.



## 4. Modification of zinc oxide nanoparticles

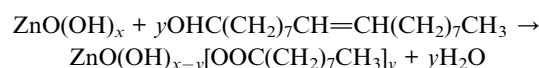
The ultimate motive behind surface modification of ZnO nanoparticles is to harness their high affinity for adhesion and agglomeration through reduction of surface energy thereby controlling their growth at the nano level. Modification of ZnO has been carried out with different classes of modifiers, namely inorganic compounds, organic compounds and polymer matrices. Inorganic compounds like  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{LiCoO}_{2.7}$ , metal ions, *etc.* when used as modifiers brought about a change of surface area or particle size, diminished the photocatalytic efficiency of the metal oxide and improved the dispersion degree of ZnO particles. Organic modifiers like silanes and carboxylic acids made the following modifications: introduction of characteristic groups on the surface of ZnO and alteration of its physicochemical properties, increase in the compatibility of ZnO with an organic matrix, reduced aggregation of particles and enhanced long-term stability in an organic matrix and improved ZnO dispersion in rubber mixtures. Modifiers such as polyethylene glycol, polystyrene, poly(methyl methacrylate), chitin, *etc.* improved the unique electrical, thermal and optical properties of ZnO/polymer composites.

Cao *et al.*<sup>90</sup> used silica and trimethyl siloxane (TMS) for modifying ZnO in order to achieve a two-fold benefit: enhancing the compatibility of ZnO and cutting down on its agglomeration in the organic phase. A chemical precipitation method using zinc sulfate heptahydrate ( $\text{ZnSO}_4 \cdot 7\text{H}_2\text{O}$ ), ammonium solution ( $\text{NH}_4\text{OH}$ ) and ammonium bicarbonate ( $\text{NH}_4\text{HCO}_3$ ) was adopted to first obtain the precursor, zinc carbonate hydroxide (ZCH). The surface of the ZCH was then successively modified by an *in situ* method using TEOS and hexamethyldisilazane (HMDS) in water. The functionalized ZCH was subjected to calcination, to yield extremely fine nanoparticles of ZnO. Reduced agglomeration was thereby effected through such functionalization of the surfaces of ZnO NPs although a lowered photocatalytic activity of the oxide was observed. Nevertheless, a marked increase in the compatibility of ZnO with the organic matrix lent credence to the method. Further, the greater shielding capacity of UV radiation renders the synthesized nanomaterial an excellent candidate for use in cosmetics. Below is a schematic representation (Fig. 4) of the synthesis of surface-modified ZnO ultrafine particles using an *in situ* modification method.

Xia and Tang<sup>91</sup> also carried out modification of ZnO using silica. They had a silica surface coated with ZnO either in the

form of a thin layer or as nanoparticles depending on the reaction parameters. ZnO clusters were generated using triethanolamine (TEOH) on a silica surface having silanol ( $\equiv\text{Si}-\text{OH}$ ) and siloxane ( $\equiv\text{Si}-\text{O}-\text{Si}\equiv$ ) groups *via* a procedure of controlled precipitation. TEOH was observed to play a dual role in the synthesis. It first induced surface modification of silica by getting adsorbed on the silica surface *via* a mechanism involving the splitting of the siloxane and silanol networks and then subsequently reacting with  $\text{Zn}(\text{OAc})_2$  to form ZnO clusters whose further growth on the silica surface is sustained *via* ripening and aggregation. The attachment of ZnO on a larger silica surface was argued to enable rapid collision with other ZnO clusters leading to the significant growth of ZnO on the silica surface.

Silica was also used by Hong *et al.*<sup>92</sup> for the modification of ZnO NPs. In a separate experiment they also used oleic acid as the capping agent. A reaction between  $\text{Zn}(\text{OAc})_2 \cdot 2\text{H}_2\text{O}$  and  $(\text{NH}_4)_2\text{CO}_3$  produced the precursor that was then subjected to a procedure culminating in calcination to get ZnO. Oleic acid was then coated on one sample of ZnO while through a separate pH controlled procedure silica was coated on another ZnO sample using sodium silicate. The oleic acid coated ZnO exhibited remarkable compatibility with the organic matrix. The FTIR spectra revealed a chemical bond between the  $-\text{OH}$  groups on the ZnO surface and the  $-\text{COOH}$  groups of the oleic acid chains. The corresponding reaction is given below:



The FTIR spectra for the  $\text{SiO}_2$ -modified ZnO revealed inter-phase bonds between ZnO and  $\text{SiO}_2$ . A thin film coating of  $\text{SiO}_2$  on the ZnO surface resulted in enhanced dispersion and reduced agglomeration of nanoparticles, a fact fairly well corroborated by HR-TEM data. The photocatalytic activity of  $\text{SiO}_2$ -modified ZnO however suffered a setback in comparison with that demonstrated by uncoated ZnO. The work further demonstrated that the thorough reduction of the crystallinity of ZnO achieved through heterogeneous azeotropic distillation of the zinc oxide precursor not only precludes aggregation but also brings about a decline in the average particle size.

Yuan *et al.*<sup>93</sup> modified ZnO using  $\text{Al}_2\text{O}_3$ . A basic carbonate of zinc was obtained from the reaction between zinc sulfate and ammonium bicarbonate followed by precipitating aluminum hydroxide over it. The resulting compound-precipitate was then

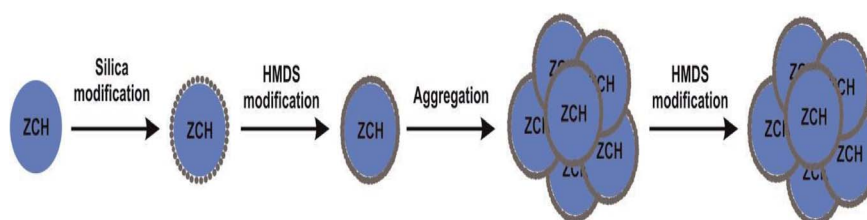


Fig. 4 Schematic representation of the synthesis of surface-modified ZnO ultrafine particles using an *in situ* modification method (reproduced from ref. 89 with permission from Elsevier).



Farzi *et al.*<sup>105</sup> carried out surface modifications of ZnO with oleic acid, trimethoxyvinyl silane and their different mixtures as capping agents. The compatibility of ZnO with different organic matrices was found to increase considerably. A greater degree of dispersion of the inorganic nanoparticle was simultaneously achieved. Modifications with trimethoxyvinyl silane were however found to be more effective than those with oleic acid clearly evident from the greater extent of compatibility with the organic phase achieved using the former. Thus, different combinations of these coupling agents were used to get a grip



Fig. 5 A schematic diagram showing the synthesis of ZnO/polystyrene nanocomposites by anchoring 4,4'-azobis(4-cyanovaleric acid) (ACVA) onto the surface of ZnO nanoparticles to initiate styrene polymerization (reproduced from ref. 97 with permission from Elsevier).

on the hydrophobicity and the dispersion capacity of the inorganic oxide in solvents that are polar to varied extents. Modifications with a 1 : 3 (by weight) mixture of trimethoxyvinyl silane and oleic acid exhibited more efficient dispersion in the highly hydrophobic styrene. Meanwhile, ZnO NPs modified with a 1 : 1 (by weight) mixture of trimethoxyvinyl silane and oleic acid

dispersed more efficiently in the moderately hydrophobic butylacrylate. Modifications attained by a 3 : 1 (by weight) mixture of trimethoxyvinyl silane and oleic acid endowed the nanoparticles with greater dispersion in the weakly hydrophobic methyl methacrylate. The magnitudes of grafting yield





Fig. 6 FESEM of pure ZnO NPs (a) and grafted ZnO/polystyrene nanocomposite particles (b) dispersed in PVC matrices (reproduced from ref. 104 with permission from Elsevier).



Fig. 7 Modification of ZnO nanoparticles with MCT-β-CD (reproduced from ref. 104 with permission from Elsevier).

and grafting density of both the coupling agents were also found to be within an acceptable range.

## 5. Potential applications

Recent years have witnessed a sharp rise in focus on ZnO NPs in the realm of research courtesy of their unique properties that have been exploited to achieve a wide range of applications in transparent electronics, ultraviolet (UV) light emitters, piezo-electric devices, chemical sensors, opto-electronics, solar cells and spin electronics.<sup>106,107</sup> Being nontoxic, ZnO has of late found extensive use as an excellent photocatalyst for the degradation of a great many emerging organic pollutants. ZnO has also attracted great interest as a raw material in the pharmaceutical and cosmetic industries, as a pigment in the paint, concrete and rubber industries, as a UV filter in products, the textile industry, and medical and biological fields, and so on (Fig. 9). A few of these applications of ZnO NPs have been presented below.

### 5.1 Concrete and rubber industries

In the rubber industry, ZnO is extensively used as an additive. ZnO like a host of other metal oxides and inorganic substances raises the thermal conductivity of silicone rubber when added even in low proportions. This it does while retaining the high electrical resistance of the silicone rubber. However, poor dispersion of the nanoparticles in the polymer matrix owing to feeble interactions between the surfaces of nanoparticles and the polymer is one area that requires improvement and therefore has been extensively investigated. Agglomeration of ZnO NPs bearing high surface energy has been reported.<sup>108</sup> This necessitates surface modification of the filler ZnO NPs in order to achieve simultaneous reduction in their agglomeration and an improved degree of dispersion in the polymer matrix.<sup>109</sup> Furthermore, ZnO is widely known to activate sulphur vulcanization *via* enhancement of the degree of cross-linking. Therefore, the mechanical properties of silicone rubber are found to improve as well when ZnO is incorporated into it as a filler.

In their attempt to enhance the interactions between the nano-sized ZnO particles and the polymer, Yuan *et al.*<sup>110</sup> by incorporating vinyl silane groups on the surfaces of ZnO NPs using vinyl triethoxysilane through a procedure premised on the hydrosilylation reaction during curing carried out their





Fig. 8 (A) Photograph of aqueous dispersions of pure ZnO (left) and f-ZnO (right), and (B) FESEM and (C) TEM micrographs of f-ZnO (reproduced from ref. 104 with permission from Elsevier).

surface modification. The vinyl silane groups on the ZnO surface enabled improved cross-linking with the rubber matrix. In order to solve this problem, surface modification techniques are applied to improve the interaction between the nanoparticles prepared by the sol-gel method and the polymer. In comparison with the nanocomposites of silicone rubber with ZnO, the nanocomposites of silicone rubber with vinyl triethoxysilane modified ZnO possessing extensive cross-linking and a higher degree of dispersion with the rubber matrix exhibited superior mechanical properties and enhanced thermal conductivity.

ZnO NPs have been widely used as an efficient material for the enrichment of cross-linking in elastic polymers.<sup>111,112</sup> The cured polymer produced through incorporation of ZnO NPs exhibited high ultimate tensile strength, tear strength, toughness and hysteresis. The slippage of polymer chains on the surfaces of ionic clusters and the renewal of ionic bonds when the sample gets externally deformed give rise to enhanced capacity of the ionic elastic polymer for stress relaxation which in turn results in its upgraded mechanical properties. Furthermore, the thermoplastic properties of such polymers enable their processing in a fused state in a manner akin to a thermoplastic polymer.<sup>113</sup> Nevertheless, carboxylic elastic polymers



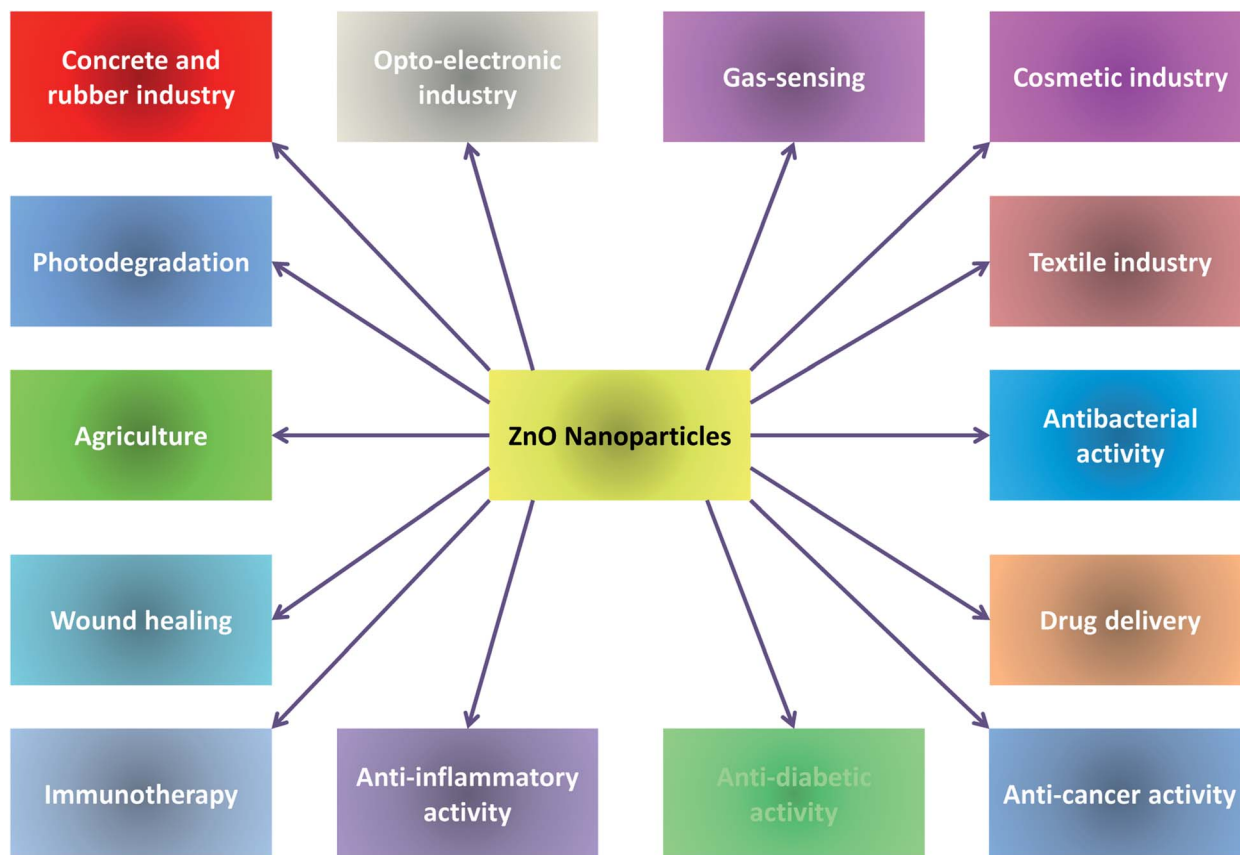


Fig. 9 Applications of ZnO NPs.

with ZnO as a cross-linker suffer from a few drawbacks prominent among which are their tendency to get scorched, feeble flex properties and high value of compression set. The tendency to get scorched is gotten rid of by the incorporation of either zinc peroxide ( $\text{ZnO}_2$ ) or  $\text{ZnO}_2/\text{ZnO}$  cross-linkers.  $\text{ZnO}_2$  serves to not only create ionic cross-links but also generate covalent cross-links as a result of peroxide action. However, prolonged curing is needed to obtain elastomers with an ultimate strength and cross-link density comparable to that of ZnO-cross-linked elastomers. The three vital processes that amount to the curing of XNBR by  $\text{ZnO}_2/\text{ZnO}$  cross-linkers are rapid creation of ionic crosslinks due to the initial ZnO present, covalent links resulting from peroxide cross-links and further ionic cross-linking due to the generation of ZnO from the decomposition of  $\text{ZnO}_2$ . Leaving aside the problem of scorching, ZnO NPs make good and therefore widely used cross-linkers in carboxylated nitrile rubbers.

The prime factors affecting the involvement of ZnO in the formation of ionic cross-links with the carboxylic groups of the elastic polymers are its particle size, surface area and morphology. They are also found to govern the dimensions of the interphase between the cross-linkers and elastomer chains.<sup>114</sup> With a view to ascertain the correlation between the characteristics of ZnO NPs and their roles in the curing of elastic polymers, Przybyszewska *et al.*<sup>115</sup> employed a variety of ZnO NPs with different morphological characteristics (spheres,

whiskers, and snowflakes) as cross-linkers in a carboxylated nitrile elastomer. It emerged from their investigation that ZnO NPs as cross-linkers imparted improved mechanical properties to vulcanizates than commercially used ZnO micro-particles. The ultimate tensile strength of vulcanizates with ZnO NPs was found to be four times higher than that of ZnO micro-particles containing vulcanizates. As a result, there is a 40% reduction of the quantity of ZnO that is put to such use. Since ZnO is known to have deleterious effects on aquatic life, an approach that reduces its usage is highly commendable from the point of view of eco-friendliness. However, ZnO cross-linked XNBR undergoes shrinkage on prolonged exposure to heat.

Among all the aforesaid morphologies, it was observed that ZnO snowflakes with a surface of approximately  $24 \text{ m}^2 \text{ g}^{-1}$  had the highest activity. However, surface area and particle size exerted little influence on the activity of ZnO cross-linkers. It was also observed that the ZnO NPs exhibited a minimum tendency to agglomerate in the rubber matrix. There gathered smaller agglomerates with ZnO NPs as cross-linkers upon sample deformation as compared to the large agglomerates observed with ZnO microparticles.

The usage of ZnO as a cross-linker in rubber has an adverse impact on the environment, particularly when it is discharged into the surroundings upon degradation of rubber.<sup>116</sup> Zinc is known to cause great harm to aquatic species<sup>117</sup> and efforts to cut down on the content of ZnO in rubber are hence being





made.<sup>118</sup> Bringing down the ZnO level in rubber, therefore, may follow any of the following three fundamental procedures:

- (i) substituting the commonly used micro-dimensional ZnO of surface area 4–10 m<sup>2</sup> g<sup>-1</sup> with nano-structured ZnO with surface area of up to 40 m<sup>2</sup> g<sup>-1</sup>;
- (ii) carrying out surface modifications of ZnO with carboxylic acids (*viz.*, stearic acid, maleic acid and the like);
- (iii) using additional activators.<sup>119</sup>

In order to get over the eco-toxicity associated with the usage of ZnO in large quantities, Thomas *et al.*<sup>120</sup> designed a few unique accelerators, namely, *N*-benzylimine amino-thioformamide (BIAT)-capped-stearic acid-coated nano-ZnO (ZOBS), BIAT-capped ZnO (ZOB), and stearic acid-coated nano-zinc phosphate (ZPS), to probe their effects on the curing of natural rubber (NR) and thereby its mechanical properties. ZnO NPs prepared by the sol-gel route were surface-decorated using accelerators such as BIAT and fatty acids such as stearic acid. The capping agents functioned to reduce the size of agglomerates leading to an improvement of vulcanization and physico-chemical properties of NR. Capping of ZnO further ensured a decline in the time and energy required for dispersion in the rubber matrix. As a result, there happened a further enhancement of the acceleration of vulcanization and a remarkable upgrade of the mechanical properties of the emerging vulcanizates. The rubber vulcanized with an optimal dose of BIAT-capped-stearic acid-coated zinc oxide (ZOBS) was found to possess superlative curing and mechanical properties in comparison with other countertypes and the reference polymer containing pristine ZnO NPs. The rigidity of vulcanizates containing ZPS was found to increase as a result of an enhanced cross-link density. The vulcanizates exhibited reduced tendency to get scorched as a result of incorporation of capped ZnO NPs and this was attributed to the delayed release of BIAT from the capped ZnO into the rubber matrix for interaction with CBS (conventional accelerator). Sabura *et al.*<sup>121</sup> adopted a solid-phase pyrolytic procedure to synthesize ZnO NPs of particle size in between 15 and 30 nm and surface area in the range 12–30 m<sup>2</sup> g<sup>-1</sup> for use in neoprene rubber as cross-linkers. Two findings emerged from this study. One, the optimal content of ZnO required was found to be low in comparison with commercially used ZnO. Two, the cure characteristic and mechanical properties of the rubber showed a marked improvement when compared with those containing conventional ZnO.

Since an improvement of the processing time and resistance of concrete against water can be made by incorporation of ZnO NPs in concrete, ZnO NPs have also been used for manufacturing concrete. Nivethitha *et al.*<sup>122</sup> have reported positively about the aforesaid fact in their work. They had the cement partially replaced by ZnO of 0, 1, 3 and 5% by weight of cement. They prepared the blended material by using a cement-sand ratio of 1 : 1 and 1 : 2 by weight while the water-binder ratio was fixed at 0.35. The compressive and split tensile strength was evaluated after 7 and 28 days. Studies were also carried out on durability parameters that include water absorption, impact strength, sorptivity and microstructure of the mortar. The results show that in general an improvement in mechanical and impact strength of the mortar was observed for

ZnO incorporation up to 3% by weight of cement. However, a slight decline of these parameters was observed for 5% by weight of cement incorporation of ZnO. SEM data revealed that the nanoparticles filled cement pores, promoted the hydration process and given their homogeneous dispersion greatly enhanced the microstructure of the cement mortar.

## 5.2 Opto-electronic industry

Because of its large exciton binding energy (60 meV), ZnO is considered as a potentially attractive material for light-emitting devices in the ultraviolet (UV) region. Due to its direct, wide bandgap of 3.37 eV, ZnO can be used as a light-emitting diode and photodetector, in sensors, and in solar cells and devices emitting a surface acoustic wave.<sup>123,124</sup> The optical and electrical properties of ZnO have been greatly exploited in recent years to generate a large photocurrent in ZnO based dye-sensitized solar cells (DSSCs). Since the performance efficiency of solar cells is considered to be substantially influenced by the varying geometries of ZnO nanostructures, a plethora of investigative studies on the effects of morphology of ZnO have unfolded in recent years. With ZnO nanosheets having inadequate surface area, a conversion efficiency of 1.55% was obtained for a DSSC.<sup>125</sup> However, the ZnO nanosheets/spheres synthesized with oxalic acid as surfactant led to an expansion of the surface area thereby enhancing the conversion efficiency of the DSSC to 2.61%.<sup>126</sup> An electrodeposited ZnO nanobelt film was used in a DSSC which was found to exhibit 2.6% photovoltaic efficiency.<sup>127</sup> DSSCs with films comprising ZnO tetrapods had conversion efficiencies in the range 1.20–3.27%.<sup>128,129</sup> Additionally, it came to the fore that incorporation of ZnO NPs into such films increased their surface area which in turn boosted their efficiency.<sup>128</sup> One-dimensional nano-sized structures have been known to reduce recombination by increasing the electron diffusion length *via* direct electron transport. A number of one-dimensional ZnO nanostructures have thus been put to use in DSSCs for enhancement of their efficiency. Notably, a DSSC with ZnO nanotube film exhibited a conversion efficiency of 2.3%.<sup>130</sup> Several dopants have also been tested for their effects on photovoltaic efficiency of ZnO based DSSCs. A dye-sensitized device with ZnO nanoporous nanosheets doped with boron was probed<sup>131</sup> and a conversion efficiency of 6.75% was obtained. A DSSC with fluorine-doped ZnO produced a conversion efficiency of 3.43%.<sup>132</sup> With a lithium-doped ZnO photoanode, the DSSC generated an efficiency of 6.1%.<sup>133</sup>

The last decade has seen an upsurge in the fabrication of ZnO-based perovskite solar cells (PSCs). Although the conventional choice for an electron transport layer has been TiO<sub>2</sub>, ZnO with higher electron mobility is increasingly replacing it as an efficient and low-cost material for electron transport in PSCs. Additionally, the power conversion efficiency of PSCs at large has exceeded 20% of late giving the necessary impetus to delve deep into the fabrication of ZnO electron transport layers (ETLs) for yet more brilliant perovskite solar devices. Bi *et al.*<sup>134</sup> fabricated a PSC with ZnO nanorods aligned vertically over the substrate. With the length of nanorods, the *J*<sub>sc</sub> (short-circuit current density), FF (fill factor) and PCE of solar cells were



found to increase. They however reported a decrease in  $V_{oc}$  with nanorod length. They reasoned that nanorod length has a bearing on the electron transport time and lifetime that in turn influence the performance of the solar cell. They achieved a maximum overall cell efficiency of 5%. Son *et al.*<sup>135</sup> substituted the single-step method used by Bi *et al.* by a two-step coating procedure. Such a treatment generated a fully filled perovskite film that covered all ZnO nanorods of varying lengths without voids and formed an overlayer on the surface of nanorods. As a further consideration, the two-step coating treatment induced optimization of the cuboid size of MAPbI<sub>3</sub> and reduced the series resistance of the solar cell.<sup>136</sup> As a result, a maximum PCE of 11.13% was obtained. Tang *et al.* designed ZnO nanowall ETLs.<sup>137</sup> The best performance PSC based on ZnO nanowalls produced a  $J_{sc}$  of 18.9 mA cm<sup>-2</sup>,  $V_{oc}$  of 1.0 V, FF of 72.1%, and PCE of 13.6%. Meanwhile, the control device shows a  $J_{sc}$  of 18.6 mA cm<sup>-2</sup>,  $V_{oc}$  of 0.98 V, FF of 62%, and PCE of 11.3%. The introduction of ZnO nanowalls led to an evident boost in the FF and PCE of the PSCs and this can be ascribed to the greater contact area between ZnO and perovskite offered by the ZnO nanowalls in comparison with the planar ZnO film which improves not only the electron collection but also transportation efficiency at the interface of the ZnO nanowalls and perovskite. Moreover, the decomposition of ZnO by perovskite triggered by the alkaline nature of the ZnO surface leads to the formation of PbI<sub>2</sub> on the perovskite/ZnO interface. The presence of PbI<sub>2</sub> can suppress the surface recombination and improve the FF.<sup>138</sup>

### 5.3 Gas-sensing

ZnO nanostructures have exhibited excellent gas sensing properties largely driven by the change in modifications in sensor resistance when gas molecules interact with their surface. The overriding feature of ZnO nanostructures when it comes to gas sensing ability is the large surface-to-volume ratio. Under ambient conditions, oxygen molecules get adsorbed on the surface through a phenomenon termed as ionosorption. In the course of this phenomenon, they collect electrons from the conduction band of ZnO and assume various forms ( $O^-$ ,  $O^{2-}$  and  $O_2^-$ ) depending on the temperature.<sup>139,140</sup> This creates a surface depletion layer that leads to an increase of resistance in ZnO. Upon approach of reductive gases in the vicinity of the ZnO surface, the adsorbed negatively charged oxygen species interact with the gas molecules and the trapped electrons are released back to the conduction band of ZnO thereby lowering the sensor resistance. On the other hand, oxidative gases undergo interaction that raises resistance in the material.

Hydrothermally obtained ZnO nanorods reported in a study by Cho *et al.*<sup>141</sup> exhibited remarkable sensitivity of NO<sub>2</sub> gas in the presence of CO. The sensor resistance elicited a change by 1.8 times its pristine value upon exposure to 1 ppm NO<sub>2</sub> while an imperceptible change was noticed for a 50 ppm CO. It also emerged from the study that ZnO nanorods with larger crystallite size turned out to be more sensitive than ones with smaller crystallite diameters. Sensitivity towards NO<sub>2</sub> was found to be more pronounced under conditions of low temperature.

Vanalakar *et al.*<sup>142</sup> also prepared ZnO nanorods with preferential orientation along the (002) direction through the wet chemical route and studied their gas sensing ability towards NO<sub>2</sub> and the parameters affecting it, such as the aspect ratio of the rod lengths, time, operating temperature and gas concentrations. The sample of ZnO nanorods with a (103)/(002) ratio of 0.036 and intermediate spacing between them produced remarkable sensing performance at a temperature of 175 °C for detection of NO<sub>2</sub> at a concentration of 100 ppm with a high sensitivity of 3100%. The mechanism of gas sensing outlined in the work is as follows:



Zhang *et al.*<sup>143</sup> constructed a Ag loaded ZnO–MoS<sub>2</sub> gas sensing film through a LBL (layer-by-layer) self-assembly route and investigated its response upon exposure to a wide range of gases such as H<sub>2</sub>, CH<sub>4</sub>, CO, C<sub>2</sub>H<sub>2</sub>, C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub>. It was found that the nanocomposite film showed remarkable selectivity towards CO (Fig. 10). Upon contact with CO gas, the Ag NPs of the nanocomposite sensing film catalyze the oxidation of CO triggering the release of electrons trapped by anionic oxygen species which in turn causes a sharp fall of sensor resistance to manifest a brilliant sensing response. The Ag–ZnO–MoS<sub>2</sub> nanocomposite's gas sensing performance towards CO was found to be higher than that of Pt–ZnO–MoS<sub>2</sub> or ZnO–MoS<sub>2</sub> or ZnO film in terms of response/recovery capability, repeatability, stability and selectivity (Fig. 11). Notably, under the same conditions of the experiment, the Ag–ZnO–MoS<sub>2</sub> sensing film functioned way better than the Pt–ZnO–MoS<sub>2</sub> sensor. This was elucidated invoking the Schottky barrier (SB). Fig. 12a and b show how a noble metal's SB would tamper with the electrical properties of the ZnO–MoS<sub>2</sub> film. The equation  $\Phi_{SB} = \Phi - \psi$  allows the calculation of the SB height  $\Phi_{SB}$ , where  $\Phi$  is the work function of the noble metal and  $\psi$  is the electron affinity

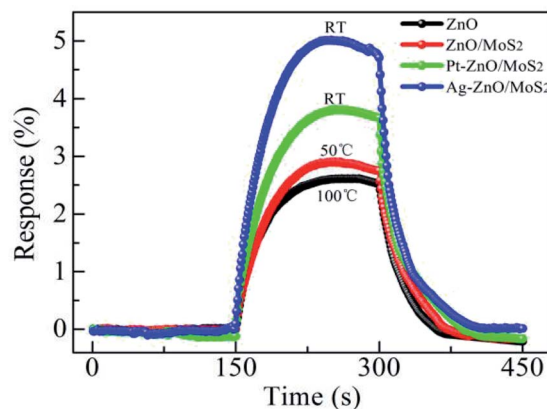


Fig. 10 Sensing responses of pure ZnO, ZnO–MoS<sub>2</sub>, Pt–ZnO–MoS<sub>2</sub> and Ag–ZnO–MoS<sub>2</sub> film sensors towards 100 ppm CO gas (reproduced from ref. 143 with permission from Elsevier).



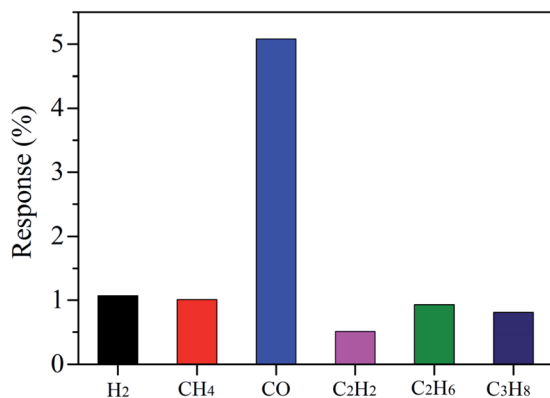


Fig. 11 Selectivity of the Ag-ZnO-MoS<sub>2</sub> nanocomposite sensor towards 100 ppm gas species of H<sub>2</sub>, CH<sub>4</sub>, CO, C<sub>2</sub>H<sub>2</sub>, C<sub>2</sub>H<sub>6</sub> and C<sub>3</sub>H<sub>8</sub> (reproduced from ref. 143 with permission from Elsevier).

of ZnO-MoS<sub>2</sub>. Metals with a lowered work function  $\Phi$  would therefore produce a SB of smaller magnitude. Hence, a metal with a small work function is desirable for a reduced SB and therefore an enhanced sensitivity of the resultant nanocomposite film. Since Ag has a lower work function (4.26 eV) than that of Pt (5.65 eV), the Ag loaded ZnO-MoS<sub>2</sub> gas sensing nanocomposite film elicited greatly elevated sensing responses towards CO. Using an improvised sol-gel procedure, Hjiri *et al.*<sup>144</sup> prepared an Al-doped ZnO gas sensor showing excellent selectivity towards CO. Samples coded A0ZO, A1ZO, A3ZO and A5ZO had [Al/Zn] ratios of 0, 0.01, 0.03 and 0.05, respectively. The response behaviour of all these samples towards 50 ppm CO at various operating temperatures was monitored. The response ( $R_0/R$ ) of all samples initially increased with temperature, reached a maximum and then exhibited a decline (Fig. 13). At lower temperatures of 200 to 300 °C, only A1ZO showed a better response than the other samples that responded almost equally. However, in the temperature range 300–400 °C, all Al-loaded samples showed a greater response than the unloaded ZnO. With different

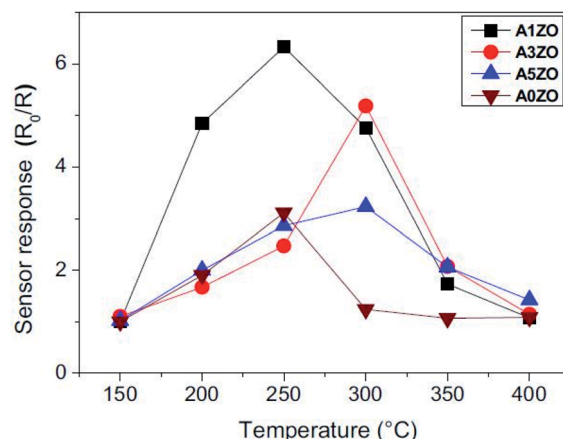


Fig. 13 Response of the Al-loaded and unloaded ZnO samples towards 50 ppm CO at different operating temperatures (reproduced from ref. 144 with permission from Elsevier).

concentrations of CO at 300 °C (Fig. 14), the A3ZO sensor showed the best response. Thus, impregnating ZnO with a low content of Al in the range of 1–3 wt% resulted in an enhanced response towards CO sensing. Besides, a significant response could be observed at a very low CO concentration of 5 ppm. Also, since exposure of the Al-loaded ZnO samples to CO produced a decrease in the resistance of the samples, it was concluded that Al-doping sustained the n-type semi-conducting properties of the pristine ZnO. Exposure to CO caused a great variation in the resistance of the sensing film (Fig. 15). The signal was further found to go back to its initial baseline value after each pulse indicating that the CO adsorption on the surface layer was reversible. The following mechanism was furnished to explain the decrease of resistance of Al-loaded ZnO upon exposure to CO:

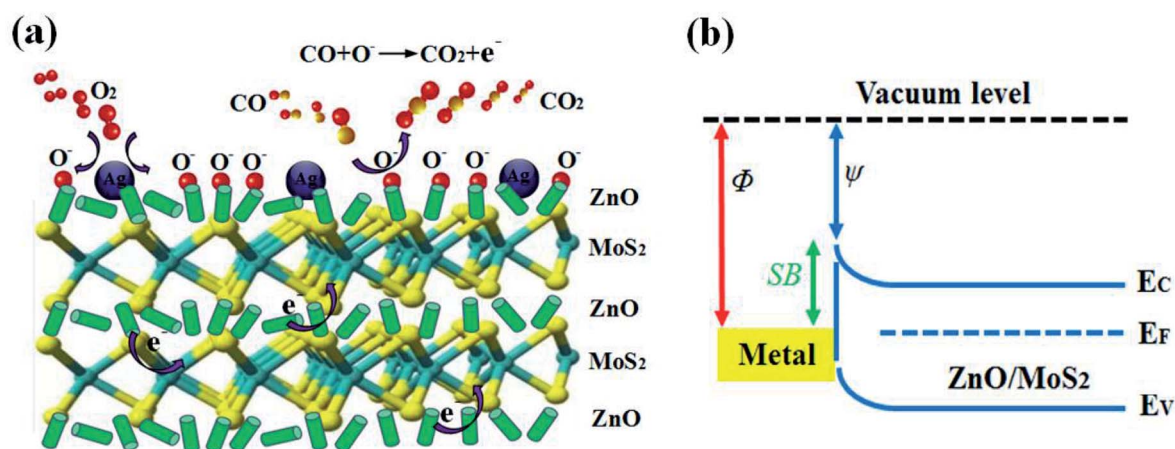
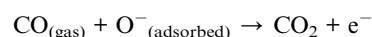


Fig. 12 (a) Schematic diagram of the Ag-ZnO-MoS<sub>2</sub> nanocomposite sensor towards CO gas; (b) energy band diagram of the Ag-ZnO-MoS<sub>2</sub> nanocomposite (reproduced from ref. 143 with permission from Elsevier).





Fig. 14 Response of AZO nanoparticles as a function of CO concentration at a temperature of 300 °C (reproduced from ref. 144 with permission from Elsevier).

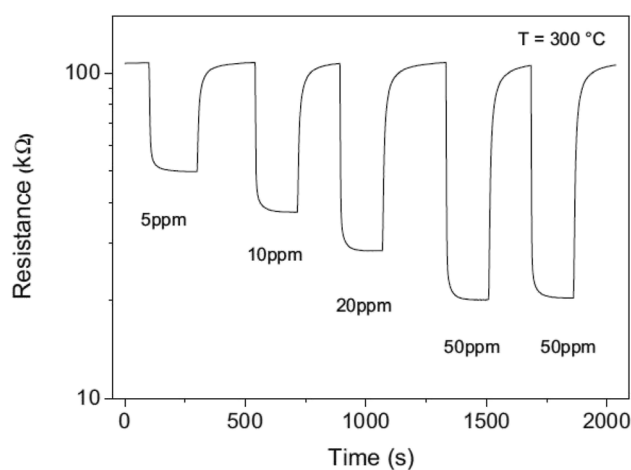


Fig. 15 Response of the A3ZO sensor as a function of CO concentration at 300 °C (reproduced from ref. 144 with permission from Elsevier).

The electrons injected into the conduction band lower the resistance of the Al-doped ZnO gas sensors. The response and recovery times observed for all Al-loaded ZnO samples were 6–8 s and 16–30 s, respectively. The unloaded ZnO sample was marked by longer response and recovery times of 30 s and 70 s, respectively. The sensing films exhibited excellent thermo-mechanical and electrical stability.

Chen *et al.*<sup>145</sup> synthesized mesoporous ZnO sensors for selectively detecting ethanol under UV light at operating temperatures lower than 150 °C. With the assistance of a low-powered UV LED (2 mW), outstanding response and selectivity towards varying concentrations (10 to 1000 ppm) of ethanol were demonstrated by the sensing layer at 80 °C. UV illumination induced electron-hole pairs that activated adsorbed oxygen leading to their extensive interaction with ethanol thereby eliciting a remarkable sensing response from the synthesized ZnO sensors. The response and recovery times were

6 s and 94 s, respectively, and the gas sensor displayed repeatability and brilliant stability. Another study<sup>146</sup> reported a facile three-stage synthesis of ZnO–SnO<sub>2</sub> composite hollow spheres for the selective sensing of ethanol by exploiting the synergic benefits of the n–n heterojunction. The basic properties of ZnO played a pivotal role in the sensing mechanism *via* dehydrogenation of ethanol as elaborated in the work.



The CH<sub>3</sub>CHO intermediate is subsequently oxidized to form CO<sub>2</sub> and H<sub>2</sub>O, as depicted in the following equation:



Therefore, the ZnO/SnO<sub>2</sub> nanocomposite gas sensor demonstrated a sharper response to ethanol gas than the pristine SnO<sub>2</sub> sensor. Moreover, a possible increase in the effective barrier height of the n–n heterojunction enabled better engagement with adsorbed oxygen causing greater depletion of electrons from the conduction band eventually leading to an enhanced gas sensing response by the system. Additionally, remarkable detection at a lower (ppb) limit was shown by the heterostructured sensor.

#### 5.4 Cosmetic industry

Due to its antibacterial and good antifungal activity, zinc oxide is widely used in production of various raw materials used in medicine such as disinfectant agents and for dermatological applications.<sup>147</sup> ZnO NPs absorb UVA and UVB radiations and can be used in sun protective creams.<sup>148</sup>

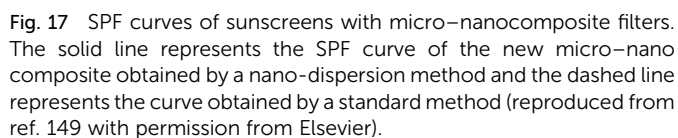
In a study by Reinosa *et al.*,<sup>149</sup> it was brought to light that a nano/micro-composite comprising nanosized TiO<sub>2</sub> dispersed on ZnO micro-particles showed a higher sun protection factor (SPF) than individual TiO<sub>2</sub> and ZnO particles. The SPF of the synthesized nano-sized TiO<sub>2</sub> was found to be higher than that of its micro-sized counterpart with the former showing maximum absorption at 319 nm while the latter showed maximum absorption at 360 nm. The synthesized micro-sized ZnO had a higher SPF than its nano equivalent. Both exhibited maximum absorption at 368 nm. These data suggested that ZnO has a higher critical wavelength because it covers the entire UV range and has a higher UVA/UVB ratio since the maximum of the SPF curve lies in the UVA region (Fig. 16a). Additionally, it was observed that TiO<sub>2</sub>, with a lower UVA/UVB ratio owing to the presence of the SPF maximum in the lower wavelength region, has a lower critical wavelength (Fig. 16b). Therefore, to boost the SPF output, a suitable combination of the two oxides was thought out. A dry dispersion procedure was adopted to prepare the composite consisting of 15 wt% TiO<sub>2</sub> NPs and 85 wt% ZnO micro-structured particles. The results obtained from this composite were compared with those obtained by the standard procedure. Raman spectroscopy revealed a superior dispersion of the NPs and their anchoring with higher quantum confinement resulting from dry dispersion by using ZnO micro-





Water-repellent and self-cleaning fibers come handy in the military world where laundering is difficult and inconvenient with regard to both time and scope. On the other hand, in the business sectors, water-repellant and self-cleaning fabrics become useful in preventing undesirable stains. Furthermore, blocking the UV portion of sunlight is another aspect that has long concerned researchers. All these facts eventually underline the massive potential offered by the textile industry for the

Zinc oxide nanowires were grown on cotton fabric by Ates *et al.*<sup>163</sup> to impart self-cleaning, superhydrophobicity and ultraviolet (UV) blocking properties. The ZnO nanowires were grown by a microwave-assisted hydrothermal method and subsequently functionalized with stearic acid to obtain a water contact angle of 150°, demonstrating their superhydrophobic nature, which is found to be stable for up to four washings. The UV protection offered by the resulting cotton fabric was also examined, and a significant decrease in transmission of radiation in the UV range was observed. The self-cleaning activity of



Jiang *et al.*<sup>183</sup> studied the potential antibacterial mechanisms of ZnO NPs against *E. coli*. They reported that ZnO NPs with an average size of about 30 nm caused cell death by coming into direct contact with the phospholipid bilayer of the membrane and destroying the membrane integrity. The significant role of ROS production in the antibacterial properties of ZnO NPs surfaced when it emerged that the addition of radical scavengers such as mannitol, vitamin E, and glutathione could block the bactericidal action of ZnO NPs. However, the antibacterial effect triggered by Zn<sup>2+</sup> released from ZnO NP suspensions was not apparent. Reddy synthesized ZnO NPs with sizes of ~13 nm and investigated their antibacterial (*E. coli* and *S. aureus*) activities.<sup>168</sup> It was discovered that ZnO NPs effected complete cessation of the growth of *E. coli* at concentrations of about 3.4 mM but induced growth inhibition of *S. aureus* at much lower concentrations ( $\geq 1$  mM). Besides, Ohira and Yamamoto<sup>184</sup> also discovered that the antibacterial (*E. coli* and *S.*

Epidemic disease cholera mainly affects populations in developing countries.<sup>169,180</sup> It is a serious diarrheal disease caused by the intestinal infection of Gram-negative bacterium *V. cholerae*. The effective antibacterial activity of ZnO NPs and their mechanism of toxicity were explored against *Vibrio*

*aureus*) activity of ZnO NPs with small crystallite sizes was far more pronounced than for those with large crystallite sizes. From ICP-AES measurement, it emerged that the amount of  $\text{Zn}^{2+}$  released from the small ZnO NPs was much higher than from the large ZnO powder sample and *E. coli* was more sensitive to  $\text{Zn}^{2+}$  than *S. aureus*. This is a further confirmation that eluted  $\text{Zn}^{2+}$  ions from ZnO NPs also play a key role in antibacterial action.

Iswarya *et al.*,<sup>185</sup> having extracted crustacean immune molecule  $\beta$ -1,3-glucan binding protein (Ph $\beta$ -GBP) from the haemolymph of *Paratelphusa hydrodromus*, successfully designed Ph $\beta$ -GBP-coated ZnO NPs. The Ph $\beta$ -GBP-ZnO NPs were spherical shaped having a particle size of 20–50 nm and halted the growth of *S. aureus* and *P. vulgaris*. *S. aureus* was found to be more prone to the bactericidal action of Ph $\beta$ -GBP-ZnO NPs than *P. vulgaris*. In addition, Ph $\beta$ -GBP-ZnO NPs could induce drastic modification in cell membrane permeability and set off outrageous levels of ROS formation both in *S. aureus* and *P. vulgaris*. This work was thus pivotal in bringing to the forefront the immensely great antibacterial hallmark of Ph $\beta$ -GBP-ZnO NPs.

The mechanism of breaking into bacterial cells by membrane disruption and then inducing oxidative stress in bacterial cells, thereby stalling cell growth and eventually causing cell death has been reported in many recent research studies.<sup>186–191</sup> Important bacterial biomolecules can also adsorb on ZnO NPs. Bacterial toxicity, in the recent past, has been heavily reported to have resulted from structural changes in proteins and molecular damage to phospholipids.<sup>192</sup> The antibacterial activity of ZnO NPs thus finds its apt application in the discipline of food preservation. As a formidable sanitizing agent, it can be used for disinfecting and sterilizing food industry equipment and containers against attack and contamination by food-borne pathogenic bacteria. ZnO NPs showed both toxicity on pathogenic bacteria (*e.g.*, *Escherichia*

*coli* and *Staphylococcus aureus*) and beneficial effects on microbes, such as *Pseudomonas putida*, which has bioremediation potential and is a strong root colonizer.<sup>193</sup>

Investigations into the antibacterial activities of ZnO micro-sized particles, ZnO NPs, and ZnO NPs capped with oxalic acid against *S. aureus* were carried out in the presence and absence of light.<sup>194</sup> It was observed that the efficiency of ZnO NPs was just 17% in the dark. However, their antibacterial properties saw a surge up to 80% upon application of light. The antibacterial behaviour was greatest for ZnO NPs while it was minimum for ZnO micro-sized particles, suggesting a higher release of  $\text{Zn}^{2+}$  ions from ZnO NPs than ZnO micro-sized particles. The examination revealed that surface defects of the ZnO NPs boosted ROS production in the presence as well as absence of light. Additionally, it was also found that capping lowers the amount of superoxide radicals generated because capping blocks the oxygen vacancies that are chiefly accountable for the generation of superoxide radicals. In another investigation, the influence of NP size on bacterial growth inhibition by ZnO NPs and the mechanistic routes of their action were demonstrated.<sup>195</sup> ZnO NPs with diameters ranging from 12 nm to 307 nm were first generated. Thereafter, they were administered to Gram-positive and Gram-negative microorganisms (Fig. 18). The results clearly illustrated the greater bactericidal efficacy of smaller ZnO NPs under dark conditions. The use of UV light resulted in an enhanced antibacterial behaviour of ZnO NPs owing to the enhanced formation of ROS from them. The antibacterial properties were rooted in the generation of ROS and the build-up of ZnO nano-sized particles in the cytoplasm and on the external membranes.

In another intriguing investigation, the toxicity induced in antibiotic resistant nosocomial pathogens such as *Acinetobacter baumannii* (*A. baumannii*) and *Klebsiella pneumoniae* (*K. pneumoniae*) by photocatalytic ZnO NPs was studied.<sup>196</sup> It was seen

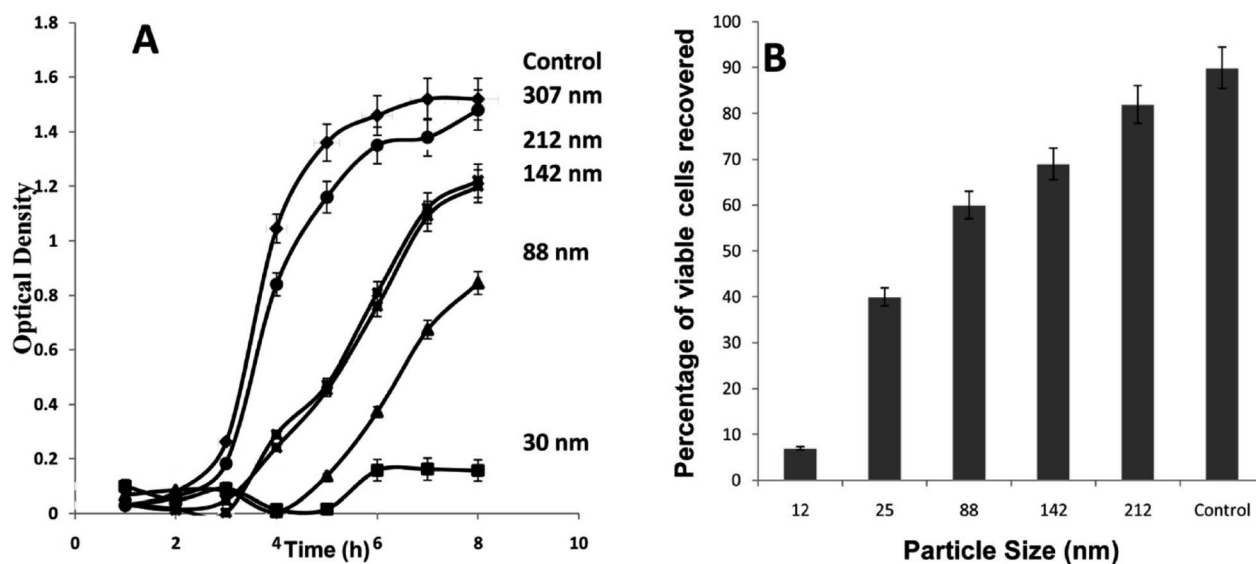


Fig. 18 The influence of different sizes of ZnO NPs on the growth of a methicillin sensitive *S. aureus* strain. (A) Growth analysis curves obtained by tracking the optical density at 600 nm. (B) Percentage of viable *S. aureus* recovered after treatment with ZnO NPs of different sizes (reproduced from ref. 195 with permission from the American Chemical Society).





of loading drugs onto the ZnO NPs directly.<sup>211</sup> Upon contact with acids, the drug molecules are released following the complete decomposition of ZnO NPs. In another investigation, a liposome-incorporated ZnO-NP based DDS (ZNPs-liposome-DNR) enclosing anticancer drug daunorubicin (DNR) was designed by Tripathy *et al.*<sup>212</sup> The incorporation of ZnO-NPs in the DDS was observed to prevent the premature release of DNR, which could be prompted only in acidic medium, thereby efficiently exerting an anticancer effect on A549 cells. The study of intracellular release in cancer cells with confocal laser scanning microscopy (CLSM) revealed that treatment with ZNPs-liposome-DNR induced a marked DNR release, causing greater cytotoxicity to cancer cells, compared to pure DNR and DNR-conjugated liposomes (liposome-DNR), as evidenced by the green fluorescence intensity. For the treatment of lung cancer, Cai *et al.* accomplished the construction of ZnO quantum dot-based drug delivery apparatus that was conjugated with a targeting agent (hyaluronic acid) and an anticancer agent (DOX).<sup>213</sup> The nano-apparatus demonstrated CD44 receptor-specific uptake and pH-driven drug release in lysosomal compartments of the cancer cells. Kumar *et al.*<sup>214</sup> also designed sub-micron sized self-assembled spherical capsules of

Although ZnO NPs make a promising antibacterial agent owing to their wide-ranging activities against Gram-positive as well as Gram-negative bacteria, the exact antibacterial pathway of ZnO NPs has not been adequately established. Hence, deep investigations into it hold a lot of important theoretical and practical value. In the future, ZnO NPs can be explored as antibacterial agents, such as ointments, lotions, and mouthwashes. Additionally, they can be overlaid on various substrates to prevent bacteria from adhering, spreading, and breeding in medical devices.

Nanotechnology has enabled the exploration of better and more efficient avenues concerning drug delivery vehicles. Various research studies suggest the potency of ZnO NPs for drug delivery in a number of diseases. ZnO NPs encapsulated with chitosan, bearing enhanced stability, were found to be effective for importing doxorubicin (DOX) into cancer cells in a recent study.<sup>195</sup> In various other studies, ZnO NPs were successfully used as vehicles for gene delivery and gene silencing (Fig. 17 and 18).<sup>198–200</sup> The use of this apparatus for gene delivery had advantages. For example, the expression of plasmid containing genes on the surface of nanoparticles could ensure safe and efficient gene targeting to the recipient tissues<sup>201,202</sup> enabling the nanoparticles to be used as suitable tools for targeting genes to various cells such as tumor cells. Nie *et al.*<sup>199</sup> in their study discovered that synthesized ZnO tetrapod-like nanostructures could be used as pioneering vehicles for gene delivery. They showed that silica-coated amino-modified tetrapod-like ZnO nanostructures could, *via* electrostatic interactions, effectively bind to DNA which in turn improved the transfection efficiency of melanoma cells. In another study conducted by Zhang *et al.*, it was unveiled that polycation-capped ZnO quantum dots had the potential to deliver DNA into COS-7 cells.<sup>203</sup> Further, the



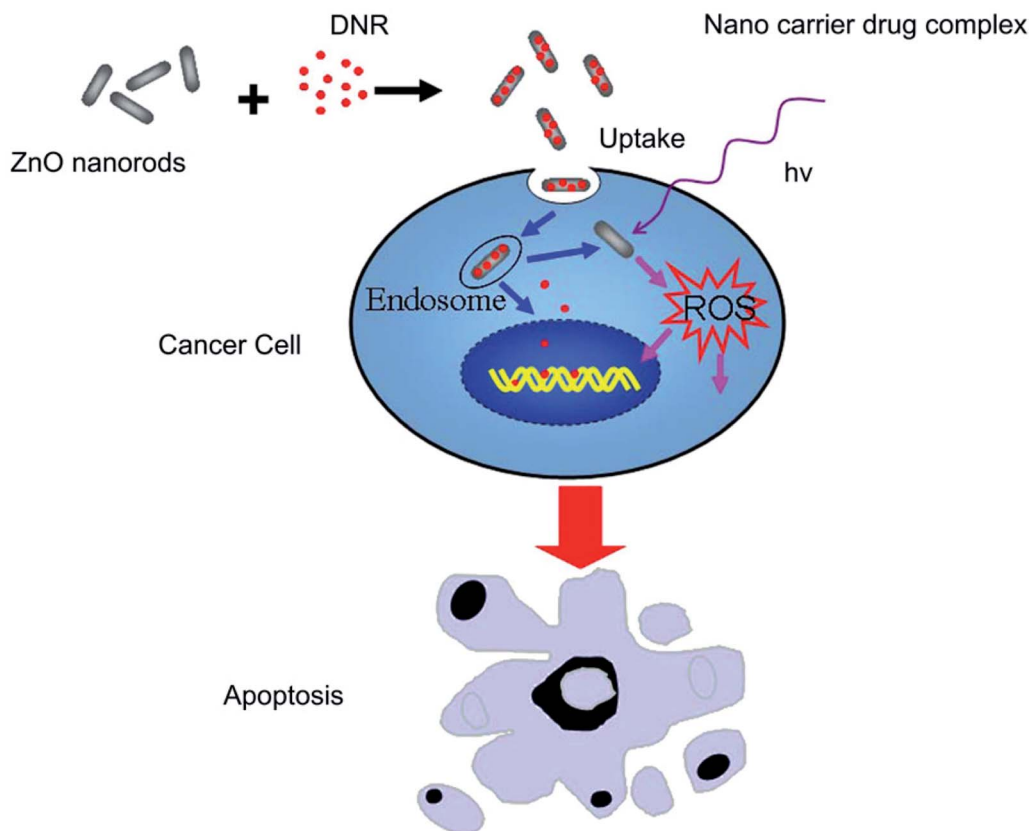


Fig. 19 Schematic illustration of possible processes of ZnO nanorods encapsulating chemotherapeutic agents for anticancer therapy (reproduced from ref. 216 with permission from Elsevier).

ZnO nanorods that successfully effected the delivery of anti-cancer agent DOX to K562 cancer cells. Furthermore, Han *et al.*<sup>215</sup> also synthesized ZnO NPs conjugated with an aptamer as a functionalization agent and DOX as an anticancer agent and demonstrated the effect of combined chemo- and radiation therapy in MCF-7 breast cancer cells employing the nano-apparatus. Recently, Zhang *et al.*<sup>216</sup> as a part of their investigation devised a new scheme to restrain the proliferation of human hepatocarcinoma cells (SMMC-7721) *via* a combined application of ZnO nanorod based DNR in photodynamic therapy (PDT), where ROS generation had the possibility to play a key role in the net anticancer behaviour of the hybrid nano-apparatus (Fig. 19). The researchers further discovered that ZnO NPs were able to transport a larger quantity of DNR *via* internalization into SMMC-7721 cells, thereby inducing outstanding restraint on the multiplication of these cancerous cells. Besides, UV irradiation on this drug delivery nano-apparatus further reinforced the arrest of cell proliferation through photocatalysis of ZnO nanorods. To look into the signaling pathway of anticancer activity of the DDS in PDT, the researchers monitored the caspase-3 activity, which is a hallmark of apoptosis. The results of immunocytochemistry study confirmed that upon treatment with a DNR–ZnO nano-composite under UV irradiation, the cells demonstrated far more pronounced activation of caspase-3 molecules in cancer-afflicted cells. It was consequently proposed that ZnO

nanorods could raise the drug's targeting efficiency and minimize the associated toxicity. Therefore, the DNR–ZnO hybrid nano-apparatus with UV irradiation was claimed to have the potential of a fruitful scheme for the treatment of cancers (Fig. 20 and 21). In another study, Ye *et al.*<sup>217</sup> using a copolymerization process also prepared water soluble ZnO–polymer core–shell quantum dots, and designed a drug delivery apparatus based on these quantum dots containing  $Gd^{3+}$  ions and anticancer drug DOX. The ZnO–Gd–DOX nano-system was found to be biocompatible, pH-responsive and led to a marked release of DOX into the acidic environment of cancer-afflicted cells and tumors. When administered to human pancreatic cancer (BxPC-



Fig. 20 Possible mechanism of ROS production by ZnO nanorods under UV irradiation (reproduced from ref. 216 with permission from Elsevier).





Fig. 21 Cytotoxicity of DNR or the DNR–ZnO nanocomposite in the absence or presence of UV irradiation against SMMC-7721 cells. The inset graph shows the  $IC_{50}$  of DNR and the DNR–ZnO nanocomposite in the absence or presence of UV irradiation for SMMC-7721 cells (reproduced from ref. 216 with permission from Elsevier).

3) tumor containing nude mice, this polymer-modified drug delivery nano-apparatus was found to display higher therapeutic efficiency compared to the FDA-approved liposomal DOX formulation DOXIL at  $2 \text{ mg kg}^{-1}$  DOX concentration. The histopathology study and ICP-AES analysis of the vital organs further confirmed that this ZnO-Gd-DOX nano-apparatus could substantially bring about growth-inhibition of tumors without exerting any toxic effects 36 days post administration. Additionally, the histopathology study of tumor sections also demonstrated severe damage to the tumor cells caused by the administration of the DDS, compared to the control, DOX and DOXIL groups.

### 5.8 Anti-cancer activity

Cancer, a condition of uncontrolled malignant cell proliferation, has typically been subjected to treatment by chemotherapy, radiotherapy, and surgery during the past several decades. Although in theory all these therapies appear to be exquisitely effective in killing cancer cells, these non-specific therapy methods also introduce a lot of serious side effects.<sup>218</sup> Recently, nanomaterial-based nanomedicine possessing high biocompatibility, easy surface functionalization, cancer targeting, and drug delivery capacity has displayed the much coveted flair to rise above these side effects. Substantial research progress has been made in biocompatible nanosized zinc oxide (ZnO) because of its unique physical properties. ZnO nanomaterials, inspired by the fact that  $Zn^{2+}$  is an essential nutrient for adults, are considered to be safe *in vivo*. Considering these leverages, ZnO NPs can be aptly chosen as biocompatible and biodegradable nano-platforms and can also be investigated for cancer treatment.<sup>219,220</sup>

Several studies have thus suggested the cytotoxic effects of ZnO NPs on cancer cells. The cancer cell viability percentage on the MCF7 cell line, A549 cell line, HL60 cell line and VERO cell line has been studied at various concentrations of ZnO. Results show that the cell viability of the above cell lines exhibits

a marked decrease with a rise in ZnO concentration<sup>221,222</sup> with minimal damage to healthy cells.

The mitochondrial electron transport chain is known to be closely linked to intracellular ROS generation, and anticancer agents accessing cancer cells could impair the electron transport chain and release huge amounts of ROS.<sup>223,224</sup> However, an inordinate amount of ROS brings about mitochondrial damage thereby resulting in the loss of protein activity balance that eventually induces cell apoptosis.<sup>225</sup> ZnO NPs introduce certain cytotoxicity in cancer cells chiefly by a mechanism that involves a higher intracellular release of dissolved  $Zn^{2+}$  ions, followed by enhanced ROS induction and induced cancer cell death by way of the apoptosis signaling pathway. The effects of ZnO NPs on human liver cancer HepG2 cells and their possible pharmacological mechanism were investigated by Sharma *et al.*<sup>226</sup> They observed that ZnO NP-exposed HepG2 cells exhibited higher cytotoxicity and genotoxicity, which were related to cell apoptosis conciliated by the ROS triggered mitochondrial route. The loss of the mitochondrial membrane potential led to the opening of outer membrane pores following which some related apoptotic proteins including cytochrome c were released into the cytosol thereby activating the caspase in due course. Mechanistic studies had proved that the loss of mitochondrial membrane potential-mediated HepG2 cell apoptosis was mainly due to the decrease in mitochondrial membrane potential and Bcl-2/Bax ratios as well as accompanying the activation of caspase-9. Besides, ZnO NPs could noticeably activate p38 and JNK and induce and attract p53<sup>ser15</sup> phosphorylation but this was not dependent on JNK and p38 pathways (Fig. 21). These results afforded valuable insights into the mechanism of ZnO NP-induced apoptosis in human liver HepG2 cells. Moghaddam *et al.*<sup>227</sup> took recourse to biogenic synthesis and successfully generated ZnO NPs using a new strain of yeast (*Pichia kudriavzevii* GY1) and examined their anticancer activity in breast cancer MCF-7 cells. ZnO NPs have been observed to exhibit powerful cytotoxicity against MCF-7



Photodynamic therapy requires the administration of a photosensitizing agent that is subjected to activation by light of a specific wavelength thereby generating ROS. The application of ZnO NPs as effective photosensitizers can be ascribed to their capability to generate ROS in response to visible light or UV light. Recent studies exhibited that photo-triggered toxicity of ZnO NPs renders them aptly suitable for targeted PDT in a spatiotemporal manner, providing a surer way to selectively

To secure concomitant intracellular drug delivery and PDT for cancer treatment, poly(ethylene glycol) (PEG)-capped ZnO NPs enclosing DOX were fabricated.<sup>233</sup> It was found that DOX-loaded PEG-ZnO NPs on exposure to UV irradiation achieved



Likewise, poly(vinylpyrrolidone) (PVP)-capped ZnO nanorods (PVP-ZnO nanorods) were designed as a drug carrying nano-apparatus for the delivery of daunorubicin (DNR), as well as a photosensitizer for PDT.<sup>234</sup> The DNR-loaded PVP-ZnO nanorods (DNR-PVP-ZnO) encouraged an exceptional upswing in the anticancer activity of DNR due to elevated cellular uptake of the DNR delivered by the nanorods. The DNR-PVP-ZnO nanorods also demonstrated efficient PDT under UV light irradiation. It has been demonstrated that NPs can furnish solutions to confront the acute demerits of conventional photosensitizers.<sup>235</sup> By a dramatic enhancement of the solubility of photosensitizers, NPs can facilitate their increased cellular internalization. They also upgrade the target-specificity of photosensitizers by way of passive targeting to tumor tissues through the enhanced permeability and retention (EPR) effect. Further, cell-specificity of photosensitizers can be remarkably increased by surface modification of the NPs to bind active targeting components. Complexation of ZnO NPs with other photosensitizers has been widely researched to increase the efficacy of ZnO NPs in PDT by synergistically enhancing the ROS generation.<sup>234,235</sup> *Meso-*

tetra(*o*-aminophenyl)porphyrin (MTAP)-conjugated ZnO nanocomposites were fabricated and examined for synergistic PDT against ovarian cancer cells.<sup>236</sup> The MTAP-ZnO NPs induced generation of ROS upon UV irradiation, the controlling parameters being concentration and light intensity. It emerged that 30  $\mu$ M MTAP-ZnO NPs wielded high light-induced toxic effects in cancer-afflicted ovarian cells under UV illumination, while they remained inactive in the dark. The cytotoxic activity of MTAP-ZnO NPs under UV illumination was markedly boosted weighed against that of porphyrin alone.<sup>235</sup> This study elucidated the targeted and synergistic PDT by nanoparticles of ZnO loaded with photosensitizing substances. ZnO NPs were combined with protoporphyrin IX (PpIX) as a drug delivery nano-apparatus for photosensitizers.<sup>237</sup> Simple ZnO NPs and PEG-capped ZnO NPs were synthesized and examined for their cancer-eliminating effect against human muscle carcinoma cells. In the absence of laser light, ZnO NPs at 1 mM concentration were found to exert very low cytotoxicity (98% viability). In the presence of 630 nm laser light, PEG-capped ZnO NPs loaded with PpIX exhibited outstanding cytotoxicity owing to the increased ROS generation. Additionally, a high build-up of PpIX in the tumor area was observed when it was delivered by ZnO NPs, exhibiting the potency of ZnO NPs as a tumor-selective drug delivery system for photosensitizers.

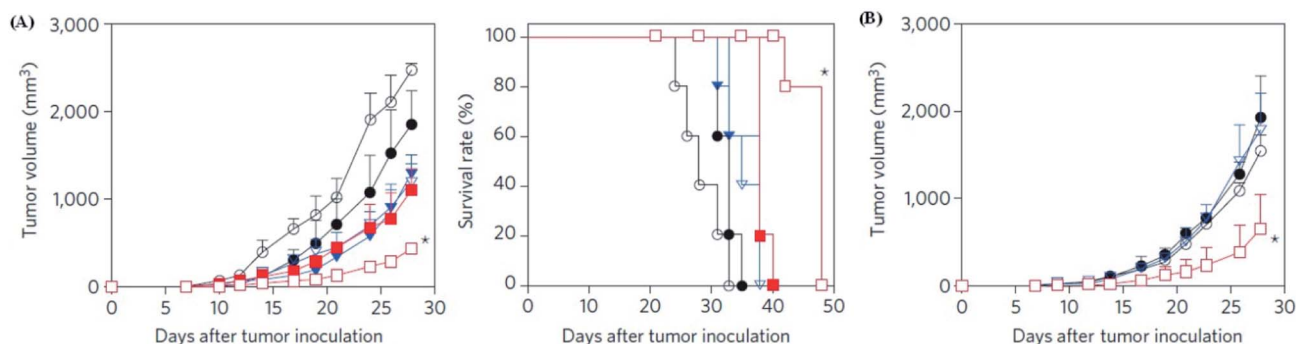


In an investigation conducted to compare the anti-diabetic activity and oxidative stress of ZnO NPs and ZnSO<sub>4</sub> in diabetic rats it was observed that ZnO NPs with small dimensions at higher doses (3 and 10 mg kg<sup>-1</sup>) had a much greater antidiabetic effect compared to ZnSO<sub>4</sub> (30 mg kg<sup>-1</sup>). The observation was backed up by a marvelous decline in the blood glucose level, a steep rise in the insulin level and a refinement of the serum zinc status in a time- and dose-dependent manner. However, it was finally inferred in the study that ZnO nanoparticles severely elicited oxidative stress particularly at higher doses corroborated by the altered erythrocyte antioxidant enzyme activity, enhancement in malondialdehyde (MDA) production, and remarkable drop in serum total antioxidant capacity.<sup>240</sup> Hyperglycemia can squarely trigger off an inflammatory state *via* activation of C-reactive protein (CRP) and cytokines, such as

ZnO nanoparticles were synthesized using a microwave-assisted method in the presence of *Vaccinium arctostaphylos* L. fruit extract.<sup>251</sup> A decrease in crystallite size was observed for the biologically synthesized ZnO compared to the chemically synthesized sample. Furthermore, the existence of organic



Wound healing is a complex and remarkable biological course of action a body resorts to in order to replace devitalized and missing cellular structures and tissue layers. The process usually consists of four phases: coagulation of blood, inflammation, proliferation, and tissue remodeling.<sup>261,262</sup> It is widely known that the healing process is subjected to a setback under several pathological circumstances such as major blood loss, diabetes, antibacterial infection, *etc.*,<sup>263</sup> which warrants the treatment of wounds urgently with utmost care.<sup>264</sup> The use of several wound dressings made of natural (alginate, chitosan, gelatin, *etc.*)<sup>265</sup> and synthetic (polyvinyl alcohol, polyurethane, polylactic acid, *etc.*)<sup>266–268</sup> polymers to heal wounds is quite common. However, synthetic polymer-based wound dressings



**Fig. 23** Tumor growth and survival of immunized mice. (A) Tumor volume (left) and survival rate (right) of mice (five mice per group) injected with MC38/CEA cells. (B) Tumor growth in human CEA-transgenic mice (five mice per group) inoculated with MC38/CEA cells (reproduced from ref. 259 with permission from Nature Publishing Group).

present some disbenefits such as toxicity of the degraded product and non-hemostatic properties. Notwithstanding that natural polymers are biocompatible, non-immunogenic and biodegradable, wound dressings containing these polymers too suffer from some disadvantages. For instance, the ionic compounds of alginate, especially calcium salts, might induce swelling and degradation while chitosan (CS) is greatly amenable to degradation by lysozyme.<sup>261</sup> In the light of these complications, researchers seek incorporation of nano-materials into wound dressing materials to prevail over these drawbacks. Among different metal-based nanomaterials, ZnO

NPs often find use in wound-healing applications. In an investigation by Diez-Pascual *et al.*<sup>269</sup> castor oil (CO)-based biocompatible polymeric films containing CS-decorated ZnO-NPs ( $\leq 5$  wt%) were developed and their wound-healing capacity in Sprague-Dawley rats was put under scrutiny. They demonstrated that the hybrid nano-wound-healing-apparatus was able to effect wound-healing in rats at a greater pace than effected by the administration of only castor oil-based polymer or gauze (control experiment), as clearly evident from the snapshots of wounds at different points of time (Fig. 24).



**Fig. 24** Photographs of the wounds treated with (1) CO/CS-ZnO (5.0 wt%), (2) CO, and (3) gauze as a control at (A) day 0, (B) day 5, and (C) day 14 (reproduced from ref. 269 with permission from the American Chemical Society).



The ensuing results revealed that, two weeks after administration, the synthesized nanocomposite induced a 90% reduction of wound area, while mere 70% wound repair was noted in the control experiment thereby bearing evidence of its commanding wound-fixing capacity. Augustine *et al.*<sup>270</sup> also fabricated ZnO NP decorated polycaprolactone (PCL) scaffolds and demonstrated that their implantation was able to boost faster wound-fixing by elevating the proliferation and migration of fibroblasts in an *in vivo* model (wound-healing model of American satin guinea pigs), without showing any marked signs of inflammation. Similarly, Bellare *et al.*<sup>271</sup> designed biocompatible ZnO NP based scaffolds of gelatin and poly(methyl vinyl ether)/maleic anhydride (PMVE/MA) with remarkable antibacterial effects. Their report threw light on the ability of the scaffolds for endothelial progenitor cell (EPC) adhesion and proliferation. Further, the topical application of the scaffolds on wounds of Swiss/alb mice displayed the potential to expedite the process of wound-fixing. Modern wound care materials suffer from several serious shortcomings that include inadequate porosity, inferior mechanical strength, lessened flexibility, lack of antibacterial properties, *etc.* Given this backdrop, a CS hydrogel/nanoparticulate ZnO-based bandage which exerted antibacterial effects against both Gram-negative (*E. coli*) and Gram-positive (*S. aureus*) bacteria was fabricated by Kumar *et al.*<sup>272</sup> The nanocomposite bandage characterized by biodegradability, microporosity and biocompatibility produced elevated wound healing in Sprague-Dawley rats and boosted re-epithelialization and collagen deposition at a remarkable pace. Taking into account the crucial factors of biocompatibility, antibacterial effects, and wound-fixing capacity, the researchers held that the hybrid nanomaterial-based bandage could be valuable for the healing of chronic wounds, burn wounds, diabetic foot ulcers, *etc.* Likewise, a porous bandage consisting of ZnO NPs conjugated with alginate hydrogel and exhibiting blood clotting capacity and bactericidal effects against *E. coli*, *S. aureus*, *Candida albicans*, and methicillin resistant *S. aureus* was fabricated by Mohandas *et al.*<sup>261</sup> The bandage made from the nanocomposite was observed to exhibit biocompatibility at a lower concentration of ZnO-NPs. Further, an *ex vivo* re-epithelialization investigation with porcine ear skin demonstrated that faster wound-fixing was effected by the hybrid nanomaterial-based bandage than only alginate control bandage. This was ascribed to the release of zinc ions that would enhance the proliferation and migration of keratinocyte cells to the wound area. Nair *et al.*<sup>273</sup> also developed a bandage consisting of a biocompatible nanocomposite of ZnO NPs conjugated with  $\beta$ -chitin hydrogel. The bandage showed efficient antibacterial activity (against *S. aureus* and *E. coli*) and had the ability of blood clotting and activation of platelets. It was elucidated that the application of the bandage on wounds of Sprague Dawley rats led to faster healing, with enhanced collagen deposition and a reduced number of bacterial colonies than in the control experiment, indicating the remarkable wound repairing capacity of the hybrid nanomaterial-based bandage. A novel, biocompatible ZnO QDs@GO-CS hydrogel was constructed by Liang *et al.*<sup>274</sup> through the simple assembly of ZnO quantum dots (QDs) with GO sheets and *via* a simple

electrostatic interaction with the loaded CS hydrogel. The antibacterial efficacy could reach 98.90% and 99.50% against *S. aureus* and *E. coli* bacteria, respectively, with a low-cost, rapid, and effective treatment. ZnO QDs in antibacterial nanoplateforms could immediately produce ROS and  $\text{Zn}^{2+}$  under acidic intracellular conditions. In parallel, when exposed to 808 nm laser irradiation, hyperthermia from GO sheets could simultaneously kill bacteria. Thus, the excellent performance of the material stems from the combined effects of hyperthermia produced under the near-infrared irradiation of GO sheets, reactive oxygen species, the release of  $\text{Zn}^{2+}$  from ZnO QDs under an acidic environment, and the antibacterial activity of the hydrogel. This work demonstrated that the synergy of antibacterial nanoplateforms could be used for wound anti-inflammatory activity *in vivo* indicated by the wound healing results. The hybrid hydrogel caused no evident side effects on major organs in mice during wound healing. Therefore, the biocompatible multimodal therapeutic nanoplateforms were proposed to possess great potential for antibacterial activity and wound healing. In a study by Dodero *et al.*,<sup>275</sup> the possibility of using for biomedical purposes alginate-based membranes embedding ZnO nanoparticles that were prepared *via* an electrospinning technique was extensively evaluated. The morphological investigation showed that the prepared mats were characterized by thin and homogeneous nanofibers (diameter of  $100 \pm 30$  nm), creating a highly porous structure; moreover, EDX spectroscopy proved ZnO-NPs to be well dispersed within the samples, confirming the efficiency of the electrospinning technique to prepare nanocomposite membranes. Mouse fibroblast and human keratinocyte cell lines were used to assess the biological response of the prepared mats; cytotoxicity tests evidenced the safety of all the samples, which overall showed very promising outcomes in terms of keratinocyte adhesion and proliferation. In particular, the strontium- and barium-cross-linked mats were characterized by similar cell viability results to those obtained with a commercial porcine collagen membrane used as a control; moreover, except for the calcium-cross-linked sample, the prepared mats exhibited a good stability over a period of 10 days under physiological conditions. Antibacterial assays confirmed the proficiency of using ZnO nanoparticles against *E. coli* without compromising the biocompatibility of the membranes. The mechanical properties of the strontium cross-linked mats were similar to those of human skin (*i.e.*, Young's modulus and tensile strength in the range 280–470 MPa and 15–21 MPa for the samples with and without nanoparticles, respectively), as well as the water vapor permeability (*i.e.*,  $3.8\text{--}4.7 \times 10^{-12} \text{ g m}^{-1} \text{ Pa}^{-1} \text{ s}^{-1}$ ), which was held to be extremely important in order to ensure gas exchange and exudate removal; furthermore, due to the low moisture content (*i.e.*, 11%), the prepared mats could be easily and safely stored for quite a long period without any negative effect on their properties. Consequently, the achieved results demonstrated that the prepared mats could be successfully employed for the preparation of surgical patches and wound healing products by using alginate as an economic and safer alternative to the commonly employed commercial animal collagen-derived membranes.



Ahmed *et al.*<sup>276</sup> fabricated chitosan/PVA/ZnO nanofiber membranes by using the electrospinning technique. The samples of chitosan/PVA and chitosan/PVA/ZnO tested for antibacterial efficacy and antioxidant potential demonstrated very encouraging results in diabetic wound healing. The nanofiber mats displayed outstanding antibacterial properties against various strains of bacteria. The samples of chitosan/PVA and chitosan/PVA/ZnO nanofiber membranes also manifested higher antioxidant properties which made them promising candidates for applications in diabetic wounds. In experiments involving diabetic rabbits, chitosan/PVA and chitosan/PVA/ZnO nanofiber mats exhibited increased performance of wound contractions in a time interval of 12 days. It was thus concluded in the study that the chitosan/PVA/ZnO nanofibrous membranes could serve as useful dressing materials for diabetic wounds, a major problem for type-2 diabetic patients worldwide.

### 5.13 Agriculture

ZnO NPs have been observed to increase the growth and yield of food crops. In peanuts, ZnO induced considerable stimulation of seed germination, seedling vigor, and stem and root growth.<sup>277</sup> A colloidal solution of ZnO is used as a nanofertilizer. Apart from providing the plants with nutrients, the nanofertilizer also revives the soil to an organic state without the harmful impacts resulting from the use of a chemical fertilizer. What gives a nanofertilizer a cutting edge is that it can be used in very low quantities. In a study by Mahajan *et al.*,<sup>278</sup> it was observed that there was stimulated growth in the roots and shoots of mung seedlings till the concentration of ZnO reached 20 ppm from 1 ppm and beyond this significant retardation was noted. In the same study an optimum ZnO concentration of 1 ppm was found to excite growth in gram seedlings. The effective growth at a specific optimum concentration and a retarded growth beyond this concentration were ascribed to the accumulation and uptake of ZnO NPs by the roots. The key role played by the concentration of nano-sized ZnO NPs was also reported by Prasad *et al.*<sup>277</sup> in their work on the benefits in the usage of ZnO NPs in agriculture. An optimum concentration of 1000 ppm of ZnO NPs led to an improvement of germination, root growth, shoot growth, dry weight and pod yield of the treated peanut seeds. Likewise, at 1.5 ppm concentration of ZnO NPs, Burman *et al.*<sup>279</sup> noted an improved shoot dry weight and antioxidant activity in chickpea seedlings. Raliya *et al.*<sup>280</sup> also reported a boost in plant growth and nutrient content in cluster bean seedlings upon treatment with 10 ppm ZnO NPs. In another study, enhanced plant height and dry weight were observed in maize plants.<sup>281</sup>

### 5.14 Photodegradation

Studies involving fabrication of metal oxide semiconductors for the photocatalytic degradation of Persistent Organic Pollutants (POPs) in water have been conducted on a regular basis of late with an eye to construct a low-cost, non-toxic and highly efficient photocatalyst. ZnO with its low cost, non-toxicity and high absorption efficiency across a large range of the solar spectrum

seems to fit the bill.<sup>282,283</sup> This resulted in various enthusiastic investigations that have been carried out in recent years to design ZnO NPs that can successfully exploit the Advanced Oxidation Process (AOP) mechanism thereby causing enhanced photodegradation of otherwise recalcitrant POPs *via* very pronounced generation of hydroxyl ( $\cdot\text{OH}$ ) radicals.

Ishwarya *et al.*<sup>73</sup> reported the degradation of methylene blue dye in the presence of ZnO NPs prepared using *Ulva lactuca* seaweed extract and solar irradiation in their study. With an optimum initial dye concentration of 25 ppm and an optimum catalyst loading of 200 mg, the dye present in 100 mL of water got degraded to 90.4% in 120 min.

Gawade *et al.*<sup>75</sup> carried out photocatalytic degradation of methyl orange dye using green fabricated ZnO NPs. 81% of the dye (20 ppm) was degraded after 100 min exposure to UV light. This they carried out after the dye solution was stirred with the catalyst for 30 min in the dark for complete equilibrium of the adsorption-desorption phenomenon when 2% of the dye was found to be adsorbed. The optimum catalyst dose was observed to be  $1.5 \text{ g dm}^{-3}$  after the dose had been varied in between 0.1 and  $2.0 \text{ g dm}^{-3}$ . The increase in degradation efficiency is ascribed to two favourable factors: (a) an increase in the number of active sites and (b) an increase in the number of photons absorbed by the catalyst. Beyond the optimal quantity of the catalyst, aggregation of the catalyst results in the active sites on the catalyst surface becoming unavailable for light absorption. The turbidity of the suspension leading to the inhibition of photon absorption on the catalytic surface of ZnO NPs because of the scattering effect was cited as an additional cause for the lowered degradation efficiency after the optimal catalyst dose.

Vidya *et al.*<sup>76</sup> performed the photodegradation of Congo red dye using ZnO NPs. With an optimal dose of 60 mg ZnO NPs in 250 mL of dye solution in which the concentration had been optimized at 30 ppm, they achieved a 90% degradation in an hour's time while maintaining an optimum pH of 9. They reported that at very low pH, the ZnO NPs display a marked tendency to agglomerate and the photon absorption on to the metal oxide surface reduces and in Congo red, the azo linkage ( $-\text{N}=\text{N}-$ ) is prone to electrophilic attack by the hydroxyl radicals. At low pH, despite excess  $\text{H}^+$  concentration, the  $\text{H}^+$  ion interacts with azo linkage decreasing the electron densities at the azo groups thereby discouraging the onslaught of hydroxyl radicals *via* the electrophilic mechanism. At high pH, electrostatic repulsion between the negatively charged ZnO catalyst and the negatively charged anionic Congo red dye hinders efficient photocatalysis. Using ZnO NPs synthesized by the sol-gel route, Senthilkumaar *et al.*<sup>284</sup> in their brief study, achieved total mineralization of Crystal Violet (CV) dye at pH 10.0 for 15 min of UV irradiation. The degradation was found to follow first order kinetics. The adsorption of the dye onto the ZnO surface in the absence of light was found to follow the Langmuir isotherm. In their study Daniel E. León *et al.*<sup>285</sup> performed photodegradation of an antibiotic cefotaxime under simulated solar irradiation. The ZnO dose was optimized at  $1.5 \text{ g L}^{-1}$ . An optimum pH of 7.5 was maintained during the experiment. The photoreaction showed 60% efficiency and attained equilibrium



after 60 min. It was found to follow pseudo-first order kinetics and the rate constant was evaluated to be  $0.466 \text{ min}^{-1}$ .

Enhanced photocatalytic activity of the Mg doped ZnO/reduced graphene oxide nanocomposite has been recently reported by Nithiyadevi *et al.*<sup>286</sup> They investigated photodegradation of cationic dyes Methylene Blue (MB) and Malachite Green (MG) under visible light irradiation. They achieved a 94.41% degradation of MB and a 99.56% degradation of MG after exposure to visible light for 75 min in each case. Both the photocatalytic degradations showed a marked increase in efficiency in comparison with that effected by bare ZnO NPs. They obeyed pseudo-first order kinetics and the rate constant assumed values of  $0.0391$  and  $0.0493 \text{ min}^{-1}$  respectively in the case of MB and MG. They cited the following reasons for the enhanced photocatalytic ability of the nanocomposite: (a) the introduction of reduced graphene oxide (RGO) enabled better adsorption of dye molecules through  $\pi$ - $\pi$  conjugation between the dye and aromatic compounds of RGO, (b) the ability of RGO to facilitate the growth of the ZnO particles on RGO sheets, (c) the availability of a large reactive surface area, (d) the greater interfacial contact between ZnO and RGO, (e) increase in the lifetime of charge carriers most probably attributed to RGO, (f) narrowing of the energy bands of ZnO due to  $\text{Mg}^{2+}$  substitution, (g) presence of oxygen vacancies and (h) the reduction of particle size.

Photodegradation of Methylene Blue (MB) was performed by Debasmita Sardar *et al.*<sup>287</sup> with an Ag-doped-ZnO nanocatalyst. On increasing the percentage of loaded Ag the rate of photocatalytic decomposition gradually increased and reached the maximum for 20% Ag loading on ZnO. The rate constant was found to be  $0.0087 \text{ min}^{-1}$  with a fairly high degradation efficiency of 55.87%. However, a sharp fall in the value of rate constant was observed for 25% Ag loading which remained almost the same on further increasing the Ag content, *i.e.* for 30 and 35% loading. It was also observed from TEM analysis that too much loading of Ag led to agglomeration and thus covered up the surface of ZnO preventing light absorption. Moreover, there were large numbers of unattached Ag nanoparticles which could be oxidized in the presence of reactive oxygen species. Oxidized silver would not initiate any charge separation in the system. It was thus assumed that silver up to this optimum amount could act as an electron-hole separation centre. Beyond the optimum amount, it could help in charge carrier recombination. In fact, a large number of negatively charged Ag particles (which had already accumulated electrons) on ZnO could capture holes and thus would start acting as a recombination site itself essentially by forming a bridge between an electron and a hole. Thus, the efficiency of charge separation and hence the photocatalytic capability declined to an appreciably large extent.

Very recently, Vaianoa *et al.*<sup>288</sup> too tried photo-catalytically favourable modification of ZnO by Ag. They too achieved similar results with regard to removal of phenol from water. The loading of Ag responded favourably in the range of 0.14–0.88 wt% but backfired beyond 1.28 wt%. Similar reasons as mentioned above were cited for the trends observed. A photocatalytic test was thus performed by using 0.15 g of the

optimized catalyst (1% Ag/ZnO) to treat drinking water containing phenol with an optimized initial concentration of  $50 \text{ mg L}^{-1}$  in 100 mL aqueous solution. Near-complete mineralization was accomplished within 180 min of exposure to UV irradiation. Photoreaction was found to fit in the pseudo-first order kinetic model. Another investigation<sup>289</sup> reported a facile microwave assisted synthesis of two-dimensional ZnO nano-triangles with a band gap of around 3.33 eV. The as-synthesized ZnO nano-triangles were applied for the reduction of noxious *p*-nitroaniline within 50 min. They were further used for the effective elimination of Rose Bengal dye within 150 min.

Likewise, ZnO-nanorods were synthesized<sup>290</sup> by adopting a facile microwave assisted green route of synthesis for the complete reduction of nitro compounds. Lauric acid was used as a complexing and capping agent in the ethanol phase. The nanorods had an average diameter of 5.5–10.0 nm with a hexagonal crystal structure and further demonstrated unusual luminescence properties wherein high intensity UV and yellow emission bands were observed along with negligible blue and green emission bands. Toxic nitro-compounds *p*-nitrophenol, *p*-nitroaniline and 2,4,6-trinitrophenol were completely reduced into amino derivatives by  $\text{NaBH}_4$  in the presence of these nanorods within 120, 45, and 18 min, respectively.

Chidambaram *et al.*<sup>291</sup> effectively constructed a ZnO/g- $\text{C}_3\text{N}_4$  heterojunction using a facile, economically viable pyrolysis synthetic route for the photodegradation of methylene blue under visible light illumination. The nanocomposites prepared using 0.1, 0.2 and 0.3 molar ratios of zinc nitrate precursor are labeled 0.1ZnO/GCN, 0.2ZnO/GCN and 0.3ZnO/GCN, respectively. The nanocomposites are found to exhibit a fall in charge recombination corroborated by their photoluminescence spectra that showed a fall in the intensity of the concerned emission peak (Fig. 25). A maximum photodegradation of 86% was achieved with 0.2ZnO/GCN in 60 min following a pseudo-first order kinetic rate constant of  $0.032 \text{ min}^{-1}$  while graphitic carbon nitride, 0.1ZnO/GCN and 0.3ZnO/GCN attained 44%, 73% and 76% degradation of methylene blue dye in the same time with lower rate constants. The loading of ZnO over g- $\text{C}_3\text{N}_4$  sheets created a heterojunction (Fig. 26). The excitation of electrons by visible light occurs from the valence band to the conduction band of g- $\text{C}_3\text{N}_4$ . The excited electrons are transferred to the conduction band of ZnO while there occurs a simultaneous movement of holes from the valence band of ZnO to the valence band of g- $\text{C}_3\text{N}_4$  *via* the smooth interface of the heterostructure. This enabled the generation of the superoxide anion radical and hydroxyl radicals that effected improved mineralization of the dye. An excess of Zn was deemed to cause recombination of photo-induced charges that led to decreased photocatalytic efficiency. In a recent investigation by the authors of the current work,<sup>292</sup> a destructive photocatalyst made up of ZnO nanorods/ $\text{Fe}_3\text{O}_4$  nanoparticles anchored onto g- $\text{C}_3\text{N}_4$  sheets was synthesized using hydrothermal synthesis and ultrasonication techniques. HRTEM micrographs shed light on the coupling of  $\text{Fe}_3\text{O}_4$  nanoparticles with ZnO nanorods and the successful formation of the intended ternary heterojunction. The g- $\text{C}_3\text{N}_4$  sheets fostered close contact between ZnO nanorods and  $\text{Fe}_3\text{O}_4$  nanoparticles







Fig. 25 Photoluminescence spectra of GCN and ZnO/GCN nanocomposites (the inset shows the enlarged PL spectra in the wavelength region of 350–450 nm) (reproduced from ref. 291 with permission from IOP Publishing).

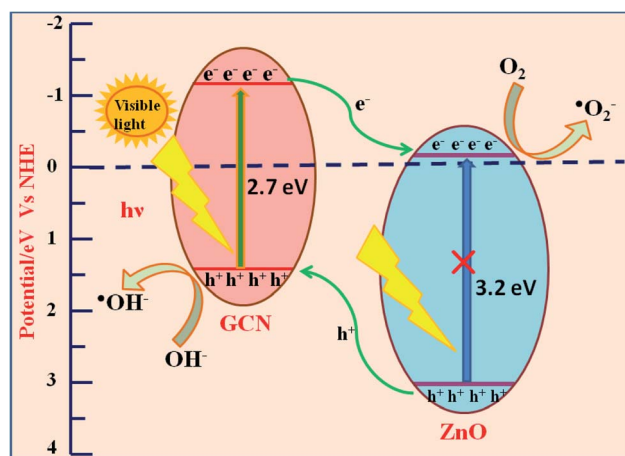


Fig. 26 Schematic depiction of the photocatalytic degradation mechanism of the ZnO/GCN heterojunction (reproduced from ref. 291 with permission from IOP Publishing).

thereby inducing a mellowed agglomeration of nanostructured ZnO/Fe<sub>3</sub>O<sub>4</sub> particles. The Tauc plot derived from UV-visible absorbance data showed that the ZnO/Fe<sub>3</sub>O<sub>4</sub>/g-C<sub>3</sub>N<sub>4</sub> nano-hybrid had a band gap of 2.48 eV. PL studies further confirmed the successful development of a staggered type II heterojunction with wide separation between light-induced charge carriers (Fig. 27). The hybrid catalyst showed remarkable photocatalytic activity under visible light, as evident from the efficient degradation of pantoprazole, a pharmaceutical

drug widely known as a recalcitrant organic water pollutant. This could be attributed to the synergistic interactions between ZnO, Fe<sub>3</sub>O<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub>. A degradation efficiency of 97.09% was achieved within 90 min with a remarkable pseudo-first order rate constant of 0.0433 min<sup>-1</sup>. The incorporation of Fe<sub>3</sub>O<sub>4</sub> expectedly facilitated the ready recovery of the catalyst and the degradation efficiency displayed fair consistency up to 4 cycles. The work thus offered a cost-efficient strategy for tackling organic water pollutants.

In another study,<sup>293</sup> a facile generation of a quaternary nano-structured hybrid photocatalyst, g-C<sub>3</sub>N<sub>4</sub>/NiO/ZnO/Fe<sub>3</sub>O<sub>4</sub>, was

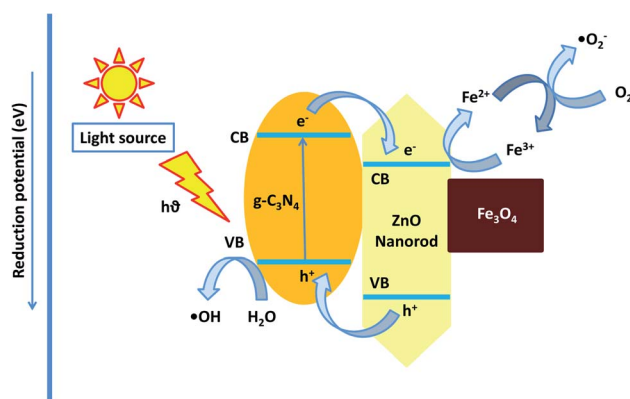


Fig. 27 Schematic depiction of the photocatalytic degradation mechanism of the g-C<sub>3</sub>N<sub>4</sub>/ZnO/Fe<sub>3</sub>O<sub>4</sub> heterojunction (reproduced from ref. 292 with permission from Elsevier).

proposed for photodegradation of an ecotoxic pharmaceutical drug, esomeprazole, in aqueous solution. The photocatalytic annihilation of esomeprazole as a prototypical organic contaminant was executed under LED irradiation. By itself the designed ternary heterojunction accomplished a maximum 95.05% photodegradation of esomeprazole and a TOC removal of 81.66% and COD reduction up to 70.68% under optimum conditions of catalyst dose, esomeprazole concentration and pH within 70 min at a superior pseudo-first order kinetic rate constant of  $0.06616 \text{ min}^{-1}$ . This actually implied an improvement of degradation over NiO/ZnO, g-C<sub>3</sub>N<sub>4</sub>/NiO and g-C<sub>3</sub>N<sub>4</sub>/ZnO up to ~74, ~57, and ~42%, respectively. The specific reaction rate also went up remarkably by almost ~3.8, ~3.18, and ~2.85 times in comparison with the values obtained for NiO/ZnO, g-C<sub>3</sub>N<sub>4</sub>/NiO and g-C<sub>3</sub>N<sub>4</sub>/ZnO, respectively. The remarkable photocatalytic potential of the heterostructured photocatalyst in practical applications was evident from its reconcilable performances under varying initial concentrations of esomeprazole and initial pH of the solution. The effect of the addition of H<sub>2</sub>O<sub>2</sub> was also put under scrutiny and it was found that the photocatalytic degradation, TOC removal and COD reduction increased to 98.43, 84.72, and 73.86%, respectively, upon addition of an optimum quantity of H<sub>2</sub>O<sub>2</sub> over the same time span. The impacts made by inorganic and organic species on photodegradation and the associated reaction kinetics were investigated and the results were reported. The inhibiting influence of water matrices on esomeprazole degradation was also evaluated for better assessment of the performance of the designed photocatalyst in a real aqueous environment.

CdS/ZnO photocatalysts were prepared by two steps *via* hydrothermal and photochemical methods for the photodegradation of rhodamine B (RhB) dye.<sup>294</sup> The UV/Vis absorption spectra revealed that the absorption performance of the heterostructure is extended toward the visible light region. The photocatalytic activities of both ZnO nanorod and CdS/ZnO heterostructures were investigated for the photodegradation of RhB dye. It was found that the CdS/ZnO heterostructure prepared with 30 min light illumination shows the best photocatalytic efficiency compared to the one at 15 min and pure ZnO nanorods. The better and enhanced photocatalytic efficiency of the CdS/ZnO heterostructure was ascribed to the high charge separation efficiency. The maximum photocatalytic efficiency of 85% was achieved within 8 h with the CdS/ZnO-30 min photocatalyst.

The photocatalytic degradation of rhodamine B (RhB) over chlorophyll-Cu co-modified ZnO catalysts (Chl-Cu/ZnO) was studied under visible-light irradiation by Worathitanon *et al.*<sup>295</sup> It was found that chlorophyll as an electron donor and copper in Cu<sup>2+</sup> form help inhibit the recombination of electron-hole pairs and improve the photoactivity of the catalyst. The synergistic effect between chlorophyll and Cu was found to improve the visible-light response of ZnO nanoparticles, resulting in excellent performance in photodegradation of RhB. The appropriate ratio of chlorophyll and Cu loadings over ZnO was 0.5Chl-0.10Cu/ZnO. At this ratio, under visible-light irradiation for 2 h, the degradation efficiency was approximately 99% (60 mg L<sup>-1</sup> of RhB solution), of which 18% of RhB adsorption

occurred under dark conditions. Moreover, outstanding reusability of Chl-Cu/ZnO, for up to six cycles, was found, with more than 80% degradation efficiency.

In yet another investigation,<sup>296</sup> ZnO nanowires (NWs) were successfully synthesized onto commercially available civil engineering materials using a hydrothermal synthesis method. This easy and low-cost method allowed obtaining an almost homogeneous repartition of nanostructures on the entirety of the surface of the substrates. The measured gap values were similar to those of the ZnO NWs grown on typical substrates, *i.e.*, ~3.18 eV and 3.20 eV for concrete and tiling, respectively. The excellent photocatalytic efficiency of our samples was demonstrated on three commonly used dyes, namely, Methyl Orange (MO), Methylene Blue (MB) and Acid Red 14 (AR 14). All of the dyes were fully degraded in less than 2 h for MB and AR 14, and less than 3 h for the more difficult to degrade MO. Investigating the durability of the samples so prepared, very promising results were found, as they showed no loss of efficiency after four experiment cycles. The ability of implementing ZnO NWs on civil engineering materials, their good photocatalytic properties, and the possibility to re-use samples with minimal efficiency losses, even after several months, were found very promising for the use of the nanostructures as road surfaces for air or water depollution.

## 6. Toxic impacts and mechanisms of ZnO NPs

Several toxicological studies have been carried out on zinc oxide nanoparticles in recent years. These studies revealed the potential adverse effects of ZnO NPs on health and the environment. ZnO NPs have been revealed to be seriously toxic to bacteria, *Daphnia magna*, freshwater microalgae, mice, and even human cells.<sup>297–301</sup> Although the exact mechanistic pathways of the interaction of ZnO NPs with living cells are yet to be known, studies have demonstrated their engagement with cell membranes, DNA and proteins and the key role they play in producing oxidative stress. Animals in certain regions face significant exposure to ZnO NPs that can have toxic effects on them. Rats on being in contact with ZnO NPs were found to demonstrate biochemical abnormalities in blood and pathological lesions in the stomach, liver and kidney.<sup>302</sup> Another study on male mice showed drastic physiological changes in them; an increase in the weight of the brain, pancreas and lungs accompanied by altered blood enzyme behaviour and mRNA expressions was observed.<sup>303</sup> The study conducted on male Sprague Dawley rats by Tang *et al.*<sup>304</sup> reported adverse effects on cytokines, enzyme activity, oxidative stress, histological and hematological parameters of liver and mRNA expression. Renal functions were also found to be disrupted by exposure to ZnO NPs. Tweaked oxidative stress and subdued catalase and superoxidase activity were observed in affected rats.<sup>305</sup> Bone marrow affected by oxidative stress and liver-DNA degradation were also reported in yet another investigation.<sup>306</sup> Exposure to ZnO NPs also leaves a host of pathological effects on aquatic organisms as has emerged from various investigations carried



## 7. Challenges and prospects

ZnO nanostructured particles have revolutionized the field of photocatalysis. And their efficacy in water splitting and degradation of recalcitrant organic water pollutants has been widely investigated and taken advantage of. However, a few concerning aspects about their photocatalytic activity still need to be dealt with through possible corrective measures. First, the photodegrading ability of a prepared ZnO nano-catalyst needs to be checked by taking the pollutant of interest *in lieu* of

a representative substance which in usual cases is a dye. This is because dye-degradation is relatively plain sailing while removal of pharmaceutical wastes, pesticides, insecticides or other endocrine disruptors offers greater challenges and complications. Moreover, the archives of scientific literature are brimming with thorough investigative reports concerning degradation of dyes. Furthermore, waste water contains a mix of different contaminants with varying ranges of pH and ionic strength. Few photodegradation studies have been conducted on organic pollutants in such a simulated sample of water while taking into account the effect of the presence of other contaminants, varying pH and ionic strength on the degradation kinetics. Second, a detailed insight into the mechanistic routes of the degradation of these compounds and their interaction with ZnO based nano-catalysts is elusive as of now and its development remains imperative and will unfold approaches to tackle other emerging contaminants. Third, many improvements in the very architecture of ZnO nanostructures are due specifically in areas such as surface area, particle size, separation and lifespan of charge carriers and so forth. Fourth, since band positions and band gaps are dependent on particle size, it becomes difficult to create heterojunctions able enough to achieve effective charge separation and thereby efficient photocatalytic activity. Systemic studies with a focus on discovering specific synthesis protocols for the achievement of ZnO based nanostructures with desired band positions and band gaps have to be embarked on. Also, there are a few difficulties associated with the operating procedures, such as loss and recovery of nano-structured photocatalysts in the course of post-synthesis treatment and photocatalytic activity. Furthermore, more sweeping research investigations are required to develop and verify the mathematical models for photocatalytic operations/systems for water/wastewater treatment in order to predict the quantum yield, kinetics and optimum conditions of the process.

ZnO nanomaterials may be outstanding candidates as biocompatible and biodegradable nanoplateforms for cancer targeted imaging and therapy. For *in vivo* imaging and therapy applications, the future of nanomedicine lies in multifunctional nanoplateforms combining both therapeutic components and multimodality imaging. Biocompatibility is also a concern for the applications of nanomaterials in biomedicine. Surface modification of nanomaterials plays a vital role in this context. Biocompatibility of ZnO nanomaterials might be enhanced by slowing down the dissolution rate through Fe doping<sup>324</sup> or surface capping.<sup>325</sup> Therefore, surface coating of ZnO NPs with biocompatible macromolecules, such as poly(lactic) acid, PEG, PEI and chitosan, was attempted to increase their suitability for further clinical usage. Another idea is the synthesis of ZnO nanoplateforms using the biodegradable and biocompatible materials already proven clinically. Some biocompatible polymers, such as liposomes and dendrimers, have been clinically approved for various pharmaceutical applications. Hence, the modification or conjugation of already approved therapeutic formulations or materials with functional ligands which will improve their diagnostic index could be essential. Much effort is needed for long-term *in vivo* toxicology studies to pave the way

for future biomedical applications of these intriguing nanomaterials. Facile conjugation of various biocompatible polymers, imaging labels, and drugs to ZnO nanomaterials can be achieved because of the versatile surface chemistry.

Some other issues of ZnO NPs concerning their biomedical application and their impact on biological systems still need further meticulous inspection. Following are a few such concerns: (a) lack of comparative analysis of the biological advantages of ZnO NPs to other metal nanoparticles, (b) the limitations imposed by the toxicity of ZnO NPs toward biological systems continue to remain a hot potato in recent research, (c) limitations of biocompatible/biodegradable ZnO nanoplateforms for tumor targeted drug/gene delivery, (d) lack of evidence-based research carrying out as its focal point a thorough survey of the therapeutic roles of ZnO NPs in improving anticancer, antibacterial, anti-inflammatory, and anti-diabetic activities, and (e) lack of extensive *in vivo* investigations into the anticancer, antibacterial, anti-inflammatory, and anti-diabetic activities of ZnO. Fresh studies focused on the above-mentioned issues would bring forth further elucidation and comprehension of the potential use of ZnO nanoparticles in biomedical diagnostic and therapeutic fields.

## 8. Conclusion

This overview on ZnO is an attempt to conduct a survey of the various strategies that have been adopted to fabricate ZnO nanostructures and their applications across a wide range of fields. One of the merits of the work lies in putting together the work done in biological and pure materials science fields. The other merit lies in the inclusion of a discussion on studies carried out on the toxicity of ZnO nanostructures. Additionally, a holistic picture of the future prospects of nano-structured ZnO *vis-à-vis* its application in the fields discussed has been presented. This overview demonstrates ZnO as a multitasking material displaying diverse properties, functionality and application in the emerging field of nanotechnology. The high chemical stability, high photostability, wide range of radiation absorption, biodegradability and biocompatibility enable ZnO NPs to be used in new materials and potential applications in a wide range of fields of technology. Moreover, the emergence of various eco-friendly bio-mediated syntheses of ZnO particles with enhanced properties which can be exploited for their better applications in various fields of technology has attracted greater interest in them in the world of research. The opto-electronic properties of ZnO NPs are largely exploited in the fabrication of ZnO based ETLs in DSSCs and PSCs. Modifications based on doping the ZnO ETLs with organic and inorganic substances and surface treatment with functionalizing agents for efficient electron transfer will surely lead to noteworthy advancements in the field of photovoltaics. The biomedical applications of ZnO have drawn great attention from researchers in recent years. Efforts are underway in designing more effective applications of ZnO NPs in regard to drug delivery systems, antimicrobial activity, anti-cancer activity, anti-inflammatory activity, and anti-diabetic activity as well as bio-imaging activity. Owing to their inherent toxicity, ZnO NPs exert profound inhibition





effects upon cancerous cells and bacteria. This they achieve by inducing intracellular ROS generation and activating apoptotic signaling pathways, which make ZnO NPs potential anticancer and antibacterial agents. Additionally, ZnO NPs have also been widely known to promote the bioavailability of therapeutic drugs or biomolecules when functioning as drug carriers to achieve enhanced therapy efficiency. Besides, the ability to lower blood glucose and raise insulin levels does attest to the promising potential of ZnO NPs in treating diabetes and abating its complications, which can be subjected to further evaluation. ZnO NPs are listed as a safe substance by the FDA. We believe that nanomaterials would dramatically promote the development of medicine, and ZnO nanoparticles are expected to make more exciting contributions in these fields. The role of ZnO nanofertilizer in inducing growth and yield of crops finds stress in recent research studies. Food demand has been mounting at an alarming rate with every passing day and regrettably the yield of staple food crops has not been able to catch up to the pace. It, therefore, is the need of the hour to design and commercialize metal oxide nanoparticles with favourable properties for sustainable agriculture. ZnO NPs have thus evinced a wide range of areas of applications in science and industry and with investigation on them still on, they offer yet newer possibilities of potential applications.

## Author contributions

S. R. wrote the main manuscript, text and figures. M. A. discussed the results and commented on, revised and corrected the whole manuscript.

## Conflicts of interest

The authors declare no competing financial or non-financial interests.

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