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Redox-sensitive ferrocene functionalised double cross-linked supramolecular hydrogels†

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Responsive double cross-linked hydrogels have proven to be a powerful approach to create smart polymer networks but unfold even greater potential if combined with supramolecular chemistry. A novel redox-sensitive ferrocene functionalised double cross-linked hydrogel is the core of this work. The explored network is based on poly(*N*-isopropylacrylamide) (PNiPAAm) and poly(2-methyl-2-oxazoline) (PMOXA) grafted chains with a β -cyclodextrin/ferrocene supramolecular host–guest system. A defined responsive behaviour was achieved using modified PMOXA macromonomers from a controlled synthesis, functionalised with β -cyclodextrin (CD-PMOXA) or with ferrocene (Fer-PMOXA), which were characterised thoroughly by ¹H NMR, GPC, FTIR and Raman spectroscopy. The association affinity between host–guest macromonomers (CD-PMOXA and Fer-PMOXA) was investigated by DLS and 2D NOESY and ROESY NMR analysis. Additional redox responsive supramolecular host–guest complexes were included as bismacromonomers and acted as additional cross-linking points in the hydrogel. These reversible changes of the degree of cross-linking in the polymer network allowed for reversible swelling and shrinkage by dissociation and re-formation of the complex, which also influenced the rheology of the hydrogel. The reproducible swelling and rheological changes could be tailored by the number of supramolecular cross-linking points in the network as well as the chain length of PMOXA macromonomers. Connecting the changes in swelling behaviour with the mechanical characteristics allowed for deeper structural insights into the network. The adjustable supramolecular redox-sensitive double cross-linked hydrogel provides an opportunity for future applications, e.g. in microfluidics and diagnostics.

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Introduction

One of the defining and key characteristic properties of cross-linked polymer networks is their ability to swell in a good solvent of network chains. A very prominent sub-type is hydrogels that swell in an aqueous solution. When they are capable of distinctive changes of their properties upon an external stimulus provided in their environment, they are called responsive hydrogels. Responsive hydrogels are hence able to absorb and displace water in response to external stimuli (temperature, solvent composition, pH, redox, light irradiation) that lead to a significant change in volume.^{1–5} Despite the change in volume, these gels retain their original shape due to isotropic swelling. Based on the type of cross-linking, hydrogels can be divided into two categories: chemically cross-linked and

physically cross-linked ones. Chemically cross-linked hydrogels possess covalent cross-linking points that allow varying of the mechanical properties of the polymer network by changing the density of the cross-linking points and they are commonly used when mechanically stable hydrogels are required.^{6,7} At the same time, the permanent nature of this type of cross-linking point deprives it of possible reversible changes, where dynamic bonds would be required.⁸ Physically cross-linked hydrogels possess non-covalent cross-linking points, which are based on physical interactions such as hydrogen bonds, van der Waals forces, and hydrophobic and coulombic interactions. Due to their non-covalent joints, physically cross-linked hydrogels show weak resistance to mechanical stimuli limiting the fields of their application. However, these types of networks can undergo reversible changes in network density based on the breakage and re-formation of physical bonds.⁸

Stronger non-covalent interactions can be achieved using dynamic supramolecular bonds, which can be broken and recovered by different chemical and physical stimuli. In recent years, the application of this type of bond as cross-linking points in hydrogels has drawn considerable and increasing attention.^{9–11} Hydrogels based on such bonds have a stronger

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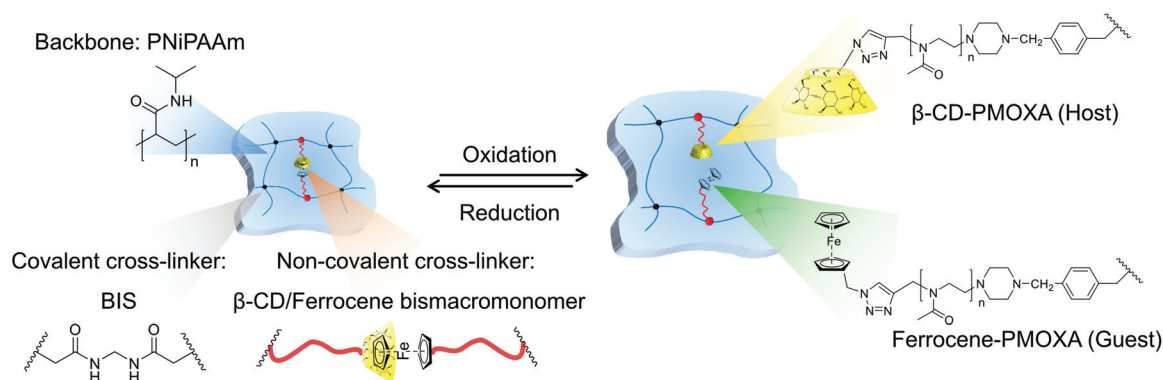
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However, the drawback of the weak control of swelling/mechanical behaviour at the swollen state restricts the development of hydrogels based on supramolecular chemistry. Double cross-linked hydrogels, bearing both covalent and dynamic

Here, we report on a novel double cross-linked hydrogel with a poly(*N*-isopropylacrylamide) (PNiPAAm) backbone cross-linked with *N*,*N*'-methylenebis(acrylamide) (BIS) as a permanent cross-linker and β -CD/ferrocene host-guest complexes as reversible cross-linking points. It should be noted that initial experiments with acrylamide did not form stable hydrogels. Poly(2-methyl-2-oxazoline) (PMOXA) bismacromonomers were used to introduce the host and guest molecules as part of additional cross-linking chains when the complex is present. Upon breaking the complex, the material is transformed into a hydrogel with grafted side chains (Fig. 1). In recent years, the number of publications that used PMOXA has increased dramatically due to its biocompatibility as well as the facility of the development of a clearly defined polymer structure by controlled polymerisation.^{34,35} PMOXA macromonomers with different chain lengths ($n = 30, 37$) and varying functionalities (alkyne (Alk- n), ferrocene (Fer- n), and β -CD (CD- n)) were synthesised *via* living cationic ring-opening polymerisation. The



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¹H NMR (500 MHz, CDCl₃): δ = 2.01–2.16 ppm (m, 105 H), 2.37–2.57 ppm (m, 8 H), 2.70–3.05 ppm (m, 3 H), 3.30–3.55 ppm (m, 142 H), 5.19 ppm (d, *J* = 10.0 Hz, 1 H), 5.70 ppm (d, *J* = 15.0 Hz, 1 H), 6.67 ppm (dd, *J* = 20.0, 10.0 Hz,

hydrogel slice were recorded for each measuring point. The same procedure was followed for the reductive step, except that a 10 mM sodium thiosulfate solution was used. This procedure was used to investigate the swelling properties of the different hydrogels.

Gel permeation chromatography (GPC). GPC measurements were performed on SECcurity² (PSS), equipped with GRAM Precolumn, Gram 30 Å, Gram 3000 Å columns under the following conditions: eluent – DMAc (5 g l⁻¹ LiBr and 1% H₂O) at 50 °C, the injection volume was adjusted to 100 µL and the selected flow rate to 1 mL min⁻¹. Before the measurement, the system was calibrated against PMOXA initiated with methyl-triflate and terminated with piperidine.

Fourier transform infrared spectroscopy (FTIR). The FTIR spectra of the macromonomers were recorded on a Vertex 80v instrument (Bruker) in the range of 4000–600 cm^{-1} with 100 scans at a resolution of 4 cm^{-1} . Analysis of the results was performed under baseline normalisation and correction using the OPUS 7 software (Bruker).

RAMAN spectroscopy. Raman spectra of macromonomers were acquired using Raman imaging system WITec alpha300R (WITec GmbH, Germany), which was generated by a pulsed 532 nm laser (laser power: 3–10 mW, integration time: 0.5 s, zoom: 20 \times).

Dynamic light scattering (DLS). The hydrodynamic diameter of the macromonomers was recorded on a Zetasizer Nano ZS instrument (Malvern Instruments, UK), equipped with a He-Ne laser (4 mW, $\lambda = 633$ nm) at a fixed angle of 1731 (non-invasive backscatter (NIBS) mode). The results were analysed using the Zetasizer software. Before measurement, the host-guest complex was prepared as follows: an aqueous solution of macromonomers was stirred 24 h at 70 °C.

Scanning electron microscopy (SEM). The hydrogels were analysed using an Ultra 55 PLUS field emission scanning electron microscope (Carl Zeiss AG, Germany) at 3 kV (magnification: 1000×). The SEM samples were freeze-dried, cut into suitable dimensions and sputter-coated with platinum before the measurement.

Swelling characteristics of hydrogels

For the swelling measurement, the respective hydrogel was prepared as an approximate 5 mm thick slice from the cylindrical hydrogel obtained after synthesis. In the first step, the initial mass value was always recorded. For this purpose, the hydrogel was taken out of the aqueous medium and carefully dried with a lint-free cloth. A glass plate was used as a wafer for hydrogels when measuring the mass on the balance. The redox cycles were then carried out in a flask with a volume of 25 mL. The hydrogel was treated with about 25 mL of the 0.1 mM hydrogen peroxide solution in a vessel. Then, the hydrogel was taken out, washed with water, and placed in the vessel with water (25 mL) for further swelling duration. Afterwards, the measurement of the mass took place as described before. In addition, the diameter and height of the

Rheology

For the rheological measurement of the hydrogels, the Advanced Rheometric Expansion System (ARES) (TA-Instrument) was used. The stainless steel plates with a diameter of 25 mm were calibrated in advance and the axial force was zeroed. Then the hydrogel was fixed between two parallel aligned stainless steel plates at RT. The previously measured hydrogel data such as the diameter and height were entered into the device for the calculations of the storage module G' . The exact determination of the diameter is crucial here. Therefore, this is measured at different points on the hydrogel slice. After the hydrogel was fixed into the rheometer at a force of 5 g, the measurement started. The storage modulus G' was examined at an oscillating frequency of 1.0 to 100 rad s^{-1} and a strain range of 2%. This strain was in the range of linear viscoelasticity as indicated by the flat G' and G'' over the entire frequency range (Fig. S15†).

Results and discussion

Polymer synthesis

The initial step of this work was the synthesis of bisfunctional PMOXA macromonomers (Fig. 2 and Fig. S1†). Using a propargyl tosylate initiator, an alkyne group was included into the PMOXA chains (Alk-*n*) and the CROP reactions were terminated with the vinylating agent *N*-(4 vinylbenzyl)piperazine (VBP) (Fig. 2a). VBP was essential to introduce a polymerising group and thus to be able to use the PMOXA chains as macromonomers during the hydrogel formation later on. The alkyne group was the basis for future modifications using the azide-alkyne click reaction. In order to study the influence of the chain length, a longer alkyne-PMOXA of around 40 repeating units and a shorter alkyne-PMOXA of around 30 repeating units (Alk-30) were targeted. Similar steps in polymer synthesis have been described in detail in our previous study.³²

The vinylating agent VBP which was used as the terminating agent in all PMOXA syntheses, was successfully derived from piperazine and 4-vinylbenzyl chloride according to the literature (confirmed by ^1H NMR, Fig. S2†).³⁶ PMOXA was synthesised *via* living CROP of 2-methyl-2-oxazoline with a ratio of $[\text{monomer}]:[\text{initiator}] = 40:1$ in acetonitrile at 70 °C for 3.5 h. After their synthesis, the macromonomers were purified by dialysis against MeOH. The chemical structure of the final polymer was investigated by ^1H NMR, which was also used to determine the degree of polymerisation (Fig. 2a). The signals $\delta = 2.81$ ppm (integral = 1.00) and $\delta = 4.01\text{--}4.20$ ppm (integral = 2.00, Fig. 2a, blue) of the ^1H NMR spectra of the polymer refer to the alkynyl protons and methylene protons in the initiator respectively. Their presence in the polymer con-



Fig. 2 ^1H NMR spectra of PMOXA macromonomers. (a) ^1H NMR spectrum of alkyne PMOXA (Alk-37) recorded in CDCl_3 , (b) ^1H NMR spectrum of CD-PMOXA (CD-37) recorded in $\text{DMSO}-d_6$, (c) ^1H NMR spectrum of Fer-PMOXA (Fer-37) recorded in $\text{DMSO}-d_6$.

confirmed the successful connection of the initiator to the macromonomers. From the integrating ratio of the signals corresponding to methylene protons and terminal ethylene proton ($\delta = 4.01\text{--}4.20$ ppm, integral = 2.00, Fig. 2a, blue; $\delta = 5.14$ ppm, integral = 1.00, Fig. 2a, red), the end-group functionality (EGF) of PMOXA with VBP was 99%. The high functionality with VBP allowed achieving a high degree of integration of macromonomers in the process of hydrogel synthesis. Analysing the integral ratio to the one of the PMOXA backbone peaks ($\delta = 3.04\text{--}3.66$ ppm, “f” in Fig. 2a–c and Fig. S3†) gave the degree of polymerisation (DP) of 37, confirming the presence of Alk-37 ($M = 3400 \text{ g mol}^{-1}$, Table 1) (for Alk-37: integral ratio of methylene protons $\delta = 4.01\text{--}4.20$ ppm, integral = 2.00 and PMOXA backbone $\delta = 3.04\text{--}3.66$ ppm, integral = 147.00). A DP of Alk-37 is close to the theoretical value confirming good adjustment of molar weight during the synthesis. The shorter PMOXA chains were synthesised using a 30:1 monomer: initiator ratio, yielding a DP of 30 (Alk-30; integral ratio $\delta = 4.01\text{--}4.20$ ppm, integral = 2.00 and PMOXA backbone $\delta = 3.04\text{--}3.66$ ppm, integral = 120.00) and had an equal EGF with VBP. GPC analysis revealed the expected monomodal distribution and low dispersities of 1.15 and 1.14 for Alk-37 and Alk-30, respectively (Table 1). Polymers for control experiments were synthesised as well. In order to test hydrogels with no host-guest complex, a methyl-terminated version with a DP of 35 was synthesised (Me-35). Since the β -cyclodextrin/adamantane host-guest complex is stronger than the one with ferrocene,³⁸ an adamantane-functionalised PMOXA with a DP of 40 (Ada-40) was synthesised following a previously published protocol.³² This polymer was intended to displace ferrocene in all host-guest complexes to gain a reference material in characterisation experiments.

CD-PMOXA and Fer-PMOXA macromonomers were obtained from Alk-PMOXA with a following copper catalysed azide-alkyne click reaction (CuAAC) with β -CD- N_3 and Fer- N_3 , respectively. Generally, the azide derivatives (1 eq.) and Alk-PMOXA (1 eq.) were used in the presence of DIPEA and CuI under argon in acetonitrile (for Fer- N_3) in acetonitrile/water (for β -CD- N_3) to implement the transition from alkynyl group to the triazole ring. In case of the CD derivatives, for the separation of the CD-PMOXA macromonomer from the free β -CD- N_3 molecules and copper catalysts, dialysis against water was performed. The obtained CD-37 and CD-30 were analysed by

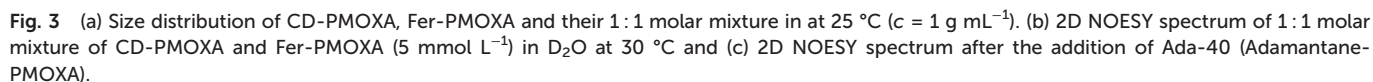
Table 1 Synthesis data and characteristics of poly(2-methyl-2-oxazoline) macromonomers

Polymer	[Monomer]/[initiator]	$M_{n,th}^a$ [g mol^{-1}]	$M_{n,NMR}^b$ [g mol^{-1}]	$M_{n,GPC}^c$ [g mol^{-1}]	D^c	EGF ^b (%)
Alk-37	40 : 1	3600	3400	2400	1.15	99
Fer-37		3800	3600	2600	1.09	98
CD-37		4800	4600	2800	1.14	94
Alk-30	25 : 1	2400	2800	2200	1.14	99
Fer-30		2600	2900	2200	1.19	99
CD-30		3500	3800	2800	1.23	97
Me-35	40 : 1	3600	3200	2900	1.11	98
Ada-40	40 : 1	3900	4100	3600	1.18	98

^a Calculated from [monomer]:[initiator]. ^b End-group functionality determined by ^1H NMR spectroscopy. ^c Determined by GPC in DMAc.



With the macromonomers in-hand, the next step was to study the β -CD/ferrocene host-guest complex forming the bismacromonomer, *i.e.* the reversible cross-linker. Knowing the association affinity of the synthesised macromonomers (CD-PMOXA and Fer-PMOXA) was the key characteristic before the hydrogel formation could be started. To investigate the coupling of ferrocene and β -CD macromonomers, DLS and 2D NOESY and ROESY NMR were conducted. DLS gave the hydrodynamic radius including the distribution of the macromonomers CD-PMOXA and Fer-PMOXA and their mixture (1:1 eq.) in water at 25 °C. The hydrodynamic diameter increased from 5.2 nm (CD-PMOXA) and 5.4 nm (Fer-PMOXA) to 8.1 nm for their mixture due to the formation of the host-guest complex (Fig. 3a). In addition to DLS measurement, 2D NOESY and



stirred for 24 h for complete complex formation (Fig. S11 and S12†). Four types of double cross-linked hydrogels with different amounts of non-covalent cross-linking points (0.5 mol% or 1 mol%) and with different PMOXA chain lengths (DP 30 or 37) were prepared (Table 2). The number of supramolecular cross-linking points in the network as well as the chain length of PMOXA macromonomers allow obtaining the hydrogels with tailor-made mechanical and redox properties. After 3 days of washing with deionised water, the homogeneous and yellow ferrocene-containing hydrogels and colourless control hydrogels were obtained. All the hydrogels will be named reflecting whether they contain grafted (G) PMOXA chains and the DP of PMOXA chains (30 or 37) after oxidation. To determine the influence of all parameters, GH-37 with 0.5 mol% of each cross-linker was defined as the basic hydrogel (GH-37a) and the conditions were altered from there. One gel contained more permanent cross-linking units (1 mol%, GH-37b), one had more responsive cross-linking points (1 mol%, GH-37c) and the last one had shorter PMOXA chains (GH-30a). Two control hydrogels were synthesised and named accordingly as GH-35Me (with Me-35 as grafted side chains) and H-PNiPAAm (no grafted side chains). It should be noted that the chain length of PMOXA bismacromonomers as the dynamic and non-covalent cross-linker (Table 2) incorporated in the polymer network was double the original size due to the formation of the host-guest complex. The bismacromonomers hence had DP ≈ 74 (CD-37 and Fer-37) and DP ≈ 60 (CD-30 and Fer-30).

Hydrogel	Type of macromonomer for host-guest complex		Permanent cross-linker BIS mol%	Host-guest cross-linker mol%
	Host	Guest		
H-PNiPAAm	—	—	0.5	—
GH-35Me		Me-35	0.5	1.0
GH-37a	CD-37	Fer-37	0.5	0.5
GH-37b	CD-37	Fer-37	1.0	0.5
GH-37c	CD-37	Fer-37	0.5	1.0
GH-30a	CD-30	Fer-30	0.5	0.5



To study all parameters of hydrogels under suitable conditions, the oxidation and reduction process had to be optimised (Fig. 4). Treating the hydrogel with 0.25 M hydrogen peroxide (H_2O_2) did break the host-guest complex and hence reduced the degree of cross-linking of the gel which was now a grafted hydrogel cross-linked by BIS only. However, the hydrogel disintegrated over time due to side reactions induced by H_2O_2 . Reducing the concentration to 0.1 mM H_2O_2 still induced the intended swelling but did not compromise the integrity of the gel. Oxidation was achieved by placing the gels for 1 h in 0.1 mM H_2O_2 and allowing them to swell for an additional 23 h in water. Choosing the correct reducing agent proved to be a challenging issue. Initial experiments with glutathione did not lead to the intended deswelling of the gel, but 0.1 M ascorbic acid did reduce the ferrocene and led to the intended deswelling of the gel. However, the deswelling was followed by strong unexpected swelling of the gel to more than 200% of the original weight. While this behavior could not be explained rationally, treating native GH-37a with 0.1 M ascorbic acid had a similar effect. Since ascorbic acid had this negative effect on the hydrogel, the conditions were changed to 10 mM sodium thiosulfate, which has been already used in the literature.³⁹ The following conditions proved to yield reproducible results: oxidation in 0.1 mM hydrogen peroxide for 1 h followed by an equilibration over 23 h in water to achieve full swelling. Reduction was conducted in 10 mM sodium thiosulfate and had to be carried out for at least 5 h, as kinetic measurements on GH-37c showed a complete deswelling only after a minimum of 5 h (Fig. S14†). Despite this complete macroscopic deswelling, a limited number of host-guest complexes might not have been re-formed, but had no effect on the macroscopic properties of the bulk material. In order to keep a reasonable working pattern, the whole deswelling step was elongated by placing the hydrogel in water for 19 h to a 24 h for the complete reduction step. The complete oxidation-reduction cycle was thus spread out over 48 h (Fig. 4 and Fig. S13† for a non-optimised 24 h cycle).

The difference in morphology of the supramolecular hydrogels in their optimised reduced and oxidised forms was also investigated by SEM (Fig. 5a and b). The average pore size of GH-37c increased (from 19 μm to 37 μm) when the oxidation of ferrocene in the hydrogel was carried out. This was easily reasoned that by oxidation the dissociation of host-guest complex took place and led to the swelling of hydrogel, resulting in the increase in pore size.

The optimised cycle conditions were then used to monitor the swelling degree (Q) over the reduction and oxidation cycle (Fig. 6a). Q was defined as the mass increase in % compared to the original value. Following the assumption that hydrogels consist mainly of water, the change in mass directly translated into a change in volume. On a molecular level, the distances between the permanent cross-linkers hence also increased by this percentage. The following characterisation took the values of at least 3 days washed and stored hydrogels in water after synthesis as references. All the graphs were normalised to the first swollen state (O1) as a starting point to emphasise the



Fig. 5 (a) SEM image of GH-37c in sodium thiosulfate (reduced, scale bar 20 μm), (b) SEM image of GH-37c in hydrogen peroxide (oxidised, scale bar 20 μm).

relative differences between the gels. While GH-37a showed Q_{ox} of 8 (O1 in Fig. 6a), GH-37b showed no swelling (data not shown). The larger amount of the permanent cross-linker in GH-37b (Table 2) reduced the mesh size of the PNiPAAm network, so that the broken host-guest interactions had no macroscopic effect on the polymer upon oxidation; also no swelling in 0.1 mM H_2O_2 over 3 days was observed. For GH-37a, the hydrogel was reproducibly swelled and deswelled over 5 cycles. In comparison with GH-30a, it is apparent that the longer PMOXA chains lead to a decrease in Q of the supramolecular grafted hydrogels (GH-30a: $Q_{\text{av. ox.}} = 15$; GH-37a: $Q_{\text{av. ox.}} = 8$). This tendency could be explained as due to the longer bismacromonomer cross-linkers interconnecting the underlying PNiPAAm network more loosely than the shorter PMOXA chains. Reducing the length of the bismacromonomer complexes allowed the incorporation of cross-linking points to the network with the size similar to or less than the PNiPAAm sub-chain. This leads to the higher Q of hydrogels similar to previously reported gels using the β -CD/adamantane system.³² With 0.5 mol% BIS in GH-37a and GH-30a, the average chain length of the PNiPAAm network in both hydrogels was expected to be about 100 units (200 C-C bonds). For GH-30a, bismacromonomers had about 180 + host-guest C-C bonds (3 bonds per repeating unit) while GH-37a had about 225 + host-guest C-C bonds for their bismacromonomers. However, the swelling and deswelling increased considerably from the third cycle for GH-30a. This hinted to a possible decomposition of the hydrogel, which could not be proved at this stage. The influence of different amounts of supramolecular cross-



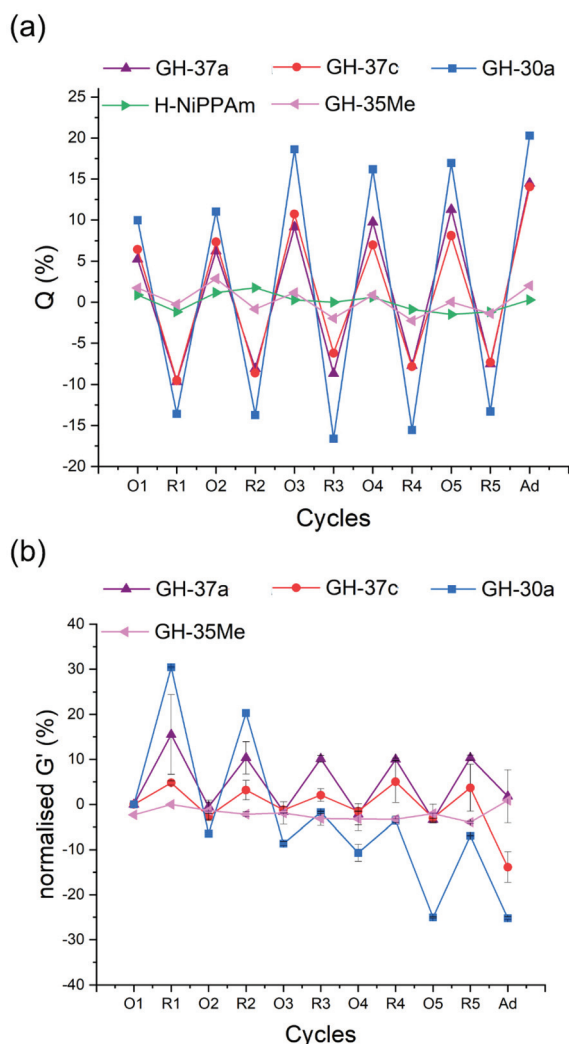


Fig. 6 (a) Swelling degree and (b) normalised storage modulus of supramolecular hydrogels GH-37a, GH-37c, GH-30a and control hydrogels H-PNiPAAm and GH-35Me. R1–R5 and O1–O5 correspond to the cycle number of reduction and oxidation, respectively.

linking points on the hydrogel properties was investigated by measuring GH-37c with 1 mol% of bismacromonomers. GH-37a and GH-37c show a constantly complete reversibility of swelling/deswelling after five redox cycles. Both hydrogels exhibited the same maximum and minimum of swelling degree in the optimised 48 h-cycle. The relative degree of swelling in the oxidised and reduced stages remain almost on the same level on each cycle for GH-37a and GH-37c (Fig. 6a).

It meant that the host-guest complexes were fully recovered after several redox cycles. It also indicated that the larger amount of responsive units did not lead to a larger response of the hydrogel (Fig. 6a), which was explained by the restrictions of the underlying PNiPAAm network as discussed for the comparison with GH-30a. To verify that specifically the supramolecular cross-linking points influenced the swelling degree of the hydrogels, the same swelling measurements of two control hydrogels (GH-35Me, H-PNiPAAm) were conducted. No differ-

ences in Q for the control hydrogels were observed upon redox cycles which confirmed the assumptions above.

At the last stage of redox cycles, the re-formation of the β -CD/ferrocene host-guest complex to β -CD/adamantane host-guest complex was used as the reference experiment to compare the number of broken cross-linking points under the influence of different stimuli (oxidation or re-forming of host-guest complex). After the last reduction process (R5 in Fig. 6a), hydrogels were placed in an aqueous solution of 2-aminoadamantane hydrochloride to form the β -CD/adamantane host-guest complex (shown in Fig. 3c through the addition of Ada-40). Forming the new complex permanently transferred the responsive double cross-linked hydrogel to a grafted hydrogel without any dynamic cross-linking points (Fig. 6a, "Ad" entry). Both the supposedly intact hydrogels GH-37a and GH-37c showed a defined increase in swelling once adamantane completely replaced the ferrocene in the complex (Q of 14 instead of 8 for both GH-37a and GH-37c). Despite the different amounts of host-guest complexes, both gels showed the same increase in Q (about 15) when treated with adamantane. This was in line with the restrictions imposed by the base PNiPAAm network already discussed. GH-30a, however, showed a similar swelling when treated with adamantane as in the oxidation-reduction cycles 3-5 of GH-30a when the suggested decomposition started.

In addition to the swelling degree measurements, the rheology of the synthesised hydrogels was investigated. Elastic deformation of the hydrogels was investigated by the storage modulus G' following a frequency sweep (see section 7 including Fig. S15 of the ESI† for details). Fig. 6b shows the normalised values of storage modulus after several reduction and oxidation cycles for GH-37a, GH-37c, and GH-30a and the GH-35Me as the control hydrogel. In order to compare all gels, the values were normalised on the first swelling cycle (no host-guest interactions) as the base value. The reduction of supramolecular hydrogels led to an increase of the storage modulus while the oxidation led to a decrease of the storage modulus. The rheological data were very much consistent with the observed trends in the swelling degree measurements. As expected, the control gel GH-35Me showed no increase or decrease in the oxidation-reduction cycles. GH-37a and GH-37c showed a reproducible decrease and increase in stiffening over all oxidation-reduction cycles. Very much unexpected, GH-37a showed the larger amplitude (10–15%) than GH-37c (about 5%). In contrast, GH-37a was weakened to a similar value of the first swelling after adamantane was added, but GH-37c was weakened to –15% of the first swelling cycle. This indicated that the larger amount of ferrocene in GH-37c led to more structural defects in the hydrogel. Breaking all host-guest interactions with the addition of adamantane then exposed these structural defects. Due to the shorter length of PMOXA, GH-30a showed a large increase in amplitude up to 30% of the reference value for GH-35Me. At the same time, the large difference in swelling and deswelling for GH-30a was constant (Fig. 6a), the G' began to decline rapidly over the following cycles (Fig. 6b), going down to –25% of the reference

value. This decline underlined the assumption of a decomposition of the hydrogel during the repeated redox cycles which was already hinted in the swelling data discussed earlier. Since this decomposition only occurred with the shortest PMOXA chain, it hinted that the valence electrons of ferrocene can interfere with the radical polymerisation leading to a less stable hydrogel with more structural defects. When the PMOXA was long enough to wrap ferrocene, this interference was suppressed in GH-37a but surfaced again in GH-37c when more ferrocene was present in the system. Summarising the results from swelling measurement, rheology and SEM, it is arguable that the presence of supramolecular cross-linking points based on the β -CD/ferrocene complex in the hydrogel structure leads to a significant, reproducible and reversible redox sensitivity.

Conclusions

In this study, novel redox-sensitive double cross-linked supramolecular hydrogels based on the host-guest interaction between β -cyclodextrin and ferrocene in a polymer chain network were synthesised. These hydrogels benefit from a combination of covalent and redox responsive non-covalent network joints (*i.e.* cross-linking points), which allowed combining of the properties of physical and chemical cross-linking points. β -CD or ferrocene units were introduced using modified PMOXA macromonomers. After the formation of the complex, the PMOXA chains had polymerisable units on both ends, effectively forming responsive bismacromonomers for a copolymerisation with NiPAAm. A shorter version (DP = 30) and a longer version (DP = 37) of clearly defined PMOXA macromonomers (CD-PMOXA and Fer-PMOXA) have been synthesised and thoroughly characterised by ^1H NMR, GPC, FTIR and Raman spectroscopy. The complexation study provided by DLS and 2D NOESY and ROESY NMR confirmed the formation of a bismacromonomer composed of a 1:1 host-guest complex. The resulting supramolecular hydrogel then contained the PMOXA chains with the β -CD/ferrocene complex as a reversible bismacromonomer cross-linker and BIS as a permanent cross-linker, which resulted in the intended double cross-linked hydrogels. These hydrogels show significant redox-sensitive properties compared to hydrogels without host-guest interactions. Both their weight and their elastic modulus responded reproducibly over various cycles to change from an oxidising to a reducing environment and *vice versa*. Increasing the amount of supramolecular cross-linking points in the hydrogel as well as decreasing the chain length of bismacromonomers demonstrated how the influence of the host-guest complex can be fine-tuned. While increasing the amount of the β -CD/ferrocene complex did increase the magnitude in response to the swelling, the rheological response revealed the existence of structural defects. Decreasing the molar mass of the bismacromonomers led to an increase in swelling degree and changes in elastic moduli of the supramolecular grafted hydrogels (GH-30a and GH-37a). However, since the ferrocene

did interfere with the radical polymerisation, hydrogels with shorter bismacromonomers or an increased content of the bismacromonomer increased the amount of structural defects of the network. The possibility to control the properties of hydrogels with the β -CD/ferrocene complex as reversible cross-linking points makes this system a promising one for various applications in microfluidics, for example, as a chemical actuator or sensor, in diagnostics, and for capturing or releasing specific molecules upon redox changes in the environment.

Author contributions

Nikolai Liubimtsev: methodology, investigation, data curation, formal analysis, writing – original draft, visualisation. Tom Kösterke: methodology, investigation, data curation, validation, formal analysis, writing – review and editing, visualisation. Yunjiao Che: conceptualisation, methodology, investigation. Dietmar Appelhaus: conceptualisation, resources, writing – review and editing, supervision, funding acquisition. Jens Gaitzsch: conceptualisation, methodology, formal analysis, writing – review and editing, visualisation, supervision, project administration. Brigitte Voit: conceptualisation, resources, writing – review and editing, supervision, project administration, funding acquisition.

Conflicts of interest

There are no conflicts to declare.

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